## **ORIGINAL PAPER**



# **Characterization of Ecofriendly Poly (Vinyl Alcohol) and Green Banana Peel Filler (GBPF) Reinforced Bio‑Films**

**B. Balavairavan<sup>1</sup>  [·](http://orcid.org/0000-0002-7646-2792) S. S. Saravanakumar1**

Accepted: 9 January 2021 / Published online: 2 February 2021 © Springer Science+Business Media, LLC, part of Springer Nature 2021

## **Abstract**

With an intention to replace the synthetic non-biodegradable films in packaging applications, the polyvinyl alcohol (PVA) blended with green banana peel fller (GBPF), the biodegradable flms were prepared by solution casting method with varying the concentrations of GBPF (5–25 wt%) in PVA matrix. The bio flms were characterized by Fourier-transform infrared spectroscopy (FTIR), X-ray difraction (XRD), thermo gravimetric analysis, transmissibility, FESEM, tensile test, flm solubility and water absorption, water vapour transmission (WVT), soil burial test. Based on results obtained, the changes evidenced in the FTIR spectrum of this PVA/GBPF bioflms suggest that strong hydrogen bonding is taking place due to interfacial exchanges of GBPF in PVA matrix. The XRD results showed that crystallinity of bio flms are greater than PVA. Thermo gravimetric analyses predicted that PVA/GBPF bio flms are thermally stable up to 300 °C. The light is 45% for transmittance in the visible light region for the PVA/GBPF (25 wt%) bio flm. The FESEM micrographs of bioflms palpable that formation of good physical interaction and compatibility between polymer matrix and GBPF up to 20 wt% of GBPF in PVA Matrix. FESEM results also confrmed that higher loading of GBPF (25 wt%) in PVA matrix, observed voids and agglomerations in flm surface. The PVA/GBPF bio flms with 20% of GBPF gave the highest tensile strength and young's modulus 44.5 MPa and 66.7 GPA respectively compared to other samples. The elongation at break decreases with increases the GBPF in PVA Matrix up to 20 wt%.The slight decrease in mechanical properties perceived due to higher loading of GBPF (25 wt%) with PVA matrix. The solubility, water absorption and WVT of the PVA/GBPF bio flms increased upon increasing the GBPF content. The biodegradation test results discovered that he highest weight loss at 42.3% (25 wt% of GBPF) probably due to the hydrophilic nature of GBPF in PVA matrix. On the whole, the present investigation confrmed that the PVA/GBPF bio flms potential for possible utilization in active packaging applications attributable to its better mechanical, thermal, optical, water absorption and biodegradation properties.

**Keywords** Green banana peel fller · Water absorption · WVP · Surface morphology: biodegradability

# **Introduction**

Petrochemical-based plastics are being replaced by bio based materials because of being widely eco-friendly. The plastics produced from sources (e.g., polyesters and polyolefn) have been commonly used in the packaging industry due to their potential features. Plastic is used as packing material on

 $\boxtimes$  B. Balavairavan balavairavanb@gmail.com S. S. Saravanakumar

sankarameena@yahoo.co.in

daily life. They are obtainable in large quantities and at low cost, displaying advantageous properties (i.e., good tensile strength, enriched barrier properties, and heat sealing) and applicability in the industry  $[1]$  $[1]$ . Even though plastic has those advantages, a high dependence on plastic could make bad effect not only on human health but also for the environment. However, these plastics are totally non-biodegradable and expose a serious ecological problem due to hydrophobic properties and very low water vapor transmission rate [\[2](#page-13-1)]. Due to that reason, there is an urgent need to fnd a new alternative packing material which is save for our body and environmentally friendly to replace plastics [[3\]](#page-13-2).

The growing public interest on the environment is induced the considerable research to develop biofilms from biopolymers. The harmful effect of conventional

<sup>1</sup> Department of Mechanical Engineering, Kamaraj College of Engineering and Technology, Madurai, Tamil Nadu 625701, India

petroleum-based plastic packaging materials overcomes by environment friendly biodegradable alternatives [\[4](#page-13-3)]. Various types of commercial polymers as well as composites are utilized for a number of industrial and daily purposes which are mainly constructed from non-biodegradable components [\[5](#page-13-4)]. Now-a-days, water soluble and biodegradable polymers got importance as an alternative source of material for using it in many purposes. Natural fbers are cheap, biodegradable and available in our surroundings and can be used to produce biodegradable composites, therefore assist to assured our environment. Manufacturing of biodegradable composites from such biodegradable plastics will enhance the demand of such materials [\[6](#page-14-0)]. The renewable natural polymers can be obtained from several sources such as starch, cellulose, chitosan, etc., while biodegradable synthetic polymers such as polyvinyl alcohol [[7\]](#page-14-1), polycaprolactone [\[8](#page-14-2)], polybutylene succinate [[9\]](#page-14-3), and copolymers are produced by using natural or petroleum-based monomers have been used. However; synthetic biopolymers have some drawbacks like a poor mechanical, barrier and thermal properties when compared to those of the petroleum –based non-biodegradable polymers. Hence, in order to overcome these drawbacks, the biopolymers are blended with fllers.

Polyvinyl alcohol (PVA) is a biodegradable polymer that has received signifcant attention because of its ability to decompose in relatively short timeframe. PVA is an important polymer having superior gas barrier properties along with higher strength, tear, and fexibility than those of natural bio based polymers [[10](#page-14-4)]. PVA is a synthetic biodegradable polymer with the technological potential as a water-processible polymer, thus allowing for flm production by solution casting [[11\]](#page-14-5). The excellent chemical resistance, good thermal barrier, good mechanical properties and physical properties of PVA have been employed in various applications, including dip coating, adhesives and solution casting flm. PVA was combined with other polymer or fller, can increase the physical, thermal and barrier properties [[12\]](#page-14-6). PVA is a costly commercial raw material. To lower down the cost of product without any compromise or partial compromise with its properties, the composites of PVA have been used. Composites are produced by blending biodegradable polymers with natural fllers to produce bio-based or biodegradable polymer composite that exhibit unique and stable characteristics. PVA has an excellent compatibility with the addition of nanocellulose filler which produces environmental friendly Nano composite. In this direction many researchers have attempted to use waste fllers such as tamarind nut powder [[13\]](#page-14-7), spent tea-leaf powder [[14](#page-14-8)], Orange peel powder [[15](#page-14-9)], coconut shell powder [[16](#page-14-10)], and Finger millet powder [[17\]](#page-14-11) etc., to make biocomposites for packaging applications.

Biodegradable plastics are a new generation of polymers emerging on the world market. Biodegradable plastics have an expanding range of potential applications, and driven by the growing use of plastics in packaging and the perception that biodegradable plastics are 'environmentally friendly'; their use is predicted to increase [\[18\]](#page-14-12). However, issues are also emerging regarding the use of biodegradable plastics and their potential impacts on the environment and effects on established recycling systems and technologies. Polyvinyl alcohol and starch are biodegradable polymers. Starch is one of the major sources in the development of biodegradable plastics. Owing to its large availability, low cost, renewability and biodegradability starches are commonly used in the production of biodegradable plastics. One of the most common waste forms of starch is the banana peels. The banana fruit's peel was selected for this experiment because it is a waste material rich in starch, the proximate composition [\[19\]](#page-14-13) of a banana peel is shown Table [1](#page-1-0).

Banana is a fruit plant grown plenty in tropical countries like India, Sri Lanka, Malaysia, and Thailand etc. The banana peel is thrown as a waste after consumption of the fruit. These peels are also used as feed for calves. However, if they are not properly disposed, they create environmental pollution by emitting toxic gases like hydrogen sulphide and ammonia [[20](#page-14-14)]. Banana peels are also used in the production of methane [[21](#page-14-15)], bio-adsorbent for dyes in wastewater and gelling agent in food processing industries. Banana peels account for 30–40% weight of the fruit and contain proteins, carbohydrates, dietary fbers, ash and traces of fat and moisture [[22](#page-14-16)]. They also contain mono and disaccharides, phenolic compounds and minerals [[23](#page-14-17)]. The banana peels are mainly composed of carotenoids, phenolic compounds, favonolsanols, biogenic amines and dietary fbers. These components are prospective antioxidants that possess a wide range of biological efects including antibacterial, anti-viral, anti-infammatory, anti-allergenic, antithrombotic, and vasodilatory actions etc [[24\]](#page-14-18). The photo chemicals present in banana peels are generally antioxidants and can help in preventing the oxidation in food materials [[25](#page-14-19)]. Banana peels exhibit excellent antimicrobial activity against bacteria, fungi, and yeast. Since banana peels contain phenolic compounds, they can be used as a source to develop the desired properties and to extend the food safety and the shelf life [\[26\]](#page-14-20). Banana peels consists high source of starch. As banana peels ripen, the glucose level increases. Therefore,

<span id="page-1-0"></span>**Table 1** Constituents of green banana peel

Constituents	$(g/100 g$ dry matter)	
Protein	$8.6 \pm 0.1$	
Fat	$13.2 + 0.2$	
Starch	$12.78 + 0.9$	
Ash	$15.25 + 0.1$	
Total dietary fat	$50.25 + 0.2$	

banana peels can be suggested as a suitable source for the manufacturing of biodegradable plastics. It is also understandable through the literature that banana peels could be a value addition to biomaterials thereby helping the reduction in environment pollution, and banana peels have enormous potential for its industrial use [[27\]](#page-14-21).

The main objective of this work is to prepare biocomposites flms from PVA and banana peel powder with diferent concentrations (5, 10, 15, 20 and 25 wt%) and characterize the fabricated bio composite flms by, FTIR, XRD, TGA, FESEM, transmissibility, % water absorption and water vapour permeability, tensile testing and soil burial test. These bio composite flms are appropriate for potential use in food packaging.

# **Materials and Methods**

## **Materials**

Poly (vinyl alcohol), PVA, 98% hydrolyzed, (MW of 1,25,000 and 1% ash max) and distilled water was purchased from Sigma–Aldrich chemicals Private limited (Bengaluru). Before the use, poly (vinyl alcohol), PVA was kept in desiccated atmosphere to do away with moisture content. Green Banana Peel Powder (GBPP) used as reinforcing filler was obtained from Venus Starch Suppliers, Salem, Tamil nadu, India. The average particle size of the GBPP was 333.3 nm measured by using Malvern Particle size Analyzer Instrument.

## **Fabrication of PVA/GBPF Bio Films**

The PVA/GBPF bio flms were fabricated using a solution casting method. First, the PVA was dissolved in distilled water by magnetic stirring for 3 h at 80 °C. Second, the GBPF (5, 10, 15, 20 and 25 wt%) added in PVA solution and magnetically stirred for about 6 h. The polymer solution (PVA/GBPF) cured at 50 °C for 30 min. The Prepared solution was casted on Petri plates of 130 and 60 mm diameter and dried at desiccators for 12 h. Ten samples were prepared for each composition of poly (vinyl alcohol) and green banana peel fller. Finally prepared PVA/GBPF bio flms were characterized and analyzed.

## **Characterization of PVA/GBPF Bio Films**

#### **Fourier Transform Infrared Spectroscopy (FTIR)**

Fourier transformation infrared spectrographs (FTIR) were recorded to categorize a number of peaks and an organic functional group on the exterior of the green banana peel fller, PVA and the prepared PVA/GBPF bio-flms. A small piece was cut from the prepared flm samples and placed onto the germanium plate. Infrared spectra in attenuated total refectance accessory of GBPF, PVA controlled flms and PVA/GBPF dry films were recorded in the mid-IR region 4000–500 cm−1 at room temperature, using an RXI Perkin Elmer FTIR Spectrophotometer at a resolution of  $4 \text{ cm}^{-1}$ .

#### **X‑ray Difraction Test (XRD)**

The crystalline structures of thin flm samples were determined by XRD technique. XRD equipment is used for recording the flm difractograms in 25 °C. X-ray difraction patterns of the GBPF, PVA controlled flm and PVA/ GBPF bio-flms were carried out by using a Bruker D8 X-ray diffractometer. The X-ray source was Ni-filtered Cu Kα radiation (40 kV and 30 mA). The composite flms were mounted on a sample holder, and the pattern was recorded in the refection mode at an angle 2θ over a range of 5.000°–80.009° at a scan rate of of 5° min−1. The conventional peak height method used to calculate the crystallinity index  $(I_C)$  of films.

$$
I_C(\%) = \left(1 - \frac{I_{am}}{I_{002}}\right) \times 100\tag{1}
$$

 $I_{002}$  is the intensity of crystalline peak (around 25°) and  $I_{am}$ is the intensity of amorphous peak (around 18°).

#### **Thermo Gravimetric Analysis (TGA)**

TG/DTG curves were obtained using thermograms Thermogravimetric analyzer (TA Instruments, TGA Q500). The thermograms of green banana peel fller, PVA matrix and PVA/GBPF bio flms were recorded using at a heating rate of 10 °C min−1 under a dynamic fow of nitrogen (50 ml min−1) in the 50–600 °C range. Approximately 10.2 mg of samples were placed on an aluminum pan for testing. Weight losses of the samples were noted for every 10 °C temperature rise.

#### **Surface Morphology (FESEM)**

The surface morphology and element mapping of the prepared flms were investigated by capturing micrographs of the representative samples using Nova Nano SEM 450, a feld—emission scanning electron microscope (FE–SEM) equipped with energy dispersive X-ray (EDX) analyzer. The operating voltage 2 kV, (Make: carl zeiss (USA), Model: sigma with Gemini column), and resolution of 1.5 nm. The samples of the prepared flms were prior coated with gold for 2 min using EMITECH SC7620 Mini Sputter Coater.

#### **Optical Absorption (Transmissibility) Analysis**

The barrier properties of PVA/GBPF composite flms against ultraviolet (UV) and visible light were measured at selected wavelength between 200 and 800 nm, using a UV–visible spectrophotometer (model 8415A, Hewlett–Packard Co., Santa Clara, CA, USA).

#### **Mechanical Properties of PVA/GBPF Bio Films**

Tensile strength (TS) and elongation at break (EAB) of the prepared flms were conducted by using an Instron 3369 testing machine. Tensile test of (PVA and PVA/GBPF bio flms) the samples were conducted according to the ASTM-D882The test was performed in the controlled room at 25 °C and 50% relative humidity (RH). Five flm samples in composition  $(10 \times 100 \text{ mm})$  with the initial grip length of 3 cm were used for testing for each flm. The flm samples were clamped and deformed under tensile loading using a 100 N load cell with the cross head speed of 2 mm s<sup> $-1$ </sup> until the samples were broken. TS values were calculated by dividing the maximum load (N) by the initial cross-sectional area  $(m<sup>2</sup>)$  of the specimen. EAB was calculated as the ratio of the fnal length at the point of sample rupture to the initial length of the specimen and expressed as a percentage.

#### **Film Solubility in Water (FS %)**

The flm Solubility was determined as the percentage of dissolved dry matter after immersion in water. Three randomly selected specimens of each type of film  $(3 \times 3 \text{ cm})$  were first dried at 60 °C for 24 h to determine the initial dry matter  $(W_1)$ . Each film was immersed into 30 ml of distilled water in a 50 ml for 24 h with slow mechanical stirring. The flm samples were removed by fltration after 24 h and dried in a drying oven at 105 °C for 24 h to determine the undissolved final dry weight  $(W_2)$ . The initial dry weight was determined from the sample moisture content (determined by gravimetric analysis), and the diference in weight was used to calculate the water soluble matter as percentage of the initial weight. Then FS of the flm sample was calculated as follows:

$$
FS\% = \frac{w_1 - w_2}{w_1} \times 100.
$$
 (2)

#### **Water Absorption**

The swelling test was performed to determine water uptake ability of the prepared films (PVA and PVA/GBPF). The water absorbing capacity of flms was conducted in accordance with ASTM D570-98 [[28](#page-14-22)]. The flm samples of each composition were cut into the dimensions of  $24 \times 10$  $mm<sup>2</sup>$ . Thin film samples were dried in on oven at 80  $°C$ prior to measurement. The dried flms were weighed and immersed in 20-ml distilled water for 12 h at 25 °C. The excessive water on the surface of the flms was removed with flter paper and then weighed immediately. The average of five values was recorded. The percentage of water absorption,  $W_a$  (%) was calculated.

$$
W_a = \frac{(W_e - W_0)}{W_0} \times 100
$$
 (3)

where  $W_e$  was the weight of the samples at equilibrium and  $W_0$  was the initial weight of the dried samples.

#### **Water Vapour Permeability or Water Transmission (WVP)**

The water vapor permeability (WVP) of PVA/GBPF flms was determined by the standard method of ASTM E96-80 [[29\]](#page-14-23). The WVP measuring cup was flled with 18 ml of distilled water and film sample  $(10 \times 10 \text{ cm})$  was placed on the top of cup and sealed tightly to prevent the leakage of water vapor. The assembled WVP cup was weighed and subsequently placed in a controlled environmental chamber set at 25 °C and 50% RH. Weight change of the cup was determined every1 h for 10 h. The water vapor transmission rate (WVP;  $g m^{-2} h$ ) of the film was calculated by using the slope of the steady-state (linear) portion of weight loss versus time plot. Then, the WVP (g m<sup>-2</sup> H pa) of the film was calculated as follows:

$$
WVP = \frac{\Delta W}{(t \times A)}\tag{4}
$$

where  $\Delta W$  was the weight gain, *t* was the time tested (h) and *A* was the area exposed to water.

#### **Bio Degradation Test—Soil Burial Test**

The soil burial test was carried out on a laboratory scale to examine the biodegradability using the method reported by [[30\]](#page-14-24). The Average initial weight and dimensions of the dry specimens were 0.65 g and  $(40 \times 10 \text{ mm})$ . The dry specimens were buried by random pattern in a perforated box to allow the samples to be attacked by the microorganisms and moisture. The moisture content was maintained at 40–50% of the soil's maximum water holding capacity. In addition, the pots were covered with plastic flm to avoid water evaporation from the soil surface. The pot containing the soil and samples were incubated at almost constant temperature of 25 °C for one month. Biodegradation was estimated by monitoring changes in weight as a function of burial time. The samples were removed from the soil every 10 days. The specimens

were then carefully washed with water several times in order to remove debris and soil from the flm and to ensure that the degradation stopped. After that, the samples were dried at room temperature to a constant weight. After drying, they were weighed using a GR200 model analytical balance in order to determine the percentage values of weight loss as by the following equation:

Weight loss 
$$
(\%) = \frac{W_0 - W_t}{W_0} \times 100
$$
 (5)

where  $W_0$  is the initial mass and  $W_t$  is the remaining mass at any given time, *t*.

# **Results and Discussion**

#### **FTIR Analysis**

The FTIR spectra of green banana peel filler (GBPF), PVA matrix and PVA/GBPF bio flms with 5, 10, 15, 20 and 25 wt% of GBPF loading are presented in Fig. [1a](#page-4-0). Figure [1b](#page-4-0) represents the FTIR spectrum of a GBPF. Figure [1](#page-4-0)b indicated the bands in the region of 3418 cm−1 indicates the presence of a stretching of strong hydroxyl groups. The band at 2980 cm<sup>-1</sup> is assigned to CH<sub>2</sub> stretching. Also, the band at 1735 cm<sup>-1</sup> corresponding to the stretching of a carbonyl group C=O (hemicellulose band). The band at 1647 cm−1 represents absorbed water and the band at 1420 cm−1 assigned to –C–H bending. In addition, the band at 1281 cm−1 corresponds to C–O stretching (lignin band) and the band at 1026 cm−1 represents C–OR stretching [[31\]](#page-14-25). The characteristics peak of PVA could be observed at 3327 cm−1 for stretching vibration of (–OH) and additional peaks of 2931 cm<sup>-1</sup> (–CH<sub>3</sub> vibration) and 1725 cm<sup>-1</sup> (C=O stretching) while PVA having the additional bands at 1240 cm<sup>-1</sup> (–CO of carbonate),  $1082$  cm<sup>-1</sup> (–O–C=O) and 855 cm<sup>-1</sup> (C–C methyl group). It is clearly understood that all the spectra are almost similar except for a few changes. Comparing the spectra of the GBPF with PVA matrix indicates that the intensity of the OH groups  $(3318 \text{ cm}^{-1} \text{ of the})$ fller is very less when compared to that of the matrix indicating lower  $\alpha$ -cellulose content in the filler). Further, the GBPF has the additional peaks corresponding to the  $CH<sub>2</sub>$ group in the range 3010–2853 cm<sup>-1</sup> with higher intensity compared to the matrix. This indicates the presence of more number of flexible CH<sub>2</sub> groups in the filler than in the matrix. There is a strong peak observed at 1735 cm<sup>-1</sup> for the fller corresponding to the CO stretching vibrations. The intensity of this peak in the matrix is shifted to 1725  $cm^{-1}$ and it is very low peak, this indicates that more amount of CO group containing hemicelluloses in the GBPF [\[32](#page-14-26)]. Thus, it clearly indicates that the fller is more amorphous than the matrix due to the presence of a large amount of



<span id="page-4-0"></span>**Fig. 1** FTIR spectra of **a** PVA, GBPF and PVA/GBPF (5–25 wt% bio flms. **b** GBPF peaks

amorphous hemicellulose in it. Another peak observed in the GBPF at  $1610 \text{ cm}^{-1}$  which is shifted to 1630 cm<sup>-1</sup> for the PVA matrix corresponds to the observed water molecules which are more in the fller. Similarly, the peak at around 1385 cm<sup>-1</sup> corresponds to CH<sub>2</sub> and CH<sub>3</sub> vibration of the fller which is having higher intensity when compared to that of the matrix. The intensity of the peak observed at 1071 cm−1 corresponding to the C–O–H and C–O–R vibrations is lower than that of the matrix indicating lower OH groups. There is an additional peak in the filler at  $1281 \text{ cm}^{-1}$ corresponding to the O–H phenolic of lignin.

The FTIR analysis of the GBPF indicates its lignocellulose nature. The –OH, C=O,  $CH<sub>2</sub>$  and  $CH<sub>3</sub>$  groups are available in both PVA and GBPF that may be possibly from compatibility and hydrogen bonding matrix and fller. It is evident from the FTIR spectra that the intensity of the bands of the PVA/GBPF bio flms mainly consist of the functional groups in between that of the matrix and the fller. It is noted that C=O peak slightly shifted toward the right side from 1725 to 1690 cm<sup>-1</sup> for the incorporation of GBPF (5–25 wt%) in the PVA Matrix. These observations confrmed that all PVA/GBPF composite flms having assimilation peak at  $1690 \text{ cm}^{-1}$  were authorized to the vibration of carbonyl (C=O) group, which facilitated the compatibility between two components. With the enhancement of GBPF  $(5–25 \text{ wt%)}$  with PVA, the characteristic peak at 3327 cm<sup>-1</sup> shifted toward lower wave numbers in bio flms. The intensity of the peak gradually decreased due to GBPF loading in the PVA matrix. There is no notable high intensity peaks in PVA/GBPF bio flms spectra due to GBPF loading in the PVA matrix. Both PVA and GBPF are hydrophilic [[33\]](#page-14-27) in nature, so good at dispersion of Nano particles occurred in bio flms.

## **XRD Analysis**

X-ray diffraction is a proven tool to study crystal lattice arrangements and yields very useful information on degree of sample crystallinity. The X-ray patterns of GBPF, PVA and PVA/GBPF bio films are shown in figure, from Fig. [2](#page-6-0)a–e, it is understandable that the matrix and the bio flms have almost similar peaks. The main common peaks at  $2\theta = 14.3^{\circ}$ , 18.8° and 19.4°. The GBPF has the peak at  $2\theta = 20.9^\circ$ . From Fig. [3a](#page-7-0), b, it is clearly evident that GBPF has higher crystallinity than PVA. Also it can be noticed that the intensity of peaks of the fller is lower than the intensity of the matrix and the intensity of PVA/GBPF bio flms reduced with increase in the fller content. The PVA diffractograms shows that higher the intensity, which divulges that semi crystallinity, crosslinking property and amorphous nature [\[34\]](#page-14-28). The difractograms of PVA/GBPF bio flms have higher intensity than that of GBPF. It is clearly evident that the PVA/GBPF bio flm consists of amorphous (PVA) and high crystalline (GBPF) components. There by fabricated PVA/GBPF bio flms have crystallinity index in between the crystallinity of fller and matrix.

Table [2](#page-7-1) shows the peak values and calculated CI value of PVA, GBPF and PVA/GBPF bio flms. The CI value of Bio flms augmented when increasing the fller content which may be due to the higher CI value of GBPF. The Peak value and crystallinity Index (CI) of PVA, GBPF and PVA/GBPF bio-flms are shown in Table [2](#page-7-1). The high crystallinity value of PVA/GBPF bio flms strongly predisposed in the thermal capacity and betterment in tensile properties.

#### **Thermal Analysis (TGA and DTG)**

The thermal stability of the poly vinyl alcohol (PVA), green banana peel fller (GBPF) and PVA/GBPF bio flms were tested using Thermogravimetric analysis and the resulting TGA and DTG curves are shown in Fig. [4](#page-7-2)a, b.

The TGA curves of the PVA/GBPF bio flms showed the weight loss pattern with increasing temperature and the DTG curves evidently exhibited the decomposition temperature at each stage of thermal degradation. The entire bio flm sample exhibited three stages of thermal degradation. The initial stage of decomposition observed around below 60—150 °C corresponds to the evaporation of moisture content with about 7–15% of weight reduction. Even though the GBPF was dried before the usage of ample amputation of moisture content is difficult due to its hydrophilic nature [[35](#page-14-29)]. The second stage of thermal degradation from 200 to 380 °C which indicated the putrefaction of polymeric network of lignin, cellulous and wide range organic molecules. The third of thermal dilapidation occurs between 400 and 550 °C indicated the deprivation of wax and ash content in PVA/GBPF bio flms 85–96% of weight reduction. PVA exhibited a two-step degradation at about 280 and 430 °C, which were attributed to the acetate group elimination at lower temperatures followed by a breakdown of polymer backbone at higher temperatures [\[36\]](#page-14-30). The main stage of weight loss or the maximum thermal decomposition  $(T_{max})$  exhibited around 300, 333, 310, 313, 316, and 320 °C for PVA, GBPF, PVA/GBPF  $(5-25 \text{ wt\%})$  bio films respectively. The DTG curve shows maximum degradation temperature (300 °C) of PVA got shifted to 320 °C in Fig. [4b](#page-7-2), mainly due to higher thermal stability of fller content (25 wt%). Further, it is evident that the GBPF demonstrated higher thermal stability than matrix and bio flms. The infection temperature corresponding to the maximum degradation rate for the PVA was found to be at 300 °C and it was increased for the PVA/GBPF bio flms with increase in the fller concentration from 5 to 25 wt%. The increased thermal stability of the PVA/GBPF bio flms may be due to presence of rigid polyphenols in the GBPF [[37\]](#page-14-31). This improvement



<span id="page-6-0"></span>**Fig. 2** X-ray difractograms of **a** PVA/GBPF (5 wt%) flm; **b** PVA/GBPF (10 wt%) flm; **c** PVA/GBPF (15 wt%) flm; **d** PVA/GBPF (20 wt%) flm; **e** PVA/GBPF (25 wt%) flm

in thermal stability of PVA/GBPF bio flms is attributed to the strong interfacial interaction and chemical bonding between PVA and GBPF. It is also evident that lowr glass transition temperature of the PVA/GBPF bio flms due to many methylene groups available in GBPF. The above declared verdicts ensured that PVA/GBPF bio flms are appropriate materials to fabricate the application with employed temperature of 300 °C.



<span id="page-7-0"></span>**Fig. 3** X-ray difractograms of **a** green banana peel powder. **b** Pure polyvinyl alcohol (PVA)

<span id="page-7-1"></span>**Table 2** Peak value and crystallinity index (CI) of PVA, GBPF and PVA/GBPF bio-flms

Bio films (CI)		Pos ( $^{\circ}$ 2Th) FWHM left ( $^{\circ}$ 2Th)	Crystal- linity index
Pure PVA	18.9388	0.6000	40.47
Green banana peel filler	20.9011	2.4000	68.54
PVA/GBPF 5 wt%)	18.9503	1.4400	53.25
PVA/GBPF (10 wt%)	18.8049	1.6800	56.56
PVA/GBPF (15 wt%)	19.0987	0.7200	60.72
PVA/GBPF (20 wt%)	19.4444	1.4400	63.88
PVA/GBPF (25 wt%)	19.1772	1.6800	67.13

## **Transmissibility**

Figure [5a](#page-8-0), shows that the PVA/GBPF bio flm has low light transmittance and the observation on the transparency level of the flm. On the surface, PVA/GBPF bio flm the green banana peel fller is spread evenly, for higher concentration of GBPF in PVA matrix it seems rough, and appears to be porous. The PVA/GBPF bio flms photographic images clearly shown in Fig. [5](#page-8-0)b. Although the value of light transmission is low, the composite flm still showed a good result. This condition causes a high chance for vapor to interact with the surface of the flm [\[38](#page-14-32)]. Thus



<span id="page-7-2"></span>**Fig. 4 a** TGA—curves of PVA, GBPF and PVA/GBPF bio flms; **b** DTG—curves of PVA, GBPF and PVA/GBPF bio flms

<span id="page-8-0"></span>



the light is not only refected and refracts by both flm and vapour surface but also absorbed toward the flm [[39](#page-14-33)].

#### **Surface Morphology**

Hence, only a few of the light transmit into this flm. PVA film is transparent with nearly 70% light transmittance at a wavelength of 800 nm as shown in Fig. [5](#page-8-0)a. However, after the green banana peel fller addition into PVA, the light transmission signifcantly decreases. As the concentration of GBPF from 5 to 25 wt% in PVA matrix, the light transmission of bio flm decreased. The light is 70% for transmittance in the visible light region for the controlled PVA flm and 38% for the flm containing 25 wt% of GBPF. All the PVA/GBPF bioflm samples are characterized by high transmission in the visible spectral range. Visible light is composed of a range of frequencies. The frequency of the radiation is proportional to its energy and the wavelength of the radiation is inversely proportional to the energy. Red is the lowest energy visible light and violet is the highest. The PVA/GBPF samples transmit predominantly in the brown region of the spectrum (700–800 nm). Figure [5](#page-8-0)a shows that, all these flms have low light transmission % (high light absorption). Increase of light absorption threshold of biodegradable flms in the UV region of the spectrum offers prospects for their use in the production of packaging material for food industry [[40\]](#page-14-34). Although the value of light transmission is low, the composite flm still showed a good result and has appropriate optical clarity for using in packaging application.

FE-SEM micrographs of Green Banana Peel Filler (GBPF) are shown in Fig. [6a](#page-9-0), b with the magnifcation of 100 and 10 µm respectively. The FE-SEM images at a magnifcation of 100  $\mu$ m of the composites with lowest (5 wt%) to highest (25 wt%) of GBPF loading in PVA matrix were presented in Fig. [6](#page-9-0)d–h respectively. From Fig. [6](#page-9-0)c, it is evident that the controlled PVA flms exhibits a very fat and smooth surface and does not show any favored alignments.

The GBPF exhibited the irregular surface morphology and a porous surface. This porous surface can cause a high absorption of water soluble PVA polymer solution [[41](#page-14-35)]. The PVA/GBPF bio flms are showing the highly layered structure and the nanoparticles of fllers dispersed in PVA matrix due to strong intermolecular hydrogen bonding [\[42](#page-15-0)]. From Fig. [6](#page-9-0)d–g, it is evident that in the case of the bio flms with 5, 10 and 15 wt% of fller loading, the GBPF was fairly uniformly distributed in the PVA matrix. However, when the fller loading was increased to 20 wt%, more agglomeration of the GBPF was evidenced in Fig. [6g](#page-9-0), h. The portion of the composite where agglomeration was identifed is shown in circles in Fig. [6g](#page-9-0). At high loading of GBPF, some parallel alignment and rough texture are also observed on the addition of GBPF which results in transition occurs from ductile to brittle. This will indirectly afect the mechanical properties at higher loading of GBPF and flms will become



<span id="page-9-0"></span>**Fig. 6** FESEM micrographs of flm surfaces **a**, **b** Green banana peel Filler; **c** PVA flm; **d** PVA/GBPF (5 wt%) flm; **e** PVA/GBPF (10 wt%) flm; **f** PVA/GBPF (15 wt%) flm; **g** PVA/GBPF (20 wt%) flm; **h** PVA/GBPF (25 wt%) flm

more brittle further incorporation of GBPF in PVA Matrix, The similar results were observed in the case of PVA/surface treated bagasse fber composites [\[43](#page-15-1)]. It is also evident that  $N<sub>2</sub>$  functional groups in filler promote the homogeneous spreading of GBPF into polar PVA matrix. From the mentioned results, it is obvious that the GBPF molecules have good compatibility with PVA and gives small voids, and agglomerations observed due to higher loading of fller and its high crystalline nature.

## **Mechanical Properties**

In order to study the efect of the GBPF on the mechanical properties of the PVA/GBPF bio flms the tensile test was carried out and the variation in tensile strength, % elongation at break and tensile modulus are presented in Fig. [7a](#page-10-0)–c. From Fig. [7a](#page-10-0), b it is clearly evident that the incorporation of GBPF increased the tensile strength and modulus of elasticity of the bio flms. From Fig. [7,](#page-10-0) it is also noted that the tensile strength of the PVA/GBPF bio flms increased from 27.1 (Controlled PVA flms) to 44.5 MPa (PVA/GBPF 20 wt%), 39.10% of increase of tensile strength. Similarly, the tensile modulus of the PVA/GBPF increased from 5.34 to 66.7 GPa, 91.99% of increase in tensile modulus from pure PVA flm to PVA/ GBPF (20wt %) bioflm. Figure [7](#page-10-0)c shows the % of Elongation at break is reduced with fller loading in PVA matrix. It is evident from the results that the GBPF truly acted as reinforcements, as tensile properties of the PVA/GBPF bio flms improved with increased fller content. This may be ascribed to the even distribution of the GBPF in PVA matrix and good adhesion between the components. The existence of polyphenols, fbers, proteins, carbohydrates, mono and disaccharides in the GBPF, may be the reason for the bonding and enhanced properties of the PVA/GBPF bio flms. Further, the increase in fller content also has increased the number of interfacial region between fller and the matrix which leads to higher load-bearing capacity. Moreover, in the case of random fller reinforcement, the strength and young's modulus increases only for some proportion of the weight fraction of fller [[44](#page-15-2)].

Subsequently, there was a small reduction in the tensile strength and modulus with 25 wt% GBPF reinforcement but the values were higher than that of the matrix. This may be ascribed to some agglomeration of the GBPF particles in PVA matrix when the concentration was further increased. The agglomeration of the GBPF particles is evident from the SEM images.



<span id="page-10-0"></span>**Fig. 7** Variations in **a** tensile strength, **b** tensile modulus and **c** % Elongation at break PVA matrix and PVA/5–25 wt% of GBPF bio flms

It is also understandable that the stress transfer from the matrix to the fllers was efective which improved the tensile properties. A drop in the % elongation of the blends could be seen with increased fller particles. This indicates that the GBPF acted as a rigid component in the PVA/GBPF bio flms. It is a common trend when there is an increase in the tensile modulus and vice versa. This shows the enhanced toughness of the bio flms. Similar results were also reported earlier [\[45](#page-15-3)]. The above stated result reveals that PVA/GBPF bio flms (5–20 wt% of GBPF in PVA matrix) have Good mechanical properties and it is suitable material for packaging applications.

#### **Film Solubility in Water (FS)**

The Water solubility (FS) of bio flm which indicates the water resistance of the film, Fig.  $8$  shows the % of film solubility in water for the controlled PVA flm and PVA/GBPF (25 wt %) bio flm were 20.5 and 25.5%, respectively.

The FS of fabricated bio flms increased linearly with increase in amalgamation of GBPF in PVA matrix. This result indicates that the GBPF based bio flms absorb water and hydrophilic more than the controlled PVA flm, which might be attributed to the hydrophilic components in the green banana peel fller such as fat, fber, and resistant starch [[39](#page-14-33), [46](#page-15-4)] also suggested that the water sensitivity of controlled PVA flm could be improved by adding the GBPF.

#### **Water Absorption**

Water absorption properties of the PVA/GBPF bio flms after immersion in a water tank at 25 °C for 12 h are shown in Fig. [9.](#page-11-1) Water absorption testing provided the water absorption characteristics of the film. The tendency of flm to absorb moisture cannot be overlooked; especially for packaging applications, because the slightest amount of moisture or water can signifcantly change the other properties. The percentage of water absorption PVA/GBPF flms is shown in Table [3](#page-12-0).

The perceived increase in water absorption for PVA/ GBPF films, with increasing GBPF composition, was due to the existence of hydroxyl groups [[47](#page-15-5)], which are extremely susceptible to water or moisture absorption. This was proven by hydrogen bonding observed in the water

<span id="page-11-0"></span>

<span id="page-11-1"></span>

FTIR spectra. A decrease in water sensitivity for bio flm is required for most applications of flms. However, due to the hydrophilic nature of the PVA and water sensitivity of the GBPF, the percentage of water uptake was increased slightly for PVA/GBPF bio flms. The same behaviour was also observed in previous studies by Liang et al. [\[48\]](#page-15-6).

## **Water Vapour Permeability (WVP)**

WVP is a measurement used to determine the ability of moisture to penetrate and pass through a material. The WVP was calculated and is shown in Table [3](#page-12-0). WVP testing determined the ability of water or moisture to penetrate and pass

<span id="page-12-0"></span>**Table 3** Film solubility in water (FS), water absorption  $(W_a)$  and water vapour permeability (WVT) of pure PVA and PVA/GBPF bio flms

Bio films $(g m^{-2} h)$	FS(%)	$W_a(\%)$	WVP
Pure PVA	20.5	10.1	7.2
PVA/GBPF 5 wt%)	20.9	13.5	7.9
$PVA/GBPF (10 wt\%)$	21.8	17.8	8.5
PVA/GBPF (15 wt%)	23.5	23.5	9.3
$PVA/GBPF (20 wt\%)$	24.2.	25.5	9.9
PVA/GBPF (25 wt%)	25.5	31.6	11.5



<span id="page-12-2"></span>**Fig. 11** Bio-degradation behaviour of PVA and GBPF bio flms

through the flm. Gontard et al. [[49](#page-15-7)] mentioned that the rate of WVP of a hydrophilic flm depends on both the difusivity and solubility of water molecules in the matrix of the flm. Figure [10](#page-12-1), shows the rate of WVP of the Controlled PVA film is lower than that of PVA/GBPF  $(5-25 \text{ wt\%})$  bio films.

The hydrophilic nature of PVA and the water sensitivity behaviour of GBPF owing to absorption of water molecules, thus resulting in a better WVP of PVA/GBPF bio flms than pure PVA flms. Bio flms, with the\ highest content of GBPF, showed the highest value of WVP. This may have been caused by the poor compatibility between PVA, GBPF during higher loading which allows more water molecules to difuse through the flms. Water molecules preferably diffuse through amorphous areas of the polymer matrix, so the degree of crystallinity is also important in the permeability

behavior of Films [\[50](#page-15-8)]. The WVP of PVA/GBPF bio flms is increased when permeable nano fllers (GBPF) are dispersed in the polymer matrix to induce tortuous path for difusion and some discontinuities were observed in higher composition of green banana peel fller in PVA matrix. It is also observed in SEM results.

<span id="page-12-1"></span>



#### **Soil Burial Test**

The biodegradation behavior of controlled PVA flm and PVA/GBPF bio flms are shown in Fig. [11.](#page-12-2) The soil burial provides qualitative indications of biodegradation. Most of the studies on the biodegradability have been based on the weight loss  $[51]$  $[51]$ . In this investigation, the degradation behavior of PVA/GBPF bio flm was evaluated by measuring its weight loss, which refers to the erosion of molecules from the solid phase.

The initial biodegradation step involves the enzymatic oxidation of secondary alcohol groups in PVA to ketone groups; then the hydrolysis of ketone groups results in chain cleavage [[52\]](#page-15-10). The degradation behavior of the flms is dependent on the presence of green banana peel fller reinforced in the PVA matrix. Figure [11](#page-12-2) shows the PVA/ GBPF bio flms have the highest biodegradability rate than the controlled PVA flms. The PVA/GBPF bio flms deserve the highest weight loss at 42.3% probably due to the hydrophilic nature of GBPF itself. Hydrophobicity increases water absorbability and hence increases degradation. In fact, the degradability also can be related to its appearance which is diminished in size and this contributed to the weight loss of the samples. Moreover, the addition of GBPF reinforcement improved properties of the PVA based flm as a whole. Both GBPF and PVA are hydrophilic. The incorporation of both components (PVA and GBPF) is believed to absorb water due to formation of hydrogen bonding between PVA and green banana peel fller evidenced by FTIR analysis. It can be concluded that the weight loss of the PVA/GBPF bio flms at 30 days proved that they are biodegradable.

## **Conclusion**

In the present report, Poly (vinyl alcohol) PVA, the agricultural waste Green Banana Peel Filler (GBPF), and PVA/GBPF bioflms with various compositions of GBPF  $(5-25 \text{ wt\%})$  with the PVA matrix were prepared by solution casting method. Based on this study, investigating the preparation and properties of PVA/GBPF bio flms, the following conclusions can be drawn. FTIR spectra designated that the presence of—OH groups on PVA and GBPF resulted in hydrogen bonding interaction which could improve the compatibility of the two components. The degree of crystallinity of all the PVA/GBPF bio flm was found to be superior to the PVA. It is manifest from the thermal analysis that the PVA/GBPF bio flms are appropriate materials to withstand the temperature of 300 °C. The light is 45% for transmittance in the visible light region for the PVA/GBPF (25 wt%) bio flms at a wavelength of 800 nm. The FESEM result revealed a good compatibility between PVA and GBPF upto PVA/20 wt% GBPF bioflms. The small size of

voids, agglomeration and cracks exhibited in the flm surface in PVA/25 wt% GBPF bioflms can be attributed to high crystalline nature of GBPF. The tensile strength and young's moduls of the PVA/GBPF bio flms were enriched by 39.10 and 91.99% respectively. The elongation at break decreased with increase in the GBPF content up to 20 wt% in PVA matrix. The FESEM results clearly show that the incorporation of 25 wt% GBPF in PVA matrix the voids, agglomeration and micro cracks are present in surface of the flm. So it is the transition stage of bioflm from ductile to brittle nature, thereby tensile properties of PVA/GBPF bioflms reduced after incorporation of 25 wt% of GBPF in PVA matrix. The water absorption test showed that the hydrophilic nature of starch molecules of GBPF increases the water uptake for all the PVA/GBPF bioflms. The presence of GBPF in the PVA matrix influenced the water uptake of the PVA/GBPF bio flms. Both water absorption and water vapour transmissibility increased with incorporation of GBPF in PVA matrix. % of Water absorption upturns from 10.1 to 31.6% and water vapour transmissibility increased from 7.2 to 11.5%, compared with the pure PVA flm. The degradation behaviour of PVA/GBPF bio flms was enhanced at 42.3% at higher composition of GBPF in PVA/GBPF bio flms. High water absorption capacity of PVA/GBPF bio flms is supportive for augmentation in biodegradation properties. Hence, with better tensile, thermal, optical, water absorption and biodegradation properties, the PVA/GBPF bio flms can be suitable for food packaging and enfolding applications. The study also deliberated to show the biodegradable flm from amalgamation of polymer and organic waste has potential to become alternative resources in plastic making industries, to reduce the amount of discarded organic wastes and to contribute to waste-to-wealth industry development.

# **References**

- <span id="page-13-0"></span>1. Sadanand V, Rajini N, Rajulu AV, Satyanarayana B (2016) Preparation of cellulose composites with in situ generated copper nanoparticles using leaf extract and their properties. Carbohyd Polym 150:32–39
- <span id="page-13-1"></span>2. Tian H, Wang K, Liu D, Yan J, Xiang A, Rajulu AV (2017) Enhanced mechanical and thermal properties of poly (vinyl alcohol)/corn starch blends by nanoclay intercalation. Int J Biol Macromol 101:314–320
- <span id="page-13-2"></span>3. Basu A, Kundu S, Sana S, Halder A, Abdullah MF, Datta S, Mukherjee A (2017) Edible nano-bio-composite film cargo device for food packaging applications. Food Packag Shelf Life 11:98–105
- <span id="page-13-3"></span>4. Khan A, Khan AAP, Asiri AM, Gupta V, Rathore M (2016) Preparation, properties and applications of organic–inorganic hybrid nanocomposite poly (aniline-co-o-toluidine) tungstomolybdate. J Mol Liq 216:646–653
- <span id="page-13-4"></span>5. Wróblewska-Krepsztul J, Rydzkowski T, Borowski G, Szczypiński M, Klepka T, Thakur VK (2018) Recent progress in biodegradable

polymers and nanocomposite-based packaging materials for sustainable environment. Int J Polym Anal Charact 23(4):383–395

- <span id="page-14-0"></span>6. Neto JSS, Lima RAA, Cavalcanti DKK, Souza JPB, Aguiar RAA, Banea MD (2019) Effect of chemical treatment on the thermal properties of hybrid natural fber-reinforced composites. J Appl Polym Sci 136(10):47154
- <span id="page-14-1"></span>7. Cano AI, Cháfer M, Chiralt A, González-Martínez C (2015) Physical and microstructural properties of biodegradable flms based on pea starch and PVA. J Food Eng 167:59–64
- <span id="page-14-2"></span>8. Pan L, Pei X, He R, Wan Q, Wang J (2012) Multiwall carbon nanotubes/polycaprolactone composites for bone tissue engineering application. Colloids Surf B 93:226–234
- <span id="page-14-3"></span>9. Zhang Y, Yu C, Chu PK, Lv F, Zhang C, Ji J, Wang H (2012) Mechanical and thermal properties of basalt fber reinforced poly (butylene succinate) composites. Mater Chem Phys 133(2–3):845–849
- <span id="page-14-4"></span>10. Khan A, Rangappa SM, Jawaid M, Siengchin S, Asiri AM (eds) (2020) Hybrid fber composites: materials, manufacturing, process engineering. Wiley, Hoboken
- <span id="page-14-5"></span>11. Vehapi M, Yilmaz A, Özçimen D (2020) Fabrication of oregano-olive oil loaded PVA/chitosan nanoparticles via electrospraying method. J Nat Fibers. [https://doi.org/10.1080/15440](https://doi.org/10.1080/15440478.2020.1774463) [478.2020.1774463](https://doi.org/10.1080/15440478.2020.1774463)
- <span id="page-14-6"></span>12. Nayak S, Khuntia SK, Mohanty SD, Mohapatra J (2020) Investigation and fabrication of thermo-mechanical properties of Ceiba Pentandra bark fber/poly (vinyl) alcohol composites for automobile dash board and door panel applications. J Nat Fibers. <https://doi.org/10.1080/15440478.2020.1745124>
- <span id="page-14-7"></span>13. SenthilMuthu Kumar T, Rajini N, Jawaid M, VaradaRajulu A, WinowlinJappes JT (2018) Preparation and properties of cellulose/tamarind nut powder green composites: (green composite using agricultural waste reinforcement). J Nat Fibers 15(1):11–20
- <span id="page-14-8"></span>14. Xia G, Reddy KO, Maheswari CU, Jayaramudu J, Zhang J, Zhang J, Rajulu AV (2015) Preparation and properties of biodegradable spent tea leaf powder/poly (propylene carbonate) composite flms. Int J Polym Anal Charact 20(4):377–387
- <span id="page-14-9"></span>15. Rathinavel S, Saravanakumar SS (2020) Development and analysis of poly vinyl alcohol/orange peel powder biocomposite flms. J Nat Fibers.<https://doi.org/10.1080/15440478.2019.1711285>
- <span id="page-14-10"></span>16. Balavairavan B, Saravanakumar SS, Manikandan KM (2020) Physicochemical and structural properties of green bio flms from poly (vinyl alcohol)/nano coconut shell fller. J Nat Fibers. [https](https://doi.org/10.1080/15440478.2020.1723778) [://doi.org/10.1080/15440478.2020.1723778](https://doi.org/10.1080/15440478.2020.1723778)
- <span id="page-14-11"></span>17. Manikandan KM, Yelilarasi A, Senthamaraikannan P, Saravanakumar SS, Khan A, Asiri AM (2019) A green-nanocomposite flm based on poly (vinyl alcohol)/Eleusinecoracana: structural, thermal, and morphological properties. Int J Polym Anal Charact 24(3):257–265
- <span id="page-14-12"></span>18. Eltayeb NE, Khan A (2020) Preparation and properties of newly synthesized polyaniline@ graphene oxide/Ag nanocomposite for highly selective sensor application. J Mater Res Technol 9(5):10459–10467
- <span id="page-14-13"></span>19. Vu HT, Scarlett CJ, Vuong QV (2017) Efects of drying conditions on physicochemical and antioxidant properties of banana (*Musa cavendish*) peels. Drying Technol 35(9):1141–1151
- <span id="page-14-14"></span>20. Astuti P, Erprihana AA (2014) Antimicrobial edible flm from banana peels as food packaging. Am J Oil Chem Technol 2(2):66–70
- <span id="page-14-15"></span>21. González-Montelongo R, Lobo MG, González M (2010) Antioxidant activity in banana peel extracts: testing extraction conditions and related bioactive compounds. Food Chem 119(3):1030–1039
- <span id="page-14-16"></span>22. Tibolla H, Pelissari FM, Martins JT, Vicente AA, Menegalli FC (2018) Cellulose nanofbers produced from banana peel by chemical and mechanical treatments: characterization and cytotoxicity assessment. Food Hydrocoll 75:192–201
- <span id="page-14-17"></span>23. Rattanavichai W, Cheng W, Chang CC (2017) Simplifed processing method of banana (*Musa acuminata*) peels possess the improvement in immunological responses of *Macrobrachium rosenbergii*. Aquac Res 48(10):5202–5213
- <span id="page-14-18"></span>24. Xue M, Lu W, Chen C, Tan Y, Li B, Zhang C (2019) Optimized synthesis of banana peel derived porous carbon and its application in lithium sulfur batteries. Mater Res Bull 112:269–280
- <span id="page-14-19"></span>25. Emaga TH, Andrianaivo RH, Wathelet B, Tchango JT, Paquot M (2007) Efects of the stage of maturation and varieties on the chemical composition of banana and plantain peels. Food Chem 103(2):590–600
- <span id="page-14-20"></span>26. Kong FB, He QL, Peng W, Nie SB, Dong X, Yang JN (2020) Eco-friendly fame retardant poly (lactic acid) composites based on banana peel powders and phytic acid: fame retardancy and thermal property. J Polym Res 27(8):1–12
- <span id="page-14-21"></span>27. Khawas P, Das AJ, Deka SC (2016) Production of renewable cellulose nanopaper from culinary banana (Musa ABB) peel and its characterization. Ind Crops Prod 86:102–112
- <span id="page-14-22"></span>28. Hassan MM (2018) Enhanced antimicrobial activity and reduced water absorption of chitosan flms graft copolymerized with poly (acryloyloxy) ethyltrimethylammonium chloride. Int J Biol Macromol 118:1685–1695
- <span id="page-14-23"></span>29. Aguirre A, Borneo R, León AE (2011) Properties of triticale four protein based flms. LWT-Food Sci Technol 44(9):1853–1858
- <span id="page-14-24"></span>30. Laxmeshwar SS, Madhu Kumar DJ, Viveka S, Nagaraja GK (2012) Preparation and properties of biodegradable flm composites using modifed cellulose fbre-reinforced with PVA. ISRN Polym Sci. <https://doi.org/10.5402/2012/154314>
- <span id="page-14-25"></span>31. Kamel NA, Abd El-messieh SL, Saleh NM (2017) Chitosan/ banana peel powder nanocomposites for wound dressing application: preparation and characterization. Mater Sci Eng C 72:543–550
- <span id="page-14-26"></span>32. Suki FM, Ismail H, Hamid ZA (2014) Preparation and properties of polyvinyl alcohol/banana frond four biodegradable flm. Prog Rubber Plast Recycl Technol 30(2):103–114
- <span id="page-14-27"></span>33. Zhong OX, Ismail H, Abdul Aziz NA, Abu Bakar A (2011) Preparation and properties of biodegradable polymer flm based on polyvinyl alcohol and tropical fruit waste four. Polym Plast Technol Eng 50(7):705–711
- <span id="page-14-28"></span>34. Wang H, Fang P, Chen Z, Wang S (2007) Synthesis and characterization of CdS/PVA nanocomposite flms. Appl Surf Sci 253(20):8495–8499
- <span id="page-14-29"></span>35. Tahir MH, Zhao Z, Ren J, Rasool T, Naqvi SR (2019) Thermokinetics and gaseous product analysis of banana peel pyrolysis for its bioenergy potential. Biomass Bioenerg 122:193–201
- <span id="page-14-30"></span>36. Peng Z, Kong LX (2007) A thermal degradation mechanism of polyvinyl alcohol/silica nanocomposites. Polym Degrad Stab 92(6):1061–1071
- <span id="page-14-31"></span>37. Vu HT, Scarlett CJ, Vuong QV (2018) Phenolic compounds within banana peel and their potential uses: a review. J Funct Foods 40:238–248
- <span id="page-14-32"></span>38. Silva VDM, Macedo MCC, Rodrigues CG, dos Santos AN, Loyola ACDF, Fante CA (2020) Biodegradable edible flms of ripe banana peel and starch enriched with extract of *Eriobotrya japonica* leaves. Food Biosci 38:100750
- <span id="page-14-33"></span>39. Pelissari FM, Andrade-Mahecha MM, Amaral Sobral PJ, Menegalli FC (2013) Comparative study on the properties of four and starch flms of plantain bananas (*Musa paradisiaca*). Food Hydrocoll 30(2):681–690
- <span id="page-14-34"></span>40. Vorobyeva OV, Andrusenko SF, Volosova EV, Avanesyan SS, Ivanova AM, Kadanova AA (2011) Modifcation of natural polymers for synthesis of biodegradable materials. Chem Sustain Dev 19(2):131–134
- <span id="page-14-35"></span>41. El Bourakadi K, Merghoub N, Fardioui M, Mekhzoum MEM, Kadmiri IM, Essassi EM, Bouhfd R (2019) Chitosan/polyvinyl alcohol/thiabendazoluim-montmorillonite bio-nanocomposite

flms: mechanical, morphological and antimicrobial properties. Compos B 172:103–110

- <span id="page-15-0"></span>42. Rajini N, Alavudeen A, Siengchin S, Rajulu V, Ayrilmis N (2019) Development and analysis of completely biodegradable cellulose/ banana peel powder composite flms. J Nat Fibers. [https://doi.](https://doi.org/10.1080/15440478.2019.1612811) [org/10.1080/15440478.2019.1612811](https://doi.org/10.1080/15440478.2019.1612811)
- <span id="page-15-1"></span>43. Guo G, Xiang A, Tian H (2018) Thermal and mechanical properties of eco-friendly poly (vinyl alcohol) flms with surface treated bagasse fbers. J Polym Environ 26(9):3949–3956
- <span id="page-15-2"></span>44. Perumal AB, Sellamuthu PS, Nambiar RB, Sadiku ER (2018) Development of polyvinyl alcohol/chitosan bio-nanocomposite flms reinforced with cellulose nanocrystals isolated from rice straw. Appl Surf Sci 449:591–602
- <span id="page-15-3"></span>45. Asad M, Saba N, Asiri AM, Jawaid M, Indarti E, Wanrosli WD (2018) Preparation and characterization of nanocomposite flms from oil palm pulp nanocellulose/poly (Vinyl alcohol) by casting method. Carbohyd Polym 191:103–111
- <span id="page-15-4"></span>46. Orsuwan A, Sothornvit R (2015) Efect of miniemulsion crosslinking and ultrasonication on properties of banana starch. Int J Food Sci Technol 50(2):298–304
- <span id="page-15-5"></span>47. Ismail H, Zaaba NF (2011) Efect of additives on properties of polyvinyl alcohol (PVA)/tapioca starch biodegradable films. Polym Plast Technol Eng 50(12):1214–1219
- <span id="page-15-6"></span>48. Chen L, Imam SH, Gordon SH, Greene RV (1997) Starch-polyvinyl alcohol crosslinked flm—performance and biodegradation. J Environ Polym Degrad 5(2):111–117
- <span id="page-15-7"></span>49. Gontard N, Duchez C, Cuq JL, Guilbert S (1994) Edible composite flms of wheat gluten and lipids: water vapour permeability and other physical properties. Int J Food Sci Technol 29(1):39–50
- <span id="page-15-8"></span>50. Guimarães M, Vagner RB, Kátia MN, Fábio GT, Gustavo HDT (2015) High moisture strength of cassava starch/polyvinyl alcohol-compatible blends for the packaging and agricultural sectors. J Polym Res 22(10):192
- <span id="page-15-9"></span>51. Guohua Z, Ya L, Cuilan F, Min Z, Caiqiong Z, Zongdao C (2006) Water resistance, mechanical properties and biodegradability of methylated-cornstarch/poly (vinyl alcohol) blend flm. Polym Degrad Stab 91(4):703–711
- <span id="page-15-10"></span>52. Campos AD, Marconato JC, Martins-Franchetti SM (2011) Biodegradation of blend flms PVA/PVC, PVA/PCL in soil and soil with landfll leachate. Braz Arch Biol Technol 54(6):1367–1378

**Publisher's Note** Springer Nature remains neutral with regard to jurisdictional claims in published maps and institutional affiliations.