**ORIGINAL PAPER**



# **A Novel, Simple Route to Produce Urea:Urea–Formaldehyde Composites for Controlled Release of Fertilizers**

**A. S. Giroto1,2 · G. G. F. Guimarães2 · C. Ribeiro2**

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#### **Abstract**

Nitrogen loss through  $NH_3$  volatilization is a primary concern for urea as fertilizer due its fast hydrolysis by soil urease. To minimize this problem herein we developed a partially-polymerized urea–formaldehyde granule as a slow-release fertilizer, by melt stage process as a viable route for large-scale production. In this product the unreacted urea fraction acts as a fast release nutrient source while the polymerized fraction acts in longer times depending on the polymerization degree. This characteristic was analyzed by means of soil incubation experiments (up to 42 days), where the available  $NH_4^+$  contents along time indicated significant lower N losses compared to conventional fertilizer, even for low-polymerized materials. Residual N in the structure was kept stored in the soil for future use by plants, as desired in many agricultural practices, showing that this simple polymerization method provides a smart fertilizer controlled by chemical structure.

**Keywords** Nitrogen losses · Partial polymerization · Controlled-release · Fertilizer

## **Introduction**

Urea is one of the main sources of N used due to its high concentration (46% N) that allows concentrated formulations at lower costs compared to other sources [\[1](#page-9-0)]. However, this fertilizer shows less efficiency than other nitrogen sources for a large number of crops in different soils and climates, due to different causes, such as  $NO<sub>3</sub><sup>-</sup>$  leaching, NH<sub>3</sub> volatilization, and its toxic effect on the plants at the beginning of the vegetative period  $[2-4]$  $[2-4]$ . These factors not only contribute to reducing the efficiency of urea, biomass production, and economic cultivation but also are an important source of environmental pollution.

One of the alternatives to reduce these losses is to reduce urea hydrophilicity, allowing its application as slow release fertilizers. In this scenario, urea formaldehyde (UF) was the first synthetic nitrogen fertilizer produced with low solubility and it belongs to the first group of materials specially

developed on a commercial scale for the slow release of nitrogen [\[5](#page-9-3)[–7](#page-9-4)]. Studies have verified that N-release from an organic polymer occurs in stages similar to those adopted for urea in agriculture  $[8-10]$  $[8-10]$ . However, most of the research into this polymer synthesis reports the production in the liquid phase, as seen by many patents  $[11-15]$  $[11-15]$ . This process is difficult to control, generally leading to fast polymerization with poor processability. Besides that, the literature does not show strategies for partial polymerization, i.e., all the available products are based on fully-polymerized systems. Therefore, the nutrient release is often too long for practical application, meaning that the nutrient is available for longer periods than the requested by aplantgrowth. This fact explains why urea–formaldehyde polymers are not popular as controlled release fertilizers, despite their higher N content compared to other commercial products. In recent research from our group, we proposed a novel nanocomposite material based on the exfoliation of montmorillonite into the matrix of a urea/urea–formaldehyde polymer [\[6](#page-9-9)]. A remarkable aspect of this process was the use of a formaldehyde solid precursor instead of the most common formaldehyde solution. Despite its simplicity, this method was limited as to the maximum amount of urea in composition (around 80%), and by the need of partial urea solubilization for a plastic mixture. Considering the basic strategy, herein we propose to investigate the polymerization in the urea melt

 $\boxtimes$  C. Ribeiro caue.ribeiro@embrapa.br

<sup>&</sup>lt;sup>1</sup> Department of Chemistry, Federal University of São Carlos, Washington Luiz Highway, km 235, São Carlos, SP 13565-905, Brazil

<sup>2</sup> Embrapa Instrumentation, 1452, XV de Novembro Street, CP: 741, São Carlos, SP 13560-970, Brazil

stage, by also using the formaldehyde precursor previously mixed as powders. The results showed that by this simple method it is possible to obtain dense urea:urea–formaldehyde composites, and with very high N contents (comparable to commercial products), and with nutrient release easily controlled by the composition.

# **Materials and Methods**

### **Materials**

Raw materials used in composite formulations wereurea (Synth) and paraformaldehyde (Sigma-Aldrich). The urea was previously milled to 300 mm in a TE-330 hammer mill (Tecnal, Brazil).

#### **Preparation of Materials**

Composites were prepared with different molar ratios of urea and paraformaldehyde (Ur/Pf) as 1:1, 1:0.5 and 0.5:1, respectively. The urea–formaldehyde polymer wasformed by reaction of formaldehyde with an excess of urea. In the first stage, urea is hydroxymethylated by the addition of formaldehyde to the amino groups. This reaction actually is a series of reactions that lead to the formation of mono-, di-, and trimethylolureas. The second stage of urea–formaldehyde resin synthesis consists of condensation of the methylolureas to low molecular weight polymers. The exact ratio, of course, is dependent on the reaction conditions employed in the addition reaction  $[6]$  $[6]$ .

Paraformaldehyde is a formaldehyde polymer with a polymerization degree between 8 and 100. This polymer depolymerizesin water and heat [[16\]](#page-9-10). Thus, the use of paraformaldehyde allows control of the urea solubility inside the composite by the extension of the polymerization reaction. The precursor materials (urea and paraformaldehyde) and water (9 wt%) were pre-homogenized and mixed-melted using a torque rheometer (Polylab RHEODRIVE Rheomix mixer and OS4) under the conditions of 60 rpm for 10 min at 90 °C. After mixing, the samples were cured at 80 °C in an oven for 12 h and subsequently stored at 90 °C until completely dry. The nomenclature given to each composition and the elementary composition ispresented in Table [1.](#page-1-0)

### **Characterizations**

The materials wereinitially characterized by scanning electron microscopy (SEM) using a JSM6510 microscope (JEOL). Samples were previouslyfixed onto carbon tapes and coated with thin layer of gold in an ionization chamber(BALTEC Med. 020). The imaging by SEM was carried out using the secondary electron mode. The optical

<span id="page-1-0"></span>



The raw materials used are indentified by initial Ur for urea and Pf for paraformaldehyde

image from paraformaldehyde was done using an optical microscopyL2800 Biological Microscope (BEL Photonics, Brazil).

X-ray diffractograms (XRD) were obtained using an XRD 6000 diffractometer (Shimadzu). The relative intensity of diffraction was registered in the angular range  $(2\Theta)$ of  $3^{\circ}-40^{\circ}$ , using a Cu K $\alpha$  incident beam (k = 0.1546 nm) and scannings peed of 1° min−1 at 30 kV voltage and 30 mA current.

Thermal degradation of samples was evaluated in the 25–600 °C range using a Q500 analyzer (TA Instruments, New Castle, DE, USA) under the following conditions: sample size  $10.0 \pm 0.5$  mg, synthetic air atmosphere (80%) N<sub>2</sub> and 20% O<sub>2</sub>) with 60 mL min<sup>-1</sup> flow, heating rate of 10 °C min−1. For differential scanning calorimetry (DSC), samples were heated from 25 to 300 °C under anitrogen flow of 60.0 mL min<sup>-1</sup> in a DSC Q100 (TA Instruments, USA). The linear heating rate used in both analyses was 10 °C min−1.

 $13<sup>13</sup>C$  solid-state nuclear magnetic resonance (NMR) measurements were performed on a Bruker (Karlsruhe, Germany) 400 MHz to  ${}^{1}H$  (9.4 T) spectrometer using  ${}^{13}C$  cross polarization magic angle spinning (CPMAS) technique in a 4 mm broadband <sup>1</sup>H/X MAS probehead. The following parameters and values were used: speed spinning of  $8$  kHz,  $90<sup>1</sup>H$  pulse length of 2.3 μs, contact time 2 ms, spectral width 50 kHz, 20 Hz of line broadening, and a recycle delay of 3 s. Hexamethylbenzene (HMB) was used as external reference standard at 17.2 ppm for methyl-carbons. All the NMR experiments were carried out with 4096 scans. Fourier Transform Infrared (FT-IR) analyses were performed on a Bucker spectrometer using aspectral resolution of  $4 \text{ cm}^{-1}$ .

#### **Evaluation of Urea Release in Water**

The urea release in water at 25 °C and natural pH for each polymer was examined following a method adapted from Tomaszewwska and Jarosiewicz. The total N organic content in each composite formulation was determined by elemental analysis (Thermo Finnigan Model Flash 1112EA). These results (Table [1\)](#page-1-0) were used to calculate the amount of material used in the release tests. Briefly, samples were soaked in water at 25 °C and were constantly homogenized by gentle stirring using an orbital shaker during 5 days. Pure urea was also tested as a control experiment [\[17](#page-9-11)]. The urea concentration in solution was determined by UV–Vis spectrophotometry in a 1601PC spectrophotometer (Shimadzu, Japan) according to the method reported by With et al. [\[18](#page-9-12)]. Curves of the percentage of urea-solubilized as a function of time were then obtained. The measurements were replicated three times under identical laboratory conditions for each type of composite formulation.

#### **Release and Transformation of N in Soils**

The release of urea from the composites was also evaluated in two Oxisols, both were collected in the surface (0–20 cm) in an area with agricultural activity in the municipality of São Carlos, Brazil. Before use, the soil samples were airdry and crushed to pass through a 2 mm screen. The soil chemical and physical properties are reported in Table [2.](#page-2-0) The pH was determined with a glass electrode (soil:water ratio, 1:1); organic C by the Walkley–Black method [[19\]](#page-9-13); cation-exchange capacity (CEC) [[20](#page-9-14)]; soil texture analysis was determined by the pipette method  $[21]$ ; water-holding capacity (WHC) [[20\]](#page-9-14); and urease activity using the buffer method of Tabatabai and Bremner [[22\]](#page-9-16).

To compare the urea release and N transformations, soil samples were incubated with urea or composites at a ratio of soil:N (1000:1 g  $g^{-1}$ ) placed in 125 mL polyethylene screwcap bottles as Urea, Ur/Pf 1:1, 1:0.5 and 0.5:1. The samples were incorporated into the soils and soil moisture content was standardized at 80% WHC with the addition of deionized water. A container with 5 mL of 4% boric acid (Sigma-Aldrich) was added in the polyethylene bottles, to capture the volatilized ammonia  $(NH<sub>3</sub>)$  during the incubation period. Samples were incubated for 0, 1, 3, 7, 14, 21 and 42 days under controlled air temperature and humidity.

<span id="page-2-0"></span>**Table 2** Chemical and physical properties of the studied soils

Oxisol	Red–Yellow	Red
pН	5.4	5.0
Organic C $(g \text{ kg}^{-1})$	7.6	7.0
$CEC$ (cmol <sub>c</sub> kg <sup>-1</sup> )	4.8	4.2
Sand $(g \text{ kg}^{-1})$	667	433
Silt $(g \text{ kg}^{-1})$	19	35
Clay $(g \ kg^{-1})$	314	532
WHC $(g \text{ kg}^{-1})$	140	200
Urease activity (mg N $\text{kg}^{-1} \text{h}^{-1}$ )	9.7	7.1

*CEC* cation exchange capacity, *WHC* water-holding capacity

Analyzes were performed after each incubation period. Determinations were carried out via volatilization of boric acid by titration with HCl 0.01 mol  $L^{-1}$  (Synth) and then the soil mineral N was extracted by KCl 1 mol  $L^{-1}$  (Vetec) at the extractant:soil ratio (10:1). In order to paralyze the urease activity, 5 mg  $L^{-1}$  of phenylmercuric acetate (Sigma-Aldrich) from the extraction stage was added to the extraction solution. The suspension remained under stirring for 1 h and was then filtered with the use of a slow filter with a diameter of 12.5 cm (Nalgon). The extract was stored in a polyethylene bottle with a volume of 100 mL at 5 °C in a refrigerator.

The ammonium  $(NH_4^+)$  and nitrate  $(NO_3^-)$  levels in the extracted soil were analyzed by the colorimetric method [[23,](#page-9-17) [24](#page-9-18)]. The contents recovered in each N fraction were expressed as percentages referring to the N applied as urea or composites. The profile of the volatilization of ammonia  $(NH_3)$  and the formation of  $NH_4^+$  in soil was presented by the average values accompanied by their respective standard deviations. Differences among treatments by total recovery as  $NH_3$  volatilized and exchangeable  $NH_4^+$  and  $NO_3^-$  after aerobic incubation were assessed by analysis of variance (ANOVA) and when the F test was significant, differences among treatments were compared by the Tukey test  $(P < 0.05)$ .

## **Results and Discussion**

In previous works urea and formaldehyde were used to synthesize UF resins and the reaction was always performed in aqueous solution  $[11-13]$  $[11-13]$ . Here, in order to improve the yield and stabilization of UF resins, paraformaldehyde was used as the raw material and the reaction was performed in urea melt. This work proposes using just the mixing–melting between urea and paraformaldehyde follow by cure and drying of materials, avoiding large amounts of water in addition to acids and basic reagents used to control the reaction medium. Therefore, the data for characterization of formed polymers are presented below.

Figure [1](#page-3-0) shows XRD patterns of the pure urea and paraformaldehyde and composites with different compositions. The pattern of neat urea and paraformaldehyde showed typical peaks at  $2\Theta = 22.5^{\circ}$ ,  $24.7^{\circ}$ ,  $29.3^{\circ}$ ,  $31.8^{\circ}$ , 35.5°, 37.3° and 22.9°, 34.3° for the precursor materials, respectively. By the XRD patterns of samples, it was possible to verify the polymerization by a wide band in the reacted samples, which is characteristic of a semi-crystalline material. This band is proportional to the molar ratio of paraformaldehyde in mixtures, i.e., the polymerization is directly influenced by the amount of this compound, as expected. This is consistent with the findings for Roumeli et al. [\[25](#page-9-20)], who observed the highest amorphous proportion



<span id="page-3-0"></span>**Fig. 1** XRD patterns for neat urea, paraformaldehyde, and formulations of urea/paraformaldehyde at different molar ratios



<span id="page-3-1"></span>**Fig. 2** DSC curves for neat urea, paraformaldehyde and polymers

in urea:formaldehyde composites at a ratio of 0.5:1 (Ur/Pf), i.e., excess formaldehyde.

DSC experiments were used to confirm the thermal behavior of the polymerization reaction among components. Figure [2](#page-3-1) shows DSC profiles for neat components and mixtures, where the endothermic peak of paraformaldehyde at 123 °C is absent in any of the mixtures—which is an indicator that this component was totally consumed during the polymerization step, as reported by Yamamoto et al. [\[6\]](#page-9-9). Regarding urea, melting at 135 °C and decomposition between  $150 < T < 210$  °C are clearly seen [[26](#page-9-21)]. As presented in Fig. [1](#page-3-0) for the Ur/Pf 0.5:1 polymer (which is supposed to be a fully polymerized material, since paraformaldehyde is in stoichiometric excess), the shift to 246 and 286 °C is confirming the complete polymerization (probably associated with degradation steps). For the other samples, peaks suggesting to free urea degradation were still present (162 and 178 °C).

Figure [3](#page-4-0) shows the TGA/DTG curves for neat precursors and polymers. This analysis confirms the polymerization degree of different samples. In the urea TGA curve, three distinct thermal events are visible, urea decomposition onset  $160 < T < 210$  °C and degradation around 237 °C (loss of amine groups) and around 292 °C degradation of polymerized fractions [\[26\]](#page-9-21). It was also observed that paraformaldehyde shows a single mass loss step at 150 °C, which is related to a direct volatilization process—confirming that the peak at 123 °C in DSC (Fig. [2](#page-3-1)) is associated with this process. For Ur/Pf 0.5:1 sample a significant (68%) mass loss at the range 174–384 °C corresponding to depolymerization/ volatilization/decomposition of Ur/Pf polymer followed by a second mass loss around 20% due to an initial degradation of the polymer chain. Considering that the proportion 0.5:1 is relative to the material more polymerized a residual of 8.54% is observed after 600 °C probably due to strong crosslinking formed between urea and formaldehyde groups in the final material. The thermal profile of sample 0.5:1 shows consequently more degradation peaks in the DTG curve, related to the presence of the urea–formaldehyde polymer. This was also seen in the Ur/Pf 1:1 sample but with a less amount of residual material (5.43%) probably owing a lower amount of urea polymerized in comparison to the previous material with a higher proportion of paraformaldehyde (0.5:1). While the material Ur/Pf 1:0.5 had no weight left after 600 °C. This was expected since Ur/ Pf stoichiometry for full polymerization is at least 0.5:1, thus the others materials should be classified as urea:UF composites due to free urea still presented. However, even in these materials, DTG data reveal that the majority of the material is polymerized since the main peak weight loss is at 280 °C for all polymers. These results showed that the 0.5:1 ratio is necessary if complete urea polymerization is desired [[27\]](#page-9-22). Furthermore, the results showed that Ur/Pf polymers synthesized by melt condensation polymerization have lower free formaldehyde content, high thermal stability, and better stability during storage [\[28](#page-9-23)].

Figure [4](#page-4-1) shows SEM micrographs of the pure urea and obtained polymers and paraformaldehyde optical image. The urea shows regular size particles larger than 50 µm. The morphology of the Ur/Pf 1:0.5 (Fig. [4](#page-4-1)b) shows that the surface is composed of rod-shaped and fibrous crystals formed above a homogenous surface, referred to as unreacted urea. For the sample Ur/Pf 1:1, the reaction leads to large amounts

of thicker needle-shaped particles, indicating the higher polymerization degree, as supposed. Finally, Ur/Pf 0.5:1 showed a smooth and homogeneous surface, which indicates the change in morphological evolution during polymerization—since previous results show the complete urea polymerization in this material. As observed by the optical image, the paraformaldehyde resembles small granular, nearspherical particles, whose morphology is not seen in the polymerized samples. This indicates that the needle-shaped re-crystallization mechanism, and are not remnants of the precursor materials. This needle-shaped structures tend to coalesce to larger rods, leading to almost continuous phases, as seen in Ur/Pf 0.5 sample. Therefore, it may be proposed that this shape results from the formation of small seeds in liquid/dissolved phase, precipitating over urea granules. Following, the growth of needle-shape structure is defined by the preferential growth in high-energy planes, which is surpassed by isotropic growth in high crystallization extension, leading to bigger crystals by classical growth or coalescence.

<span id="page-4-1"></span>**Fig. 4** SEM micrographs for **a** pure urea, **b** Ur/Pf 1:0.5, **c** Ur/Pf 1:1, **d** Ur/Pf 0.5:1 and **e** optical image from paraformaldehyde

<span id="page-4-0"></span>

This is almost similar to the crystal growth from melt, typically seen for metals and many other polymers crystallized from solution [\[27\]](#page-9-22).

The  $^{13}$ C NMR analyses were performed upon the synthesis of polymers in order to better understand the chemical structure formed after curing process, as seen in Fig. [5](#page-5-0)a. Based on  $^{13}$ C chemical shifts it is possible to evaluate the presence of various functional groups on the polymers. Table [3](#page-8-0) summarizes the chemical shifts confirming the proposed reactions. In the first step, the hydroxymethylation refers to the addition of up to three formaldehyde molecules with a urea bifunctional molecule thereby producing hydroxymethylureas, as noted by the chemical shift around 65–66 and 71–72 ppm. In some cases, hydroxymethylation conditions (slightly alkaline pH and temperatures around 70 °C) also favor the formation of methylene ether bridges between hydroxymethylureas as observed for the polymer Ur/Pf 1:1 at 76 ppm. These ether bridges may later undergo rearrangement to form methylene bridges  $(-CH<sub>2</sub>-)$ , thereby releasing formaldehyde molecules.

The second stage, acid condensation, epitomizes the oligomer Ur/Pf. The hydroxymethylureas, the free urea, and free formaldehyde react to form linear or partially branched molecules with average or high molecular weights. The type of bridge predominantly formed at this stage is the methylene bridge with chemical shifts around 47–48, 53–54 and 60–61 ppm as observed for all materials. In the third synthesis step, the second portion of urea reacts with the free formaldehyde producing more hydroxymethylureas. It was possible to verify the formation of species with different degrees of free urea substitution. For the chemical shifts shown there is a priority formation of mono-substituted urea for materials with the 1:1 and 1:0.5, materials with lower polymerization degrees and the formation of di-substituted urea for the polymer Ur/Pf 0.5:1, the polymer with the highest polymerization degree. This material also exhibited chemical shifts regarding branched chemical groups [\[27–](#page-9-22)[30](#page-9-24)]. It was not observed any paraformaldehyde band left (in the region of 80–90 ppm) which is attributed to single carbon present (reactive formaldehyde oligomer) as also observed by Hoong et al., Fan et al. and Tomita et al., [[31](#page-9-25)[–33](#page-9-26)] showing that all paraformaldehyde was polymerized.

Figure [5](#page-5-0)b shows a comparison of FT-IR spectra from 4000 to 400 cm−1 between pure urea, neat paraformaldehyde, and Ur/Pf polymers. All the changes of amine stretching vibration frequency indicate the formation of Ur/Pf polymer, which was consistent with previous work [\[25](#page-9-20), [27](#page-9-22), [34](#page-9-27)]. It was possible to see that multiple and broad peaks in the Ur/Pf resin spectra are mainly due to the polymer structure complexity. The bands at 3439 and 3336 cm−1 correspond to N–H asymmetric and symmetric stretching of urea as shown for pure urea and Ur/Pf 1:0.5 clearly exhibit that the same

<span id="page-5-0"></span>

NMR of urea and polymers, **b** FTIR transmittance spectra of pure urea, neat paraformaldehyde, Ur/Pf 1:0.5, Ur/Pf 0.5:1 and Ur/Pf 1:1

stretching frequency disappears and shifts to higher regions (3422 cm−1) after the reaction for Ur/Pf 1:1 and Ur/Pf 0.5:1. The broad peak around 3350–3450 cm<sup>-1</sup> can be also attributed to the hydrogen bonded O–H and N–H. The fact that this band is rather broad may be attributed to monomers such as water and formaldehyde, whose O–H group may form hydrogen bonds with reactive functional groups such as CH<sub>2</sub>OH, NH<sub>2</sub> and N–H, as observed by Jada  $[36]$  $[36]$ . But no band was found in the regions of 1097–905 cm<sup>-1</sup> (C–O aliphatic ether) referent to paraformaldehyde indicating that no residue of this material was presented in final polymers [\[37](#page-10-1)]. Also, it is important to mention that the free  $-NH<sub>2</sub>$  group has a characteristic peak at  $3440 \text{ cm}^{-1}$ , while the bonded –NH group at 3340 cm<sup>-1</sup> [\[25\]](#page-9-20). In spectra for urea, the characteristic bands of amide I, II, and III are around 1680, 1612 and  $1463 \text{ cm}^{-1}$ , respectively. These peaks are assigned to the C–O stretching of amide I and II, as well as the –N–H scissors of amide I. The multiple peaks at  $1460-1470$  cm<sup>-1</sup> may be attributed to C–H bending vibrations of the  $CH<sub>2</sub>–N$ group, while the small peaks in the area of  $1320-1450$  cm<sup>-1</sup> can be assigned stretching of groups like  $CH<sub>2</sub>OH$ ,  $CH<sub>3</sub>$  and CN as observed also by Zhong et al. [[38\]](#page-10-2). The shift has also been assigned to C–H stretching and –O–H bending vibrations of alcohol, but in Ur/Pf 1:1 theses peaks almost disappear. The fact that in the pure UF resin spectrum, the peak is centered at the area of 3410 cm<sup>-1</sup> in the space of Ur/Pf 0.5:1, indicates that the amount of bonded –NH is higher compared to the free –NH<sub>2</sub>. The 1255 cm<sup>-1</sup> peak is attributed to both the asymmetric stretch of  $N-CH_2-N$  and the asymmetric stretch of –C–O–C– of ether linkages [[35\]](#page-9-28). Also, the peak at 2958 cm<sup>-1</sup> becomes sharper and more distinct, which along with the absence of the 2900  $cm^{-1}$  peak of the asymmetric C–H alcohol stretches, indicates that quantity of ether and methylene linkages is increased in the cured resin and thus, a cross-linking process has evolved.

Figure [6](#page-6-0) shows the urea release profile in water (full immersion) for all samples. It is noteworthy that all materials had a slowed behavior in the urea release, even the non-completely polymerized sample (Ur/Pf 1:0.5) when compared to urea. The degree of polymerization plays a key role in this process since the release time and the equilibrium value over 50 h was directly dependent on the polymer content. It was observed that Ur/Pf 1:0.5 showed a behavior very close to pure urea. Considering that this sample is characterized by a partially polymerized structure, this release may correspond to the solubilization of that unreacted urea, but also to the polymer itself—since this is probably a low molecular weight polymer, it is possible that the structure presents some solubility and then, part of the solute correspond to oligomers or soluble polymeric chains. However, this suggests that the time for total urea release for this polymer may be greater than 120 h, i.e., 5 days of complete immersion in water. For other materials, urea releases only 30 and 20%



<span id="page-6-0"></span>**Fig. 6** Kinetics of urea release at 25 °C in an aqueous medium (neutral pH) for commercial urea and Ur/Pf polymers

for Ur/Pf 1:1 and 0.5:1 were observed, respectively, at the end of 50 h (equilibrium time). This value is almost constant until the end of the experiment, suggesting that the maximum solubilization degree was attained. Altogether, this full immersion test is very aggressive and may not correspond to real conditions in the field, where the moisture level is limited by the soil uptake capacity.

Therefore, in order to analyze the release of nitrogen for the prepared materials in near-real conditions, incubation experiments were performed, and the influence of soil moisture and temperature on the volatilization of  $NH<sub>3</sub>$  and formation of  $NH_4^+$  and  $NO_3^-$  were analyzed—as seen in Figs. [7,](#page-7-0) [8](#page-7-1) and Table [4,](#page-8-1) respectively. The samples behavior was clearly influenced by the polymerization extent. Figure [7](#page-7-0) shows the  $N$  recovery as  $NH<sub>3</sub>$  volatilized during aerobic incubation of Ur or polymers (Ur/Pf) applied in both type soils studied. The rapid urea release from the Ur treatment demonstrated the high  $NH_3$  volatilization exceeded 60%, an amount poorly affected by the soil type (Red–Yellow or Red Oxisol). However, the hydrolysis of urea was faster in the Red–Yellow than Red, which can be observed by the over 50% ammonia volatilization until the first week for the Red–Yellow soil, while Red soil attained this level only at the third week. This result can be attributed the much higher urease activity and higher sand content of the Red–Yellow soil, as reported in Table [2](#page-2-0), which intensifies the  $NH<sub>3</sub>$  volatilization trend. The high  $NH<sub>3</sub>$  losses from urea in the incubation experiment can be attributed to low CEC of both soils (Table [2](#page-2-0)) agreeing with Guimarães et al. [\[39](#page-10-3)].

The polymerization of urea with paraformaldehyde provided control of urea release, even in the partially polymerized sample (Ur/Pf 1:0.5) in both soils and inhibited the N losses (Fig. [7](#page-7-0)). This sample was characterized by



<span id="page-7-0"></span>**Fig. 7** Fertilizer N recovery as  $NH<sub>3</sub>$  volatilized during the aerobic incubation of urea (Ur), or polymers (Ur/Pf) applied to the **a** Red– Yellow Oxisol and **b** Red Oxisol. Data are shown as a mean of triplicate incubations with bars to indicate the standard error of the mean



<span id="page-7-1"></span>**Fig. 8** Fertilizer N recovery as ammonium  $(NH_4^+)$  during the aerobic incubation of urea, or polymers (Ur/Pf) applied to the **a** Red–Yellow Oxisol and **b** Red Oxisol. Data are shown as an average of triplicate incubations with bars to indicate the standard error of the mean

a reduced polymerization degree evidenced by structural analysis (Table [3\)](#page-8-0) which intensified the release of urea during the incubation period. Before urea hydrolysis, the breaking of the fragment structure and release of the urea molecule into the soil is necessary. This breaking can be

attributed to the microbial activity in soil that was faster for the treatment with low rate paraformaldehyde. It is important to notice that, even with this sample presenting a possible significant amount of low molecular weight polymers, by the synthesis conditions, these soluble molecules were also effective to reduce the total  $NH<sub>3</sub>$  volatilization.

Figure [8](#page-7-1) shows recovery of exchangeable  $NH_4^+$  during the incubation period in the Oxisols. The  $NH_4^+$  content reflects production through urea hydrolysis, primarily consumption via NH<sub>3</sub> volatilization and low effect of nitrification. For both soils, the unamended urea shows a fast  $NH_4^+$ content increase until the first week after application in the soil, however, it is followed by a decrease provided by the loss of N from ammonia volatilization. The polymers provide a more constant and homogeneous  $NH_4^+$  release during the incubation period. Unfortunately, the content of  $NH_4^+$  from Ur/Pf composites was low during the 6-week study attributed to high control urea release, which is consistent with the evidence of the increase rate of paraformaldehyde in the treatments. The smallest polymerization of Ur/Pf 1:1 and Ur/Pf 1:0.5 provided higher of  $NH_4^+$ release than Ur/Pf 0.5:1 during incubation in Red–Yellow and more consistent for Ur/Pf 1:0.5 than other treatments in Red Oxisol. No definite explanation can be offered for low nitrification, which remained lower than 3.5% during the incubation period for both soils.

Table [4](#page-8-1) compares the fertilizer N recovery as total volatilized  $NH_3$ , and total exchangeable  $NH_4^+$  and  $NO_3^-$  after aerobic incubation intervals of urea and composites (Ur/ Pf) in two Oxisols. In both soils, the total N losses from urea alone showed similar behavior exceeding 60%. In contrast, the Ur/Pf 1:0.5 treatment did not exceed 8%, while the treatment with the highest polymerization degree showed insignificant N losses. This advantage evidences the efficiency of the paraformaldehyde to control urea release and N losses.

Similarly, the lower total  $NH_4^+$  content of composites compared to the Ur treatment can be attributed to the more gradual release of urea. Therefore, the increase of the polymerization paraformaldehyde percentage decreases  $NH_4^+$  content in the soil. However, the Ur/Pf polymers present high urea content without release to the soil after 6 weeks of incubation, which could be released in the longer period.

The total N recovery as  $NH_3$ ,  $NH_4^+$ , and  $NO_3^-$  from unamended urea was 96% for Red–Yellow and 92% for Red Oxisol (Table [4\)](#page-8-1). In contrast, the total N recovery of composites does not exceed 24%, which demonstrates that these compounds have more than 75% N to be released after 42 days. The N recovery as exchangeable  $NO<sub>3</sub><sup>-</sup>$  did not show differences between the treatments, with values not exceeding 3.5% for Red–Yellow and 2.8 for Red Oxisol.

<span id="page-8-0"></span>**Table 3** 13C NMR solid chemical shifts of some of the structural fragments that appear in the Ur/Pf resins showed in Fig. [4](#page-4-1)



<span id="page-8-1"></span>Table 4 Fertilizer N recovery as NH<sub>3</sub> volatilized and exchangeable  $NH_4^+$  and  $NO_3^-$  after aerobic incubation of Red–Yellow or Red Oxisol

Treatment	Total NH <sub>3</sub> $(\%)$	Total $NH_4^+$ (%)	Total $NO_3$ <sup>-</sup> $(\%)$
Red–Yellow			
Ur	60.4 a <sup>1</sup>	33.2 a	3.24a
Ur/Pf $1:0.5$	5.02 <sub>b</sub>	15.5 <sub>b</sub>	3.51 a
Ur/Pf 1:1	0.35c	12.2c	3.38 a
Ur/Pf 0.5:1	0.35c	7.1 <sub>d</sub>	2.84a
Red			
Ur	62.16a	27.26a	2.75a
Ur/Pf $1:0.5$	7.51h	8.47 b	1.86 a
Ur/Pf 1:1	0.28c	4.33 c	2.80a
Ur/Pf 0.5:1	0.28c	2.06c	2.30a

1 Mean values reported from triplicate incubations. Values within a column followed by the same letter do not differ significantly by the Duncan test at  $p < 0.05$ 

# **Conclusion**

The results showed a simple method, based on mixing–melt of urea with different amounts of paraformaldehyde, that was effective to produce N controlled release materials. The composition plays a key role in the release profiles since the polymerization extent was shown as the mechanism needed to delay N solubilization. This was only possible by controlling the in-situ production of formaldehyde through the degradation of a solid precursor, paraformaldehyde, in processing conditions comparable to the expected for melt extrusion. It is worthy to mention that, in this strategy, the conventional setup for polymer extrusion processing may be easily adapted to produce this kind of fertilizer, which is an important technological advance compared to the conventional fertilizing products. This also influences on the actual chain length, i.e., the partial polymerization was shown as an effective strategy to produce oligomers and low-molecular-weight polymers, which will interact faster to the soil biota—releasing the nutrient. Also, the soil profile, analyzed in two Oxisols, was seen as an important factor for release: soils with higher urease activity induced higher polymer solubility, however at low levels—indicating that these materials can avoid volatilization effects, even in the low-polymerized materials (with higher free urea contents). Finally, considering that the polymers are expected to continue their degradation behavior, the residual N in the structure is maintained for further use in the soil, as desired in many agricultural practices.

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