

Composites of BiVO₄ and g-C₃N₄: Synthesis, Properties and Photocatalytic Decomposition of Azo Dye AO7 and Nitrous Oxide

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Abstract

The composites of BiVO₄ and g-C₃N₄ (BiVO₄/g-C₃N₄) were synthesised by the calcination of a mixture of monoclinic BiVO₄ and bulk g-C₃N₄ at 300 °C for 4 h. Both components were previously prepared by the precipitation of Bi(NO₃)₃ with NH₄VO₃ and annealing of melamine. X-ray photoelectron spectroscopy (XPS) identified the presence of C–O and C=O bonds as well as metal nitrides which confirmed the formation of a heterojunction between BiVO₄ and g-C₃N₄. The heterojunction was also indicated by UV–Vis diffuse reflectance (DRS) and photoluminescence (PL) spectroscopy. The band gap energies were determined at 2.42–2.46 eV of BiVO₄ and 2.75–2.82 eV of bulk g-C₃N₄, respectively. The photocatalytic activity of the composites was investigated by the decomposition of Acid Orange 7 (AO7) and nitrous oxide. In case of AO7, the BiVO₄/g-C₃N₄ (1:3) composite was the most active one and the main role in the reaction was played by photoinduced holes forming hydroxyl radicals. At the decomposition of N₂O, the most important species were the photoinduced electrons and the BiVO₄/g-C₃N₄ (1:1) composite was the most active photocatalyst.

Keywords $BiVO_4 \cdot G-C_3N_4 \cdot Composites \cdot Heterojunction \cdot Photocatalysis$

1 Introduction

Graphitic carbon nitride $(g-C_3N_4)$ has attracted attention of many researchers from all over the world as a metal-free polymeric semiconducting material. Although, $g-C_3N_4$ was already synthetized by Liebig in 1834 and its properties were studied later, e.g. [1, 2], an immense interest of this material has started after a paper published by Wang et al. [3] about

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a photocatalytic water splitting under visible light. $G-C_3N_4$ has diamond-like properties, such as thermal, chemical and photochemical stability, due to tri-s-triazine (heptazine) (C_6N_7) -based building blocks and strong covalent bonds between carbon and nitrogen atoms [2]. $G-C_3N_4$ has been synthetized by the thermal condensation of nitrogen-rich precursors, such as cyanamide, dicyandiamide, melamine and so forth [4]. An interesting feature of $g-C_3N_4$ is its band gap energy of 2.7 eV which makes it possible working under visible light [5–7]. However, its photocatalytic applications are limited by a fast recombination of photoinduced electrons and holes. It can be overcome by doping of $g-C_3N_4$ structures with metals and non-metals [7, 8] and coupling with metal [9] and semiconductor nanoparticles forming heterojunction photocatalysts [6, 10–15].

Recently, we studied some composites of $g-C_3N_4$ with semiconductor nanoparticles, such as TiO₂ [16, 17], WO₃ [18, 19] and SnO₂ [20]. The composites of $g-C_3N_4$ with BiVO₄ particles were studied by several authors [21–24]. BiVO₄ exists in monoclinic, tetragonal and orthorhombic modifications. Monoclinic BiVO₄ is able to absorb visible light due to the band gap of 2.4 eV in contrast to tetragonal one with the band gap of 2.9 eV [25]. $BiVO_4$ has been investigated due to its low toxicity, photocatalytic [26, 27] and coloristic properties [28, 29]. $BiVO_4$ can be prepared by several methods, such as a hydrothermal [26, 28–30], mechanochemical [25], ultrasonic [31] and surfactant-assisted precipitation [32, 33] and an ultrasonic spray pyrolysis [34]. Graphitic carbon nitride as well as its heterostructures with other semiconductors has been tested for the decomposition of risky environmental pollutants in gas and liquid phase.

Nitrous oxide is one of compounds depleting the stratospheric ozone layer and also strongly contributing to the greenhouse effect. The most important sources of N_2O are agriculture (fertilized soils), fossil fuels combustion, biomass burning, chemical industry, the production of nitric and adipic acids, and all combustion processes including emissions from car engines. The direct decomposition of N_2O to nitrogen and oxygen could be a suitable for its removal but from a thermodynamic point of view it is spin-forbidden with a high activation energy and feasible at a temperature about 1000 K [35]. The photocatalytic decomposition of N_2O using appropriate photocatalysts is an alternative way which can be performed at ambient temperatures and atmospheric pressure [36].

Besides gaseous pollutants such as N_2O there is a significant amount of organic pollutants present in waters, for instance, biocides, pharmaceuticals, dyes etc. [37]. Photocatalysis is also capable to oxidatively decompose them due to high oxidation potentials of photoinduced holes. The photocatalytic decomposition of synthetic dyes, which are an important source of water contamination coming mainly from textile industry, has been intensively studied [38].

The aim of this paper was to synthetize some composites of BiVO₄ and g-C₃N₄ and to examine their physico-chemical properties including the photocatalytic activity. They are able to absorb visible light and, therefore, have been investigated in various technical disciplines. In this work, BiVO₄/g-C₃N₄ composites were obtained by calcination of the mixtures of both pure components at 300 °C and resulting materials were characterized by common instrumental methods. Their photocatalytic properties were tested by the decomposition of azo-dye Acid Orange 7 [39] and nitrous oxide.

2 Experimental

2.1 Chemicals

All the chemicals used were of analytical reagent grade. Melamine was purchased from Sigma-Aldrich (USA), NH_4VO_3 , citric and nitric acids and 1,4-benzoquinone (1,4-BQ) were purchased from Lach-Ner (Czech Republic), $Bi(NO_3)_3 5 H_2O$ was purchased from Penta (Czech Republic), triethanolamine (TEOA) and tertial butanol (t-BuOH) from Lachema (Czech Republic). Azo-dye Acide Orange 7 was purchased from Synthesia (Czech Republic).

Water deionized by reverse osmosis (Aqua Osmotic, Czech Republic) was used for preparation of all solutions.

2.2 Preparation of g-C₃N₄

Melamine was heated in a ceramic crucible air in a muffle furnace. The heating temperature was gradually elevated with the heating rate of 15 °C min⁻¹ at the final temperature of 620 °C and kept for 1 h [4]. The resulting product was ground into powder and store in a closed vessel at a laboratory temperature.

2.3 Preparation of BiVO₄

BiVO₄ was prepared by a sol–gel method [40]: Bi(NO₃)₃ 5 H_2O was dissolved in 2 mol l⁻¹ HNO₃ and precipitated with NH₄VO₃ in the presence of citric acid. The resulting BiVO₄ was dried at 110 °C overnight and consequently heated at 500 °C for 5 h in a muffle furnace. The resulting product was ground into powder stored in a closed vessel at the laboratory temperature.

2.4 Preparation of BiVO₄/g-C₃N₄ Composites

The powders of BiVO₄ and g-C₃N₄ were mixed in mass portions of 1:1 and 1:3 and calcined at 300 °C for 4 h [40]. Together with the composites both pure components were heated in the same manner. The calcination products were ground into powder and stored in closed vessels at the laboratory temperature. The mass loss caused by heating of the materials was checked by their weighting after cooling; a weight loss of 0.06–0.70% was found. The composites were labelled as BiVO₄/g-C₃N₄ (1:1) and BiVO₄/g-C₃N₄ (1:3) according to the portion of both components. The content of Bi in the composites was determined by an atomic absorption spectrometer ContraAA 700 (Analytik Jena AG, Germany).

2.5 X-Ray Diffraction

The phase composition and microstructural properties of the prepared materials were determined using X-ray powder diffraction (XRD) technique. XRD patterns were obtained using a Rigaku SmartLab diffractometer (Rigaku, Japan) with a detector D/teX Ultra 250. A source of X-ray irradiation was a Co tube (CoK α , $\lambda_1 = 0.178892$ nm, $\lambda_2 = 0.179278$ nm) operated at 40 kV and 40 mA. Incident and diffracted beam optics were equipped with 5° Soller slits; incident slits were set up to irradiate area of a sample 10 × 10 mm (automatic divergence slits) constantly. Slits on a diffracted beam were set up to a fixed value of 8 and 14 mm. The samples rotated (30 rpm) during measurements to eliminate preferred orientation effect. The XRD patterns were collected in a 2θ range of 5° – 90° with a step size of 0.01° and speed 0.5°/min. The XRD patterns were evaluated using a PDXL 2 software (version 2.4.2.0) and compared with a database PDF-2, release 2015 and were analysed using the LeBail method to refine the lattice parameters of bismuth vanadate. Background of the XRD patterns was determined using the B-Spline function. Peak shapes were modelled with the pseudo-Voigt function accounting for a peak asymmetry due to axial divergence.

The crystallite size *L* was calculated according to Scherrer's equation for broadening $B(2\theta)$ (in radians) at a half maximum intensity (FWHM) of a diffraction peak as

$$B(2\Theta) = \frac{K\lambda}{L\cos\Theta}$$
(1)

where λ is the wavelength of X-rays, θ is Bragg's angle and *K* is the constant equal to 0.94 for cube or 0.89 for spherical crystallites. In this study *K* was rounded to 0.9.

2.6 Fourier Transform Infrared Spectrometry

Fourier transform infrared (FTIR) spectroscopy was performed with a FT-IR spectrometer Nexus 470 (ThermoScientific, USA) equipped with a DTGS detector using the KBr pellet technique. The spectra were recorded in the region of $4000-400 \text{ cm}^{-1}$ with a resolution of 4 cm⁻¹ (64 scans; the Happ-Genzel apodization).

2.7 Raman Spectrometry

Raman spectra were recorded using a dispersive Raman spectrometer DXR SmartRaman (ThermoScientific, USA) with a CCD detector. A 180° degree sampling was used as a measurement technique of the Raman spectroscopy. The measurement parameters were performed in the spectral region of 1800–50 cm⁻¹ with a excitation laser 780 nm, a grating of 400 lines mm⁻¹ and a aperture of 50 μ m (exposure time 1 s, the number of exposures was 1000). The spectra were treated with fluorescence correction (6th order).

2.8 XPS Spectrometry

X-ray photoelectron spectra were recorded using a Phoibos 100 (SPECS) hemispherical analyser operated in the FAT mode. The spectral line AlK_{α} of SPEXS XR50 was used to study a surface chemistry. High-resolution spectra for pass energy of 10 eV were utilized for quantification, and survey spectra were recorded at 40 eV for qualitative analyses. The sample composition was calculated from selected high-resolution spectra and corresponding RSF from the CasaXPS

library. The C–C component of the C 1s peak was set to a binding energy of 284.5 eV.

2.9 UV–Vis Diffuse Reflectance Spectrometry

UV–Vis DRS spectra were recorded with a Shimadzu UV-2600 (IRS-2600Plus) spectrophotometer at the room temperature in the range of 220–1400 nm. Reflectance was re-calculated to absorbance using Schuster-Kubelka-Munk's equation as follows

$$F(R_{\infty}) = \frac{(1 - R_{\infty})^2}{2R_{\infty}}$$
(2)

where R_{∞} is the diffuse reflectance from a semi-infinite layer. These DRS spectra were transformed to the dependencies of $(F(R_{\infty})h\nu)^2$ on $h\nu$ in order for us to obtain band gap energies of the prepared materials.

2.10 Photoluminescence Spectroscopy

Photoluminescence spectra were recorded by a spectrometer FLSP920 Series (Edinburgh Instruments, Ltd.) in the range from 390 to 640 nm. The spectrometer was equipped with a Xe900 arc non ozone lamp 450 W (Steady State Lamp) and a R928P type detector (PMT detector). The excitation wavelength of 325 nm was used for all measurements; a width of excitation and emission slits was 0.3 nm.

2.11 Physisorption of Nitrogen

Specific surface area (SSA) of the samples was determined using the N₂ adsorption isotherms recorded at 77 K and calculated by means of the Brunauer, Emmett and Teller (BET) method using a 3Flex (Micromeritics Instruments, USA) apparatus after sample degassing at 150 °C for 24 h under less than 1 Pa vacuum. Adsorption–desorption isotherms were evaluated according to the BET theory for the p/p^0 range of 0.05–0.25.

2.12 Transmission Electron Microscopy

Transmission electron microscopy was performed with a JEOL 2100 microscope with LaB6 electron gun and an X-ray energy-dispersive spectrometer (EDS) as a detector. The accelerating voltage of 200 kV was applied. Micrographs were taken by a camera Tengra (EMSIS). Before analysis the samples were dispersed in ethanol and sonicated for 5 min. One drop of this solution was placed on a copper grid with a holey carbon film and dried at the room temperature.

2.13 Photoelectrochemical Measurement

Photoelectrochemical measurements were performed using a photoelectric spectrometer equipped with a 150 W xenon lamp, monochromator and potentiostat (Instytut Fotonowy, Poland) [41].

2.14 Photocatalytic Decomposition of AO7

Photocatalytic activity of the composites and pure $BiVO_4$ and $g-C_3N_4$ was tested using the azo-dye AO7. The photocatalytic decomposition was performed under irradiation of a Narva Black light blue lamp with the maximum intensity at 368 nm (36 W).

The photocatalytic procedure was described recently [42]. The 0.05 g of the powder sample was mixed with 150 ml of the AO7 solution (150 ml of the AO7 stock solution of 3×10^{-5} mol 1^{-1}). The prepared suspension was stirred (400 rpm) for 30 min in dark and after 30 min an UV light was switched on. From the start of irradiation the suspension was sampled in periodic intervals. The suspension was sampled with a 5 ml syringe (2 ml of the suspension was taken) and filtered (syringe filter with 0.45 mm pore size). 1 ml of the filtered solution was diluted with demineralized water to 10 ml and absorbance was measured for this dilute solution. The AO7 absorbances were measured using a UV–VIS spectrometer Specord 250 Plus with software WinASPECT PLUS (version 4.1.0.0) in 1 cm quartz glass cuvettes at the absorption maximum of 483 nm.

2.15 Photocatalytic Decomposition of N₂O

The N₂O photocatalytic decomposition was conducted in a stainless steel batch photoreactor with an 8 W Hg lamp (λ =365 nm) as a source of the irradiation. A photoreactor with the photocatalyst inside was purged with a N₂O/He mixture (N₂O concentration was 1050 ppm). The photocatalytic procedure was described recently [41]. The reproducibility of N₂O photocatalytic decomposition was checked by repeated tests in the interval of 0–22 h.

3 Results and Discussion

The structure, texture, optical and photocatalytic properties of the $BiVO_4/g-C_3N_4$ composites as well as pure $BiVO_4$ and $g-C_3N_4$ were studied by several instrumental methods. The structure was studied by X-ray diffraction, FTIR and Raman spectroscopy. The texture was investigated by TEM and the measurements of SSA. The photocatalytic properties were studied by means of the decomposition of AO7 and nitrous oxide. The observed results were also compared with measured photocurrents.

3.1 X-Ray Diffraction Analysis

The XRD patterns are shown in Fig. 1. Monoclinic BiVO₄ was the dominant phase in all the composites (ICDD PDF card No. 00-014-0688): the space group I2/a and lattice constants a = 0.51950 nm, b = 1.17010 nm, c = 0.50920 nm, a



= $\gamma = 90^\circ$, $\beta = 90.38^\circ$. It was obtained from a precipitated tetragonal one [43, 44] and by heating at 500 °C.

The presence of g- C_3N_4 was confirmed by several weak and broad reflections, but the most typical are those around 14.9° and 32.2° 2theta. The first reflection at 14.9° 2theta (d=0.695 nm) is indexed as the (100) plane and is attributed to in-plane tri-s-triazine units forming one-dimension melon strands. The second reflection at 32.2° 2theta (d=0.322 nm) is interpreted as a long range interplanar stacking of aromatic systems with the (002) plane.

Refined lattice parameters and crystallite sizes are shown in Table 1. From the collected data, it was determined that the lattice parameters a, b and c of BiVO₄ were changed with the increasing content of $g-C_3N_4$. They slightly increased compared to pure BiVO₄ and BiVO₄ crystal matrix of the composites slightly expanded. It can be explained by the formation of a heterojunction between $g-C_3N_4$ and BiVO₄ structures. The crystallite sizes of BiVO₄ were calculated using Scherrer's formula as an average of three strongest diffractions at the (011), (121) and (040) planes. The other strong diffractions of the (042), (161) and (200) planes were also indicated in the XRD patterns [21, 26]. It is obvious that the crystallite sizes of $BiVO_4$ in the composites were significantly lower than that in pure $BiVO_4$ likely due to disintegration of large agglomerates into smaller particles by their connection with g-C₃N₄ ones. One of such agglomerates was examined by TEM as given below.

3.2 FTIR Spectrometry

The BiVO₄/g-C₃N₄ composites were studied by FTIR spectrometry as shown in Fig. 2. The spectra of g-C₃N₄ contained the three broad spectral bands at 3251 cm⁻¹, 3158 cm⁻¹, and 3082 cm⁻¹, which can be attributed to the stretching vibrations of N–H bonds. Other significant bands can be observed at 1636 cm⁻¹, 1574 cm⁻¹, 1461 cm⁻¹, 1411 cm⁻¹, 1321 cm⁻¹, 1241 cm⁻¹ and 812 cm⁻¹, which are typical for g-C₃N₄, e.g. [4, 45–47]. The strong bands between 1636 cm⁻¹ and 1241 cm⁻¹ correspond to the C=N and C–N bonds of heterocyclic rings. A medium band at 812 cm⁻¹ can be attributed to the breathing mode of triazine units. Some bands typical for BiVO₄ were recorded at 737 cm⁻¹ and 826 cm⁻¹ which can be assigned to the antisymmetric stretching vibrations of V-O-V units [25, 26, 48].

Table 1 Structural properties of $BiVO_4$, $g-C_3N_4$ and their composites	Material	L (nm), g-C ₃ N ₄	L (nm), BiVO ₄	a (nm)	b (nm)	c (nm)
	BiVO ₄	-	240	0.51941	1.17016	0.50933
	g-C ₃ N ₄	9.6	_	-	-	-
	BiVO ₄ /g-C ₃ N ₄ (1:1)	12.0	192	0.51961	1.17064	0.50957
	BiVO ₄ /g-C ₃ N ₄ (1:3)	10.2	116	0.51971	1.17086	0.50968



Wavenumber (cm⁻¹)

3.3 Raman Spectrometry

The structure of BiVO4/g-C3N4 composites was also studied by the Raman spectrometry, see Fig. 3. The Raman spectra of BiVO₄ show a strong band at 815 cm⁻¹ with the shoulders at 706 cm⁻¹, 341 cm⁻¹ and 210 cm⁻¹. The band at 815 cm⁻¹ can be assigned to the antisymmetric stretching of bridging of VOV units [28, 48] as already mentioned above. The shoulders were explained by the stretching modes of two different types of VO bonds [49]. The band at 341 cm⁻¹, the doublet at 328 cm⁻¹ and 364 cm⁻¹ and the band at 201 cm⁻¹ indicate the bending vibrations of VO₃ units and the stretching ones of BiO bonds, respectively [27, 48, 49]. Some bands corresponding to g-C₃N₄ were observed at 1230 cm⁻¹, 768 cm⁻¹, 706 cm⁻¹, 473 and 487 cm⁻¹ (a double band) and 212 cm⁻¹. The band at 1230 cm⁻¹ was ascribed to the $=C(sp^2)$ bending vibrations [50]. The bands at 768 cm⁻¹ and 706 cm⁻¹ could be ascribed to the ring breathing mode I and bending NCN and stretching CN vibrations, respectively. The bands at 473 cm⁻¹ and 212 cm⁻¹ correspond to the ring breathing mode II and stretching CN vibrations, respectively [51].

3.4 XPS Analysis

The $BiVO_4/g$ - C_3N_4 composites showed typical C 1s and N 1s peaks demonstrated in Fig. 4. The C 1s spectrum shows mainly one carbon peak at the binding energy (BE) of



Fig. 4 High resolution XPS C 1s (left) and N 1s spectra of $g-C_3N_4$, BiVO₄ and BiVO₄/g-C₃N₄ (1:3) composite

287.4 eV corresponding to the C–N–C coordination with a small amount of C–C contamination at the BE of 284.5 eV (energy reference). In the N 1s spectrum (right Fig. 4) several binding energies can be found. The main peak shows the occurrence of C–N–C groups (397.8 eV) and tertiary nitrogen N–(C)₃ groups (399.2 eV). Deconvolution of the peaks also revealed a weak additional one at 400.5 eV which is indicative for amino groups. However, it is important to underline that the peak of tertiary amines is about four-times stronger than that of hydrogen bounds in amines. This proves a high degree of condensation beyond a linear melon structure which would possess no tertiary amines. The peak at 403.5 eV is attributed to charging effects. Integration of the C and N signals provided the atomic C/N ratio of 0.8 close to a theoretical one of 0.75 [52].

The BiVO₄/g-C₃N₄ composite (1:3) showed C 1s and N 1s peaks with the above given components and, additionally, there was a component at 285.7 eV representing C-O bonds in left Fig. 4 (the blue peak). Also, the presence of C=O bonds is probable but a corresponding peak was overlapped by the C-N-C one at 287.4 eV. Both C-O and C=O bonds confirm the formation of the heterojunction between BiVO₄ and g-C₃N₄. The N 1s peak was modified significantly. Although, all main components of g-C₃N₄ were confirmed there was a new dominant peak with the BE of 396.2 eV in which about 50% of N was involved, see the blue peak in right Fig. 5. This component should represent metal nitrides, such as bismuth and vanadium nitrides, in the composite. It indicates the formation of the heterojunction as well. In the literature, samples containing V and N are referred to show a N 1s main peak at 397.5 eV, a V 2p3/2 peak at 513.5 eV and an O 1s one at 531 eV [53]. The observed O 1s peak at 529.8 eV was supposed to indicate VO_x systems [54]. The N 1s peak at 396.3 eV and the peak of V 2p3/2 at 512.3 eV were also referred in the literature [53]. The presence of C and N in pure BiVO₄ was negligible with the small C-C component at 284.5 eV in the C 1s spectrum (energy reference). There was the main O 1s peak at 529.8 eV what is typical for OV [54] and also OBi bonds [55]. A small contribution of –OH groups at the BE of 533 eV was observed. These components were present in the $BiVO_4/g-C_3N_4$ composite as well.

The vanadium XPS spectra displayed in Fig. 5 are widely discussed in the literature [54, 56]. In pure BiVO₄ a wide peak of V 2p was observed and well fitted by a system of components V 2p3/2(IV) at 515.8 eV, V 2p3/2(V) at 517.2 eV, V 2p1/2(IV) at 523.2 eV and V 2p1/2(V) at 524.5 eV (spin split at 7.33 eV) proposed in [56]. Therefore, there is a mixture of vanadium IV and V oxidation states in the ratio of V(V)/V(IV)=2. Since an achromatic X-ray source was used the fitting model also includes a main satellite of O 1s separated by 8.4 eV with intensity of 9.2% typical for achromatic sources.

The examined BiVO₄/g-C₃N₄ composites contained these components again but also the additional peaks observed at 519.4 eV and 526.6 eV are spin split at 7.2 eV as mentioned above. Those components cannot be vanadium nitrides [53], however, some similar components were observed in VO_x thin films after intensive argon ions sputtering [54]. They could be related to some defect of the VO_x structures. The spin split for vanadium oxides at 7.33 eV and at 7.5 eV for pure metal V were reported [54].

Bismuth in the BiVO₄ sample exhibited some core level peaks located at the BE of 159.0 eV and 164.3 eV which can be attributed to Bi 4f7/2 and Bi 4f5/2 orbitals, respectively, of a bismuth center in Bi₂O₃ [53]. A small broadening toward the lower BE was observed at 157 eV and 163.5 eV, which can be explained by the presence of metallic Bi–Bi bonds in amount of about 12% as a result of some defects in Bi–O–Bi structures of bismuth oxide. The BiVO₄/g-C₃N₄ composite exhibited a significant broadening of Bi 4f peaks toward the higher BE likely due to remaining Bi(NO₃)₃



Fig. 5 High resolution XPS O 1s and V 2p spectra (left) and Bi 4f spectra of g-C₃N₄, BiVO₄ and BiVO₄/g-C₃N₄ (1:3)

impurities [55]. Therefore, it is highly probable a N 1s component presented in $BiVO_4/g-C_3N_4$ was related to the $Bi(NO_3)_3$ impurities as well.

3.5 TEM Analysis and SSA Measurements

The BiVO₄/g-C₃N₄ composites were studied by TEM as well. The micrographs of pure BiVO₄ and g-C₃N₄ are shown in Fig. 6a, b. Two micrographs and a EDS spectrum of BiVO₄/g-C₃N₄ (1:3) are shown in Fig. 6c–e, respectively. Large BiVO₄ particles with size of several hundred nanometres were observed. The TEM micrographs also show some bulk and exfoliated structures of g-C₃N₄. The EDS spectrum of BiVO₄/g-C₃N₄ confirmed the presence of both components.

SSA was measured by physisorption of nitrogen at 77 K and evaluated according to the BET isotherm. The SSA of BiVO₄ and g-C₃N₄ was determined at 6 m² g⁻¹ and 35 m² g⁻¹ and the SSA of BiVO₄/g-C₃N₄ (1:1) and BiVO₄/g-C₃N₄ (1:3) was determined at 23 m² g⁻¹ and 28 m² g⁻¹, respectively, as summarized in Table 2. It is obvious that by adding BiVO₄ to g-C₃N₄ the SSA of 35 m² g⁻¹ of g-C₃N₄ was higher than that of bulk g-C₃N₄ and lower than that of exfoliated one [4, 57]. This means that g-C₃N₄ was partly exfoliated which was also observed by the TEM.

The real content of $BiVO_4$ in the prepared samples was determined at 100, 44 and 21 wt% for $BiVO_4$, $BiVO_4/g-C_3N_4$ (1:1) and $BiVO_4/g-C_3N_4$ (1:3), respectively (Table 2).

3.6 UV–Vis Diffuse Reflectance Spectrometry

UV–Vis DRS spectra were recorded for us to find the optical band gap energies of the BiVO₄/g-C₃N₄ composites as shown in Fig. 7. The reflectance values decreased with the decreasing content of g-C₃N₄. The spectra of pure g-C₃N₄ were composed from two bands corresponding to bulk and exfoliated g-C₃N₄ during the composites calcination which is also visible in the TEM micrographs (Fig. 6b, d). The composite spectra also contained these two bands of g-C₃N₄ and one band of BiVO₄. The optical band gap energies were evaluated by means of commonly employed Tauc's plot [58].

$$\varepsilon h \nu = C (h \nu - E_o)^p \tag{3}$$

where ε is the molar extinction coefficient, $h\nu$ is the energy of incident photons, E_g is the optical band gap energy (hereinafter the band gap energy), C is the constant and p is the power depending on the type of electron transition: p=2and p = 1/2 are for direct and indirect semiconductors, respectively.

Based on the literature data, both $BiVO_4$ and $g-C_3N_4$ were evaluated supposing that to be direct semiconductors

[22, 26, 29, 40]. The E_g values of g-C₃N₄ were evaluated at 2.88 eV and 3.09 eV corresponding to bulk and exfoliated g-C₃N₄. All band gap energies are summarized in Table 2. In the composites, the E_g of bulk g-C₃N₄ was determined at 2.75 eV, 2.81 eV and 2.82 eV in accordance with the increasing content of g-C₃N₄. The E_g values related to exfoliated g-C₃N₄ varied in the narrow interval of 3.08–3.09 eV [4, 57] corresponding to electron transitions between highest occupied (HOMO) and lowest unoccupied (LUMO) molecular orbitals.

The E_g of BiVO₄ was determined at 2.51 eV in pure BiVO₄ and 2.46 eV and 2.43 eV in BiVO₄/g-C₃N₄ (1:1) and BiVO₄/g-C₃N₄ (1:3), respectively. These bad gap energies correspond to the transition of 6 s electrons of Bi to 3d orbitals of V in monoclinic BiVO₄. The light absorption of tetragonal BiVO₄ was due to the charge-transfer transitions between vanadium and oxygen in VO₄³⁻ and the corresponding E_g is about 2.9 eV [43, 44].

3.7 Photoluminescence Spectrometry

The photoluminescence spectra of g-C₃N₄, BiVO₄ and the BiVO₄/g-C₃N₄ composites were excited at the wavelength of 325 nm and are shown in Fig. 8. A broad band of g-C₃N₄ with a maximum at 470 nm can be explained by electron $\pi^* \rightarrow$ lone pair and $\pi^* \rightarrow \pi$ transitions [59]. The PL intensity decreased from pure g-C₃N₄ to pure BiVO₄ according to the decreasing portion of g-C₃N₄ in the composites. A partial splitting of one large band at 470 nm into two bands at 460 nm and 500 nm indicates the separation of photoinduced electron and holes via the heterojunction avoiding their recombination. The low PL intensity of BiVO₄ can be explained by the formation of structure defects working as electron traps as a result of the calcination. It was observed in the PL spectra of pure BiVO₄.

The PL spectra of pure $BiVO_4$ showed two bands at 450 and 500 nm, see Fig. S1 (Supplementary Materials). The first band can be ascribed to recombination of holes formed in the hybrid orbitals of Bi 6s and O 2p and electrons generated from V 3d orbitals [60]. The second one corresponds to recombination of electrons from the 3d orbitals of V with holes from the 6s orbitals of Bi as given above (Fig. 8).

3.8 Photoelectrochemical Measurement

The photocurrent generation at 365 nm was recorded for each material with longer times for opened and closed shutter. The lowest amounts of photoinduced electron and holes were detected in pure $BiVO_4$ and the composite containing the highest amount of $BiVO_4$ - $BiVO_4/g$ - C_3N_4 (1:1). The second composite and pure g- C_3N_4 showed the same photocurrent which was higher than that of pure $BiVO_4$.



Fig. 6 TEM micrographs of BiVO4 (a), g-C3N4 (b) and BiVO4/g-C3N4 (1:3) (c, d), EDS spectrum of BiVO4/g-C3N4 (1:3) (e)

Table 2Characteristics of $g-C_3N_4$, BiVO4 and theircomposites

Sample	SSA (m2 g-1)	Real content of BiVO ₄ (wt%)	E_g bulk g- C_3N_4 (eV)	E _g BiVO ₄ (eV)
BiVO ₄	6	100	_	2.51
g-C ₃ N ₄	35	0	2.88	-
BiVO ₄ /g-C ₃ N ₄ (1:1)	23	44	2.75	2.46
$BiVO_4/g-C_3N_4$ (1:3)	28	21	2.82	2.43

Band gap energy of exfoliated g-C₃N₄ was 3.08-3.09 eV



Fig.7 UV–Vis diffuse reflectance spectra of $g\text{-}C_3N_4,\ BiVO_4$ and $BiVO_4/g\text{-}C_3N_4$

3.9 BiVO₄/g-C₃N₄ Heterojunction Study

The band gap energies of $BiVO_4$ and $g-C_3N_4$ were used for the estimation of their band edge potentials according to empirical equations

$$E_{VB} = X - E^0 + 0.5E_g$$
(4)

$$E_{CB} = E_{VB} - E_g \tag{5}$$

where E_{VB} is the valence band (VB) edge potential, X is electronegativity of a semiconductor obtained as a geometric mean of constituent atoms, $E^0 = 4.5$ eV is the energy of free electrons vs vacuum, E_{CB} is the conduction band (CB) edge potential. The semiconductors' electronegativity X = 4.67 eV [24] for g-C₃N₄ and 6.04 eV for BiVO₄ [22, 29] were adopted from the literature. Taking into account the determined band gap energies of pure g-C₃N₄ and BiVO₄, their edge potentials were calculated as follows: $E_{VB} = 1.61$ eV and $E_{CB} = -1.27$ eV for g-C₃N₄ and $E_{VB} = 2.80$ eV and $E_{CB} = 0.29$ eV for BiVO₄; they are demonstrated in Fig. S2. The edge potentials were compared with the standard redox



Fig. 8 Photoluminescence spectra of $g-C_3N_4$, BiVO₄ and BiVO₄/ $g-C_3N_4$ composites

potentials of hydroxyl and superoxide radicals: $E^0(OH/$ H_2O = 2.74 V [61], $E^0(O_2/O_2^{\bullet-}) = -0.33$ V [61] and $E^{0}(OH-/OH \bullet) = 1.99 V [62]$. The redox potential of $E^{0}(O_{2}/$ $O_2^{\bullet-}$) = -0.16 V was also referred for the concentration of $1 \mod 1^{-1} \text{ of } O_2[61].$

3.10 Photocatalytic Activity

composites

3.10.1 Photocatalytic Decomposition of AO7

The photocatalytic activity of the composites and pure $g-C_3N_4$ and BiVO₄ was studied by means of the decomposition of AO7 under irradiation with the maximum intensity at 368 nm as shown in Fig. 9. Unlike the prepared materials (photocatalysts), AO7 absorbed the irradiation very little and, therefore, its sensitization was not supposed [63, 64]. The photocatalytic activity of BiVO₄ was negligible and similar to a blank solution with no photocatalyst. The activity of the other materials increased in the order $BiVO_4/g-C_3N_4$ (1:1) < $g-C_3N_4$ < $BiVO_4/g-C_3N_4$ (1:3). These results are in accordance with those of photoelectrochemical measurement (Fig. 10). The material with the smallest photocurrent response (BiVO₄) proved the smallest activity in the photooxidation reaction (the photocatalytic decomposition of AO7).

The very low activity of BiVO4 was also observed by other authors [21, 22, 24, 30, 65] and was explained by particles morphology and size [30] and their negative surface charge [65]. In this case, a very likely reason is recombination of photoinduced electrons and holes in BiVO₄ defects



g-C,N,

 $BiVO_{4}/g-C_{3}N_{4}(1:1)$

0.020

Fig. 10 Time based photocurrent response plots of $g-C_3N_4$, BiVO₄ and BiVO₄/g-C₃N₄ composites

[44] formed during the calcination which was documented by its low photoluminescence as already discussed. The heterojunction between $BiVO_4$ and $g-C_3N_4$ was supposed to facilitate the separation of electrons and holes and, thus, to increase the photocatalytic activity [21-24, 40].

Based on the calculated edge potentials of $g-C_3N_4$ and BiVO₄ the formation of hydroxyl radicals was assumed by the reactions of holes and electrons as follows



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 $BiVO_4/g-C_3N_4$ (1:3)

400

BiVO,

$$h^+ + H_2 O \rightarrow OH^{\bullet} + H^+ \tag{6}$$

$$h^+ + OH^- \to OH^{\bullet} \tag{7}$$

$$e^- + O_2 \to O_2^{\bullet -} \tag{8}$$

$$O_2^{\bullet-} + H^+ \leftrightarrows HO_2^{\bullet} (pK_a = 4.8)$$
⁽⁹⁾

$$2 \operatorname{HO}_2^{\bullet} \to \operatorname{H}_2\operatorname{O}_2 + \operatorname{O}_2 \tag{10}$$

$$H_2O_2 + e^- \to OH^{\bullet} + OH^- \tag{11}$$

Since BiVO₄ was photocatalytically inactive, only g-C₃N₄ and BiVO₄/g-C₃N₄ (1:3) were examined for us to confirm the formation of superoxide and hydroxyl radicals and holes using some scavengers. For this purpose, the scavengers 1,4-BQ for superoxide radicals, t-BuOH for hydroxyl radicals and TEOA for holes were employed. The photocatalytic decomposition of AO7 in the presence of BiVO₄/g-C₃N₄ (1:3) and the scavengers is shown in Fig. 11. In all cases, the scavengers reduced (t-BuOH) or nearly inhibited (1,4-BQ and TEOA) the decomposition of AO7. It confirmed the formation of the radicals and holes according to Eqs. (6–11).

The CB of BiVO₄ is more positive than the standard redox potentials of the $O_2/O_2^{\bullet-}$ couple, therefore, from a view of thermodynamics electrons in this CB could not form the $O_2^{\bullet-}$ radicals [24]. The VB of BiVO₄ is more positive than the standard redox potentials of the OH•/H₂O and OH⁻/OH• couples, thus, the holes in BiVO₄ were able to form the OH• radicals. However, the negligible photocatalytic activity and low luminescence of BiVO₄ were already explained by its defect structure.

Since the $O_2^{\bullet-}$ radical were formed in the presence of BiVO₄/g-C₃N₄ (1:3) a direct Z-scheme heterojunction [66] was only possible to be created during the composites



Fig.11 Photocatalytic decomposition of AO7 in the presence of $BiVO_4/g\text{-}C_3N_4$ (1:3) and scavengers

synthesis. Afterward, electrons from the CB of $BiVO_4$ recombined with holes in the VB of $g-C_3N_4$ and remaining charge carriers were able to take part in the photocatalytic reactions. The Z-scheme heterojunction in $BiVO_4/g-C_3N_4$ photocatalysts was also reported by Tian et al. [24]. On conditions that a type-II heterojunction between the conduction and valence bands of $g-C_3N_4$ and $BiVO_4$ was formed the redox potentials of both electrons and holes would be too low to take part in the reactions Eqs. (6–11).

3.11 Photocatalytic Decomposition of N₂O

The effect of irradiation time on the N₂O photocatalytic decomposition was probed in the presence of both the BiVO₄/g-C₃N₄ composites and pure BiVO₄ and g-C₃N₄ for the time period of 0–22 h. The time dependences of the N₂O conversion for photocatalysis and photolysis were depicted in Fig. 12. It is evident the conversion of nitrous oxide was increasing over the time. The photolysis led only to the 6% conversion of N₂O after 22 h of irradiation and each prepared material was found to be more effective. The highest N₂O conversion was observed in the presence of BiVO₄/g-C₃N₄ (1:3). On the other hand, pure BiVO₄ demonstrated the smallest photoactivity similarly as at the decomposition of AO7.

3.12 Comparison of Photocatalytic Reactions

As already mentioned, the photocatalytic activity of the prepared composites was investigated in two different reactions: (i) the photocatalytic decomposition of AO7 and (ii)



Fig. 12 The time dependence of N₂O conversion over prepared photocatalysts under UVA irradiation (λ =365 nm)

the photocatalytic decomposition of N_2O . The photocatalytic decomposition of AO7, where a main role was played by photoinduced holes, can be marked as a photooxidation reaction. The first step of this photooxidation is the reaction of holes with water and the production of hydroxyl radicals which are one of the most powerful oxidizing agents. The mechanism and photoactivity of that reaction was described above.

In case of the N_2O photocatalytic decomposition, the most important species are photoinduced electrons. The reaction is started by electrons according the equation (Eq. 12) [67]:

$$N_2O + e^- \rightarrow N_2 + O^-_{ads} \tag{12}$$

The adsorbed oxygen on the photocatalyst's surface is oxidized by holes (Eq. 13) as

$$2O_{ads}^{-} + 2h^{+} \rightarrow 2O_{2} \tag{13}$$

In general, the photocatalytic activity can be influenced by light-absorption ability, surface properties and photoinduced charge separation efficiency. The SSA of the prepared materials is summarized in Table 1 and the dependence between the SSA and the N_2O conversion is demonstrated in Fig. 13. These findings indicate that the SSA was not the crucial factor that determined the photocatalytic activity of the materials.

The PL spectra (Fig. 8) served for the investigation of separation efficiency of electron-hole pairs. The weaker PL intensities indicate the better separation of electrons and holes which can lead to their higher photocatalytic activity. As already mentioned, the decreasing PL intensity of the



Fig. 13 Dependence between the photocatalytic activity of the N_2O decomposition and SSA for different photocatalysts

composites can be explained by the lower content of $g-C_3N_4$ and also by supressing the recombination of photoinduced charge *via* the heterojunction between $g-C_3N_4$ with BiVO₄. In addition, the wide band splitting at 470 nm supports idea of the electron–holes separation. The smallest intensity of PL was observed at BiVO₄/g-C₃N₄ (1:1) which is in agreement with its highest activity of the photocatalytic decomposition of N₂O. The similar findings were observed by other authors [22, 68, 69].

It follows from the previous results that the Z-scheme heterojunction of BiVO₄/g-C₃N₄, which affected their properties, such as light absorption, crystal structure, specific surface area and charge carrier mobility, was successfully created. The formation of V⁴⁺ in the composites could increase mobility of electrons and holes. Furthermore, the V⁴⁺ produced oxygen vacancy acted as the centre of photocatalytic sites and chemisorbed species (Eq. 13) resulting in better photoelectrochemical properties. That hypothesis was confirmed by Safei et al. [70] who dealt with the photoelectrochemical performance of Z-scheme BiVO₄/g-C₃N₄ photocatalysts and supported with the experimental results of Density Functional Theory (DFT) simulations. The photoinduced electrons from the CB of g-C₃N₄ directly reduced nitrous oxide (Eq. 12) while the holes from the VB of $BiVO_4$ oxidized adsorbed oxygen coming from the decomposition of N_2O (Eq. 13). At the same time the electrons from the CB of BiVO₄ were recombining with the holes from the VB of $g-C_3N_4$.

4 Conclusion

The composites of BiVO₄/g-C₃N₄ were prepared by the calcination of mixed monoclinic BiVO₄ and bulk g-C₃N₄ at 300 °C for 4 h. During this process bulk g-C₃N₄ was partly exfoliated. The presence of both components was confirmed by the FTIR and Raman spectroscopy and TEM (EDS). The specific surface area was determined using the physisorption of nitrogen (the BET method) at 23–28 m² g⁻¹ of the composites and 6 m² g⁻¹ and 35 m² g⁻¹ of pure BiVO₄ and g-C₃N₄, respectively.

The band gap energies were determined by Tauc's method at 2.42–2.46 eV of BiVO₄ and 2.75–2.82 eV and 3.08–3.06 of bulk and exfoliated g-C₃N₄, respectively. It means that both BiVO₄ and bulk g-C₃N₄ were able to absorb the visible irradiation. The band gap energies of the composites shifted due to the heterojunction between BiVO₄ and g-C₃N₄. It was confirmed by the XPS which identified the presence of C–O and C=O bonds and metal (bismuth and vanadium) nitrides in the composites. The existence of the heterojunction was also confirmed by the PL spectroscopy.

The photocatalytic activity of the composites was investigated by the decomposition of AO7 and nitrous oxide. The BiVO₄/g-C₃N₄ (1:3) composite was the most active one for the decomposition of AO7. The experiments with the scavengers 1,4-BQ, t-BuOH and TEOA proved that the AO7 decomposition was mainly performed by holes and following hydroxyl radicals which were formed only due to the direct Z-scheme heterojunction between both components. The different situation was observed in case of N₂O when the BiVO₄/g-C₃N₄ (1:1) composite was the most active photocatalyst and photoinduced electrons were the most important species.

The formation of the Z-scheme heterojunction between $BiVO_4$ and $g-C_3N_4$ creates a promising area for further research, especially in the field of environmental technologies. This study demonstrates suitability of the $BiVO_4/g-C_3N_4$ composites for the photocatalytic decomposition of risky environmental pollutants, such as the azo-dye AO7 and nitrous oxide. However, there are a lot of various harmful organic and inorganic compounds in the environment which could be removed by the photocatalytic decomposition as well. It will be the aim of our next research.

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