

The Effect of the Temperature–Time Mode of Crystallization on the Morphology and Thermal Properties of Nanocomposites Based on Polypropylene and Magnetite (Fe₃O₄)

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Abstract

In the present study, the influence of the temperature–time mode of crystallization (TTC) on the morphology and thermal properties of PP/Fe₃O₄ nanocomposite materials was investigated. The morphology of the nanocomposites prepared in different TTC mode was studied by atomic force microscope. AFM study shows that the root mean square roughness of samples is 90–95, 50, 21 nm for PP/Fe₃O₄@20, PP/Fe₃O₄@200 and PP/Fe₃O₄@20000 respectively. Thermo gravimetric analysis was employed to investigate the thermal stability of PP/Fe₃O₄ nanocomposite samples (PP/Fe₃O₄@200) is higher than the thermal stability of samples obtained with other two modes. Crystallization and melting behaviors of nanocomposite samples prepared in different TTC mode have been studied with DSC method and the degree of crystallinity of samples was calculated. It was found that, degree of crystalization decreases with increasing of cooling rate. The XRD patterns of samples produced in different TTC modes also correlate well with this result.

Keywords Polypropylene · Magnetite · Thermal properties

1 Introduction

In polymeric based composites, the transition from microsized fillers to nano-sized ones significantly changes a number of operational and technological properties associated with local chemical interactions, including: curing speed, polymer chain mobility, polymer chain deformability, structure ordering (degree of crystallization of the polymer matrix) [1]. Nanocomposites differ from conventional composite materials due to the much more developed surface area of the filler particles. The properties of such materials

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depend substantially on their morphology as well as the structure and properties of the interface between the matrix and the filler [2, 3].

Nanocomposite materials can be used in all aspects of information technology, such as transducer material for obtaining information, chip packaging materials and circuit boards in information processing, magnetic materials for information storage, composite fiber, sheath tube, antenna reflector panels for information transmission, and mechanical structural materials in information implementation [4]. Thermophysical properties are of great importance for determining the practical value of polymer materials and these properties depend on the crystal structure of the polymer. Regularity of the arrangement of macromolecules around filler particles plays main role for formation of different structures of polymer. Depending on the conditions of crystallization, concentration, chemical structure and molecular weight of the polymer, the molecules can fold in different crystallographic directions. As a result of such treatment, deep structural transformations take place in them, which significantly affect the properties of the resulting materials. The orientation of macromolecules also has a significant effect on the thermal properties of thermoplastics. Thermal

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methods for the analysis of polymer composites are in the list of the most effective methods for studying phase and other physicochemical transformations occurring between individual compounds under the influence of heat [5].

In the present work, we studied the influence of the temperature–time mode of crystallization (TTC) on the structures and thermal properties of polymer based $PP + Fe_3O_4$ nanocomposites.

2 Experimental Part

2.1 Materials

Isotactic polypropylene (PP-MoplenHF500N, Homopolymer) with the density of 0.92 g/cm³ at 25 °C was used in this study [Mw=250,000, Mn=67,000, Melt Mass-Flow Rate (MFR)=11.5 g/10 min (230 °C, 2.16 kg), melting T=162 °C].

Magnetite nanoparticles were obtained by co-precipitation in an alkaline medium. The average nanoparticle size is 7–15 nm [6].

2.2 Composite Fabrication

Composites were prepared by mixing in solution and hot pressing. Polypropylene was dissolved in toluene by stirring the mixture for about 30 min on magnetic stirrer at the temperature of 120 °C [6, 7]. The filler was added into the solution of polymer and was mixed during 2 h until the homogeneous system was obtained. Then solvent was evaporated and in the final step nanocomposites films were prepared by hot pressing method [8]. The processing temperature and time were160 °C and 5 min correspondingly while consolidation pressure was 10 MPa in the heat press. The molds were then cooled at three modes: cooling in liquid nitrogen (cooling rate 20,000 °C/min)-(PP/Fe₃O₄@2000); cooling in water (cooling rate 200 °C/min)-(PP/Fe₃O₄@2000) and the slow cooling under press for 24 h (cooling rate 20 °C/min)-(PP/Fe₃O₄@20).

2.3 Research Methods of Polymer Nanocomposites

2.3.1 AFM Analysis

The morphology of the nanocomposites was studied using atomic force microscopy Integra Prima (NT-MDT, Zelenograd). For the scan special silicon cantilevers fabricated by plasma etching method with the needle with the radius of curvature of 20 nm and the resonance frequency of 1-5 Hz were used. Scansize was 2×2 mm. The measurements were performed in the semicontact microscopy mode in air and needle change of the cantilever oscillation amplitude was fixed, determining the surface topography. The scanning speed and the number of scanned lines of the image were 256 and 1969 Hz respectively.

2.3.2 XRD

X-ray diffraction analysis was performed on Rigaku Mini Flex 600 XRD diffractometer at ambient temperature. In all the cases, Cu K α radiation from a Cu X-ray tube (run at 15 mA and 30 kV) was used. The samples were scanned in the range of angles 2 θ of 20°–70°.

2.3.3 IR Study

IR study IR spectra of the samples were recorded on a spectrometer FT-IR Varian-3600 Excalibur Series, allowing to record the spectra in the range of $4000-400 \text{ cm}^{-1}$.

2.3.4 Thermogravimetric Analysis (TGA)

Thermogravimetric analysis of samples was conducted in a thermogravimetric analyzer (TGA) Model Seiko Exstar 6000 TG/DTA 6300. Nanocomposites samples were heated from 30 to 650 °C with a heating rate of 100 °C/min in a nitrogen atmosphere.

2.3.5 Differential Scanning Calorimetric Analysis (DSC)

Differential scanning calorimetric analysis of nanocomposites was performed on nanocomposites by using DSC 6100 (Seiko Instruments Japan) model of Differential Scanning Calorimeter (DSC). Samples were placed into aluminum sample pans and experiments were carried out under nitrogen atmosphere with a purge rate of 20 ml/min. Samples were heated from 20 to 250 °C then cooled to 25 °C.

3 Results and Discussion

Figure 1 shows the AFM 2D images obtained for nanocomposites PP + Fe₃O₄ prepared in different temperature–time modes of crystallization. Figure 2 shows histogram of rootmean-square roughness of surface of PP + Fe₃O₄ nanocomposites. The root mean square roughness of samples are 90–95, 50, 21 nm for PP/Fe₃O₄@20, PP/Fe₃O₄@200 and PP/Fe₃O₄@20,000 respectively. It can be concluded from the AFM images that, quenching in liquid nitrogen (PP/Fe₃O₄@20,000) formed smaller crystallites than in the case of other two modes. During slow cooling of alloy, the polymer macromolecules find ways of creating a regular and large supramolecular structure around nanoparticles and this is consistent with literature. With the increasing of the cooling rate of alloy, the polymer macromolecules are without



Fig. 1 AFM surface images of PP + Fe_3O_4 nanocomposites prepared in different TTC. a PP/ $Fe_3O_4@20,000$ b PP/ $Fe_3O_4@200$ c PP/ $Fe_3O_4@200$



Fig.2 Histogram of surface heights of PP+Fe₃O₄ nanocomposites prepared in different TTC. **a** PP/Fe₃O₄@20,000 **b** PP/Fe₃O₄@200 **c** PP/Fe₃O₄@20

the possibility of to form a regular supramolecular structure and ultimately the structure elements on the surface of the nanocomposite become smaller.

Thermal behaviour of PP + Fe₃O₄ nanocomposites obtained through different temperature–time modes of crystallization (TTC) was investigated by two methods [9]. Figure 3 shows TGA curves of the PP + Fe₃O₄ nanocomposites produced in different TTC modes. As it can be seen from the figure, the decomposition of PP + Fe₃O₄ nanocomposites in all three mode occurs at one-stage [10, 11].The initial stage of thermal degradation of PP/Fe₃O₄@20,000, PP/Fe₃O₄@200 and PP/Fe₃O₄@20 samples started at 241.1, 290.1, 227.3 °C respectively. The results of TGA analysis of the samples are given in Table 1.

It was found that thermal stability of water-cooled nanocomposite samples (PP/Fe₃O₄@200) is higher than the thermal stability of samples obtained using other two modes. This is explained by the establishment of an optimal regular structure during the formation of the water-cooled nanocomposite samples. The semi-crystallinepolymers, depending on the cooling rate of the polymer melt, can exhibit two kinds of structures: amorphous and crystalline. During the cooling of nanocomposite in different TTC modes amorphous phase is formed around initial nucleus. Due to the high cooling rate, the polymeric macromolecule can not reach the most thermodynamically stable state during quenching in liquid nitrogen. In other words, around the structural centers, shortrange order is formed. In contrast to this, at the slow-cooling mode there is enough time for formation of crystallites (PP/ Fe₃O₄@20) and thus they have long-range-order. In this case, the structure obtained in water cooling mode can be assumed thermodynamically more favorable conditions for semicrystalline polymer around iron oxide nanoparticles which play the role of heat transfer centers.

DSC curves of PP + Fe_3O_4 nanocomposites obtained through different TTC modes are shown in Fig. 4.

Defining the heat of fusion, the degree of crystallinity of the $PP + Fe_3O_4$ polymer nanocomposites can be calculated with following relationship:

degree crystallinity(%) =
$$\frac{H}{H_0} \times 100\%$$
 (1)

H is the melting heat, determined by the area of peak corresponding to the melting of the polymer, H_0 is the heat released at the melting of 100% crystalline phase (for polypropylene, it is $H_0 = 207$ mJ/mg). The results of DSC analysis of the PP + Fe₃O₄ nanocomposites obtained through different temperature–time modes of crystallization are given in Table 2.



Fig. 3 TGA curves of PP + Fe₃O₄ nanocomposites obtained through different TTC. a PP/Fe3O4@20 b PP/Fe3O4@200 c PP/Fe3O4@20,000

Table 1TGA data of $PP + Fe_3O_4$ nanocompositesobtained through different TTC	Sample	Initial decomposition tem- perature (°C)	Half decomposition tem- perature (°C)	Final decomposi- tion temperature (°C)
	PP/Fe ₃ O ₄ @20	227.3	457.36	487.29
	PP/Fe ₃ O ₄ @200	290.1	457.33	522.94
	PP/Fe ₃ O ₄ @20,000	241.1	455.87	484.25

According to the Table 2, the calculated degree of crystallinity is 51.69% for PP/Fe₃O₄@20 samples, 47.82% for PP/Fe₃O₄@200 samples and 44.15% for PP/Fe₃O₄@20,000. As already noted above, at the high cooling rate (PP/ Fe₃O₄@20,000) forming crystallites have a short-range order, but in slow-cooling mode (PP/Fe₃O₄@20) they comprise long- range-order, which in turn increases percentage of crystalline phase. In this respect PP/Fe₃O₄@200 samples which quenched in water, occupy an intermediate position. It also can be seen from degree of crystallinity of samples (Table 3).

The XRD patterns of PP/Fe₃O₄ nanocomposites obtained through different temperature–time modes of crystallization

are presented in Fig. 5. From the patterns of all three modes, a series of characteristic peaks (3 1 1), (4 0 0), (5 1 1), and (4 4 0), which are in well accordance with the inverse cubic spinel phase of Fe₃O₄ (= 35.4°, 43.12°, 56.98°, 62.74° magnetite), were observed [5]. Comparing obtained patterns, we can conclude that an increase in the rate of cooling of the molten materials leads to a decrease in the intensity and number of reflexes. So when the PP alloy is quenched in liquid nitrogen, reflections which are typically characteristic for smectic structures of PP are observed [12].

Figure 6 shows the IR spectra of PP and nanocomposites films based on PP + Fe_3O_4 . The assignment of bands are as shown below [13, 14].

Table 3 Assignment for absorption bands of polypropylene



Fig. 4 DSC curves of PP + Fe₃O₄ nanocomposites obtained through different TTC: a depending of crystallization on temperature b depending of melting on temperature. (1) PP/Fe₃O₄@20,000 (2) PP/Fe₃O₄@200 (3) PP/Fe₃O₄@20

Table 2DSC data of \overline{S} PP + Fe3O4 nanocompositesobtained through different TTC	Sample	Melting tempera- ture (K)	Crystallisation tem- perature (K)	Melting enthalpy (mJ/mg)	Degree of crystallinity (%)
	PP/Fe ₃ O ₄ @20	434.67	388.8	107	51.69
	PP/Fe ₃ O ₄ @200	433.9	387.9	99	47.82
	PP/Fe ₃ O ₄ @20,000	434.28	387.8	91.4	44.15

Frequency (cm ⁻¹)	Phase	Assignment
808	Crystalline	CH ₂ rocking, C–C chain stretching, C–CH stretching
899	Crystalline	CH ₃ rocking, CH ₂ rocking, CH bending
940	Crystalline	CH ₃ rocking, C–C chain stretching
972	Amorphous	CH ₃ rocking, C–C chain stretching
997	Crystalline	CH ₃ rocking, CH ₂ wagging, CH bending
1044	Crystalline	C-CH ₃ stretching, C-C chain stretching, CH bending
1102	Crystalline	C-C chain stretching, CH ₃ rocking, CH ₂ wagging
1167	Crystalline	C-C chain stretching, CH ₃ rocking, CH bending
1219	Crystalline	CH ₂ twisting, CH bending, C-C chain stretching
1255	Crystalline and amorphous	CH bending, CH ₂ twisting, CH ₃ rocking
1378	Crystalline and amorphous	CH ₃ symmetric bending, CH ₂ wagging

In contrast to pure PP, In the spectrum of $PP + Fe_3O_4$ based nanocomposites film produced by different temperature-time mode demonstrated two intense peaks, in 584 and 591 cm^{-1} bands, that are due to the stretching vibration mode associated to the metal-oxygen absorption band (Fe–O bonds in the crystalline lattice of Fe_3O_4) [15]. The absorption spectra of samples obtained through different temperature-time crystallization mode differ from each another only by intensity. This is explained by the fact that the samples taken from various modes differs from each other only by the supramolecular structure.



Fig. 5 XRD patterns of PP+Fe₃O₄ nanocomposites obtained through different TTC. (1) PP/Fe₃O₄@20,000 (2) PP/Fe₃O₄@200 (3) PP/Fe₃O₄@20

4 Conclusions

The AFM analysis of PP + Fe₃O₄ based nanocomposite samples produced by different TTC modes revealed that the root-mean-square roughness of the surface of samples which was cooled (PP/Fe₃O₄@2000) in liquid nitrogen is smaller than this of the samples obtained in the other two modes. This explains smaller size of the crystallites in the samples cooled in liquid nitrogen. At the same time, the results of the TGA analysis exhibited that the thermal stability of water-cooled nanocomposite samples $(PP/Fe_3O_4@200)$ is higher than the thermal stability of nanocomposite samples produced in other two regimens. The results of DSC analysis of samples produced through application of various TTC modes show that, crystallinity decreases with increasing the cooling rate of alloy. The XRD patterns of samples produced in defferent TTC modes are also in accordance with this result. Analytical results show that, increase of cooling time of samples causes the ratio of crystalline phase in nanocomposite materials based on polycrystalline polymer, such as isotactice polypropylene to increase. During formation of supramolecular structure of the polymer at the different cooling rates, the structure of polymer and polymer/filler interaction cause a veriosity of the thermal conductivity and, accordingly, the thermal stability of the materials. The higher thermal stability of samples cooled in water $(PP/Fe_3O_4@200)$ can be explained by this fact. IR spectra of samples produced by different temperature-time mode shows that, this nanocomposite samples differ each other only by supramolecular structure.



Fig. 6 IR-spectra of of PP+Fe₃O₄ nanocomposites obtained through different TTC. (1) Pure PP (2) PP/Fe₃O₄@20 (3)PP/Fe₃O₄@200 (4) PP/Fe₃O₄@20,000

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