

Three-Dimensional Ordered CeO₂ Hollow Spheres (3DOHSs-CeO₂) from Polymethylmethacrylate/CeO₂ Core/Shell Microsphere Colloidal Crystals

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Abstract The composite microspheres containing polymethylmethacrylate (PMMA) cores and ceria (CeO₂) nanoparticles shells were synthesized by in situ chemical precipitation based on electrostatic interaction. The PMMA/CeO₂ colloidal crystal templates were self-assembled by a modified two-substrate vertical deposition method. After removal of the polymer cores by calcination, the three-dimensional ordered CeO₂ hollow spheres (3DOHSs-CeO₂) films were fabricated. The resulting 3DOHSs-CeO₂ materials were characterized by X-ray diffraction, field emission scanning electron microscopy, high resolution transmission electron microscopy, and N₂ adsorption-desorption analysis. Results indicate that the obtained samples presented a typical face-centered cubic structure, and the CeO2 hollow spheres (HSs) formed shared necks with each other. The average CeO₂ HSs diameters and center-to-center distances were 230-240 nm, and the porous shells (20-30 nm in thickness) of CeO₂ HSs were consisted of tiny CeO₂ nanoparticles with a size of 10-20 nm. This work suggested that the 3DOHSs architectures would be valuable for photonic crystals, advanced optical and catalytic applications.

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1 Introduction

Ceria (CeO₂) materials with a cubic fluorite-type crystallographic symmetry has attracted much attention due to their potential applications in photocatalysts, fuel cells, heterogeneous catalysts, polishing abrasives, and UV absorbers [1–3]. Many efforts have been devoted to improving the performance of CeO₂ materials through structural and morphological control, including nano-particles [3, 4], nano-wires [5, 6], nano-rods [7], nano-tubes [8], nano-plates [9], nano-spheres [10], nano-cubes [2, 11], nano-films [12], nano-octahedrons [13, 14], and hollow spheres [15].

Three-dimensionally (3D) ordered inorganic materials have attracted great interests for numerous applications in photonic devices, catalysis, sensors, batteries, supercapacitors, fuel cells, sorption, and thermal insulation [16]. The conventional approach is the hard-templating process with colloidal crystals as the sacrificial templates. Up to date, 3D ordered CeO₂ materials are mainly focused on 3D ordered macroporous (3DOM) products or inverse opals due to their bicontinuous nanostructure (a continuous nanostructured framework with large interfacial area and a 3D interconnected pore system). Waterhouse et al. [17] fabricated inverse opal ceria films and powders, exhibiting 3DOM structures and a photonic band gap in the visible region using a colloidal crystal template approach. The obtained 3DOM-CeO₂ powders showed an improved thermal stability compared to non-networked CeO₂ nanoparticles of comparable initial crystallite size and surface area. Liu et al. [18] also prepared 3DOM-

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CeO₂ materials with hierarchical pore structure via a dual hard-soft templating strategy. Compared to conventional bulk CeO₂ materials, the hierarchical 3DOM-CeO₂ supported iridium catalysts exhibited an enhanced catalytic activity in CO oxidation.

Colloidal crystal templates method can also be applied in fabrication of 3D ordered inorganic hollow spheres (3DOHSs). Marlow et al. [19] demonstrated a method to coat a uniform vinyl-SiO₂ shell on negatively charged polystyrene particles. The core-shell hybrids were selfassembled into colloidal crystals using a capillary deposition method, and then converted into 3DOHSs-SiO₂ films by calcination. Dinh et al. [20] reported a novel Au/TiO₂ nanostructured photocatalyst, which was constructed by the 3D ordered assembly of thin-shell Au/TiO₂ hollow nanospheres. The novel 3DOHSs-Au/TiO₂ photocatalysts exhibited an improved photocatalytic activity for the photocatalytic decomposition of isopropanol under visiblelight illumination by comparison with than conventional Au/TiO₂ nanopowders. In our previous work [21], highquality 3D ordered TiO₂ hollow-spheres films were fabricated using polystyrene spheres colloidal crystals as sacrificial templates and titanium tetrachloride as titanium sources. To the best of our knowledge, there are few reports on the fabrication of 3DOHSs-CeO₂ materials using colloidal crystal templates.

Herein, we synthesized well-defined core-shell structured polymethylmethacrylate (PMMA)/CeO₂ composites, and PMMA/CeO₂ microspheres colloidal crystals were obtained via a modified two-substrate vertical deposition method [22, 23]. After removal of PMMA cores by calcination, 3DOHSs-CeO₂ materials with well-ordered periobtained. odicity were А detailed structural characterization of the as-prepared samples indicated the CeO₂ hollow spheres (HSs) exhibited a face-centered cubic packing structure, and the CeO2 HSs were connected with shared necks.

2 Experimental Section

2.1 Materials

Methyl methacrylate (MMA) was obtained from Shanghai Lingfeng Chemical Reagent Co. (China) and purified with an inhibitor removal column. Ammonia (25–28 wt%), poly(sodium-*p*-styrene-sulfonate) (PSS), sodium chloride (NaCl), potassium persulfate (KPS), absolute ethanol, hexamethylenetetramine (HMT), and cerium nitrate hexahydrate (Ce(NO₃)₃·6H₂O) were purchased from Shanghai Chemical Reagent Co. (China) and used as received.

2.2 Preparation of PMMA/CeO₂ Composite Microspheres

Monodisperse PMMA spheres were synthesized by a modified soap-free emulsion polymerization [24]. Typically, 8 g of MMA and 150 g of deionized (DI) water were charged into a 250 mL round bottom flask equipped with a Graham condenser, magnetic stirrer, and thermometer with a temperature controller. The mixture was heated to 70 °C at a stirring rate of 250 rpm for 2 h, and then injected by the mixture of 0.2 g KPS and 30 g DI water to initiate polymerization. The polymerization was continued at the same temperature for another 7 h.

For the preparation of PMMA/CeO₂ microspheres, 4 mL PMMA colloids, Ce(NO₃)₃·6H₂O (1.5 g) and HMT (2.4 g) were charged into 200 mL DI water. The mixture was stirred for 10 min to form homogeneous suspension. Subsequently, the mixture was reacted at 75 °C for 2 h. These operations were performed under constant magnetic stirring at 200 rpm. The resulting precipitates were centrifuged, washed with DI water and absolute ethanol, and dried at 80 °C.

2.3 Surface Modification of PMMA/CeO₂ Composites with PSS

According to the literature [25], NaCl (1.7 g) and PSS (0.2 g) were charged into the 40 mL PMMA/CeO₂ suspensions (3 wt%), the mixture was stirred at room temperature for 2 h. Subsequently, the particles were collected by centrifugation and washed with DI water three times to ensure removal of residual polyelectrolyte and salt. Finally, the PSS-stabilized PMMA/CeO₂ composites were obtained, and dispersed in DI water for self-assembly.

2.4 Fabrication of PMMA/CeO₂ Colloidal Crystals and 3DOHSs-CeO₂

The PMMA/CeO₂ colloidal crystals were obtained by a modified two-substrate vertical deposition [26] through electrostatic colloid stabilization by adsorption of PSS on the surface of PMMA/CeO₂ composites. Oxidized silicon wafers and glass slides $(10 \times 25 \text{ mm}^2)$ were boiled in the piranha solution (3:1 v/v 98 % H₂SO₄/35 % H₂O₂) at 60 °C for 1 h to create a clean hydrophilic surface. A piece of pretreated oxidized silicon wafer and cover glass were vertically placed together and then dipped into the PMMA/CeO₂ composites suspension (3.0 wt%). In the process of the evaporation of the solvent, the microspheres self-assembled to form ordered colloidal crystals under induction of the capillary force. Subsequently, the cover glass

was peeled off carefully after the colloidal crystals were dried completely. Finally, the colloidal crystals on the oxidized silicon wafer were calcinated at 500 °C (1 °C/ min) for 2 h to remove the PMMA templates, subsequently formed 3D ordered CeO₂ hollow spheres.

2.5 Characterization

The crystal phases of the products were measured on a Rigaku D/Max 2500 PC X-ray diffractometer (XRD). The morphologies and composition of the samples were characterized using a JSM-6360LA scanning electron microscope (SEM), Hitachi S-4800 field emission SEM [equipped with an energy-dispersive X-ray spectrometer (EDS)], and a JEOL-2100 high-resolution transmission electron microscope [TEM, coupled with a selected area electron diffractometer (SAED)]. The surface area and the pore size of samples were determined using the Brunauer–Emmett–Teller (BET) model and the Barrett–Joyner–Halenda (BJH) method, respectively, derived from N_2 adsorption–desorption measurements performed on an automatic micropore physisorption analyzer (Micromeritics ASAP 2020).

3 Results and Discussion

The morphologies and composition of the as-prepared PMMA and PMMA/CeO₂ particles are displayed in Fig. 1. As shown in Fig. 1a, the monodisperse PMMA colloids exhibit a uniform size of 250 nm with a deviation <5 %. After coating, the PMMA/CeO₂ samples (Fig. 1b) are large-scale uniform spheres similar to the PMMA cores. Furthermore, the surfaces of the composites are much rough, indicating that the CeO_2 particles have been coated on the surfaces of PMMA cores. According to the EDS spectrum (inset in Fig. 1b), the surfaces of the composites consist of Ce, O and C elements. As confirmed by the TEM image of the composites (Fig. 1c), CeO_2 nanoparticles are uniformly immobilized on PMMA colloids. The strawberry-like PMMA/CeO₂ microspheres exhibit a well-defined core-shell structure. A possible mechanism for the formation of PMMA/CeO₂ microspheres is described in our previous work [27].

The XRD pattern of the 3DOHSs-CeO₂ products (Fig. 2) is indexed to the CeO₂ fluorite structure (space group = Fm3m, JCPDS card #34-0394) [2]. The characteristic XRD peaks located at different 2θ values matched well with the corresponding *hkl* planes of bulk CeO₂. No additional impurity was found in the pattern, implying the high purity of the product. The average grain size determined by the Scherrer formula of CeO₂ powders was ca. 9.7 nm.

Figure 3 shows the representative FESEM images of the $3DOHSs-CeO_2$ samples. The top-view image (Fig. 3a)



Fig. 1 SEM (**a**, **b**), TEM (**c**) images and EDS spectrum (*inset*) of the obtained PMMA colloids and PMMA/CeO₂ composites

indicates that the products present a typical face-centered cubic packing structure with (111) plane parallel to the substrate. The average sphere diameters and center-to-center distances were 230–240 nm, which was determined by the high-magnification FESEM image (Fig. 3b). From the cross-section image (Fig. 3c), some holes could be clearly observed on the surfaces, indicating the hollow structure of CeO_2 spheres. And the films over ten layers of



Fig. 2 XRD pattern of the 3DOHSs-CeO₂ samples

CeO₂ hollow sphere arrays are formed. Furthermore, these hollow spheres are connected with shared necks, and the highly 3D ordered structure of CeO₂ hollow spheres are obtained after removing the PMMA cores of the hybrids colloidal crystals by calcination at 500 °C.

The structural details of the 3DOHSs-CeO₂ samples were further investigated by HRTEM and SAED analyses, and the representative results are illustrated in Fig. 4. The low-magnified TEM image (Fig. 4a) also confirms that the samples are consisted of closely packed, hexagonal array of CeO₂ hollow spheres. Furthermore, the SAED pattern of as-synthesized samples (inset in Fig. 4a) reveals the pure crystalline nature of as-synthesized particles with cubic fluorite structure, where (111), (200), (220) and (311) lattice planes were clearly indexed [28]. As confirmed by the high-magnified TEM image (Fig. 4b), the walls of CeO₂ hollow spheres are composed of small nanoparticles (10-20 nm), and the shell thickness is 20-30 nm. The CeO₂ HSs exhibited loose state and even some interparticle voids could be clearly observed. According to the HRTEM analysis, the measured interplanar distances of 0.31 nm achieved from the circled region in Fig. 4c are in correspondence with the lattice spacing of the (111) plane of CeO₂ [29].

Figure 5 shows the N_2 adsorption-desorption isotherms and the corresponding pore-size distribution curves (inset) of the 3DOHSs-CeO₂ products. The samples present isotherms of type IV, indicating the presence of mesopores (2–50 nm) [30]. The shapes of hysteresis loops are type H3 at a high relative pressure range of 0.8–1.0, indicating the presence of slit-like pores. The isotherms show high absorption at high relative pressure (P/P₀) range (approaching 1.0), implying the formation of large mesopores and macropores [31]. The specific



Fig. 3 SEM images and EDS spectrum (*inset*) of the 3DOHSs-CeO₂ samples: **a**, **b** *top view* image, **c** cross-section image

surface area determined by BET model is 84.61 m^2/g . The pore size distribution curve (inset in Fig. 5) calculated from the desorption branch based on the BJH method is a wide range of 2–10 nm, further confirming the presence of mesopores. It is noted that more macroporous



Fig. 4 TEM (a, b), HRTEM (c) images and SAED pattern (*inset*) of the 3DOHSs-CeO₂ samples

information (larger than 100 nm) cannot be directly obtained by N_2 adsorption-desorption analyses [32]. However, the hollow structures are clearly observed by cross-sectional FESEM image of the sample (Fig. 3c). Therefore, the results of FESEM, HRTEM and N_2 adsorption-desorption analysis reveal that the as-prepared 3DOHSs-CeO₂ materials possess a hierarchically multimodal pore-size distribution from mesopores to macro pores.



Fig. 5 Nitrogen adsorption-desorption isotherms and pore size distribution (*inset*) of the as-synthesized 3DOHSs-CeO₂ samples

4 Conclusions

PMMA polymer cores were uniformly coated with CeO_2 nanoparticles, and well-defined core-shell structured PMMA/CeO₂ composite particles were obtained. The PMMA/CeO₂ composites colloidal crystals were fabricated through a modified two-substrate vertical deposition method. The three-dimensional ordered CeO₂ hollow spheres (3DOHSs-CeO₂) were obtained by calcination to remove PMMA cores. Colloidal crystal templating is a simple and efficient method for the fabrication of the films composed of a closely packed, hexagonal array of the CeO₂ HSs without complicated apparatuses and procedures. These films will be important for promissing applications in photonic crystals, advanced optical and catalytic materials. This method may provide an alternative to the fabrication of other high-quality 3D ordered inorganic HSs.

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