



# Synthesis of Bi<sub>2</sub>S<sub>3</sub>-sensitized TiO<sub>2</sub> photoanodes for enhanced photovoltaic performance in solar cell application

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**Received:** 4 March 2024

**Accepted:** 9 July 2024

**Published online:**  
18 July 2024

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## ABSTRACT

The synthesis of Bi<sub>2</sub>S<sub>3</sub> nanoparticles for sensitizing TiO<sub>2</sub> photoanodes were synthesized through a cost-effective and straightforward approach using modified chemical bath deposition (M-CBD) or successive ionic atomic layer adsorption reactions (SILAR) at room temperature. Initially, a TiO<sub>2</sub> seed layer was synthesized at room temperature via the chemical bath deposition method, followed by deposition of a mesoporous TiO<sub>2</sub> layer using the doctor blade method. This study investigated the influence of the number of SILAR cycles and the choice of counter electrodes on the performance of Bi<sub>2</sub>S<sub>3</sub>/TiO<sub>2</sub>-based photoelectrodes. Characterization of the prepared Bi<sub>2</sub>S<sub>3</sub>/TiO<sub>2</sub> photoanode involved various techniques, including X-ray diffraction, UV–Vis spectroscopy, scanning electron microscopy, and Raman spectroscopy, enabling the analysis of its structural, optical, and morphological properties. The Bi<sub>2</sub>S<sub>3</sub>/TiO<sub>2</sub>-based cell exhibits a maximum conversion efficiency of 0.8%, demonstrating the potential of this combination for photovoltaic applications. This study contributes to the field of solar cell technology by presenting a novel approach for sensitizing TiO<sub>2</sub> photoanodes with Bi<sub>2</sub>S<sub>3</sub> nanoparticles, offering insights into the optimization of fabrication parameters and performance enhancement strategies for future device design and development.

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## 1 Introduction

The utilization of conventional energy resources leads to environmental pollution and energy crises, and overcoming the utilization of renewable and clean energy resources is an alternate. The utilization of solar energy helps reduce the energy crisis. The utilization of solar energy is essential because it is both clean and eco-friendly. Solar cell devices can convert light energy into electrical energy. Semiconductor-sensitized solar cells possess many advantages, such as a high absorption coefficient, constant power output, tunable energy bandgap, and generation of multiple electron–hole pairs with high-energy excitons [1]. Metal-oxide semiconductors are wide bandgap materials with large surface areas for quantum dot absorption. These semiconductors were deposited on transparent conducting oxides, such as indium-doped tin oxide and fluorine-doped tin oxide, with thicknesses of 10–13  $\mu\text{m}$ . Metal oxides such as titanium oxide ( $\text{TiO}_2$ ) [2], zinc oxide ( $\text{ZnO}$ ) [3, 4], indium oxide ( $\text{In}_2\text{O}_3$ ) [5, 6], niobium oxide ( $\text{Nb}_2\text{O}_5$ ) [7], tungsten oxide ( $\text{WO}_3$ ) [8], and tin oxide ( $\text{SnO}_2$ ) [9, 10] have been widely studied for their fabrication.  $\text{TiO}_2$  has been extensively studied owing to its potential optical and electronic properties, which increase the adsorption of quantum dots and enhance photoconversion efficiency. However, the hopping mechanism limits the efficient charge transfer from  $\text{TiO}_2$ .  $\text{TiO}_2$ , widely studied metal oxide, is known for its unique optoelectronic properties and wide bandgap (3.2 eV) structure, and is classified into Rutile, Anatase, and Brookite crystal structures. Many narrow bandgap materials such as CdSe, PbS, CdS,  $\text{Sb}_2\text{S}_3$ , and  $\text{Bi}_2\text{S}_3$  have been used as sensitizers [11–16].  $\text{Bi}_2\text{S}_3$  was mainly in the form of bismuthinite. The color of  $\text{Bi}_2\text{S}_3$  is lead gray to brown, and the melting point and density of  $\text{Bi}_2\text{S}_3$  are 850  $^\circ\text{C}$  and 6.780  $\text{gm}/\text{cm}^3$ , respectively.  $\text{Bi}_2\text{S}_3$  is a direct bandgap material that belongs to the V–VI group. The bandgap of  $\text{Bi}_2\text{S}_3$  is suitable for light harvesting; it is less toxic than PbS- and CdS-based sensitizers, the dispersibility of  $\text{Bi}_2\text{S}_3$  is good, and either powder or thin films can be synthesized. Various techniques are available for the synthesis of  $\text{Bi}_2\text{S}_3$  such as electrochemical deposition, [17] chemical bath deposition, [18] and successive ionic layer adsorption and reaction (SILAR) [19–21].

Among these synthetic methods, SILAR/M-CBD is one of the most suitable and common, owing to its facile processing, versatile application to different surfaces and nanostructures, proper coating thickness,

and packing density.  $\text{Bi}_2\text{S}_3$  deposited using the SILAR method is either a thin film or QDs, and the bandgap of  $\text{Bi}_2\text{S}_3$  is 1.3 – 2.2 eV [22, 23]. It is near the near-infrared region and has a relatively large absorption coefficient; it has applications in solution-processed photodetectors, thermoelectric devices, photosensitizers, solar cells, and supercapacitors.  $\text{Bi}_2\text{S}_3$  is also an exciting candidate for use in polymer-nanoparticle hybrid solar cells. The objective of this study was to improve the performance of the  $\text{Bi}_2\text{S}_3$ -based solar cells. According to a literature review, the performance of  $\text{Bi}_2\text{S}_3$ -based solar cells does not exceed 2% [19]. To improve the performance of  $\text{Bi}_2\text{S}_3$ -based solar cells, a variety of research strategies have been used, such as surface passivation, co-sensitization, and optimization of the electrolyte and counter electrodes, and details of the strategy and comparative boost in the performance were mentioned in our previously published review article [19]. Here, we focus on an optimized number of SILAR cycles for loading  $\text{Bi}_2\text{S}_3$  on a mesoporous  $\text{TiO}_2$  layer and the impact of the counter electrode on solar cell performance.

The present work demonstrates the synthesis of  $\text{TiO}_2$  photoelectrodes for solar cell applications. The  $\text{Bi}_2\text{S}_3$  nanoparticles were synthesized using the SILAR method. Sandwiched  $\text{TiO}_2/\text{Bi}_2\text{S}_3$  solar cell was measured using a carbon counter electrode and polysulfide electrolyte under a solar simulator assembly. The effect of counter electrodes on the performance of solar cells is significant; therefore, optimization of the counter electrodes is an essential parameter. A good counter electrode has the following properties: 1. high catalytic activity, 2. Good conductivity, and 3. Stability [24, 25]. The solar cell performance was measured using carbon (C) and copper tin sulfide (CTS) counter electrodes. CTS is a promising candidate for replacing Pt counters in DSSC [26]. In this study, we used a SILAR-synthesized CTS/FTO thin film as the counter electrode for a  $\text{Bi}_2\text{S}_3/\text{TiO}_2$ -based solar cell. Synthesis of  $\text{TiO}_2$  compact film was done by chemical bath deposition [27]. Deposition of mesoporous  $\text{TiO}_2$  layers on the seed layer of the  $\text{TiO}_2$  film was done using the doctor blade method. For the sensitization of  $\text{TiO}_2$  films,  $\text{Bi}_2\text{S}_3$  was used as a sensitizer. Deposition of  $\text{Bi}_2\text{S}_3$  on  $\text{TiO}_2$  films was done using the SILAR method. A polysulfide electrolyte was used to measure IV performance. Fabrication of CTS counter electrodes was done using the SILAR method [28]. The direct bandgap material like CTS possesses a bandgap within the range of 1.45 eV and the absorption coefficient of CTS is high. CTS can

be synthesized using several methods such as Spray Pyrolysis, chemical bath deposition, SILAR, and sputtering. [29].

In this study, synthesis of the  $\text{Bi}_2\text{S}_3/\text{TiO}_2$  photoelectrodes using cost-effective and simple methods such as chemical bath deposition, doctor blades, and SILAR was performed. SILAR is one of the cheapest methods for the synthesis of metal chalcogenide materials at room temperature. Here, we synthesized  $\text{Bi}_2\text{S}_3$  nanocrystals for the sensitization of  $\text{TiO}_2$  by SILAR method at room temperature. The sensitizer loading on the  $\text{TiO}_2$  mesoporous layer was directly proportional to the number of SILAR cycles. Here, we used CTS as the counter electrode for JV characteristic measurements. As per a literature survey, the utilization of CTS as a counter for the JV measurement of  $\text{Bi}_2\text{S}_3/\text{TiO}_2$ -based photoelectrodes has not been reported. Here, the Conversion efficiency was higher than that of the carbon counter electrode. The present report also highlights the role of counter electrodes in the performance of  $\text{Bi}_2\text{S}_3$ -coated  $\text{TiO}_2$ -based solar cells.

Deposition methods such as SILAR and CBD are cost-effective and simple; however, these methods have limitations such as complexity in multilayer deposition, which is time-consuming and requires careful control of the reaction conditions. Choosing the appropriate method depends on cost and compatibility, and methods such as spray pyrolysis, spin coating, thermal evaporation, and hydrothermal methods are useful for material synthesis and film deposition. High-quality crystalline films were obtained using these methods.

## 2 Experimental details

### 2.1 Chemicals used

All the chemicals used in this study were without further purification and bought from SRL, Sigma-Aldrich, Thomas Baker, and High Purity Laboratory Chemicals HPLC.

### 2.2 Preparation of $\text{TiO}_2/\text{Bi}_2\text{S}_3$ electrode

This study used all the chemicals without further purification and bought from SRL, Sigma-Aldrich, Thomas Baker, and HPLC. The following steps were involved in the formation of the  $\text{TiO}_2/\text{Bi}_2\text{S}_3$  electrode:

#### 2.2.1 Deposition of $\text{TiO}_2$ compact layer

In this study, we used a  $\text{TiO}_2$  compact film to avoid back contact. The chemical bath deposition method is very cost-effective and does not require sophisticated instruments, and compact  $\text{TiO}_2$  is deposited on transparent conducting oxides such as (FTO). The substrates were cleaned via ultrasonication in absolute ethanol, and thoroughly cleaned with acetone. The seed layer is prepared by the chemical bath deposition method using 5 ml  $\text{TiCl}_3$ , 1 M NaOH, and 20 ml double-distilled water (DW). The prepared seed layer was annealed at 450 °C [27].

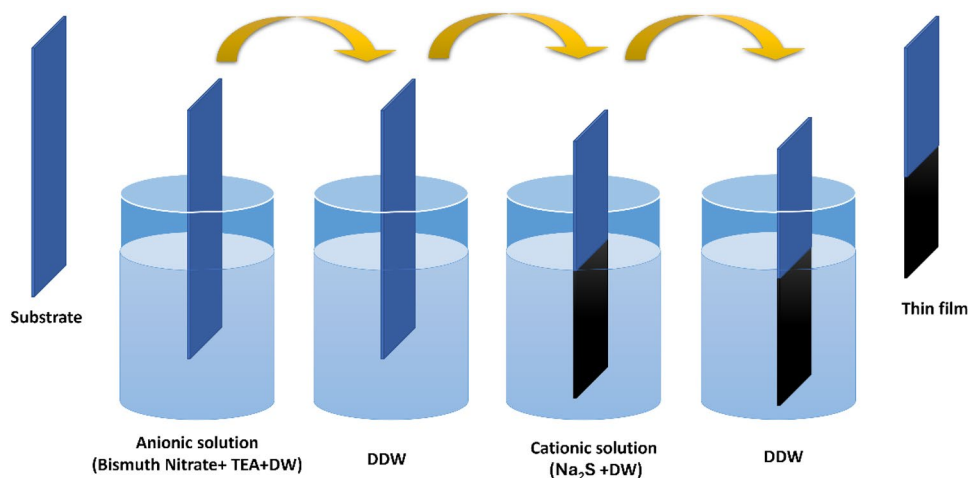
##### 2.2.1.1 Preparation of paste and deposition of mesoporous $\text{TiO}_2$ layer on FTO

The preparation of the paste is an important step in the fabrication of metal-oxide layers. Commercial  $\text{TiO}_2$  [P25] nanopowder (0.5 g), ethyl cellulose (0.30 g), terpineol (2.72 g), and Acetyl Acetone (0.5 ml) were used to prepare the  $\text{TiO}_2$  paste, all of which were of AR grade and were purchased from Loba Chemie. Pvt. Ltd. One of the advantages of using ethanol for the preparation of paste is the volatile nature of ethanol, which easily evaporates when subjected to a small amount of heat treatment in an incubator. The prepared  $\text{TiO}_2$  slurry was deposited on a transparent conductive oxide (FTO) substrate using the doctor blade method. The wet  $\text{TiO}_2$  films were dried in an incubator for 24 h. Subsequently, the dried mesoporous  $\text{TiO}_2$  layer was annealed at 450 °C. for 1 h in a muffle furnace [30].

#### 2.2.2 Sensitization of $\text{TiO}_2$ photoelectrode by using $\text{Bi}_2\text{S}_3$ nanoparticles

$\text{Bi}_2\text{S}_3$  nanoparticles sensitized  $\text{TiO}_2$  Films using the SILAR method; 0.003 M Bismuth Nitrate ( $\text{Bi}(\text{NO}_3)_2$ ) and 0.015 M Sodium Sulfide  $\text{Na}_2\text{S}$  were the sources of cationic and anionic precursors, respectively, in double-distilled water. The sensitized films were then annealed. The steps involved in the deposition of  $\text{Bi}_2\text{S}_3$  are shown in Fig. 1. Here, mesoporous  $\text{TiO}_2$  film is dipped in beaker 1 for 20 s after that, the film is rinsed in beaker 2. The third beaker contained an anion source film dipped for 20 s, and the same film was rinsed for 10 s in beaker 4, completing the one SILAR cycle. The same procedure was done 5-25

**Fig. 1** Steps involved in the sensitization of  $\text{TiO}_2$  films using  $\text{Bi}_2\text{S}_3$  nanoparticles by SILAR method



times with very fine and uniform deposition of  $\text{Bi}_2\text{S}_3$  occurring on the  $\text{TiO}_2$  film [31, 32].

### 2.3 Preparation of polysulphide electrolyte

Polysulfide electrolytes are preferred for semiconductor-sensitized solar cells. Polysulfides were prepared using  $\text{Na}_2\text{S}$  and sulfur powder in ethanol, methanol, and water.

### 2.4 Synthesis of CTS counter electrode on FTO

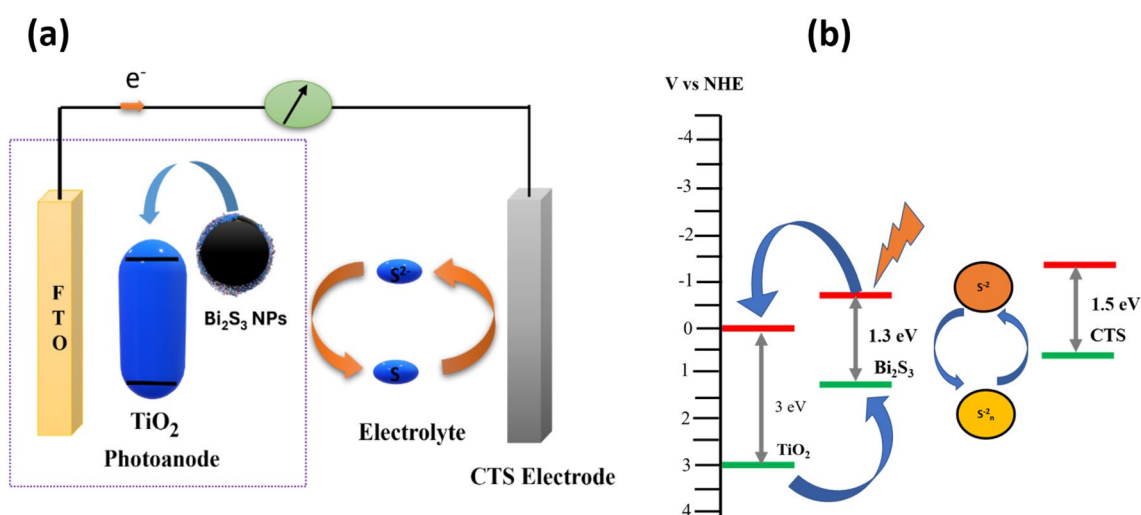
The counter electrode directly influences  $R_{\text{series}}$  and thus, the Fill Factor, which is why optimization and choosing an appropriate counter electrode is the most important part of the fabrication of solar cells [33].

A CTS counter electrode was synthesized using the SILAR method [28].

### 2.5 Device fabrication

The  $\text{Bi}_2\text{S}_3/\text{TiO}_2$  photoanode and CTS counter electrode were combined to form a sandwich solar cell spacer separating the two electrodes. Polysulfide electrolytes are used as redox mediators in photoanodes and counter electrodes. The performance of the  $\text{Bi}_2\text{S}_3/\text{TiO}_2$ -based solar cells was measured using a solar simulator assembly. A schematic representation of the  $\text{Bi}_2\text{S}_3/\text{TiO}_2$ -based solar cells is shown in Fig. 2a.

Figure 2b shows a Schematic Representation of the band alignment of the  $\text{Bi}_2\text{S}_3/\text{TiO}_2$  electrode: the conduction band potential of  $\text{Bi}_2\text{S}_3$  (0.7) is more negative



**Fig. 2** Schematic representation (a)  $\text{Bi}_2\text{S}_3/\text{TiO}_2$  solar cell (b) band alignment of  $\text{Bi}_2\text{S}_3/\text{TiO}_2$  solar cell [34]

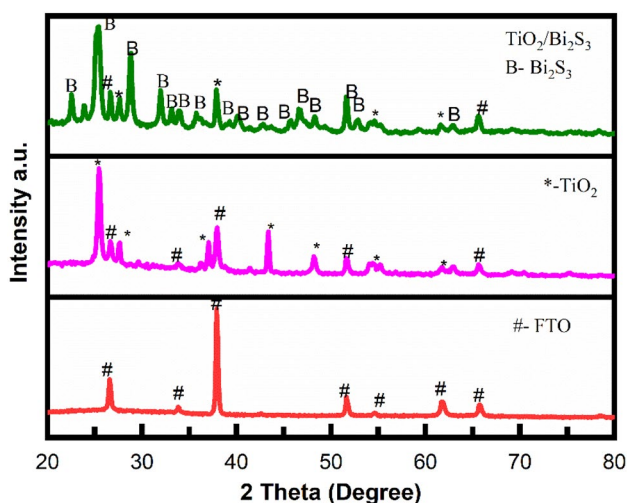


than that of  $\text{TiO}_2$ , and the valence band (VB) is more cathodic than the VBs of  $\text{TiO}_2$ ; therefore, photoexcited electrons from the conduction band of  $\text{Bi}_2\text{S}_3$  jump towards  $\text{TiO}_2$ , which is useful for enhancing the photovoltaic performance of  $\text{Bi}_2\text{S}_3/\text{TiO}_2$ -based solar cell [34].

### 3 Result and discussion

#### 3.1 Structural properties analysis (X-ray diffraction techniques)

Figure 3 shows the XRD patterns of FTO,  $\text{TiO}_2$ , and  $\text{Bi}_2\text{S}_3$ -sensitized  $\text{TiO}_2$  film using an X-ray diffractometer Bruker D8 with  $\text{CuK}\alpha$  radiation ( $1.54 \text{ \AA}$ ) in the range of  $20^\circ$ – $80^\circ$ . Peaks attributed to # are the peaks of FTO, XRD analysis of  $\text{TiO}_2$  film revealed that peaks corresponding to 25.5, 27.6, 37.8, 48.2, 53.9, 62.5 degrees are the peaks of  $\text{TiO}_2$  (Peaks of  $\text{TiO}_2$  were denoted by \*) having tetragonal anatase and rutile structure. Peaks are matched with JCPDS 21–1272. Average crystallite size was estimated to be 27 nm, calculated by Scherrer formula, Fig. 3 also shows XRD graphs of  $\text{Bi}_2\text{S}_3/\text{TiO}_2$  photoelectrode peaks of  $\text{Bi}_2\text{S}_3$  denoted by (B), and the observed characteristics peak at 22.4, 23.7, 28.7, 31.9, 33.05, 34.02, 35.7, 39.2, 40.23, 42.8, 45.6, 46.6, 51.06, and  $52.78^\circ$  are assigned to the (220), (101), (230), (221), (301), (330), (240), (041), (340), (421), (002), (431), (441), and (132) planes of orthorhombic  $\text{Bi}_2\text{S}_3$  (JCPDS No. 75–1306), it shows



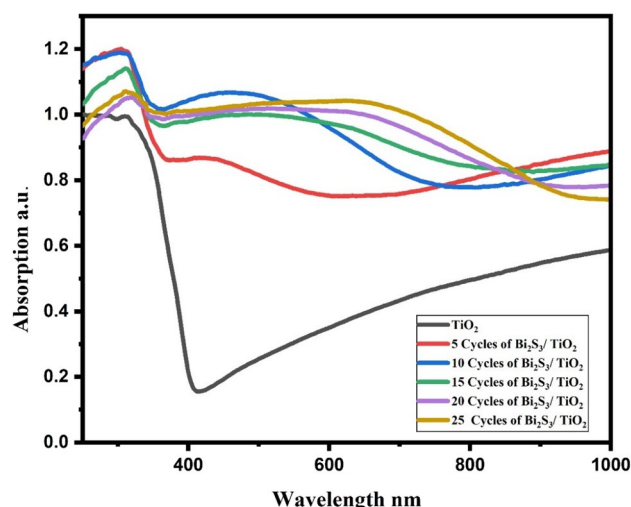
**Fig. 3** XRD of bare FTO,  $\text{TiO}_2$  film deposited by doctor blade method and  $\text{Bi}_2\text{S}_3$ -sensitized  $\text{TiO}_2$  film

$\text{Bi}_2\text{S}_3$  is successfully deposited on  $\text{TiO}_2$ . The average crystallite size was estimated to be 25 nm.

#### 3.2 Optical properties study

JASCO UV–Vis Spectrometers were used to measure the optical properties of  $\text{Bi}_2\text{S}_3/\text{TiO}_2$  films on FTO, and the UV–Vis spectra were measured in the range of 200–1000 nm. The absorption spectra of the  $\text{TiO}_2$  films are shown in Fig. 4. Spectral analysis revealed that the  $\text{Bi}_2\text{S}_3$ -sensitized  $\text{TiO}_2$  photoanode had an absorption band that extended from the ultraviolet to the visible region, indicating that the sensitization of  $\text{TiO}_2$  with  $\text{Bi}_2\text{S}_3$  nanoparticles improved its light absorption capabilities across a wider range of wavelengths. This is in contrast to the unsensitized  $\text{TiO}_2$  film, which primarily absorbs light in the ultraviolet region, thus limiting its efficiency in harnessing light energy for potential applications. When  $\text{Bi}_2\text{S}_3$  nanoparticles were loaded onto the  $\text{TiO}_2$  photoanode, the absorption width extended towards the visible wavelength region, enabling the sensitized  $\text{TiO}_2$  photoanode to capture a broader range of light wavelengths, potentially improving its efficiency and performance in solar cell applications. [35]

The shift in the absorption peak from 421 to 630 nm as the number of deposition cycles increased from 5 to 25 in the  $\text{Bi}_2\text{S}_3/\text{TiO}_2$  films indicates a significant change in the absorbed energy and optical properties of the material. To calculate the shift in the absorbed energy, we can use the following formula:



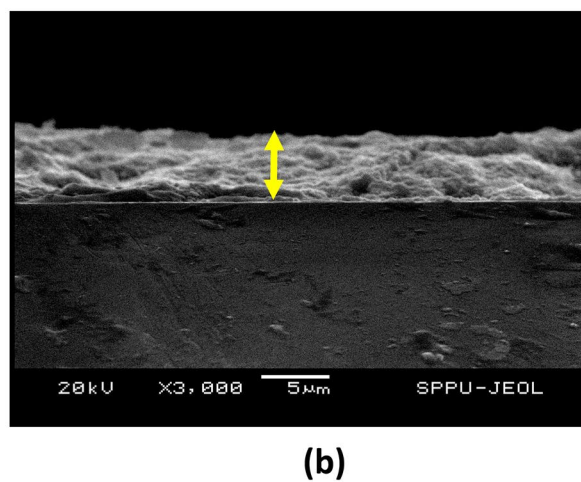
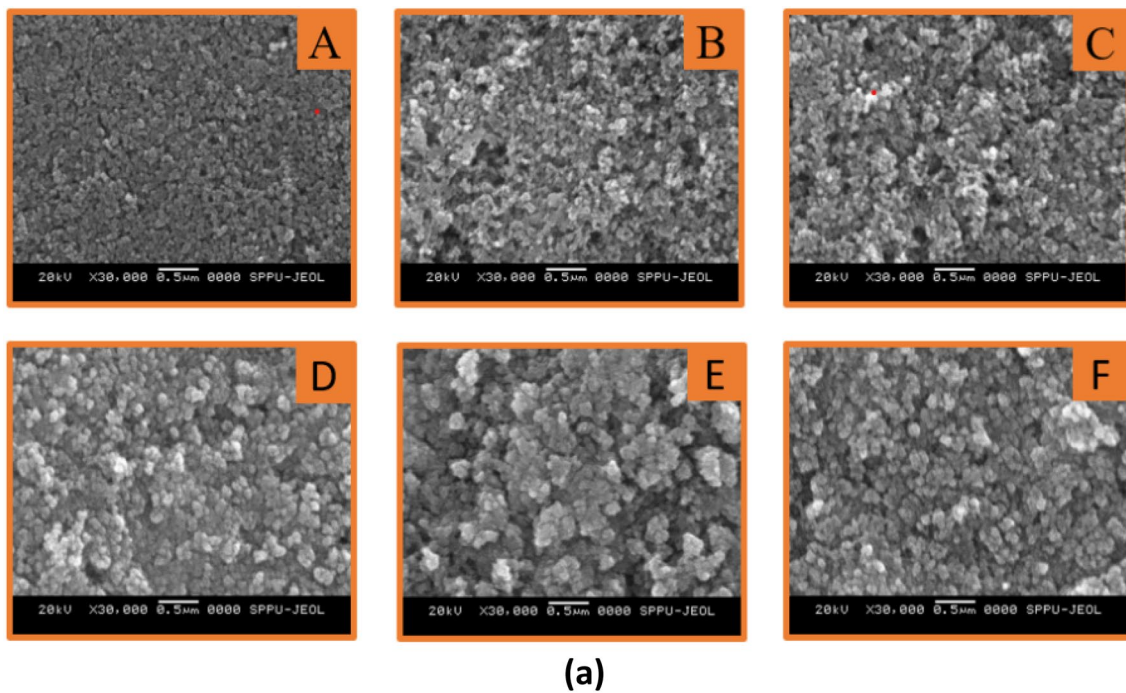
**Fig. 4** UV–Visible spectra of  $\text{Bi}_2\text{S}_3/\text{TiO}_2$  films

$$\Delta E = \frac{hc}{\lambda_2} - \frac{hc}{\lambda_1} \quad (1)$$

The equation relates the shift in the absorbed energy ( $\Delta E$ ), Planck's constant ( $h$ ), speed of light ( $c$ ), initial wavelength (421 nm), and final wavelength (630 nm). The shift in the absorbed energy was 0.98 eV.

### 3.3 Morphological analysis by scanning electron microscopy (SEM)

The surface morphologies of the TiO<sub>2</sub> compact and doctor-bladed films were recorded using a JEOL JSM-6360A Scanning Electron microscope [SEM]. Figures 5a (A to F) show SEM micrographs of the TiO<sub>2</sub> film deposited by the simple Doctor Blade method on a chemical bath-deposited compact TiO<sub>2</sub> layer showing granular morphology. Bi<sub>2</sub>S<sub>3</sub> nanoparticles were successfully loaded onto the TiO<sub>2</sub> film, and figures



**Fig. 5** **a** SEM micrographs of **A** TiO<sub>2</sub>, **B** Bi<sub>2</sub>S<sub>3</sub>/TiO<sub>2</sub>, – 5 SILAR cycles, **C** Bi<sub>2</sub>S<sub>3</sub>/TiO<sub>2</sub>, – 10 SILAR cycles, **D** Bi<sub>2</sub>S<sub>3</sub>/TiO<sub>2</sub>, – 15 SILAR cycles, **E** Bi<sub>2</sub>S<sub>3</sub>/TiO<sub>2</sub>, – 20 SILAR cycles, **F** Bi<sub>2</sub>S<sub>3</sub>/TiO<sub>2</sub>, – 25 SILAR cycles. **b** SEM cross section of TiO<sub>2</sub> film

**Table 1** Atomic percentage of Ti, O, Bi, and S

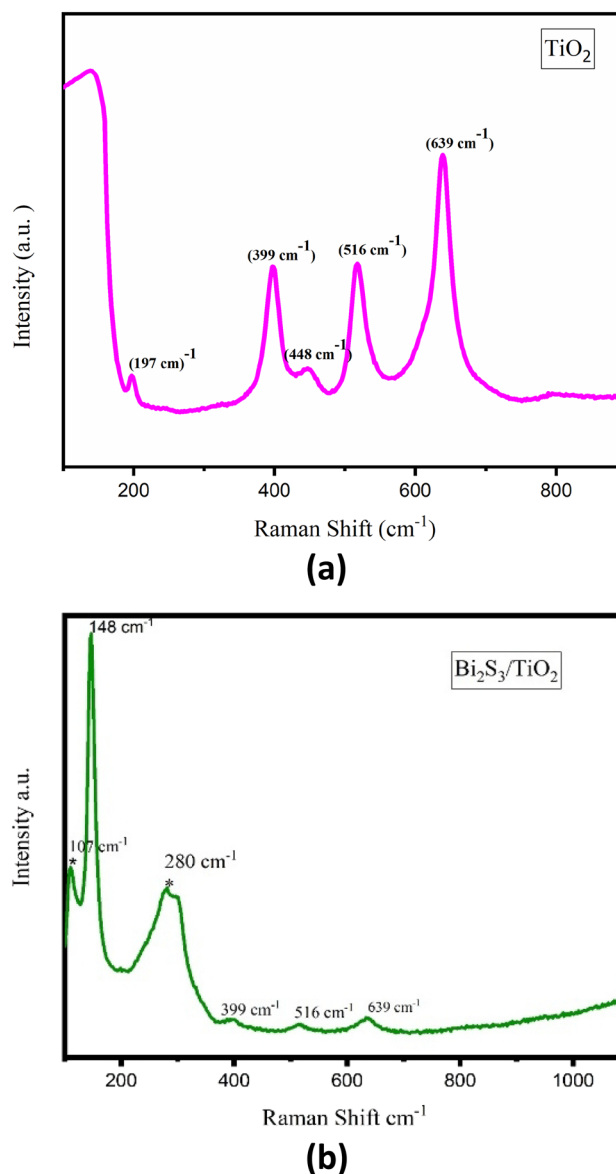
Material composition	No. of SILAR cycles	Ti At %	O At%	Bi At %	S At %
TiO <sub>2</sub>	–	18.80	81.20	–	–
Bi <sub>2</sub> S <sub>3</sub> /TiO <sub>2</sub>	5	17.30	81.66	0.77	0.27
Bi <sub>2</sub> S <sub>3</sub> /TiO <sub>2</sub>	10	15.90	81.48	1.55	1.07
Bi <sub>2</sub> S <sub>3</sub> /TiO <sub>2</sub>	15	16.37	77.72	3.23	2.67
Bi <sub>2</sub> S <sub>3</sub> /TiO <sub>2</sub>	20	16.61	73.34	4.84	5.21
Bi <sub>2</sub> S <sub>3</sub> /TiO <sub>2</sub>	25	15.36	76.36	4.32	3.97

B–F show that Bi<sub>2</sub>S<sub>3</sub> deposited TiO<sub>2</sub> films by various SILAR cycles (5–25). It indicates that as the number of SILAR cycles increased, the size of Bi<sub>2</sub>S<sub>3</sub> increased. Figure 5b shows the SEM cross section of the TiO<sub>2</sub> film; the thickness of the doctor-bladed TiO<sub>2</sub> film is approximately 5–7 μm. The deposition of Bi<sub>2</sub>S<sub>3</sub> was confirmed by energy-dispersive spectroscopy (EDS) analysis, as shown in Figure S1 (Supporting Information). The atomic percentages of Bi and S increased as the number of SILAR cycles increased, as shown in Figure S1a–f. The variation in the atomic percentages of Bi and S (the number of SILAR cycles varied from (5–25)) is shown in Table 1.

This study demonstrates that we successfully deposit Bi<sub>2</sub>S<sub>3</sub> nanoparticles on TiO<sub>2</sub> using the SILAR method. The number of SILAR cycles was varied from 5 to 25, and the atomic percentage of Bi<sub>2</sub>S<sub>3</sub> increased with the number of cycles. The EDS results confirmed the deposition of Bi<sub>2</sub>S<sub>3</sub>, showing increasing atomic percentages of Bi and S with more cycles. The atomic percentages of Ti and O decreased as the Bi<sub>2</sub>S<sub>3</sub> content increased, indicating the uniform coating of Bi<sub>2</sub>S<sub>3</sub> on TiO<sub>2</sub>. SEM and EDS results showed that the SILAR technique could tune the Bi<sub>2</sub>S<sub>3</sub> loading on TiO<sub>2</sub> by adjusting the number of cycles. In summary, the characterization techniques confirmed the controlled coating of TiO<sub>2</sub> with Bi<sub>2</sub>S<sub>3</sub> nanoparticles using the SILAR method by varying the number of deposition cycles.

### 3.4 Raman spectroscopy

Raman spectroscopy was used to determine the phases and crystallinity of the prepared samples. The Raman spectra of the TiO<sub>2</sub> and Bi<sub>2</sub>S<sub>3</sub>/TiO<sub>2</sub> samples are shown in Figs. 6a and b, respectively. Raman Spectra of TiO<sub>2</sub>/Bi<sub>2</sub>S<sub>3</sub> samples were collected in the range of 100–1000 cm<sup>-1</sup>. Figure 6a shows mixed phases of



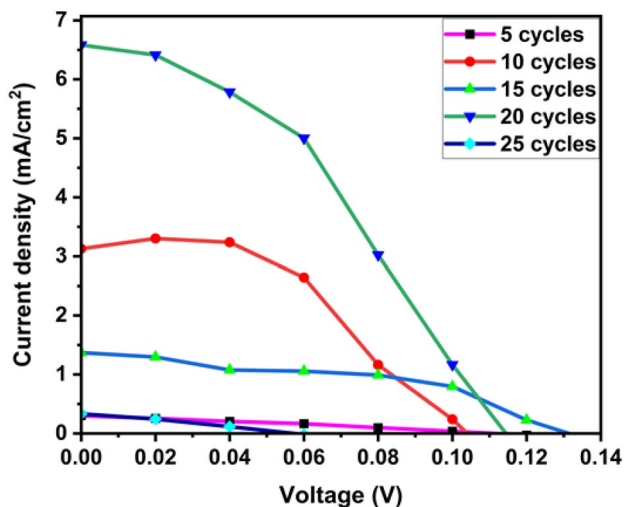
**Fig. 6** a Raman Spectra of TiO<sub>2</sub> film (Deposited by doctor blade method). b Raman spectra of Bi<sub>2</sub>S<sub>3</sub>/TiO<sub>2</sub> photoanode

anatase and rutile TiO<sub>2</sub>. Figure 6b) shows the Raman spectra of TiO<sub>2</sub>/Bi<sub>2</sub>S<sub>3</sub>. Peaks of TiO<sub>2</sub> and Bi<sub>2</sub>S<sub>3</sub> at 148 cm<sup>-1</sup>, 399 cm<sup>-1</sup>, 516 cm<sup>-1</sup>, and 639 cm<sup>-1</sup> confirming the formation of the anatase TiO<sub>2</sub> phase. The peaks at 148 cm<sup>-1</sup> and 639 cm<sup>-1</sup> arise mainly because of the symmetric stretching of O–Ti–O in TiO<sub>2</sub>, the peak at 399 cm<sup>-1</sup> shows the B<sub>1g</sub> vibrational mode caused by symmetric bending vibration, and the peak at 516 cm<sup>-1</sup> shows the A<sub>1g</sub> vibrational mode, which is mainly attributed to the antisymmetric bending vibration of TiO<sub>2</sub> [36, 37]. The peak at 107 cm<sup>-1</sup> corresponds to the

$A_{1g}$  and  $280\text{ cm}^{-1}$  to the  $B_{1g}$  modes, which correspond to  $\text{Bi}_2\text{S}_3$ .

### 3.5 JV curves

The JV curves of the  $\text{Bi}_2\text{S}_3/\text{TiO}_2$  film using the carbon electrode (Fig. 7) shows improved photovoltaic behavior with increasing  $\text{Bi}_2\text{S}_3$  thickness. The cell parameters of  $\text{Bi}_2\text{S}_3/\text{TiO}_2$  films using the Carbon electrode is shown in Table 2. The open-circuit voltage ( $V_{oc}$ ) increases from 0.11 to 0.14 V as the number of SILAR cycles increases from 5 to 25 cycles. The short-circuit current density ( $J_{sc}$ ) peaks at 20 SILAR cycles, reaching  $6.58\text{ mA/cm}^2$ , suggesting optimal light absorption and carrier generation. The fill factor ranged from 29.83 to 48.77%, and the highest value was achieved after 15 cycles. The overall power conversion efficiency peaked at 0.30% for 20 SILAR cycles, likely owing to non-optimal cell design and poor charge collection. Comparing the IV parameters for 15 cycles, there was a large difference in the fill factor and efficiency, possibly owing



**Fig. 7** Comparative JV curves of various SILAR cycles (5–20) of  $\text{Bi}_2\text{S}_3/\text{TiO}_2$  photoanode using C counter electrode

**Table 2** Solar cell parameters of  $\text{Bi}_2\text{S}_3/\text{TiO}_2$  photoanode using C counter electrode

SILAR Cycle	$I_{sc}$ (mA)	$V_{oc}$ (V)	Fill Factor (%)	$J_{sc}$ ( $\text{mA/cm}^2$ )	Efficiency (%)
5	0.01	0.11	29.83	0.30	0.01
10	0.05	0.13	44.20	1.37	0.07
15	0.25	0.10	48.77	3.12	0.15
20	0.13	0.11	39.83	6.58	0.30
25	0.07	0.14	34.54	2.57	0.12

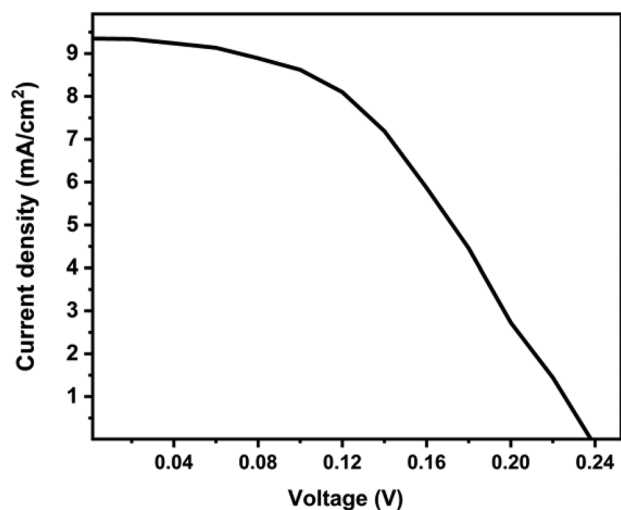
to measurement errors or variability. Overall, the IV results suggest that increasing the  $\text{Bi}_2\text{S}_3$  thickness improves the photovoltaic behavior until the optimal point is reached after 20 SILAR cycles. Here, we varied the SILAR cycles from 5 to 25 to obtain variable performance of the solar cells. Five cycles of  $\text{Bi}_2\text{S}_3$  on the mesoporous  $\text{TiO}_2$  layer resulted in incomplete sensitizer coverage. As the number of deposition cycles increased, the performance of the cell increased; however, after 25 cycles, the conversion efficiency of  $\text{Bi}_2\text{S}_3$  on  $\text{TiO}_2$  was lower than that at 20 cycles. This may cause aggregated deposition of sensitizers. Here, we evaluate the performance of a  $\text{Bi}_2\text{S}_3$  (20 cycles)/ $\text{TiO}_2$  photoanode with a CTS counter electrode. Twenty deposition cycles of  $\text{Bi}_2\text{S}_3$  on  $\text{TiO}_2$  resulted in the optimal loading of the sensitizer (neither incomplete nor aggregated) on the  $\text{TiO}_2$  layer. This optimized film (based on performance) was used to measure the performance of solar cells with a CTS counter electrode.

The solar cell parameters for 20 SILAR cycles of  $\text{Bi}_2\text{S}_3/\text{TiO}_2$  were obtained using a CTS counter electrode is shown in Table 3 and JV curve as shown in Fig. 8 showing a fill factor of 0.59, an open-circuit voltage of 0.238 V, a short-circuit current density of  $5.70\text{ mA/cm}^2$ , and a power conversion efficiency of 0.80%. The low efficiency indicates significant losses in the cell, while a fill factor of 0.59 indicates efficient charge carrier collection and low internal resistance. However, the low  $J_{sc}$  and efficiency suggest high recombination and low light absorption, possibly because of the insufficient thickness or poor quality of the  $\text{Bi}_2\text{S}_3$  absorber layer. The performance parameters

**Table 3** Solar cell parameters of  $\text{Bi}_2\text{S}_3/\text{TiO}_2$  photoanode using CTS counter electrode

SILAR cycle	$V_{oc}$ (V)	Fill factor (%)	$J_{sc}$ ( $\text{mA/cm}^2$ )	Efficiency (%)
20	0.23	0.59	5.70	0.80





**Fig. 8** JV curve of  $\text{Bi}_2\text{S}_3/\text{TiO}_2$  with CTS counter electrode (20 SILAR cycles)

suggest that the 15 SILAR cycle  $\text{Bi}_2\text{S}_3/\text{TiO}_2$  cell can generate a decent photovoltage, but improvements in photocurrent and power conversion are needed, likely through optimization of the  $\text{Bi}_2\text{S}_3$  layer and interface quality.

## 4 Conclusion

This study reports the feasibility of using a SILAR-deposited  $\text{Bi}_2\text{S}_3$  layer on a  $\text{TiO}_2$  Photoanode as a sensitizer for the conversion of photons in SSSC. In this study, a  $\text{Bi}_2\text{S}_3/\text{TiO}_2$  photoelectrode was synthesized on an FTO substrate via SILAR. SILAR-deposited  $\text{Bi}_2\text{S}_3$  is amorphous, which is a major drawback of this method. Annealing was essential for obtaining the crystalline orthorhombic phase of  $\text{Bi}_2\text{S}_3$ . In this study, we investigated the effect of the number of SILAR cycles on the performance of  $\text{Bi}_2\text{S}_3/\text{TiO}_2$ -based solar cells. XRD analysis showed the successful deposition of orthorhombic  $\text{Bi}_2\text{S}_3$  on tetragonal anatase/rutile  $\text{TiO}_2$ . UV-Vis absorption was extended to the visible range with  $\text{Bi}_2\text{S}_3$  sensitization. SEM and EDS confirmed the uniform  $\text{Bi}_2\text{S}_3$  coating on the  $\text{TiO}_2$ . The IV measurements showed increasing  $V_{oc}$ ,  $J_{sc}$ , FF, and efficiency with SILAR cycles, peaking at 20 cycles. Comparing the C and CTS counters, CTS provided a higher 0.80% efficiency versus 0.30% for C. In summary, SILAR-deposited  $\text{Bi}_2\text{S}_3$ -sensitized

$\text{TiO}_2$  to visible light absorption. Increasing the  $\text{Bi}_2\text{S}_3$  thickness improved the performance until it reached an optimal value after 20 cycles. The CTS counter electrode provides better efficiency than C. Further improvements in the cell design are needed to increase the efficiency. The  $\text{Bi}_2\text{S}_3/\text{TiO}_2$  heterostructure can be used for visible light-induced  $\text{H}_2$  evolution. The eco-friendly nature and low production cost are the most promising features of  $\text{Bi}_2\text{S}_3/\text{TiO}_2$  photoelectrodes for application in PEC hydrogen generation and photocatalysis.

## Acknowledgements

Abhijit T. Supekar is thankful to the State Government of Maharashtra for financial support through the CSMNRF-2020 fellowship. Shoyebmohamad F. Shaikh extends his sincere appreciation to the Researchers Supporting Project number (RSP2024R370), King Saud University, Riyadh, Saudi Arabia, for their financial support.

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## Funding

Open access funding provided by Manipal Academy of Higher Education, Manipal. Shoyebmohamad F. Shaikh extended his sincere appreciation to the Researchers Supporting Project number (RSP2024R370), King Saud University, Riyadh, Saudi Arabia, for their financial support for the project.

## Data availability

The datasets generated or analyzed during the current study are available from the corresponding author upon reasonable request.

## Declarations

**Conflict of interest** The authors declare that they have no known competing financial interests to declare.

**Supplementary Information** The online version contains supplementary material available at <https://doi.org/10.1007/s10854-024-13181-4>.

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