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Construction of a three-dimensional rGO/CoFe₂O₄ nanorods composite with enhanced microwave absorption performance

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ABSTRACT

Nowadays, designing and fabricating electromagnetic (EM) wave absorbers with lightweight and strong reflection loss (RL) is still urgent. Herein, rGO/ CoFe₂O₄ composite combined with magnetic CoFe₂O₄ nanorods and reduced graphene oxide (rGO) was fabricated via molten salt method in combination with a simple etching route. The EM absorption ability of rGO/CoFe₂O₄ nanorods composite can be regulated by changing the rGO addition amount. Particularly, the absorber for rGO/CoFe₂O₄ nanorods composite with the rGO addition amount of 70 mg (S-70) displays favorable EM wave absorption properties. The optimal RL value is -56.3 dB at the frequency of 16.9 GHz, when the corresponding thickness is 1.4 mm. The effective absorption bandwidth (EAB, RL < -10 dB) is 4.3 GHz, when the thickness is 1.6 mm. Consequently, the rGO/CoFe₂O₄ nanorods composite is a potential candidate for microwave absorbing materials.

1 Introduction

Due to the widespread use of communication devices and radar technology, EM radiation has become a tricky pollution question in many fields. It even has been regarded as a serious threat to electronic stability, healthcare and military fields [1–3]. So, it brings the great demand for developing high-performance EM wave absorbers which possess strong absorption, wide absorption bandwidth, thin thickness and lightweight [4–6]. According to theoretical analysis and experimental verification, the EM absorption materials absorbing microwave energy is basically impacted by dielectric and/or magnetic loss [7–9]. However, materials with only one loss factor are hard to obtain excellent EM wave absorption properties owing to inappropriate impedance matching and limited loss mechanism [1, 10]. Therefore, through artificially adjusting the EM

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parameters, fabricating the magnetic/dielectric composites with strong absorption capacity, high efficiency and wide absorption frequency has become a credible strategy to obtain excellent EM wave absorbers [11–14].

rGO has drawn considerable attention owing to two-dimensional (2D) single-layer structure, large specific surface area, abundant residual defects and appropriate dielectric loss [15, 16], which may lead to excellent electrical conductivity, thermal and mechanical properties. Nevertheless, the EM wave absorption properties of rGO are very poor because its high electric conductivity and low magnetic permeability lead to impedance mismatching characteristic [17, 18]. Therefore, many researchers have reported that modifying rGO surface with magnetic components can moderate its impedance matching and improve the absorption of the EM wave [19, 20]. Zhang et al. investigated that ternary rGO/MnFe2-O₄/PVDF composites exhibited enhanced EM wave absorption performance because of the synergetic effects between rGO, MnFe₂O₄, and PVDF [2]. Chen et al. presented a simple synthetic strategy for expanded graphite/polyaniline/CoFe₂O₄. The composite displays the best absorption of -19.1 dB and a thickness of 0.5 mm [21]. Zhang et al. reported that combined CoFe₂O₄ with rGO can ameliorate EM wave adsorbing performances of the former, and the RL_{max} of composite is -53.6 dB at 11.4 GHz [22]. Hence, introducing magnetic ferrites onto the surface of rGO sheets undoubtedly becomes а suitable strategy.

As a typical spinel ferrite, cobalt ferrite (CoFe₂O₄) has outstanding properties, such as moderate roomtemperature saturation magnetization, high chemical stability and mechanical hardness [23], which makes it to be a promising candidate for microwave absorbers. Many efforts have been made to design CoFe₂₋ with O₄-based composites controllable electromagnetic parameters. Li et al. successfully fabricated CoFe₂O₄/graphene oxide hybrids and obtained a maximum RL of - 10.8 dB at 4.0 mm with frequency of 9.2 GHz [24]. Zhang et al. prepared lightweight PANS@ CoFe2O4-RGO epoxy composites. The composites display the best absorption of - 14.8 dB at 13.6 GHz with a matching thickness of 2 mm [25]. Moitra et al. manufactured multifunctional CoFe₂O₄-reduced graphene oxide nanocomposites by a simple 'in situ' co-precipitation method and the RL_{max} of this nanocomposites is -31.3 dB at 9.1 GHz [26]. Zhang et al. obtained porous rugbyshaped $CoFe_2O_4$ particles and obtained optimal RL of - 34.1 dB at the thickness of 2.5 mm and the EAB is 2.6 GHz [23]. Fu et al. prepared hollow $CoFe_2O_4$ sphere/graphene composites and studied their possible absorbing mechanism [27]. However, the density and thickness is still unsatisfied, which heavily restricts its practical application.

Herein, we synthesized a three-dimensional rGO/ CoFe₂O₄ nanorods composite by using a facile-etching method. The corresponding synthetic route is displayed in Scheme 1. The hybrids show significant microwave absorption ability in terms of both the reflection loss and thickness because of moderate impedance matching of materials. Combining the two materials not only takes advantage of both but also increases the impedance matching and multiple interface loss between different materials [28]. Hence, this composite can be used as a potential lightweight microwave absorber.

2 **Experimental section**

2.1 Materials

Flake Graphite was provided by Qingdao Jinhui graphite Co. Ltd. Graphene oxide (GO) was prepared via Hummer's method with some modification. Iron nitrate nonahydrate (Fe(NO₃)₃·9H₂O), sodium chloride (NaCl), cobalt nitrate hexahydrate (Co(NO₃)₂·6H₂O), dimethyl formamide (DMF) and methyl mercaptoacetate (C₃H₆O₂S) were obtained from Sinopharm Chemical Reagent Co. Ltd. Hydrazine (NH₂NH₂·H₂O) was purchased from Aladdin Co. Ltd. All the chemicals were used without further purification.

2.2 Preparation of CoFe₂O₄ particles

CoFe₂O₄ particles were synthesized by molten salt method. According to the stoichiometry of CoFe₂O₄, the proper amount of Fe(NO₃)₃.9H₂O and Co(NO₃)₂.6H₂O were mixed in 50 mL ultrapure water and stirred for 2 h. The mixture were transferred into 100 mL kettle and heated at 180 °C for 12 h. A black precipitate of CoFe₂O₄ was obtained after 12 h. The precipitate was washed by centrifuge machine and dried by vacuum oven. Then, the





Scheme 1 The schematic illustration of the preparation of the rGO/Co2FeO4 nanorods composite

precipitate and NaCl were mixed according to the mass ratio of 1:1 and calcined at 1000 °C for 4 h. The mixture of $CoFe_2O_4$ and NaCl was washed with ultrapure water to remove the NaCl. At last, the $CoFe_2O_4$ particles were obtained.

2.3 Preparation of rGO/CoFe₂O₄ nanorods composite

0.2 g above CoFe₂O₄ particles and a proper amount (50, 60, 70, 80 mg) of prepared GO were sufficient dissolved in DMF in a three-necked bottle at 80 °C. Subsequently, 10 mL of NH₂NH₂·H₂O and 2.5 mL of C₃H₆O₂S were added to the above solution under N₂ atmosphere dropwisely with continuously stirring. After 2 h, the reaction was terminated with cold ethanol. Finally, the precipitate was pulpified alternately with ethanol and ultrapure water, followed by freeze-dried for 24 h. According to the different amounts of GO nanosheets (50, 60, 70, 80 mg), the obtained rGO/CoFe₂O₄ nanorods composites were named S-50, S-60,S-70 and S-80, separately. The preparation of the rGO/CoFe₂O₄ nanorods composites is investigated in Scheme 1.

2.4 Materials characterization

The morphologies and structures of $rGO/CoFe_2O_4$ nanorods composite were recorded by scanning electron microscopy (SEM, Hitachi, S-4800, Japan) equipped with energy dispersive X-ray analysis (EDX), and transmission electron microscopy (TEM, JEOL-2010, Japan). The specific surface area and pore size distribution were determined by the Brunner– Emmet–Teller method (BET, ASAP 2020, Micromeritics). The crystal structures of obtained products were observed by X-ray diffraction (XRD, D/max-2200, Japan). The carbon graphitization degree was investigated by Raman spectra (Raman, Horiba JobinYvon, France).

The EM parameters (ε' , ε'' , μ , μ'') of samples were characterized by vector network analyzer (VNA, HP8720ES, Agilent, USA) through the coaxial wire method in the frequency range of 2–18 GHz. rGO/ CoFe₂O₄ nanorods composites (30 wt%) were mixed with molten wax (70 wt%) and compacted into coaxial rings (d_{in} =3.04 mm, d_{out} =7 mm, and t = 3 mm).

3 Results and discussion

3.1 Structure and morphology of rGO/ CoFe₂O₄ nanorods composite

Figure 1a illustrates the XRD patterns of the rGO, CoFe₂O₄ particles, CoFe₂O₄ nanorods and rGO/ CoFe₂O₄ nanorods composite. Apparently, it appears a broad diffraction peak at about $2\theta = 25^{\circ}$ that correlates to the graphite-like structure (002) with an interlayer spacing of 0.36 nm in the XRD pattern of rGO, indicating that the GO has been changed to rGO [29]. The diffraction peaks for CoFe₂O₄ particles and CoFe₂O₄ nanorods are well matched with CoFe₂O₄ cubic spinel crystal structure (JCPDS card no. 22-1086). Remarkably, the intensity for the CoFe₂O₄ nanorods presents a significant drop compared with CoFe₂O₄ particles because the porous structure contains abundant tiny fragments. For rGO/CoFe2O4 nanorods composite, all the observed peaks correspond to CoFe₂O₄ and there are no other peaks, demonstrating the purity of rGO/CoFe₂O₄ nanorods composite. The diffraction pattern of rGO is not observed in rGO/CoFe2O4 nanorods composites



Fig. 1 a XRD spectra and b Raman spectra of the as-prepared samples

because the diffraction intensity of rGO may be too weak compared with that of the $CoFe_2O_4$ crystals.

The Raman spectra are investigated to reveal the structural information of composites (Fig. 1b). The Raman spectrum of GO has two prominent peaks at 1352 cm⁻¹ (D band) and 1589 cm⁻¹ (G band). The former represents the defect and disorder of carbon materials and the latter stands for the in-plane vibration modes of carbon atoms [30]. Three clear and prominent peaks located at about 311, 468 and 690 cm^{-1} in the spectrum of CoFe₂O₄ particles are detected, which is consistent with the characteristic bands of $CoFe_2O_4$ [31, 32]. The peak positions of CoFe₂O₄ nanorods are still in accordance with that of CoFe₂O₄ particles, but the intensities decrease, which corresponds to the XRD results. Further, the intensity ratio (I_D/I_G) is used to estimate the defect densities [33]. The intensity ratio of $rGO/CoFe_2O_4$ nanorods composite (S-70) $(I_D/I_G = 1.02)$ is bigger than that of GO ($I_D/I_G = 0.91$), indicating that many defects and disorders exist in S-70 [15, 20]. Meanwhile, the disorder nature is important for superior EM absorption materials [22].

The SEM and TEM images of $CoFe_2O_4$ particles, CoFe₂O₄ nanorods and rGO/CoFe₂O₄ nanorods composite (S-70) are displayed in Fig. 2. As shown in Fig. 2a, d, CoFe₂O₄ particles have a typical spinel morphology with a size of ~ 500 nm. And the nanorod-like CoFe₂O₄ powders with a length of about 300 nm were observed after etching process (Fig. 2b, e). As can be seen from Fig. 2c, CoFe₂O₄ nanorods are densely embedded in the flexible rGO sheets, meaning good adhesion between rGO and $CoFe_2O_4$ nanorods. The force between the rGO and $CoFe_2O_4$ is molecular force. The evolution of nanorods of $CoFe_2O_4$ may be attributed to the loss of $CoFe_2O_4$ which due to the reduction of Fe^{3+} to Fe^{2+} by hydrazine. Then, Fe^{2+} is immediately coordinated with methyl mercaptoacetate in DMF, which results in the removing of $CoFe_2O_4$. Thus, $CoFe_2O_4$ nanorods are formed on surfaces of rGO. TEM images of rGO/ $CoFe_2O_4$ nanorods composite (S-70) further reveal $CoFe_2O_4$ nanorods wrapped by rGO sheets.

The nitrogen absorption–deposition isotherm analysis of the CoFe₂O₄ particles, CoFe₂O₄ nanorods and rGO/CoFe₂O₄ nanorods composite (S-70) was investigated to study the surface area. As presented in Fig. 3, the samples showed typical IV isotherms, demonstrating their mesoporous character [34]. The surface areas are 0.94, 9.05 and 15.10 m² g⁻¹ for CoFe₂O₄ particles, CoFe₂O₄ nanorods and rGO/ CoFe₂O₄ nanorods composite (S-70), respectively. The larger the surface area, the more active sites for the effective reflecting and scattering EM waves can be provided, which may be good for the multiple absorption processes of EM waves.

The phase compositions of the as-synthesized $rGO/CoFe_2O_4$ nanorods composite with different rGO contents are measured by XRD (Fig. 4). All the diffraction peaks are well assigned to the standard card of the spinel structured CoFe₂O₄. And the peak at around 25° belonging to (002) crystal face of rGO cannot be indexed in composites, confirming that the intensity of rGO/CoFe₂O₄ nanorods composite peaks



Fig. 2 SEM images of a $CoFe_2O_4$ particles, b $CoFe_2O_4$ nanorods and c $rGO/CoFe_2O_4$ nanorods composite (S-70); TEM images of d $CoFe_2O_4$ particles, e $CoFe_2O_4$ nanorods and f $rGO/CoFe_2O_4$ nanorods composite (S-70)



Fig. 3 N_2 adsorption-desorption isotherms of the prepared CoFe₂O₄ particles, CoFe₂O₄ nanorods and rGO/CoFe₂O₄ nanorods composite (S-70)

is very strong. Meanwhile, no other unclear peaks appear, suggesting the purity of $rGO/CoFe_2O_4$ nanorods composite.

Typical morphology of the as-synthesized rGO/ $CoFe_2O_4$ nanorods composite (S-70) is shown in Fig. 5a–d. The composite is consisted with flexible 2D



Fig. 4 XRD patterns of $rGO/CoFe_2O_4$ nanorods composite with different rGO contents

sheet-like rGO and 3D nanorods. The larger magnification SEM images of $rGO/CoFe_2O_4$ nanorods composite (S-70) clearly show that the composite is assembled by numerous $CoFe_2O_4$ nanorods grown on rGO nanosheets, generating multiple interfaces in 3D hierarchical microstructures. Further, the composition of the rGO/CoFe_2O_4 nanorods composite further

analyzed by elemental mapping through EDS (Fig. 5e). The distributions of C, O, Fe and Co elements with different colors, implying the interconnection of $CoFe_2O_4$ nanorods and rGO.

3.2 EM wave absorption properties of rGO/ CoFe₂O₄ nanorods composite

The RL values for the $rGO/CoFe_2O_4$ nanorods composite were obtained by using the transmission line theory. They are calculated by experimentally obtaining electromagnetic parameters. The equations are as below [35–38]:

$$Z_{\rm in} = Z_0 \sqrt{\frac{\mu_{\rm r}}{\varepsilon_{\rm r}}} \tan h \left[j \frac{2\pi}{c} \sqrt{\mu_{\rm r} \varepsilon_{\rm r}} f d \right] \tag{1}$$

$$\mathrm{RL}(\mathrm{dB}) = 20lg \left| \frac{Z_{in} - Z_0}{Z_{in} + Z_0} \right|,\tag{2}$$

where Z_{in} represents the input impedance of the absorber, Z_0 means the characteristic impedance of free space, ε_r and μ_r are complex permittivity and permeability of the absorber, respectively. *f* is the frequency, *d* is the thickness of the specimen, and *c* is the velocity of light. Figure 6 is the RL–*f* curves of composites with different thicknesses. The amount of rGO introduced has significant influences on the EM wave absorbing capacities of samples. From Fig. 6a

Fig. 6 Frequency dependence of reflection loss $(\mathbf{a}-\mathbf{d})$ and the corresponding 3D contour maps of reflection loss versus frequency and thickness $(\mathbf{e}-\mathbf{h})$ for rGO/CoFe₂O₄ nanorods composite with different mass ratios of rGO

and e, the RL of S-50 is relatively poor with only -6.9 dB at 2.4 mm. Further increasing the rGO content to 60 mg (S-60), the RL_{max} is -16.2 dB at 14.4 GHz with 2 mm and the corresponding EAB is 3.1 GHz (from 12.5 to 15.6 GHz) (Fig. 6b and f). As shown in Fig. 6c and g, the composite of S-70 exhibits the best absorption performance than other composites. The RL_{max} reaches – 56.3 dB and the EAB is 3.4 GHz (14.6-18 GHz) at 1.4 mm. In addition, the EAB range can be well adjusted by changing the thickness. When the thickness is increased to 1.6 mm, the EAB is 4.3 GHz and the RL_{max} value is -26.4 dB. According, the excellent microwave absorption properties of S-70 may be ascribed to impedance matching characteristic and/or EM wave attenuation. Figure S1 shows the RL curves of CoFe₂O₄ particles and CoFe₂O₄ nanorods with various thicknesses. The RL_{max} value of CoFe₂O₄ particles and CoFe₂O₄ nanorods is - 14.1 dB and - 33.1 dB at 3 mm. Compared with CoFe₂O₄ particles, the RL_{max} value of CoFe₂O₄ nanorods is significantly increased, which may be due to its larger specific surface area. CoFe₂O₄



Fig. 5 SEM images with different magnifications (a-d), elemental mapping (e) of C, O, Fe and Co of rGO/CoFe₂O₄ nanorods composite (S-70)





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nanorods possess high surface area, which provide more active sites for the effective reflection and scattering of incident electromagnetic. Nevertheless, the RL_{max} values of both are lower than those of S-70. For S-80 in Fig. 6d and h, compared with S-70, the maximum RL value and the effective absorption frequency width are significantly reduced. Consequently, only when the amount of rGO is suitable can the absorbing property of the absorber be improved.

The microwave absorption properties of the absorbers usually are dependent on relative complex permittivity $(\varepsilon_r = \varepsilon' - j\varepsilon'')$ and relative complex permeability $(\mu_r = \mu' - j\mu'')$. The real part and the imaginary part represent the storage capability and loss ability [39–41]. The dielectric loss tangent (tan $\delta_{\varepsilon} = \varepsilon'' / \varepsilon'$) and the magnetic loss tangent (tan $\delta_{\mu} = \mu'' / \mu'$) indicate the dielectric loss and magnetic loss [42], respectively.

The function of ε' , ε'' , μ' , μ'' , tan δ_{ε} and tan δ_{μ} with frequency over 2-18 GHz for the rGO/CoFe₂O₄ nanorods composites (S-50, S-60, S-70 and S-80) is shown in Fig. 7, respectively. The ε' and ε'' values of rGO/CoFe₂O₄ nanorods composite are proportional to the introduction of rGO sheets, but inversely proportional to the frequency, which is a normal for dielectrics (Fig. 7a and b) [43–47]. The ε' values for S-50, S-60, S-70 and S-80 range from 3.8 to 6.6, 5.8–14.3, 8.2-25.7, and 8.7-30.5 in 2-18 GHz, respectively. And the curve of ε'' has a similar trend to the curve of ε' . The ε' and ε'' of the CoFe₂O₄ particles and CoFe₂O₄ nanorods are shown in Fig. S2 (a and b), which are much lower than those of the rGO/CoFe₂O₄ nanorods composite. This phenomenon reveals that introducing rGO nanosheets can greatly improve the permittivity of $rGO/CoFe_2O_4$ nanorods composite due to the high electrical conductivity property of rGO. Besides, the residual defects and groups of rGO surface bring the relaxation and polarization that increase the permittivity over 2-18 GHz. Figure 7c and d presents relationship between complex permeability (μ' and μ'') and frequency for rGO/CoFe₂O₄ nanorods composites. The μ' and μ'' values of rGO/CoFe₂O₄ nanorods composites do not differ significantly with different rGO proportions. This is because the introduced rGO is a non-magnetic material that will not influence the magnetic properties of the rGO/CoFe₂O₄ nanorods composite. The μ' and μ'' of the CoFe₂O₄ particles and CoFe₂O₄ nanorods are illustrated in Fig. S2 (c and d). They do not change a lot with increasing the frequency. The curves of tan δ_{ε} and tan δ_{μ} show obviously fluctuation over 2–18 GHz (Fig. 7e and f). And as rGO nanosheets are increased, tan δ_e is strengthened. Here, as to S-70, the dielectric loss is much higher than magnetic loss in the frequency range of 2–15 GHz, which denotes that the EM loss in the lower frequency mainly comes from dielectric loss but the magnetic loss mainly contributes in the higher frequency. The results display that the adjustment of rGO amount has great tunable potential for EM impedance matching.

Generally speaking, the microwave absorption ability mainly relies on the impedance matching $(|Z_{in}/Z_0|)$ and attenuation constant (α). The impedance matching characteristic is a prerequisite for achieving excellent microwave absorber [48]. When value of $|Z_{in}/Z_0|$ approaches to one, most of the EM waves can enter the absorber. Then they are effectively transformed to other forms of energy, or dissipated at the air–absorber interface. Nevertheless, the α is another important factor to determine the absorbing properties of absorbing materials. High α can effectively attenuate or transform the incident microwave. The α can be expressed as the following equation [49]:

$$\alpha = \frac{\sqrt{2}\pi f}{c} \times \sqrt{\left(\mu''\varepsilon'' - \mu'\varepsilon'\right) + \sqrt{\left(\mu''\varepsilon'' - \mu'\varepsilon'\right)^2 + \left(\mu'\varepsilon'' + \mu''\varepsilon'\right)^2}}.$$
(3)

Figure 8 presents the EM $|Z_{in}/Z_0|$ and α of rGO/ CoFe₂O₄ nanorods composite with different rGO proportions. The $|Z_{in}/Z_0|$ values of S-70 are closer to 1 among four composites in the whole frequency range at 1.4 mm (Fig. 8a), indicating that S-70 has the best impedance matching. From Fig. 8b, S-70 and S-80 show much higher α than other composites. However, the EM absorption performance of S-80 is not as good as that of S-70, illustrating that the superior microwave absorber not only influenced by comprehensive attenuation ability, but also by impedance matching characteristics.

The magnetic loss of rGO/CoFe₂O₄ nanorods composite mainly comes from the CoFe₂O₄ nanorods. Normally, the magnetic loss is basically originated from hysteresis loss, domain wall resonance, eddy current loss, natural resonance and exchange resonance [50–52]. Hysteresis loss and domain wall resonance loss always occur at low frequency (MHz). The eddy current loss donation contribution is decided from $C_0 = \mu''(\mu')^{-2}f^{-1}$ [53]. When the magnetic



Fig. 7 Frequency dependence of the real part (a) and imaginary part (b) of relative complex permittivity, real part (c) and imaginary part (d) of relative complex permeability, dielectric

loss (e) and magnetic loss (f) for $rGO/CoFe_2O_4$ nanorods composite with different mass ratios of rGO



Fig. 8 The impedance matching (a) and attenuation constants α (b) versus frequency for all samples

loss originates from the eddy current loss, the value of C_0 remains unchanged [54]. Figure 9 illustrates the C_0-f curves of four samples. A noticeable variation can be detected, suggesting the inexistence of eddy current effect. From the above results, the natural resonance and exchange resonance are the main forms of magnetic loss of rGO/CoFe₂O₄ nanorods composite. Meanwhile, the significant variation of C_0 at 2–7 GHz is basically because of the natural resonance, while at higher frequency the exchange resonance loss dominates.

Scheme 2 shows the proposed EM wave absorbing mechanism of the rGO/CoFe₂O₄ nanorods absorber. The absorbing property comprehensively depends on the impedance matching, dielectric loss and magnetic loss mechanisms [55]. Firstly, combining CoFe₂O₄ nanorods with rGO provides dielectric loss and magnetic loss. The optimal impedance matching is a prerequisite to allow EM waves to get into the absorbers to the maximum extent [56]. Secondly, the interfaces among rGO/rGO, rGO/air and rGO/CoFe2O4 nanorods induce lots of interfacial polarization. The interface polarization caused by multiple interfaces converts EM wave energy into thermal energy [55]. Thirdly, the defects and function groups in rGO provide many dipoles resulting in polarization and relaxation loss [50, 57]. Finally, the special structure of $rGO/CoFe_2O_4$ nanorods composite is prone to multiple reflections, which is beneficial to improve the EM wave absorbing performance [58–60].



Fig. 9 $\mu''(\mu')^{-2}f^{-1}$ values of rGO/CoFe₂O₄ nanorods composite with different mass ratios of rGO

EM wave absorbing performances of $rGO/CoFe_2$. O₄ nanorods composite and other metal oxide composites are displayed in Table 1. Compared with most of them [61–69], rGO/CoFe₂O₄ nanorods composite (S-70) revealed outstanding performance at a thin thickness, which meet the requirement of "thin, strong, light, wide". Thus, rGO/CoFe₂O₄ nanorods composite can be theoretically used as a high-efficient and controllable microwave absorbers.



Scheme 2 Possible mechanism for the microwave absorption of rGO/CoFe₂O₄ nanorods composite

Material	Matrix	Absorber content	Optimum frequency (GHz)	Optimum thickness (mm)	$R_{\rm L}$ (dB)	Bandwidth (GHz)	References
rGO/Fe ₃ O ₄	Paraffin	50 wt%	12.07	2.5	- 18.2	2	[61]
rGO/ZnO	Paraffin	50 wt%	9.7	2.2	- 45.05	2.5	[62]
rGO/β-LiFe5O8	Paraffin	40 wt%	16.5	9	- 26.3	2.5	[63]
rGO/ZnFe ₂ O ₄	Paraffin	30 wt%	16.7	1.6	- 29.3	2.6	[64]
rGO/BiFeO3	PVA	10 wt%	10.68	1.55	- 28.68	2.1	[65]
rGO/Ni _{0.8} Zn _{0.2} Fe ₂ O ₄	Paraffin	50 wt%	10.6	2.0	- 18.4	3.32	[66]
PEDOT/rGO/Co ₃ O ₄	Paraffin	50 wt%	10.7	2.0	- 51.1	3.1	[67]
MoS ₂ /Fe ₃ O ₄ /graphene	Paraffin	30 wt%	5.9	2.5	- 45.8	5.1	[68]
rGO-Fe ₃ O ₄ @ZnO	Epoxy	30 wt%	11.4	5	- 38.0	2.4	[<mark>69</mark>]
rGO/CoFe ₂ O ₄ nanorods composite	Paraffin	30 wt%	16.9	1.4	- 56.3	3.4	This work

Table 1 EM wave absorbing performances of rGO/CoFe₂O₄ nanorods composite and other metal oxide composites

4 Conclusions

CoFe₂O₄ nanorods anchored on rGO nanosheets have been successfully prepared via etching method. The as-prepared rGO/CoFe₂O₄ nanorods composite presents superior microwave absorption properties. Different amounts of GO nanosheets are introduced in the composite to support the CoFe₂O₄ nanorods. The results suggest that the optimal RL value of rGO/CoFe₂O₄ nanorods composite (S-70) could reach - 56.3 dB with a thickness of 1.4 mm along with an EAB of 3.4 GHz. And the heterostructure of rGO/CoFe₂O₄ nanorods composite could be widely applied as strong capacity and lightweight microwave absorbing materials.

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