

Study of the optical and gas sensing properties of In₂O₃ nanoparticles synthesized by rapid sonochemical method

Hafeez Ullah^{1,*} , Zain H. Yamani², Ahsanulhaq Qurashi^{2,5}, Javed Iqbal³, and Kashif Safeen⁴

¹ Institute of Physics & Electronics, Gomal University, Dera Ismail Khan 29220, KP, Pakistan

²Center of Excellence in Nanotechnology (CENT), KFUPM, Dhahran 31261, Saudi Arabia

³Laboratory of Nanoscience and Technology (LNT), Department of Physics, Quaid-I-Azam University Islamabad, Islamabad, Pakistan

⁴Department of Physics, Abdul Wali Khan University Mardan, Mardan, Pakistan

⁵Department of Chemistry, Khalifa University, Abu Dhabi 127788, United Arab Emirates

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ABSTRACT

Indium oxide (In_2O_3) nanoparticles were synthesized via a facile rapid sonochemical method. Detailed spectroscopic techniques were used to investigate optical, structural and chemical properties of the synthesized In_2O_3 nanoparticles. The structural analysis shows that In_2O_3 nanoparticles have cubic structure and are polycrystalline in nature. The morphology of the In_2O_3 nanoparticles examined by field emission scanning electron microscopy revealed spherical and uniformly distributed particles. Bruno emit Teller surface analyzer demonstrated that the surface areas of In_2O_3 nanoparticles is 45 m²/g and also confirmed that the synthesized nanoparticles are mesoporous. Raman spectra also revealed that the synthesized nanoparticles have cubic structure. In–O band stretching of the synthesized In_2O_3 nanoparticles was confirmed using Fourier Transform Infrared Spectroscopy. Photoluminescence spectra of the In_2O_3 nanoparticles showed broad and intense UV emission peak at 358 nm. Moreover, the synthesized In_2O_3 nanoparticles showed good sensitivity and fast response toward the hydrogen gas at lower temperature.

1 Introduction

Nanostructure semiconducting materials demonstrate very interesting chemical, optical properties compared to those of bulk materials. These nanostructures semiconductor materials exhibit extraordinary properties towards the novel devices [1]. In last decade, various type of nanostructures were synthesized to optimize different kinds of properties such as chemical, electrical and optical

Address correspondence to E-mail: hafeezullah83@gmial.com

properties of these nanostructure materials. In₂O₃ has a wide band gap (3.6 eV) n-type semiconductor having large applications in many fields including solar cell, gas sensor, field emission display, biosensor, optoelectronics and photo catalysis [2–8]. Various chemical and physical vapor deposition techniques including sol gel, co precipitation, pulse laser depositions, thermal evaporation and sputtering have been used for the preparation of Indium oxide nanostructure. However, these techniques involved complicated procedure, deployed expansive equipments and required high energy for operation. On the other hand, wet chemistry method used in the present work required low energy, economical and is easy to handle. The wet chemistry method includes spray pyrolysis, spin- and dip-coating [9]. Furthermore, the indium oxide nanostructure materials have been synthesized using the surfactant templates method. However, the sol-gel method and co-precipitation method are the mostly used method for synthesis of these nanoparticles. Using the wet chemistry approach, the surface morphology, shape, crystallinity and particles size of indium oxide nanoparticles can be controlled. Further, highly crystalline nanoparticles can be obtained using the wet chemistry approach at moderate temperature. Amongst wet chemistry synthesis approach, recently sonochemistry has been attracted great attention for the synthesis of ultrafine metal oxide nanostructure. In this technique, the cavitation and microbubbles are generated with powerful ultrasound radiation. Parameters of the process can be controlled easily in the sonochemical method. The important parameters of the sonochemical reaction are the bob vibrating frequencies, temperature plays significant role in the size of the particles, shape morphology and the crystallinity of indium oxide nanoparticles [10–12].

In this paper, In_2O_3 nanoparticles were synthesized by simplest sonochemical method for the gas sensor applications. The synthesized In_2O_3 nanoparticles were characterized using various techniques demonstrated well excited optical properties, structural and shows good response toward the hydrogen gas. Moreover, the synthesized In_2O_3 nanoparticles can be used in high conducting transparent oxide in other optoelectronics devices.

2 Experimental

In₂O₃ nanoparticles were synthesized using indium acetylacetonate (high purity Sigma-Aldrich) as a precursor. The appropriate quantity of indium acetylacetonate precursors were dissolved in the distilled water. 4 mL NH₄OH was added to the prepare solvent in order to maintain pH around 9-10 during the sonochemical process. The obtained yields in the solution form have been treated using a high ultrasonic intensity processor (Power = 750 W, frequency = 20 kHz and Ti-probe tip = 6 mm) via direct-immersion way for 60 min. During the sonication's the temperature of the precursor's solution was fixed in the range of 80–90 °C using a water bath. After the sonication the homogeneous and transparent solution were obtained. In order to removal the acetate and other residues, the final solutions have been centrifuged and subsequently washed numerous times using deionized water. The obtained powders were at 70 °C in the oven. Finally, the dried powders were calcined at different temperatures 400 °C, 700 °C, and 800 °C for 2 h.

The crystalline phase and crystallite size of the synthesized In2O3nanoparticles was investigated using Mini-X-ray Diffraction (Mini-XRD) with CuKa X-ray radiation ($\lambda = 0.15406$ nm). The morphology, size of In₂O₃nanoparticles and chemical composition has been examined with (FESEM/FIB/GIS (Tescan Lyra-3) and EDX equipped with FESEM. The surface area and porosity of In₂O₃ nanoparticles were carried out by surface area and porosity analyzer (ASAP2020). The photoluminance spectrum of the In₂O₃ nanoparticles were measured at room temperature by the fluorescence spectrometer (FLS920) Xe lamp with excitation wavelength using λ_{ex} = 315 nm. The Raman spectrum was taken at room temperature with IHR320 Horiba Spectrometer with CCD.

3 Results and discussion

3.1 XRD analysis

Figure 1 shows the XRD patterns of In_2O_3 nanoparticles calcined at 400 °C. In the XRD spectrum peak positions at 21.39, 30.89, 35.59, 51.45, and 60.72 are assigned to reflections of the planes (211), (222), (400), (440) and (622), respectively. The highest intensity of the (222) diffraction peak clearly shows that the prepared In_2O_3 nanoparticles are cubic polycrystalline structure. The peak position in the XRD pattern shows the cubic structure by comparing with standard pattern of cubic structure of In_2O_3 (JCPDS No. 06-0416). The lattice parameter and cell volume of the In_2O_3 nanoparticles is calculated using the following equations.

$$a = d_{hkl}(l^2 + h^2 + k^2)^{1/2} \tag{1}$$

$$V = a^3 \tag{2}$$

where "*a*" is the lattice parameter, *h*, *k* and *l* are the miller indices of the plane and d_{hkl} is the inter-planar spacing. The experimentally obtained lattice parameter "*a*" and cell volume "*V*" of the In₂O₃ nanoparticles are 10.07 Å and 1021.14 Å³ respectively.

The crystallite size of the In_2O_3 nanoparticles determined using the Williamson–Hall plot of $\beta \cos \theta$ along axis and put $4\sin \theta y$ axis, using Eq. (2) was applied to the (222), (400), and (440) peaks. The plot of $4\sin \theta$ versus $\beta \cos \theta$ taking (222), (400), and (440) lattice planes. The crystallite size D was extracted from the linear fit from the intercept at y-axis.

$$\beta_{\rm hkl}\cos\theta = \frac{K\lambda}{D} + 4\varepsilon\sin\theta \tag{3}$$

Here β_{hkl} is the full width at half-maximum (FWHM) of the peak considered (in rad) corresponding to be induced by both the average crystallite size D and strain ε . Where λ is the X-ray wavelength (0.1541 nm) for CuK α radiation while *K* is approximately equal to 0.9. The crystallite sizes



Fig. 1 XRD pattern of indium oxide nanoparticles calcined at 400 $^{\circ}\mathrm{C}$

and strain of the In_2O_3 nanoparticles are 15.2 \pm 1.8 nm and 6.5 \times 10^{-3} respectively.

3.2 SEM and EDX analysis

Figure 2a, b shows the morphology of In_2O_3 nanoparticles taken at resolution of 1 µm and 500 nm, respectively. FESEM images indicate that the synthesized In_2O_3 nanoparticles are in spherical shape and homogeneously spread. The FESEM micrographs are also demonstrated that the synthesized In_2O_3 nanoparticles size in the range from 15 to 30 nm and well disperse. The sizes of the In_2O_3 nanoparticles from the FESEM micrograph are good agreement with XRD finding.

The energy-dispersive spectroscopy has been used for the chemical and compositional analysis of In_2O_3 nanoparticles. Figure 3 shows the EDX spectrum that the presence of oxygen and indium elements. An extra peak also appeared in the EDX spectrum which might be associated with gold induced from coating of the sample for analysis.

3.3 Photoluminescence properties

Figure 4 shows the room temperature PL spectra of In₂O₃ nanoparticles, measured by means of Xenon laser of wave length 250 nm used as excitation source. The strong UV emission of the In₂O₃ nanoparticles have been obtained at broad band centered at \sim 358 nm (3.47 eV). Such kind of emission is not reported for bulk In_2O_3 [13]. In the present case, it can be linked with the nanostructure In_2O_3 as validated from the FESEM and XRD analysis. One possibility of this emission could be oxygen vacancies as reported in literature [14–17]. The arises of oxygen vacancies in the synthesized In₂O₃ nanoparticles can be explained as follow, during calcination of the In_2O_3 nanoparticles, some sight of oxygen become incomplete or might be some intrinsic defects arises, which could be related to the oxygen vacancies [17]. The suppression of defect-related emission of In₂O₃ is correlated to the reconstruction of defect structures. These oxygen vacancies formed new energy level near or in the band gap of the In_2O_3 , which could be normally acted as deep defects donors' level. The strong UV emission of the In₂O₃ nanoparticles might be the radiative recombination of the photo-excited holes and electrons occupied by oxygen vacancies. Similar mechanism is already explained for the ZnO











Fig. 4 Photoluminescence of In₂O₃ nanoparticles

and SnO_2 semiconductor [18]. The other possibilities of the oxygen vacancies could the size of the indium oxide nanoparticles. It is already mentioned in the FESEM analysis that the synthesized In_2O_3 nanoparticles size is in the range of 15–30 nm. The small size would favor the existence of oxygen vacancies for the high surface to volume ratio of In_2O_3 nanowires [19].

3.4 Brunauer–Emmett–Teller

Brunauer–Emmett–Teller (BET) method was employed to determine the specific surface area and pore size of the In_2O_3 nanoparticles in the N_2 adsorption–desorption isotherm. Figure 5 illustrates the isotherm of In_2O_3 nanoparticles. The synthesized In_2O_3 nanoparticles are related to the type IV profile of the adsorption–desorption isotherm indicating the mesoporous nature of the nanoparticles. The estimated surface area, pore size and volume of the synthesized In_2O_3 nanoparticles were found around of 45 m²/g, 10.5 nm and 0.28 cm³/g, respectively.

3.5 FTIR spectra

Room temperature FTIR spectra of In_2O_3 were recorded (range 400–4000 cm⁻¹) to confirm the stretching vibration of In–O bonds (see Fig. 6). The spectra contain high adsorption stretching. The stretching band near to 1990 cm⁻¹arises due to C–H stretching confirmed [20]. The nitrate group band stretching and bends deformation of water appeared at 1630 cm⁻¹ [21]. The presence of high intensity stretching vibration band of OH and C–H might be the adsorption of the moister from the surrounding





Fig. 5 Nitrogen adsorption–desorption isotherms of In_2O_3 nanoparticles

after the calcination of the synthesized samples. The In–O band stretching is appeared at about 630 cm⁻¹. Concluding the FTIR analysis, it is clearly confirmed that the synthesized indium oxide has In–O band stretching at about 630 cm⁻¹.

3.6 Raman spectroscopy

Raman spectroscopy were used to identify the low energy modes of In_2O_3 nanoparticles, the disorderness induce in the modes due to the strain after the calcination of the synthesized In_2O_3 nanoparticles. Cubic In_2O_3 belongs to I_a^3 and T_h^1 space group. For the cubic In_2O_3 , the total expected low energy active Raman mode in the infrared region were 22 and 16. Wang et al. and Zhang et al. reported 6 Raman modes and 11 infrared modes for cubic In_2O_3 [22, 23]. The Raman mode which is always active in vibration symmetry are A_{g} , E_{g} , T_{g} and T_{u} [22, 24]. Figure 7 shows the Raman spectrum of the prepared samples calcinated at 400 °C, 600 °C and 800 °C. Five vibrational modes were observed in the Raman spectra calcined at 600 °C, which appeared at 303 cm⁻¹, $364 \text{ cm}^{-1},435 \text{ cm}^{-1}, 492 \text{ cm}^{-1}$ and 619 cm^{-1} . From the literature, these modes are belonging to In₂O₃ cubic structure [25]. The shift in Raman peaks towards the lower wave numbers observed with increasing the calcination temperature. This shift may be due to the presence of defects as function of calcination temperature such as strain and other structural defects induced in the structure of In₂O₃ with the increasing calcination temperature. Structural disordering in the Raman spectroscopy is mainly due to defects induced in the metal oxide structure which may be caused to break the selection rule.

3.7 Gas sensing properties

Figure 8 shows the measurement setup for gas sensing properties of In_2O_3 nanoparticles. The gas sensing response of the samples was investigated with variation of time versus the electrical resistance for ambient of hydrogen gas. The response time, defined as the time needs to drop the resistance of the sensing material by 90% of the initial baseline resistance with



Fig. 6 The FTIR spectra of In₂O₃ nanoparticles

Deringer



Fig. 7 Raman spectra of In_2O_3 nanoparticles calcined at 400 °C, 600 °C and 800 °C

introducing of the H_2 gas to the sensing material. The response time is important characteristic of the sensor after changing the gas ambient. The response time of the In_2O_3 nanoparticles and the changing the ambient gas of these nanoparticles are illustrated in Fig. 9.

Figure 9 presents the gas sensing properties of In_2O_3 nanoparticles for hydrogen (H₂) gas. Very quick response toward the hydrogen gas was observed by In_2O_3 nanoparticles at 200 °C. Correspondingly, the sensor returns again very quickly to its original position after purging out the H₂ gas. The quick and fast response of the In_2O_3 nanoparticle toward the H₂ gas could be full surface area exposure of the In_2O_3 molecules to the chemical environment.

As from the BET measurement is demonstrated that the In₂O₃ nanoparticles have large surface area to volume. Due to this large surface area of the In₂O₃ nanoparticles, a greater number of In₂O₃ molecules are available to contribute the surface reaction. It is well known that as the size of particles decreases, the surface area is increased vice versa. The behavior of the response of the In₂O₃ nanoparticles is related to the surface reaction of the oxygen species and the reducing gas in contact [26-28]. Increasing surface area of In₂O₃ nanoparticles increases adsorption sites and hence large number of oxygen species is available for the reaction. Significant change in the resistance due the interaction of reducing gas shows higher sensitivity of the sensor [29]. This feature can contribute to the change in the resistance of In₂O₃ nanoparticles, with enhance sensing response and lower temperature. Moreover, working for

Fig. 8 Measurement setup for gas sensing properties of In₂O₃ nanoparticles



Fig. 9 Gas sensing properties of In₂O₃ nanoparticles at 200 °C

improving the sensor sensitivity, particle size and porosity of the prepared material structure must be taken into consideration, which depends upon the sensor surface area [30]. The sensitivity of the gas response of In_2O_3 semiconductor can be measured through the power-law relation [31].

$$S(\%) = \frac{R_{air} - R_{hydrogen}}{R_{air}} \times 100$$
(4)

The sensitivity can be calculated from the power law. The sensitivity of the of In_2O_3 nanoparticles was around $13.5\% \pm 1\%$. The response time estimated was around 30 to 35 s while the recovery time was found 40 to 45 s. When H_2 gas reacts with In_2O_3 nanoparticles, it has been found that the baseline



resistance of the In_2O_3 nanoparticles is much low (in the range of 50–250 Ω). It has been established that low resistance of the sensing material layer is very suitable for the reduction of the noise in the analysis and consequently shows the high signal/noise ratio. The decrease in the resistance of In_2O_3 in H_2 gas sensing might be the n-type semiconducting behavior of these nanoparticle also having high mobility and more intrinsic electronic concentrations [32].

4 Conclusion

The In₂O₃ nanoparticles were successfully prepared using sonochemical techniques. The X-ray diffraction result demonstrated that the synthesized In₂O₃ nanoparticles have crystallite nature with cubic structure. The crystallite size and strain of the In₂₋ O₃nanoparticles estimated from XRD were 15.2 ± 1.8 nm and 6.5×10^{-3} . FESEM micrographs show that In₂O₃ nanoparticles have spherical and uniformly distributed particles. EDX spectrum confirmed the presence of indium and oxygen elements. The photoluminescence spectra of the In₂O₃ nanoparticles showed broad and intense UV emission peak at 358 nm (3.47 eV). The Raman active modes at 629 cm⁻¹ (E_{2g}), 494 cm⁻¹ (A_{1g}), 306 cm⁻¹ (E_{1g}) of the calcined sample at 400 °C confirmed the cubic structure of In2O3. BET of the nanoparticles shows that the N2 adsorption-desorption has isotherm of type IV. The BET specific area of the In₂O₃ nanoparticles is 45 m²/g. FTIR results confirm nature of molecular bonding in In₂O₃. The gas sensitivity of In₂O₃ nanoparticles measured at 200 °C showed good response for hydrogen gas. This would contribute to the safe development of hydrogen energy for various applications.

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