

One-step hydrothermal synthesis of Ni₃S₄@MoS₂ nanosheet on carbon fiber paper as a binder-free anode for supercapacitor

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Abstract $Ni₃S₄@MoS₂$ nanosheets grown on the carbon fiber paper ($Ni₃S₄@MoS₂/CFP$) were successfully prepared via a facile one-step hydrothermal technique. Serving as a supercapacitor electrode, the obtaining $\text{Ni}_3\text{S}_4@\text{MoS}_2/\text{CFP}$ exhibits a remarkable specific capacitance of 1296 F/g, high rate capability of 57.8%, and excellent cycling stability of 96.2% retention after 5000 cycles. The superior electrochemical properties of $Ni₃S₄@MoS₂/CFP$ nanocomposite is attributed to the synergistic effects of the layered $Ni₃S₄$, $MoS₂$ and the conductive CFP.

1 Introduction

As one of the most promising energy storage devices, supercapacitor (SC), also named electrochemical capacitor, has attracted worldwide attention due to its excellent properties, such as rapid recharge ability, high power density, long-life cycles, eco-friendly nature and low-cost preparation [\[1](#page-7-12)[–3](#page-7-13)]. Based on the different charge storage mechanism, supercapacitor is normally divided two different categories: double-layer capacitor and pseudo-capacitor. The former usually uses carbon material as electrode material [\[4](#page-7-14)[–7](#page-7-15)], while the latter employs some metal oxide and conducting polymers as electrode material based on redox capacitive mechanism $[8-11]$ $[8-11]$. The continuous exploration of high-capacity materials and the design of optimal

 \boxtimes Yezeng He hyz0217@hotmail.com electrode architecture have been an indispensable part of the significant development of SC.

Metal sulfide has recently aroused the attention due to their unique electrochemical property [[12–](#page-7-0)[14\]](#page-7-1). As a typical example, molybdenum sulfide $(MoS₂)$ where Mo atom can easily prompt redox faradaic reaction in the interlayer space due to the oxidation states from the $+2$ to $+6$ has been proved to be an promising electrode material and its theoretical specific capacitance is high up to 1000 F/g [\[15](#page-7-2), [16](#page-7-3)]. Ramadoss et al. prepared mesoporous $MoS₂$ nanostructure by a hydrothermal method and obtained maximum capacitance of 376 and 403 F/g at a scan rate of 1 mV/s in 1 M $Na₂SO₄$ and KCl electrolyte solutions respectively [\[17](#page-7-4)]. Wang et al. have synthesized a hierarchical $MoS₂$ nanospheres by hydrothermal method for supercapacitors, with their capacitance of 142 F/g at current density of 0.59 A/g [\[18](#page-7-5)]. However, the specific capacitance of $MoS₂$ is still very poor due to low electronic conductivity and small quantities of accessible active sites of $MoS₂$, which limited the further application and development.

On the other hand, another transition metal sulfide, nickel sulfides, have many advantages for the application of supercapacitors because of good inductivity, natural abundance and versatility [[13,](#page-7-6) [19](#page-7-7)[–21](#page-7-8)]. Wang et al. adopted hydrothermal synthesis to prepare rigid three-dimensional $Ni₃S₄$ nanosheet frames with a specific capacitance of 1213 F/g and capacitance retention around 60% after 2000 cycles [\[22](#page-7-9)]. Zhang et al. have prepared hierarchical structure $Ni₃S₂$ (710.4 F/g and 84% cycling retention after 2000 cycles) [\[23](#page-7-10)]. Yang et al. have synthesized a hierarchical flower-like β-NiS electrode, which show high perfor-mance 857.76 F/g and 44% retention after 1000 cycles [\[24](#page-7-11)]. Although these nickel sulfides show very high specific capacitance, the cycle stability is poor.

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In this work, we synthesized a novel $\text{Ni}_3\text{S}_4@\text{MoS}_2$ nanocomposite grown on carbon fiber paper (CFP) with excellent electric conductivity and larger specific surface via facile one-step hydrothermal technique. The Ni₃S₄^{*@*} $MoS₂/CFP$ electrode shows a high specific performance of 1296 F/g at 1 A/g in 2 M KOH solution. In addition, the $Ni₃S₄@MoS₂/CFP$ electrode showed excellent cyclic stability (retention 96.2% after 5000 cycles at 5 A/g) and high rate capability. When the current density increase from 1 to 10 A/g, the specific capacitance retention of $\text{Ni}_3\text{S}_4@\text{MoS}_2/$ CFP can maintain 57.8%.

2 Experimental section

2.1 Material

All the reagents used in this experiment were of analytical grade and used without purification. Thiourea (NH_2CSNH_2) , Sodium molybdatedihydrate
(Na₂MoO₄·2H₂O), nickel(II) chloride hexahydrate $(Na_2MO_4.2H_2O),$ (NiCl₂·6H₂O) and Anhydrous ethanol (C₂H₆O) were purchased from Aladdin Industrial Corporation. Carbon fiber paper (CFP) (product of Toray, Japan; Teflon treated: 5% wt. wet proofing; thickness: 0.11 mm), used as working electrode, was obtained from Fuel Cell Store, USA. The water was purified through a Lab Pure Water System $(18.25 \text{ m}\Omega)$.

2.2 Synthesis of Ni3S4@MoS2/CFP

Before the synthesis, the CFP was hydrothermally pretreated with 0.1 M nitric acid (HNO₃) in 9 $\rm{°C}$ for 5 h and subsequently rinsed with deionized water and absolute ethanol. Finally, the CFP was dried at 70°C in vacuum for 6 h then the mass of CFP was measured by using balance. Typically, the $Ni₃S₄@MoS₂/CFP$ nanocomposite was prepared through a facile solvothermal method. Initially, 1.25 mmol $Ni(Cl)_2·6H_2O$, 2.5 mmol $Na_2MoO_4·2H_2O$ and 6.25 mmol $NH₂CSNH₂$ were dissolved in a 30 ml mixed solution of DI water and anhydrous ethanol (1:1 in volume).Then, two pieces of carbon fiber paper $(1 \times 2 \text{ cm}^2)$ vertically insert into the Teflon holder were subsequently soaked in the solution, and the precursor solution was magnetically stirred by ultrasonication for 0.5 h following by heated the solution to 180 °C in an electric oven, and kept at that temperature for 48 h. After natural cooling to room temperature, the as synthesized electrodes were then taken out, ultrasonically cleaned for 5 min in the DI water and rinsed with ethanol several times, dried in vacuum at 70°C over night. The mass loading of $Ni₃S₄@MoS₂/CFP$ nanosheets on carbon fiber paper was 0.5 mg/cm^2 . For properties comparison, pristine $MoS₂/CFP$ and $Ni₃S₄/CFP$ were prepared to follow

the similar procedure in the absence of $Ni(Cl)_{2}·6H_{2}O$ and $Na₂MoO₄·2H₂O$, respectively.

2.3 Characterization

The MoS_2/CFP , Ni_3S_4/CFP and $Ni_3S_4@MoS_2/CFP$ sample were characterized for structural properties using X-ray diffraction (D8 Advance, Bruker, Germany) with Cu-Kα radiation from 10°–80°with scan rate of 8°/min. The morphology and structure of the products were observed by field emission SEM (Hitachchi SU-8000) equipped with highresolution TEM (HRTEM, Tecnai G2F20).

2.4 Electrochemical measurements

Electrochemical measurements were performed on a CHI660D (Chenhua, shanghai, China) electrochemical working station in 2 M KOH aqueous electrolyte solution under three electrode systems consisted of a working electrode, a platinum plate counter electrode, and a saturated calomel electrode (SCE) reference electrode at room temperature. Cyclic voltammetry (CV), galvanostatic charge–discharge curves, and electrochemical impedance spectroscopy (EIS) were conducted in 2 mol/L KOH electrolyte with typical three-electrode cells (Pt as the counter electrode, Ag/AgCl as the reference electrode, and Ni_3S_4 @ $MoS₂/CFP$ as the working electrode). CV curves were collected over a voltage range from 0 to 0.5 V, and galvanostatic charge–discharge curves were measured over a voltage range from 0 to 0.44 V. The specific capacitance can be calculated from the galvanostatic charge–discharge curve according to Eq. ([1\)](#page-1-0)

$$
C = \frac{I\Delta t}{m\Delta v} \tag{1}
$$

where C is the specific capacitance, I is the current, Δt is the discharge time, Δv is the potential window, and m is mass of the electroactive material. EIS was recorded in the frequency range from 0.01 to 100 kHz at an open-circuit potential.

3 Results and discussion

The $Ni₃S₄@MoS₂/CFP$ nanostructure was successfully prepared through a facile and low temperature hydrothermal method. The fabrication of $Ni₃S₄@MoS₂/CFP$ nanostructures is schematically depicted in Fig. [1](#page-2-0). During the reaction, the chemical reaction occurred at the surface of the cleaned carbon fiber pager, and the active species (S ions) will be released from $NH₂CSNH₂$ according Eq. ([2\)](#page-2-1). At last the S ions with MoO_4^{2-} and Ni^{2+} ions to produce $\text{Ni}_3\text{S}_4\text{\textcircled{e}}$ MoS_{2} .

Fig. 1 Schematic the reaction process of the $\text{Ni}_3\text{S}_4@\text{MoS}_2/\text{CFP}$

Fig. 2 XRD patterns of $Ni₃S₄@MoS₂/CFP$, $MoS₂/CFP$ and $Ni₃S₄/$ CFP, respectively

$$
NH_2CNH_2 + H_2O \to 2NH_3 + 2H_2S + CO_2 \tag{2}
$$

The composition and purity of the as-obtained samples are analyzed by XRD technique. Figure [2](#page-2-2) displays the XRD pattern of MoS_2 , $Ni₃S₄$ and $Ni₃S₄@MoS₂$ loaded on the CFP, respectively. The typical diffraction peak around 26.6° of all pattern arise from the CFP substrate (JCPDS card no. 26-1080). From the XRD pattern of $\text{Ni}_3\text{S}_4@\text{MoS}_2/$ CFP, the diffraction peaks located at $2\theta = 26.5^{\circ}$, 31.8°, 47.3ºand 54.8° can be perfectly indexed to (022), (113), (224) and (044) planes of $Ni₃S₄$ (JCPDS card no. 43-1469), which is agreed with those of $Ni₃S₄/CFP$. For $MoS₂$, the curve of $Ni₃S₄@MoS₂$ shows serious sharp peaks at 29.2°, 32.6°, 35.8°and 39.5°which could be ascribed to the (004), (100) , (102) and (103) phase of hexagonal MoS₂ (JCPDS) 37-1942) in accordance with MoS2/CFP, respectively. In addition, no obvious peak of (002) diffraction peak in the $MoS₂/CFP, Ni₃S₄@MoS₂/CFP are attributed to the low$ crystal of $MoS₂$ [[25\]](#page-7-18).

The morphological studies of the MoS₂/CFP, Ni₃S₄/ CFP and $Ni₃S₄@MoS₂/CFP$ were carried out by electron microscopy techniques. Figure [3](#page-3-0) displays the surface morphology after the MoS₂/CFP, Ni₃S₄/CFP andNi₃S₄[@] $MoS₂composite deposition on CFP. Figure 3b shows$ $MoS₂composite deposition on CFP. Figure 3b shows$ $MoS₂composite deposition on CFP. Figure 3b shows$ that $MoS₂$ flower-like sheets uniformly covered on the surface of the CFP. As shown in Fig. [3c](#page-3-0), the pristine $Ni₃S₄$ nanospherescompactlyaggregated on surface of CFP with size of 100–300 nm. However, it can be observed that numerous $Ni₃S₄$ nanoparticles are homogeneously distributed on the surface of $MoS₂$ the Fig. [3d](#page-3-0). In addition, Fig. [3](#page-3-0)e, f indicate that the morphology of $Ni₃S₄ changes from compact sphere to cluster. During the$ reaction, the $MoS₂$ layers have severely affected nuclei formation and crystal growth of $Ni₃S₄$ nanoparticles. The highly magnified SEM image in Fig. [3](#page-3-0)f shows that the $Ni₃S₄@MoS₂$ is composed of cluster with the size of 100–150 nm by piling up intertwined sheet-like structure. The flower-like sheets $MoS₂$ might provide more efficient contacts between active material and electrolyte. But the poor conductivity of $MoS₂$ might limit charge transition. In this case, the conductive $Ni₃S₄$ nanoparticles could efficiently lower the internal resistance and provide perfect continuous channels for electron transportation.

More details of the morphological and structural features of as-obtained $Ni₃S₄@MoS₂$ separated from the CFP by ulrasonication are studied by TEM technique. Figure [4](#page-4-0) shows the TEM images of the synthesized $Ni₃S₄@MoS₂$ nanocomposite which is composed of many nanosheets. The result is consistent with what has been indicated by the SEM pattern in Fig. [3](#page-3-0)d–f. As shown in Fig. [4a](#page-4-0), b, the $Ni₃S₄@MoS₂$ display a morphology of folded and tangled layers with some folded edges exhibiting parallel lines corresponding to the different layers of $MoS₂$ sheets. In Fig. [4](#page-4-0)c, d, the high-resolution TEM (HRTEM) images of $Ni₃S₄@MoS₂$ indicate that the interlayer spacing of $Ni₃S₄$ sheets is 0.29 nm and interplanar spacing of the $MoS₂$ in the composite is 0.62 nm, which

Fig. 3 SEM images of **a** pure CFP, **b** MoS₂/CFP, **c** Ni₃S₄/CFP, **d**–**f** Ni₃S₄@MoS₂/CFP

match well with the (113) plane of $Ni₃S₄$ and (002) plane of $MoS₂$, respectively [[22,](#page-7-9) [25\]](#page-7-18).

The electrochemical performances of the as-prepared $MoS₂/CFP, Ni₃S₄/CFP$ and $Ni₃S₄@MoS₂/CFP$ nanocomposite were carried out in a three electrode system in 2 M KOH aqueous solution as shown in Fig. [5](#page-5-0)a. Clearly, $Ni₃S₄@MoS₂/CFP$ has larger area CV curves than that of the MoS_2/CFP and $Ni₃S₄/CFP$, which suggests the higher capacitive capability of the $Ni₃S₄@MoS₂/CFP$. In addition, the peaks at 0.35 and 0.21 V of $Ni₃S₄@MoS₂/CFP$ are too strong compared with bare $MoS₂/CFP$. This is the main reason why the $Ni₃S₄@MoS₂/CFP$ nanocomposite only exhibits one pair of redox peaks. The appearance of a pair of remarkable redox peaks indicates the nanosheets of $Ni₃S₄@$ $MoS₂/CFP$ offering more active sites for redox charge transfer. For the curves of $Ni₃S₄@MoS₂/CFP$ composite, the observed characteristic oxidation and reduction position (oxidation peak around +0.35 V and reduction peak around +0.21 V) exhibited in the CV curves are related to faradic redox reactions reacting at the $Ni₃S₄@MoS₂/CFP$ surface tively

in alkaline KOH electrolyte [[17,](#page-7-4) [26–](#page-7-19)[28\]](#page-7-20). The main reaction mechanism can be described as follows,

 $NiS_x + 4OH^- - 2e^- \rightarrow NiO_2 + S_x^{2-} + 2H_2O$ $NiO_2 + H_2O + e^- \leftrightarrow NiOOH + OH^ NiOOH + H₂O + e^- \leftrightarrow Ni(OH)₂ + OH^ MoS₂ + K⁺ + e⁻ \leftrightarrow MoS - SK⁺$

Figure [5](#page-5-0)b illustrates the charge–discharge curves ofMoS₂/CFP, Ni₃S₄/CFP and Ni₃S₄@MoS₂/CFP in the potential range of 0–0.44 V at a current density of 1 A/g $Ni₃S₄@MoS₂/CFP$ nanocomposite shows the best charge storage properties, compared to $MoS₂/CFP$ and $Ni₃S₄/CFP$. The specific capacitances of the prepared $\text{Ni}_3\text{S}_4\text{@MoS}_2/$ CFP were calculated to be 1296 F/g lager than those of $MoS₂/CFP (150 F/g)$ and $Ni₃S₄/CFP (364 F/g)$ at 1 A/g. These results are in good accordance with the CV curves. Figure [5c](#page-5-0) shows the CV curves of the $Ni₃S₄@MoS₂/CFP$ composite at a san rate of 5–30 mV/s in the potential window of 0.0–0.5 V versus SCE in 2 M KOH. The CV curves of three electrodes show an obvious redox but rectangular shape, which indicate that electrodes at certain scan rate possess a typical redox capacitive mechanisms pseudocapacitance. It is obvious that the integrated area gradually increased and the shape of CV curves remained similar, implying good rate capability.Moreover, the current density increases and the oxidation peak shifts to a more positive position, while the reduction peak shifts to a more negative position. This is due to the increased internal resistance within the pseudoactive material with the increase in scan rate.For comparison, the CV of the pristine CFP and $Ni₃S₄@MoS₂/CFP at 5 mV/s are also shown in Fig. 5d. It is$ $Ni₃S₄@MoS₂/CFP at 5 mV/s are also shown in Fig. 5d. It is$ $Ni₃S₄@MoS₂/CFP at 5 mV/s are also shown in Fig. 5d. It is$ noted that the CFP is of almost no electrochemical performance. The supercapacitive characteristic can also be measured by constant current charge–discharge cycling test. Figure [5](#page-5-0)c shows the galvanostatic charge–discharge curves of atNi₃S₄@MoS₂/CFP at current densities of 1–10 A/g with a platform during process of charge and discharge, respectively, corresponding to a pair of redox peaks in CV curves. The specific capacitance can be calculated by Eq. ([1\)](#page-1-0). The calculated results indicate the specific capacitances of the material are 1296, 936, 893, 863, 829, 763 and 750 F/g at

Fig. 5 a Cyclic voltammograms curves of $MoS₂/CFP$, $Ni₃S₄/CFP$ and $\text{Ni}_3\text{S}_4\text{@MoS}_2/\text{CFP}$ at scan rate of 5 mV/s, respectively; **b** galvanostatic charging/discharging curves of $MoS₂/CFP$, $Ni₃S₄/CFP$ and Ni₃S₄@MoS₂/CFP at current density of 1 A/g, respectively; **c** cyclic voltammograms curves of $Ni₃S₄@MoS₂/CFP$ at different scan rate in

2 M KOH; **d** CV curves of Ni₃S₄@MoS₂/CFP and CFPat scan rate of 5 mV/s, respectively; **e** galvanostatic charging/discharging curves at a serious current densities for $Ni₃S₄@MoS₂/CFP$ in 2 M KOH; **f** rate capability curves of $Ni₃S₄@MoS₂/CFP$ at various current density

current densities of 1, 2, 3, 4, 5, 8 and 10 A/g respectively, which is better than that of pristine $Ni₃S₄$ reported elsewhere [\[22](#page-7-9)]. The large specific capacitance can be attributed to unique sheet structure. Furthermore, Fig. [5d](#page-5-0) exhibits the $\text{Ni}_3\text{S}_4@ \text{MoS}_2/\text{CFP}$ electrodes as a function of current density. It indicates that the specific capacitance decreases with the increase of current density. However, the specific capacitance retention of $Ni₃S₄@MoS₂/CFP$ can maintain

57.8% when the current density increase from 1 to 10 A/g, which suggests that efficient surface redox reaction take place on the $Ni₃S₄@MoS₂/CFP.$ The results indicate that the $\text{Ni}_3\text{S}_4@ \text{MoS}_2/\text{CFP}$ composite could be recognized as one of the important potential electrode material in application of electrodes and batteries.

The cycle stability of $\text{Ni}_3\text{S}_4@ \text{MoS}_2/\text{CFP}$ composite was evaluated by repeating the constant current charge/discharge test in a potential window of 0–0.44 V (vs. SCE) at a current density 5 A/g for 5000 cycles. Figure [6](#page-6-0)a shows that the specific capacitance has been increased during the first 1000 cycles, mainly due to the electrode material activated during the continuing charge–discharge process. Finally, the capacitance retention of $\text{Ni}_3\text{S}_4@\text{MoS}_2/$ CFP in 2 M KOH electrolytes remains around 96.2% after 5000 cycles. EIS is one of the prominent tools to understand the charge-transfer behavior and capacitive nature of an electrode–electrolyte interface system with respect to the applied frequency [[29\]](#page-7-21). Figure [6](#page-6-0)b shows Nyquist plots for the $Ni₃S₄@MoS₂/CFP$ electrode with a parallel model before and after 5000 cycles. In the high-frequency region, the first intercept of the semicircle with the real axis corresponds to the intrinsic ohmic resistance of an equivalent series resistance (Rs) of the electrolyte and electrode. Obviously, the Rs of $Ni₃S₄@MoS₂/CFP$ electrodes was measured to be 0.7718 and 0.611 Ω , respectively. Then, the semicircle in the high-medium frequency region is attributed to the charge-transfer impedance (Rct) at the electrode and electrolyte interface. The value of Rct is measured as semicircular arc diameter .Therefore, the Rct of $Ni₃S₄@$ MoS₂/CFP electrodes increased from 2.32 to 3.88 Ω after the 5000 cycles. In the low frequency region, the straight line in the low frequency range is related to the diffusive

resistance of the electrolyte into interior of the electrode and the ion diffusion in to the electrode, suggesting the ideal capacitive behavior of the $\text{Ni}_3\text{S}_4@ \text{MoS}_2/\text{CFP}$. The high performance of the hybrid composite electrodes is attributed to the following some advantages of the electrode. (1) Carbon fiber paper (CFP), a network of microsized carbon fibers, has many excellent characteristics such as large surface area, high porosity, good electric conductivity, and excellent chemical stability in a wide variety of liquid electrolytes; (2) $Ni₃S₄@MoS₂/CFP$ accumulate to form pores for ion-buffering reservoirs to improve the diffusion rate of ions; (3) the conductive $Ni₃S₄$ could efficiently lower the internal resistance; (4) the sheeted $\text{Ni}_3\text{S}_4\text{@MoS}_2$ / CFP offer large surface area with short electrons and ions diffusion path and the unique structure of $\text{Ni}_3\text{S}_4@\text{MoS}_2$ increase the amount of electroactive sites for the effective utilization of $Ni₃S₄@MoS₂$ active material.

4 Conclusion

In summary, $Ni₃S₄@MoS₂$ nanosheets grown on the carbon fiber paper $(Ni_3S_4@MoS_2/CFP)$ composites were successfully prepared via a facile one-step hydrothermal technique. The specific capacitance of $Ni₃S₄@MoS₂/CFP$ nanocomposite has been measured to be 1296 F/g at a discharge density of 1 A/g. the specific capacitance retention of $Ni₃S₄@$ $MoS₂/CFP$ can maintain 57.8% with the current densities from 1 to 10 A/g. Furthermore, the $\text{Ni}_3\text{S}_4\text{@MoS}_2/\text{CFP}$ nanocomposite displays an excellent long-time cycling stability, retaining 96.2% of the initial capacitance after 5000 cycles at current density of 5 A/g. These excellent results indicate that the $Ni₃S₄@MoS₂/CFP$ nanocomposite is a

Fig. 6 a Cycling performance of Ni₃S₄@MoS₂/CFP at the current density of 5 A/g; **b** Nyquist plots of Ni₃S₄@MoS₂/CFP electrode in 2 M KOH solution in the frequency range from 0.01 to 100,000 Hz

promising and suitable electrode material for high-performance supercapacitors.

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References

- 1. P. Simon, Y. Gogotsi, Nat. Mater. **7**, 845 (2008)
- 2. M.D. Stoller, S. Park, Y. Zhu, J. An, R.S. Ruffo, Nano Lett. **8**, 3498 (2008)
- 3. L.L. Zhang, X.S. Zhao, Chem. Soc. Rev. **38**, 2520 (2009)
- 4. E. Frackowiak, Chem. Soc. Rev. **9**, 1774 (2007)
- 5. E. Frackowiak, K. Metenier, V. Bertagna, F. Beguin, Appl. Phys. Lett. **77**, 2421 (2000)
- 6. Q. Xiao, X. Zhou, Electrochim. Acta **48**, 575 (2003)
- 7. C.X. Guo, C.M. Li, Energy Environ. Sci. **4**, 4504 (2011)
- 8. Z. Huang, Z. Zhang, X. Qi, X. Ren, G. Xu, P. Wan, Nanoscale **8**, 13273 (2016)
- 9. X. Xia, J. Tu, Y. Zhang, X. Wang, C. Gu, X. Zhao, ACS Nano **6**, 5531 (2012)
- 10. M. Mastragostino, C. Arbizzani, F. Soavi, Solid State Ionics **148**, 493 (2002)
- 11. Y.T. Weng, N.L. Wu, J. Power Sources **238**, 69 (2013)
- 12. Z. Zhang, X. Liu, X. Qi, Z. Huang, L. Ren, J. Zhong, RSC Adv. **4**, 37278 (2014)
- 13. X. Rui, H. Tan, Q. Yan, Nanoscale **6**, 9889 (2014)
- 14. Z. Xing, Q. Chu, X. Ren, J. Tian, A.M. Asiri, K.A. Alamry, Electrochem. Commun. **32**, 9889 (2013)
- 15. C. Largeot, C. Portet, J. Chmiola, J. Am. Chem. Soc. **130**, 2730 (2008)
- 16. L. Cao, S. Yang, W. Gao, Z. Liu, Y. Gong, L. Ma, Small **9**, 2905 (2013)
- 17. A. Ramadoss, T. Kim, G.S. Kim, J.K. Sang, New. J. Chem. **38**, 2379 (2014)
- 18. L. Wang, M. Ying, Y. Min, Y. Qi, Electrochim. Acta **186**, 391 (2015)
- 19. X. Liu, X. Qi, Z. Zhang, L. Ren, Y. Liu, L. Meng, Ceram. Int. **40**, 8189 (2014)
- 20. S. Shen, Q. Wang, Cheminform **25**, 1166 (2012)
- 21. Z. Xing, Q. Chu, X. Ren, A.M. Asiri, K.A. Alamry, Electrochem. Commun. **32**, 9 (2013)
- 22. L. Wang, J. Liu, L.L. Zhang, B. Dai, M. Xu, M. Ji, RSC Adv. **5**, 8422 (2015)
- 23. Z. Zhang, Z. Huang, L. Ren, Y. Shen, Q.I. X., J. Zhong, Electrochim. Acta **149**, 316 (2014)
- 24. J. Yang, X. Duan, Q. Qin, W. Zheng, J. Mater. Chem. A **1**, 7880 (2013)
- 25. K.J. Huang, L. Wang, Y.J. Liu, Y.M. Liu, H.B. Wang, T. Gan, Int. J. Hydrog. Energy **38**, 14027 (2013)
- 26. L. Hou, C. Yuan, D. Li, L. Yang, L. Shen, F. Zhang, Electrochim. Acta **56**, 7454 (2011)
- 27. W. Wei, L. Mi, Y. Gao, Z. Zheng, W. Chen, X. Guan, Chem. Mater. **26**, 3418 (2014)
- 28. C. Yuan, B. Gao, L. Su, L. Chen, X. Zhang, J. Electrochem. Soc. **156**, A199 (2009)
- 29. Z. Zhang, Y. Liu, Z. Huang, L. Ren, X. Qi, X. Wei, Phys. Chem. Chem. Phys. **17**, 20795 (2015)