



Developing porous organic polymers as precursors of nitrogen-decorated micro-mesoporous carbons for efficient capture and conversion of carbon dioxide

Xiankun Wu^{1,3}, Rong Guan¹, Wen-Tao Zheng^{1,*}, Kuan Huang^{2,*} , and Fujian Liu^{4,*}

¹School of Chemistry and Environmental Engineering, Yancheng Teachers University, Yancheng 224007, Jiangsu, China

²School of Resources Environmental and Chemical Engineering, Nanchang University, Nanchang 330031, Jiangxi, China

³Jiangsu Nanda Huaxing Science and Technology of Environmental Protection Co. Ltd, Yancheng 224001, Jiangsu, China

⁴National Engineering Research Center of Chemical Fertilizer Catalyst, Fuzhou University, Fuzhou 530002, Fujian, China

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ABSTRACT

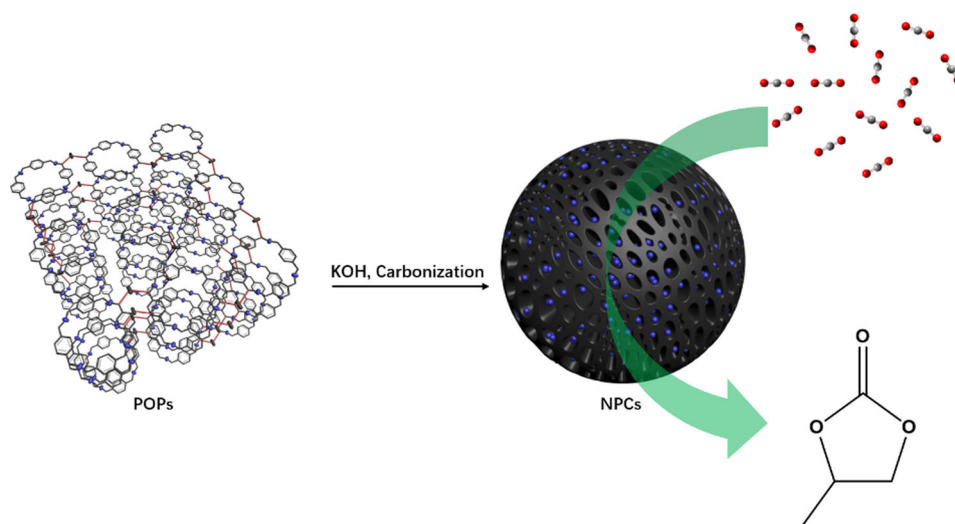
Porous carbons with large surface areas, abundant mesopores and weak base sites are promising materials for the capture and conversion of CO₂. However, it is still challenging to obtain such porous carbons in a facile and template-free way. Herein, nitrogen-decorated micro-mesoporous carbons were synthesized by direct carbonization of porous organic polymers, which were developed through alkylation-induced hyper-crosslink of rigid organic bases without the use of any templates. The synthesized carbons have ultrahigh surface areas of 2366–3580 m²/g, total pore volumes of 1.74–3.38 cm³/g and N contents of 1.50–3.24 wt%. As a consequence, the synthesized carbons enable highly efficient and selective adsorption of CO₂ from CO₂/N₂ mixed gas, with the CO₂ capacities of 1.50–2.03 mmol/g at 0 °C and 15 kPa, and IAST selectivities of 69–78 for CO₂/N₂ (0.15/0.85 vol) mixed gas at 0 °C and 100 kPa. After loaded with metal salts, the synthesized carbons also exhibit high activities for the catalytic conversion of CO₂, with the TOFs of > 600 h⁻¹ for cycloaddition of CO₂ with propylene oxide at 100 °C and 1.0 MPa.

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Address correspondence to E-mail: wentaozheng1990@163.com; huangk@fzu.edu.cn; fjliu@fzu.edu.cn

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GRAPHICAL ABSTRACT



Introduction

Carbon dioxide (CO₂) is a greenhouse gas and mainly emitted in the tail gas of thermal power plants, which consume large quantity of coals. The popularization of automobiles, which achieve energy through burning gasoline, also makes a great contribution to the emission of CO₂. According to the report of Intergovernmental Panel on Climate Change (IPCC), the atmospheric concentration of CO₂ has increased to a high level in comparison with a hundred years ago [1]. As a consequence, the global warming issue is more and more serious, thus inducing the continuous melting of glaciers and desertification of lands. Given the urgent situation, a consensus has been reached across the world that the emission of CO₂ must be significantly reduced, so as to preserve the living environment of human beings. [2] Traditionally, the reducing of CO₂ emission is realized by scrubbing with aqueous alkanolamine solutions [3, 4]. However, aqueous alkanolamine solutions are associated with some defects such as high volatility, large heat capacity and strong corrosion. Recently, ionic liquids (ILs) with extremely low volatility and structural designability were proposed as alternatives to aqueous alkanolamines for application in the reducing of CO₂ emission [5–7]. Unfortunately, the

high cost and large viscosity of ILs disfavor the practical use of ILs in the industry.

On the other hand, CO₂ is a widely used C1 feedstock for the synthesis of value-added chemicals such as cyclic carbonates, carboxylic acids, urea-derivates, formamides, etc. [8–12]. Therefore, it is of great significance to capture CO₂ from industrial tail gas and simultaneously convert it into value-added chemicals. Many advanced materials have been investigated for the capture and conversion of CO₂, such as zeolites [13, 14], porous carbons [15–17], porous organic polymers (POPs) [18–20] and metal–organic frameworks (MOFs) [21–23]. Among these materials, porous carbons are particularly attractive, because they have intriguing features such as high thermal stability, tunable porous and chemical structure. The precursors for porous carbons are also widely available, including natural compounds [24–27], organic wastes [28–30] and synthetic polymers [31–33]. Ideal porous carbons for CO₂ capture and conversion should have large surface areas, abundant mesopores and weak base sites. The large surface areas are favorable for the exposure of materials to reactants. The abundant mesopores are favorable for the diffusion of reactants in nanochannels. The weak base sites can not only selectively attract CO₂ from gas

phase through Lewis base–acid interaction, but also anchor metal ions that have catalytic activity for CO₂ conversion.

Porous carbons with large surface areas are normally synthesized by chemical or physical activation of pre-carbonized solids [34–40]. The most frequently used activation agent is KOH, while others include NaNH₂, K₂CO₃, ZnCl₂, NH₃, H₂O and CO₂. However, the activation process mainly results in micropores that are narrow in pore width. To construct mesopores, hard or soft templates should be employed in the preparation of carbon precursors [41–45]. The hard templates (e.g., SiO₂) are removed by HF etching after carbonization process, while the soft templates (e.g., surfactants and block copolymers) are removed through thermal decomposition during carbonization process. The introduction of weak base sites can be realized by choosing precursors with weak base groups for carbonization [46, 47], or treating porous carbons with weak base compounds [48, 49]. Though with the progress, it is still challenging to synthesize porous carbons with large surface areas, abundant mesopores and weak base sites in a facile and template-free way.

The choice of precursors has great impact on the porosity and function of resultant carbons, thereby influencing their performance for CO₂ capture and conversion. In this work, we developed a class of porous organic polymers (POPs) with abundant mesopores and weak base sites by alkylation-induced hyper-crosslink of rigid organic bases, without the use of any templates. It is then envisioned that the direct carbonization of such POPs may result in porous carbons with abundant mesopores and weak base sites: (1) the permanent mesopores formed by crosslinking structure are likely to be preserved during carbonization process; (2) the weak base sites

can be *in situ* decorated into resultant porous carbons. To increase the surface areas of resultant porous carbons and simplify the synthetic process, POPs were mixed with KOH for direct carbonization. Scheme 1 illustrates the synthetic route for porous carbons with large surface areas, abundant mesopores and weak base sites. The porous carbons were systematically characterized for porous and chemical structure, and also examined for CO₂ capture and conversion performance.

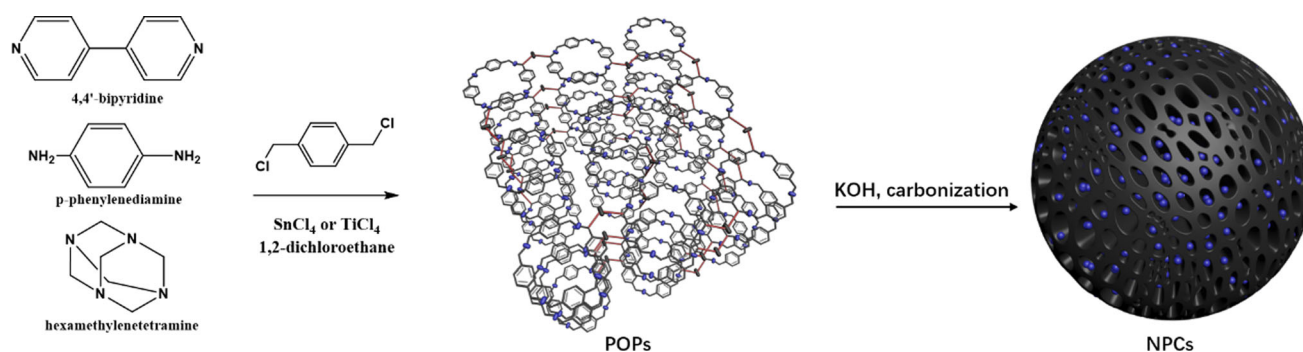
Experimental

Chemicals

CO₂ (99.99 vol%) and N₂ (99.99 vol%) were supplied by Jiangxi Huasheng Co. Ltd., China. 1,4-Bis(chloromethyl)benzene (98 wt%) and *p*-phenylenediamine (99 wt%) were supplied by Shanghai Sigma Aldrich Co. Ltd., China. 1,2-Dichloroethane (99.5 wt%), SnCl₄ (99 wt%), TiCl₄ (99 wt%), 4,4'-bipyridine (98 wt%), hexamethylenetetramine (HMTA, 99 wt%), ethanol (99.7 wt%), iso-propylamine (99 wt%), KOH (90 wt%), CoCl₂·6H₂O (99 wt%), Zn(OAc)₂·2H₂O (99 wt%), *N,N*-dimethylformamide (DMF, 99 wt%), propylene oxide (99 wt%), tetrabutylammonium bromide (TBAB, 99 wt%) and methanol (99.9 wt%) were supplied by Shanghai Adamas Co. Ltd., China. All the chemicals were used as received.

Synthesis of POPs

POPs were synthesized by alkylation-induced hyper-crosslink of rigid organic bases according to the literature [50]. In brief, 1,4-bis(chloromethyl)benzene (2.0 g) was dissolved in 1,2-dichloroethane (30 mL),



Scheme 1. Synthetic route for porous carbons with large surface areas, abundant mesopores and weak base sites.

and then, SnCl_4 or TiCl_4 (6.0 mL) was added to the mixture under vigorous stirring at 0 °C. Subsequently, 4,4'-bipyridine, *p*-phenylenediamine or HMTA (1.6 g) was added to the mixture under static condition. The reaction was performed at 75 °C for 24 h under the protection of flowing N_2 . The formed solid was filtered out and washed with warm ethanol to remove residual reactants. The product was activated with iso-propylamine, followed by drying at 60 °C and 0.1 kPa for 24 h. The synthesized samples were denoted as POP-*x*, where *x* is the rigid organic base.

Synthesis of porous carbons

Porous carbons were synthesized by direct carbonization of POP-*x* samples. In brief, POP-*x* (1.0 g) was mixed with KOH (4.0 g) by manual grinding. The mixture was then loaded in a nickel crucible, transferred to a tube furnace, heated to 800 °C at a rate of 5 °C/min and kept at 800 °C for 2 h. The whole carbonization process was protected by flowing N_2 . After cooling down to the room temperature naturally, the product was washed with deionized water until the pH of water approaches 7. The synthesized samples were denoted as NPC-*x*, where NPC stands for nitrogen-decorated porous carbon and *x* is the rigid organic base.

Synthesis of metal-loaded catalysts

Metal-loaded catalysts were synthesized by impregnation of metal salts into the nanopores of porous carbons. In brief, $\text{CoCl}_2 \cdot 6\text{H}_2\text{O}$ (0.41 g) or $\text{Zn}(\text{OAc})_2 \cdot 2\text{H}_2\text{O}$ (0.34 g) was dissolved in DMF (75 mL), and then, NPC-*x* (0.40 g) was added to the mixture. The mixture was vigorously stirred at 120 °C for 12 h under the protection of flowing N_2 . The solid product was filtered out, washed with warm DMF and finally dried at 60 °C and 0.1 kPa for 24 h. The synthesized samples were denoted as M@NPC-*x*, where M is the metal and *x* is the rigid organic base.

Characterizations

Before characterizations, each sample was degassed at 160 °C and 0.1 Pa for 6 h. N_2 adsorption isotherms at -196 °C were determined by a Micromeritics Gemini 2390a surface area analyzer to calculate the pore width distributions and porosity parameters.

Pore width distributions were calculated using the Barrett–Joyner–Halenda (BJH) method. Surface areas were calculated using the Brunauer–Emmett–Teller (BET) equation in the relative pressure range of 0.02–0.15. Total pore volumes were calculated according to the amounts of N_2 adsorbed at the relative pressure of 0.975. Micropore volumes were calculated using the *t*-plot method in the thickness range of 0.45–0.6 nm. Mesopore volumes were calculated by integrating the pore size distributions in the pore size range of 2–50 nm.

Thermogravimetric analysis (TGA) was performed on a Seiko 6300 TG/DTA system under the protection of flowing N_2 , with the heating rate set to 10 °C/min. Elemental analysis was performed on an Elementar Vario EL II system. Scanning electron microscope (SEM) images were taken on a Zeiss Auriga SEM/FIB Crossbeam system at an acceleration voltage of 5 kV. Transmission electron microscope (TEM) images were taken on a Zeiss Libra 200 FE system at an acceleration voltage of 200 kV. X-ray diffraction (XRD) patterns were collected on a PANalytical X'Pert Pro X-ray powder diffractometer with $\text{Cu-K}\alpha$ radiation. The scanning range was set to $2\theta = 15^\circ$ – 75° , and the step size was set to 0.02° . Raman spectra were collected on a Princeton Acton Trivista 555 spectrometer with $\lambda = 532$ nm laser excitation. X-ray photoelectron spectroscopy (XPS) spectra was collected on a Thermo Scientific ESCALAB250 X-ray photoelectron spectrometer with $\text{Al K}\alpha$ radiation, and the binding energies were calibrated using the C 1s peak at 284.9 eV. Inductively coupled plasma (ICP) analysis was performed on a Jarrel-AshJ-A1100 spectrometer.

Single gas adsorption

Before experiments, each sample was degassed at 160 °C and 0.1 Pa for 6 h. CO_2 and N_2 adsorption isotherms at 0 and 25 °C were determined by a Micromeritics Tristar II 3020 surface area analyzer.

Mixed gas adsorption

Mixed gas adsorption experiments were performed on a home-made fixed-bed adsorption system. In a typical run, a certain amount of NPC-*x* was loaded in a quartz tube, and then, the sample was stabilized at target temperature for 1 h under the protection of flowing Ar. Subsequently, the Ar flow was switched

to a CO₂/N₂ mixed gas flow. The composition of inlet gas was adjusted by two Horiba Metron S500 mass flow controllers. The compositions of outlet gas were recorded online by an Agilent 7890B gas chromatograph.

Catalytic CO₂ conversion

Catalytic CO₂ conversion experiments were performed in a stainless steel autoclave with a Teflon tube (25 mL). The autoclave was first purged with CO₂ to replace air and immersed in ice/water bath for 10 min. Then, specific amounts of M@NPC-*x*, propylene oxide and TBAB were loaded into the autoclave quickly. Subsequently, the autoclave was sealed and charged with CO₂ to target pressure. After stirring at target temperature for a specific period of time, the excess CO₂ was vented out and the autoclave was immersed in ice/water bath to halt the reaction. The liquid mixture was analyzed by an Agilent 7890A gas chromatography (GC) equipped with a flame ionization detector (FID), and the yields and selectivities were calculated using the internal standard method. Specifically, dodecane was used as the internal standard to calculate the correction factor of GC peaks. The amounts of reactant and product after reaction were then calculated by comparing the GC peak areas of reactant and product with that of dodecane. To examine the recyclability of catalysts, M@NPC-*x* was filtered out from reaction system, washed with methanol, dried at 60 °C and 0.1 kPa for 24 h and reused for catalytic CO₂ conversion experiments.

Results and discussion

Structural characterizations

The carbon precursors (i.e., POP-*x* samples) developed by alkylation-induced hyper-crosslink of rigid organic bases were first characterized by N₂ adsorption isotherms at –196 °C to calculate the pore width distributions and porosity parameters (see Fig. 1, Table 1). It can be seen that the N₂ adsorption isotherms display type II profiles, with considerable increase in N₂ uptakes at the relative pressure of > 0.9. The pore width distributions cover a wide range of 0–50 nm, suggesting that both micropores and mesopores are present in POP-*x* samples. POP-*x* samples have moderate surface areas of 209–596 m²/g and total pore volumes of 0.21–0.73 cm³/g, following the sequence of POP-4,4'-bipyridine > POP-*p*-phenylenediamine > POP-HMTA. The mesopore volumes account for > 65% of the total pore volumes for POP-*x* samples, suggesting the predominantly mesoporous structure of POP-*x* samples. In addition, the elemental analysis results show that POP-*x* samples have the N contents of 3.25–5.82 wt%, with the highest value for POP-HMTA (see Table 1). This is within expectation since HMTA has higher N content than the other two organic bases. Furthermore, the TGA curves show that POP-*x* samples start to decompose at ~175 °C, and the loss of weight gradually slows down at > 250 °C (see Figure S1).

The porous carbons (i.e., NPC-*x* samples) were then synthesized by direct carbonization of POP-*x* samples with the assistance of KOH at 800 °C. Actually, the effects of carbonization temperatures on the porosity parameters, N contents and CO₂ uptakes of resultant NPC-4,4'-bipyridine were examined (see Figure S2, Table S1 and Figure S3). It is found that

Figure 1 N₂ adsorption isotherms at –196 °C (a) and pore width distributions (b) of POP-*x* samples.

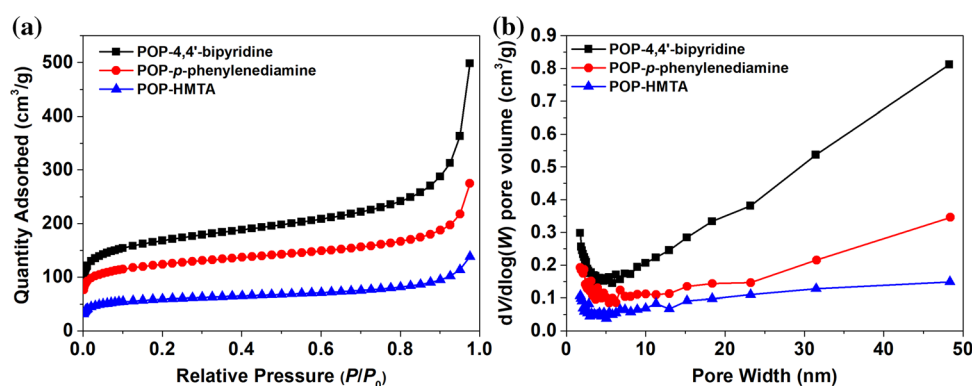


Table 1 Porosity parameters and N contents of POP-*x* samples

Sample	S_{BET} (m ² /g) ^a	V_{total} (cm ³ /g) ^b	V_{micro} (cm ³ /g) ^c	$V_{\text{micro}}/V_{\text{meso}}$ (%/%) ^d	N content (wt%) ^e
POP-4,4'-bipyridine	596	0.73	0.17	23/77	3.34
POP- <i>p</i> -phenylenediamine	437	0.41	0.14	34/66	3.25
POP-HMTA	209	0.21	0.07	33/67	5.82

^aBET surface area^bTotal pore volume^cMicropore volume^dMicro-/mesopore volume ratio^eDetermined by elemental analysis

when the carbonization temperatures increase from 500 to 800 °C, the porous parameters of resultant porous carbons increase, while the N contents of resultant porous carbons decrease. However, further increasing the carbonization temperatures from 800 to 900 °C results in no product. The CO₂ uptakes of NPC-4,4'-bipyridine synthesized at 800 °C are also higher than those of NPC-4,4'-bipyridine synthesized at other carbonization temperatures. Therefore, the carbonization temperature was set to 800 °C for all the carbon precursors in this work.

The synthesized NPC-*x* samples were systematically characterized for porous and chemical structure. Figure 2 shows the N₂ adsorption isotherms at −196 °C and pore width distributions of NPC-*x* samples, which display similar profiles as those of POP-*x* samples. Therefore, both micropores and mesopores are also present in NPC-*x* samples. The calculated porous parameters for NPC-*x* samples are summarized in Table 2. It is notably found that NPC-*x* samples have ultrahigh surface areas of 2366–3580 m²/g and total pore volumes of 1.74–3.38 cm³/g, following the sequence of NPC-4,4'-bipyridine > NPC-*p*-

phenylenediamine > NPC-HMTA. This is consistent with the sequence of surface areas and total pore volumes for corresponding POP-*x* samples. Obviously, the surface areas and total pore volumes of NPC-*x* samples are much higher than those of POP-*x* samples, because KOH etching creates additional nanochannels in NPC-*x* samples during the carbonization process. The mesopore volumes account for > 50% of the total pore volumes for NPC-*x* samples, still suggesting the predominantly mesoporous structure of NPC-*x* samples. However, NPC-*x* samples have slightly higher percentages of micropore volumes than POP-*x* samples, since the nanochannels created by KOH etching are mainly microporous.

Figures 3 and 4 show the SEM and TEM images of NPC-*x* samples, respectively. In the SEM images, rough surfaces embedded with sponge-like mesopores can be clearly observed. In the TEM images, micropores and mesopores with disordered arrangements can be clearly observed. The observations from SEM and TEM images indicate that NPC-*x* samples are amorphous, and have micro-mesoporous structure. The micro-mesoporous structure is

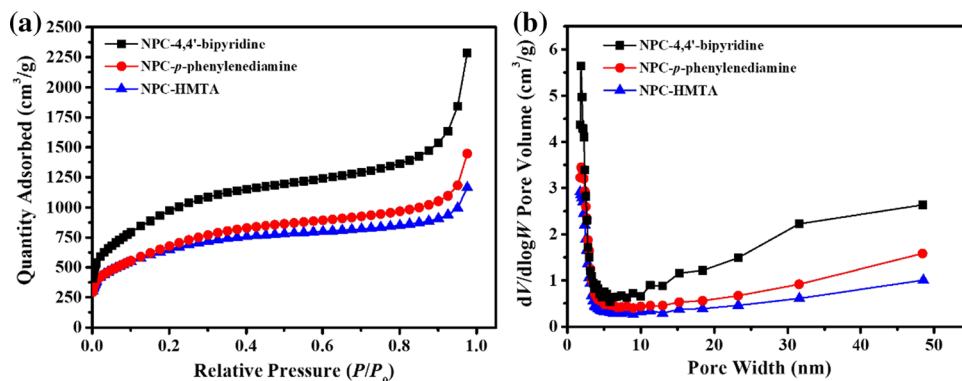
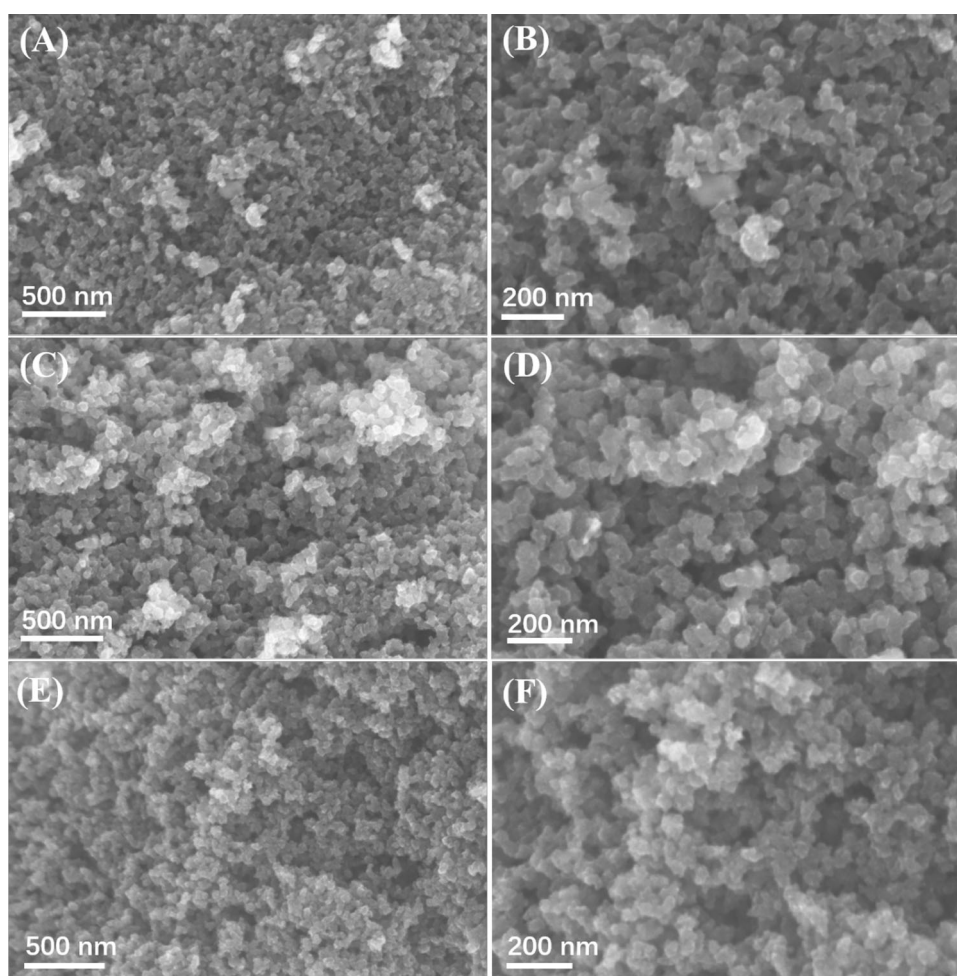
Figure 2 N₂ adsorption isotherms at −196 °C (a) and pore width distributions (b) of NPC-*x* samples.

Table 2 Porosity parameters and N contents of NPC-*x* samples

Sample	S_{BET} (m ² /g) ^a	V_{total} (cm ³ /g) ^b	V_{micro} (cm ³ /g) ^c	$V_{\text{micro}}/V_{\text{meso}}$ (%/%) ^d	N content (wt.%) ^e
NPC-4,4'-bipyridine	3580	3.38	1.15	34/66	2.46
NPC- <i>p</i> -phenylenediamine	2486	2.15	0.75	35/65	1.50
NPC-HMTA	2366	1.74	0.82	47/53	3.24

^aBET surface area^bTotal pore volume^cMicropore volume^dMicro-/mesopore volume ratio^eDetermined by elemental analysis**Figure 3** SEM images of NPC-4,4'-bipyridine (a, b), NPC-*p*-phenylenediamine (c, d) and NPC-HMTA (e, f).

consistent with the pore size distributions and porosity parameters calculated from N₂ adsorption isotherms. Figure 5 shows the XRD patterns and Raman spectra of NPC-*x* samples. In the XRD patterns, there are peaks arising from the (002) planes of graphitic carbons at $\sim 24^\circ$. However, the peaks are

very weak in intensities and broad in shapes, suggesting that the graphitic degrees of NPC-*x* samples are quite low. In the Raman spectra, there are peaks arising from the D bands of sp³ carbons at $\sim 1350\text{ cm}^{-1}$, and the G bands of sp² carbons at $\sim 1590\text{ cm}^{-1}$. The peaks of D bands are weaker in

Figure 4 TEM images of NPC-4,4'-bipyridine (a, b), NPC-*p*-phenylenediamine (c, d) and NPC-HMTA (e, f).

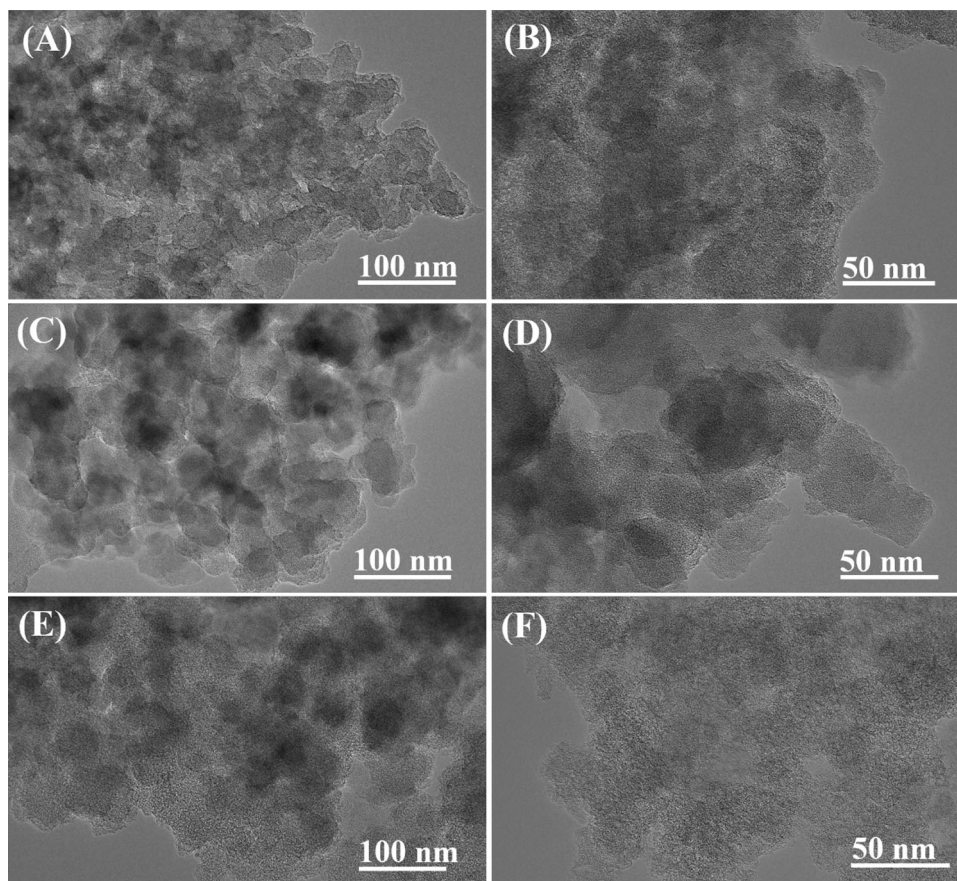
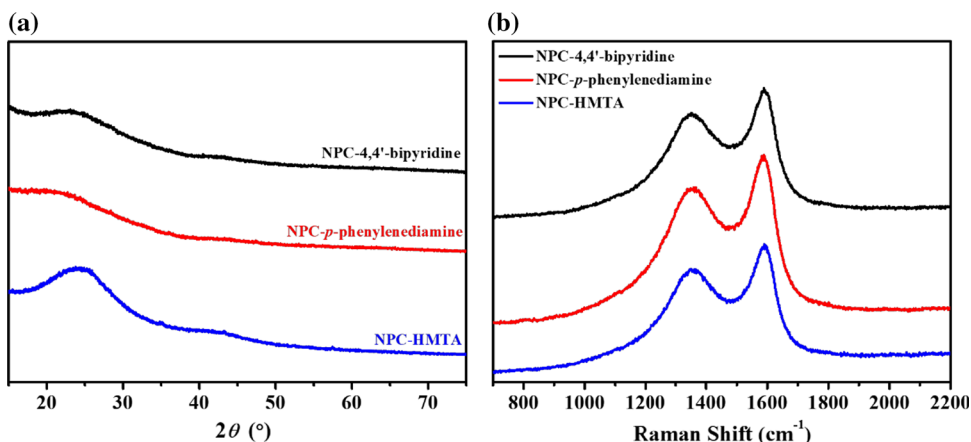


Figure 5 XRD patterns (a) and Raman spectra (b) of NPC-*x* samples.

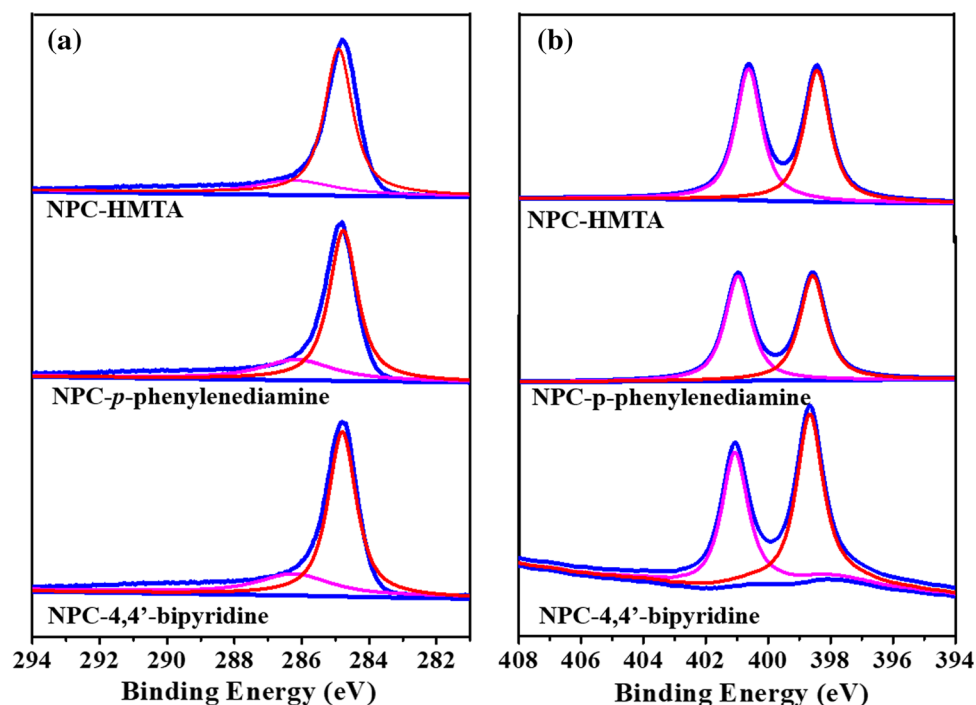


intensities than those of G bands, implying that most carbon atoms are assembled within conjugated rings in NPC-*x* sample.

The N contents of NPC-*x* samples determined by elemental analysis are also summarized in Table 2. NPC-*x* samples have the N contents of 1.50–3.24 wt%, with the highest value for NPC-HMTA. This is consistent with the highest N content for corresponding POP-HMTA among POP-

x samples. The N contents of NPC-*x* samples are lower than those of POP-*x* samples, because some N species are lost during carbonization process. However, the loss of N species for POP-4,4'-bipyridine (from 3.34 to 2.46 wt%) is not that significant as those for POP-*p*-phenylenediamine (from 3.25 to 1.50 wt%) and POP-HMTA (from 5.82 to 3.24 wt%). This can be explained by the fact that the N species in POP-4,4'-bipyridine are assembled within conjugated rings,

Figure 6 C 1 s (a) and N 1 s (b) spectra of NPC-*x* samples.

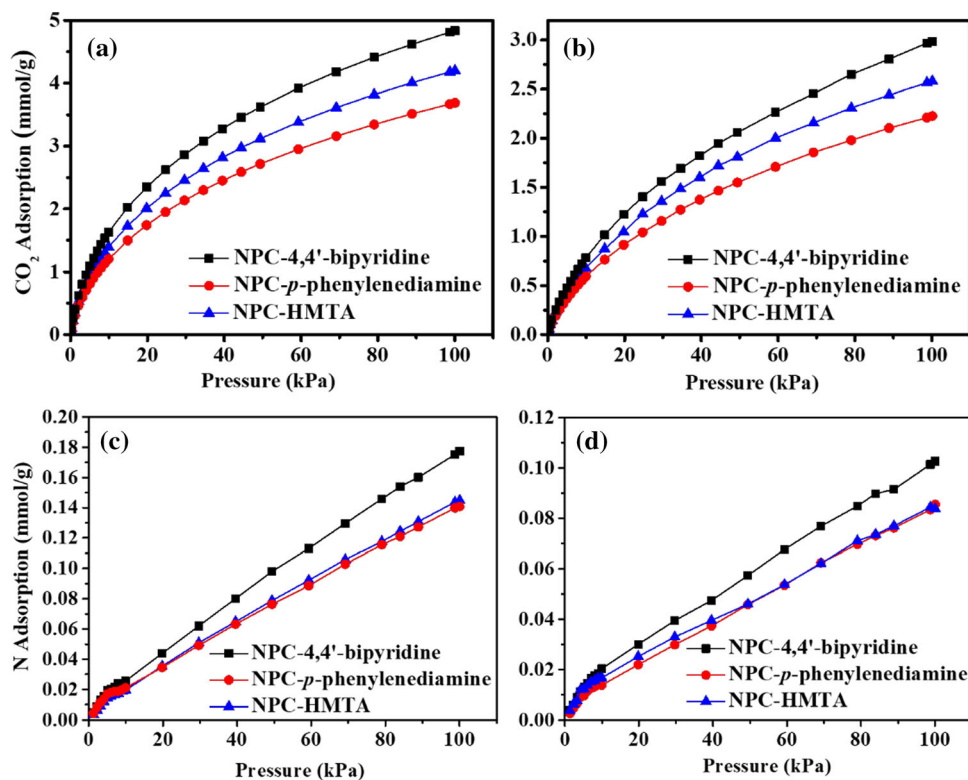


while those in POP-*p*-phenylenediamine and POP-HMTA are not. The N species assembled within conjugated rings are more stable than those connected by single bonds [51]. As a result, the N species in POP-4,4'-bipyridine are more favorable to be preserved during carbonization process. Figure 6 shows the C 1 s and N 1 s spectra of NPC-*x* samples. The C 1 s spectra can be deconvoluted into peaks at ~ 284.6 and ~ 286.2 eV. The former ones are attributed to the signals of C–C bonds, while the latter ones are attributed to the signals of C–N bonds. The N 1 s spectra can be deconvoluted into peaks at ~ 398.6 and ~ 401.1 eV. The former ones are attributed to the signals of pyridinic N, while the latter ones are attributed to the signals of quaternary N. It is found that the percentage of pyridinic N in NPC-4,4'-bipyridine is higher than those in NPC-*p*-phenylenediamine and NPC-HMTA (see Table S2). This is understandable since the N species in POP-4,4'-bipyridine are pyridinic N, while those in POP-*p*-phenylenediamine and POP-HMTA are not. Therefore, more pyridinic N can be decorated into NPC-4,4'-bipyridine during carbonization process. Since there are lone pair electrons in pyridinic N, pyridinic N is more attractive to CO₂ than quaternary N.

Selective CO₂ capture

Considering the ultrahigh surface areas, micro-mesoporous structure and N-decorated feature of NPC-*x* samples, they are believed to exhibit promising application in the capture and conversion of CO₂. The selective CO₂ capture performance of NPC-*x* samples was examined by both single and mixed gas adsorption experiments. Figure 7 shows the CO₂ and N₂ adsorption isotherms of NPC-*x* samples at 0 and 25 °C. The CO₂ adsorption isotherms display non-ideal profiles, which is a result of the strong interaction between N species of NPC-*x* samples and CO₂. The CO₂ uptakes follow the sequence of NPC-4,4'-bipyridine > NPC-HMTA > NPC-*p*-phenylenediamine. NPC-4,4'-bipyridine has the highest CO₂ uptakes among the three NPC-*x* samples because it has the highest surface area, micropore volume and percentage of pyridinic N. There is no significant difference in surface areas and micropore volumes for NPC-*p*-phenylenediamine and NPC-HMTA, but the N content of NPC-*p*-phenylenediamine is lower than that of NPC-HMTA. Therefore, NPC-HMTA has the second highest CO₂ uptakes, while NPC-*p*-phenylenediamine has the lowest CO₂ uptakes among the three NPC-*x* samples. The isosteric heats of CO₂ adsorption on NPC-*x* samples were calculated by using the CO₂ uptakes at different temperatures

Figure 7 CO₂ (a, b) and N₂ (c, d) adsorption isotherms of NPC-*x* samples at 0 °C (a, c) and 25 °C (b, d).



(see Figure S4). The CO₂ adsorption heats are in the range of -30 to -34 kJ/mol, suggesting the moderately strong interaction between NPC-*x* samples and CO₂. Therefore, the CO₂ adsorbed by NPC-*x* samples may be easily stripped out with low energy penalty. The CO₂ adsorption heats of NPC-4,4'-bipyridine are slightly more negative than those of NPC-*p*-phenylenediamine and NPC-HMTA, because NPC-4,4'-bipyridine has higher percentage of pyridinic N than the other two samples (see Table S2). In addition, the CO₂ uptakes of NPC-*x* samples are significantly improved in comparison with those of POP-*x* samples (see Figure S5), which is mainly attributed to the much higher surface areas of NPC-*x* samples than POP-*x* samples.

However, the N₂ adsorption isotherms display nearly ideal profiles, which is a result of the weak interaction between NPC-*x* samples and N₂. Due to the inert nature of N₂, the N species in NPC-*x* samples have little contribution to the adsorption of N₂. The N₂ uptakes follow the sequence of NPC-4,4'-bipyridine > NPC-*p*-phenylenediamine \approx NPC-HMTA. NPC-4,4'-bipyridine has the highest N₂ uptakes among the three NPC-*x* samples because it has the highest surface area. The N₂ uptakes of NPC-*p*-phenylenediamine and NPC-HMTA are similar,

because there is no significant difference in surface areas for NPC-*p*-phenylenediamine and NPC-HMTA. The CO₂ and N₂ adsorption isotherms were then fitted by the dual-site Langmuir–Freundlich (DSLFL) equation. The fitted parameters (see Tables S3–S6) were used to calculate the selectivities of CO₂/N₂ for NPC-*x* samples according to the ideal adsorption solution theory (IAST) [52], and results are shown in Fig. 8. The CO₂/N₂ selectivities of NPC-*x* samples follow the sequence of NPC-HMTA > NPC-4,4'-bipyridine > NPC-*p*-phenylenediamine. This is consistent with the sequence of N contents for NPC-*x* samples. In addition, the CO₂/N₂ selectivities of NPC-*x* samples are significantly improved in comparison with those of POP-*x* samples (see Figure S6), which is mainly attributed to the much higher CO₂ uptakes of NPC-*x* samples than POP-*x* samples. The CO₂ uptakes and CO₂/N₂ selectivities of NPC-*x* samples are summarized in Tables 3 and 4, respectively. Overall, these values are quite impressive and superior to most porous materials reported in the literature (see Tables S7 and S8).

Figure 9 shows the breakthrough curves for CO₂/N₂ mixed gas adsorption on NPC-HMTA at 25 °C. The breakthrough time for CO₂ is ~ 10 min/g while that for N₂ is almost instant, when the mixture of

Figure 8 IAST selectivities of CO_2/N_2 for NPC-*x* samples at 0 °C (a) and 25 °C (b).

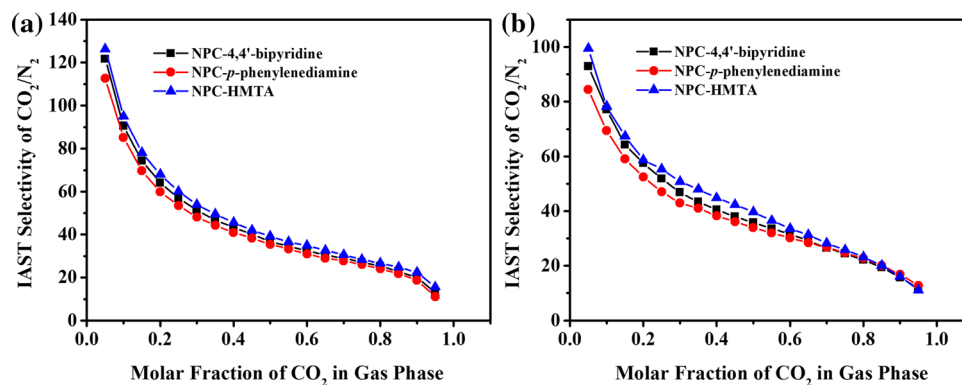


Table 3 CO_2 capacities of NPC-*x* samples at different temperatures and pressures

Sample	CO_2 capacity at 0 °C (mmol/g)		CO_2 capacity at 25 °C (mmol/g)	
	15 kPa	100 kPa	15 kPa	100 kPa
NPC-4,4'-bipyridine	2.03	4.83	1.01	2.98
NPC- <i>p</i> -phenylenediamine	1.50	3.68	0.78	2.22
NPC-HMTA	1.73	4.19	0.88	2.58

Table 4 IAST selectivities of CO_2/N_2 (0.15/0.85 vol) at 100 kPa for NPC-*x* samples

Sample	IAST selectivity at 0 °C	IAST selectivity 25 °C
NPC-4,4'-bipyridine	74	64
NPC- <i>p</i> -phenylenediamine	69	61
NPC-HMTA	78	67

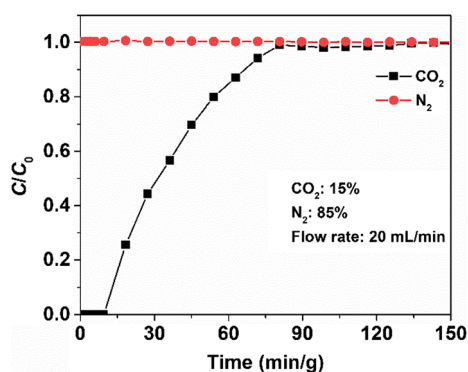


Figure 9 Breakthrough curves for CO_2/N_2 mixed gas adsorption on NPC-HMTA at 25 °C.

CO_2/N_2 (0.15/0.85 vol) was used for adsorption. It costs ~ 80 min/g for the concentrations of CO_2 in outlet gas reaching the level of inlet gas. Therefore, NPC-*x* samples do enable highly efficient and selective adsorption of CO_2 from CO_2/N_2 mixed gas. Given that there usually exists some moisture in the industrial stream, the breakthrough curves for humidified CO_2/N_2 mixed gas (RH = 20%) adsorption on NPC-HMTA at 25 °C were also determined

(see Figure S7). They are similar to those for dry gas adsorption, indicating that the presence of minor moisture has no obvious effect on the selective adsorption of CO_2 by NPC-*x* samples. The recyclability of NPC-*x* samples for CO_2 adsorption is very important for the long-term use of these materials. Figure S8 shows the CO_2 uptakes of NPC-HMTA at different adsorption–desorption cycles. The adsorption was performed at 25 °C and 100 kPa, and the desorption was performed at 80 °C and vacuum for 1 h. The loss of CO_2 uptakes for NPC-HMTA is negligible after five adsorption–desorption cycles. Therefore, the adsorption of CO_2 by NPC-*x* samples is highly reversible.

Catalytic CO_2 conversion

The synthesized M@NPC-*x* samples were first characterized by N_2 adsorption, ICP and XPS to give the porosity parameters (see Figure S9 and Table S9), metal contents (see Table S9) and chemical nature of loaded metals (see Figure S10). The porosity parameters of M@NPC-*x* samples are decreased in

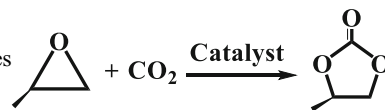
comparison with those of NPC-*x* samples, which is within expectation because some of the nanopores in NPC-*x* samples are occupied by loaded metals. The M@NPC-*x* samples also have predominantly mesoporous structure, with the mesopore volumes accounting for > 50% of the total pore volumes. Especially for Co@NPC-HMTA, the mesopore volume of which accounts for 91% of the total pore volume. The metal contents of M@NPC-*x* samples are similar, with the values of 0.62–0.71 mmol/g. The Co 2p spectra of Co@NPC-4,4'-bipyridine show the Co 2p_{3/2} and 2p_{1/2} peaks at 780.6 and 795.8 eV, respectively, and two satellites peaks at 786.9 and 802.7 eV. This agrees with the Co 2p spectra of Co²⁺, suggesting that the loaded Co does not undergo oxidation. The Co 2p spectra of Co@NPC-4,4'-bipyridine show the Co 2p_{3/2} and 2p_{1/2} peaks at 780.6 and 795.8 eV, respectively, and two satellites peaks at 786.9 and 802.7 eV. This agrees with the Co 2p spectra of Co²⁺, suggesting that the loaded Co does not undergo oxidation. The Zn 2p spectra of Zn@NPC-4,4'-bipyridine show the Zn 2p_{3/2} and 2p_{1/2} peaks at 1022.4 eV and 1045.4 eV, respectively, also suggesting that the chemical state of loaded Zn is Zn²⁺.

The catalytic CO₂ conversion performance of M@NPC-*x* samples was then examined by taking the cycloaddition of CO₂ with propylene oxide as a

model reaction. The yields, selectivities and TOFs of CO₂ conversion reaction under different conditions are summarized in Table 5. It can be seen that both Co@NPC-*x* and Zn@NPC-*x* samples show excellent performance for catalytic CO₂ conversion, giving the product yields of 97.6–99.5%, selectivities of 98.2–99.5% and TOFs of 614.1–716.8 h⁻¹ at 100 °C and 1.0 MPa in 1.5 h (runs 1–4). The activities of M@NPC-*x* samples for catalytic conversion of CO₂ are affected by many factors, such as the CO₂ uptakes and porosity parameters. It is difficult to determine that which factor contributes more than others to the catalytic conversion of CO₂. Herein, we can only make a qualitative analysis. Co@NPC-*p*-phenylenediamine has the lowest catalytic activity among the three Co@NPC-*x* samples, probably because NPC-*p*-phenylenediamine has the lowest CO₂ uptakes among the three NPC-*x* samples. Although the CO₂ uptakes of Co@NPC-HMTA are lower than those of NPC-4,4'-bipyridine, the percentage of mesopores in Co@NPC-HMTA is higher than that in NPC-4,4'-bipyridine (see Table S9). Since mesopores with large pore width are favorable for the diffusion of reactants, Co@NPC-HMTA has slightly higher catalytic activity than NPC-4,4'-bipyridine.

Co@NPC-4,4'-bipyridine still gives a product yield of 99.1% and TOF of 647.7 h⁻¹ even after recycled for 5 times (run 5), which is comparable to the activity of

Table 5 Cycloaddition of CO₂ with propylene oxide catalyzed by M@NPC-*x* samples



Run	Catalyst	Additive	Time (h)	Yield (%)	Selectivity (%)	TOF (h ⁻¹)
1	Co@NPC-4,4'-bipyridine	n-Bu ₄ NBr	1.5	99.5	> 99.5	640.9
2	Co@NPC- <i>p</i> -phenylenediamine	n-Bu ₄ NBr	1.5	98.1	98.2	614.1
3	Co@NPC-HMTA	n-Bu ₄ NBr	1.5	99.2	98.6	658.0
4	Zn@NPC-4,4'-bipyridine	n-Bu ₄ NBr	1.5	97.6	> 99.5	716.8
5	Co@NPC-4,4'-bipyridine ^a	n-Bu ₄ NBr	1.5	99.1	> 99.5	647.7
6	None	n-Bu ₄ NBr	1.5	34.9	98.5	–
7	Co@NPC-4,4'-bipyridine	None	1.5	21.5	99.1	138.5
8	Co@NPC-4,4'-bipyridine ^b	n-Bu ₄ NBr	48	97.8	> 99.5	19.7

Reaction conditions: 100 °C, propylene oxide (20 mmol), TBAB (7.2 mol%), CO₂ (1.0 MPa), catalyst (30 mg)

^arecycled for 5 times

^breaction conditions: 30 °C, propylene oxide (20 mmol), TBAB (7.2 mol%), CO₂ (0.3 MPa), catalysts (30 mg)

fresh sample. However, the product yield is only 34.9% if no catalyst was used (run 6), and the product yield is only 21.5% if no TBAB was used (run 7). Therefore, the cooperation between M@NPC-*x* samples and TBAB is essential for the efficient conversion of CO₂. It is noteworthy that Co@NPC-4,4'-bipyridine can give a product yield of 97.8% and TOF of 19.7 h⁻¹ when the conversion reaction was performed at 30 °C and 0.3 MPa for 48 h (run 8). Therefore, M@NPC-*x* samples are also efficient for catalytic CO₂ conversion under mild conditions. Overall, the activities of M@NPC-*x* samples are superior to most heterogeneous catalysts reported in the literature (see Table S10). The superior activities of M@NPC-*x* samples are believed to be attributed to ultrahigh surface areas, micro-mesoporous structure and N-decorated feature of NPC-*x* samples.

Conclusions

In summary, nitrogen-decorated micro-mesoporous carbons were developed by direct carbonization of porous organic polymers, which were pre-synthesized through alkylation-induced hyper-crosslink of rigid organic bases without the use of any templates. The developed carbons were systematically characterized for porous and chemical structure and also examined for CO₂ capture and conversion performance. It is found that the developed carbons have ultrahigh surface areas, abundant mesopores and weak base sites. As a result, the developed carbons enable highly efficient and selective adsorption of CO₂ from CO₂/N₂ mixed gas. The CO₂ capacities and CO₂/N₂ selectivities of developed carbons are superior to most porous materials reported in the literature. After loaded with metal salts, the developed carbons also exhibit high activities for the catalytic conversion of CO₂. The activities of metal-loaded carbons for cycloaddition of CO₂ with propylene oxide are also superior to most heterogeneous catalysts reported in the literature. Based on the results obtained in this work, it is concluded that the developed carbons have promising application in the capture and conversion of CO₂.

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Compliance with ethical standards

Conflict of interest The authors declare that they have no conflict of interest.

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