




# Biomass applied in supercapacitor energy storage devices

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## ABSTRACT

The ever-increasing energy demand and fossil energy consumption accompanied by the worsening environmental pollution urge the invention and development of new, environmentally friendly and renewable high-performance energy devices. Among them, the supercapacitor has received massive attention, and the various electrode materials and polymer electrolytes have been exploited. The carbon-based electrodes and electrolytes derived from biomass are highly trusted as idea candidates for supercapacitors due to their attractive structure, abundance, low cost, renewability, and environmentally friendliness. This review will highlight the available characteristics of materials, synthetic strategies, and improvement approach of biomass-derived electrodes and electrolytes for application in supercapacitors. Future relative research trends also will be briefly discussed.

## Introduction

In recent years, as the energy demand and fossil energy consumption is increasing rapidly and environmental pollution is getting worse, it is urgent to invent and develop new, environmentally friendly, and renewable high-performance energy conversion and storage devices [1, 2]. Supercapacitor is a new type of energy storage system between secondary battery and physical capacitor, which has received massive attention as energy storage owing to its ultrahigh power density, long life cycles, and rapid

charge/discharge rate. It is usually categorized into pseudocapacitors, the capacitance of which originates from reversible Faradaic reactions of electroactive species, and electrical double-layer capacitors with capacitance arising from the reversible electrostatic adsorption of ions at the interface of electrode/electrolyte based on the charge-storage mechanism [3]. However, there is still a certain gap to be narrowed between the electrochemical performance of supercapacitors and the increasingly stringent market requirements, such as fast discharge speed, high self-discharge rate, and low specific capacitance.

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The electrochemical performance of supercapacitors is significantly affected by the electrode materials. Metal oxides (such as  $\text{Co}_3\text{O}_4$ ,  $\text{MnO}_2$ , and  $\text{NiO}$ ), conductive polymers (such as polyaniline, polypyrrole, and polythiophene), and carbon materials are widely used for supercapacitor electrodes. Besides, novel electrode materials, such as metal phosphites, metal sulfide, and porous organic polymer, have been developed at present. Metal phosphites [4] show high electrical conductivity and high specific capacitance as pseudocapacitive materials for electrodes. Metal sulfides [5] possess superior electrical conductivity and are commonly compounded with other materials. Porous organic polymers exhibit electronic conductivity for its extended  $\pi$ -conjugated systems and significant pseudocapacitance for good redox activity, but its rate performance and cycling stability are inferior because of the low electrochemical stability, and much efforts have been made to overcome it [6]. Among these electrode materials, carbon materials, including activated carbons [7], carbon fibers [8], templated carbons [9], carbon aerogels [10], carbon nanotubes [11], and graphene [12], are most widely studied for its much better cycle performance, along with the large specific surface area, various pore structure, good electrochemical conductivity, and stability. Carbon materials derived from renewable biomass have drawn tremendous attention in recent years, such as cellulose, chitin, chitosan, and lignin, which have low cost, attractive structure, abundance, renewability, and environmental friendliness compared to traditional carbon materials. Especially, many biomass resources are rich in heteroatoms such as O, N, and S to be converted into heteroatom self-doped carbon materials [13–17] with higher specific capacitance.

At the same time, new electrolytes with high conductivity and superior electrochemical stability have been studied to improve the electrochemical performance of supercapacitors [18]. The supercapacitors are widely assembled with liquid electrolytes, which are usually toxic and corrosive involving the risk of leakage. To reduce the risk of electrolyte leak, polymer electrolytes formed by placing liquid electrolytes into polymer matrix are developed as ionic conductors and separators between electrodes [19, 20], possessing a wide potential window, a high ionic conductivity, excellent stability, and mechanical strength. The formation of polymer electrolytes can cause huge resource consumption and environmental

concerns. Thus, biomass-based electrolytes from cellulose, starch, chitosan, alginate, soybean protein, and so on have gained more and more attention recently for the advantages of renewability, low cost, environmentally friendliness, and superior mechanical properties.

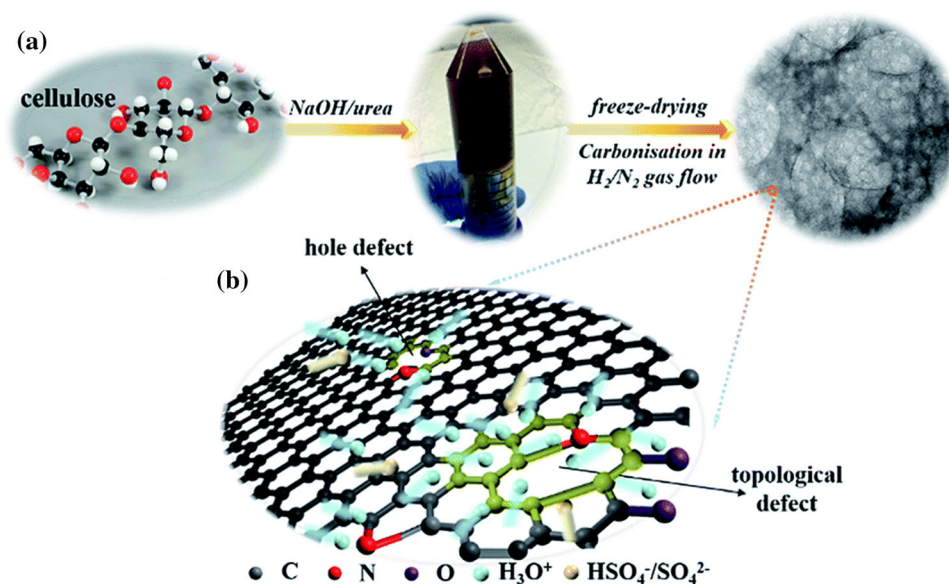
In this review, we provide a summary of the recent development in research of carbon-based electrode and polymer electrolyte derived from biomass materials, such as cellulose, chitin, chitosan, lignin, alginate, starch, and gelatin, emphasizing the characteristics, synthetic strategies, improvement approach, and their relationship with electrochemical performance of biomass-derived electrodes and electrolytes in supercapacitors. Finally, the current challenges and future directions of biomass-derived carbon electrodes and electrolytes are briefly discussed.

### Biomass-derived carbon materials in supercapacitor electrodes

Carbon materials possess various structures, low density, ideal stability, nontoxic, and high conductivity and are most widely used for supercapacitor electrodes, including activated carbons, template carbons, carbon nanotubes, graphene, and carbon aerogels. To reduce the energy shortages and environmental pressure, the carbon materials derived from biomass, such as cellulose, lignin, alginate, starch, chitin, and chitosan, have been considered as significant candidates of electrode materials with the advantages of abundance, unique nature structure, low cost, and environmental friendliness.

### Cellulose-derived carbon

Cellulose is the most abundant natural polymer with low density, low cost [21], biocompatibility, and large amount of hydroxyl groups for chemical modification. Cellulose-derived carbon has been applied in sustainable energy storage such as supercapacitor electrodes. Chen et al. [22] prepared hierarchical porous carbon using straw cellulose by carbonization and activation, exhibiting excellent electrochemical performance with a specific capacitance of 358 F/g and outstanding cycling stability (capacitance retention of 95.6% over 10,000 cycles) in  $\text{H}_2\text{SO}_4$  electrolyte. Lu et al. [23] produced porous carbon from



**Figure 1** Schematic illustration of the preparation of microcrystalline cellulose-derived porous carbon with intrinsic defects and N, O doping. Republished with permission of Royal Society of Chemistry, from Microcrystalline cellulose-derived porous carbons with defective sites for electrochemical

applications, Zhao, X. S.; Zhu, Zhonghua; Duignan, Timothy; Xiao, Changlong; Sun, Xiaoming; Gaddam, Rohit Ranganathan; Zhuang, Linzhou; Lu, Hao, 7, 39, 2019; permission conveyed through Copyright Clearance Center, Inc [23].

microcrystalline cellulose with intrinsic defects and nitrogen, oxygen doping by carbonization and thermal treatment in a reducing environment as Fig. 1 showed, which was used in a supercapacitor electrode delivering a high specific capacitance of 426 F/g at 0.25 A/g in 1 M  $\text{H}_2\text{SO}_4$  electrolyte and capacitance retention about 90% after 60,000 cycles at a current density of 5 A/g. Cui et al. [24] developed methylcellulose/ $\text{NaHCO}_3$  flake precursors and obtained wing-like porous carbon sheets after carbonization giving an obvious enhancement in potassium storage for their large surface area, open framework, and enlarged interlayer spacing. The assembled potassium-ion hybrid supercapacitor with porous carbon sheets as electrode materials showed an excellent energy density of 108 Wh/kg at a power density of 280 W/kg. Zhang et al. [25] assembled an all-nanofiber asymmetric supercapacitor using  $\text{NiCo}_2\text{O}_4$  as cathode, a nanocellulose membrane as separator, and nanocellulose-derived porous carbon as anode with a three-dimensional hierarchical porous structure and an ultrahigh surface area of 2046  $\text{m}^2/\text{g}$ , demonstrating a good electrochemical performance with the capacitance of 64.83 F/g at 0.25 A/g. The rich O self-doping in cellulose-derived carbon offers excellent capacity for electrodes.

Carbon aerogels based on cellulose possess high porosity and interconnected 3D networks as electrode materials with good mechanical stability. Wan et al. [26] developed a highly conductive cellulose-derived carbon aerogel matrix with self-stacked multilayer  $\text{FeOCl}$  for an anode material, delivering an ultrahigh specific capacitance of 647 F/g at 2 mA/ $\text{cm}^2$  in 1 M  $\text{Na}_2\text{SO}_4$  electrolyte and good cycle stability with capacitance loss less than 10% after 10,000 cycles, in which the carbon aerogel matrix significantly increased the mechanical stability and charge-storage kinetics. The nanofiber and nanoplate structures of carbon aerogels can be manipulated through the control of carbonization temperature. Zhang et al. [10] used wood cellulose to fabricate carbon aerogels by freeze-drying and high-temperature carbonization and revealed that the independent nanofibers structure and volume shrinkage increased with temperature while the nanoplate structure decreased. The carbon aerogels carbonized at 800 °C possessed the highest specific surface and total pore volume with a specific capacitance of 172.7 F/g and a specific capacitance of 89.43% after 5000 cycles. Yang et al. [27] firstly fabricated 3D hybrid aerogels from 2,2,6,6-tetramethylpiperidine-1-oxyl radical-oxidized cellulose nanofibril (TOCN)/GO aqueous solution by ion-

**Table 1** The capacity and cycle stability of cellulose-derived activated carbons and carbon aerogels

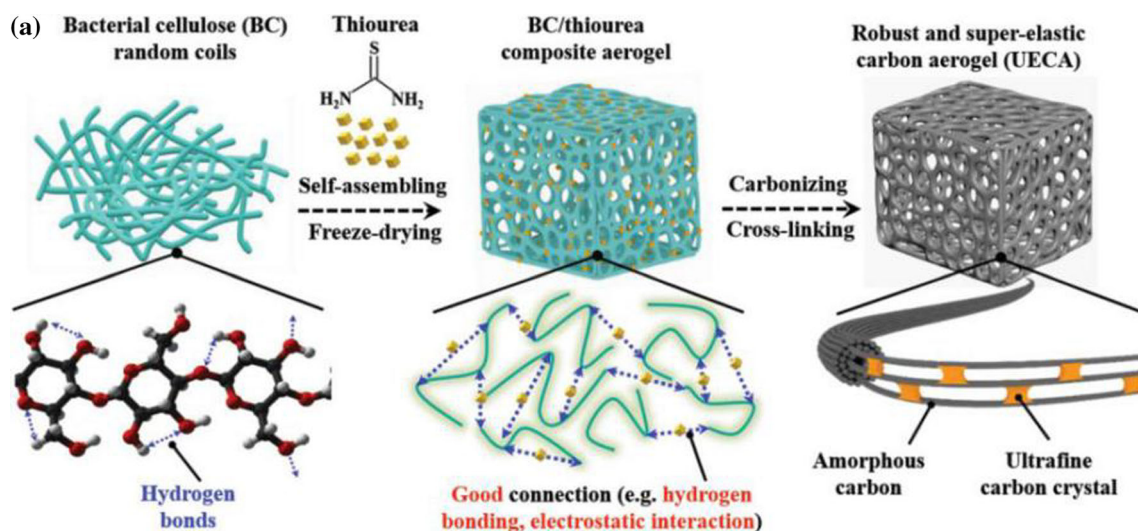
Types of carbon	Carbon materials	Specific capacitance	Cycle stability: capacitance retention	References
Activated carbons	Methylcellulose-derived porous carbon sheets	178 mAh/g at 2 A/g	75.8% after 3000 cycles	[24]
	Microcrystalline cellulose-derived N-doped carbon	426 F/g at 0.25 A/g	90% after 60,000 cycles	[23]
	Straw cellulose-derived carbon	358 F/g at 0.5 A/g	95.6% over 10,000 cycles	[22]
	Porous carbon from CMC, BC, and citric acid	350 F/g at 0.5 A/g	96% after 10,000 cycles	[28]
Carbon aerogels	Wood cellulose-derived carbon aerogels	172.7 F/g	89.43% after 5000 cycles	[10]
	Cellulose-derived carbon aerogel with FeOCl	647 F/g at 2 mA/cm <sup>2</sup>	90% after 10,000 cycles	[26]
	BC-based N-self-doped carbon nanofiber aerogels	224 F/g at 0.5 A/g	97% at 5 A/g after 10,000 cycles	[29]
	TOCN/RGO carbon aerogels	394.5 F/g at 0.5 A/g	99.77% after 10,000 cycles	[27]

exchange and freeze-drying and further obtained carbonized TOCN/RGO aerogels through carbonization at 1100 °C showing high specific capacitance of 394.5 F/g at 0.5 A/g and 99.77% capacitance retention after 10,000 cycles. The capacity and cycle stability of cellulose-derived activated carbons and carbon aerogels are shown in Table 1. Compared to activated carbons, carbon aerogels based on cellulose have more developed pore structure and 3D networks providing rapid ion transport channels and thus carbon aerogels tend to show more outstanding electronic conductivity and cycle stability.

Among various cellulose and its derivatives, bacterial cellulose and carboxymethyl cellulose-derived carbon exhibit relatively excellent electrochemical performance and get more extensive attention. Bacterial cellulose (BC) consists of pure cellulose produced by microorganism fermentation of *Gluconacetobacter xylinus* without hemicellulose and lignin. Besides the advantages of lightweight, nontoxicity, high flexibility, industrial-scale production, and excellent biodegradability, BC has an ultrahigh water holding capacity to hold dopants because of the hydrophilic nature, and an ultrafine three-dimensional (3D) network structure composed of nanofibers as a desired precursor to produce porous carbons. Lei et al. [30] prepared a highly N/S dual-doped carbon derived from BC with 3D structure by absorption swelling, possessing excellent wettability, fast ion-transportation, and a high

specific capacitance of 202 F/g at a current density of 1.0 A/g. Chen et al. [29] prepared N-self-doped carbon nanofiber aerogels via in situ growth of zeolitic imidazolate frameworks (ZIF8) nanocrystals on the surface of BC gel due to electrostatic interaction followed by dry-freezing and carbonization at 90 °C, exhibiting specific capacitance about 224 F/g at 0.5 A/g, high energy density of 31.04 Wh/kg at a power density of 250 W/kg, and good capacity retention about 97% at 5 A/g after 10,000 cycles.

Carboxymethyl cellulose (CMC), one of the cellulose derivatives, has water solubility and sustainability and is rich in carboxyl and hydroxyl groups with a high content of oxygen, based on which CMC has been considered as a promising candidate for the creation of O-self-doping carbon electrodes. Lee et al. [31] prepared porous carbon turbostratic structures from an aqueous CMC solution by ice-templating and carbonization, possessing a high specific area of 724.35 m<sup>2</sup>/g and well-developed pore structures without activation, as well as excellent specific capacitance about 210 F/g at 1A/g and an outstanding capacitance retention about 100% after 10,000 cycles at 10 A/g in a 6.0 M KOH electrolyte. Shu et al. [28] fabricated hierarchical porous carbon materials from carboxymethyl cellulose, bacterial cellulose, and citric acid through a one-step carbonization/activation process, exhibiting a high specific capacitance of 350 F/g at 0.5 A/g, good



**Figure 2** Schematic illustration of the preparation process of BC/thiourea carbon aerogel. Reproduced from Robust, Superelastic Hard Carbon with In Situ Ultrafine Crystals, Xiaoping Yang, Wei-

Hong Zhong, Yunhua Yu, et al., 30, 3, 2019 with permission from John Wiley and Sons [37].

capacitance retention of 96% after 10,000 cycles, and excellent rate capability of 254 F/g at 15 A/g.

The porous carbon derived from biowaste rich in cellulose is considered as a prospective electrode material due to low cost and sustainable characteristics. The specific surface area of them is limited to hundreds of square meters per gram, calling for templates or activation with the activation reagent such as KOH and  $K_2CO_3$  to enhance the porous structure. Cigarette butts containing lots of cellulose acetate have been used to prepared nitrogen-doped carbon with urea as the nitrogen source via carbonization, activation, and hydrothermal method, and the obtained hierarchical porous carbon had a high specific surface area of  $1633.37 \text{ m}^2/\text{g}$ , advantageous specific capacitance of 330.1 F/g at 0.5 A/g, and high cycling stability of 93.48% after 10,000 cycles [32]. Chestnut shell-derived hierarchical porous carbon was prepared through the activation with  $KHCO_3$ , possessing a BET surface area of  $2298 \text{ m}^2/\text{g}$ , a good specific capacity of 387 F/g at 2 A/g, an excellent rate capability of 303 F/g maintained at 50A/g, and high capacity preservation of 98.68% after 10,000 cycles at 30 A/g [33]. Chaturvedi et al. [34] prepared porous carbon for supercapacitor electrode materials using anaerobically treated fruit and vegetable waste as a source containing calcium and silicon-based inorganic impurities as hard templates to create hierarchical pores during

carbonization, and the porous carbon had a specific surface area of  $2502 \text{ m}^2/\text{g}$  and a capacitance of 235 F/g at 1 A/g in acidic aqueous.

In terms of performance improvement, heteroatoms doping of cellulose-derived carbon can provide extra pseudocapacitances and increase the surface wettability. It is a facile and low-cost approach to enhance the electrochemical performance of carbon-based electrodes. Urea and melamine are widely used as the main nitrogen source to prepared N-doped carbon [16, 35]. Kim et al. [36] prepared N-doped porous carbon using Mg/K/Mg center dot K-nitrate-urea-cellulose mixture by the efficient and exothermic pyrolysis, high-temperature carbonization, and subsequent washing treatment, showing a specific capacitance of 279 F/g at 1 A/g and a good rate capability of 235 F/g at 30 A/g in the two-electrode system. Chen et al. [12] fabricated graphene@mesoporous nitrogen-doped carbon nanosheets with the sandwich-like structure using cellulose as precursor and melamine as the nitrogen source, which exhibited good specific capacitance of 264 F/g at 1 A/g, and outstanding cyclic stability with 99.9% retention after 2500 cycles. Thiourea is another additive to achieve N, S co-doping of the cellulose-derived electrode material. Ding et al. [37] prepared an ultralight and superelastic hard carbon aerogel by carbonizing self-assembling BC/thiourea aerogel as shown in Fig. 2, and further fabricated the

**Table 2** The capacity and cycle stability of cellulose-derived carbon, heteroatom-doped carbon materials and carbons compounded with pseudocapacitive materials

Types of carbon	Carbon materials	Specific capacitance	Cycle stability: capacitance retention	References
Cellulose-derived carbons	Methylcellulose-derived porous carbon sheets	178 mAh/g at 2 A/g	75.8% after 3000 cycles	[24]
	Wood cellulose-derived carbon aerogels	172.7 F/g	89.43% after 5000 cycles	[10]
	CMC-based porous carbon	210 F/g at 1 A/g	about 100% after 10,000 cycles	[31]
	Straw cellulose-derived carbon	358 F/g at 0.5 A/g	95.6% over 10,000 cycles	[22]
Heteroatom-doped carbons	BC-based N-self-doped carbon nanofiber aerogels	224 F/g at 0.5 A/g	97% at 5 A/g after 10,000 cycles	[29]
	Graphene@mesoporous nitrogen-doped carbon nanosheets	264 F/g at 1 A/g	99.9% after 2500 cycles	[12]
	N and S co-doped porous carbon	288 F/g at 0.5 A/g	96% after 10,000 cycles	[38]
	Porous carbon from CMC, BC, and citric acid	350 F/g at 0.5 A/g	96% after 10,000 cycles	[28]
	Microcrystalline cellulose-derived N-doped carbon	426 F/g at 0.25 A/g	90% after 60,000 cycles	[23]
	Carbons compounded with pseudocapacitive materials	Porous carbon aerogel loaded PPy	387.6 F/g at 0.5 A/g	92.6% after 10,000 cycles
Cellulose-based carbon aerogel loaded Co <sub>3</sub> O <sub>4</sub> and NiMoO <sub>4</sub>		436.9 C/g at 0.5 A/g	82.7% after 5000 cycles	[39]
Cellulose-derived carbon aerogel with FeOCl		647 F/g at 2 mA/cm <sup>2</sup>	90% after 10,000 cycles	[26]
BC-derived carbon Co <sub>3</sub> O <sub>4</sub> with Co <sub>3</sub> O <sub>4</sub> nanoparticles		785 F/g at 0.5 A/g	93.0% after 5000 cycles	[40]

supercapacitor presenting areal capacitance of 0.97 F/cm<sup>2</sup> and capacitance retention of 86% at 20 A/g. Song et al. [38] produced N and S co-doped porous carbon utilizing cellulose as precursors, a 9.5% NaOH and 4.5% thiourea aqueous as doping reagent and solvent by an all-in-one pyrolysis strategy, and the porous carbon showed appropriate heteroatom functionalities with 3.82% N and 1.94% S, and a high specific capacitance of 288 F/g at 0.5 A/g in KOH electrolyte for the three-electrode system. The comparison of cellulose-derived carbon and heteroatom-doped carbon materials is shown in Table 2. The redox reaction of heteroatoms offers extra pseudocapacitances to improve the capacity of carbon materials, and the increased surface wettability can accelerate the electron transport rate of electrode materials.

Besides, the electrochemical properties of cellulose-derived carbon electrode can also be improved by the

incorporation of nanostructured pseudocapacitive materials with redox kinetics. The hybridization of metal oxides, such as NiMoO<sub>4</sub> and Co<sub>3</sub>O<sub>4</sub> with carbonaceous species, has been widely applied. Wang et al. [39] developed a hierarchical porous material by loading Co<sub>3</sub>O<sub>4</sub> nanoparticles and NiMoO<sub>4</sub> nanorods on cellulose-based carbon aerogel frameworks, delivering an improved specific capacitance of 436.9 C/g at 0.5 A/g and capacitance retention of 70.7% at 5.0 A/g. Zhao et al. [40] used BC adsorbing organophosphorus pesticides as a carbon precursor to fabricate a loose 3D carbon skeleton, which then absorbed cobalt salt further transformed into Co<sub>3</sub>O<sub>4</sub> nanoparticles anchored on the carbon skeleton for electrode material. The electrode showed an ultra-high specific capacitance of 785 F/g at 0.5 A/g and excellent cycling stability with capacity retention over 93.0% after 5000 cycles. Compared with metal oxides, conducting polymers feature lightweight, low cost,

and simple preparation process for composite electrodes. Zhuo et al. [41] prepared a hierarchical porous carbon aerogel from cellulose loading conductive polymer polypyrrole (PPy), and the obtained hybrid exhibited a good specific capacitance of 387.6 F/g at 0.5 A/g and high cycling stability with a capacitance retention of 92.6% after 10,000 cycles in 1.0 M H<sub>2</sub>SO<sub>4</sub> electrolyte. Zhang et al. [42] fabricated compressible carbon and mechanically robust aerogels using bamboo cellulose nanofibers as precursor via low-temperature lyophilization and high-temperature pyrolysis, and further coated carbon aerogels with polypyrrole by in situ oxidative polymerization. The prepared core-sheath composites were applied in a symmetrical supercapacitor delivering a good specific capacitance of 268.5 F/g at a current density of 0.5 A/g, and an energy density of 23.8 Wh/kg at a power density of 450.4 W/kg. As shown in Table 2, the capacity of the compound of carbon and pseudocapacitive materials is much higher than carbon materials thanks to the redox kinetics of metal oxides or conducting polymers. However, the cycle stability is not satisfactory because of the restricted solid attachment between two species and the agglomeration of iron oxide nanoparticles during the charging-discharging processes, which need further optimization.

### Chitin/chitosan-derived carbon

Chitin, a polysaccharide mainly found in zygomycetes fungi, crustaceans (such as the shell of shrimp and crab), insects, and other arthropods, received extensive attention as a carbon precursor due to its high carbon content, abundant resource, low cost, nontoxicity, renewability, and biodegradability. It is composed of  $\beta$ -(1,4)-linked-2-acetamido-2-deoxy-d-glucopyranose units, possessing numbers of amido-gen and hydroxyl groups and a considerable amount of nitrogen ( $\sim$  6.9 wt%) to be a promising precursor of N-enriched porous carbon. The nitrogen-containing carbon materials derived from chitin have been successfully synthesized and employed as electrode materials for supercapacitors. Wang et al. [9] synthesized a hierarchical porous carbon from chitin with KMnO<sub>4</sub> as template precursor and activating agent, and it had a huge specific surface area and substantial nitrogen and oxygen functional groups as well as an outstanding specific capacitance of 412.5 F/g at 0.5 A/g, and excellent electrochemical

stability with only 0.4% capacitance loss after 10,000 cycles. Recently, chitin nanofibers have attracted much attention for fabricating nanogels and areogels as precursors for N-doped carbon with hierarchical porosity, large surface area, and good electrochemical performance. Chitin-derived carbon tends to keep excellent morphology after pyrolyzation thanks to the stiffness of chitin chains providing great opportunities for well-done nanostructure. Zheng et al. [43] prepared chitin nanogels by sol-gel transition of chitin in NaOH/urea solvent and pyrolyzed them at 800 °C obtaining N-doped porous carbon nanospheres having ordered micropores centered at about 0.6 nm, high surface area of 1363 m<sup>2</sup>/g, specific capacitance of 192 F/g at 0.5 A/g, and capacitance retention of 81% at 10 A/g. Ding et al. [44] fabricated nanofibers aerogel from chitin isolated from marine biomass by hydrogen bond crosslinking as the precursor of N-enriched carbon nanofiber aerogel with a high specific surface area up to 1597 m<sup>2</sup>/g, high nitrogen content up to 7.65%, the largest electrochemical capacitance of 221.0 F/g at 1.0 A/g, and good retention capability of 92% over 8000 cycles at 5.0 A/g in a 6.0 mol/L KOH electrolyte.

Direct pyrolysis of heteroatom-rich biomass is an alternative method for heteroatom-containing carbon electrode materials. Chitin-based heteroatom-doped carbons derived from shells of crab and shrimps, cicada slough, squid gladius, and so on are utilized as electrode materials for supercapacitors. Cicada slough contains plentiful chitin along with proteins, amino acids considered as nitrogen sources, sulfur, and phosphorous. Jia et al. [45] used it to prepare porous carbon with high heteroatom functionalities at about 12.06% atomic ratio of N, O, S, P, and high specific surface area of 1676 m<sup>2</sup>/g, showing a high specific capacitance of 355 F/g at 1 A/g and an excellent rate capability of 284 F/g at 30 A/g. Zhou et al. [46] prepared cicada slough-derived carbon by pre-carbonization in air and KOH activation and the as-prepared porous carbon with KOH/char weight ratio of 2/1 showed greatest electrochemical performance with a specific capacitance of 266.5 F/g at 0.5 A/g and a good rate capability of 196.2 F/g at 20 A/g. The squid gladius has considerable amount of chitin and less impurities requiring short process to be converted into chitin compared with shells of crab and shrimps. It was carbonized into O, N-doping activated carbon via KOH activation with a nitrogen content of 4.04 wt% and large surface area of

1129 m<sup>2</sup>/g for supercapacitor electrodes delivering maximum specific capacitance of 204 F/g and 100% capacitance retention after 25,000 cycles in 1 mol/L H<sub>2</sub>SO<sub>4</sub> electrolyte [7].

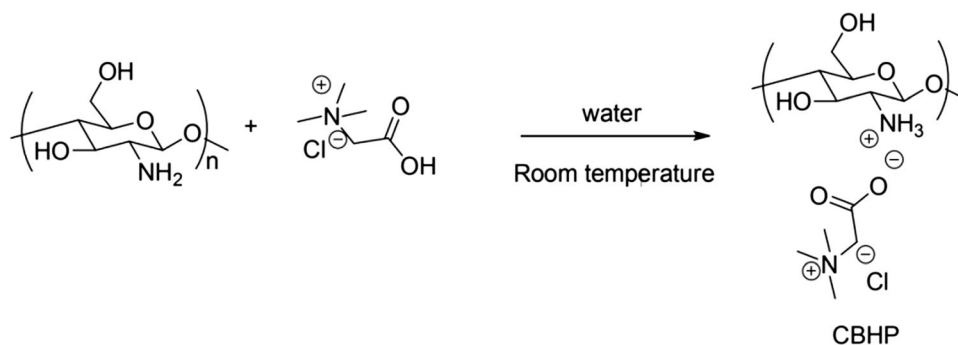
Chitin is insoluble in common solvents such as dilute acid, and it requires special solution systems to dissolve chitin at low temperature including LiOH/KOH/urea aqueous solution and NaOH/urea aqueous system. Compared to chitin, chitosan, a partially deacetylated derivative of chitin possesses better solubility with substantial amino group (-NH<sub>2</sub>), and the chitosan with low molecular weight or high degree of deacetylation (not less than 55%) is soluble in kinds of dilute acid. Taking this into account, Gao et al. [47] prepared carbon microspheres by pyrolyzing the chitin microspheres from a chitin/chitosan blend solution, and the improved nanopores/nanochannels of microspheres were generated via dissolution and removing chitosan with acetic acid. The as-prepared carbon microspheres had improved specific surface area of 1450 m<sup>2</sup>/g as a substrate of polyaniline nanoclusters for symmetric supercapacitor exhibiting excellent cycling stability of 90.6% capacitance retention after 10,000 cycles. The solubility of chitosan also makes it easy to prepare uniform blends with other soluble raw materials. Yuan et al. [48] synthesized the hybrid porous carbon via activating the mixture of chitosan and gelatin prepared through simple solution blending and freeze-drying. Gelatin can not only bring heteroatoms into the carbon skeleton for kinds of functional groups (-OH, -NH<sub>2</sub>, -COOH) on its main chain, but also change the microstructure and increase the porosity of chitosan-based carbon. The obtained carbon material exhibited an ultrahigh specific capacitance of 331 F/g at 1 A/g and well-done cycle stability with 90% capacitance retention after 10,000 cycles at 10 A/g. Up till now, the carbonization of chitosan into carbon electrode materials has attracted great attention for the high N content of chitosan to introduce N into the carbon framework. Genovese et al. [49] studied chitosan-derived activated carbon electrode material in various aqueous liquid and solid polymer electrolytes, proving that chitosan-activated carbon is compatible with neutral, acidic, and alkaline liquid electrolytes and advanced solid polymer electrolytes such as Li<sub>2</sub>SO<sub>4</sub>-polyacrylamide solid electrolyte.

Chitosan has massive amino groups (-NH<sub>2</sub>) and hydroxyl groups (-OH) on its chain which give it a

strong affinity for transition-metal ions. The composites of chitosan and metal have been studied as precursors for stable carbon electrode materials. Zhang et al. [50] fabricated graphene-like carbon nanosheets through simple one-step carbonization of chitosan-containing textile sludges, obtained by using chitosan derivative to remove reactive dye from aqueous solution, with Fe<sub>2</sub>(SO<sub>4</sub>)<sub>3</sub> as graphitization catalyst. The as-prepared graphene-like carbon nanosheets showed a good capacitance of 195 F/g at 1 A/g and outstanding cycling stability with 3.4% capacitance loss after 2500 cycles as electrode materials for supercapacitors, attributed to the ultrathin (near 3.2 nm thickness) graphene-like nanosheets structure and co-doping of Fe and N. Hao et al. [11] used chitosan-coated oxygen-containing carbon nanotubes to remove toxic metal ions of Cu(II) and Cr(VI) and further carbonized the metal ions-adsorbed adsorbents producing Cu- and CrN-loaded carbon composites with a specific capacitance of 114.9 F/g at 2 mV/s, respectively, superior to pure chitosan-coated carbon nanotubes.

Pyrolysis of chitosan or chitin through several conventional methods, such as direct activation and two steps carbonization followed by activation has been widely applied in producing N-doped porous carbons for supercapacitor. Cheng et al. [51] prepared N-doped hierarchically porous carbon by carbonization and activation of chitosan and betaine hydrochloride polyelectrolyte formed via reaction between amino groups in chitosan and carboxylic acid group in betaine hydrochloride (Fig. 3). The obtained N-doped porous carbon displayed a pore volume of 2.24 cm<sup>3</sup>/g, an ultrahigh specific area of 3300 m<sup>2</sup>/g, large amount of O (17.16 at.%) and N (2.23 at.%), and excellent electrochemical performance with high capacitance of 367 F/g at 0.1 A/g, and superior retention capability of 94% after 20,000 cycles at 4 A/g. In addition, hydrothermal carbonization (HC) is another traditional method to fabricate N-doped porous carbon from chitosan or chitin for supercapacitor using water as the carbonization medium at mild processing temperature, and the control of HC time can increase doped N content, active N species, turbostratic structure, mesoporosity, and specific surface area resulting in significant effect on the supercapacitance performances. Tong et al. [52] prepared an N-doped porous carbon for supercapacitor from chitosan by controlling HC with highly time-dependent





**Figure 3** The synthesis of chitosan and betaine hydrochloride-based polyelectrolyte precursor. Republished with permission of Royal Society of Chemistry, from ultrahigh-surface-area nitrogen-doped hierarchically porous carbon materials derived from chitosan and betaine hydrochloride sustainable precursors for

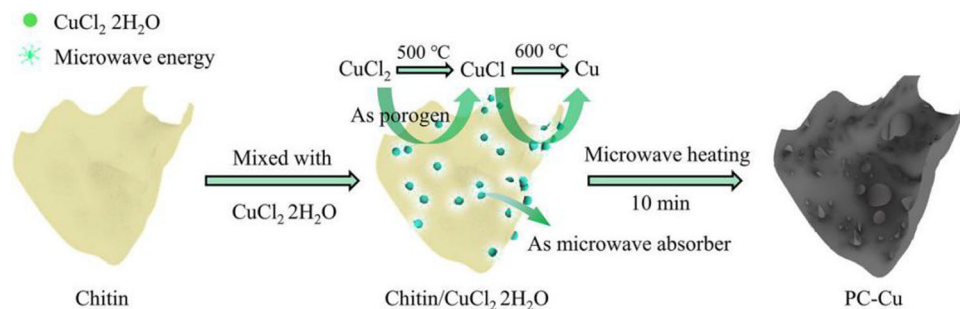
physicochemical and electrochemical properties, exhibiting outstanding specific capacitance around 406 F/g in a three-electrode system and 274 F/g in a two-electrode system. Wu et al. [53] fabricated oxygen- and nitrogen-containing carbon via one-pot ionic liquid-assisted HC carbonization of chitosan, during which ionic liquids played as nitrogen sources, structure-directing agent, and dispersant. The ordered lattice of graphite-like structures with carbon quantum dots on the surface provided supercapacitor performance with a specific capacitance of 355 F/g at 0.2 A/g in 1 M H<sub>2</sub>SO<sub>4</sub> in a three-electrode system. Zhou et al. [54] prepared supramolecular hydrogels by mixing chitosan and zinc chloride (as a chemical activation agent) solution and then made nitrogen-rich hierarchically porous carbon via HC having a high specific surface area of 1067 m<sup>2</sup>/g and nitrogen content of 6.36 at% as well as delivering a good specific capacitance of 228.7 F/g at 1 A/g, superior rate capability of 84.9% retention at 10 A/g and long-term electrochemical stability with 1.7% capacity loss after 5000 cycles.

Microwave heating is an excellent carbonization method with high heating efficiency and rate [55, 56] and has been used in the rapid pyrolysis of chitin and chitosan. For the poor microwave-absorbing ability of biomass to be transformed into porous carbon in this way, ZnCl<sub>2</sub> is widely introduced as microwave absorber and porogen for its good microwave-absorbing ability. However, the pore-creating ability of ZnCl<sub>2</sub> is limited due to its nonetching activation mechanism. Thus, various alternative methods have been reported to obtain chitin or chitosan-derived

high-performance supercapacitors, Jian Cheng, Qinqin Xu, Xia Wang, Zaiquan Li, Fuzhong Wu, Jiaojing Shao and Haibo Xie, 3, 5, 2019; permission conveyed through Copyright Clearance Center, Inc [51].

porous carbon with the absence of ZnCl<sub>2</sub> via microwave heating. Chen et al. [57] used wheat-straw-derived char as a microwave absorber and KOH as an activation agent to prepare porous carbons via microwave heating, and post-oxidation reactions between oxygen in the air and active sites in the carbons resulted in the doping of oxygen (contents up to 21.6–28.2%). The as-prepared carbon materials had a high specific surface area of 1905 m<sup>2</sup>/g, a proper hierarchically pore distribution with mesopore ratios of 45.6–59.0%, good capacitive performances with a high specific capacitance of 325 F/g at 0.5 A/g, and a superior capacitance retention of 81.0% at 10 A/g. Chen et al. [58] used chitin as carbon precursor and chloride dihydrate (CuCl<sub>2</sub>·2H<sub>2</sub>O) as microwave absorber and porogen to directly synthesize nitrogen-rich porous carbon by only microwave heating process within 10 min (Fig. 4). The obtained porous carbon gained nitrogen retention up to 5.2%, a superior specific capacitance of 227 F/g at 0.5 A/g, a capacitance retention of 84.8% at 10 A/g and 67.7% at 50 A/g in 6 mol/L KOH electrolyte. Chen et al. [59] converted chitosan hydrogel beads into porous carbons with KOH as a chemical porogen and water wrapped in hydrogel beads as a physical porogen transformed into steam to create mesopores, and the porous carbons material possessed N content of 3.4% and high O content of 17.5% introduced into carbons by the post-oxidation in air atmosphere, as well as high capacitance retention of 87.5% from 0.5 to 10 A/g.

As mentioned above, the incorporation of heteroatoms is a promising approach to enhance the



**Figure 4** The scheme of the rapid microwave-assisted synthesis of chitin-based porous carbon. Reproduced from Rapid synthesis of chitin-based porous carbons with high yield, high nitrogen

retention, and low cost for high rate supercapacitors, Weimin Chen, Min Luo, Xin Wang, et al., 44, 2, 2019 with permission from International Journal of Energy Research [58].

electrochemical performance of biomass-derived carbon. Furthermore, multiple heteroatoms doping, such as nitrogen/sulfur co-doping and nitrogen/boron co-doping, is proved to have a synergistic effect. Huo et al. [60] fabricated N/S co-doped hierarchical porous carbons from freeze-dried mixture containing chitosan as carbon precursor,  $\text{CaCl}_2$  as activator, urea as foaming agent, and methanesulfonic acid as sulfur source by one-step carbonization and activation, in which methanesulfonic acid reacted with chitosan molecules forming hydrogen bonds to dope controllable sulfur and to construct a uniform planar network, further giving porous carbons good electrochemical performance such as a specific capacitance of 272 F/g at 1.0 A/g and an excellent rate capability of 172 F/g at 100 A/g in 6.0 M KOH electrolyte. Lin et al. [61] prepared N/B co-doped porous carbon by one-step carbonization and activation of the chitosan/KOH mixture at high temperature, followed by hydrothermal reaction with boric acid to dope boron atoms. The electrochemical double-layer supercapacitor with N/B co-doped porous carbon as electrode material delivered a high specific capacitance of 316 F/g at 0.2 A/g and an excellent capacitance retention of 94.08% after 10,000 cycles at 10 A/g in the 6 M KOH electrolyte. More heteroatoms co-doped chitosan-derived carbon materials and its electrochemical properties are listed in Table 3.

Transition-metal oxides are considered as excellent pseudocapacitive materials to be integrated with chitosan-derived carbon for effectively facilitating the surface wettability, the pore structure, and electrochemical performance of carbon materials in supercapacitor field. Ken et al. [62] used N-doped chitosan nanofibers and nickel nitrate as precursors and  $\text{KNO}_3$

as a pore-forming agent to form N/NiO<sub>x</sub> co-doped hierarchical porous carbons nanosheets through freeze-drying and carbonization, which exhibited a higher specific capacitance of 614.6 F/g at 1 A/g than that of carbon nanosheets undoped Ni, and the assembled supercapacitor achieved a high energy density of 20.3 Wh/kg at a power density of 240.9 W/kg and retained 85.61% of the initial capacitance after 10,000 cycles. Liu et al. [63] used graphene-dispersed chitosan hydrogel as precursor to prepare the B, N-doped porous carbon and further anchored Co and Ni oxide nanoparticles on it via chemical reduction and calcining at high temperature. The Co and Ni species significantly improved the graphitic degree and the composite exhibited extremely high capacitance of 1266.7 F/g at 1 A/g and good stability. Lin et al. [64] prepared chitosan-derived porous carbon through directly carbonized with KOH as the activator and grow  $\text{MnO}_2$  on the surface of it through the redox reaction between  $\text{KMnO}_4$  and it to form  $\text{MnO}_2$ -doped holey carbon showing an outstanding capacitance of 460 F/g at 0.2 A/g and an excellent capacitance retention of 91.67% over 10,000 cycles at 10 A/g in 6.0 M KOH electrolyte in a three-electrode system. Wang et al. [66] fabricated Co/N-doped porous carbon material using chitosan, urea, and cobalt acetate as the precursor of carbon, nitrogen, and cobalt, respectively, with F127 as dispersant for Co/CoO<sub>x</sub> particles at high pyrolysis temperature. It delivered an overpotential of 259 mV, small Tafel slope of 99 mV/dec, and a high stability of 83% current density retention after 6 h electrochemical test as electrode in hydrogen evolution reaction in 1 M potassium hydroxide electrolyte. Al-Farraj et al. [65] embedded  $\text{NiFe}_2\text{O}_4$  nanoparticles into nitrogen-doped carbon matrix

**Table 3** The electrochemical properties of heteroatom co-doped and transition-metal oxide-doped chitosan-derived carbon materials

Types of carbon materials	Carbon materials	Specific capacitance	Cycle stability: capacitance retention	References
Heteroatoms co-doped chitosan-derived carbon materials	N/O-doped chitosan and gelatin-derived carbon	331 F/g at 1 A/g	90% after 10,000 cycles	[48]
	N/S co-doped hierarchical porous carbons	272 F/g at 1.0 A/g	97.1% after 5000 cycles	[60]
	N/B co-doped porous carbon	316 F/g at 0.2 A/g	94.08% after 10,000 cycles	[61]
	N/O-doped porous carbon	367 F/g at 0.1 A/g	94% after 20,000 cycles	[51]
	O/N-doped carbon	355 F/g at 0.2 A/g	94% after 10,000 cycles	[53]
	O/N-doped wheat-straw-derived porous carbon	325 F/g at 0.5 A/g	96% after 10,000 cycles	[57]
Transition-metal oxide-doped chitosan-derived carbon materials	N/NiO <sub>x</sub> co-doped hierarchical porous carbons nanosheets	614.6 F/g at 1 A/g	85.61% after 10,000 cycles	[62]
	B, N-doped porous carbon with Co and Ni oxide nanoparticles	1266.7 F/g at 1 A/g	73.2% after 5000 cycles	[63]
	MnO <sub>2</sub> -doped holey carbon	460 F/g at 0.2 A/g	91.67% over 10,000 cycles	[64]
	Nitrogen-doped carbon-embedded NiFe <sub>2</sub> O <sub>4</sub> nanoparticles	958.33 F/g at 5 mV/s	91.25% after 6000 cycles	[65]

derived from chitosan, and the obtained nanocomposite exhibited good specific capacitance of 958.33 F/g at 5 mV/s, excellent cyclic stability of 91.25% after 6000 cycles, and energy density of 43.75 Wh/kg at a power density of 516.25 W/kg. The electrochemical performance of transition-metal-doped chitosan-derived carbon materials is generalized in Table 3. Compared to heteroatom co-doped carbon materials, the transition-metal oxides/heteroatom co-doped carbons tend to exhibit much higher capacitance indicating that the transition-metal oxides doping has higher efficiency to provide pseudocapacity, but the cycle stability are inferior for the restricted solid attachment between two species and the agglomeration of iron oxides nanoparticles.

The porous carbons from chitosan probably lack 3D interconnected porous framework. To improve the structure stability in the preparation of chitosan-derived carbon materials is a challenging task for extensive applications as electrode materials. Tan et al. [67] fabricated N-doped carbon using nanocomposites of chitosan and graphene quantum dots (GQDs) as the precursor, preserving the 3D interconnected frameworks due to GQDs, a zero-

dimensional carbon with extraordinary chemical stability and mechanical flexibility and delivering an ultrahigh specific capacitance of 545 F/g at 1 A/g and superior cyclic stability of 88.9% capacitance retention after 5000 cycles at 10 A/g as the electrode materials for supercapacitors. MOFs, a framework bridging metal ions/clusters with organic ligands, have been employed as self-sacrificial templates to prepare chitosan-derived nanoporous carbon with a desirable structure. Zhong et al. [68] prepared hierarchically porous carbon by pyrolysis of MOF-5/chitosan composites followed by a wash using acid, and it exhibited a high ratio of meso-/macropore volume to micropore of 1.47 and an outstanding specific surface area of 2375 m<sup>2</sup>/g, as well as a good specific capacitance of 199.9 F/g. Liu et al. [69] synthesized carbon nanosphere@vanadium nitride derived from MOF of chitosan as carbons source, NH<sub>4</sub>VO<sub>3</sub> and F127, and the prepared nanohybrid material delivered a high specific capacitance of 300.4 F/g. The hard template strategy is an important way to enhance the structure stability during the pyrolysis of chitosan precursor utilizing silica, metal oxide, and advanced zeolite imidazole frameworks

(ZIF). Liu et al. [70] prepared honeycomb-like hierarchically porous carbon derived from chitosan with nanosized SiO<sub>2</sub> particles as hard template for mesopores by a one-pot template removal utilizing polytetrafluoroethylene binder and carbonization process. The as-prepared porous carbon had superior specific surface areas of 1011 m<sup>2</sup>/g and a high specific capacitance of 250.5 F/g at 0.5 A/g in 6 mol/L KOH electrolyte in a three-electrode system, and the assembled supercapacitor exhibited outstanding cycling stability with capacity enhanced to 107% after 5000 cycles at 10 A/g. Li et al. [71] used chitosan cryogel as basic material added ZIF-7 and carbonized it at 950 °C obtaining coral structural ZIF-7-coated chitosan fibers with an internal resistance lower than 2 Ω and a high specific capacitance of 173.1 F/g.

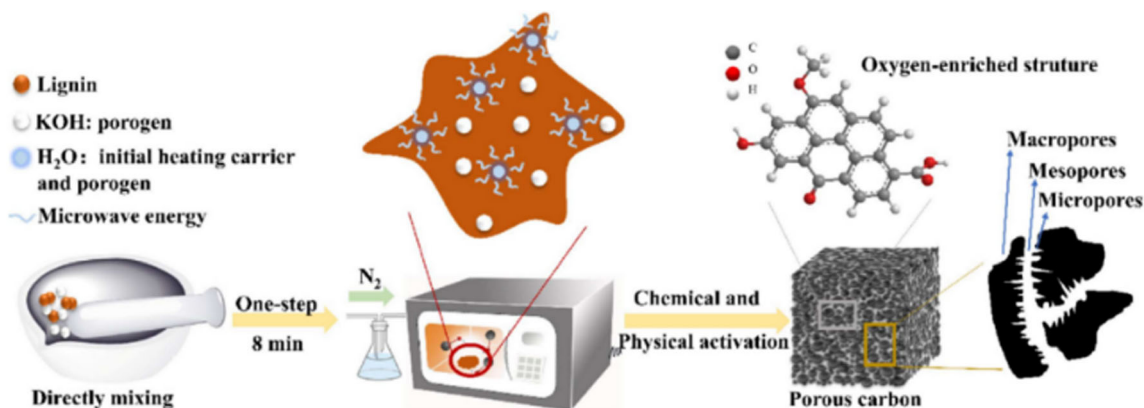
In recent years, flexible electrode materials attracted more and more attention for flexible supercapacitors with excellent electrochemical performance when repeatedly folded. Chitin/chitosan-derived carbon materials have been explored for flexible electrode. Yue et al. [72] developed a carbon fibrous film through direct carbonization of chitin fiber cloth, possessing polyhedron carbon-scale stacking microstructure and mechanically foldable flexibility as well as specific capacitance of 114.9 F/g at 1 A/g and outstanding cycling stability with capacitance retention about 94.0% after 5000 cycles at 1 A/g. Liu et al. [73] fabricated a foldable composite electrode containing N-self-doped carbon sheet derived from chitosan, microfibrillated cellulose fiber framework, and in situ PANI, which could perform well when folded with a high specific capacitance of 1688.8 mF/cm<sup>2</sup> (139.6 F/g), long-term cycling stability more than 84% capacitance retention after 5000 cycles, and an energy density of 11.75 mW·h·cm<sup>-3</sup> at the power density of 25 mW·cm<sup>-3</sup>.

### Lignin-derived carbon for supercapacitor electrode

Lignin is the second most abundant biomass on earth after cellulose as a main constituent (15%–30% by weight) of plant skeleton and byproduct of the pulp and papermaking process, with advantages of low cost, ready availability, and renewability. Unlike cellulose and hemicelluloses, it possesses three-dimensional structures consisting of methoxylated phenylpropane composed of sinapyl, coniferyl, and p-coumaryl alcohols, leading to its very

stable molecular structure that is hard to break down into available carbon precursors for electrodes. Much progress has been made to prepare lignin-derived carbon as supercapacitor electrode materials, such as activated carbon and porous carbon-based materials, commonly using a two-step approach (carbonization and activation). Zeng et al. [74] prepared porous activated carbon from lignin and carbonaceous mudstone mixture by activation treatment with HNO<sub>3</sub>, exhibiting a high specific capacitance of 110.1 F/g at 1 A/g which is three times of the untreated carbon electrode. Zhang et al. [75] proposed a novel bacterial activation method to depolymerize the macromolecular lignin through removing numbers of thermally stable G units and side chain structures, resulting in more developed pore structures and graphitization during the carbonization, and thus the obtained porous carbon exhibited an excellent specific capacitance of 428 F/g at 1 A/g and good cycling stability with a capacitance retention of 96.7% at 5 A/g after 10,000 cycles.

The biomass containing a large amount of lignin is widely used as a raw material of porous carbons for electrodes without extraction process. Tea waste has high content (70–80%) of lignin, cellulose, hemicelluloses, and polyphenol, with rich protein as nitrogen-containing compounds. It has been used to fabricate activated carbon with high graphitization degree via carbonization and activation with KOH, showing excellent cycle stability with capacitance retention 96.66% after 16,000 cycles at 4 A/g and ultrahigh power density of 33,494.70 W/kg at an energy density of 19.45 Wh/kg [76]. Tisawat et al. [77] prepared activated carbon from coconut shell functionalized with quinone derivatives to provide pseudocapacitance behavior via solvent-free method, exhibiting an outstanding specific capacitance of about 485 F/g at 1.0 A/g, low resistance of 2.25 Ω and excellent cycling performance with a capacitance loss of 1.2% after 1000 cycles. Wang et al. [78] obtained ultrathin graphene-like porous carbon nanosheets from sustainable pine bark precursors with potassium formate as nontoxic activation agent, which possessed very small thickness of about 3.8 nm, high mesopore proportion and a huge specific surface area of 1816.4 m<sup>2</sup>/g, as well as good capacitive performance in supercapacitor with a specific capacitance of 128.1 F/g at 1 A/g, excellent rate capability of 76.3% at 40 A/g, and outstanding cycling stability with only 6.1% capacitance loss over



**Figure 5** Schematic illustration of synthesis of enzymatic hydrolysis lignin-derived porous carbon [81]. Reproduced from Direct Microwave Conversion from Lignin to Micro/Meso/

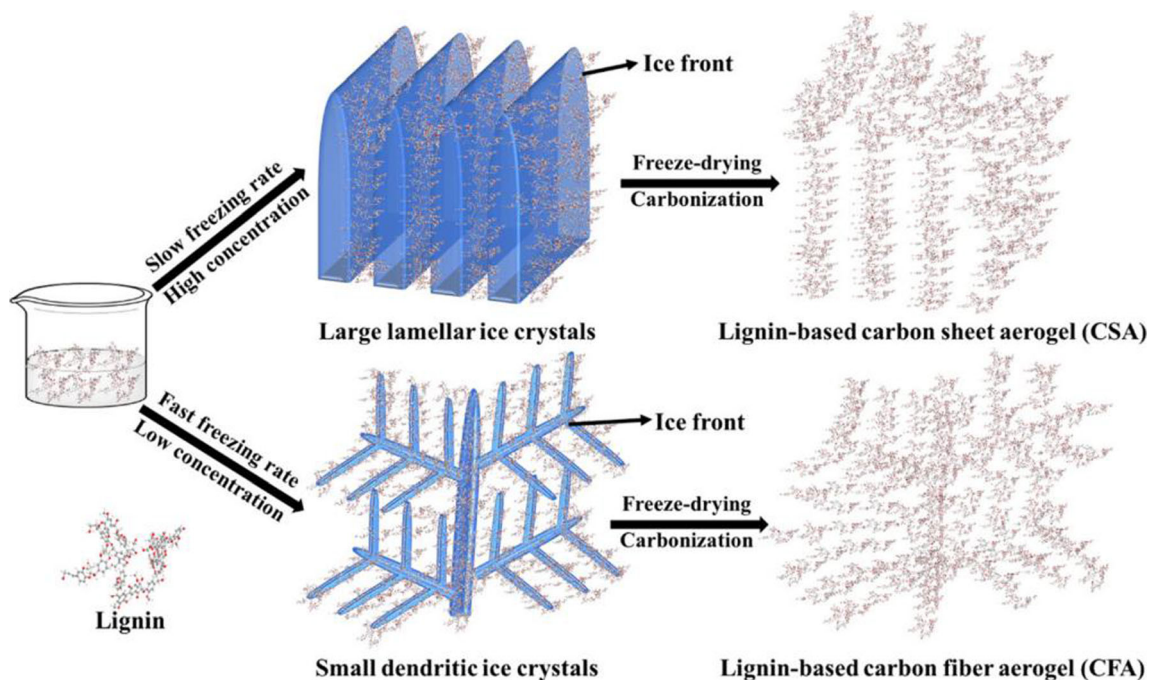
Macroporous Carbon for High-Performance Symmetric Supercapacitors, Xiaoyan Zhou, Weimin Chen, Pei Yang, et al., 6, 18, 2019 with permission from ChemElectroChem [58].

10,000 cycles in TEABF<sub>4</sub>/AN electrolyte. Zou et al. [79] produced nitrogen-doped hierarchical porous carbon from biomass bamboo fungus by two-step pyrolysis processes, which was performed in a large specific surface area of 1708 m<sup>2</sup>/g and high mesopore percentage about 25.5% along with a high specific capacitance of 228 F/g.

The traditional two-step approach tends to require a long production duration up to 16 h and produce microporous carbons with undeveloped mesopore/macropore structure. More effective one-step approaches have been developed to produce lignin-derived carbon electrode materials. Sun et al. [80] made lignin self-assemble onto the pore internal surface of the wood as well as the surface of Ni<sup>2+</sup> catalyst and activator crystals, and completed catalytic graphitization, Na<sub>2</sub>CO<sub>3</sub> activation, and Ni-doping synchronously. The obtained electrode materials had a specific capacitance of 163.7 F/g. Microwave heating combined with humidified nitrogen is proposed as a one-pot method to prepare lignin-derived porous carbons with oxygen-enriched hierarchical structures. Wang et al. [81] synthesized high-performance porous carbon from enzymatic hydrolysis lignin directly via only of microwave heating in 8 min with solid KOH to make sufficient micropores and to reduce the time-consuming drying process (Fig. 5). The fabricated carbon electrode showed a high specific capacitance (338 F/g at 1 A/g) and capacitance retention of 86% at 10 A/g, and the assembled supercapacitor delivered an energy density of 17.1 Wh/kg in Na<sub>2</sub>SO<sub>4</sub> electrolyte. Chen et al. [82] converted lignin into hierarchical porous

carbon via microwave heating in humidified N<sub>2</sub>, using KOH as the chemical activating agent. The obtained porous carbon had hierarchical pore distribution with a mesopore ratio of 65.8% and a surface oxygen content of 16.5%. They also combined microwave heating in humidified N<sub>2</sub> with water vapor plasma modification to prepare lignin-derived hierarchical porous carbon exhibiting a high specific capacitance (254.6 F/g at 0.5 A/g) and an excellent rate capability (retention rate of 75.6% at 10 A/g), in which water vapor plasma modification led to further development of the porosity, 11.6% increase in specific surface area, and 33.43% oxygen-doping [83].

There are several reports on carbon aerogels (CA) derived from lignin with well-developed porous structures and excellent conductivity. Cai et al. [84] prepared 3D structural CA via dissolving lignin and cellulose in BMIMCL, regenerating in deionized water, freeze-drying, and carbonization, showing a specific capacitance of 166 F/g at 0.1A/g and outstanding capacitance retention of 98.6% after 1000 cycles. Zhang et al. [85] produced lignin-derived carbon aerogels by ultrafast freezing of lignin/KOH solutions, freeze-drying, and in situ activation (Fig. 6), possessing abundant mesopores, macropores, and oxygenated groups, a high specific surface area (1681.6 m<sup>2</sup>), and a good specific capacitance (189 F/g at 1 A/g), a high energy density (26.25 Wh/kg at 1000 W/kg), and excellent capacitance retention (97.4% after 10,000 cycles). Nevertheless, there are a small number of reports on the composite membrane from lignin used as electrodes. Wang et al. [86] transformed lignin into porous conductive carbon



**Figure 6** Schematic illustration of the preparation of lignin-derived carbon-sheet aerogel (CSA) and carbon-fiber aerogel (CFA). Adapted with permission from Ultrafast-Freezing-Assisted Mild Preparation of Biomass-Derived, Hierarchically Porous,

Activated Carbon Aerogels for High-Performance Supercapacitors, Youfang Zhang, Chenyang Zhao, Wee Kit Ong, et al., 7, 1. Copyright (2019) American Chemical Society [85].

structures by femtosecond laser direct writing and prepared a composite membrane by mixing MoS<sub>2</sub> into the lignin/PAN composite for supercapacitors exhibiting a high specific capacitance of 16 mF/cm<sup>2</sup> at 10 mV/s. Cui et al. [87] synthesized MnO<sub>2</sub> nanoparticles on a RGO/lignin-based porous carbon composite membrane by electrodeposition for supercapacitor electrode, showing a mass specific capacitance of 135 F/g and an area specific capacitance of 1136 mF/cm<sup>2</sup> at 1 mA/cm<sup>2</sup> attributed to the conductive pathway of the porous carbon framework and the large surface area of the MnO<sub>2</sub> nanoparticles. These reports show the potential of a composite membrane containing lignin-derived carbon for supercapacitor electrodes.

Carbon nanofibers (CNFs) have an ultra large accessible surface area, structural stability, and excellent ion conductivity to be applied for supercapacitor electrodes. Recently, the electrospinning method is proposed for preparing carbon nanofibers. It has the advantages of low cost, convenient operation, simple equipment, and continuous production compared with traditional preparation methods such as self-assembly, hydrothermal synthesis, and

chemical vapor deposition. However, lignin is poorly soluble and refractory with low plasticity limiting its application in electrospinning and calling for liquefaction technology to convert it into electrospinning precursor liquid. Jiang et al. [88] converted lignin to liquefied carbon by mixing it with phenol and liquefied at 160 °C, and then doped it into polyacrylonitrile as an electrospinning precursor to prepare activated carbon nanofibers by electrospinning and high-temperature activation, exhibiting a good specific capacitance of 273 F/g at 0.5 A/g and a capacitance retention of 210 F/g under 10 A/g. Polyacrylonitrile (PAN) is commonly selected to blend with lignin as electrospinning precursor for its high char yield, easy spinnability, high mechanical strength, and good flexibility. Jayawickramage et al. [89] prepared carbon nanofibers for electrodes from electrospun blends of PAN with lignin through thermal stabilization, carbonization, and activation by CO<sub>2</sub>, and the assembled supercapacitor exhibits a specific capacitance of 128 F/g and an energy density of 59 Wh/kg at a power density of 15 kW/kg in an ionic liquid electrolyte. Dai et al. [90] prepared N, S co-doped graphene-modified lignin/PAN-based

**Table 4** The electrochemical properties of lignin-derived carbon materials

Types of carbon materials	Carbon materials	Specific capacitance	Rate capability	Cycle stability: capacitance retention	References
Activated carbon	Activated carbonaceous mudstone and lignin-derived carbons	110.1 F/g at 1 A/g	68.1% at 5 A/g	Above 40% after 5000 cycles	[74]
	Ultrathin graphene-like porous carbon nanosheets from pine bark	128.1 F/g at 1 A/g	76.3% at 40 A/g	93.9% over 10,000 cycles	[78]
	Activated carbon from tea waste	167 F/g at 1 A/g	81.42% at 30 A/g	96.66% after 16,000 cycles	[76]
	Hierarchical porous carbon from lignin	254.6 F/g at 0.5 A/g	75.6% at 10 A/g	94% after 20,000 cycles	[83]
Carbon aerogels	3D structural CA from lignin and cellulose	166 F/g at 0.1 A/g	Above 50% at 20 A/g	98.6% after 1000 cycles	[84]
	Lignin-derived CA	189 F/g at 1 A/g	55.02% at 20 A/g	97.4% after 10,000 cycles	[85]
Carbon nanofibers fabricated by electrospinning	CNFs from blends of PAN with lignin	128 F/g at 1 A/g	–	75% after 1000 cycles	[89]
	N, S co-doped graphene-modified lignin/PAN-based CNFs	267.32 F/g at 1 A/g	–	96.7% after 5000 cycles	[90]
	CNFs from liquefied lignin-derived carbon	273 F/g at 0.5 A/g	76.9% at 10 A/g	92.76% after 5000 cycles	[88]
	Lignin and phosphoric acid-derived CNFs	346.6 F/g at 0.1 A/g	82.3% at 5 A/g	–	[92]

CNFs by electrospinning, carbonization, and activation, during which graphene could capture  $\text{NH}_3$ ,  $\text{SO}_2$ , and HCN released from lignin and PAN to increase the heteroatoms content. The lignin/PAN CNFs-based supercapacitor exhibited a high specific capacitance of 267.32 F/g and an excellent cycling stability with a capacitance retention of 96.7% after 5000 cycles in KOH electrolyte. Yun et al. [91] fabricated ZnO doping CNFs from lignin, pitch, PAN, and zinc acetate by one-step electrospinning and heat treatment without any activation, in which pitch increased the carbon yield and conductivity and reduced the cost. Nevertheless, PAN is expensive and nonrenewable, restricting its further application of green and bio-renewable CNFs. Thus, developing alternative bio-renewable materials for CNFs, such as cellulose, is worth studied to produce CNFs by electrospinning technology. Cao et al. [92] used phosphoric acid to improve the flexibility and spinnability of lignin through the esterification reducing hydrogen bond interaction between lignin molecules, and further prepared CNFs from modified lignin and cellulose-acetate with high flexibility,

possessing high surface areas, good flexibility, and high specific capacitance of 346.6 F/g, which is a bio-renewable CNF for electrodes. The lignin-derived carbon materials mentioned above are listed in Table 4, including normal activated carbon, CA, and CNFs. As shown in Table 4, the CNFs prepared by electrospinning methods deliver superior specific capacitance along with excellent rate capability, which benefits from larger accessible surface area of nanofibers and stable porous structure brought by electrospinning.

Kraft lignin differs from native lignin in chemical structure and its complexity is preserved and even enhanced in the extraction process. During the Kraft process, cellulose and hemicellulose are separated from lignin in a digester with a strong base and sodium sulfite under high temperatures and pressures. Carbon electrode materials from kraft lignin gain increasing attention for its plentiful heteroatoms. Liu et al. [93] prepared the O, N, and S co-doped hierarchical porous carbons via direct self-activated pyrolysis of kraft lignin, possessing 3D highly interconnected structure with high specific surface areas

(1307 m<sup>2</sup>/g), plentiful oxygen, nitrogen, and sulfur co-doping (9.84–19.91 wt%), and the assembled symmetric supercapacitor showed a high specific capacitance (244.5 F/g at 0.2 A/g), good rate capability (initial capacitance retention of 81.8% at 40.0 A/g), and excellent cycling stability (retention of 91.6% over 10,000 cycles). Schlee et al. [8] developed purely Kraft lignin-derived carbon fiber mats by electrospinning, stabilization in air, subsequently carbonization in N<sub>2</sub>, and an activation in CO<sub>2</sub>, and the carbon fiber mats possessed many oxygen functional groups, a specific capacitance of 155 F/g at 0.1 A/g and good cycle stability with retention of 94% after 6000 cycles in KOH electrolyte.

Lignosulfonate is a water-soluble derivative of lignin and a cheap bio-material as wastes of the sulfite pulping process, providing sulfur self-doping for supercapacitor electrodes. Tian et al. [94] developed multilevel porous carbon from sodium lignosulfonate with polyaniline-coated polystyrene spheres as the nitrogen source and template of macropores, KOH as activation agent, and the multilevel porous carbon showed a high specific surface area of 1454.7 m<sup>2</sup>/g, high nitrogen and sulfur content of 2.1 and 4.3 wt% respectively, and the fabricated supercapacitor delivered a high specific capacitance of 269 F/g and excellent cycling stability with capacitance retention of 98.4% after 10,000 cycles. Zhang et al. [95] prepared compact biochar-based materials for electrodes from sodium ligninsulfonate, p-phenylenediamine, and graphene oxide, through the evaporation process at 90 °C and low-temperature pyrolysis treatment at 400 °C, showing ultra-high specific capacitance of 480 F/g at 1 A/g and good cycling stability in 1 M H<sub>2</sub>SO<sub>4</sub> electrolyte.

## Other biomass-derived carbons

### *Alginate-derived carbon*

Alginate, a polysaccharide linear anionic biopolymer consists of β-D-mannuronate (M) and α-L-guluronate (G) residues, can be easily extracted from brown algae and bacteria with cheap price, nontoxicity, environmentally friendliness, and large production about 30,000 metric tons annually. Besides, alginate possesses rich oxygen-containing functional groups such as carboxyl and hydroxyl groups and superior carbon content to be used as an ideal O-self doped carbon precursor for supercapacitor electrodes with a

self-activation effect. Sun et al. [96] constructed 3D interconnected porous carbons derived from potassium alginate via self-activation process, and the porous carbons delivered a high specific capacity of 279 F/g at 1 A/g and excellent capacitance retention of 96.6% after 10,000 cycles for its high oxygen content, large surface area, and 3D interconnected structure. Sodium alginate (SA) has been widely studied as precursor for carbon electrodes since it is easily obtained from marine biomass and displays a 3D hierarchical porous framework after carbonization. Wang et al. [97] employed sodium alginate as carbon precursor and KNO<sub>3</sub> as the oxidant by integrated gunpowder chemistry-assisted exfoliation and KOH activation, obtaining a porous carbon nanosheet with the ultrahigh specific surface area of 2788 m<sup>2</sup>/g, a specific capacitance of 226.9 F/g at 1 A/g, excellent cycling stability of 94% capacitance retention after 10,000 cycles, and well-done energy density of 65.6 Wh/kg at 43.75 kW/kg in ionic liquid electrolyte. Bai et al. [98] fabricated hierarchical oxygen-enriched porous carbon by carbonization of SA and BC composite followed by KOH activation at 700 °C for high-performance supercapacitor electrodes exhibiting outstanding specific capacitance of 302 F/g at 0.5 A/g, a good capacitance retention of 75.2% at 10 A/g, and high cycling stability with a 93.8% capacitance retention in 6 M KOH electrolyte due to the macro/mesoporosity and massive oxygen functional groups.

As mentioned above, the co-doping of two heteroatoms, such as N,O co-doping, N,S co-doping, has been employed to further improve the electrochemical performances of alginate-derived carbons due to the synergistic effect. Zhao et al. [99] prepared crosslinked SA gel via the electrostatic interaction between carboxylate in SA and cationic diammonium obtained from diammonium cations reacting with hydrochloric acid, and carbonized the SA gel particles obtaining N–O co-doped hierarchically porous carbon with an ultrahigh specific area of 3794 m<sup>2</sup>/g, high specific capacity of 269.0 F/g at 1 A/g, and superior cycling stability of 92.09% after 5000 cycles at 5 A/g in the three-electrode system. Hu et al. [100] prepared N,O co-doped hierarchically porous carbons by freeze-drying-assisted carbonization of SA and urea as porogenic agent and nitrogen source without any other activator, which exhibited a high specific surface area of 1179 m<sup>2</sup>/g, desirable oxygen and nitrogen contents of 13.8 wt% and 6.7 wt%, high capacitance of 305 F/g at 1 A/g, and well-done rate



ability of 73.8% retention at 20 A/g. Huang et al. [101] fabricated polyaniline (PANI)/SA hydrogel using hydrogen bonding and entanglement between SA and PANI and then carbonized it to form N,S co-doped hierarchically porous carbon (NSC), in which PANI served as the N source and the SA provided massive oxygen and facilitated the porosity by  $\text{Na}^+$ , and ammonium sulfate mineral from oxypolymerization served as N,S-enriched active agent. The NSC had high contents of oxygen (12.8 atom%), nitrogen (8.4 atom%) and sulfur (1.4 atom%) exhibiting excellent specific capacitance of 309 F/g at 0.5 A/g. Ye et al. [102] prepared 3D interconnected hierarchical porous activated carbon aerogels from SA and PANI by oxypolymerization, freeze-drying, and carbonization/activation at 700 °C, possessing high surface areas of 1695 m<sup>2</sup>/g and proper content of O (13.76%) and N (2.84%) and ultrahigh specific capacitance of 342 F/g at 2 A/g in 3 M H<sub>2</sub>SO<sub>4</sub> electrolyte. Cui et al. [103] also used urea as nitrogen source adding KOH as activating agent to large-scale produced nitrogen-doped SA-derived nanocarbons displaying interconnected porous, ultrahigh specific surface area of 3313 m<sup>2</sup>/g, abundant nitrogen (about 7.2 wt%) and oxygen (about 7.4 wt%) elements, high specific capacitance of 267 F/g at 1 A/g and outstanding capacitance retention of 76.8% at 100 A/g.

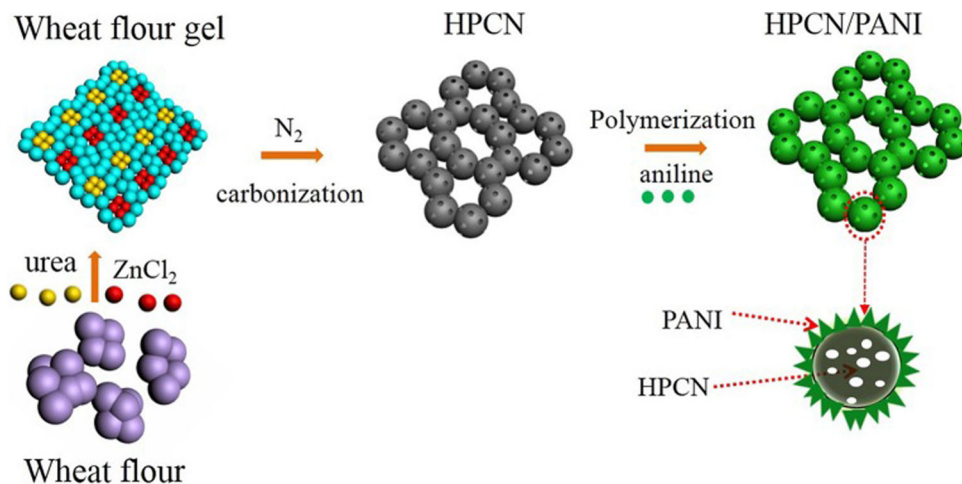
Iron oxides are proved to be a promising material to fabricate hybrid composites with carbon for supercapacitor electrodes because of its high theoretical capacitance. Conventional methods such as thermal decomposition, hydrothermal decomposition, chemical precipitation and co-precipitation have been widely applied. Wang et al. [104] synthesized MnO<sub>2</sub> nanoflakes on Ni foam by chemical deposition, then coated a thin layer of SA via electrophoretic deposition and calcined them into carbon enhancing the conductivity and protecting the crystal from dissolving. The as-prepared composite exhibited an ultrahigh specific capacitance of 412 F/g at 1 A/g and the excellent capacitance retention of 95.7% after 10,000 cycles. Conventional methods are based on deposition restricting solid attachment between two species resulting in poor durability for the agglomeration of iron oxide nanoparticles during charging-discharging processes. Inspiringly, the  $-\text{COO}^-$  groups in the G units of SA possess a strong capability to chelate various metal ions such as  $\text{Co}^{2+}$ ,  $\text{Ni}^{2+}$ ,  $\text{Cu}^{2+}$ ,  $\text{Cr}^{3+}$ , and  $\text{Fe}^{3+}$  by ion-exchange process forming an “egg-box” model structure that can be

used to synthesize hybrid composites. Zhai et al. [105] prepared Cu-doped carbon aerogels (Cu-CA) via freeze-drying and one-step carbonization of hydrogels formed by chelating SA with metal cations  $\text{Cu}^{2+}$ , and the Cu-CA had a hierarchical nanostructure, uniform Cu doping, and an outstanding specific capacitance up to 414.4 F/g with a potential voltage window of -0.6–0 V maintaining at 224.8 F/g after 1000 cycles in the solution of 6 M KOH. Li et al. [106] used SA to coordinate with  $\text{Fe}^{3+}$  via chelation effect preparing a carbon and Fe<sub>3</sub>O<sub>4</sub> composite by the sol-gel method, which had two-dimensional sheet-like structure, and nanosize and homogeneously distributing Fe<sub>3</sub>O<sub>4</sub> particles on the surface as well as excellent electrochemical performance as a supercapacitor electrode with an outstanding specific capacitance of 550 F/g at 1 A/g and good capacitance retention of 89% after 2000 cycles.

#### *Starch-derived carbon*

Starch is a polysaccharide comprising branched amylopectin with  $\alpha$  (1 → 4),  $\alpha$  (1 → 6) glycosidic linkages and helical amylose with  $\alpha$  (1 → 4) glycosidic linkages and contains 49% of oxygen. It is cheap, renewable, biodegradable and environmentally friendly, and it can be produced from a variety of crops allowing high application prospects as precursor to derive porous carbon materials for supercapacitors. Gao et al. [107] used starch-derived carbon foam material as negative electrode and hierarchical porous sphere MnCo<sub>2</sub>O<sub>4</sub> as positive electrode fabricating asymmetric supercapacitor displaying excellent specific energy of 42.27 Wh/kg at the specific power of 400 W/kg under an output potential difference of 1.6 V and good cycling stability. Cao et al. [108] synthesized hierarchical porous carbons (HPCs) employing starch magnesium nitrate raw materials via an exothermic pyrolysis, high-temperature thermal treatment, and subsequent acid washing, in which magnesium nitrate acted as an oxidant and provided MgO template removed by acid for creating micro- and mesopores. The HPCs had a huge specific surface area about 2300 m<sup>2</sup>/g and delivered superior specific capacitance of 229 F/g at 1 A/g, good rate capability with capacitance of 211 F/g at 10 A/g, and excellent cycling stability with capacitance retention of 94% after 10,000 cycles at 2 A/g in a 6 M KOH electrolyte. The iron oxides doping can improve the electrochemical performance

**Figure 7** Schematic illustration of synthesizing HPCN/PANI composites. Reproduced from Construction of Ultrathin Nitrogen-Doped Porous Carbon Nanospheres Coated With Polyaniline Nanorods for Asymmetric Supercapacitors, Pingping Yu, Qunliang Wang, Lingxia Zheng, and Yanfeng Jiang, 7, 2019 with permission from Front Chem [116].



of starch-derived carbon in the same way as mentioned. Samdani et al. [109] synthesized the composite of nanostructured Co<sub>3</sub>O<sub>4</sub> and bio-inspired carbon (BIC) via a pyrolysis of corn starch and subsequently hydrothermal synthesis, showing ultrahigh specific capacitance of 473 F/g at 1 A/g as electrode material and assembling all-solid-state symmetric supercapacitor with an energy density of 17 Wh/kg at a power density of 184 W/kg.

Starch grains tend to swell, split, and form a homogeneous gelatinized solution in water at an appropriate temperature (generally 60–80 °C), which is called gelatinization accompanied by the disconnection of hydrogen bonds between crystalline and amorphous starch molecules. Based on this, the sol-gel method is appropriately applied to the preparation of starch-derived porous carbon-based electrodes. Liu et al. [110] prepared starch-based gel by mixing starch with the KOH solution at 85 °C and removing the water, then fabricated 3D-reticular porous carbon via the carbonization of the gel and the in situ activation of KOH solution, which delivered a high specific capacitance of 372 F/g at 0.5 A/g at a wide potential window of 0–2.6 V in 2 M KOH electrolyte and an outstanding energy density of 24.5 Wh/kg at a power density of 695 W/kg in 2 M [BMIm]BF<sub>4</sub>/AN electrolyte. Zhong et al. [111] fabricated starch-derived porous carbon-based electrode by the carbonization of carbon cloth soaked in the gel from the mixture of starch and KOH, and this binder-free electrode exhibited 272 F/g at 1 A/g and a capacitance retention of 75.9% at 50 A/g.

Recently, many kinds of starch with different origination, such as potato starch, sweet potato starch, and corn starch, have been studied as

precursors of porous carbon in supercapacitor electrodes with excellent performance. Wang et al. [112] fabricated 2D amorphous carbon nanosheets (CNs) via burning corn starch above a Ni foam and further coated CNs with a CoO/CoF<sub>2</sub> composite film significantly enhancing the electron collection efficiency, and the as-prepared electrode had a high specific capacitance of 12.00 F/cm<sup>2</sup> at 5 mA/cm<sup>2</sup>, superior rate capability, and good cycling stability. Xing et al. [113] employed three raw materials, taro, sweet potato, and potato as the carbon precursor to prepare starch-based porous carbons by carbonization and KOH activation, and found that degrees of graphitization of porous carbon was influenced by the amylopectin content in starch. The taro starch had the highest amylopectin content (84%) among three raw materials and taro starch-derived carbon possessed the highest degree of graphitization and the best electrochemical performance with a good specific capacitance of 397 F/g, outstanding capacity retention of 97% over 20,000 cycles, and a superior energy density of 22.59 Wh/kg at a power density of 148 W/kg in 6 M KOH electrolyte.

Chemical activation is a widely used approach to facilitate the porous structures of starch-derived carbon with the chemical agents such as KOH, NaOH, and ZnCl<sub>2</sub>. Guo et al. [114] prepared HPCs through one-step carbonization/activation of starch as precursors and KOH as activating agents, which possessed 3D interconnected hierarchically porous structure and an ultrahigh specific surface area of 2259.54 m<sup>2</sup>/g as well as an excellent specific capacitance of 385.7 F/g at 1 A/g in a three-electrode system. However, the strong activation effect of KOH and NaOH may degrade the starch into small

molecular and obtain a smaller proportion of carbon atoms resulting in the low yield of carbon. Compared to KOH,  $\text{ZnCl}_2$  is a milder activation reagent to create hierarchical pores and high yield of carbon materials by dehydrating and crosslinking reaction with precursor when temperature increase. Kasturi et al. [115] employed *Artocarpus heterophyllus* seed as carbon precursor containing a rich source of starch, vitamin, minerals, phytonutrients, and protein, and  $\text{ZnCl}_2$  as activating agents to produce biomass-derived material possessing abundant nitrogen species (2.1 at. wt%), high specific surface area (1841  $\text{m}^2/\text{g}$ ), and exceptional porous architecture. The coin cell all-solid-state supercapacitor with the as-prepared electrode and *Manihot esculenta* starch-based biopolymer electrolyte showed an outstanding specific capacitance of 240 F/g at 0.5 A/g with a coulombic efficiency of 97% after 2000 cycles and superior specific energy of 17 Wh/kg at a power density of 3823 W/kg. Yu et al. [116] synthesized ultrathin N-doped porous carbon nanospheres/polyaniline (HPCN/PANI) composite by one-step pyrolysis of wheat flour, urea and  $\text{ZnCl}_2$  followed by in situ polymerization of aniline (Fig. 7). The HPCN/PANI possessed extremely high specific capacitance of 783 F/g and the assembled asymmetric supercapacitor with it as anode delivered a desirable specific capacitance of 81.2 F/g.

#### *Agar-derived carbon*

Agar is a kind of macromolecular polysaccharide with galactose as the main component, which can be extracted from red algae. It is low cost, easily available, biodegradable, and contains a certain amount of oxygen possessing potential as a carbon precursor to derive O-doped carbon-based electrode materials. Vijayakumar et al. [117] used agar as carbon source deriving carbon electrode material and utilized it in a symmetric supercapacitor which exhibited a high specific capacitance of 170 F/g at 0.5 A/g, good rate property with 70 F/g capacitance at 50 A/g, and cyclic stability with high capacitance retention about 85% after 15000 cycles in  $\text{Na}_2\text{SO}_4$  electrolyte. Hu et al. [118] synthesized N/O co-doped porous carbon from the mixture of agar, potassium citrate, and urea through one-step pyrolysis. The prepared carbon delivered an excellent specific capacity of 357 F/g at 1 A/g in 6 M KOH electrolyte, good rate

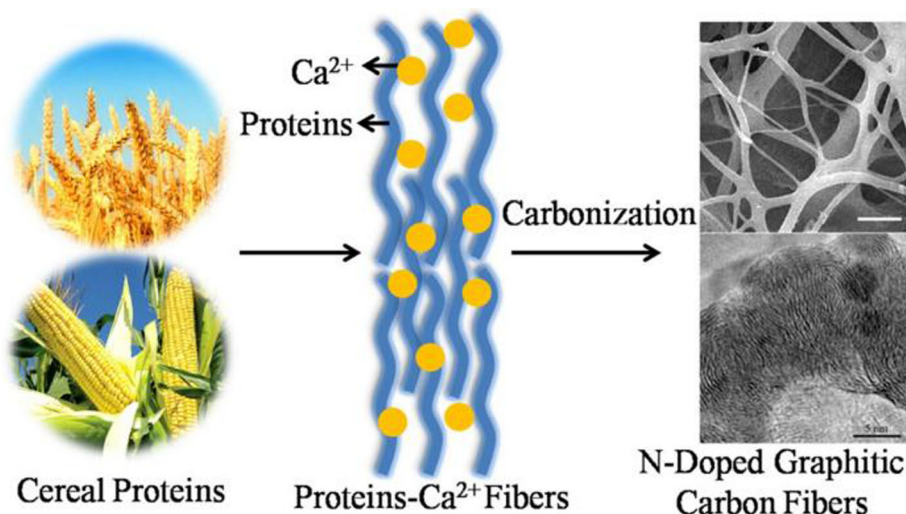
performance of 267 F/g at 50 A/g, and long-term stabilization of 95.6% retention after 10,000 cycles.

Agar can form a stable hydrogel to stabilize the dispersed state of nanoparticles and the multitudinous  $-\text{OH}$  groups of agar can make strong intermolecular interaction between them to avoid aggregation and shrinkage. Thus, it have been used to compound with various nanoparticles. Xie et al. [119] prepared a carbon material through carbonization of agar with graphene oxide nanosheets (GONs). Agar hydrogel could not only fix the monolayer-dispersed state of GONs but also provided abundant  $-\text{OH}$  groups strong interacting with the functional groups on GONs averting aggregation and shrinkage of the carbonized particles, resulting in the large surface area of about 1200  $\text{m}^2/\text{g}$  of the carbon materials exhibiting a good capacitance of 115 F/g, high rate performance with 70% at 80 A/g, and outstanding cycling stability retaining almost 100% after 10,000 cycles at 5 A/g. Huang et al. [120] used a mixture of agar and b-cyclodextrin (b-CD) as carbon precursor,  $\text{MnO}_2$  nanorods as the template to form nanotunnels inserted hierarchical-porous carbon, in which agar hydrogel stabilized the dispersed state of  $\text{MnO}_2$  nanorods and  $-\text{OH}$  groups of agar built strong interactions with b-CD. The resulted carbon material showed a high specific surface area of 1441  $\text{m}^2/\text{g}$ , excellent capacitance of 253.1 F/g at a scan rate of 5 mV/s, good rate performance of 80.3% at 100 mV/s, and cycling stability higher than 100% after 6000 cycles at 20 mV/s.

#### *Protein-derived carbon*

Despite the extensive application on converting carbohydrates into carbon electrode materials, proteins have not been widely studied in energy storage. Proteins contain abundant N and O atoms on their polypeptide chains as potential precursor of heteroatom-doped carbon with enhanced capacitance, surface wettability, and electronic conductivity. Demir et al. [121] used pea protein as precursor to fabricate N and O dual-doped porous carbons by two-step pyrolysis and KOH activation, featuring a large surface area of 3500  $\text{m}^2/\text{g}$ , high N/O contents of 2.5/17.9 at% as well as a remarkable capacitance of 413 F/g at 1.0 A/g in 1 M KOH electrolyte, good cycle stability of 91.6% after 20,000 cycles, and high rate capability of 210 F/g at 30 A/g. Song et al. [122] produced protein-derived N/O-doped hierarchically

**Figure 8** Schematic illustration of synthesizing N-doped graphitic carbon fibers. Reproduced from Highly nitrogen-doped graphitic carbon fibers from sustainable plant protein for supercapacitor, Jingqi Yang, Yixiang Wang, Jingli Luo, Lingyun Chen, 121, 2018 with permission from Industrial Crops and Products [125].



porous carbon materials from sericin, a waste protein, by pre-carbonization and KOH activation, which had excellent electrochemical performances as supercapacitor electrode with a high specific surface area of 2723 m<sup>2</sup>/g, a high specific capacitance of 287 F/g at 0.5 A/g, a good cyclic stability with 93% capacitance retained after 10,000 cycles at 5 A/g, and excellent rate capability with 74% capacitance retained at 0.5 A/g. Niu et al. [123] used commercial protein powder as precursor to obtain protein-based microporous carbon by KOH activation with high nitrogen content of 15.29 at.%, outstanding capacitance of 336 F/g at 1 A/g, good rate capacity of 173 F/g at 10 A/g, and cycling stability of 83% after 10,000 cycles at 10 A/g.

However, most proteins would lose their structure during carbonization since the polypeptide chains of protein become flexible above glass-transition temperature. To preserve their structure during carbonization is one of the biggest challenges in applying proteins as carbon precursors. The silica template has been introduced to preserve the hierarchically porous structure of N-doped carbon from protein [124], but its preparation is time and energy consuming and the hydrogen fluoride used to remove the template is corrosive and harmful for the environment. In another way, Yang et al. [125] incorporated Ca<sup>2+</sup> into the protein fiber matrix to obtain well maintained interconnected fibrous structure during pyrolysis (Fig. 8), in which Ca<sup>2+</sup> improved the thermal stability of the protein and the calcium salt/oxide served as a template to both generate pores and provided a solid support. The obtained carbon fibers showed a high areal specific

capacitance of 64 μF/cm<sup>2</sup> at 0.5 A/g and excellent cyclic stability with capacitance retention of 98% after 5000 cycles at 10 A/g.

Protein-derived carbons have been compounded with high-conductive graphene materials for higher performance in supercapacitors. Xie et al. [126] coated graphene with the soy-protein-derived N/O co-doped amorphous carbon forming a sandwiched porous architecture through a one-pot hydrothermal process followed by carbonization in an argon atmosphere and further thermally oxidation in air. The as-prepared material exhibited a high specific capacitance of 221 F/g at 0.2 A/g and retained 153 F/g at 20 A/g in 6 M KOH electrolyte as well as superior cycling performance with a retention of 96% after 2000 cycles at 2 A/g. Ma et al. [127] used graphene oxide sheets (GOs) as both template and catalyst for the hydrothermal carbonization of egg proteins and the resulting egg protein-derived carbon/rGO composite was activated by KOH to obtain highly porous carbon material, showing an outstanding specific capacitance of 482 F/g at 0.1 A/g, excellent rate capability of 280 F/g at 20 A/g and cycling stability about 100% capacitance retention after 10,000 cycles at 5 A/g.

Gelatin is fat-free, high-protein and cholesterol-free. As a noncrystalline mixture is obtained by irreversibly partial hydrolyzation of collagen in animal skin, connective tissues, and bones, it is low cost, abundant, sustainable, and rich of nitrogen as an idea precursor of nitrogen-doped carbon applied in supercapacitor electrodes. Niu et al. [128] prepared a ternary (O, N, S)-doped microporous carbon nanosheets by low temperature (600 °C) pyrolysis of

fish skin and KOH and the fish skin was firstly transformed into gelatin by KOH. The as-prepared carbon nanosheets delivered a high volumetric capacitance of  $300 \text{ F}\cdot\text{cm}^{-3}$  at the  $1 \text{ A/g}$  and maintained capacitance of  $235 \text{ F}\cdot\text{cm}^{-3}$  even at  $200 \text{ A/g}$ . Li et al. [129] fabricated a N-self doped 3D cross-coupled macro-mesoporous carbon network from the  $\text{NaNO}_3$ -gelatin biopolymer aerogel by salt-template strategy, which exhibited an ultrahigh specific surface area of  $2872.2 \text{ m}^2/\text{g}$ , high energy density of  $92 \text{ Wh/kg}$  at a power density of  $1 \text{ kW/kg}$  as electrode in  $4 \text{ V}$  EMIBF<sub>4</sub> ILs, and retained an energy density of  $39 \text{ Wh/kg}$  at an ultrahigh power density of  $200 \text{ kW/kg}$ . Hu et al. [130] prepared mesoporous nitrogen-doped carbon films via carbonization of gelatin/copper hydroxide nanostrands film. This free-standing carbon film was employed as a binder-free supercapacitor electrode showing superior specific capacity of  $316 \text{ F/g}$  at  $0.5 \text{ A/g}$ , and excellent capacitance retention of  $92.9\%$  after  $11,000$  cycles.

The composites of gelatin-derived carbon with iron oxide and graphene oxide perform better applied in supercapacitor electrodes. Zeng et al. [131] fabricated a nitrogen-doped hierarchical porous carbon derived from composite of gelatin and graphene oxide by a combined nano- $\text{CaCO}_3$  template, solvation, volatilization, and KOH activation, which possessed a large surface area of  $1091 \text{ m}^2/\text{g}$  and combination of electrical double layer with pseudocapacitance capacitance and delivered an excellent specific capacitance of  $234 \text{ F/g}$  at  $0.5 \text{ A/g}$ . Peng et al. [132] synthesized nickel phosphide nanoparticle self-inlaid carbon nanosheet frameworks using a nickel nitrate-assisted gelatin, in situ substrate-free chemical blowing pyrolysis and low-temperature phosphorization method, and the obtained carbon nanosheet as a positive electrode of supercapacitor exhibited an excellent specific capacity of  $145 \text{ mAh/g}$  at  $0.5 \text{ A/g}$ , a superior rate capability of  $70\%$  capacity retention at  $10 \text{ A/g}$ . Sun et al. [133] prepared composite electrode of nitrogen-doped hierarchical porous carbon and  $\alpha\text{-MnO}_2$  nanowires via a template method, had structures of inserting one-dimensional nanowires into 3D interconnected porous, outstanding specific capacitance of  $204.6 \text{ F/g}$  at  $1 \text{ A/g}$  and well-done cycling stability with  $97.2\%$  retention after  $5000$  cycles. Yang et al. [134] used gelatin and melamine as the carbon sources, cobalt acetate as the source of  $\text{Co}_3\text{O}_4$  obtaining 2D carbon nanosheets with cobalt oxide nanoparticles on the surface ( $\text{GM-C@Co}_3\text{O}_4$ )

through the hydrothermal reaction and carbonization. The morphology of  $\text{GM-C@Co}_3\text{O}_4$  composite could be tuned by the amount of cobalt acetate.  $\text{GM-C@Co}_3\text{O}_4$  displayed an ultrahigh specific capacitance of  $1503 \text{ F/g}$  at  $1 \text{ A/g}$  and assembled  $\text{GM-C@Co}_3\text{O}_4//\text{GM-C}$  supercapacitor exhibited an energy density of  $48.4 \text{ Wh/kg}$  at a power density of  $1502 \text{ W/kg}$ .

Table 5 displayed the electrochemical performance of porous carbon-based electrode materials derived from alginate, starch, or gelatin. They are classified into three types: porous carbons without external doping, heteroatoms co-doped porous carbons, and composites of transition-metal oxides/conductive polymers with carbon materials to show the benefit brought by two main methods on improvement of electrochemical performance. Both external heteroatom doping and composite with transition-metal oxides/conductive polymers are proved to significantly enhance the specific capacitance of carbon materials through offering pseudocapacitance and the latter method performs much better. Nevertheless, the rate capability of composites is lower than other two types for the relatively slow ion transfer caused by restricted solid attachment between two species. Besides, the composites exhibited inferior cycle stability caused by complex factors related to structural integrity, electronic properties, and electrochemical behavior of pseudocapacitive materials. The phenomenon is applicable to other biomass-derived carbon materials combined with the data reports and data in the first few parts. Thus, external heteroatom doping is a more comprehensive approach to improve the electrochemical properties of biomass-derived carbon materials, and composites need further improvement on the rate capability and cycle stability through changing the structure, composition, and so on.

### Biomass applied in supercapacitor electrolytes

Supercapacitors are widely assembled with liquid electrolytes, but the liquid electrolytes are usually toxic, corrosive, and have a risk of leakage. One of the methods to overcome the problem is the formation of polymer electrolytes with a wide potential window, high ionic conductivity, excellent stability, and mechanical strength. However, the formation of

**Table 5** The electrochemical properties of alginate/starch (bold parts)/gelatin (italics parts)-derived carbon materials

Types of carbon materials	Carbon materials	Specific capacitance	Rate capability	Cycle stability: capacitance retention	References
Porous carbon without external doping	Potassium alginate-derived 3D interconnected porous carbons	279 F/g at 1 A/g	71.7% at 50 A/g	96.6% after 10,000 cycles	[96]
	SA-derived porous carbon nanosheet	226.9 F/g at 1 A/g	68% at 50 A/g	94% after 10,000 cycles	[97]
	<b>Starch magnesium nitrate-derived HPCs</b>	<b>229 F/g at 1 A/g</b>	<b>92.1% at 10 A/g</b>	<b>94% after 10,000 cycles</b>	[108]
	<b>Starch-derived 3D-reticular porous carbon</b>	<b>372 F/g at 0.5 A/g</b>	<b>64.5% at 10 A/g</b>	<b>93.2% after 10,000 cycles</b>	[110]
	<b>Starch-derived porous carbon</b>	<b>272 F/g at 1 A/g</b>	<b>75.9% at 50 A/g</b>	<b>97% after 10,000 cycles</b>	[111]
	<b>Starch-derived HPCs</b>	<b>385.7 F/g at 1 A/g</b>	<b>78.72% at 20 A/g</b>	<b>About 100% after 60,000 cycles</b>	[114]
Heteroatoms co-doped porous carbon	<i>Gelatin-derived carbon nanosheets</i>	<i>300 F/cm<sup>3</sup></i>	<i>78.3% at 200 A/g</i>	<i>About 100% after 20,000 cycles</i>	[128]
	N,O co-doped SA-derived porous carbon	269.0 F/g at 1 A/g	74.7% at 7 A/g	92.09% after 5000 cycles	[99]
	N,O-doped porous carbons derived from SA and urea	305 F/g at 1 A/g	73.8% at 20 A/g	95.8% after 10,000 cycles	[100]
	N,S co-doped porous carbon from PANI/SA hydrogel	309 F/g at 0.5 A/g	63.4% at 10 A/g	90% after 3000 cycles	[101]
	O,N co-doped activated carbon aerogels from SA and PANI	342 F/g at 2 A/g	52.4% at 8 A/g	73.8% after 2000 cycles	[102]
	N,O co-doped SA-derived nanocarbons	267 F/g at 1 A/g	76.8% at 100 A/g	97% after 10,000 cycles	[103]
Composites of transition-metal oxides/conductive polymers with carbon materials	MnO <sub>2</sub> nanoflakes coated with a thin layer of SA	412 F/g at 1 A/g	About 66.7% at 20 A/g	95.7% after 10,000 cycles	[104]
	Cu-doped SA-derived carbon aerogels	414.4 F/g	45.6% at 1.5 A/g	54.2% after 1000 cycles	[105]
	SA-derived carbon and Fe <sub>3</sub> O <sub>4</sub> composite	550 F/g at 1 A/g	47.4% at 10 A/g	89% after 2000 cycles	[106]
	<b>Composite of nanostructured Co<sub>3</sub>O<sub>4</sub> and corn starch-derived carbon</b>	<b>473 F/g at 1 A/g</b>	<b>About 63.4% at 10 A/g</b>	<b>77% after 5000 cycles</b>	[109]
	<b>N-doped wheat flour-derived porous carbon nanospheres/PANI composite</b>	<b>783 F/g</b>	<b>About 70.2% at 10 A/g</b>	<b>89.3% after 5000 cycles</b>	[116]
	<i>Composite electrode of nitrogen-doped gelatin-derived porous carbon and alpha-MnO<sub>2</sub> nanowires</i>	204.6 F/g at 1 A/g	72.1% at 10 A/g	97.2% retention after 5000 cycles	[133]
<i>Gelatin and melamine-derived carbon@Co<sub>3</sub>O<sub>4</sub> composite</i>	1503 F/g at 1 A/g	86.5% at 20 A/g	83.3% after 5000 cycle	[134]	

polymer electrolytes can cause huge resource consumption and environmental concerns. Hence, developing renewable, low cost, environmentally friendly biomass-based electrolytes has gained more and more attention. Biomass materials, such as cellulose, starch, chitosan, alginate, and soybean protein, have been applied in supercapacitor electrolyte in recent years.

### Cellulose-based electrolytes

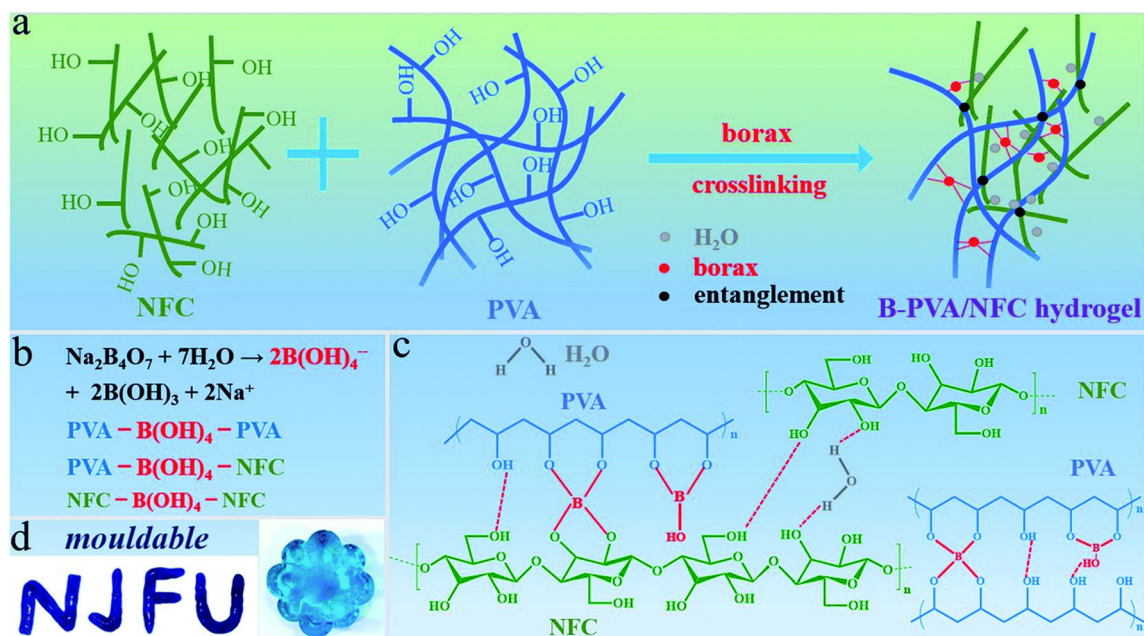
Cellulose possesses the advantages of high chemical/thermal stability, good mechanical strength, and flexibility, indicating its possibility to be involved in polymer electrolyte. Kim et al. [135] incorporated lithium trifluoromethanesulfonate into modified methylcellulose to synthesize a transparent, flexible, and eco-friendly polymer electrolyte, with good conductivity of nearly 1 mS/cm and negligible change in its performance after being recycled and reused 20 times. Jiao et al. [136] prepared the electrolyte separator made of nanofibrillated cellulose (NFC) and poly(4-styrenesulfonic acid) with high conductivity of 0.2 S/cm at 100% relative humidity. The NFC worked as three-dimensional template providing good ionic conductivity through nanopores in the electrolyte.

Polymer electrolytes are classified into solid polymer electrolytes and gel polymer electrolytes. The chemically modified cellulose have been used to synthesize solid polymer electrolytes in some reports. Wang et al. [137] used carboxylated ginger cellulose nanofibers and polyaniline to organize self-assembly of the 3D conductive supramolecular aerogel (SA) with a conductivity of 0.372 mS/cm and a capacitance of 59.26 mF/cm<sup>2</sup>, fast charge–discharge performance, and excellent capacitance retention in a green aqueous medium. Then, the SA solid electrolyte was combined with PVA/H<sub>2</sub>SO<sub>4</sub> and low-cost nonwoven polypropylene to form a supercapacitor without adhesive, and the supercapacitor possessed a high normalized capacitance of 291.01 F/g. Ko et al. [138] synthesized a polymer proton-conducting electrolyte through attaching sulfonic acid functional groups to hydroxypropyl cellulose, and this cellulose-based solid electrolyte exhibited a proton conductivity of 172 mS/cm at room temperature and 100% relative humidity. Aluminum fumarate metal-organic framework (A520) was introduced into the electrolyte to enhance the low humidity performance

and the conductivity of the obtained sulfonated cellulose/A520 hybrid electrolytes enhanced more than an order of magnitude when the relative humidity was below 40%.

However, the rigidity of all-solid-state electrolyte-based supercapacitors can hardly meet the requirements of emerging flexible and wearable supercapacitor energy storage devices. Thus, developing quasi-solid-state electrolytes is necessary, such as gel electrolytes. Parveen et al. [139] assembled an asymmetric solid-state supercapacitor by sandwiching a gel electrolyte-soaked cellulose paper between a red phosphorus/graphene and Ni<sub>2</sub>P as the negative and positive electrodes, attaining a high operating window of up to 1.6 V. Ji et al. [140] prepared cellulose/poly(vinyl alcohol) composite gels electrolyte through the freeze-thawing method for electrical double-layer capacitors. The composite gels showed enhanced mechanical properties and improved network structure compared with the PVA gel, and attained a voltage window of 0–1.8 V and high capacitance retention of nearly 88% after 1500 cycles.

The research on cellulose-based hydrogel electrolytes is the most in the study of various cellulose-based gel electrolytes in supercapacitors, and good results have been achieved. Chen et al. [141] developed a zinc-salt-containing borax-crosslinked polyvinyl alcohol/nanocellulose hydrogel electrolyte as shown in Fig. 9, which possessed great mechanical properties, self-healing feature, and high ionic conductivity, and the flexible supercapacitor assembled from it, zinc metal anode, and cellulose paper cathode exhibits high capacity (504.9 mF/cm<sup>2</sup> at 0.5 mA/cm<sup>2</sup>) and excellent cyclic performance (95.3% capacity retention over 5000 cycles). Wang et al. [142] prepared a cellulose-based hydrogel electrolyte through low temperature in situ crosslinking approach and applied it in flexible supercapacitors with the N-doped graphene hydrogel as the electrode. The assembled supercapacitors exhibited a high capacitance (98.8 F/g at 1 A/g), good rate capability (89.5% at 16 A/g), and cyclic performance (capacitance retention 93.9% after 5000 cycles). Li et al. [143] obtained double-crosslinked hydrogels through a facile solution-phase method using polydopamine as a crosslinker between cellulose and polyacrylamide, and the mechanical and self-healing properties of hydrogels are superior to other hydrogels at the premium ratio of 0.4 with hierarchical porosity (max. 80%). The micro-supercapacitor



**Figure 9** **a** The preparation process of borax-crosslinked polyvinyl alcohol/nanocellulose hydrogel; **b** Multi-complexation between PVA and nanocellulose dynamically crosslinked by borax; **c** Formation mechanism of the 3D network in hydrogel; **d** Photographs of the hydrogel electrolyte moulded into various shapes. Republished with permission of Royal Society of Chemistry, from High-performance flexible and self-healable

quasolid-state zinc-ion hybrid supercapacitor based on borax-crosslinked polyvinyl alcohol/nanocellulose hydrogel electrolyte. Minfeng Chen, Jizhang Chen, Weijun Zhou, Junling Xub and Ching-Ping Wong, 3, 5, 2019; permission conveyed through Copyright Clearance Center, Inc [141].

fabricated by this hydrogel membrane and activated carbon materials showed areal capacitance of 275.8 mF/cm<sup>2</sup> and volumetric capacitance of 394.1 F/cm<sup>3</sup> at 10 mV/s. Wei et al. [144] prepared a water-deactivated polyelectrolyte hydrogel electrolyte using an anionic polymer carboxymethyl cellulose and a cationic monomer methacrylamido propyl trimethyl ammonium chloride. The flexible supercapacitor assembled with this hydrogel electrolyte containing 1 M Li<sub>2</sub>SO<sub>4</sub> and activated carbon electrodes possessed a high operating voltage of 2.1 V.

Compared with hydrogels, cellulose-based ionogels are more stable because ionic liquids (ILs) dispersed in 3D polymer networks are nonvolatile with negligible vapor pressure. Certain ILs with strong nucleophilic anions can effectively dissolve cellulose by disrupting hydrogen bonding and enhance the processability of cellulose. Rana et al. [145] phosphorylated and dissolved a microcrystalline cellulose network by a tailor-made 1,3-dimethylimidazolium methyl phosphite ionic liquid mixture, then polymerized the 2-hydroxyethyl methacrylate monomer to synthesize a cellulose-based dual network ionogel

electrolyte, exhibiting a high ionic conductivity of 2.6–22.4 mS/cm at 30–120°C and a max toughness of 1.46 MJ/m<sup>3</sup> at 30 °C. Guo et al. [146] prepared a halloysite nanotubes-doped liquid crystalline ionogels by in situ crosslinking of cellulose/ionic liquid solutions. The ionic conductivity increased as HNTs concentration increasing and shearing could improve the room-temperature ionic conductivity.

### Starch-based electrolytes

Starch is another type of biomass applied in environmentally friendly supercapacitor electrolytes due to its abundance, renewability, inexpensiveness, excellent solubility, easy degradability, good compatibility, and superior mechanical strength. Kasturi et al. [147] prepared a biopolymer electrolyte film for supercapacitor employing Manihot Esculenta starch (MES) through an eco-friendly solvent casting process without any ionic liquid additive. The fabricated MESF exhibited a high conductivity of 3.1 × 10<sup>-3</sup> S·cm<sup>-1</sup> and gradually degraded by the microorganisms when buried under the soil with a weight loss of



**Table 6** The conductivity of biomass-based polymer electrolytes and composite electrolytes

Polymer electrolyte composition	The conductive ions	Conductivity (mS/cm)	References
Modified methylcellulose	Li <sup>+</sup>	1	[135]
Nanofibrillated cellulose and PSSH	DMSO	200	[136]
Carboxylated ginger cellulose nanofibers and PANI	H <sup>+</sup>	0.372	[137]
Microcrystalline cellulose and poly 2-hydroxyethyl methacrylate	ILs	2.6	[145]
Manihot Esculenta starch	H <sup>+</sup>	3.1	[147]
Corn starch	ILs	0.321	[149]
Corn starch crosslinked with CA	H <sup>+</sup>	2.3	[148]
Starch	Na <sup>+</sup>	71.5	[151]
Corn starch crosslinked with glutaraldehyde	Na <sup>+</sup>	100	[152]
PSSA and starch and glycerol as plasticizer	Li <sup>+</sup>	5.7	[156]
Chitosan, starch and glycerol as plasticizer	Li <sup>+</sup>	0.37	[155]
Chitosan and PDDA	OH <sup>-</sup>	24	[164]
Alginate	Li <sup>+</sup>	8.4	[167]
Alginate crosslinked by dynamic catechol-borate ester bonding	K <sup>+</sup>	6.6	[171]
Alginate and chitosan	Na <sup>+</sup>	51	[165]
SPI and poly(ethylene oxide)	Li <sup>+</sup>	0.1	[172]

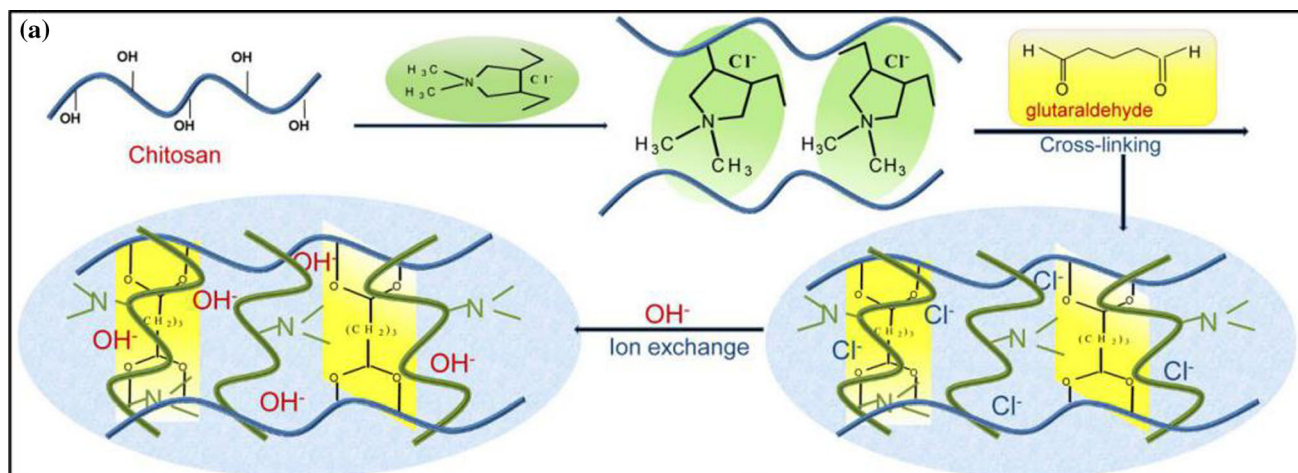
6% after 40 days. Willfahrt et al. [148] established a novel screen printable hydrogel using corn starch with citric acid (CA) as crosslinking agent and the electrolyte by melt-blending and found that the hydrogel with 10 wt% CA proved to be suitable for electrolyte in printed supercapacitors, providing an ionic conductivity about  $2.30 \text{ mS}\cdot\text{cm}^{-1}$ , prolonged stability against degradation, and superb printability.

Lithium salts are widely added as ion providers in the starch-based electrolyte systems to increase the electrical conductivity by enhancing the amorphous region significant in supporting the segmental motions. Liew et al. [149] prepared biopolymer electrolytes employing corn starch as polymer electrolyte with two different counteranions of ionic liquids, viz. 1-butyl-3-methylimidazolium trifluoromethanesulfonate (BmImTf) and 1-butyl-3-methylimidazolium hexafluorophosphate (BmImPF(6)), achieving the maximum ionic conductivities about  $3.21 \times 10^{-4}$  and  $1.47 \times 10^{-4} \text{ S}\cdot\text{cm}^{-1}$  with adulteration of 80 wt% of BmImTf and 50 wt% of BmImPF(6) at ambient temperature, respectively. The supercapacitor assembled with the triflate-based biopolymer electrolyte exhibited a wider electrochemical stability window and a higher specific capacitance than the hexafluorophosphate system. Lithium perchlorate (LiClO<sub>4</sub>) has less electronegativity than other fluoro-type lithium salts, along with ease of complex formation in the fabrication of solid polymer electrolytes, good ion

conductivity, and stability on application aspect. Thus, LiClO<sub>4</sub> is usually chosen as an ion provider salt added in the starch-based electrolytes. Teoh et al. [150] fabricated corn starch-based biodegradable polymer electrolytes with the addition of the nano-sized silica (SiO<sub>2</sub>) and lithium perchlorate (LiClO<sub>4</sub>), then assembled solid-state polymeric capacitors, in which SiO<sub>2</sub> and LiClO<sub>4</sub> successfully improved the ionic conductivity of electrolytes and the capacitance of supercapacitors.

However, lithium salts are limited with high cost and handling problems and sodium is recognized as an alternate salt in some studies. Railanmaa et al. [151] prepared starch hydrogels with NaCl of various concentrations for supercapacitor electrolytes through physical crosslinking, and the conductivity values were 71.5 mS/cm at 2 mol/L. Chauhan et al. [152] obtained a transparent film employing corn starch and NaClO<sub>4</sub> salt by crosslinking with glutaraldehyde, possessing a high-conductivity of  $10^{-2} \text{ S/cm}$ , the large electrochemical stability window around 2.4 V, and a small equivalent series resistance about 6.252 Ω for 0.8-mm-thickness. The conductivity of starch-based electrolytes with lithium salts and sodium salt is summarized in Table 6, and the sodium salts offer starch-based electrolytes with much higher conductivity than lithium salts.

The electrolytes produced from starch tend to be brittle and sensitive to moisture [153]. Blending and



**Figure 10** Schematic preparation procedure for the CS-PDDA-OH- membrane. Adapted with permission from Alkaline Exchange Polymer Membrane Electrolyte for High Performance of All-Solid-State Electrochemical Devices, Yanan Wei, Min Wang,

Nengneng Xu, et al. Copyright (2018) American Chemical Society [164].

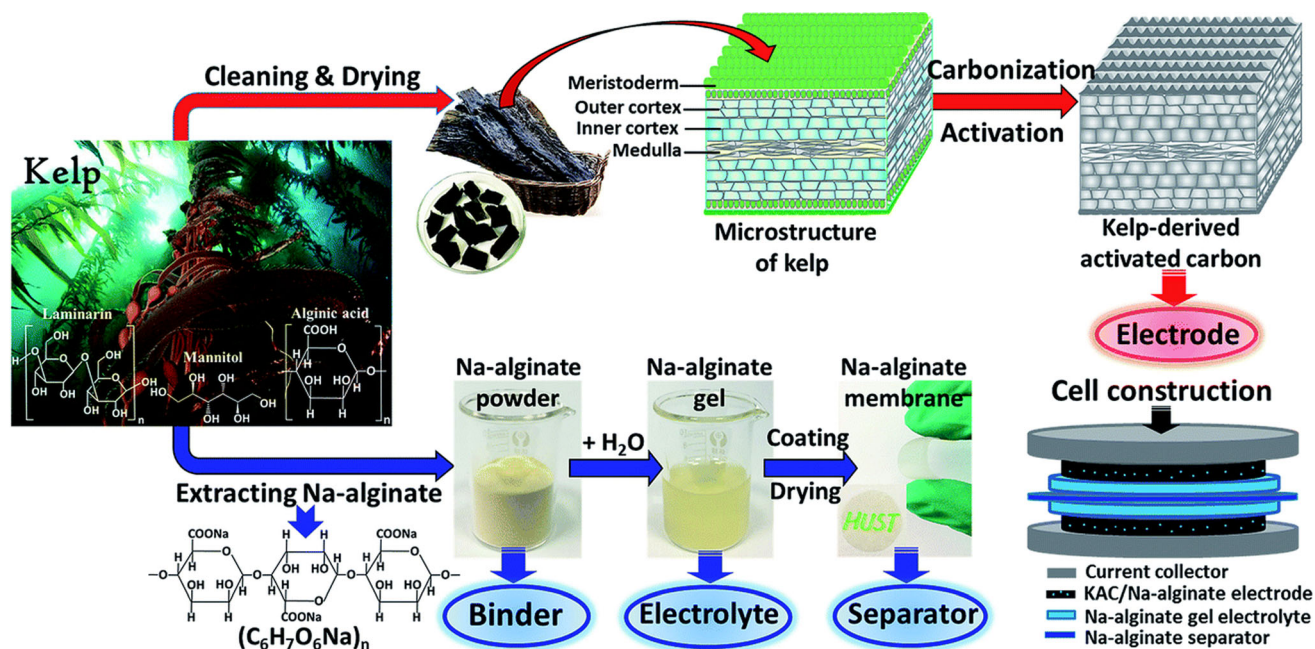
plasticizing are two effective methods applied widely to improved physical, mechanical, and electrical properties of starch-based electrolytes. Glycerol is a proper plasticizer used in some studies. Yusof et al. [154] prepared a solid polymer electrolyte using polymer blend of potato starch and methylcellulose with glycerol as plasticizer and  $\text{LiClO}_4$  as dopant salt, and then employed it in the carbon-based electrical double-layer capacitor with the highest specific capacity value of 28.04 F/g after 1000 charge–discharge cycles. Sudhakar et al. [155] used blends of starch and chitosan,  $\text{LiClO}_4$  as dopant and glycerol as plasticizer to prepare biodegradable solid polymer electrolytes. The polymer electrolyte with 60:40 (CS/starch) blend ratio exhibited the maximum conductivity of  $3.7 \times 10^{-4}$  S/cm, and the carbon–carbon supercapacitor with it had an excellent specific capacitance of 133 F/g at 10 mV/s. In another study, they [156] took poly(styrene sulphonic acid) to replace chitosan, and the obtained polymer electrolyte with 80/20 (PSSA/starch) blend ratio exhibited the maximum conductivity of  $5.7 \times 10^{-3}$  S/cm, and the carbon–carbon supercapacitor with it exhibited a specific capacitance of 115 F/g at 10 mV/s.

### Chitin and chitosan-based electrolytes

Chitin has the semi-crystalline structure due to its highly extended hydrogen bond, leading to the difficulty to dissolve in water or common organic

solvents and limiting the applications of chitin in polymer electrolytes. Several solvent systems have been employed to enhance the dissolution efficiency for chitin, such as NaOH/Urea [157],  $\text{CaCl}_2$ -MeOH [158], LiCl/DMAc [159], and ionic liquids. The ionic liquids with an imidazolium ring has higher dissolution efficiency for chitin, and that with a more basic anion (such as acetate) has better ability to break a strong hydrogen bond [160]. Deng et al. [161] dissolved chitin in 1-butyl-3-methylimidazolium acetate, a room-temperature ionic liquid suitable for chitin with different molecular weights and origins, to prepare a weak ion gel. [BMIM]Ac could dissolve chitin efficiently at a relatively low temperature of about 80 °C in 10 min.

Chitosan is more widely used in the preparation of polymer electrolytes compared to chitin for its better solubility. Keskinen et al. [162] employed chitosan both as ionic conducting electrolyte separator and activated carbon binder to manufacture nontoxic supercapacitors with the capacitance of 0.26–0.43 F and the equivalent series resistance of 12–32  $\Omega$ . Perez-Madrugal et al. [163] tried to fabricate a solid-state organic electrochemical supercapacitor using poly(3,4-ethylenedioxythiophene) electrodes, polyaniline fibers as redox additive, and chitosan hydrogels as electrolyte. However, the chitosan hydrogels exhibited poor mechanical integrity when submitted to stress, limiting its applicability in bio-hydrogel electrolyte.



**Figure 11** Graphical illustration of the synthesis for individual components in supercapacitors from kelp. Republished with permission of Royal Society of Chemistry, from Bio-inspired high-performance solid-state supercapacitors with electrolyte,

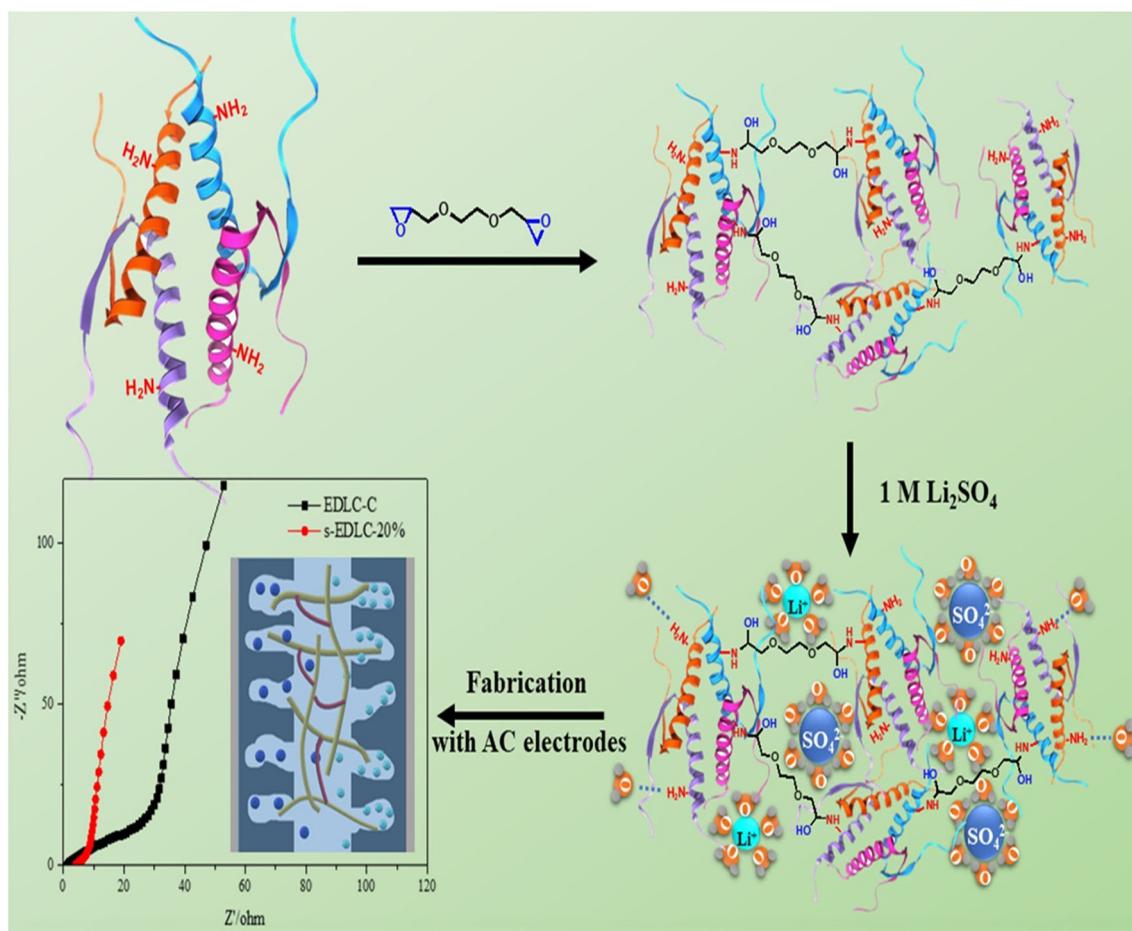
separator, binder and electrodes entirely from kelp, Juan Zeng, Lu Wei and Xin Guo, 5, 48, 2017; permission conveyed through Copyright Clearance Center, Inc [169].

A pure chitosan electrolyte has a poor mechanical property and low ionic conductivity. Many studies have been developed to enhance the mechanical property and ionic conductivity, such as compounding. Wei et al. [164] explored anion-exchange membranes (AEMs) using the chitosan and poly(diallyldimethylammonium chloride) (PDDA) composites for all-solid-state electrochemical devices, in which PDDA was water-soluble and offered conductive anions ( $OH^-$ ) as shown in Fig. 10, and thus AEMs exhibited high ion conductivity of  $0.024 S\cdot cm^{-1}$ , a low water-swelling ratio, as well as thermal, alkaline, and oxidation stability with strong membrane structure. Zhao et al. [165] developed semi-dissolution acidification sol-gel transition method for preparing polyelectrolyte complex hydrogels of chitosan and sodium alginate with uniform composition presenting a high ionic conductivity ( $0.051 S/cm$ ) and proper mechanical properties (elongation at break of 109.5% and a tensile strength of 0.29 MPa). The solid-state supercapacitors assembled with these hydrogels and polyaniline nanowire electrodes exhibited a high specific capacitance ( $234.6 F/g$  at  $5 mV/s$  and good cycling stability (capacitance retention of 95.3% after 1000 cycles).

Cao et al. [166] prepared a supramolecular hydrogel electrolyte (SHE) via a crosslinking between chitosan and  $Li^+ / Ag^+$  with excellent moldable and flexible properties as well as a high thermal stability, and an assembled supercapacitor with active carbon and  $MnO_2$  as negative and positive electrodes survived with an area capacity of  $10 mF/cm^2$  at  $1.8 mA/cm^2$  after more than 10,000 cycles.

### Other biomass-based electrolyte

Sodium alginate is a high-modulus natural polysaccharide extracted from brown algae, containing massive carboxylic groups. It forms a viscous gel with ionic conductivity, nontoxicity, and environmental friendliness when binding with water. In recent reports, sodium alginate has been employed to fabricate bio-hydrogel electrolytes for supercapacitors. Ojha et al. [167] prepared sodium alginate hydrogels as a  $Li^+$  conducting electrolytes to assemble symmetric and asymmetric supercapacitors, showing an ionic conductivity about  $8.4 mS\cdot cm^{-1}$  at room temperature with electrochemical stability over a wide operational voltage window around 2.5 V. Zhao et al. [168] used a highly concentrated sodium



**Figure 12** Schematic construction process of the crosslinked SPI gel polymer electrolyte and the solid-state supercapacitor. Reproduced from A Crosslinked Soybean Protein Isolate Gel Polymer Electrolyte Based on Neutral Aqueous Electrolyte for a

High-Energy-Density Supercapacitor, Pengfei Huo, Shoupeng Ni, Pu Hou, Zhiyu Xun, Yang Liu and Jiyou Gu, 11, 5, 2019 with permission from Polymers [173].

alginate hydrogel as a biocompatible electrolyte providing a wider electrochemical stability window (about 2.8 V) compared with conventional aqueous electrolytes (1.23 V). The fabricated solid-state micro-supercapacitors exhibited good mechanical stability, excellent electrochemical performances with a high areal capacitance of 47.20 mF/cm<sup>2</sup>, good capacitance retention of 95% after 10,000 cycles and high energy density of 21.20 μWh/cm<sup>2</sup> at a power density of 0.18 mW/cm<sup>2</sup>. Zeng et al. [169] developed a hydrogel electrolyte, a membrane separator, binder and electrodes using sodium alginate extracted from kelp to assemble a solid-state supercapacitor (Fig. 11) presenting excellent electrochemical performance with a high specific capacitance of 277 F/g or 88.2 F/cm<sup>3</sup>, good rate capability (capacitance of 156 F/g at 20 A/g) and good cycle stability. Thanks to the good

hydrophilicity of sodium alginate as well as the same origin, the electrolyte and separator had a good wettability and low interface resistance.

To enhance the mechanical property and cold resistance of the alginate hydrogels, the approach of improving the crosslinking structure have been applied in some studies. Zeng et al. [170] prepared both electrodes and electrolytes using the polyacrylamide/sodium alginate dual-network hydrogel matrix possessing superb self-adhesion because of hydrophilic interaction and hydrogen bonds as well as highly softness and toughness due to the energy-dissipative mechanism. The assembled supercapacitor was highly stretchable and compressible to be deformed arbitrarily under severe stress-strain deformation conditions, delivering stable energy output and high areal capacitance of 128 mF/cm<sup>2</sup> at

1 mA/cm<sup>2</sup>. Tao et al. [171] developed a sodium alginate-based hydrogel crosslinked by dynamic catechol-borate ester bonding for the electrolyte of a cold-resistant and self-healable supercapacitor, with the ability to keep at least 80% of room-temperature capacitance at  $-10\text{ }^{\circ}\text{C}$  and recover its capacitive performance in 10 min without external stimulus during 10 cut/healing cycles thanks to the dynamically crosslinked alginate networks.

Soybean protein isolate (SPI), a biomass made out of amino acids covalently linked by peptide bonds, is renewable, biodegradable, and hydrophilic. It has been investigated as the matrix of membranes having ionic conductivity. Fu et al. [172] prepared a soy protein-based solid conductor and added a little poly(ethylene oxide), exhibiting an ionic conductivity about  $10^{-5}\text{ S/cm}$ , demonstrating that SPI can be applied in polymer electrolytes. Huo et al. [173] successfully formed a crosslinked SPI gel polymer electrolyte by the solution casting method, swelling the SPI membrane in 1 mol/L  $\text{Li}_2\text{SO}_4$  (Fig. 12), and applied it in the solid-state electric double-layer supercapacitor with activated carbon electrodes, which delivered a specific capacitance of 115.17 F/g at a current density of 1.0 A/g higher than that of the supercapacitor with the commercial separator in 1 mol/L  $\text{Li}_2\text{SO}_4$ .

Table 6 lists the electrical conductivity of biomass (cellulose, starch, chitosan, alginate) based and composite electrolytes. Uniform composite biomass with proper materials, such as conductive polymers, crosslinking agents, and plasticizers, can greatly increase the conductivity of the biomass-based electrolyte by an order of magnitude. Starch and alginate-based electrolytes tend to exhibit higher conductivity compared to other biopolymers with the same conductive ions.

## Conclusion

Biomass-derived carbon materials, including activated carbons, carbon fibers, templated carbons, carbon aerogels, carbon nanotubes, and graphene, have been developed for supercapacitor electrodes. For preparing carbon precursors from raw biomass materials, the sol-gel method and electrospinning are new methods corresponding with the properties of some biomass. Pyrolysis of biomass through several conventional methods, such as high-temperature

carbonization and hydrothermal carbonization, have been widely applied in producing porous carbons for supercapacitors. Besides, microwave heating is a new method used in the rapid pyrolysis of biomass with very high heating efficiency and rate. For the poor microwave-absorbing ability of biomass, it is significant to utilize and develop the microwave absorber, such as  $\text{ZnCl}_2$ ,  $\text{CuCl}_2 \cdot 2\text{H}_2\text{O}$ , biochars, and humidified nitrogen.

To facilitate the porous structures of biomass-derived carbon, chemical activation is a widely used approach with the chemical agents such as KOH,  $\text{K}_2\text{CO}_3$ , NaOH, and  $\text{ZnCl}_2$ , besides which the MOF, ZIF frameworks and hard template are introduced to both fabricate and preserve the 3D interconnected hierarchically porous frameworks during pyrolysis of biomass. The current techniques to produce biomass-derived porous carbon materials possess own advantage and limitations, and the improvement on each production technique is significant for the optimization of the properties and the future commercial production of biomass-derived carbons. Compared to two-step pyrolysis and activation, one-step methods, achieving carbonization and activation at the same time, are much more efficient, time and energy-saving, and worth further research and improvement.

To further enhance the electrochemical performance of biomass-derived carbon materials, heteroatoms (such as O, N, S, and B) doping is a frequently used approach providing extra pseudocapacitances and increasing the surface wettability. Various biomass resources, like cellulose, chitosan, lignin, and so on, are rich in heteroatoms such as O, N, and S to be converted into heteroatom self-doped carbon materials, and multiple heteroatoms doping is proved to have a synergistic effect. Incorporation of pseudocapacitive materials with redox kinetics is another widely used way by compounding carbon materials with iron oxides/sulfides/phosphites and conductive polymers like PPy and PANI. The latter method performs much better on the improvement of specific capacitance. Nevertheless, the rate capability and cycle stability of composites are inferior, which need further improvement through changing their structures, composition, searching more appropriate pseudocapacitive materials, and other innovative methods. Heteroatom doping and incorporation with pseudocapacitive materials will continue being the main methods to improve the electrochemical

performance of biomass-based carbon electrode materials. In addition, the burgeoning biomass-derived carbon nanofibers have great potential to improve the electrochemical performance of supercapacitors and worth further research.

Polymer electrolytes based on biomass, such as cellulose, chitin, chitosan, alginate, and soybean protein isolate, have been applied in supercapacitors with a wide potential window, a high ionic conductivity, excellent stability, and mechanical strength in recent years. Among them, starch and alginate-based electrolytes tend to exhibit higher conductivity with the same conductive ions. Compared to all-solid-state electrolytes, biomass-based quasi-solid-state electrolytes, such as gel electrolytes, better meet the requirements of emerging flexible and wearable supercapacitor energy storage devices and will receive constant attention in the future. Among them, biomass-based hydrogel electrolytes are widely studied in supercapacitors and good results have been achieved, and biomass-based ionogels are more stable than hydrogels because ionic liquids dispersed in 3D polymer networks are nonvolatile with negligible vapor pressure.

Although progress has been achieved on the application of biomass in electrolytes, it is still a relatively new research topic. For biomass-based electrolytes, the enhancement of their ionic conductivity and mechanical properties is still the focus of future research for flexible and wearable supercapacitors. Lithium salts and sodium salts can be added as ion providers in the biomass-based electrolyte systems to increase the electrical conductivity by enhancing the amorphous region significant in supporting the segmental motions. Besides, blending and plasticizing are two effective methods to improve not only physical and mechanical properties but also the electrical properties of biomass-based electrolytes, which have been applied widely recently. Glycerol is a proper plasticizer used in many studies. Composite electrolytes of biopolymers and proper materials, such as conductive polymers, crosslinking agents, and plasticizers are proved to exhibit greatly increased conductivity by an order of magnitude. These methods provide reference direction for the improvement of biomass-based electrolytes in the future but there are still large gaps to be filled on the variety and effectivity of approaches to achieve this aim.

## Author contributions

The manuscript was written through the contributions of all authors. All authors have given approval to the final version of the manuscript.

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## Compliance with ethical standards

**Conflicts of interest** No potential conflict of interest was reported by the authors.

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