# **Computation & theory**



# Atomistic insight into shell-core evolution of aluminum nanoparticles in reaction with gaseous oxides at high temperature

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# ABSTRACT

Aluminum nanoparticles (ANPs), as an economical and effective metal fuel, are widely applied in energetic formulations. The objective of this research was to gain insights into the oxidation of ANPs in gaseous oxides  $(CO_2, CO, NO_2, and$ NO). The reactive molecular dynamics (RMD) simulations were performed to elucidate the detailed mechanisms of surface oxidation, chain-like products formation, and hollow formation in the evolution of ANPs. The O atoms in gaseous oxides are adsorbed on the ANPs' surface followed by the cleavage of O-C/N bond. The nucleation and growth of chain-like products occur in dense gaseous oxides. The four forms of chain products include tilted chain, twisted chain, branched chain, and cyclic chain were observed in CO atmosphere. A similar chain structure is also formed in NO atmosphere, but the chain length is significantly reduced. The oxide shell of ANPs is formed and expands rapidly in CO<sub>2</sub> atmosphere, resulting in voids between oxide shell and Al core. Al atoms are transported from the core to the oxide shell through a bridge composed of Al atoms. The Al core gradually diffused outward and was eventually hollowed out. In addition, the final product has carbon deposits (C48 and C98) on the surface and core.

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### **GRAPHICAL ABSTRACT**



## Introduction

Metal powders that can transiently release a large quantity of combustion heat by oxidation are widely used as propellants and fuels [1-3]. Currently, aluminum powder has been recognized as one of the ideal energetic metals due to its high energy density, abundant raw materials, and environmentally friendly combustion products [4, 5]. However, the oxide films on aluminum powder have significant impacts on its ignition. The alumina layer will reduce the energy release of aluminum combustion [6]. To improve these problems, aluminum nanoparticles (ANPs) with higher activity are used to replace the micro-sized ones because of its specific properties, such as size effect and surface-interface effect [7–10]. Meda et al. used scanning electron microscopy, specific surface area measurements, X-ray photoelectron spectroscopy, and X-ray diffraction to study the combustion characteristics of nano-aluminum powder in solid propellants. It was found that the combustion efficiency of the propellant containing nanoparticles was significantly improved, and the smaller the particle size of the aluminum powder, the lower the ignition temperature of the propellant and the shorter the combustion time [11]. Meda et al. also tested the content of unburned aluminum powder in the combustion products of solid propellants. It is found that the content of unburned aluminum powder in the combustion products of nano-aluminum powder is lower than that of micron powder [12]. Ivanov et al. also found that the burning rate of Alex propellants (an ultra-fine aluminum powder produced by plasma explosion process) is higher than that of ordinary micron aluminum powder propellants [13].

The atomic insights into the ignition mechanism of ANPs have been investigated in the extremely small scales of space and time [14, 15]. Hong et al. studied the oxidation mechanism of ANPs by using ReaxFF molecular dynamics simulations. It is proposed that the ANPs oxidation depends on combined effects of the temperature and the oxygen gas pressure [15]. Campbell et al. determined that the large pressure gradients exist inside the oxidized ANPs by using the molecular dynamics simulations. The results indicated that the aluminum core is in positive pressure, while the oxide shell is under negative pressure [16, 17]. However, most previous researches on ignition of ANPs have been performed in oxygen atmosphere, and only a few studies have been performed to investigate the ignition of ANPs in other atmospheres. Actually, ignition and combustion of



ANPs under various conditions are also of significance. Carbon oxides and nitrogen oxides are the main components of solid rocket propellant combustion products that can interact with metal aluminum powder. Furthermore, it is well known that Martian atmosphere contains about 96% of CO<sub>2</sub>, which makes it possible to achieve in situ resource utilization. Therefore, the Al/CO<sub>2</sub> mixture is considered a promising fuel for future Mars exploration missions. Palopoli et al. studied the pyrolysis gaseous products of nitramine compounds and found that these gas products contained a large amount of nitrogen oxides such as N<sub>2</sub>O, NO<sub>2</sub>, NO [18]. Yan et al. found that a large amount of nitrogen oxides are emitted during the combustion of nitramine propellants. It is of great significance to study the effect of these nitrogen oxides on aluminum oxidation, especially the exploration of the combustion mechanism of high-energy solid fuels [19]. Therefore, in the combustion of aluminum-based fuels, it is of great importance to study the interaction between gaseous molecules and aluminum, especially to explore the combustion mechanism of high-energy solid fuel propellants. Meanwhile, keeping ANPs stable under various conditions of manufacture, transportation, stock, and application is also of significance. Therefore, the detailed understanding of the transient structure evolution of ANPs as well as its interplay with the small gaseous molecules at the ignition stage is very meaningful. In this work, we performed reactive molecular dynamics (RMD) simulations to show the ignition of ANPs in four different atmospheres (CO<sub>2</sub>, CO, NO<sub>2</sub>, and NO). Our results would provide important insights into understanding the initial ignition and combustion mechanisms of metallic nanoparticles.

### **Computational details**

The reactive force field (ReaxFF), developed and trained from quantum mechanical calculations, employs connectivity, bond-order formalism and polarizable charge descriptions to compute precisely both reactive and non-reactive interactions [20–24]. In this study, a ReaxFF suitable for ANP was applied to investigate the ignition and combustion mechanism of ANP under four gaseous oxides [25]. The ANP structure (Al<sub>3997</sub>, 5.0 nm in diameter) was established and placed at the center of the simulation

box measured  $10.1 \times 10.1 \times 10.1$  nm by using the materials visualizer tool. Different atmospheres were randomly filled into the remaining empty spaces in the box. To study the effect of gas pressure on ignition and combustion, the low density, medium density, and high density for each gas were filled in boxes. Details on the four gaseous oxides are listed in Table 1. The thickness of the vacuum layer between ANP and gases component was set as 0.2 nm. The models were equilibrated at 298 K for 10 ps and then were heated rapidly to the target temperatures within 30 ps, and were equilibrated finally at target temperatures for 200 ps. We performed the RMD simulations in the canonical (NVT) ensemble using the Nosé-Hoover thermostat [26, 27]. These initial model settings were repeated infinitely by imposing periodic boundary conditions in all three dimensions. The detailed information of dynamic trajectory was recorded every 50 fs and applied to analyze the evolution of ANP at five temperatures (1500, 2000, 2500, 3000, and 3500 K). The initial atomic velocities based on the required temperatures in different systems were determined according to the Maxwell-Boltzmann distribution. Newton's equation of motion was calculated using the velocity Verlet algorithm with a time step of 0.1 fs, and atomic charges were obtained by the charge equilibration (QEq) method with the update of each 10th time step. All RMD simulations were performed using largescale atomic/molecular massively parallel simulator (LAMMPS) program package [28].

### **Results and discussion**

### Dynamic morphologies of ANPs

To investigate the reaction mechanism of  $CO_2$  oxidation, the evolution of the charge distribution is shown in Fig. 1. At 0 ps, the charge of Al atoms tends to be zero. When it reaches 20 ps (heating stage),  $CO_2$ molecules move to and react with Al atoms on the surface of ANPs. With the prolonged heating time, the low-coordinated Al transitions to the high-coordinated oxide, and the oxide gradually grows to the inside of ANP. In addition, the higher the density of  $CO_2$ , the more intense the oxidation reaction. The large pressure gradients accumulate inside the oxide at this time. The generation of staggered stress fields causes the nanoparticle shape to change. Therefore,

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Atmospheres	Number of molecules/density (low)	Number of molecules/density (medium)	Number of molecules/density (high)
CO <sub>2</sub>	690/0.055	1380/0.11	2770/0.22
СО	690/0.035	1380/0.07	2770/0.14
NO <sub>2</sub>	690/0.058	1380/0.115	2770/0.23
NO	690/0.0375	1380/0.075	2770/0.15

Table 1 The number of molecules of the filling gas and the corresponding gaseous density (g/cm<sup>3</sup>)



Figure 1 Evolution of charge distribution of ANPs–CO<sub>2</sub> systems with time at 3500 K. The CO<sub>2</sub> densities of **a** 0.055 g/cm<sup>3</sup>, **b** 0.11 g/cm<sup>3</sup>, and **c** 0.22 g/cm<sup>3</sup> were selected as examples.

Figure 2 a The cross section images of Connolly surface area of the solid, the hollow cavity with a radius of 9.67 A in the particles (240 ps). Thumbnail image of cavity of 3-µm Al in the bottom right [31]. **b** The nanoparticle shape of the solid region at 2000 K (40 ps). c Surface carbon deposition of nanoparticles and external long chain of  $Al_5C_{12}$ . **d** Core carbon deposition of nanoparticles (sectional drawing). The blue, red and yellow atoms represent the Al, O and C atoms, respectively.





when the density of  $CO_2$  is  $0.22 \text{ g/cm}^3$ , the nanoparticles could not maintain the spherical morphology at 40 ps.

The Connolly surface defines the boundary between the solid area and its environment. The Connolly surface calculations were obtained by atom volumes and surfaces (Material studio, version 6.0). Figure 2a shows the hollow oxide spheres in the oxidation reaction of ANPs and CO2 with a density of  $0.22 \text{ g/cm}^3$  at 1500 K. By calculating the Connolly surface, the radius and area of the hollow region are 9.67 Å and 2.885 nm<sup>2</sup>, respectively. Hollow structures have been reported by researchers [29–31]. They proposed the oxide developed on the shell of the sphere. The new phase expanded around the exterior as temperature increased. The presence of hollow structures is because outward Al diffusion has resulted in a core hollow. The evolution of hollow region is shown in Fig. 3. The thin slices with a thickness of 0.5 nm show the details of the internal structural evolution. The initial reaction mainly occurred in the surface area. The oxidation of the surface causes the atoms to rearrange and to release a large amount of heat. The oxide shell rapidly expands and grows into the outer space, causing a significant void between oxide shell and the Al core. At 40 ps, Al atoms diffuse and form clusters in the void. The Al core and the oxide shell are connected by some bridges composed of Al atoms. The Al core transport Al atoms to the oxide shell through them, which promoted further oxidation of Al in oxide shell. The

Al atoms in the core gradually diffused outward and was eventually hollowed out. Hence there is a big central void in the final nanoparticle product. Chakraborty et al. studied the diffusion mechanism of oxide-coated ANPs at 2000 K. Hollow particles are considered to be in a transient state and can be considered to the higher outward diffusion of aluminum ions compared to the inward diffusion of the shell ions [29]. Figure 2b shows the nanoparticle shape of the same system at 2000 K. Both the surface and internal structure of ANPs undergo significant changes in CO<sub>2</sub> oxidation and generate a large number of defects. Due to the misfit dislocations on the surface and the emission of atomic chain oxides, the ANPs are more conducive to being oxidized by  $CO_2$  [32]. Sarou et al. observed the carbon dissolution process in combustion of Al-CO<sub>2</sub>. The carbon concentration reached the saturation limit ( $x_{\rm C} = 0.23$  at T = 2600 K), the excess carbon was ejected on the surface and formed a solid coating [33]. Carbon coats the aluminum surface to suppress the burning of aluminum. C deposits are easily formed under hightemperature reactions. Some results further confirm carbon deposition [31, 33, 34]. Figure 2c illustrates the formation of C<sub>98</sub> deposited on the nanoparticle surface at 2000 K. At the same time, a chain-like product is generated in a gaseous environment. The chemical formula of long chain is  $Al_5C_{12}$ , and the C=C and <sup>[4]</sup>Al are arranged alternately, and the starting and ending ends are  $C \equiv C$ . In addition to carbon surface deposition, there is also internal carbon deposition.



Figure 3 The evolution of the morphologies of the hollow region at 1500 K. The blue, red and yellow atoms represent the Al, O and C atoms, respectively.

As shown in Fig. 2d, C48 exists at the core of the nanoparticles at 3500 K. As the nanoparticles are fully oxidized at high temperatures, agglomeration occurs between the same phases. Most C atoms in the nanoparticles move outward and accumulate on the surface, and the other part of the C atoms move inward. Eventually, these C atoms are mainly distributed on the core or surface of the nanoparticles. The possibility of forming C deposits on the surface or core is greater. As the aluminum oxide expands outward, the stress inside the nanoparticles is reduced, and the inward C atoms aggregate into C deposits such as C<sub>98</sub> and C<sub>48</sub>. The presence of a large amount of AlC<sub>4</sub> in the gaseous environment indicates that high temperature is not conducive to the growth of the  $Al_n C_{2n+2}$  chain.

Compared with the combustion of ANPs–CO<sub>2</sub>, the severity of ANPs–CO combustion was significantly reduced. Similarly, taking three CO densities at 3500 K as an example, the evolution of charge distribution of ANPs are shown in Fig. 4. When the CO densities are  $0.035 \text{ g/cm}^3$  and  $0.07 \text{ g/cm}^3$ , ANPs oxidized from the shell to the core. The ANPs are not completely oxidized due to the limitation of the oxygen concentration in the external atmosphere. When the CO density reaches  $0.14 \text{ g/cm}^3$ , the oxidation of ANPs increases significantly. In addition, the thermal energy released by the surface reactions could trigger greater stress on the oxide coating, causing the oxide of the shell to be broken down at 40 ps. In addition, the chain-like product in the

atmosphere environment gradually nucleates and grows from 120 to 240 ps.

Figure 5a shows the interaction of CO molecules with ANPs in the initial heating stage. Since the electronegetivity of oxygen is much greater than that of carbon, the CO molecules adjacent ANPs adjust their orientation so that the O-terminal faces ANPs. Figure 5b shows chain oxides ejecting out from the nanoparticles. The chain product is composed of repeated Al<sub>4</sub>O, and its core consists of oxygen atoms and is externally bonded with four Al, which damages the oxide layer and facilitates further oxidation. The chain-like nucleation and growth of oxides were observed on the ANPs surface during the oxidation and microexplosion-accelerated oxidation [32, 35]. Our results demonstrated these viewpoints further. Figure 5c shows a snapshot of the ANPs–CO system with the CO density is  $0.14 \text{ g/cm}^3$ . There are a lot of carbon-containing chain products in the ANPs-CO system, far more than the ANPs-CO<sub>2</sub> system. This is mainly due to the increase in the relative concentration of C. Four forms of chain products including tiled chain (I), twisted chain (II), branched chain (III), and cyclic chain (IV) were observed. The chemical formula of tiled chain (I), twisted chain (II) and branched chain (III) is  $Al_nC_{2n+2}$ , and that of the cyclic chain (IV) is  $Al_n C_{2n}$ .

Figure 6a shows the morphological evolution of ANPs in  $NO_2$  atmosphere. The ambient atmosphere was composed of  $NO_2$  at first, and the amount of  $N_2$  formed increased significantly with time. Figure 6b



Figure 4 Evolution of charge distribution of ANPs–CO systems with time at 3500 K. The CO densities of a 0.035 g/cm<sup>3</sup>, b 0.07 g/cm<sup>3</sup>, and c 0.14 g/cm<sup>3</sup>.

Figure 5 a The interaction between CO molecules and ANPs in the range of 0.2 nm when the density of CO in the environment is 0.035 g/cm<sup>3</sup>. **b** The snapshot of the nanoparticles when the CO density is 0.07 g/cm<sup>3</sup>. The enlarged view is a crosssectional view of the chain ejected by the nanoparticles. c The snapshot of the system at 3500 K with a CO density of  $0.14 \text{ g/cm}^3$ . The enlarged ball-stick skeleton and polyhedron diagrams are Al<sub>6</sub>C<sub>14</sub> (I), Al<sub>5</sub>C<sub>12</sub> (II),  $Al_{10}C_{22}$  (III) and  $Al_6C_{12}$  (IV), respectively.



Figure 6 Morphological evolutions of ANPs in the a nitrogen dioxide and b nitric oxide with the densities of  $0.23 \text{ g/cm}^3$  and  $0.15 \text{ g/cm}^3$ , respectively. Blue, red and green represent Al, O and N atoms, respectively.

1

120 ps

80 ps

shows the morphological evolution of ANPs in NO atmosphere. Consistent with the reaction of ANPs– $NO_2$ , oxygen in NO as the main component participates in the reaction with ANPs. The difference is the

0 ps

distribution of aluminum nitride clusters that increase with time in the external environment. Although the density of  $NO_2$  and NO is different, the molecular numbers of two are the same. What causes

160 ps

**2**00.ps

 $240 \,\mathrm{ps}$ 

(a)

**(b)** 



Figure 7 Sectional view of nanoparticle evolution in  $\mathbf{a}$  NO<sub>2</sub> and  $\mathbf{b}$  NO atmosphere. Blue, red and green represent Al, O and N atoms, respectively.

the difference in the final ambient gas? The crosssectional drawings of the nanoparticles in the reaction are showed in Fig. 7. It is found from the RMD trajectory that N diffuses faster than O in ANPs. In the early stage, although aluminum oxide was formed and remained in the shell, N atoms were not hindered and diffused to the center of the nanoparticle first. Driven by thermodynamics, Al-O bond formation is more favorable at high temperature. Therefore, the O atoms gradually replace N atoms in the nitride phase of nanoparticles. For the ANPs-NO<sub>2</sub> system, we observe a clear phenomenon of competition between oxide phase and nitride phase from 60 to 80 ps. The oxide phase expands and the nitride phase continuously disappears in the ANPs-NO<sub>2</sub> system. The formation of an aluminum oxide cap moves on the particle surface in the combustion experiment [36]. Our simulation results are consistent with experimental phenomena. We found that the aluminum nitride clusters in the environmental phase increase significantly during this period, as shown in the snapshot of 80 ps in Fig. 6a. Because the boiling point of Al<sub>2</sub>O<sub>3</sub> (2980 °C) is higher than the melting point of AlN (2249 °C), the aluminum nitride easily enters the environmental phase as a gaseous molecule. For the ANPs-NO system, the concentration of O is not sufficient to oxidize completely the ANPs. As a result, significant phase separation and phase agglomeration occur. Small aluminum nitride clusters are released into the environmental phase continuously because of the lower-boiling aluminum nitride. Figure 8 shows a snapshot of the ANPs–NO system. There are a large number of chain products in



**Figure 8** The snapshot of the system at 3500 K with a NO density of 0.15 g/cm<sup>3</sup>. The enlarged ball–stick and polyhedron diagrams are AlN<sub>4</sub> (I),  $Al_2N_6$  (II) and  $Al_3N_8$  (III), respectively. The enlarged image (IV) is chain products in surface.

the ambient atmosphere, including  $AlN_4$  (I),  $Al_2N_6$  (II) and  $Al_3N_8$  (III). There are also chain products to be released from the shell.

The presence of chain products promotes the oxidation reaction of ANPs. On the one hand, the n(AI)/m(X) ratio of  $AI_nX_{2n+2}$  is greater than that of AIN and  $AI_4C_3$ , hence more C and N atoms participate in the reaction. On the other hand, the nanoparticles can provide AI and N atoms for the chain formation reaction. Hence the  $AI_nX_{2n+2}$  that ejects from the shell of nanoparticles is more conducive to nucleation and growth, rather than a chain-like product generated in



Figure 9 The peak number of chain products in CO atmosphere (a) and NO atmosphere (b).

the environmental phase. Figure 9 shows the maximum number of various chain products in oxidation reaction. The number of chains decreases as the number of *n* in  $Al_n X_{2n+2}$  increases. For the ANPs–CO and ANPs–NO systems, their maximum n values were observed in the simulations to be 5 and 3, respectively. Although the numbers of AlN<sub>4</sub> and  $Al_2N_6$  (135 and 37, respectively) is more than the number of AlC<sub>4</sub> and Al<sub>2</sub>C<sub>6</sub> (53 and 23, respectively), longer C-containing chains appear in CO atmosphere such as  $Al_4C_{10}$  and  $Al_5C_{12}$ . In addition, a small amount of AlN<sub>3</sub>O was detected in the ANPs-NO system. To the best of our knowledge, there has been no research on chain products focused on this oxidation in CO/NO atmosphere. Thus, our results offer new insights into the chain product formation mechanisms.



Figure 10 Reaction steps extracted from the initial reaction trajectories of ANPs with different atmospheres.

# Initial decomposition pathway of different gas with ANPs

In the initial reaction, the reaction trajectory of gas molecules attacking ANPs is shown in Fig. 10. There are two pathways of  $CO_2$  attacking ANPs. One is that the Al atoms extract an oxygen atom from  $CO_2$  and then the remaining CO is released; the other is that the  $CO_2$  lies flat on the surface of the ANPs and reacts with three Al, and generates 2AlO and AlC. The reaction of NO<sub>2</sub> with ANPs is similar to that of  $CO_2$ . When CO attacks the surface of ANPs, AlC and AlO are generated. In addition to the reaction of NO similar to that of CO, it was also observed that with the assistance of another Al, the Al breaks the N–O bond to form NAIO.

# Evolution of the potential energy and total species of the system

Evolution of potential energy (PE) of the ANPs–CO<sub>2</sub> system is shown in Fig. 11. The PE curve initially decreases rapidly and then smoothly, indicating a release of energy during CO<sub>2</sub> oxidation. Taking 1500 K as an example, when an equilibrium is reached, the total potential energies released in the CO<sub>2</sub> density of (a) 0.055 g/cm<sup>3</sup>, (b) 0.11 g/cm<sup>3</sup>, and (c) 0.22 g/cm<sup>3</sup> are 2.412 × 10<sup>5</sup>, 2.575 × 10<sup>5</sup>, and  $3.305 \times 10^5$  kcal/mol, respectively. The greater the ambient atmosphere concentration, the higher the energy released. Additionally, the higher the temperature, the greater the asymptotic value of PE, indicating that increased temperature accelerates the rate of heat release.

In order to obtain the initial reaction information of the ANPs, a series of C++ scripts were applied to obtain the evolution of the fragments with time. The



Figure 11 Evolution of PE with CO<sub>2</sub> atmosphere at different temperature. The CO<sub>2</sub> densities of **a** 0.055 g/cm<sup>3</sup>, **b** 0.11 g/cm<sup>3</sup>, and **c** 0.22 g/cm<sup>3</sup> were selected as examples.



cutoff bond length is 1.1 times that of the standard single fragment in fragment analysis. Figure 12 shows the reduction of the four gaseous oxides with

time. As temperature rises, the number of  $CO_2$ , CO,  $NO_2$ , and NO shows an accelerated decline, indicating a dependence on temperature. In addition, the



Figure 13 a The number of CO gases formed by the CO<sub>2</sub>– ANPs system with time at 0.22 g/cm<sup>3</sup>. b The number of NO gases formed by the NO<sub>2</sub>– ANPs system with time at 0.23 g/cm<sup>3</sup>. c The number of N<sub>2</sub> gases formed by the NO<sub>2</sub>– ANPs system with time at 0.23 g/cm<sup>3</sup>. d The number of N<sub>2</sub> gases formed by the NO– ANPs system with time at 0.15 g/cm<sup>3</sup>.



curves of the different densities of NO<sub>2</sub> and NO distinctly intersect with each other. Especially for NO<sub>2</sub>, the curves of 0.22 and  $0.11 \text{g/cm}^3$  cross at 234 ps, indicating that high density promotes the oxidation of ANPs. CO2 reacts with ANPs to generate CO during ignition and combustion. Figure 13a shows the number of CO formed at high CO<sub>2</sub> density. CO molecules formed by CO<sub>2</sub> continues to increase from 0 to 240 ps. The higher the temperature, the more CO molecules are formed. NO and N2 are formed during the ignition and combustion of NO<sub>2</sub> and ANPs. Figure 13b shows the number of NO formed at high NO<sub>2</sub> density. Because NO<sub>2</sub> is more active than CO<sub>2</sub>, the amount of NO varies dramatically. At high density, the number of NO molecules initially increases rapidly and then decreases with time. Figure 13c, d shows the number of  $N_2$  formed under NO<sub>2</sub> and NO atmospheres, respectively. For the  $NO_2$  atmosphere, the number of  $N_2$  generated in NO<sub>2</sub> atmosphere slowly rises and then equilibrates. The higher the temperature, the more  $N_2$  is generated. For the NO atmosphere, the number of N2 rises and then decreases at 2500, 3000 and 3500 K.

However, the reaction of ANPs with NO hardly produces  $N_2$  at 1500 and 2000 K. The product evolutions at low1 and medium density are illustrated in Supporting Information (Figure S5 and S6).

#### Analysis of chemical bonds

The formation of three chemical bonds Al-O, Al-C, and Al-N is the main source of exothermic heat. Therefore, the evolutionary details of these three chemical bonds are helpful to understand the details of the reaction between ANPs and gaseous oxides. Figure 14 shows the evolution of these three chemical bonds with temperature. In CO2 atmosphere, the amount of Al-O is twice that of Al-C at a low density. With the increase in  $CO_2$  density, the number of Al–O is more than twice that of Al–C. This is because ANPs are more favorable for bonding with O under sufficient CO<sub>2</sub> supply. Due to the formation of carbon-containing chain products and carbon deposition, the number of Al–C slowly decreases at 0.22 g/ cm<sup>3</sup> when it reaches its highest peak. In CO atmosphere, the number of Al–O is substantially equal to



Figure 14 Evolution of Al-O, Al-C, and Al-N with time at different temperature.

the number of Al–C at low density. At 0.14 g/cm<sup>3</sup>, the formation of Al-O gradually dominates. This is also because the bonding strength of Al-O is much higher than that of Al-C, hence Al-O is more favorable to be formed when the amount of CO is sufficient. Since N atoms are feasible to form stable  $N_2$ , the amount of Al-O is not twice or the same as the amount of Al-N in the nitrogen oxides atmosphere. As shown in Fig. 14g-l, the amount of Al-O is more than three times that of Al–N in NO<sub>2</sub> atmosphere. The amount of Al-O is less than twice that of Al-N in NO atmosphere. In these two nitrogen oxide gases, as the density increases, more amount of Al-O is formed. In addition, under high-density gas, the amount of Al-N also shows a downward trend after reaching the peak similar to the carbon oxides.

# Conclusions

In this work, we performed a series of RMD simulations to study the oxidation processes of ANPs in various atmospheres. Based on the morphological evolution of ANPs, the mechanism of surface oxidation, the formation mechanism of chain-like products, and the principle of product void formation are explained during ignition and combustion. Fragment analysis and chemical bonds are further used to analyze changes in reactants, intermediates, and final products. The main results are summarized as follows:

- The ignition of ANPs in various atmospheres is 1. dependent on the combined effects of temperature and pressure. At the same time, it is also affected by changes in its morphology. In the CO atmosphere, the four forms of chain products include tilted chain, twisted chain, branched chain, and cyclic chain were observed. The chain products are repeatedly arranged in the form of AlC<sub>2</sub>, and the head and tail of the chain are  $C \equiv C$ . For the NO atmosphere, a similar chain structure is also formed, but the length of the chain is significantly reduced. The aluminum nitride on the surface enters the environmental phase in chains form when the aluminum oxide phase excludes the aluminum nitride phase at high density.
- 2. In  $CO_2$  atmosphere, ANPs form an oxide shell and rapidly expand into the outer space,

resulting in a void between oxide shell with Al core. Al atoms are transported from the core to the oxide shell through a bridge composed of Al atoms. The Al core gradually diffused outward and was eventually hollowed out. In addition, the final product has carbon deposits ( $C_{48}$  and  $C_{98}$ ) on the surface and core.

3. The analysis of Al–O, Al–C, and Al–N is helpful to understand the details of the reaction between ANPs and gaseous oxides. We found that ANPs ignited and burned Al–O and Al–C in a CO<sub>2</sub> and CO atmosphere at almost 2:1 and 1:1 ratios, respectively. However, the oxidation reaction of ANPs in NO<sub>2</sub> and NO does not show such a rule. This is because N<sub>2</sub> is generated as a stable structure during ignition and combustion compared to nitrogen oxide gas.

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# **Author contributions**

Conceptualization was performed by LS and XHJ; data curation was performed by LS; formal analysis was carried out by LS, FQZ, and SYX; project administration was done by XHJ; XHJ carried out supervision; writing of original draft was done by LS; writing, review, and editing was done by XHJ and FQZ.

# **Compliance with ethical standards**

**Conflict of interest** The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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