Chemical routes to materials



Self-supported phosphorus-doped CoMoO₄ rod bundles for efficient hydrogen evolution

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ABSTRACT

Herein, a series of phosphorus-doped CoMoO₄ rod bundles electrocatalysts on 3D nickel foams have been successfully synthesized and exploited as efficient hydrogen evolution reaction (HER) electrocatalyst in 1 M KOH. The incorporation of phosphorus into CoMoO₄ can significantly promote the HER activity, and the CoMoO₄ bundles phosphorized at 350 °C exhibited the optimum activity, which only required low overpotentials of 56 and 148 mV to deliver cathodic current densities of 10 mA cm⁻² and 100 mA cm⁻², respectively. This electrode also presented considerable long-term electrochemical stability with negligible delay after 30 h operation and 3000 accelerated cyclic voltammetry cycles. Our study provides a superior earthabundant catalyst for efficient HER in alkaline media, which also suggests that P doping engineering is an effective way to boost the HER activity of transition metal-based oxides.

Introduction

Water electrolysis driven by renewable energy sources is widely considered as a clean technique to produce pure hydrogen [1–3]. Generally, electrochemical water splitting proceeds through two half-reactions, namely hydrogen evolution reaction (HER) and oxygen evolution reaction (OER) [4]. As for HER, by far, the benchmark electrocatalysts are still limited to Pt-based precious metal catalysts [5, 6]. Their large-scale industrial application, however, is largely restricted

by the scarcity and high cost. Consequently, the development of effective HER catalysts using earthabundant materials is urgently needed [1, 2].

Recent years, transition metals oxides, phosphides, and sulfides have shown a great promise [7–19]. Among them, Co- and Mo-based oxides have been attracting increasing attention for designing low-cost electrocatalysts, due to the unique electron configurations [10]. Particularly, bimetallic CoMoO₄ composite oxide exhibits promising electrochemical activity because of the synergistic interactions

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between different elements [15–24]. However, for pure CoMoO₄, large overpotentials (> 200 mV) are usually required to deliver a current density of -10 mA cm^{-2} [16–18].

To boost the HER activity, there are some strategies like designing special microstructure and constructing heterostructures. For example, $CoMoO_4/Co_9S_8$ arrays and heterostructured $CoMoO_4-Co_2Mo_3O_8$ composite have been constructed for HER catalysts with enhanced performance [15, 22]. Further, elemental doping with a variety of element choice is proven to be an efficient way to modulate the physicochemical properties of parent oxides. For example, Li et al. [21]reported that phosphorusdoped β -CoMoO₄ nanoplates on Ti foil show greatly enhanced HER activity, which only requires an overpotential of 138 mV at -10 mA cm^{-2} in 1 M KOH.

Encouraged by this work, herein, we fabricated and evaluated 3D free-standing P-doped CoMoO₄ rod bundles anchored on NF as HER electrocatalysts. Phosphorus heterodoping was achieved through phosphorization with NaH₂PO₂ at higher temperatures. Electrochemical measurement confirmed that phosphorus incorporation can dramatically enhance the electrocatalytic activity toward HER in alkaline media. Among treated samples, the CoMoO₄ bundles phosphorized at 350 °C exhibited the highest activity with a low overpotential of 56 mV @ - 10 mA cm⁻² and a small Tafel slope of 78 mV dec⁻¹. High HER activity with excellent stability makes it a very promising Pt-free catalyst for hydrogen evolving application.

Experimental section

Chemicals and materials

Analytical grade of $(NH_4)_6Mo_7O_{24} \cdot 4H_2O$, $Co(NO_3)_2 \cdot 6H_2O$, $CO(NH_2)_2$, NH_4F , $NaH_2PO_2 \cdot H_2O$, KOH were all purchased from Aladdin Ltd. Nickel foam (NF, 1 mm in thickness) was obtained from Heze Tianyu Technology Co. Ltd. Deionized (DI) water from a Millipore system was used throughout the experiment.

Catalyst synthesis

Co(OH)F precursors on NFs were first synthesized via a facile hydrothermal method. Nickel foams $(2 \times 4 \text{ cm}^2)$ with a thickness of 1 mm were cleaned with 6 M HCl, acetone, ethanol and DI water for 10 min each in an ultrasound bath consecutively. Then, 2 mmol Co(NO₃)₂·6H₂O, 8 mmol NH₄F, and 10 mmol CO(NH₂)₂ were slowly dropped into 36 ml DI water to form a pink solution. NFs and the solution were transferred to a 50-ml Teflon-lined autoclave and kept at 100 °C for 6 h. After cooling, the NFs were dried at 60 °C for 6 h to obtain a Co(OH)F coated on NF. After that, 0.48 mmol of $(NH_4)_6$ Mo₇O₂₄·4H₂O was dissolved in 40 mL water and the solution was then transferred to a 50-mL Teflon-lined autoclave with Co(OH)F on NF. The autoclaves were sealed and heated at 140 °C for 4 h. After cooling naturally, NFs coated with CoMoO₄ were washed by DI water for 3 times and dried at 60 °C overnight. Finally, the foam specimens (at downstream) and NaH₂PO₂ (1 g, at upstream) were placed in 2 separate quartz boats in a tube furnace, which were calcined at 300 °C, 350 °C and 400 °C for 2 h using a ramping rate of 2 °C min⁻¹ in N₂ flow. As-obtained samples were taken out after cooling and named accordingly as P-CoMoO₄-300, P-CoMoO₄-350 and P-CoMoO₄-400. The loadings of CoMoO₄-based catalysts on NFs were estimated by a Sartorius BP211D high-precision balance (about 9–11 mg cm^{-2}).

Characterization

The crystal phases were examined by a Rigaku D/max-2600 X-ray diffraction diffractometer with Cu-K α radiation. Raman measurement was carried out using a Hobriba J-Y LabRAM spectroscope using 532 nm radiation. The sample morphologies were recorded by using a Hitachi SU-70 field-emission scanning electron microscopy (FE-SEM) and an FEI Tecai G2 F20 transmission electron microscopy (TEM). Further, the chemical states in catalysts were determined by a Thermo-fisher K-Alpha X-ray photoelectron spectroscopy (XPS) with Al K α radiation, and all the binding energies were referenced using the adventurous carbon (C1*s*) peak at 284.8 eV.

Electrochemical evaluation

A typical three-electrode configuration consisting of a foam electrocatalyst as working electrode, a graphite rod and a saturated calomel electrode (SCE) as counter electrode and reference electrodes were used. Electrochemical tests were carried out with a Chenhua CHI 660E electrochemical workstation in 1 M KOH electrolyte. Linear sweep voltammetry (LSV) curves were collected at a slow scan rate of 2 mV s⁻¹. All LSV curves were iR-corrected unless noted, and all potentials were converted to reversible hydrogen electrode (RHE) scale using the equation of RHE = Eversus SCE + $0.059 \times pH + 0.2415(V)$. Electrochemical impedance spectroscopy (EIS) measurements were recorded within a frequency range of 100 kHz to 0.01 Hz at an AC amplitude of 5 mV. Cyclic voltammetry (CV) test was carried out at non-faradaic region to probe the electrochemical doublelayer capacitance (C_{dl}) of various samples. A series of CV sweeps were performed at various scan rates (20, 40, 60, 80, 100 and 120 mV s⁻¹) in the region of -0.45 to -0.35 V versus SCE. By plotting the difference of current density $(j_a - j_c)$ between the anodic and cathodic sweeps at - 0.4 V versus SCE against the scan rate, a linear trend was observed. The slope of the fitting line is equal to twice of the geometric $C_{\rm dl}$, which is proportional to the effective electrode surface area (ECSA) of the catalysts. Therefore, the ECSAs of different samples can be compared based on their C_{dl} values. Preliminary electrochemical stability of the P-CoMoO₄-350 catalyst was investigated by galvanostatically at a current density of -10 mA cm⁻² for 30 h. Accelerated degradation test was performed using CV scans within 0 to -0.2 V versus RHE at 100 mV s⁻¹ for 3000 continuous scans.

Results and discussion

A synthesis procedure of P-doped CoMoO₄ bundles on NF is schematically shown in Fig. 1. The crystal phases of obtained catalysts on NF foams were determined by XRD (Fig. 2a). After the first hydrothermal process, the color of NF changes to reddish-brown (Fig. 1) and its diffraction peaks for the precursor could be perfectly indexed as Co(OH)F (JCPDS No. 50-0827) except for the peaks of NF support. The second hydrothermal process changed the sample color to purple, and all the other peaks

revealed a mixture of two phases, which can be indexed as the orthorhombic CoMoO₄ (JCPDS No. 15-0439) and CoMoO₄·0.9H₂O (JCPDS No. 14-0086) [25, 26]. Further phosphorization with NaH₂PO₂ at higher temperatures resulted in black foams and the corresponding XRD patterns suggested the samples treated at 350 and 400 °C transformed to monoclinic CoMoO₄ (β-phase, JCPDS No. 21-0868). No additional peaks from phosphides like CoP and MoP are detected, indicating no secondary phases were yielded during phosphorization treatment. XRD patterns of P-CoMoO₄-350 and pure CoMoO₄ annealed at 350 °C in the air (named as CoMoO₄-350-air) were also compared as shown in Fig. 2b. Phosphorization treatment did not change the diffraction peaks obviously. A closer look at the main peak of CoMoO₄ at $\approx 26.5^{\circ}$ shows a positive shift of 0.2°, suggesting the lattice contraction due to the incorporation of smaller P ions into CoMoO₄ lattice [27]. Raman spectra were further probed, and the results are shown in Fig. 2c. The Raman spectrum of pristine CoMoO₄ exhibits five Raman bands. The intense doublet at ~ 938 cm⁻¹ could be assigned to Mo=O bond, and the bands at 817 and 857 cm^{-1} are associated with O-Mo-O bond. Moreover, the peaks at 334 and 358 cm⁻¹ can be assimilated to Co-O-Mo bond [6, 28]. No obvious new peak is observed in phosphorized and air-annealed CoMoO₄ bundles. Compared with CoMoO₄-350-air sample, the peaks between 800 and 1000 cm⁻¹ (P-CoMoO₄-350) show slight blueshift by about 3 cm⁻¹ that could originate from P doping in CoMoO₄ lattice. These further confirm the formation of CoMoO₄ phase and P doping did not change the crystal structure, consistent with the XRD analysis.

The morphology and microstructure evolution of catalysts were investigated by SEM and TEM techniques. As shown in Fig. 3a, b, the Co(OH)F precursor clearly shows nanosaw feature which uniformly covers on NF surface. After the introduction of Mo ions in hydrothermal condition, the precursor transformed into bundles, as revealed in Fig. 3c, d. These bundles consist of several microrods with a square-like top of about 2 μ m in width and 10–15 μ m in length. The bundle surface is relatively smooth. Further phosphating treatment retains the bundle character, and a closer look at enlarged image of P-CoMoO₄-350 reveals that some particles are formed (Fig. 3e, f). With the increasing of phosphorization temperature, the bundle surface appears to be more

Figure 1 A diagram for the fabrication process of P-CoMoO₄ rods on Ni foam (NF). First, reddish-brown Co(OH)F nanosaws were hydrothermally grown on Ni foam, which were then hydrothermally transferred to purple CoMoO₄ rods. Finally, black P-CoMoO₄ rods were obtained via phosphorization treatment.





coarse (Figure S1), probably due to the hydrate dissociation [21]. The granule size also increases for samples at higher temperatures. In Fig. 3g, EDS mapping images from selected area clearly show even distribution of Mo, Co, O and P elements across the bundles, which also confirms the successful doping of P into CoMoO₄ lattice [29], and P doping content is estimated to be 5.6% in P-CoMoO₄-350 catalyst (Figure S2). The microrod character with some nanoparticles on the surface is also confirmed by TEM analysis (Fig. 4a). Figure 4b presents a high-resolution TEM (HR-TEM) image of P-CoMoO₄-350. Lattice fringes with spacings of 0.69 nm and 0.21 nm can be ascribed to the (001) and (222) planes of CoMoO₄, respectively, which are in good agreement with XRD analysis [30, 31]. It is anticipated that the unique bundle morphology constructed with many microrods could



(b)200 nm (\mathbf{d}) 1 μm 0 Co (g) Mo Р 5 µm (a) 1 μm 5 nm

Figure 3 SEM images of **a**, **b** Co(OH)F, **c**, **d** CoMoO₄, **e**, **f** P-CoMoO₄-350 and **g** EDS elemental mapping images of Co, Mo, O, and P in P-CoMoO₄-350.



facilitate both mass transfer and diffusion of gaseous products and provide a high concentration of active sites that are very beneficial to catalytic activity enhancement.

The chemical states and surface electronic structure of the $CoMoO_4$ and $P-CoMoO_4$ -350 samples were

examined by XPS. The survey spectrum of fresh CoMoO₄ in Figure S3 clearly verifies the presence of Co, Mo, O and Ni elements. For P-CoMoO₄-350 catalyst, two new weak peaks at ~ 134.1 eV and 191.7 eV can be ascribed to P 2*p* and P 2*s* signals, respectively. High-resolution P 2*p* spectrum in Fig. 5a

demonstrates a strong peak at 134.1 eV in treated CoMoO₄, which is associated with surface oxidized P–O species [32]. This is another solid evidence of P doping, in accordance with EDS analysis. In the Co 2p region (Fig. 5b), two peaks at ~ 781.1 and 797.5 eV are attributed to the Co 2p3/2 and 2p1/2levels, respectively. [19, 33] After phosphating, the Co 2p3/2 peak of P-CoMoO₄-350 shifts to higher energy, which implies that the electronic structure of Co centers has been modified through P doping [34]. The Mo 3d spectrum of bare sample (Fig. 5c) exhibits two main peaks. The one at low binding energy can be specified as Mo 3d5/2, while the other one is Mo 3d3/22. The splitting width of two peaks is 3.1 eV, signifying a majority of Mo^{6+} oxidation state [35]. Those peaks shift positively by about 0.2 eV in P-doped CoMoO₄, and more importantly, two new peaks located at 230.4 eV and 233.6 eV due to Mo⁵⁺ species can be deconvoluted. In O 1s spectra (Fig. 5d), two fitted peaks at 530.5 and 532.1 eV can be assigned to the typical metal-oxygen bond (Mo-O-Co) and surface absorbed hydroxyls, respectively [22, 36]. Whereas a new peak at the high energy of 531.4 eV appeared after phosphating which is attributed to P-O [37].

The HER catalytic performance of pristine CoMoO₄ and P-CoMoO₄ samples treated at different temperatures were evaluated and compared. The NF support, commercial 20% Pt/C (Johnson Matthey), Co(OH)F precursor and CoMoO₄-350-air were also examined in the same condition for reference. LSV polarization curves after iR drop compensation are plotted in Fig. 6a and Figure S4. It can be found that benchmark Pt/C exhibits superior activity with zero onset overpotential, while the NF and Co(OH)F precursor show very poor HER performance. Apparently, the phosphated CoMoO₄ bundles display much better activity toward HER than undoped one. Among CoMoO₄-based catalysts, the P-CoMoO₄-350 demonstrates the highest activity (Table S1) with the lowest overpotential of 56 mV at a cathodic current density of -10 mA cm^{-2} . For comparison, the required overpotentials to supply the same current density are 169 mV, 73 mV and 63 mV for pure CoMoO₄, P-CoMoO₄-300 and P-CoMoO₄-400, respectively. To afford a higher current density of -100 mA cm^{-2} , the P-CoMoO₄-350 requires an overpotential of 148 mV, still much lower than other tested catalysts except Pt/C (Table S1). Such superior HER electrocatalytic activity of P-CoMoO₄-350

Figure 5 High-resolution XPS spectra of CoMoO₄ and P-CoMoO₄-350, **a** P 2p, **b** Co 2p, **c** Mo 3*d* and **d** O 1*s*.







electrode outperforms other reported Mo-based oxides electrocatalysts, for example, $CoMoO_4$ - $Co_2Mo_3O_8$ (58 mV@ – 10 mA cm⁻²) [18], P-doped- β -CoMoO₄ nanoplates (138 mV@ – 10 mA cm⁻²) [21] and CoMoO₄ nanowires (121 mV@ – 10 mA cm⁻²) [23], (see Table S2 for details).

The HER kinetics of these electrodes is analyzed by corresponding Tafel plots. The smallest Tafel slope of 20% Pt/C (38 mV dec⁻¹) is in agreement with a previous report [38], which therefore confirms the reliability of our electrochemical test. As expected, P-CoMoO₄-350 exhibits a Tafel slope of 78 mV dec⁻¹, which is smaller than that of P-CoMoO₄-300 (108 mV dec^{-1}), P-CoMoO₄-400 (90 mV dec^{-1}), pure-CoMoO₄ (112 mV dec $^{-1}$). A smaller Tafel slope implies the higher HER catalyst activity, [39] suggesting the higher intrinsic activity of P-CoMoO₄-350. Meanwhile, the Tafel slope is usually used to analysis the HER mechanism and rate-determining step. In this work, the experimentally obtained Tafel slope of P-CoMoO₄-350 implies that the HER reaction takes place via a Volmer-Heyrovsky mechanism where Heyrovsky process is considered as the rate-limiting step.

To further study the electron transfer kinetics at catalyst–electrolyte interface, impedance spectra

were collected and the Nyquist plots are given in Fig. 6c. By fitting the impedance spectra with a simplified Randles equivalent circuit (insert in Fig. 6c), the charge transfer resistances (R_{ct}) were obtained and listed in Table S1. It is obvious that P-CoMoO₄-350 shows a much smaller R_{ct} value (2.5 Ω) than that of P-CoMoO₄-300 (6.3 Ω), P-CoMoO₄-400 (2.9 Ω), bare CoMoO₄ (30.1 Ω) and CoMoO₄-350-air (29.0 Ω), indicating the fastest charge transfer process among the CoMoO₄-based catalysts, confirming the abovementioned rapid HER kinetics in P-CoMoO₄-350 catalyst. To redouble study the intrinsic catalytic activity of P-CoMoO₄-350, the turnover frequencies (TOFs) that reflect all the active sites involved in the HER reaction were measured according to references [40, 41]. As shown in Figure S5, the TOF value of P-CoMoO₄-350 catalyst is 0.59 s^{-1} at an overpotential of 100 mV, much greater than that of CoMoO₄ (0.092 s^{-1}) . The high TOF value of P-CoMoO₄-350 indicates its high intrinsic activity due to the successful P doping, which can improve the electrical conductivity and reduce the hydrogen adsorption free energy, as supported by theoretical calculations by Li et al. in P-doped $CoMoO_4$ nanoplates [42]. These are also evidenced by impedance spectra and electrochemical performance in P-doped $CoMoO_4$ rod bundles in our study.

In order to measure the density of active sites, the electrochemically active surface areas (ECSAs) of the foam catalysts were determined by using electrochemical double-layer capacitance (C_{dl}) , which were calculated using CV scans from - 0.35 to - 0.45 V versus SCE at scan rates range of 20–120 mV s⁻¹ (Figure S6). As depicted in Fig. 6d, P-CoMoO₄-350 exhibits the highest C_{dl} value (4.03 mF cm⁻²) among tested samples, which is about 4.3 and 70 times higher than that of pristine $CoMoO_4$ (0.94 mF cm⁻²) and Co(OH)F (0.06 mF cm⁻²). The dramatically increased ECSA of P-CoMoO₄-350 reflects the more accessible active sites, due to the P uniformly dope in rods; the rough surface also enables greatly increased exposed surface area [42]. The ECSAs were calculated from C_{dl} values using the specific capacitance value of a standard 1 cm² flat surface which is usually between 20 and 60 μ F cm⁻². An average value of 40 μ F cm⁻² was used for calculation [17]. Moreover, LSV plots normalized by ECSA and mass of typical electrodes are given in Figure S7a, b, which further confirms the high specific activity of P-CoMoO₄-350 catalyst.

In addition to HER activity, considerable stability of a catalyst is also very important for practical application. The durability was first tested by accelerated CV cycling at a scan rate of 100 mV s⁻¹. As compared in Fig. 7a, the polarization curve of P-CoMoO₄ approximates to the initial one even after 3000 CV cycles. At the current densities of -10 and -100 mA cm^{-2} , the initial overpotentials are 48 and 154 mV, respectively. After test, the corresponding values reached 51 and 157 mV. Furthermore, chronopotentiometric test at -10 mA cm^{-2} was carried out (insert in Fig. 7a). Its overpotential shows a slight increase of 0.7 mV after testing for 30 h. Both CV and chronopotentiometric tests prove its strong stability. The XRD peaks turn to be weak after test (Fig. 7b) but a comparison of Raman results (Fig. 7c) for P-CoMoO₄-350 catalyst before and after test suggests no detectable phase change occurred. SEM images in Fig. 7d show that the rod bundle feature is maintained, and the enlarged view in the insert reveals a more rough surface with the formation of more particles. The excellent stability of provides more evidence of P-CoMoO₄-350 to be a reliable catalyst in alkaline water reduction.

Based on the discussion above, it is clear that phosphorus doping is an effective strategy in

Figure 7 Comparisons of P-CoMoO₄-350 catalyst before and after 3000 CV cycles, **a** polarization curves (insert: chronoamperometric curve without iR correction at the constant overpotential of -1.13 V, **b** XRD patterns, **c** Raman spectra and **d** SEM images.



engineering HER activities and the greatly enhanced activity of phosphorized $CoMoO_4$ bundles can be ascribed to the following combined reasons: (1) The unique bundle morphology with rough surface enables the exposure of more active sites with high accessibility and facile diffusion of the electrolyte and gaseous product, (2) direct anchoring of bundles on conductive NF eliminates the use of additional polymer binder and thus enables strong interfaces at the electrocatalyst/electrode, and (3) P doping modulates the electronic structure that enables greatly enhanced HER kinetics.

Conclusion

Phosphorus-doped CoMoO₄ rod bundles were successfully prepared by combined hydrothermal and phosphorization processes. The introduction of P ions into CoMoO₄ lattice is evidenced by combined XRD, Raman, EDS and XPS techniques. The electrochemical test confirmed that phosphorus incorporation can efficiently promote the electrocatalytic performance. The self-supported CoMoO₄ bundles phosphorized at 350 °C exhibited the highest activity in 1 M KOH with low overpotentials of 56 mV @ -10 mA cm^{-2} and 148 mV @ -100 mA cm^{-2} as well as a low Tafel slope of 78 mV dec^{-1} . The excellent HER activity and high stability enable P-CoMoO₄-350 a promising HER electrocatalyst in basic media. Our work suggests that P doping engineering is an effective route to tune the HER performance of transition metal oxides.

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Compliance with ethical standards

Conflicts of interest The authors declare that they have no conflict of interest.

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