## **Energy materials**



## Synthesis of 3D flower-like cobalt nickel phosphate grown on Ni foam as an excellent electrocatalyst for the oxygen evolution reaction

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Received: 1 August 2017 Accepted: 21 September 2017 Published online: 10 October 2017

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### ABSTRACT

Low-cost and highly active water oxidation electrocatalysts are increasingly demanded to improve the efficiency of water splitting. Herein, for the first time, we developed a facile strategy to design a three-dimensional flower-like cobalt nickel phosphate as such an electrocatalyst. The cobalt nickel phosphate was directly supported on Ni foam  $((Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni)$  via a simple two-step hydrothermal method. This method offers several advantages including low cost, facile synthesis and binder-free. Benefiting from its three-dimensional flower-like nanostructure with a big specific surface, high conductivity and the synergic effect of metal ions are the as-synthesized  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$  reveals excellent oxygen evolution reaction activity in alkaline electrolyte. It exhibits a low overpotential of 273 mV to reach a current density of 10 mA cm<sup>-2</sup> and a Tafel slope of 59.3 mV dec<sup>-1</sup> in 1.0 mol L<sup>-1</sup> KOH. Notably, the  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$  also operates efficiently in 30 wt% KOH, capable of affording an anodic current of 10 mA cm<sup>-2</sup> at a lower overpotential of 246 mV and retains its catalytic performance for at least 30 h.

## Introduction

Developing efficient electrocatalysts for oxygen evolution reaction (OER) is imperative for energy storage and conversion systems, including rechargeable metal–air batteries, reversible fuel cells and water splitting [1, 2]. Unfortunately, the OER is kinetically sluggish, needing large overpotential and high activation energy to form the O=O double bond [3–5]. Up

Transition metal phosphates [9–12], phosphides [13–15], oxides [16–18], hydroxides [19–21] and

to now, the noble metal oxides, such as  $RuO_2$  and  $IrO_2$ , are still considered as the most active electrocatalysts to promote the OER, while the scarcity and prohibitive cost have heavily hindered their widespread commercialization [6–8]. Accordingly, many efforts are focused on exploring earth-abundant, costeffective and highly efficient OER electrocatalysts.

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sulfides [22-24] have exhibited a tremendous potential to replace those precious electrocatalysts for OER. For metal phosphates, they have  $PO_4^{3-}$  or  $HPO_4^{2-}$  to carry and accelerate the motion of protons (H<sup>+</sup>) and thereby increase the OER catalytic activity [4, 9]. In 2008, a Co-Pi reported by Kanan and Noreca [9], showing superior OER catalytic performance in neutral solution. After that, the earth-abundant and low-cost metal phosphates electrocatalysts have been extensively designed [25-34], such as NiCo<sub>2</sub>(PO<sub>4</sub>)<sub>2</sub> [35], LiNi<sub>1-x</sub>Fe<sub>x</sub>PO<sub>4</sub>/mesoporous carbon (LiNi<sub>1-x</sub>)  $Fe_xPO_4/C$  [36],  $Co_3(PO_4)_2/N$ -doped carbon layers (Co<sub>3</sub>(PO<sub>4</sub>)<sub>2</sub>@N–C) [37], Co- and Fe-doped nickel phosphate-Na-nanotubes (NiPNa(Co<sub>2</sub>Fe<sub>2</sub>)-NTs) [38], iron-doped nickel phosphate (Ni:Pi-Fe/NF) [39], nickel-iron phosphates (NiFe-P) [40], and so on. Nonetheless, the operations of these electrocatalysts usually employ the polymeric binders. These will reduce the conductivity, block the active sites and finally decrease the catalytic performance of electrocatalysts. In order to address this challenge, Sun et al. synthesized a Co-Pi nanoarray on Ti mesh (Co-Pi NA/Ti) by an electrodeposition method [41]. The Co– Pi NA/Ti showed excellent OER catalytic activity in neutral media. The authors concluded that the threedimensional (3D) nanoarray configuration of Co-Pi NA/Ti not only exposed more active sites but also accelerated the diffusion of electrolytes and oxygen. Although the above excitingly advances have been reported, further studies of metal phosphates or the optimization of their OER catalytic performance are undoubtedly crucial. In this regard, an efficient method to prepare these materials is required. Recently, we have noticed that the hydrothermal anion exchange method is effective to synthesize materials [42]. Following this way, the products will retain the unique morphology of the corresponding precursors after an anions exchange process. However, it is still a challenge to utilize this method to prepare the metal phosphates.

In this work, we demonstrated a facile two-step hydrothermal method to synthesize 3D flower-like  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$ . This 3D flower-like  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$  combines the advantages of 3D porous architecture with a large surface area for fast mass transport, efficient electron transfer from conductive Ni foam to  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2$  and the synergistic effect between the optimized ratio of Co and Ni. It presents highly activity and stability for OER in alkaline medias, out-performing the majority reported non-noble metal materials and even comparable to the active  $RuO_2$  electrocatalyst.

## **Experimental sections**

## Materials

CoCl<sub>2</sub>·6H<sub>2</sub>O, NiCl<sub>2</sub>·6H<sub>2</sub>O, Na<sub>2</sub>HPO<sub>4</sub>·12H<sub>2</sub>O, urea and NH<sub>4</sub>F were purchased from Aladdin. Ni foam was purchased from Kunshan Electronic Limited Corporation. All the reagents were used without any further purification.

## Synthesis of precursor Co<sub>0.5</sub>Ni<sub>0.5</sub>-LDH/Ni

The precursor  $Co_{0.5}Ni_{0.5}$ -LDH/Ni [XRD pattern is exhibited in Fig. S1, Supporting Information (SI)] was prepared by the following processes. Firstly, Ni foam (3 cm × 3 cm) was cleaned by sonication sequentially in 3 mol L<sup>-1</sup> HCl solution and absolute ethanol for 15 min each. Then, 1.5 mmol CoCl<sub>2</sub>·6H<sub>2</sub>O, 1.5 mmol NiCl<sub>2</sub>·6H<sub>2</sub>O, 5.0 mmol urea and 0.5 mmol NH<sub>4</sub>F were fully dissolved in 50 mL distilled water to form a homogeneous solution under agitated stirring. The obtained solution was transferred into a Teflonlined autoclave (80 mL), in which contained a cleaned Ni foam, sealed and kept at 120 °C for 6 h. After cooling down to room temperature, the Ni foam was taken out and washed with deionized water thoroughly before being dried at 60 °C.

## Synthesis of (Co<sub>0.5</sub>Ni<sub>0.5</sub>)<sub>3</sub>(PO<sub>4</sub>)<sub>2</sub>/Ni

To obtain the  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$ , 5.0 mmol Na<sub>2</sub> HPO<sub>4</sub>·12H<sub>2</sub>O was mixed with 50 mL distilled water under stirring to form a homogeneous solution. Then, 3 mol  $L^{-1}$  HCl was dropped into the above solution to adjust the pH value to 8.5, which was tested by the pH meter. The obtained solution was placed into an autoclave, containing the precursor Co<sub>0.5</sub>Ni<sub>0.5</sub>-LDH/ Ni. Moreover, the autoclave was heated at 180 °C for 12 h and cooled down to room temperature naturally. The product was severally washed by distilled water and ethanol for three times to remove the impurities on the surface and then dried at 60 °C overnight. The mass loading of (Co<sub>0.5</sub>Ni<sub>0.5</sub>)<sub>3</sub>(PO<sub>4</sub>)<sub>2</sub>/ Ni is  $4.4 \text{ mg cm}^{-2}$ . The sample synthesized without Ni foam was denoted as  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2$ nanoparticles.

Scanning electron microscopy (SEM) images were obtained by a Philips SEM-XL30S microscope operated at 15 kV. High-resolution transmission electron microscope (HRTEM, JEOL JEM-2100F) coupled with an energy-dispersive X-ray spectroscopy (EDS) analyzer was performed at an accelerating voltage of 200 kV. Nitrogen sorption isotherm was achieved by a Micromeritics TriStar 3000 analyzer at 77 K. The X-ray diffraction (XRD) pattern was conducted by a MSAL-XD2 X-ray diffractometer with Cu K $\alpha$  radiation ( $\lambda = 1.5406$  Å). The inductively coupled plasma optical emission spectrometer (ICP-OES) was measured on Perkin-Elmer Optima 2000DV. The X-ray photoelectron spectroscopy (XPS) was tested by an ESCALab250.

#### **Electrochemical measurements**

Electrochemical measurements were conducted in a standard three-electrode system, in which the (Co<sub>0.5</sub>Ni<sub>0.5</sub>)<sub>3</sub>(PO<sub>4</sub>)<sub>2</sub>/Ni, Pt foil and Hg/HgO electrodes were severally used as the working, counter and reference electrodes. 1 mol L<sup>-1</sup> KOH solution was employed as the electrolyte. All potentials were converted via the Nernst equation  $(E_{\rm RHE} = E_{\rm Hg})$  $_{HgO}$  + (0.098 + 0.059pH) V). The polarization curves were obtained by linear sweep voltammetry (LSV) at a scan rate of 1 mV s<sup>-1</sup>. The Tafel slope was obtained according to Tafel equation  $(n = b \log j + a)$ , where *a* is a constant, *j* is the current density, *b* is the Tafel slope and  $\eta$  is the overpotential). The overpotentials (n) were calculated according to the equation  $(\eta = E_{RHE} - 1.23 \text{ V})$ . Electrochemical impedance spectroscopy (EIS) was analyzed in the frequency range of 10 kHz to 10 mHz with an amplitude of 5 mV. Chronoamperometry was performed at certain potentials.

The roughness factor ( $R_f$ ) was determined from a ratio of the electrochemically active surface area (ECSA) to the electrode geometrical surface area (1.0 cm<sup>2</sup>) [43]. The ECSA was evaluated by the electrochemical double-layer capacitance ( $C_{dl}$ ) according to the equation (ECSA =  $C_{dl}/C_s$ , where  $C_s$  is 40 uF cm<sup>-2</sup>, the specific electrochemical double-layer capacitance of an atomically smooth surface [43, 44]). The  $C_{dl}$  was evaluated by cyclic voltammograms measured in a non-Faradaic region from 1.004 to 1.078 V versus RHE at scan rates ranging from 5 to

30 mV s<sup>-1</sup>.  $2C_{dl}$  is equal to the linear slope of capacitive currents versus scan rates [45].

## **Results and discussion**

# Structure and morphology of (Co<sub>0.5</sub>Ni<sub>0.5</sub>)<sub>3</sub>(PO<sub>4</sub>)<sub>2</sub>/Ni

The flower-like  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2$  grown on Ni foam is prepared through a simple two-step hydrothermal process, as shown in Scheme 1. In step I, the  $CO_3^{2-}$ and OH<sup>-</sup> ions are released by the hydrolysis of urea and gradually co-precipitated with  $Co^{2+}$  and  $Ni^{2+}$ ions to form the  $Co_{0.5}Ni_{0.5}$ -LDH/Ni (yellow–green), which has a 3D flower-like structure. Furthermore, the  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$  (deep brown) is obtained through a facile in situ anions exchange process using the  $Co_{0.5}Ni_{0.5}$ -LDH/Ni as a sacrificial template (step II). In step II, the  $Na_2HPO_4$  is employed as the source of  $PO_4^{3-}$  and the related reactions may follow Eqs. (1)–(4):

$$H_2 PO_4^- = H PO_4^{2-} + H^+$$
(1)

$$HPO_4^{2-} = PO_4^{3-} + H^+$$
(2)

$$\begin{array}{l} Co_{0.5}Ni_{0.5}(CO_3)_{0.125}(OH)_6\cdot 0.38H_2O \ + \ 6.25H^+ \\ = \ 0.5Co^{2+} + \ 0.5Ni^{2+} + \ 0.125CO_2 + \ 6.505H_2O \end{array} \eqno(3)$$

$$1.5 \text{Co}^{2+} + 1.5 \text{Ni}^{2+} + 2 \text{PO}_4^{3-} = (\text{Co}_{0.5} \text{Ni}_{0.5})_3 (\text{PO}_4)_2$$
(4)

From the SEM images revealed in Fig. 1a, b, the 3D flower-like structure has still retained after exchanging the  $Co_{0.5}Ni_{0.5}$ -LDH/Ni to  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$ . Figure 1c, d shows the "petals" of "flower" are composed of ultrathin nanosheets, while the thicker nanoplates grew on "petals" are also observed. Meanwhile, the structure of the whole "flower" is exhibited in Fig. S2, revealing the diameter of "flower" is about 3.6 um. Moreover, the  $N_2$  adsorption-desorption isotherm (Fig. S3) analysis indicates the mesoporosity and a large specific area for (Co<sub>0.5</sub>  $Ni_{0.5}$ )<sub>3</sub>(PO<sub>4</sub>)<sub>2</sub>/Ni of about 58 m<sup>2</sup> g<sup>-1</sup>. The porous architecture of (Co<sub>0.5</sub>Ni<sub>0.5</sub>)<sub>3</sub>(PO<sub>4</sub>)<sub>2</sub>/Ni along with a large surface area will facilitate the mass diffusion/transport and expose a large number of catalytic active sites for OER [35, 36, 46, 47]. The HRTEM image in Fig. 1e clearly presents the lattice fringes of d = 0.344 nm, corresponding to the (111) crystal plane of  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$ . In Fig. 1f, the



Scheme 1 Schematic diagrams of the synthesis of  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$ . The fuchsia, green, aquamarine blue, violet and yellow spheres severally represent the  $Co^{2+}$ ,  $Ni^{2+}$ , urea,  $NH_4F$  and  $Na_2HPO_4$ .



Figure 1 a SEM images of  $Co_{0.5}Ni_{0.5}$ -LDH/Ni; **b**-e are severally the SEM, TEM and HRTEM images of  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$ ; and **f** EDS element mappings of  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$ .



uniform distribution of elements of Co, Ni, P and O in  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$  is also observed.

As the XRD pattern shown in Fig. 2a, the characteristic peaks of the as-synthesized sample are matched well with the standard diffraction patterns of monoclinic  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2$  (JCPDS No. 75-1328) and Ni foam (JCPDS No. 65-2865), suggesting the successful synthesis of  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$  after

the phosphorylation. The EDS analysis of  $(Co_{0.5} Ni_{0.5})_3(PO_4)_2/Ni$  in Fig. 2b gives the mole ratio of  $Co^{2+}$  and  $Ni^{2+}$  is 48.6: 51.4 (~ 0.5:0.5), which can be further confirmed by the ICP results exhibited in Table S1, but the Cu element is coming from the copper mesh.

In addition, the XPS spectra in the Co 2p region (Fig. 3a) show two sharp peaks at 782.8 eV ( $2p_{3/2}$ )



and 798.8 eV  $(2p_{1/2})$  with two strong satellite peaks separately at 787.2 and 804.5 eV, indicating the presence of Co<sup>2+</sup> [4, 37]. Simultaneously, Fig. 3b exhibits two strong peaks located at 858.1 and 875.7 eV in the Ni 2*p* region are severally corresponding to Ni  $2p_{3/2}$  and Ni  $2p_{1/2}$ , attributed to Ni<sup>2+</sup> [3]. Moreover, the peaks at 863.7 and 882.5 eV are the satellite peaks of Ni  $2p_{3/2}$  and Ni  $2p_{1/2}$ . The oxidation

states of Co and Ni presented in  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$  are benefited for OER. As shown in Fig. 3c, the peaks at 133.6 and 134.4 eV of P 2*p* are related to the pentavalent tetra-bonded phosphorous in  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$  [4, 37]. In Fig. 3d, the peaks at 531.6 and 532.9 eV are corresponding to the core levels of O 1*s* in phosphate species, adsorbed water and/or possibly adsorbed O<sub>2</sub> [3, 37].

### OER activity of (Co<sub>0.5</sub>Ni<sub>0.5</sub>)<sub>3</sub>(PO<sub>4</sub>)<sub>2</sub>/Ni

The above characterizations imply that the as-synthesized  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$  may be active for OER. Therefore, we further estimate its electrocatalytic properties for OER in 1 mol L<sup>-1</sup> KOH solution by the LSV measurement at a scan rate of 1 mV s<sup>-1</sup>. For comparison, the  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2$  nanoparticles and commercial RuO<sub>2</sub> loaded on Ni foam (all loading 4.4 mg cm<sup>-2</sup>, denoted as  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2$  Nps/Ni and RuO<sub>2</sub>/Ni [3]) and bare Ni foam were also tested. Figure 4a exhibits the polarization curves. The oxidation peaks at ~ 1.3 V versus reversible hydrogen electrode (RHE) are corresponding to the transition from  $M^{2+}$  (Co<sup>2+</sup> and/or Ni<sup>2+</sup>) to  $M^{3+}$  (Fig. S4) [3, 6]. Owing to the large oxidation peaks, the onset potentials of (Co<sub>0.5</sub>Ni<sub>0.5</sub>)<sub>3</sub>(PO<sub>4</sub>)<sub>2</sub>/Ni, (Co<sub>0.5</sub>Ni<sub>0.5</sub>)<sub>3</sub>(PO<sub>4</sub>)<sub>2</sub> Nps/ Ni are inaccurate. Therefore, we report the overpotentials at 10 mA cm<sup>-2</sup> here. As shown in Fig. 4a, the bare Ni foam has poor OER catalytic performance with an overpotential of 502 mV to achieve a current density of 10 mA cm<sup>-2</sup> ( $\eta_{10}$ ), while (Co<sub>0.5</sub>Ni<sub>0.5</sub>)<sub>3</sub>(PO<sub>4</sub>)<sub>2</sub> Nps/Ni is active for OER with a lower  $\eta_{10}$  of 305 mV. What should be noted is that the (Co<sub>0.5</sub>Ni<sub>0.5</sub>)<sub>3</sub>(PO<sub>4</sub>)<sub>2</sub>/ Ni possesses significantly enhanced catalytic activity, because it just requires an overpotential of 273 mV to generate a current density of 10 mA cm<sup>-2</sup>, which even



Figure 4 a Polarization curves (the inset exhibits the expanded region around the electrocatalytic onset of polarization curves); b Tafel plots; c EIS of electrocatalysts measured at a potential of

1.57 V; **d** chronoamperometric response of  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$ ; and **e**, **f** SEM images after stability measurement in different magnifications.

lower than 296 mV for RuO<sub>2</sub>/Ni. In addition, although the  $\eta_{10}$  of  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$  at 273 mV is larger than 220 mV for Ni:Pi-Fe/NF, it is superior to those of the  $Co_3(PO_4)_2$  (380 mV) [4],  $NiCo_2(PO_4)_2$  (347 mV) [35], LiNi<sub>1-x</sub>Fe<sub>x</sub>PO<sub>4</sub>/C (311 mV) [36], Co<sub>3</sub>(PO<sub>4</sub>)<sub>2</sub>/N-C (317 mV) [37], NiPNa(Co<sub>2</sub>Fe<sub>2</sub>)-NTs (300 mV) [38],  $Ni_{r}Co_{2r}(OH)_{6r}@Ni (> 295 \text{ mV})$  [48],  $NiCo_{2}O_{4}/CNTs$ (> 320 mV) [49] and the other recently noble metal electrocatalysts in Table S2. Efficient performance of  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$  may be attributed to the uniquely 3D flower-like structure along with a large surface area. Interestingly, we found that the  $(Co_{0.5})$  $Ni_{0.5})_3(PO_4)_2/Ni$  is not a simple mixture after synthesized, while the synergistic effect between the optimized ratio of Co and Ni can also improve the OER activity of this material, as shown in Fig. S5.

To further evaluate the OER kinetics of the asprepared materials, the Tafel plots are modeled from the corresponding LSV curves and shown in Fig. 4b. It can be judged that the  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$  has more favorable OER catalytic kinetics and higher performance with a small Tafel slope of  $59.3 \text{ mV} \text{ dec}^{-1}$  comparing to those of 119.0 mV  $\text{dec}^{-1}$ for  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2$  NPs/Ni and 87.0 mV dec<sup>-1</sup> for RuO<sub>2</sub>/Ni. Moreover, the efficient OER activity of (Co<sub>0.5</sub>Ni<sub>0.5</sub>)<sub>3</sub>(PO<sub>4</sub>)<sub>2</sub>/Ni is supported by the electrochemical impedance spectroscopy (EIS) presented in Fig. 4c. The impedance responses of EIS are fitted by the equivalent circuit inset in Fig. 4c [50]. This equivalent circuit includes a solution resistance  $(R_s)$ with two parallel comprised of constant phase elements (CPE1, CPE2) and resistors (R<sub>ct</sub>, R<sub>p</sub>). Especially, the  $R_{\rm ct}$  represents the charge transfer resistance, caused by redox reactions on the interfaces of electrocatalyst and electrolyte. It is observed that the  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$  possesses the lower  $R_{\rm ct}$  value (1.63  $\Omega$ ) with respect to that of (Co<sub>0.5</sub>  $Ni_{0.5}$ <sub>3</sub>(PO<sub>4</sub>)<sub>2</sub> Nps/Ni (3.76  $\Omega$ ). This result implies the (Co<sub>0.5</sub>Ni<sub>0.5</sub>)<sub>3</sub>(PO<sub>4</sub>)<sub>2</sub>/Ni has better charge transfer ability in the electrochemical process of OER due to it directly grown on Ni foam with binder-free.

To further understand the superior performance of  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$ , the electrochemically active surface areas (ECSA) and roughness factors ( $R_f$ ) of the as-prepared samples were evaluated by the electrochemical double-layer capacitance ( $C_{dl}$ , Fig. S6a–d). Figure S6e shows the ESCA and  $R_f$  values of ( $Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$  are severally 1385.0 cm<sup>2</sup> and 1385.0, which are obviously larger than those of the ( $Co_{0.5}Ni_{0.5})_3(PO_4)_2$  Nps/Ni (247.5 cm<sup>2</sup> and 247.5),

 $RuO_2/Ni$  (550.0 cm<sup>2</sup> and 550.0) and bare Ni foam (42.5 cm<sup>2</sup> and 42.5). Therefore, it can be also concluded that the superior OER catalytic activity of  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$  is partially associated with its uniquely flower-like mesoporous architecture that possesses the large electrochemically active surface area to expose many active sites on the surface of the electrocatalyst.

The chronoamperometry measurement was carried out at the potential of 1.57 V versus RHE to probe the stability of  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$  in 1 mol L<sup>-1</sup> KOH solution, as shown in Fig. 4d. One can see that the current density of  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$  just reveals a little decay after 30 h continuous measurements, demonstrating the superior stability of  $(Co_{0.5}Ni_{0.5})_3$  $(PO_4)_2/Ni$  for OER. The great water oxidation stability of  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$  may benefit from its high structure durability, because the 3D flower-like morphology of  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$  reveals unobviously change even after long-term OER durability measurement, as SEM images shown in Fig. 4e, f.

Moreover, because the practical water electrolysis of electrocatalysts usually operates in 30 wt% KOH [3], the OER catalytic activity of  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/$ Ni was also evaluated under this circumstance. In Fig. 5a, b, the  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$  exhibits a larger OER current in 30 wt% KOH with respect to that of in 1 mol  $L^{-1}$  KOH. The  $\eta_{10}$  of  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$  in 30 wt% KOH is 246 mV, which is smaller than the 273 mV in 1 mol  $L^{-1}$  KOH. As shown in Fig. 5c, the observation of small OER activity decays for continuing OER electrolysis for 30 h at different operating potentials, suggesting the great durability of  $(Co_{0.5})$ Ni<sub>0.5</sub>)<sub>3</sub>(PO<sub>4</sub>)<sub>2</sub>/Ni even under strongly alkaline solution. Simultaneously, Fig. 5d shows the volume-time curves reveal the Faradaic efficiency of  $(Co_{0.5})$ Ni<sub>0.5</sub>)<sub>3</sub>(PO<sub>4</sub>)<sub>2</sub>/Ni is mostly 100% with the volume of O<sub>2</sub> generating equal to the theoretical one. The high catalytic activity and remarkable stability imply the considerable potential of  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$  for practical application.

In general, the excellent OER activity and stability of  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$  in alkaline electrolytes could be involved the following factors: (1) the obtained  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$  has a unique 3D flower-like mesoporous architectures and big specific surface area, which not only allow the random motion of the electrons and electrolytes, but also accelerate the oxygen molecules evolution/transportation [35, 51]; (2) the  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$  was directly grown on the





high conductivity Ni foam substrate along with an open 3D network, which affords a stable structure to promote the durability of materials and results in low charge transfer resistance in OER electrolysis process; and (3) the synergistic effect between the optimized ratio of Co and Ni is believed to be an another factor for the efficient OER catalytic performance of  $(Co_{0.5} Ni_{0.5})_3(PO_4)_2/Ni$ .

## Conclusions

In summary, we have demonstrated a facile hydrothermal method to synthesize a  $(Co_{0.5}Ni_{0.5})_3$   $(PO_4)_2/Ni$ . This method is low cost and effective. The  $(Co_{0.5}Ni_{0.5})_3(PO_4)_2/Ni$  displays highly catalytic activity toward OER in a basic electrolyte with a low overpotential of 273 mV to drive an anodic current of 10 mA cm<sup>-2</sup> and a Tafel slope of 59.3 mV dec<sup>-1</sup> in 1.0 mol L<sup>-1</sup> KOH. More importantly, this novel electrocatalyst even reveals higher catalytic activity in 30 wt% KOH, demonstrating considerable potential for the commercial application. It is believed that this work not only offers us a cost-effectively, active and stable OER electrocatalyst, but also opens up a new

avenue to prepare metal phosphates and the other electrocatalysts, which have excellent activity toward water electrolysis and/or electrochemical devices.

## Acknowledgements

The authors wish to acknowledge financial support from the National Natural Science Foundation of China (21376105, 21576113).

**Electronic supplementary material:** The online version of this article (doi:10.1007/s10853-017-1631-3) contains supplementary material, which is available to authorized users.

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