

# Few-layer thick WS<sub>2</sub> nanosheets produced by intercalation/exfoliation route

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## ABSTRACT

Thin two-dimensional (2D) nanostructures of layered materials are of importance for both science and technology due to their unique properties and promising applications. Few-layer tungsten disulfide (WS<sub>2</sub>) presents some amazing optoelectronic behaviors that have attracted great interest. In this work,  $WS<sub>2</sub>$  nanosheets with few-layer ( $<10$  layers) thickness have been successfully produced by Li ions intercalation/exfoliation of bulk  $WS_2$  via a solvothermal technique. X-ray diffraction indicates that hexagonal crystal structure of bulk  $WS<sub>2</sub>$  remains the same after the intercalation/exfoliation treatment. Scanning electron microscopy and transmission electron microscopy characterizations reveal that the exfoliated products are single-crystalline thin nanosheets with high crystallinity. Atomic force microscopy reveals that 8-layer thick  $WS_2$ nanosheets can be fabricated by the present approach. Raman phonon modes of the prepared  $WS_2$  nanosheets present the redshifts due to their decreased thickness. The work provides a facile and affective route to high-quality  $WS_2$ thin 2D nanostructures that would have great potentials for optoelectronics.

## Introduction

Two-dimensional (2D) nanostructure of carbon, i.e., graphene, has inspired great interesting in the last decade because of its unique structure and physical properties [\[1–4](#page-4-0)]. Meanwhile, as the analogues of graphene, some atomically thick 2D nanostructures of layered inorganic compounds, such as transition metal dichalcogenides (TMDs, including  $M_0S_2$ , WS<sub>2</sub>,  $WSe<sub>2</sub>$ , TiS<sub>2</sub>, etc.), transition metal oxides (TMOs, including  $MoO<sub>3</sub>$ , WO<sub>3</sub>, etc.), and hexagonal boron nitride (h-BN), have drawn considerable attention

from both fields of sciences and technologies [\[1](#page-4-0), [5](#page-4-0)[–8](#page-5-0)]. These 2D nanostructures present unique physical and chemical properties, and some applications in optoelectronics and photonics have already been demonstrated [[9,](#page-5-0) [10\]](#page-5-0).

Among those materials, it is known that hexagonal  $2H-WS<sub>2</sub>$  has typical layered structure in which each layer consists of a plane of tungsten atoms between two close-packed planes of sulfur atoms [\[11](#page-5-0)]. Bulk  $WS<sub>2</sub>$  is a narrow band semiconductor with an indirect band gap of 1.35 eV that limits its applications in optoelectronics, while a direct band gap can be

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<span id="page-1-0"></span>achieved in 2D monolayer  $WS_2$  [[12–14\]](#page-5-0), which makes great change in its physical properties and enables it to possess a series of potential applications [[15](#page-5-0), [16\]](#page-5-0). Nowadays, some reports have shown that  $2D$   $WS_2$ can be obtained by different methods including direct pyrolysis [[17\]](#page-5-0), atmospheric pressure chemical vapor deposition [[18\]](#page-5-0), thermal reduction sulfurization [[19\]](#page-5-0), chemical exfoliation [[20\]](#page-5-0), modified <sup>L</sup>-cysteine-assisted solution-phase method [\[21](#page-5-0)], and so on.

For layer-structured compounds, it is reasonable to expect that the exfoliation approaches are simple and effective to realize their 2D nanostructures. In particular, Li-ion-assisted wet chemical intercalation/ exfoliation method has been successfully used to fabricate a series of layered materials. Zhong et al. developed a hydrothermal Li intercalation and exfoliation route using ethylene glycol as both reductant and solvent to obtain ultrathin  $Bi<sub>2</sub>Te<sub>3</sub>$  nanosheets from  $Bi<sub>2</sub>Te<sub>3</sub>$  bulk [[22\]](#page-5-0). Similarly, the ultrathin  $MoS<sub>2</sub>$ nanosheets with good quality and high yield were produced [[15\]](#page-5-0). Xie et al. reported that freestanding five-atom-thick  $Bi<sub>2</sub>Se<sub>3</sub>$  single layers were first synthesized via a scalable intercalation/exfoliation strategy  $[23]$  $[23]$ . Using *n*-butyllithium as intercalation precursor, Rao's group synthesized  $WS_2$  layers by a 72-h intercalation treatment in hexane solvent under  $N<sub>2</sub>$  atmosphere and ultrasonication [[6\]](#page-4-0).

In this work, we report the fabrication of two-dimensional  $WS_2$  thin nanosheets by Li-ion intercalation/exfoliation route [[23](#page-5-0)]. Li ions were intercalated in bulk  $WS_2$  through a solvothermal technique, and the  $2DWS_2$  nanosheets were finally obtained by the exfoliation process using sonication treatment. The



Figure 1 Schematic illustration of the formation of  $WS_2$ nanosheets.

phase, morphology, microstructure, and Raman scattering of the obtained products were characterized. The results show that the as-prepared  $WS_2$ nanosheets are well-crystallized single crystals and their thickness can be down to 8-layer  $WS_2$ . Compared with the previous reports on liquid or chemical exfoliation of WS<sub>2</sub> [[6,](#page-4-0) [20\]](#page-5-0), the present method is low cost and facile because no dangerous n-butyllithium or supercritical  $CO<sub>2</sub>$  was involved. The work provides an effective route to high-quality  $WS_2$  thin layers with promising applications in optoelectronics.

## Experimental procedure

#### Materials preparation

The starting materials, commercial available  $WS_2$ (99 wt%) and  $Li<sub>2</sub>CO<sub>3</sub>$  (99 wt%) powders, were directly used without further purification. The synthesis of the WS2 nanosheets involves two steps, i.e., Li intercalation and exfoliation procedures, which is similar with the previous report  $[23]$ . In a typical run, 352 mg  $WS_2$ powders and 142 mg  $Li<sub>2</sub>CO<sub>3</sub>$  were mixed and put into a Teflon-lined stainless-steel autoclave with the capacity of 50 mL. Then 40 mL benzyl alcohol was added into the autoclave and vigorously stirred for 60 min. After that, the autoclave was sealed and maintained at 200  $^{\circ}C$ for 48 h. Cooled to room temperature naturally, the precipitates, i.e., Li-intercalated  $WS_{2}$ , were collected at the bottom of the autoclave and repeatedly rinsed by deionized (DI) water, acetone, and alcohol to remove soluble inorganic and organic impurities. Then the Liintercalated WS<sub>2</sub> were dried at 50 °C for 6 h in vacuum. 30 mg Li-intercalated  $WS_2$ , 30 mL distilled water, and 30 mL dimethyl formamide (DMF) were added into a jar with a capacity of 125 mL, which was sealed and sonicated at for 12 h. The resultant dispersions were centrifuged at 600 rpm for 30 min and then the supernatant was collected. After centrifuging the supernatant at 6000 rpm for 10 min, the collected products were orderly washed with 3 % HCl and DI water to remove the residual  $Li<sub>2</sub>CO<sub>3</sub>$ . Finally, the as-prepared products were collected by repeatedly washing with ethanol and drying at 50  $\degree$ C for 6 h in vacuum.

#### Characterization of the materials

The final products were characterized by X-ray diffraction (XRD), scanning electron microscopy





Figure 2 SEM images of WS<sub>2</sub> bulk (a) and the exfoliated WS<sub>2</sub> nanosheets (b).



**Figure 3** The XRD patterns of the exfoliated  $WS_2$  nanosheets and commercial  $WS_2$  powders.

(SEM), transmission electron microscopy (TEM), high-resolution transmission electron microscopy (HRTEM), atomic force microscopy (AFM), and Raman scattering.

#### Results and discussion

Figure [1](#page-1-0) schematically depicts the formation processes of two-dimensional tungsten disulfide  $(WS_2)$ nanosheets produced by Li intercalation/exfoliation of layered bulk  $WS_2$  via the solvothermal and sonication treatment, which is similar with the cases of  $Bi_2Te_3$  [\[21](#page-5-0)],  $Bi_2Se_3$  [\[23](#page-5-0)], and  $MoS_2$  [[15\]](#page-5-0). Briefly, because the bulk  $WS_2$  layers are connected by weak van der Waals interaction, lithium atoms of  $Li<sub>2</sub>CO<sub>3</sub>$ dissolved in benzyl alcohol can be easily intercalated into the interlayers of bulk  $WS_2$  under appropriate solvothermal condition. When the Li-intercalated  $WS_2$  were sonicated in the mixed solution of distilled water and DMF for 12 h, the lithium atoms between layers of bulk  $WS_2$  were extracted out, leading to the production of the exfoliated  $WS_2$  layers. Finally, twodimensional WS<sub>2</sub> nanosheets powders were collected from the suspension by centrifuging.

As shown in Fig. 2a, b, the morphologies of bulk  $WS_2$  and the exfoliated  $WS_2$  layers are characterized by SEM. It can be obviously observed that the bulk  $WS_2$ precursor consists of many thick plates with native layered structure. On the contrary, the exfoliated products are thin two-dimensional  $WS_2$  nanosheets with flat and smooth surfaces. There is no distinct difference in the lateral sizes of the synthesized  $WS_2$ nanosheets and the starting bulk  $WS_2$  plates.

Figure 3 shows the XRD patterns of the exfoliated  $WS_2$  products and commercial  $WS_2$  powders. The main reflections of the  $WS_2$  nanosheets can be well indexed to hexagonal  $2H-WS<sub>2</sub>$  with lattice constants of  $a = 3.1532$  Å and  $c = 12.323$  Å (ICDD PDF Card No. 08-0237) except for an unknown peak marked by square. The sharp profiles of the reflections suggest good crystallization of the sample. Compared with the XRD pattern of the commercial  $WS_2$  powders, the exfoliated  $WS_2$  products show the same crystal structure except for the size-related broadening of the reflections.

Figure [4a](#page-3-0), b shows the low-magnification TEM images of exfoliated  $WS_2$  nanosheets. The individual WS<sub>2</sub> nanosheets are randomly stacked and look transparent under the irradiation of electron beam, which implies that the obtained  $WS_2$  nanosheets are very thin. Microstructures of the  $WS_2$  nanosheets are

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further checked by HRTEM. Representative HRTEM crystal lattice images taken from the as-prepared samples are shown in Fig. 4c. Clear lattice fringes with a spacing of 0.270 nm shown in the HRTEM image can be indexed to the  ${100}$  plane of  $2H-WS<sub>2</sub>$ , revealing the single-crystal nature of the prepared nanosheets. The well-arranged lattice with hexagonal symmetry indicates that the top surface of the nanosheet is {001} plane. The SAED pattern shown in Fig. 4d can be indexed to  $2H-WS<sub>2</sub>$ , further demonstrating the nanosheet is well-crystallized single crystal. The zone axis of the SAED pattern is along [001], which reveals the [001] preferential orientation of the nanosheet, too.







<span id="page-4-0"></span>

Figure 6 Raman spectra for  $WS_2$  nanosheets (*a*) and  $WS_2$  bulk (b).

As shown in Fig. [5a](#page-3-0), a typical AFM topography image is shown to confirm the thickness of the asprepared nanosheets. The nanosheet has smooth and flat surface with a uniform thickness across the lateral dimension. Figure [5b](#page-3-0) shows the height profile of the nanosheet. It can be seen that the  $WS_2$  nanosheet has a thicknesses of about 6 nm, which corresponds to about 8 times the thickness of  $WS_2$  monolayer [\[24](#page-5-0)].

Room-temperature Raman spectrum of the prepared  $WS_2$  nanosheets (Fig. 6) shows two peaks at 349.5 and 416.5  $\rm cm^{-1}$ , which can be assigned to  $\rm E_{2g}^{1}$ and  $A_{1g}$  phonon modes of 2H-WS<sub>2</sub> [\[11](#page-5-0)], respectively. Compared with the corresponding phonon modes of bulk  $WS_2$  shown in Fig. 6, there is slightly redshifts in those of nanosheets. It is known that  $A_{1g}$  and  $E_{2g}^1$ modes are due to the vibrations perpendicular and parallel to the S–W–S layer, respectively [\[11](#page-5-0), [25](#page-5-0)].  $A_{1g}$ mode is sensitive to the thickness of  $WS_2$  nanosheets, and the observed redshifts are consistent with the previous reports on the few-layer WS<sub>2</sub>  $[11, 24, 25]$  $[11, 24, 25]$  $[11, 24, 25]$  $[11, 24, 25]$  $[11, 24, 25]$  $[11, 24, 25]$  $[11, 24, 25]$ which indicates the weakened interaction between  $WS<sub>2</sub>$  layers with decreasing the thickness. On the other hand,  $E_{2g}^{1}$  mode has less dependence on the thickness and is thought to be influenced by the longrange Coulomb interactions [[25\]](#page-5-0). The frequency difference of the two modes of the prepared  $WS_2$ nanosheets is smaller than that of bulk  $WS_2$ , which also demonstrates their decreased thicknesses [\[11](#page-5-0)].

## **Conclusions**

In summary, thin two-dimensional tungsten disulfide (WS<sub>2</sub>) nanosheets have been successfully produced by Li intercalation/exfoliation of bulk  $WS_2$  via solvothermal technique. The as-prepared  $WS_2$  nanosheets are high-quality single crystals with hexagonal structure and have thin thicknesses of few-layer  $WS_2$ . The work provides a facile and efficient route to large-scale preparation of two-dimensional  $WS_2$  nanosheets with potential applications in optoelectronic nanodevices.

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## Compliance with ethical standards

Conflict of interest The authors declare that they have no conflict of interest.

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