

# One-step synthesis of flower-like WO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub> heterojunction with enhanced visible light photocatalytic activity

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**Abstract** WO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub> heterojunctions with flower-like structure were prepared by a facile hydrothermal process without any surfactants or templates. In the heterojunctions, WO<sub>3</sub> nanoparticles were incorporated into three-dimensional flower-like hierarchical Bi<sub>2</sub>WO<sub>6</sub>. The WO<sub>3</sub>/ Bi<sub>2</sub>WO<sub>6</sub> samples showed much higher photocatalytic activity than pure Bi<sub>2</sub>WO<sub>6</sub> did for rhodamine B degradation under visible light irradiation. This could be ascribed to the formation of *n*–*n* type heterojunction, which resulted in the high transfer rate of the photo-generated electron– holes between WO<sub>3</sub> and Bi<sub>2</sub>WO<sub>6</sub> and confirmed by photoluminescence and photocurrent measurements.

# Introduction

Semiconductor photocatalysts have received much attention due to their abilities of solar energy conversion and environmental purification [1–3]. The semiconductor TiO<sub>2</sub> is considered as one of the best photocatalysts [4, 5]. However, the light response range of TiO<sub>2</sub> is limited to ultraviolet because of the wide band gap (3.2 eV) [6], and the ultraviolet accounts for only about 4 % of entire sunlight [7–9]. So it is significant to prepare new photocatalysts, which can response in large range of sunlight including the visible light which accounts for about 46 % of sunlight.

 $Bi_2WO_6$ , an *n*-type semiconductor [10], is exactly a visible-light-driven photocatalyst with a relatively narrow

band gap of 2.7 eV [11] ( $E_{CB} = 0.24$  eV vs NHE), which has attracted a great deal of attention [12-14]. However, as a single-phase photocatalyst, some drawbacks restrict its further practical application, such as high recombination of the photo-generated electron-hole pairs and low photo quantum efficiency. Fortunately, as preparing a heterojunction is a well-known way to suppress the photo-carriers recombination rate and enhance the photocatalytic property, many materials have been coupled with Bi<sub>2</sub>WO<sub>6</sub> to form the heterojunction, such as BiOBr [15], Bi<sub>2</sub>O<sub>2</sub>CO<sub>3</sub> [16], CdS [17], Co<sub>3</sub>O<sub>4</sub> [18], WO<sub>3</sub> [19, 20], or ZnFe<sub>2</sub>O<sub>4</sub> [21]. Among these,  $WO_3$  is an inexpensive and promising material reported to have good photocatalytic performance. As an *n*-type semiconductor with band gap of 2.8 eV  $(E_{CB} = 0.74 \text{ eV vs NHE})$  [22], WO<sub>3</sub> can match well with  $Bi_2WO_6$  to form an *n*-*n* type heterojunction.

Morphology-controlled synthesis of  $Bi_2WO_6$  has been also considered as an efficient way to enhance photocatalytic activities under visible light. Three-dimensional flower-like  $Bi_2WO_6$  photocatalysts have large surface areas and plenty of meso-pores which can effectively harvest visible light due to multiple scattering [23], In addition, the flower-like  $Bi_2WO_6$  structure have been found to increase the active sites and improve the photoenergy conversion efficiency [24]. He et al. [20] have prepared the  $WO_3$ (core)/ $Bi_2WO_6$  (shell) photocatalyst through a hydrothermal reaction and heat treatment. Unfortunately, the preparation process was complex and the advantage of flower-like superstructure  $Bi_2WO_6$  could not survive.

Therefore, for the first time, we successfully prepared the flower-like  $WO_3/Bi_2WO_6$  heterojunction by one-step hydrothermal method without any surfactants or templates. The experimental results show that the flower-like  $WO_3/Bi_2WO_6$  heterojunction has much greater activities for photocatalytic degradation of rhodamine B under visible

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light irradiation than pure  $Bi_2WO_6$  does. Moreover, a possible mechanism of  $WO_3/Bi_2WO_6$  on the degradation of RhB is discussed and confirmed by photoluminescence spectra and photocurrent measurements.

# **Experimental**

# Preparation of WO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub>

All chemicals are reagent grade and used without further purification. A typical procedure (molar ratio of WO<sub>3</sub>/  $Bi_2WO_6$  samples designed as 0.3:1) is as follows: 2 mmol Bi(NO<sub>3</sub>)<sub>3</sub>·5H<sub>2</sub>O was firstly dissolved in a 30 mL 0.4 mol/L HNO<sub>3</sub> solution, and then the solution was stirred for 10 min at 40 °C. 30 mL aqueous solution containing 1.3 mmol Na<sub>2</sub>WO<sub>4</sub>·2H<sub>2</sub>O was added into the above solution drop by drop and then stirred for 24 h at 40 °C. The obtained white suspension was transferred into a 100-mL Teflon-lined autoclave. The autoclave was heated at 160 °C for 20 h and then cooled to room temperature naturally. The light yellow precipitate was collected and washed with distilled water and ethanol for several times. Then the light yellow precipitate was dried at 60 °C in the air for 10 h. According to this method, WO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub> samples with different molar ratios of 0.15:1, 0.3:1, and 0.6:1 were also obtained and denoted as 5, 10, and 20 %  $WO_3/Bi_2WO_6$  (mass ratio), respectively. The pure  $Bi_2WO_6$ and WO<sub>3</sub> were also synthesized by a similar procedure.

# Characterization

#### The structure and morphology characterization

The crystal structure of the samples were investigated by X-ray diffraction (XRD, Model No: D/max2200pc, Japan) with Cu  $K_{\alpha}$  radiation ( $\lambda = 1.54$  Å) over the range of  $10^{\circ} \le 2\theta \le 70^{\circ}$ . The morphology and microstructure of the samples were obtained by field emission scanning electron microscope (FE-SEM, Model No: Hitachi S-4800) and transmission electron microscope (TEM, Model No: FEI Tecnai G2 F20 S-TWIN).

# N<sub>2</sub> physisorption analysis

According to the Brunauer–Emmett–Teller analysis (BET, ASAP 2460, Micromeritics, USA), the specific surface areas were determined by nitrogen adsorption–desorption isotherms at 77 K. Prior to measurements, all Bi<sub>2</sub>WO<sub>6</sub>-based powders were degassed at 473 K for 24 h under a vacuum to ensure a clean, dry surface, free of any loosely bound adsorbed species. The pore size distributions of all samples were calculated from desorption branches of the

corresponding nitrogen isotherm by the Barrett-Joyner-Halenda (BJH) method.

# Photoluminescence and photoelectrochemical measurements

The photoluminescence measurements were carried out on a Hitachi F-4600 fluorescence spectrophotometer. The photocurrent was conducted by an electrochemical workstation (Model No: CH1660D instruments, shanghai). A threeelectrode system was employed using the Ag/AgCl (3 M KCl) electrode as a reference electrode, a Pt wire as the counter electrode and an indium tin oxide (ITO) conducting glass coated with the WO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub> film as the working electrodes. The WO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub> film was deposited on ITO glass by the dip-coating method. Briefly, a piece of ITO glass was washed with distilled water and ethanol in an ultrasonic bath for 30 min, and dried at room temperature. In the meantime, WO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub> (0.02 g) was dispersed in 1 mL distilled water containing 0.5 mL Nafion and ultrasonically treated for 10 min. The as-resulted WO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub> suspension was deposited onto the surface of ITO using a microsyringe and dried at room temperature. 1 M Na<sub>2</sub>SO<sub>4</sub> was used as the electrolyte and a 300 W Xe lamp (wavelength range, distance between sample and lamp) was used as light source. All measurements were carried out at room temperature.

# Photocatalytic activity

The photocatalytic activities of the samples were evaluated via the degradation of RhB in an aqueous solution under visible light irradiation. A xenon long-arc lamp GXU 500 with a UV 420-nm cutoff filter was used as the light source to simulate visible light irradiation. In every experiment (seven quartz tubes), 5-mg sample was added into 5 mL RhB aqueous solution with a concentration of 10 mg/L in a quartz tube. Before illumination, the suspensions were sonicated for 30 min and magnetically stirred in the dark for 30 min to reach the adsorption-desorption equilibrium between the photocatalyst and RhB solution. Then the mixture was exposed to the stimulated visible light under magnetic stirring. At given time intervals, 5 mL suspension was taken and centrifuged to remove the photocatalyst. Then the RhB concentration was detected by recording the absorbance at the characteristic band of 553 nm using a Shimadzu UV-2550 UV-vis spectrophotometer.

# **Results and discussion**

# **Crystal structures**

Figure 1 shows XRD patterns of the samples. The pure  $Bi_2WO_6$  is identified as the orthorhombic  $Bi_2WO_6$  (JCPDS



Fig. 1 XRD patterns of WO<sub>3</sub>,  $Bi_2WO_6$ , and  $WO_3/Bi_2WO_6$  samples. The *inset* is partial magnification of XRD pattern of 20 % WO<sub>3</sub>/ $Bi_2WO_6$  sample

NO. 39-0256) without any impurity. The pure WO<sub>3</sub> is identified as monoclinic WO<sub>3</sub> (JCPDS NO. 72-1465) without any impurity. For the WO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub> samples, XRD

patterns consist of  $Bi_2WO_6$  and  $WO_3$  diffraction peaks. As shown in the inserted figure for the 20 %  $WO_3/Bi_2WO_6$ sample, the diffraction peaks of  $WO_3$  can be observed. The strongest peak of  $WO_3$  is attributed to (002) plane, which is in good agreement with the following HRTEM analyses. No impurity peak is also found in the  $WO_3/Bi_2WO_6$ heterojunction. This suggests that the photocatalyst is only composed of orthorhombic  $Bi_2WO_6$  and monoclinic  $WO_3$ .

Figure 1 shows XRD patterns of the samples. The pure  $Bi_2WO_6$  is identified as orthorhombic  $Bi_2WO_6$  (JCPDS NO. 39-0256) without any impurity. The pure  $WO_3$  is identified as monoclinic  $WO_3$  (JCPDS NO. 72-1465) without any impurity. For the  $WO_3/Bi_2WO_6$  samples, XRD patterns consist of  $Bi_2WO_6$  and  $WO_3$  diffraction peaks. As shown in the inserted figure for the 20 %  $WO_3/Bi_2WO_6$  sample, the diffraction peaks of  $WO_3$  can be observed. The strongest peak of  $WO_3$  is attributed to (002) plane, which is in good agreement with the following HRTEM analyses. No impurity peak is also found in the  $WO_3/Bi_2WO_6$  heterojunction. This suggests that the photocatalyst is only composed of orthorhombic  $Bi_2WO_6$  and monoclinic  $WO_3$ .



Fig. 2 SEM images of a overall morphology and b a detailed view on an individual sphere, c TEM and d HRTEM images of the 5 % WO<sub>3</sub>/ $Bi_2WO_6$ 

#### Morphology characterization

The morphology and microstructure of the 5 % WO<sub>3</sub>/Bi<sub>2-</sub> WO<sub>6</sub> sample were investigated by SEM and TEM. Figure 2a shows that the 5 % WO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub> sample consists of spherical particles with diameter ranging from 3 to 5 µm. All the spherical particles are well dispersed. Figure 2b is image with high magnification of an individual 5 % WO<sub>3</sub>/ Bi<sub>2</sub>WO<sub>6</sub> spherical particle which reveals more detailed structural characteristics of the heterojunction photocatalysts. We can see that the spherical particle is flower-like and consists of two-dimensional thin nanoplates with many different pores, which can serve as transport paths for small molecules and then improve the photocatalytic property. More in-depth information of the sample can be observed by TEM (Fig. 2c) and HRTEM micrographs (Fig. 2d). There are many nanoparticles with the diameter about 10 nm attached to the nanoplate. By carefully measuring the lattice parameters with Digital Micrograph and comparing with the data in JCPDS, three different kinds of lattice fringes with spacing of 0.273, 0.315, and 0.33 nm are obtained. It can be sure that the lattice fringes with spacing of 0.273 and 0.315 nm belong to the (200) and (113) crystallographic plane of orthorhombic Bi<sub>2</sub>WO<sub>6</sub> (JCPDS NO. 39-0256) and the lattice fringe with spacing of 0.33 nm belongs to (120) plane of monoclinic  $WO_3$ (JCPDS NO. 72-1465). So the nanoplates could be Bi<sub>2</sub>WO<sub>6</sub>

and the nanoparticles on the surface of the nanoplates could be  $WO_3$ .

Figure 3 is SEM micrograph of all the samples: (a) pure  $Bi_2WO_6$ ; (b) 5 %  $WO_3/Bi_2WO_6$ ; (c) 10 %  $WO_3/Bi_2WO_6$ ; and (d) 20 %  $WO_3/Bi_2WO_6$ . All the samples exhibit flower-like structure which consists of nanoplates. From the inserted figures, we can see that with increasing  $WO_3$  doping content, the nanoplates become denser and thinner, at the same time, the pores among the nanoplates become smaller and smaller. The thickness of the nanoplates is about 27, 20, 17, and 12 nm for the  $WO_3$  content of 0, 5, 10, and 20 %, respectively.

# N<sub>2</sub> physisorption analysis

The analysis of surface area and pore size are carried out to establish correlations, which derived from the nitrogen molecules that are physisorbed by these composite photocatalysts [25]. The nitrogen adsorption/desorption isotherms and the corresponding BJH pore size distribution curves of the pure Bi<sub>2</sub>WO<sub>6</sub> and 5 % WO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub> composite are shown in Fig. 4. It can be seen that the pure Bi<sub>2</sub>WO<sub>6</sub> and 5 % WO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub> heterojunction exhibit the typical type-III curves for mesoporous materials according to IUPAC classification (Fig. 4a). The specific surface areas are calculated using the BET method and the pore size distribution of the each sample is calculated from the



Fig. 3 SEM micrographs of the samples: a pure Bi<sub>2</sub>WO<sub>6</sub>; b 5 % WO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub>; c 10 % WO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub>; and d 20 % WO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub>





**Table 1** Adsorption parametersand the pseudo-first-orderkinetic (k) value of  $Bi_2WO_6$  andall  $WO_3/Bi_2WO_6$  samples

Sample	BET surface area (m <sup>2</sup> /g)	Pore size (nm)	$K (\min^{-1})$
Bi <sub>2</sub> WO <sub>6</sub>	17.41	13.92	0.006
5 % WO3/Bi2WO6	26.01	9.73	0.012
10 % WO3/Bi2WO6	30.18	8.59	0.030
20 % WO <sub>3</sub> /Bi <sub>2</sub> WO <sub>6</sub>	45.75	6.42	0.018

desorption branch of the isotherms using the BJH algorithm (Fig. 4b). As seen in Table 1, with increasing the content of  $WO_3$ , the BET specific surface area becomes larger and larger and the BJH pore size becomes smaller and smaller.

# Photoluminescence spectra analysis

PL spectrum is useful to reveal the migration, transfer, and recombination process of the photo-generated electronhole pairs in the semiconductor. In order to demonstrate the separated efficiency of the electron-hole pairs, the room temperature photoluminescence (PL) spectra were examined for Bi2WO6 and all WO3/Bi2WO6 samples with an excitation wavelength of 300 nm. As shown in Fig. 5, all samples show an apparent characteristic emission peak from 400 to 500 nm. The strongest emitting peak at 455 nm can be attributed to the radiative recombination process of self-trapped excitations [26]. Compared with pure Bi<sub>2</sub>WO<sub>6</sub>, all WO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub> samples have a weaker PL intensity and the 5 % WO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub> sample shows the lowest emission peak. A weaker intensity of the peak represents a lower recombination probability of free charges. So couple WO<sub>3</sub> with Bi<sub>2</sub>WO<sub>6</sub> has improved the separated efficiency of the electron-hole pairs and the 5 %  $WO_3/Bi_2WO_6$  sample has the best performance.

# Transient photocurrent analysis

We also have measured transient photocurrent to gain an insight into the high separation efficiency of the photogenerated electrons and holes in the  $Bi_2WO_6$  and  $WO_3/Bi_2WO_6$  samples. Figure 6 displays the photocurrent-time



Fig. 5 Photoluminescence (PL) spectra of  $Bi_2WO_6$  and  $WO_{3'}Bi_2WO_6$  samples

(*I*–*t*) curves of the four samples with typical on–off cycles of intermittent visible light irradiation. As shown in Fig. 6, the photocurrent boosts rapidly once the light is turned on and returns quickly to its dark current state when the light is turned off. The initial current is due to the separation of the electron–hole pairs at the semiconductor/electrolyte interface: holes are trapped by the reduced species in the electrolyte, while electrons transport to the back contact substrate [27]. As WO<sub>3</sub> concentration increases, the photocurrent increases until WO<sub>3</sub> doping content is 5 % and then decreases. This dependence corresponds well to the above PL results.



Fig. 6 Photocurrent transient responses for  ${\rm Bi}_2 WO_6$  and  $WO_3/$   ${\rm Bi}_2 WO_6$  samples

# Photocatalytic activity

Figure 7a shows the photocatalytic degradation efficiency of RhB in the presence of different samples under visible light irradiation. The blank test confirms that the photocatalytic degradation of RhB can be ignored without catalyst. All heterojunctions show higher photocatalytic activity than pure Bi<sub>2</sub>WO<sub>6</sub> dose and the 5 % WO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub> shows the highest photocatalytic activity. It can be observed that only 48.9 and 32.1 % RhB is photodegraded by pure Bi<sub>2</sub>WO<sub>6</sub> and WO<sub>3</sub> under visible light in 100 min, respectively. 5 % WO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub> can result in 98.5 % degradation rate in the same condition. However, to further increase the amount of WO<sub>3</sub>, photocatalytic activity of RhB is decreased to 93 % in the same condition. According to the above SEM, BET specific surface areas and BJH pore size distribution results, we think that the suitable amount of WO3 can uniformly distribute on the surface of flower-like Bi<sub>2</sub>WO<sub>6</sub> and make the nanoplates become density and thin which is helpful to improve the photocatalytic activity. However, the excess WO<sub>3</sub> makes

Fig. 7 a Photocatalytic degradation curve and b firstorder plots for the photodegradation of RhB in the presence of WO<sub>3</sub>,  $Bi_2WO_6$ , and  $WO_3/Bi_2WO_6$  samples under visible light irradiation flower-like  $Bi_2WO_6$  overlap and agglomerate which prevent the transport paths for organic molecules and then lower photocatalytic property. The results indicate that the suitable content of  $WO_3$  plays an important role in the enhancement of photocatalytic performance.

For a better comprehension of the photocatalytic activity of catalysts, the reaction kinetic of the RhB degradation was investigated. The experimental data were fitted by the relevant equation as below:  $-\ln(C/C_0) = kt$ , where  $C_0$  and C are the concentrations of RhB at adsorption–desorption equilibrium and the reaction time t, respectively, and k is the apparent first-order rate constant. As shown in Fig. 7b, upon varying the WO<sub>3</sub> content within 5.0–20 %, the plots of the ( $C_0/C$ ) versus irradiation time (t) display a nearly straight line. All WO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub> heterojunctions show higher photodegraded efficiency than pure Bi<sub>2</sub>WO<sub>6</sub> and WO<sub>3</sub> do. The 5 % WO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub> exhibits the highest photodegraded efficiency and is about fivefold compared to that of the pure Bi<sub>2</sub>WO<sub>6</sub> sample (Table 1).

In order to demonstrate the photocatalytic activities of WO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub> samples with WO<sub>3</sub> mass ratio between 0 and 5 %, we do experiment about 3 %  $WO_3/Bi_2WO_6$  sample. As is shown in Fig. 8, all strong peaks can be ascribed to the orthorhombic Bi<sub>2</sub>WO<sub>6</sub> (JCPDS NO. 39-0256). No clear diffraction peak belonging to WO<sub>3</sub> is observed, due to the small amount of WO<sub>3</sub> and their highly dispersion in WO<sub>3</sub>/  $Bi_2WO_6$  sample (Fig. 8a). It can be observed that the flower-like superstructure cannot be destroyed and also consist of two-dimensional nanoplates (Fig. 8b). The specific surface areas and pore size distribution is 23.68 m<sup>2</sup>/g and 11.37 nm, respectively (Fig. 8c). The photocatalytic activity indicates that the 3 % WO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub> heterojunction exhibits lower photocatalytic activity (90.2 % RhB be photodegraded) than 5 % WO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub> does due to the small amount of WO<sub>3</sub> loading (Fig. 8d).

# **Circulation experiment**

(a) b) WO<sub>3</sub> 3.5 . 1.0 Bi2WO6 3.0 5% WO3/Bi2WO6 0.8 10% WO3/Bi2WO6 2.5 20% WO3/Bi2WO6 In (C<sub>o</sub>/C) C/C0 2.0 0.6 blank 1.5 WO3 0.4 Bi2WO6 1.0 5% WO3/Bi2WO6 0.2 10% WO3/Bi2WO6 0.5 20% WO3/Bi2WO6 0.0 0.0 -20 20 40 60 80 40 60 0 100 20 80 100 0 Irradition time (min) Irradiation time (min)

Figure 9a shows the variations of the UV-vis absorption spectra of RhB solutions in the presence of  $5 \% \text{ WO}_3$ /

**Fig. 8 a** XRD pattern, **b** SEM micrograph, **c** nitrogen adsorption/desorption isotherm curve and BJH pore size distribution curve, **d** photocatalytic degradation curve of 3 % WO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub> sample





 $Bi_2WO_6$  heterojunction under visible light irradiation ( $\lambda > 420$  nm), which shows a maximum absorption band at 553 nm. With the increase of irradiation time, a rapid decrease of RhB absorption band is observed and the color of the suspension gradually changes from pink to light green. In addition, the maximum absorption peak of RhB gradually shifts to blue region, which can be attributed to a successive de-ethylation from the aromatic rings and destruction of the conjugated structure process, according to the formation of the different de-ethylated rhodamine intermediates [28]. To evaluate the stability of 5 % WO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub>, the circulating runs

checked. As shown in Fig. 9b, after every 100 min of

photodegradation, the separated photocatalysts were washed with deionized water and dried to remove the

ions absorbing on its surface after every reaction. There

is no obvious catalyst deactivation after four cycling

runs and the flower-like structure do not change. The



## The role of WO<sub>3</sub>

About the influence of WO<sub>3</sub> concentration on the photocatalytic activity, two effects should be considered: on one hand, WO<sub>3</sub> concentration affects morphology of the sample (as shown in Fig. 3). As WO<sub>3</sub> concentration increases, the nanoplates become denser and thinner and the pores among the nanoplates become smaller and smaller. This may be helpful to increase the specific surface area and the active sites in porous microsphere. On the other hand, WO<sub>3</sub> concentration affects photo-carriers recombination rate of the sample (as shown in Figs. 5, 6). The addition of  $WO_3$ has suppressed photo-carriers recombination due to the formation of n-n type heterojunction, which is helpful to improve the sample photocatalytic activity. However, the suppression effect of heterojunction does not monotonically increase with the WO<sub>3</sub> concentration increasing. From Figs. 5 and 6, we can see that 5 % WO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub> exhibits the highest separated efficiency of the photo-carriers. In above two respects, the latter effect is expected to dominate. So the sample photocatalytic activities show the dependence on WO<sub>3</sub> concentration: as WO<sub>3</sub> concentration increases, the photocatalytic activities increase until WO<sub>3</sub> mass ratio is 5 % and then decreases.

# Photocatalytic mechanism

To better understand the photocatalytic mechanism of the n-n type WO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub> heterojunction, a possible mechanism on the degradation of RhB is shown in Fig. 10. The optical band gap energy of Bi<sub>2</sub>WO<sub>6</sub> and WO<sub>3</sub> is 2.7 eV  $(E_{CB} = 0.24 \text{ vs NHE})$  and 2.8 eV  $(E_{CB} = 0.74 \text{ eV vs})$ NHE), respectively. The band gaps of the two semiconductors match well with each other. Under visible light irradiation, both the  $Bi_2WO_6$  and  $WO_3$  are excited by absorbing photons, and then electron-hole pairs are produced. The WO<sub>3</sub> acts as electron-accepting semiconductor. Photo-generated electrons transfer from the conduction band (CB) of  $Bi_2WO_6$  to that of WO<sub>3</sub>. Simultaneously, holes shift from the valence band (VB) of WO<sub>3</sub> to that of Bi<sub>2</sub>WO<sub>6</sub>. The effectively separation of photo-generated electrons and holes can be enhanced, which result in higher photocatalytic performance.

# Conclusions

In summary, the flower-like WO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub> heterojunction with different WO<sub>3</sub> concentrations was prepared via the one-step hydrothermal method. The flower-like superstructure is composed of Bi<sub>2</sub>WO<sub>6</sub> nanoplates and WO<sub>3</sub> nanoparticles. The WO<sub>3</sub> concentration has a great influence on the morphology and photocatalytic activity of WO<sub>3</sub>/ Bi<sub>2</sub>WO<sub>6</sub> heterojunction. As WO<sub>3</sub> concentration increases, the nanoplates become denser and thinner, and the photocatalytic activities increase until mass ratio of WO<sub>3</sub> is up to 5 % and then decreases. The result indicates that n-n type heterojunction has better photocatalytic activity than pure  $Bi_2WO_6$  does, because electrons can be injected from the CB of Bi<sub>2</sub>WO<sub>6</sub> to that of WO<sub>3</sub> under visible irradiation. Thus, the photo-generated electrons and holes are efficiently separated at the intimate interface of heterojunction in time. Therefore, the flower-like WO<sub>3</sub>/Bi<sub>2</sub>WO<sub>6</sub> heterojunction has good potential for application to organic pollutants purification.

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