In-situ synthesis of CdS/g-C₃N₄ hybrid nanocomposites with enhanced visible photocatalytic activity

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Abstract A novel and simple in situ synthetic strategy was used to fabricate CdS/g-C₃N₄ hybrid nanocomposite catalysts with visible-light-driven photocatalytic activity from cadmium-containing carbon nitride compounds. X-ray diffraction measurements, high-resolution transmission electron microscopy images, and Fourier transform infrared spectra showed heterojunctions with a close interface between the g-C₃N₄ and the CdS nanoparticles and nanorods in the composite. Ultraviolet visible diffuse reflectance spectra exhibited a red shift that further presented the CdS in the polymer g-C₃N₄ skeleton, which allowed the efficient utilization of the solar spectrum for creating photogenerated electrons and holes. The photoluminescence spectra of the nanocomposites suggested charge transfer from g-C₃N₄ to CdS. The photocurrent intensity of hybrid nanocomposites was 2.3 times than that of pure g-C₃N₄ sample, and photocatalytic activity for the photodegradation of methyl orange was 2.5 times, and hydrogen evolution reaction was 2.8 times. Enhanced photocatalytic activity and photocurrent for the CdS/g-C₃N₄ hybrid nanocomposites were achieved.

Introduction

Serious environmental pollution and global energy shortage have recently led to increasing interest in studying the application of photocatalyst technology on water pollution treatment and hydrogen evolution [1–3]. In 1972, Fujishima [4] reported TiO_2 as the electrode for water in photolysis, and several researchers have exerted great efforts in searching for suitable semiconductor materials to apply as photocatalysts since then [5, 6]. A typical semiconductor material CdS has a wide range of application in photocatalysis and electronic devices [7]. CdS has shown promise in enhancing photodegradation because it can efficiently separate photogenerated charge carriers [8]. However, CdS is unstable under photo-corrosion because the photogenerated holes self-oxidize during the photocatalytic process.

Wang [9] has recently produced a polymeric carbon nitride material (C_3N_4) , which has been used as a metalfree photocatalyst for water splitting produce hydrogen application. Carbon nitride $(g-C_3N_4)$, which is essentially composed of covalent bonds, is the most promising candidate to complement carbon in phocatalytic applications. Carbon nitride posses a high thermal and chemical stability and has attracted attention because of its outstanding mechanical, electrical, thermal, and optical properties. Nevertheless, the low efficiency caused by the high recombination rate of photogenerated charges limits the practical application of the current material. Several methods, including fabrication porous structure doped with metal or nonmetal elements, have been developed to enhance the phtocatalytic capability of carbon nitride [10-20]. Furthermore, Bai et al. fabricated a $g-C_3N_4$ from nanoplates to nanorods, which enhanced the photocatalytic activity and photocurrent intensity about 1.5 and 2.0 times those of g-C₃N₄ nanoplates, respectively [21]. Combining C₃N₄ with other semiconductors to form heterostructures particularly provides a feasible route to inhibit the recombination of photogenerated electron-hole pairs. Efforts are exerted to design organic and inorganic hybrid structures that exhibit improved selectivity and efficiency toward light energy conversion [22].

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The energy levels of C₃N₄ and CdS are compared and their well-matched overlapping band-structures are suitable to construct heterostructures that can effectively separate and transfer photogenerated charges. Therefore, combining C₃N₄ and CdS overcomes the disadvantages that exist in C₃N₄ and CdS simultaneously [23]. Feng et al. recently prepared CdS/g-C₃N₄ composites through chemical impregnation and subsequent calcination of separately prepared CdS particles and $g-C_3N_4$ [24]. Cao et al. reported an in situ growth CdS/g-C₃N₄ composites using a solvothermal method, the combination of as-prepared g-C₃N₄ and Cd resource in dimethyl sulfoxide as the S resource solution [25]. Fu et al. prepared CdS/g-C₃N₄ composites through an in situ precipitation-deposition method; the composites of the Cd and S resources were first prepared then combined with as-prepared $g-C_3N_4$ [26]. Jiang et al. fabricated g-C₃N₄-CdS composite through a precipitation method; the as-prepared g-C₃N₄ was first combined with the Cd resource and then combined with the S resource [22]. Lu et al. have also reported that the as-prepared ultrathin g-C₃N₄ nanosheets-CdS nanocomposite exhibited significantly enhanced photocatalytic activity for methyl orange (MO) degradation under visible light irradiation [27]. Significant advances have recently been made to the design and concern loaded type CdS-C₃N₄ composite particle. However, the phtocatalytic activity of the loaded type CdS-C₃N₄ needs further improvement because of the fast recombination of photogenerated charge carriers.

Thus, we report a novel and simple in situ synthetic approach toward CdS/g-C₃N₄ hybrid nanocomposites, which are based on the growth of CdS nanoparticles and nanorods within the g-C₃N₄ polymers from metal-containing carbon nitride compounds, similar to the synthesis of Fe-g-C₃N₄ [28] and ZnO-g-C₃N₄ [29]. CdS/g-C₃N₄ hybrid nanocomposites are synthesized using low-cost melamine and Cd(CH₃₋ COO)₂·2H₂O as precursors to obtain the Cd-g-C₃N₄, which are combined through a polycondensation process at elevated temperatures, and combined with sulfur. This process assures the two active components are efficiently in contact with each other. The synergistic effects of the two components significantly enhance the photocatalytic performance.

Experimental section

Chemicals

All reagents were analytical grade without any further purification before the experiment.

Synthesis of Cd/g-C₃N₄ compounds

First, $Cd/g-C_3N_4$ compounds were synthesized through a hydrothermal method. An amount of 4 g of melamine

mixed with 100 mL of deionized water and 1 g of $Cd(CH_3COO)_2 \cdot 2H_2O$ was heated and stirred at 80 °C for 40 min to create $Cd/g-C_3N_4$. The mixture was transferred into a Teflon-lined stainless autoclave (50 mL), maintained at 150 °C for 4 h, and then naturally cooled at room temperature. The precipitation was washed several times with deionized water to remove residual reactants and then dried in a vacuum oven at 60 °C for 3 h. The product was ground into powder and heated in a covered crucible under nitrogen atmosphere for 4 h to reach 550 °C. The Cd/g-C₃N₄ precursors were obtained by naturally cooling down to room temperature under nitrogen gas.

Synthesis of CdS/g-C₃N₄ hybrid materials

The obtained light yellow bulk was ground into powder and dispersed in saturated thiourea solution (35 mL)through ultrasonic treatment for 10 min. The mixed suspension was transferred into Teflon-lined stainless autoclave, kept in an oven at 150 °C for 4 h, and then naturally cooled to room temperature. Finally, the product was collected, washed with distilled water, vacuum dried, grounded, and stored for characterization.

The pure $g-C_3N_4$ was obtained through the same conditions but without adding the Cd(CH₃COO)₂·2H₂O powder.

Characterization

The structures and compositions of the products were analyzed using X-ray diffraction (XRD) on a BRUKER D8 ADVANCE X-ray powder diffractometer with Cu K α_1 radiation ($\lambda = 1.5406$ Å) within the 2 θ range of 5°–80°. The accelerating voltage and emission current were 40 kV and 40 mA, respectively. Transmission electron microscopy (TEM) and high-resolution transmission electron microscopy (HRTEM) images were measured on FEI Tecnai G2 (F30 S-TWIN/X-MAX80) at 200 kV. Ultraviolet visible (UV-vis) diffuse reflectance spectra were measured using a UV-vis spectrophotometer (Agilent Cary 5000, USA). A Hitachi F-2500 spectroscope recorded the products of the photoluminescence (PL) spectra and is equipped with an integrated sphere attachment. Fourier transform infrared (FT-IR) spectra were recorded using Spectrum100 (PerkinElmer, USA). FT-IR spectrometry within the wavenumber range of $600-4000 \text{ cm}^{-1}$ was applied using the KBr pellet technique. The Brunauer-Emmett-Teller (BET) surface area was determined by a multipoint BET method using Gemini VII 2390. Electrochemical measurements were performed with a CHI 660E electrochemical workstation (Chenhua Instrument, Shanghai, China) in a standard three-electrode system. The visible irradiation was obtained from a 500 W Xe lamp (CHF-MX, Beijing).

Photocatalytic activity for pollutant degradation

Photoelectrochemical measurements

Photocurrent measurements were performed on a CHI 660E electrochemical workstation in a conventional threeelectrode configuration using a Pt foil and Ag/AgCl (saturated KCl) as counter and reference electrodes, respectively. A 500 W Xe arc lamp served as the light source. A 0.5 M Na₂SO₄ aqueous solution was used as the electrolyte. The working electrodes were prepared as follows: 10 mg of the prepared product was ground with 20 μ L of a PEDOT-PSS (Sigma-Aldrich, 1.3-1.7 %) aqueous solution and 100 µL of distilled water to create a slurry. The slurry was then spread on a $2 \times 1.5 \text{ cm}^2$ FTO glass substrate, with an active area of approximately 0.5 cm^2 , using the doctor-blade method; adhesive tape was used as the space. The film was air dried and annealed in a vacuum oven at 150 °C for 30 min. The photoresponses of the samples as the light was turned on and off were measured at 0.2 V.

Measurement of photocatalytic activity

Degradation of methyl orange

The photocatalytic activities were evaluated through the decomposition of methyl orange (MO) under visible light irradiation ($\lambda = 400$ nm), which was obtained from a 300 W mercury lamp (XGY-1 Nanjing University) with cut-off filter of 400 nm. An amount of 0.1 g of the photocatalyst was dispersed in 200 mL of MO solution (the initial concentration was 10 mg L^{-1}) in the dark and stirred for 30 min to reach the adsorption equilibrium before the photocatalytic reaction. Approximately 10 mL of the mixture was sampled at selected times during the photoreaction process, and the suspension was then centrifuged and filtered to separate the photocatalyst particles. The temperature of reaction solutions was sustained at 25 °C with cooling water circulating during the reaction. The concentration of MO was determined by the UV-vis spectrophotometer using a Shimadzu UV-2450 spectrophotometer at 464 nm.

Evolution of hydrogen

The photocatalytic hydrogen evolution was conducted in a closed gas circulation system, and a top window Pyrex cell was used as the reactor. The irradiation light source was a 300 W Xe lamp with a cut-off filter ($\lambda > 420$ nm), which was positioned 20 cm away from the reactor. In a typical photocatalytic experiment, the photocatalyst g-C₃N₄ (0.1 g) reaction system contained 10 ml of triethanolamine (TEOA) [30] aqueous solution, and the reaction system of

CdS/g-C₃N₄ (0.1 g) was the aqueous solution with 1.0 M Na₂SO₃ and 0.35 M Na₂S [31]. TEOA, Na₂SO₃, and Na₂S were used as sacrificial reagents for photocatalytic hydrogen generation. The Pt cocatalyst (2 wt%) was photode-posited on the photocatalyst particles by directly dissolving H₂PtCl₆·6H₂O into the reactant suspension. The temperature of the reaction solutions was sustained in an ambient temperature because of the flow of cooling water during the reaction. The other test conditions were equivalent. The evolved gas was determined by the online gas chromatography (Shimazu GC-14, with nitrogen as a carrier gas).

Results and discussion

Hybrid structure and optical properties

Figure 1(left) shows the XRD patterns of $g-C_3N_4$ -CdS composite photocatalysts with different ratios of CdS/g- C_3N_4 , pure g- C_3N_4 , and Cd/g- C_3N_4 . The XRD patterns reveal graphitic stacking of C_3N_4 layers in pure g- C_3N_4 , Cd/g-C₃N₄, and CdS/g-C₃N₄. The peaks of 13.1° and 27.77° were the two characteristic diffractions of the g-C₃N₄ sample. The peak at 13.1° corresponded to the inplane structural packing motif of the tristriazine unit and is indexed as the (100) peak [9]. The strong diffraction peak at 27.77° indicated the graphite-like stacking of the conjugated aromatic units of CN, which was indexed to the (002) plane of the hexagonal $g-C_3N_4$ (JCPDS# 87-1526). Both g-C₃N₄-CdS samples exhibited diffraction peaks corresponding to C₃N₄ and hexagonal wurtzite-structured CdS (JCPDS# 65-3414). No other impure peaks were observed, which suggests a two-phase composition of g-C₃N₄ and CdS in these composites. The two peaks of 13.1° and 27.77° gradually decreased in intensity when CdS was introduced, which is similar to the report of Fu Jie [26]. The XRD pattern of 1:4 CdS/g-C₃N₄ composites revealed distinct reflections at 24.925°, 26.450°, 28.217°, 36.424° , and 43.858° that corresponded to the (100), (002), (101), (102), and (110) planes of CdS, respectively.

Figure 1(right) shows the typical EDS spectrum of the CdS/g-C₃N₄ composites. Peaks associated with Cd, S, C, and N were observed in the EDS spectrum. The C/N molar ratio is nearly 0.75, and the S/Cd molar ratio is nearly 1.1. Both of them are close to the theoretically predicted of the C₃N₄ and CdS empirical stoichiometry, respectively. Similar observations were also reported for CdS nano-structured materials synthesized by solvothermal and hydrothermal methods [32]. The BET surface areas of g-C₃N₄, CdS, and CdS/g-C₃N₄ (1:4) photocatalysts were shown in Table 1. The BET surface areas of the pristine g-C₃N₄ and CdS were about 21.1 m² g⁻¹ and 12.8 m² g⁻¹, respectively. After CdS was doped, the BET of CdS/g-



Fig. 1 XRD patterns of (a) pure $g-C_3N_4$, (b) Cd/ $g-C_3N_4$, (c) $1/10CdS/g-C_3N_4$, (d) $1/4CdS/g-C_3N_4$ (*left*) and EDS spectrum of $1/4CdS/g-C_3N_4$ (*right*)

Table 1 BET surface areas of g-C₃N₄, 1:4CdS/g-C₃N₄ and CdS

Sample	S(BET) (m ² /g)	Pore volume (cm^3/g)
g-C ₃ N ₄	21.1	0.15
1:4CdS/g-C ₃ N ₄	20.8	0.13
CdS	12.8	0.06

 C_3N_4 (1:4) was 20.8 m² g⁻¹. The change in BET may be related to the hybrid of CdS and g-C₃N₄. Accordingly, the pore volume of the g-C₃N₄–CdS (1:3) is 0.13 cm³/g, between that of g-C₃N₄ or CdS.

Figure 2a shows the TEM image of pure $g-C_3N_4$. The pure g-C₃N₄ had an obviously thin laminar structure. Figure 2b-d shows the lower and higher magnification TEM images of the obtained CdS/g-C₃N₄ (1:4) composites. The obtained CdS/g-C₃N₄ nanocomposites were composed of two phases: hexagonal CdS and g-C₃N₄ polymer. The CdS/ g-C₃N₄ hybrid nanocomposites were randomly distributed on the surface of g-C₃N₄ sheet (Fig. 2b). Several nanoparticles and nanorods were embedded in the obtained $CdS/g-C_3N_4$ nanocomposites. The particles were 3–5 nm in size and the rods had diameters of 4-5 nm and the length of 25-30 nm were gathered and intertwined to form sphere-like nanocomposites (Fig. 2c, d). These hybrid nanocomposites had a size of 65 nm (Fig. 2c). Similar observations were also reported for CdS microspheres prepared through a self-assembly method [32-35]. The HRTEM image (Fig. 2c-d) revealed the exact structure of the obtained CdS/g-C₃N₄ hybrid nanocomposites: the single crystalline CdS nanoparticles were embedded into the $g-C_3N_4$ polymers and nanorods located at the nanosphere border. The lattice part was CdS nanocrystals and the amorphous part was g-C₃N₄ polymers. The HRTEM image clearly revealed a close interface between the g-C₃N₄ and the CdS nanoparticles and nanorods in the composite,

which formed heterojunctions. Consequently, the charge transfer between the two semiconductors was spatially smooth, which was fundamental to improving photocatalytic activity. The CdS crystallite in the hybrid nanocomposites exhibited three lattice planes with spacings of 0.357, 0.336, and 0.181 nm, which corresponded to the (100), (002), and (103) lattice planes of the hexagonal structured CdS nanocrystal, respectively. The lattice spacings of 0.336 and 0.181 nm observed in Fig. 2d also agreed well with the interplanar distances of the (002) and (103) direction parallel in the wurtzine phase of CdS.

Figure 3 shows the FT-IR spectra of three kinds of samples. Several strong bands were found in the 1200–1650 cm⁻¹ region, which corresponded to the typical stretching modes of the aromatic CN heterocycles as reported [2, 36]. The peaks were approximately at 1236, 1312, 1397, and 1636 cm⁻¹. Moreover, the characteristic breathing mode of the triazine units was observed at 802 cm⁻¹ [2, 37]. The broad bands at approximately 3000 cm⁻¹ corresponded to the stretching mode of the terminal NH₂ or NH groups at the defect sites of the aromatic ring [38]. The bands at 1533, 1457, 1129, and 711 cm⁻¹ were attributed to the Cd–S bond [26, 29]. The FT-IR spectrum further confirmed the formation of composites that contained the two phases of CdS and g-C₃N₄.

Figure 4 shows the UV–Vis spectrum of the pure $g-C_3N_4$, 1:4 CdS/ $g-C_3N_4$, and 1:10 CdS/ $g-C_3N_4$ composites. The pure $g-C_3N_4$ exhibited an absorption edge at 457 nm, which corresponded to the band gap of 2.71 eV. The samples of CdS/ $g-C_3N_4$ with mass ratio of 1:10 and 1:4 had absorption edges at 490 and 497 nm, which corresponded to the band gaps of 2.53 and 2.49 eV, respectively. The absorption value and band gap of the pure CdS sample was 514 nm, which corresponded to the band gap of 2.41 eV according to the report of Cao [24]. Thus, the CdS/ $g-C_3N_4$ composites exhibited hybrid absorption



Fig. 2 TEM images of a pure g-C₃N₄ and b, c, and d CdS/g-C₃N₄ (1:4)



Fig. 3 FT-IR spectrum of (a) g-C₃N₄, (b) 1:10CdS/g-C₃N₄, and (c) 1:4CdS/g-C₃N₄

features of $g-C_3N_4$ and CdS, which allowed for efficiently utilizing the solar spectrum to create photogenerated electrons and holes. Figure 4 shows a red shift at -40 nm of the absorption edge (from pure $g-C_3N_4$ to CdS/ $g-C_3N_4$) and the enhanced absorption intensity of CdS/ $g-C_3N_4$ samples, which cause photocatalytic activity [39].

Figure 5 shows the photoluminescence spectra of pure $g-C_3N_4$, 1:10 CdS/ $g-C_3N_4$, and 1:4CdS/ $g-C_3N_4$ with an excitation wavelength of 275 nm. The pure $g-C_3N_4$ sample exhibited a strong emission peak centered at 460 nm. The intensity of the emission spectral bands significantly decreased when CdS was introduced, which indicated an



Fig. 4 Absorption spectra of (a) pure $g-C_3N_4$, (b) 1:10CdS/ $g-C_3N_4$, and (c) 1:4CdS/ $g-C_3N_4$

efficient transferring of photoexcited electrons from $g-C_3N_4$ to CdS. This spectra were useful in the migration, transfer of charge carriers, and separation and recombination processes of the photogenerated electron-hole pairs as Samanta S. reported [40, 41].

Photocatalytic activity and photocurrent response

Figure 6 shows the MO degradation curves of the photocatalysts of g-C₃N₄, CdS, and CdS/g-C₃N₄ under light irradiation ($\lambda = 400$ nm). Adsorption tests of photocatalysts were performed before running the photocatalytic



Fig. 5 Photoluminescence spectra of (*a*) pure g-C₃N₄, (*b*) 1:10 CdS/ g-C₃N₄, and (*c*) 1:4 CdS/g-C₃N₄



Fig. 6 Photocatalytic degradation of MO over (*a*) g-C₃N₄, (*b*) CdS/g-C₃N₄ and (*c*) CdS under visible light ($\lambda > 400$ nm)

reaction. All photocatalytic systems completely reached adsorption equilibrium with light irradiation after 30 min. The pure g-C₃N₄ solution and CdS solution was fully faded at 25 min and 60 min, respectively, but the suspension was fully faded at 15 min when the CdS/g-C₃N₄ hybrid nanocomposites were under light irradiation. To quantitatively investigate the reaction kinetics of the MB photocatalytic degradation, the photocatalytic data are analyzed using pseudo-first-order kinetics [26], which can be described as follows:

$$\ln C_0 / C_t = k_{\rm obs} t, \tag{1}$$

where C_0 is the initial MO concentration after stirring the suspension solution for 30 min, C_t is the MO concentration at irradiation time *t*, and k_{obs} is the observed pseudo-first-order rate constant. The observed first-order rate constant

 $k_{\rm obs}$ can be obtained from the linear relationship of $\ln(C_0/C_t)$ versus *t*. The calculated $k_{\rm obs}$ of g-C₃N₄–CdS (1:4) is 0.2745 min⁻¹, which is about 2 times as much as that of g-C₃N₄ (0.1187 min⁻¹), indicative of an enhanced photocatalytic activity of g-C₃N₄–CdS photocatalysts. This result was due to the significant difference in the interface of the samples; the heterojunction photocatalyst exhibited a closely contacted interface, which is fundamental to improving photocatalytic activity.

The photocatalytic hydrogen evolution activity of the prepared CdS/g-C₃N₄ composite was evaluated under visible light irradiation ($\lambda > 420$ nm) using Pt (2.0 wt%) as the cocatalyst to reduce the overall potential for H₂ evolution. The sacrificial reagents of 1.0 M Na₂SO₃ and 0.35 M Na₂S were used to consume the photo-induced holes. In a similar experiment with pure g-C₃N₄, the triethanolamine was the sacrificial reagent. Other test conditions were equivalent. Figure 7 shows the H₂ evolution rate over the sample under visible light irradiation. The catalytic g-C₃N₄ and CdS/g-C₃N₄ were uniformly loaded Pt (2.0 wt%). The hydrogen yield performance capacity of $CdS/g-C_3N_4$ was obviously higher than the g-C_3N_4. The maximum hydrogen production rate of CdS/g-C₃N₄ reached 259 μ mol h⁻¹, which was 2.8 times more than that of pure g-C₃N₄. A photocatalyst must have good stability to convert solar energy. Thus, the catalyst was subjected to continuous illumination experiments to evaluate its stability. A two-cycle experiment was performed. The amount of hydrogen reached approximately 1300 µmol after 6 h in the first cycle, and the two-cycle experiment had a hydrogen yield of 1000 µmol. This result demonstrated that photocatalysis performed better under visible light.

Figure 8 shows the results of the transient photocurrent response obtained from the pure $g-C_3N_4$ and CdS/ $g-C_3N_4$ samples. The photocurrent intensity remained at a constant



Fig. 7 Plots of photocatalytic H₂ evolution amount vs. irradiation $(\lambda > 420 \text{ nm})$ times for (a) g-C₃N₄ and (b) CdS/g-C₃N₄



Fig. 8 Transient photocurrent responses for (a) pure $g-C_3N_4$ and (b) CdS/g-C_3N_4



Fig. 9 Schematic of the photogenerated charge-transfer processes between $g-C_3N_4$ and CdS in photocatalytic hydrogen under visible light

value when the light was on and rapidly decreased when the light was turned off. The photocurrent over CdS/g- C_3N_4 greatly improved, which was 2.3 times better than that of pure g- C_3N_4 .

Figure 9 shows the schematic of the photogenerated charge-transfer process between g-C3N4 and CdS in photocatalytic hydrogen under visible light. Both g-C₃N₄ and CdS were excited by the visible light and produced photogenerated electrons and holes. The conduction and valence band edge potentials of $g-C_3N_4$ were located at -1.12 and 1.57 eV [42] and CdS was located at -0.5 and 1.9 eV. Under light irradiation, the photo-induced electrons on the conduction band (CB) of g-C₃N₄ directly transferred to the CB of CdS. These electrons, together with the electrons excited from the valent band (VB) of CdS, accumulated on the Pt deposited on CdS nanoparticle surfaces. Pt can effectively capture electrons. Meanwhile, holes on the VB of CdS spontaneously transferred to the VB of $g-C_3N_4$. Therefore, an effective charge separation was achieved, which resulted in a prolonged lifetime for the photogenerated electrons and holes to

enhance photocatalytic activity. Consequently, the photoinduced holes oxidized SO_3^{2-} and S^{2-} on the g-C₃N₄ surface. Meanwhile, the separated electrons had ample time to reduce H₂O to H₂ on the Pt particle surface. However, CdS is unstable in aqueous solutions under irradiation [10]. Thus, CdS/g-C₃N₄ heterojunctions can effectively promote the charge separation and transfer corrosive holes from CdS to g-C₃N₄. This phenomenon makes the charge separation efficient, reduces the probability of charge recombination, and prevents the oxidation of CdS by holes, which results in enhanced photocatalytic activity.

Conclusions

Heterojunctions between polymers g-C₃N₄ and semiconductors CdS were successfully fabricated using a simple in situ growth method. Heterojunctions with close interfaces between the g-C₃N₄ and the CdS nanoparticles and nanorods in the composite were confirmed by the XRD, TEM, and FT-IR measurements. The UV-vis diffuse reflectance spectra exhibited a red shift that showed the CdS was introduced into the polymer g-C₃N₄, which allowed for the efficient utilization of the solar spectrum to create photogenerated electrons and holes. The PL spectra of the nanocomposites indicated a charge transfer from $g-C_3N_4$ to CdS. The photocatalytic activity and photocurrent for the CdS/g-C₃N₄ hybrid nanocomposites were significantly enhanced, which was due to an increased interfacial contact between the CdS and g-C₃N₄ polymer, fundamental to improving photocatalytic activity.

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