Mechanochemically assisted synthesis of pristine Ca(II)Sn(IV) layered double hydroxides and their amino acid intercalated nanocomposites

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Abstract Syntheses of Ca(II)Sn(IV)-layered double hydroxides (LDHs) are attempted by the traditional coprecipitation as well as mechanochemical methods. Both the co-precipitation method and the one-step milling operation proved to be unsuccessful; these methods only produced physical mixtures of hydroxides and carbonates of the two metal ions. However, a two-step milling operation (dry milling followed by milling in the presence of minute amount of water) led to successful synthesis, verified by a range of characterisation methods. Surprisingly, it was found that ball-milling was not even necessary; the reaction proceeded on manual grinding of the components

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in an agate mortar with a pestle. The preparation of nanocomposites through intercalation of the anions of cystine or valine into Ca(II)Sn(IV)-LDH could also be achieved by the two-step milling method verified again by a range of instrumental methods.

Introduction

Layered double hydroxides (LDHs) are receiving increasing attention in recent years due to their unique structure and wide-ranging technological applications. The structure of LDHs consists of brucite-like layers, in which some of the divalent cations are isomorphously substituted with trivalent cations, thus providing a layer with positive charge. In these layers, the cations are surrounded by hydroxide ions in an octahedral arrangement. To compensate the positive charge of the layers, different types of anions together with water molecules are situated in the interlayer space [[1\]](#page-8-0).

The compounds can be described by the general formula of $[M(II)_{1-x}M(III)_x(OH)_2]^{x+}$ $[(A^-)_{x/n} y \cdot H_2O]^{x-}$, in which M(II) and M(III) are di- and trivalent cations, respectively, and A^- is the charge-compensating anion. A large variety of $M(II)$, $M(III)$, and A^- can be used to synthesize LDHs. The most frequently used divalent cations are Mg(II), $Cu(II), Ca(II), Co(II), Zn(II), etc.,$ and the trivalent ones are Al(III), Fe(III), Cr(III), etc., while the most common anions are CO_3^2 , NO₃, and Cl⁻ [\[2](#page-8-0)].

It was proven that LDH structure was formed, even if $M(II)$ was replaced by Li^+ . Then, the layers have the composition of $[LiAl_2(OH)_6]^+$; these materials have been studied and characterized thoroughly [[3\]](#page-8-0).

Partial or full replacement of the M(III) ions for M(IV) ones is an entirely different matter. The first mention of M(IV)-containing LDHs was as early as 1997 [\[4](#page-8-0)]. They reported the successful synthesis of $Mg(II)Al(III)Zr(IV)$ -LDH. Soon after that, they have published a paper, in which they claimed the success of the preparation of $Mg(II)Al(III)Sn(IV)$ -LDH using the simple co-precipitation method [\[5\]](#page-8-0). The claim was based on the changes in the lattice parameters due to the presumed incorporation of Sn(IV), obtained from the X-ray diffraction patterns. They also suggested a mechanism for the incorporation of Sn(IV). The theory was that some of the trivalent cations in the LDH layer were substituted by the tetravalent cations. After this, the syntheses of other similar systems (NiAlSn and CoAlSn) were also reported by this group $[6, 7]$ $[6, 7]$ $[6, 7]$ $[6, 7]$.

Soon after that, another group [[8,](#page-8-0) [9](#page-8-0)] embarked on studying this topic again, and attempted the synthesis of Mg(II)Al(III)Sn(IV)-LDH with the method given earlier [\[5](#page-8-0)], and for characterisation, 119 Sn Mössbauer and X-ray absorption spectroscopies were applied. It was found that the tetravalent cations were not incorporated in the layers of the LDH, instead, they formed M(IV) oxide-like particles, which were segregated from the layers.

After these findings were published, the scientific community with an interest in M(IV)-type LDHs has been divided into two main groups. Some years later, the first one continued to study $M(II)M(III)M(IV)$ systems [\[10](#page-8-0), [11](#page-8-0)]. They also found that the Sn(IV) ions were not incorporated in the layers, nevertheless, the segregated Sn(IV) oxidelike particles improved the catalytic properties of these materials. The other one tried a different, and in a way safer approach, and attempted to prepare $M(II)Sn(IV)$ -LDHs [\[12](#page-8-0), [13](#page-8-0)]. It is safer, since there is no added M(III); thus, there is no competition with the usually more favored M(II)M(III)-LDH formation. Just like in the other cases, solution-phase chemical method (hydrolysis with urea) was applied, and success in the syntheses of Co(II)Sn(IV)-LDH samples were reported.

Obviously, these results are contradicted, and it seemed worth trying to validate them and find a resolution. This is the major objective of this contribution.

The most common technique used for the preparation of LDHs is co-precipitation, a solution-phase chemical method. In this, aqueous solutions of M(II) and M(III) cations are used as precursors. These solutions are mixed together; then, with a base (usually aqueous NaOH or urea solution), the corresponding pH value is set; thus, the metal ions precipitate in the form of LDH [[14\]](#page-8-0).

Another not so commonly used preparative technique is the mechanochemical method, which was applied for LDH synthesis only in recent years. A two-step milling operation was developed [\[15](#page-8-0)], in which $M(II)(OH)_2$ and $M(III)(OH)_3$ were used as precursors. The first step was dry milling, which was followed by the addition of minute amount of water, and then, milling was continued (wet milling). The addition of water

was necessary, because the precursors did not provide enough hydroxide groups to form the LDH. This method has not yet been used for LDHs containing M(IV) cations.

The fact that anions situated (intercalated) between the layers can be replaced with other anions is one of the most important features of LDHs. A great variety of anions can be intercalated to LDHs: simple inorganic anions, oxo- and polyoxometalate anions, anionic complexes of transition metals, organic anions, anionic polymers, etc. Many methods have been developed for the intercalation of the various anions; however, the most frequently applied ones are intercalation using co-precipitation (e.g., [\[16](#page-8-0)]), direct anion exchange (e.g., [[17\]](#page-8-0)), and the dehydration-rehydration method (e.g., [\[18](#page-8-0)]). Intercalation of the anionic form of methotrexatum (an antifolate drug) was intercalated mechanochemically into MgAl-LDH [[19\]](#page-8-0), and since this method has not been applied for M(IV)-containing LDH systems yet, to see, whether it works or not for our system was the other objective of our work.

Studying the synthesis possibilities for the Sn(IV)-containing LDHs is important, because they can be used as catalysts in a large variety of reactions like Baeyer–Villiger oxidation of cyclohexanone [[20\]](#page-8-0), photocatalysis [[21,](#page-8-0) [22](#page-8-0)], liquid-phase methanol carbonylation [\[23](#page-8-0)], etc.

In this study, our observations collected during the attempted preparation, and intercalation of M(II)M(IV)- LDHs, using the traditional co-precipitation method and the novel mechanochemically assisted route, are described. We are going to show that the solution-phase chemical method does not give M(II)Sn(IV)-LDHs, while the twostep milling as well as the manual grinding methods lead to the formation of the LDH structure, and the synthesis of nanocomposites through the intercalation of amino acid anions can also be performed this way.

Experimental

Materials and methods

Concentrated NaOH (\sim 20 M) stock solution was prepared from MilliQ water (Millipore) and analytical grade solid NaOH (Spectrum 3D), and its carbonate content was minimized. At this base concentration, the solubility of the $Na₂CO₃$ precipitate is negligible, and after filtration (polysulfone Nalgene filter) under inert (N_2) atmosphere, carbonate-free NaOH solution could be obtained. This stock solution was kept in airtight, caustic-resistant Pyrex bottle, and was then diluted to the desired concentration under inert (N_2) atmosphere.

 $CaCl₂, CoCl₂ 6H₂O, Ca(OH)₂, L-cystine, DL-value, and$ SnCl4 6H2O (Sigma-Aldrich) were of analytical grade and were used as received.

The first technique applied was the co-precipitation method. LDHs were prepared via dropwise addition of the two metal salt solutions (CaCl₂ or CoCl₂ and SnCl₄) in various molar ratios to a vigorously stirred NaOH solution. The pH of the solution containing the two metal ions was set to 1.71 using a 20 % m/m HCl solution (Aldrich), in order to prevent the hydrolysis of the Sn salt. The concentration of NaOH was 1 M, and the quantity was calculated to have a final pH of 9. The mixture of the two solutions was stirred for 24 h. The precipitate was filtered until air dry with the aid of caustic-resistant vacuum filter unit (Nalgene, USA) equipped with an appropriate membrane (Versapor $0.45 \mu m$, Pall Corporation). The solid reaction products were kept at room temperature in desiccators over dry $SiO₂$.

The $Ca(II)$ to $Sn(IV)$ molar ratios were varied in a wide range (from 3:1 to 6:1), while for the $Co(II)Sn(IV)$ samples, it was kept constant at 3:1. In the latter case, the samples were aged for 6 h at a temperature of 70 $^{\circ}$ C.

The second method, which is a novel technique for these systems, was the mechanochemical route. The mechanochemical syntheses were applied in two ways: by a mixer mill or a simple agate mortar. The mill used was a Retsch MM 400 mixer mill, with two stainless steel grinding jars of 50 cm³ internal volumes each, and stainless steel grinding balls of 20 mm in diameter. The rotation frequency was 11.6 s^{-1} and was kept constant throughout the syntheses, just like the total weight of the precursors $(3 g)$. Two methods were investigated: the one-step and the twostep millings. In the former one, only dry milling was performed, and the precursors were ground for 3 h. In the latter one, in addition to dry milling, wet milling was also applied. The precursors were ground when dry for 1 h, and then 0.7 ml of water was added, which was followed by a 2-h long wet grinding. After the synthesis, the products were kept at room temperature in a desiccator over dry $SiO₂$.

When using the agate mortar, the results of the abovementioned methods were meant to be reproduced; thus, the methods applied were analogous to the one mentioned above in both the one-step (dry) and two-step (dry and wet) millings. In this case, the starting materials (3 g altogether) were ground until a uniform, fine powder was obtained, and then, the mixture was divided into two parts. One part was placed in a desiccator and was studied without further treatment. To the other part, 0.35 ml of water was added, and then grinding was continued until a uniform and homogenous mixture was obtained. The product thus received was dried and kept in a desiccator.

The intercalation of various amino acid anions was performed using a method similar to the two-step milling process mentioned above. The first step was dry milling, where the precursors were ground for 1 h; then, a certain amount of 0.1 M NaOH and the amino acid was added, which was followed by 2 h of wet grinding. Preliminary experiments were performed prior to attempt the intercalation, which showed that the ball to sample weight ratio (B/S) had a major effect on the formation of the LDH phase. The results revealed that the $B/S = 100$ value was optimal for the LDH formation. This value can be achieved by reducing the weight of the samples. In the intercalation experiments, $B/S = 100$ was used, and therefore, the amount of NaOH had to be reduced. The amount of amino acid was set in the way that the molar ratio of Sn(IV) and the amino acid was 1:1. The addition of the precursors, the amino acids, and the NaOH into the grinding jars, as well as the sealing of the jars were performed under N_2 atmosphere in a glove box. In this way, the disturbing $CO₂$ could be excluded. After the synthesis, the products were kept in desiccators, at room temperature.

Apparatus and equipment

Powder X-ray diffraction (XRD) patterns of the dry solid samples were recorded in the $2\theta = 3^{\circ} - 60^{\circ}$ range (θ is the incidence angle of the X-ray beam) on a Philips PW1710 instrument, using CuK α ($\lambda = 1.5418$ Å) radiation.

The morphologies of the substances obtained were studied using a Hitachi S-4700 scanning electron microscope (SEM) at various magnifications. The ratios of the di- and tetravalent ions in the solid samples and elemental mapping were determined using a Röntec QX2 energy dispersive X-ray (EDX) analyzer coupled to the microscope.

The Fourier-transform infrared (FTIR) spectra of the pristine, the organic anion-intercalated LDH nanocomposite, and the intercalated anion were recorded on a BIO-RAD FTS-65A/896 spectrometer equipped with a DTGS detector in diffuse reflectance. Spectral resolution was 4 cm^{-1} , and 256 scans were collected for a spectrum. The spectra were baseline corrected and smoothed using the WIN-IR software package. The samples were finely ground and combined with KBr without pressing into pellets.

Raman spectra were recorded on a BIO-RAD Digilab Division dedicated FT-Raman spectrometer equipped with liquid nitrogen-cooled germanium detector and $CaF₂$ beam-splitter. The excitation line was provided by a Spectra Physics T10-106C Nd:YVO4 laser at 1064 nm. The spectra were recorded in the $3600-100$ cm⁻¹ range with 4 cm^{-1} resolution. 1024 scans were collected for each spectrum. The excitation power was 280 mW at the sample position. The spectrometer was controlled by the BIO-RAD Win-IR 3.3 software. The samples were placed in a standard NMR tube. Spectra were recorded at room temperature. Data were processed by the GRAMS/AI 7.00 software.

Results and discussion

In order to prepare Ca(II)Sn(IV)-LDHs, two preparation methods (co-precipitation and mechanochemistry) were tried. Samples prepared by the co-precipitation method, with their acronyms, molar ratios and synthesis parameters, are presented in Table 1.

The X-ray diffractograms of these samples are presented, without baseline correction or smoothing in Fig. 1.

The typical diffractogram of an LDH displays three major reflections (003, 006, and 009), from which the 003 is the most significant—it appears around 10° 2 θ value. The presence or lack of these reflections was investigated to determine whether the LDH formation was successful or not.

The diffractograms of the Ca(II)Sn(IV) samples do not exhibit drift in the baseline, i.e., the crystallinity is good. The Ca:Sn molar ratio was varied in a wide range. In neither case could the specific reflections of the LDHs be recognised, however. In order to identify the reflections, a randomly chosen diffractogram was analyzed (Fig. 2).

The typical reflections of $Ca(OH)_2$ and $CaCO_3$ could be identified $(CO₂$ is most probably airborne). This proves that $Ca(OH)$ ₂ was precipitated, but the incorporation of the Sn(IV) ions into its layers did not happen. The presence of $CaCO₃$ could be explained by the fact that during the

Table 1 Samples prepared with the co-precipitation method

Ca(II):Sn(IV) molar ratios	c(SnCl ₄)	pН
3:1	0.04	Ч
4:1	0.04	
5:1	0.04	
6:1	0.04	
	c(Ca(II)Cl ₂) (mol L^{-1}) 0.12 0.16 0.20 0.24	(mol L^{-1})

Fig. 1 Powder X-ray diffractograms of Ca(II)Sn(IV) samples prepared by the co-precipitation method: (a) $Ca₃Sn$, (b) $Ca₄Sn$, (c) Ca₅Sn, (d) Ca₆Sn

Fig. 2 Powder X-ray diffractogram of the Ca₃Sn sample prepared by the co-precipitation method

synthesis, the air was not excluded. It can be concluded that using the co-precipitation method, Ca(II)Sn(IV)-LDHs could not be synthesized.

It is worth mentioning that our attempt for the preparation of $Co(II)Sn(IV)$ -LDH $(Co(II)/Sn(IV) = 3)$ by the co-precipitation method was also unsuccessful, in spite of aging the freshly precipitated material for 6 h at 343 K.

After these experiments, we gave up using the solutionphase chemical method and turned to mechanochemistry. Samples prepared using the various mechanochemical and mechanochemically assisted methods, with their designations, molar ratios and synthesis parameters, are presented in Table [2](#page-4-0).

The XRD results for the $Ca(II)Sn(IV)$ samples synthesized via the pure mechanochemical (dry or one-step milling) and the mechanochemically assisted (dry plus wet or two-step milling and wet grinding) methods are presented in Fig. [3](#page-4-0).

In the one-step milling operations (B and D curves), there were no LDH formation; the LDH-specific reflections are missing. However, in the two-step milling operations (traces A and C) at 10° 2 θ values, the 003 reflection specific to the LDH structure appeared. Thus, it is strongly indicated that the two-step milling method produced Ca(II)Sn(IV)-LDH. The observation that the synthesis was only successful in the two-step milling process is due to the fact that using only dry milling, the crystal water introduced with the precursors could not generate sufficient amounts of hydroxide group necessary for the LDH structure.

Using the Bragg equation, the interlayer spacing of the LDHs could be calculated from the positions of the 003 reflections, and they were 8.38 and 9.05 Å for the Ca4Sn_mm_w and Ca4Sn_m_w samples, respectively.

For providing further proofs of LDH formation, SEM measurements were performed in order to gain information

Sample	Ca(II):Sn(IV)	Frequency (s^{-1})	Added $H2O$ (mL)	Dry milling (h)	Wet milling (h)
$Ca_4Sn_mm_d$	4:1	11.6			
$Ca_4Sn_mmm_w$	4:1		0.7		
Ca_4Sn _{-m} _d	4:1	$\qquad \qquad -$		Na	Na
Ca_4Sn _m_w	4:1		0.35	Na	Na

Table 2 Samples prepared with the mechanochemically assisted methods: using mixer mill with dry (mm_d) and wet (mm_w) milling and using agate mortar with dry (m_d) and wet (m_w) grinding

Fig. 3 Powder X-ray diffractogram of Ca(II)Sn(IV) samples prepared via mechanochemical methods (a) Ca₄Sn_mm_w, (b) $Ca_4Sn_mm_d$, (c) $Ca_4Sn_m_w$, (d) $Ca_4Sn_m_d$

about the morphology of the sample. The hexagonally shaped lamellar structure, the typical morphology of LDHs was sought in the samples. As Figs. 4 and 5 attest the SEM images of $Ca(II)Sn(IV)$, samples prepared with two-step milling or the wet grinding methods show these characteristics.

The particle sizes are larger than what one would expect for LDHs; however, it may be assumed that the amorphous phase is due to smaller particles sticking together to form bigger aggregates. The SEM images have confirmed the results of the XRD measurements: LDH was formed

Fig. 5 SEM image of the Ca_4Sn_m w sample at a magnification of 50000

indeed, during the two-step mechanochemically assisted syntheses.

Using EDX, the elemental composition could be obtained and projecting it on the corresponding SEM image, the elemental map, and thus the elemental distribution of the samples could be determined. For the $Ca₄Sn$ mm w sample, the elemental analysis was performed, from which it could be concluded that during the preparation of the samples, iron contamination (iron was the main component of the grinding balls and grinding jars) did not occur.

On the elemental map of both the $Ca_4Sn_mm_w$ and Ca_4Sn_m w samples (Figs. [6,](#page-5-0) [7\)](#page-5-0), it can be observed that

Fig. 4 SEM images of the Ca₄Sn_mm_w samples at magnifications of 13000 (a), and 50000 (b)

Fig. 6 Elemental distribution map (a) on the Ca₄Sn_mm_w sample (b)

Fig. 7 Elemental distribution map (a) on the Ca_4Sn_m w sample (b)

Table 3 Samples prepared with the mechanochemically assisted intercalation method: using mixer mill with two-step milling with the intercalated anion of cystine (mm_w_Cys) or valine (mm_w_Val)

Sample	Ca(II):Sn(IV) molar ratio	$Sn(IV): A^{-a}$ molar ratio	Frequency (1/s)	B/S	Added 0.1 M NaOH (uL)	Dry milling (h)	Wet milling (h)
$Ca3Sn_mm_w_Cys$	3:1	l:1	11.6	100	200		
$Ca3Sn_{mm}$ w Val	3:1	1:1	11.6	100	200		

 A^- represents the intercalated anion

the distributions of the Ca and Sn are uniform within the particles, and no aggregation or segregation can be noticed. This is also another piece of evidence supporting that LDH was formed during the two-step milling and also the wet grinding procedures.

For the preparation of nanocomposites, various organic anions (of cystine and valine) were intercalated in the LDHs formed, using the two-step milling method. The samples prepared in this way, with their designations, molar ratios and synthesis parameters, are presented in Table 3.

The diffractograms of the intercalated samples and that of the pristine LDH is displayed in Fig. [8](#page-6-0).

One of the most important proofs that the intercalation of the anions between the layers of the LDH succeeded is the change in the interlayer spacing. This change can be calculated from the position of the 003 reflection, using the Bragg equation. The lower the position of the 003 reflection in terms of 2θ angle, the larger the interlayer distance is. Thus, in the samples prepared, one is looking for the increased interlayer spacing, i.e., the presence of the 003 reflection at lower 2θ values relative to the pristine $Ca(II)Sn(IV)-LDH$ sample. In both cases (using the anions of cystine or valine), this phenomenon could be observed. For the samples containing the anions of valine and cystine, the 003 reflection shifted to 8.8° and 5.6° from 9.8° in the pristine LDH. The calculated interlayer spacings are summarized in Table [4.](#page-6-0)

Additional support for the success of intercalation is provided by SEM–EDX measurements. Using this method, the elemental distribution maps were taken, shown in Figs. [9](#page-6-0) and [10](#page-7-0).

The C–S and N–C elemental maps revealed that in both cases, the distributions of the amino acid anions were

Fig. 8 Powder X-ray diffractograms of the amino acid anion– Ca(II)Sn(IV)-LDH samples intercalated by the mechanochemically assisted pathway: (a) $Ca_3Sn_mm_w$, (b) $Ca_3Sn_mm_w_Val$, (c) Ca₃Sn_mm_w_Cys (the "original" 003 reflection of the LDH and the 003' reflection of the intercalated LDH are indicated)

Table 4 The interlayer spacings of the pristine and intercalated Ca(II)Sn(IV)-LDHs

Sample	Interlayer spacing (\check{A})		
$Ca3Sn_{mm}$ w	9.01		
$Ca3Sn_mm_w$ Val	9.99		
$Ca3Sn_{mm}$ w Cys	15.65		

uniform. The Ca–Sn elemental maps show that these two elements are also evenly distributed within the particle (which is one of the characteristics of the LDHs).

Other conclusive information could be gained regarding the success of the intercalation by IR spectroscopy. The IR spectra of the intercalated samples were taken and compared to those of the pristine LDH and the amino acids in zwitter ionic forms (Figs. [11](#page-7-0), [12](#page-7-0)).

At around 1100 cm^{-1} , the symmetric stretching vibration of the C–N bond is present in the spectra of both the intercalated sample and the zwitter ionic valine, and is not seen in the pristine LDH. This finding verifies that an organic–inorganic composite is in our hands. The presence of the vibrations of the carboxylate ion is additional evidence. Actually, the band corresponding to the asymmetric vibration can only be identified in the composite, since the symmetric vibration coincides with a band of the pristine LDH at 1645 cm^{-1} . Nevertheless, this peak shifted from 1492 to 1520 cm^{-1} indicating interaction with the layer of the LDH, i.e., the carboxylate ion is situated among the layers.

A similar reasoning may be given for the interaction of the LDH and the anion of cystine, since here, the symmetric carboxylate band can be identified too, and it also shifts in the composite compared to the zwitter ionic form (the shift is smaller though than in the previous case, it is from 1417 to 1428 cm^{-1}). In order to support the conclusion that the anion of cystine is also intercalated, measurements with Raman spectroscopy were also performed Fig. [13](#page-7-0).

The C–S bond stretching vibration at 499 cm^{-1} appeared in the Raman spectrum of the zwitter ionic Lcystine, and it is seen and shifted to 472 cm^{-1} in that of the composite sample as well. This confirms the intercalation of the anion of cystine into Ca₃Sn-LDH.

Finally, let us say few words about the possible mechanistic steps of the synthesis of the $Ca(II)Sn(IV)$ -LDH in the two-step milling procedure. Direct observation of events occurring in the mill was not possible; therefore, the results of measurements after milling are used for offering a mechanism. It is generally the case in mechanochemistry, however, this may change soon, since very recently, a Raman spectrometer translucent mill setup has been published [\[24](#page-8-0)] allowing in situ measurements during milling. Dry milling could not produce the LDH, but it was a necessary prerequisite of the successful synthesis. It provided intimate mixing and activation for the reaction to occur. The second step, wet milling that is, further activated the components, and water was necessary for the formation of the LDH, which possibly was a kind of hydrolysis-co-precipitation.

Fig. 9 Ca–Sn (b) and C–S (c) elemental distribution maps on the Ca₃Sn_mm_w_Cys sample (a)

Fig. 10 Ca–Sn (b) and N–C (c) elemental distribution maps on the $Ca₃Sn_mmm_w$ Val sample (a)

Fig. 11 FTIR spectra of $Ca₃Sn_mmm_w$ (a), $Ca₃Sn_mmm_w$ Val (b), and the zwitter ionic DL -valine (c)

Fig. 12 FTIR spectra of $Ca_3Sn_mm_w$ (a), $Ca_3Sn_mm_w$ Cys (b), and the zwitter ionic L -cystine (c)

Fig. 13 Raman spectra of $Ca_3Sn_mm_w$ (a), $Ca_3Sn_mm_w$ Cys (b), and zwitter ionic L -cystine (c)

Conclusions

The synthesis of Ca(II)Sn(IV)-LDH was attempted using the co-precipitation, a pure mechanochemical (one-step milling), mechanochemically assisted methods (two-step milling), and wet grinding. The first two techniques proved to be unsuccessful; however, both the two-step milling and the wet grinding procedures produced Ca(II)Sn(IV)-LDH.

Furthermore, the two-step milling technique performed superbly in the intercalation of the anions of cystine or valine producing organic–inorganic nanocomposites.

To sum up, the combination of dry and wet milling or wet grinding can be viable procedures in the preparation and intercalation of LDHs that cannot be easily approached, if at all, by solution-phase methods.

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