Double-sided transparent electrodes of $TiO₂$ nanotube arrays for highly efficient CdS quantum dot-sensitized photoelectrodes

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Abstract A novel double-sided CdS quantum dots-sensitized $TiO₂$ nanotube (TNT)/ITO photoelectrode is fabricated to improve the energy conversion efficiencies of quantum dots-sensitized solar cells (QDSCs). Our experimental results show that the double-sided CdS quantum dots-sensitized TNT/ITO photoelectrodes show enhanced light absorption. As a consequence, the photoelectrochemical response of the CdS/TNT/ITO photoelectrode is much improved compared with single-sided CdS sensitized TNT arrays on Ti substrate (i.e., CdS/TNT/Ti photoelectrode). An optimum conversion efficiency of 7.5 % is achieved by the double-sided CdS(15)/TNT/ITO photoelectrode, which is an enhancement of about 120 % when compared with the single-sided CdS/TNT/Ti photoelectrode. Our results demonstrate that the energy conversion efficiencies of QDSCs can be improved by designing a new photoelectrode structure.

Introduction

Quantum dot-sensitized solar cells (QDSCs) have attracted a great deal of attention as an alternative to conventional thin film photovoltaic (PV) devices due to the outstanding optoelectronic properties of quantum dots (QDs), their

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simple fabrication procedure and low cost. The photoelectrodes of QDSCs are typically $TiO₂$ nanoparticle porous films. In recent years, although QDSCs are progressing very rapidly $[1–5]$ $[1–5]$ $[1–5]$, the energy conversion efficiencies of QDSCs still lag behind those of dye-sensitized solar cells. The main reasons for the low efficiencies of QDSCs are low light-harvesting efficiency of QDs, low QDs loading amount due to small surface area in nanoparticle porous photoelectrode, slow electron transport, and the charge recombination at the photoelectrode surface [\[6](#page-6-0)]. For increasing the efficiency of QDSCs, considerable efforts have been made to improve the development of more efficient photoelectrode materials, such as highly ordered one-dimensional (1D) $TiO₂$ nanotube (TNT) photoelectrodes. Compared with $TiO₂$ nanoparticle porous films, TNT arrays were found to be superior in light-harvesting efficiency and electron transport ability due to its 1D channel [\[7](#page-6-0)], in which the internal light scattering was enhanced and the electron transfer through TNT photoelectrodes was promoted. Furthermore, the photogenerated charge recombination at the photoelectrode surface was suppressed by promoting faster electron transport. It has been demonstrated that the QDSCs-based TNT photoelectrodes showed higher energy conversion efficiencies than those of QDSCs-based $TiO₂$ nanoparticle electrodes [\[8](#page-6-0)]. Nevertheless, to further improve the conversion efficiencies of QDSCs based TNT photoelectrodes, there are still some issues to be resolved, such as low QDs loading amount due to the insufficient surface area for QDs and low light-harvesting efficiency.

The TNT arrays can be fabricated by a variety of methods including electrochemical anodization [\[9–12](#page-6-0)], hydrothermal synthesis [\[13](#page-6-0)], and template-assisted synthesis [[14\]](#page-6-0). For the TNT photoelectrodes in QDSCs, most of the TNT arrays were directly grown on the titanium

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Fig. 1 Scheme for fabricating a double-sided CdS sensitized

TNT/ITO photoelectrode

substrate (TNT/Ti) through electrochemical anodization of titanium metal in non-aqueous electrolytes [\[10–12](#page-6-0)]. However, only the QDs deposited on the illuminated side of such TNT/Ti-based photoelectrode can contribute to the light absorption due to the opaque Ti substrate, which suggests that the incident light can not be absorbed by the QDs on the other side of TNT/Ti-based photoelectrode and therefore leads to the waste of light absorbing materials (i.e., QDs) and the energy loss of incident light. Here, aiming at these issues, the transparent double-sided CdS QDs-sensitized TNT arrayed photoelectrodes are reported, in which the TNT arrays are oriented perpendicular to the double-sided indium tin oxide (ITO) glass substrates (i.e., double-sided TNT/ITO photoelectrode). Such photoelectrode was first fabricated using a process involving flaking the anodic TNT-array film off the Ti substrate and adhering it onto double-sided ITO glass by sol–gel processed NP- $TiO₂$ paste [\[15](#page-6-0)] and then sintering at high temperature. Several methods have been used to detach the TNT films from the Ti substrate, such as ultrasonication $[16, 17]$ $[16, 17]$ $[16, 17]$ $[16, 17]$, N₂ gas blowing combined with methanol wetting [[18\]](#page-6-0), solvent-evaporation [\[19](#page-6-0)], solution immersion [[20\]](#page-6-0), and twostep anodization [\[15](#page-6-0)] methods. In this study, ultrasonication method was selected. Moreover, the detached TNT arrays can also be transferred and adhered onto the singlesided FTO glass using Ti isopropoxide solution as a paste [\[21](#page-6-0)]. After the fabrication of double-sided TNT/ITO photoelectrodes, the as-prepared photoelectrodes were sensitized with CdS QDs by chemical-bath deposition (CBD). The detailed synthetic strategy is illustrated in Fig. 1.

In comparison with the commonly used opaque TNT-Ti photoelectrode with single-sided CdS sensitized structure, such fabricated photoelectrode with double-sided CdS sensitized structure provides more space to deposit CdS QDs and more channels to transfer electron to ITO glass substrate, which enhances simultaneously the light absorption and carrier collection of photoelectrodes. To our knowledge, there is no work dedicated to the CdS QDs-sensitized solar cells based on the transparent double-sided TNT/ITO photoelectrode. Only several similar double-sided photoelectrodes, such as double-sided $TiO₂/Ti/TiO₂$ [\[22\]](#page-6-0), ZnO/ITO/ZnO [\[3](#page-6-0)], and $TiO₂/FTO/Cu-Ti-O [23]$ $TiO₂/FTO/Cu-Ti-O [23]$ $TiO₂/FTO/Cu-Ti-O [23]$, are designed and fabricated for photoelectrochemical hydrogen generation. Therefore, our study demonstrates an innovative approach for improving the photoelectrode structure for highly efficient QDSCs.

Experiment section

Materials

ITO (\leq 15 Ω/\square , Wuhu Token Sci. Co., Ltd., China) sheet glass, Titanium foil (Ti, Sigma-Aldrich, 0.25 mm thickness, 99.7 % purity), cadmium chloride $(CdCl₂, Kanto Chemical)$ Co., 98.0 %), ammonium fluoride (NH4F, Sigma-Aldrich, 98+ %), sodium sulfide nonahydrate (Na₂S, Sigma-Aldrich, 98.0 %), $Ti(OCH_2CH_2CH_2CH_3)_4$ $(Ti(OBu)_4,$ Sigma-Aldrich, 97 %), and ethylene glycol (Junsei Chemical Co., 99.0 %) are used as received.

Synthesis of double-sided TNTs arrays/ITO and TNTs/ Ti films

Titanium foils were sonicated successively in acetone, ethanol and deionized (DI) water for 10 min respectively, and dried in air. The growth of TNT arrays was accomplished by direct anodization of the as-prepared Ti foil in a two electrode electrochemical cell with a platinum foil as the counter electrode at 55 V for 16 h at room temperature, which contained 0.25 wt% NH₄F and 0.75 wt% DI water in ethylene glycol (99.0 wt%). The experimental procedure is similar to our previous papers [\[10](#page-6-0), [24](#page-6-0)]. The anodized samples were washed successively in ethanol and DI water to remove the occluded ions and dried in a N_2 stream. After that, the dried

samples were ultrasonicated with DI water until the Ti nanotube-array films were detached from the Ti substrate. Then, the Ti nanotube-array films were adhered onto both sides of ITO glass with a drop of $TiO₂$ sol containing Ti(OBu)4 and polyethylene glycol and then annealed at 450 °C for 1 h in air to form crystalline anatase $TiO₂$ phase, which is similar to that described by Kuang et al. [\[25\]](#page-6-0). After annealing, the as-prepared TNT-array films on double-sided ITO glass substrates (i.e., double-sided TNT/ITO films) were naturally cooled down to room temperature in air. For comparison, the TNT-arrays on Ti substrates (i.e., TNT/Ti films) for the single-sided opaque $CdS(n)/TNT/Ti$ electrodes were also fabricated by the same process.

Synthesis of doubled-sided CdS/TNT/ITO and CdS/ TNT/Ti electrodes

Doubled-sided CdS-sensitized TNT/ITO and CdS/TNT/Ti electrodes were synthesized by sequential CBD, as described in our previous papers [[10,](#page-6-0) [24\]](#page-6-0). Typically, one CBD process was performed at room temperature by dipping the crystallized doubled-sided TNT/ITO electrodes in a 0.05 M $Na₂S$ aqueous solution for 1 min, rinsing it with DI water, and then dipped in a 0.05 M CdCl₂ aqueous solution for 1 min, followed by washing it again with DI water. The doubled-sided CdS-sensitized TNT/ITO and CdS/TNT/Ti electrodes after n cycles of CdS deposition were denoted as $CdS(n)/TNT/ITO$ and $CdS(n)/TNT/Ti$, respectively. In this study, the cycle number $n = 5$, 10, 15, and 20.

Characterization

The surface morphology and structure of the as-prepared electrodes were characterized using a scanning electron microscope (SEM) (JSM-7001F, Japan Electron Optics Laboratory Co., Ltd., Tokyo, Japan) and power X-ray diffractometry (XRD) (DX-2500; Dandong Fangyuan Instrument Co., Ltd., Dandong, China), respectively. The UV–Vis absorbance spectroscopy was obtained from a UV–Vis spectrophotometer (UV-2550, Shimadzu Corporation, Kyoto, Japan). The current–voltage characteristics of as-prepared samples were measured in a 250-mL quartz cell with an electrochemical workstation (CHI660E, Shanghai Chenhua Instruments Co., Ltd., Shanghai, China) using a three-electrode configuration with the as-prepared samples as working electrode, a Pt foil counter electrode and a saturated Ag/AgCl reference electrode under simulated AM 1.5G illumination (100 mW cm^{-2}) provided by a solar simulator equipped with a 500 W Xe lamp. The electrolyte was $1.0 M Na₂S$ aqueous solution. The photocurrent responses of the working electrodes with a surface area of 0.04 cm^{-2} were recorded during a voltage sweep from -1.3 to 0 V.

Fig. 2 SEM images showing a top view, b bottom view, and c crosssectional view of free-standing $TiO₂$ nanotube arrays

Results and discussion

Figure 2a shows a typical surface SEM image of as-prepared free-standing crystallized TNT membrane, which shows that the TNT arrays still retain a regularly arranged pore structure after they are detached from the Ti substrate. The average diameter of the TNTs is about 142 nm and the

thickness of the wall of the TNTs is about 12 nm. Figure [2](#page-2-0)b shows the bottom view of the free-standing TNT membrane. It can be seen that the TNTs were packed in approximately hexagonal symmetry. Figure [2](#page-2-0)c shows the cross-sectional view of the TNT arrays showing that the TNTs of about 26 µm in length are compact and stand vertically.

Figure 3a shows the surface SEM image of the TNT/ITO film covered by CdS QDs after 15 CBD cycles (i.e., CdS(15)/ TNT/ITO film). For a comparison, the surface SEM image of the CdS(10)/TNT/ITO film is shown in the inset to Fig. 3a. It can be found after 10 cycles of CBD process, individual CdS QDs with average diameters of \sim 8 nm have been assembled uniformly onto the TNTs and the aggregation of CdS QDs is not obvious. Besides, the inner diameter of the TNT decreased slightly due to the deposition of CdS QDs on the nanotube wall. Similar phenomenon is also reported by Lin et al. [[26](#page-6-0)]. However, for the CdS(15)/TNT/ITO electrode, it can be found that a large ratio of TNTs is covered by the aggregation of the CdS QDs when the CBD cycle number n increases up to 15 and the size of CdS nanoparticle increases with the cycle number n , which is consistent with the previously reported results [\[10](#page-6-0), [26](#page-6-0)].

Figure 3b clearly displays the cross-section of the CdS(15)/TNT/ITO film, which shows a structure of three layers: the top layer of CdS-sensitized TNT arrays, the middle layer of TiO₂ film with a thickness of about 1 μ m adhered onto ITO glass, and the bottom layer of ITO substrate. The crystallinity of the CdS(15)/TNT/ITO film was characterized by XRD, as shown in Fig. 3c. The diffraction peaks of $TiO₂$ (marked with T) are clearly observed, corresponding to an anatase $TiO₂$ (JCPDS 84-1286) [[27\]](#page-6-0). The observed diffraction peaks of CdS (marked with C) can be referred to a hexagonal CdS (1 0 1), (2 2 0), which is in agreement with our previous results [[10,](#page-6-0) [24](#page-6-0)].

The UV–Vis absorption spectra of the double-sided CdS QDs-sensitized TNT/ITO films, i.e., CdS(n)/TNT/ITO films $(n = 0, 5, 10, \text{ and } 15)$ are shown in Fig. [4a](#page-4-0). It can be seen that, the plain double-sided TNT/ITO film absorbs mainly the light with wavelengths smaller than 400 nm, which is similar to the case of TNT film on Ti substrate [[26,](#page-6-0) [28](#page-6-0)]. After CdS deposition, the absorbance of the spectra of the $CdS(n)/TNT/ITO$ films increase significantly in the UV region as well as the visible region, which is mainly due to the light absorption of deposited CdS QDs. In addition, the absorbance of the spectra of the $CdS(n)/TNT/ITO$ films $(n = 5$ and 10) increase with the cycle number *n*, which is due to an increased adsorption amount of CdS QDs. For the CdS(15)/TNT/ITO film after 15 CBD cycles, the light absorption property is further improved in the 483–600 nm wavelength region. However, compared to the CdS(10)/ TNT/ITO film, the absorbance of the spectra of the CdS(15)/TNT/ITO film in the 350-483 nm wavelength

 20 $\overline{30}$ 40 $\overline{50}$ 60 70 80 Two-Theta (deg)

Fig. 3 a Top view and b cross-sectional SEM images of the CdS(15)/ TNT/ITO film. c The XRD pattern of the free-standing double-sided CdS(15)/TNT film. The surface SEM image of the CdS(10)/TNT/ITO film is shown in the inset to Fig. 3a

region decreases slightly. For the $CdS(10)/CuInS_2(2)/TNTs$ film [[10\]](#page-6-0), similar phenomenon has also been observed. This phenomenon is probably due to the quantum size effect. The size of the individual CdS QD will increase with the increasing of cycle number, which might decrease the band gap of CdS QD due to gradually increased CdS particle size.

Fig. 4 UV–Vis absorption spectrum of a the plain TNT/ITO, doublesided CdS(n)/TNT/ITO films ($n = 5$, 10, and 15) and **b** the doublesided CdS(10)/TNT/ITO film compare to the single-sided CdS(10)/ TNT/IT film

The decreased band gap of the CdS QDs would inevitably lead to a red shit of the absorption of the CdS(15)/TNT/ITO film and result in an increased light absorption in the visible region (\600 nm) and a decreased light absorption in the UV region. It can be clearly seen from Fig. 4a that the spectral photoresponse of the CdS(15)/TNT/ITO film is extended to about 600 nm. In fact, due to the quantum size effect, the significant redshift of the spectral photoresponse can be observed with the increase in the cycle number n not only for the CdS(15)/TNT/ITO film but also for the CdS(n)/ TNT/ITO films ($n = 5$ and 10), as shown in Fig. 4a, which is similar to previously reported results [[24,](#page-6-0) [29](#page-6-0)]. For example, the spectral photoresponse of the CdS(5)/TNT/ ITO, CdS(10)/TNT/ITO films is extended to about 525 and 550 nm, respectively.

In addition, compared to the plain TNT/ITO film, a slight increase in the light absorption of the $CdS(n)/TNT/$ ITO films ($n = 5$, 10, and 15) films in near-infrared (NIR) region (wavelength >700 nm) is observed in Fig. 4a, which may be probably due to the electron coupling between $TiO₂$ and CdS heterostructure [[30,](#page-6-0) [31\]](#page-6-0). After the CdS QDs were deposited on $TiO₂$ film, a heterojunction is formed between the $TiO₂$ nanoparticle and CdS quantum dot, which may make the band gap of the $TiO₂/CdS$ composite system different from that of $TiO₂$ or CdS due to the electron coupling between $TiO₂$ and CdS heterostructure. When the band gap of the $TiO₂/CdS$ composite system is even smaller than that of bulk CdS, it is possible that the $TiO₂/CdS$ film has a weak absorption in NIR region.

For a comparison, the absorption spectra of the singlesided opaque $C dS(n) / TNT/Ti$ films were also measured and used to compare with that of double-sided $CdS(n)/TNT/$ ITO film. As an example, Fig. 4b shows the absorption spectrum of the CdS(10)/TNT/Ti and CdS(10)/TNT/ITO films. It can be found that the absorption edge of both the as-prepared CdS(10)/TNT/Ti and CdS(10)/TNT/ITO films showed no obvious difference, which appears at about 550 nm. However, the absorbance of the spectra of the double-sided CdS(10)/TNT/ITO film are significantly enhanced compared with the CdS(10)/TNT/Ti film. Apparently, this phenomenon can be attributed to the enhanced light harvesting capability of double-sided CdSsensitized TNT/ITO films compared with the single-sided CdS-sensitized TNT/Ti film due to an increased amount of CdS nanoparticles, which shows the potential of this doubled-sided structure in improving the current density of the CdS-sensitized TNT photoelectrode due to the increased light absorption.

To study the photoelectrochemical properties of the plain TNT/ITO and double-sided $CdS(n)/TNT/ITO$ photoelectrodes ($n = 5$, 10, 15, and 20), photoelectrochemical measurements were performed versus a saturated Ag/AgCl electrode under AM 1.5 G (100 m W cm^{-2}) illumination. Figure [5](#page-5-0) shows the $J-V$ characteristics of the CdS(n)/TNT/ ITO electrodes. It can be observed that the dark current density of the plain TNT/ITO electrode is negligible. Under AM 1.5 G (100 m W cm^{-2}) illumination, all electrodes showed pronounced photoresponse. The saturated photocurrent density of the plain TNT/ITO electrode is about 0.36 mA cm^{-2} , which is comparable with that of the plain TNT/Ti electrode [[24\]](#page-6-0). The open circuit voltage (V_{oc}) of the plain TNT/ITO electrode is about 1.0 V. After the deposition of CdS QDs, the open circuit voltage of the $CdS(n)/TNT/ITO$ photoelectrodes does not change obviously. However, the photocurrent density of the $CdS(n)/$ TNT/ITO photoelectrode is enhanced substantially compared to the plain TNT/ITO photoelectrode, which can be attributed to the improved visible light absorption by the CdS QDs, as shown in Fig. 4a. Moreover, the photocurrent density of the CdS(n)/TNT/ITO photoelectrode increases markedly with the cycles of CdS deposition $(n<15)$, resulting from an higher incorporated amount of CdS that can induce a higher photocurrent density. However, when the cycle number *n* increases further $(n > 15)$, the

Fig. 5 J–V characteristics of the plain TNT/ITO and double-sided $CdS(n)/TNT/ITO$ photoelectrodes ($n = 5$, 10, and 15)

photocurrent density for an applied potential decreases, which may be explained by the fact that the light absorption of CdS QDs in the nanotubes would be less effective due to the aggregation of CdS QDs on the surface of the TNT film. This explanation can be supported by Fig. [3](#page-3-0)a. A similar result was also observed in previous TNT/Ti-based CdS QDSCs [[24,](#page-6-0) [28\]](#page-6-0). The highest photocurrent density at short-circuit $(J_{\rm sc})$ of 6.95 mA cm⁻² is achieved for the CdS(15)/TNT/ITO photoelectrode, which is much larger than the previously reported value (about 4.8 mA cm^{-2}) of the single-sided opaque TNT/Ti-based CdS QDSCs [[24\]](#page-6-0) and the highest $J_{\rm sc}$ of about 6.0 mA cm⁻² for the doublesided CdS/CdSe QDs-sensitized ZnO nanowire arrays [[3\]](#page-6-0).

The photoconversion efficiency η of the double-sided $CdS(n)/TNT/ITO$ photoelectrodes ($n = 5$, 10, 15, and 20) as a function of applied potential (vs. Ag/AgCl) is calcu-lated as [[32\]](#page-6-0) η % = [(total power output-electric power input)/light power input] \times 100 % = j_p [($E_{\text{rev}}-|E_{\text{app}}|$)/ $I_0 \propto 100$, where j_p is the photocurrent density (mA cm⁻²), E_{rev} (1.23 V/NHE) is the standard state-reversible potential, E_{app} is the applied potential, and I_0 is the power density of incident light (mW cm⁻²). In addition, E_{app} equals the differences between E_{means} and E_{acc} (i.e., $E_{\text{app}} = E_{\text{means}} - E_{\text{acc}}$), where E_{means} is the electrode potential (vs. Ag/AgCl) of the working electrode at which photocurrent was measured under illumination and E_{acc} is the electrode potential (vs. Ag/ AgCl) of the same working electrode under open circuit conditions, under the same illumination and in the same electrolyte. Our calculation result shows that the doublesided CdS(15)/TNT/ITO photoelectrode demonstrates the best PV performance. Only the calculation result for the double-sided CdS(15)/TNT/ITO photoelectrode is shown in Fig. 6 for clarity. It clearly shows that a maximum photoconversion efficiency of about 7.5 % was achieved at an applied potential of -0.73 V versus Ag/AgCl, which is about 120 % higher than the highest efficiency (3.3 %) of the

Fig. 6 The photoconversion efficiencies of the double-sided CdS(15)/TNT/ITO electrode

single-sided opaque TNT/Ti-based CdS QDSCs [[10\]](#page-6-0) and far higher than that $(\leq 2 \%)$ of the QDSCs based on the CdSsensitized mesoscopic $TiO₂$ films [\[33](#page-6-0), [34](#page-6-0)]. The largely enhanced photoconversion efficiency clearly demonstrates the advantage of the transparent TNT/ITO photoelectrodes with double-sided structure compared to the opaque TNT/Ti photoelectrodes with single-sided structure, indicating that the obtained double-sided photoelectrode has potential application in QDSCs and photocatalysis.

Conclusions

In conclusion, double-sided CdS-sensitized TNTs arrayed photoelectrodes were successfully fabricated via ultrasonic treatment of anodized TNT/Ti foil, adhesion of free-standing TNT membrane to the double-sided ITO glass using a drop of a mixture solution of Ti $(OBu)_{4}$ and polyethylene glycol. In comparison to single-sided opaque TNT/Ti-based electrode, this double-sided sensitized photoelectrode shows enhanced light absorption. Moreover, an optimum conversion efficiency of 7.5 % can be achieved by the double-sided CdS(15)/TNT/ITO photoelectrode, which is about 120 % higher than the highest efficiency of the single-sided CdS/ TNT/Ti photoelectrode. The results demonstrate that the photoelectric conversion efficiency of QDSCs can be improved by designing a new photoelectrode structure.

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