

Dielectric properties of $\text{BaTiO}_3\text{--Bi}(\text{Zn}_{1/2}\text{Ti}_{1/2})\text{O}_3\text{--NaNbO}_3$ solid solutions

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Abstract In order to develop dielectric ceramics with temperature-stable permittivity characteristics, perovskite $\text{BaTiO}_3\text{--Bi}(\text{Zn}_{1/2}\text{Ti}_{1/2})\text{O}_3\text{--NaNbO}_3$ ceramic solid solutions were investigated with a particular focus on effects of BaTiO_3 and NaNbO_3 contents on the dielectric properties of ternary compounds. Keeping the ratios of the other two constituents constant, decreasing the BaTiO_3 content leads to a broadening of the temperature-dependent permittivity maximum and a decrease in the overall permittivity. For compositions of constant BaTiO_3 content, replacing $\text{Bi}(\text{Zn}_{1/2}\text{Ti}_{1/2})\text{O}_3$ with NaNbO_3 shifts the temperature of the maximum permittivity to lower temperatures (e.g., to -103°C for a composition of 70BT–5BZT–25NN) while maintaining a broad permittivity peak with temperature, which for the 50BT–25BZT–25NN composition also satisfies the X9R standard. Thus, the investigation of BT–BZT–NN compounds resulted in promising dielectric properties with broad temperature ranges of high permittivity, which is of interest for advanced capacitor applications.

Introduction

Perovskite materials are known to be of interest for a wide variety of electroceramic applications. Various groups of

materials have recently been investigated including lead-based [1–3], non-lead-based [4], and bismuth-based perovskite [5]. In capacitor technologies, high temperature-stable permittivity, low dielectric loss, and high insulation resistance are desirable. Solid solutions of $\text{BaTiO}_3\text{--BiMO}_3$ and $\text{PbTiO}_3\text{--BiMO}_3$, in which the BiMO_3 compounds are known to be unstable in the perovskite structure under ambient conditions, have recently received a great deal of interest because of their attractive dielectric and piezoelectric properties, where M represents $\text{Zn}_{1/2}\text{Ti}_{1/2}$ [6, 7], $\text{Mg}_{1/2}\text{Ti}_{1/2}$ [8–11], $\text{Ni}_{1/2}\text{Ti}_{1/2}$ [12, 13], and Sc [11, 14]. Of particular interest to this study, it has been shown that BiMO_3 additions transform the characteristic sharp phase transition—and associated permittivity maximum—of pure BaTiO_3 into a less-distinct phase transition with a broad and strongly dispersive permittivity maximum characteristic of relaxor dielectrics. In addition, the temperature of maximum permittivity (T_{max}) decreased significantly (to below room temperature) with low concentrations of BiMO_3 and then increased, to higher than room temperature, as the BiMO_3 content increased [6, 8, 9, 12, 14]. This phenomenon has been observed both in the pseudo-binary systems mentioned earlier as well as in pseudo-ternary systems such as $\text{BaTiO}_3\text{--Bi}(\text{Zn}_{1/2}\text{Ti}_{1/2})\text{O}_3\text{--BiInO}_3$ [15] and $\text{BaTiO}_3\text{--Bi}(\text{Zn}_{1/2}\text{Ti}_{1/2})\text{O}_3\text{--BiScO}_3$ (BT–BZT–BS) [16, 17]. The latter compound, BT–BZT–BS, showed excellent dielectric and electrical properties for high-temperature capacitor applications exhibiting a flat temperature dependent of relative permittivity over a broad temperature range. However, the large temperature and frequency dependence of permittivity at temperatures below $\sim 100^\circ\text{C}$ are undesirable for the use of these materials in applications requiring ambient operation.

In order to decrease T_{max} to temperatures well below room temperature and to improve the temperature stability

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of the relative permittivity of $\text{BaTiO}_3\text{--Bi}(\text{Zn}_{1/2}\text{Ti}_{1/2})\text{O}_3$ -based ceramics, sodium niobate (NaNbO_3) was selected as an end-member for the ternary system. NaNbO_3 has the perovskite structure and exhibits both orthorhombic symmetry and antiferroelectricity at room temperature [18]. Complex phase transitions driven by temperature were also seen in this material [19]. It has also been known as the host materials for developing high-performance lead-free piezoelectric $(\text{K}_{1/2}\text{Na}_{1/2})\text{NbO}_3$ [20, 21].

NaNbO_3 has been shown to form a complete substitutional solid solution with BaTiO_3 [22]. While the end members of this $\text{BaTiO}_3\text{--NaNbO}_3$ solid solution have Curie temperatures (and associated permittivity maxima) at 126 and 347 °C, respectively, compositions containing 20–30 mol% NaNbO_3 have been shown to have T_{max} values below -73 °C [22]. In addition, a single-phase perovskite could be formed in $\text{NaNbO}_3\text{--Bi}(\text{Zn}_{1/2}\text{Ti}_{1/2})\text{O}_3$ solid solutions when the concentration of $\text{Bi}(\text{Zn}_{1/2}\text{Ti}_{1/2})\text{O}_3$ is no more than 10 mol% [23]. Low additions of $\text{Bi}(\text{Zn}_{1/2}\text{Ti}_{1/2})\text{O}_3$ (BZT) also transformed the antiferroelectric behavior of pure NaNbO_3 (NN) to relaxor ferroelectric behavior. It should be also noted that the multiple phase transitions found in pure NaNbO_3 transformed to a diffuse phase transition for NN–BZT compositions close to the solubility limit. Importantly, compositions containing between 7 and 10 mol% BZT exhibited T_{max} values below -50 °C.

Thus, in order to develop dielectric material with a low transition temperature (T_{max}), the $\text{BaTiO}_3\text{--Bi}(\text{Zn}_{1/2}\text{Ti}_{1/2})\text{O}_3\text{--NaNbO}_3$ solid solutions were investigated. Two groups of compositions were studied (Fig. 1) as follows;

- 1) $x\text{BaTiO}_3\text{--}(100 - x)(0.5\text{Bi}(\text{Zn}_{1/2}\text{Ti}_{1/2})\text{O}_3\text{--}0.5\text{NaNbO}_3)$, where $x = 50\text{--}80$.
- 2) $70\text{BaTiO}_3\text{--}(30 - y)\text{Bi}(\text{Zn}_{1/2}\text{Ti}_{1/2})\text{O}_3\text{--}y\text{NaNbO}_3$, where $y = 5\text{--}25$.

The dielectric properties and polarization hysteresis behaviors of these solid solutions are demonstrated here.

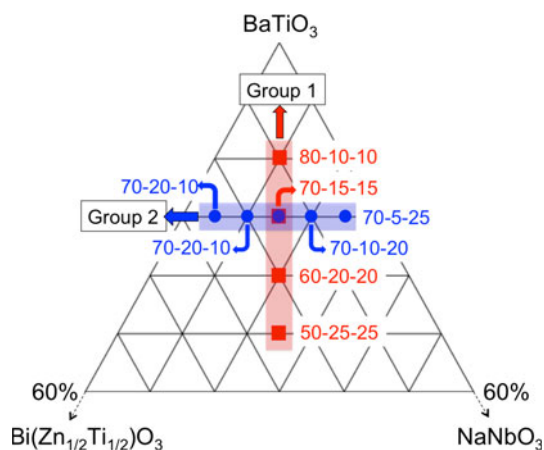


Fig. 1 Ternary diagram showing specific compositions investigated

Experimental procedure

A solid-state reaction method was used to prepare the $\text{BaTiO}_3\text{--Bi}(\text{Zn}_{1/2}\text{Ti}_{1/2})\text{O}_3\text{--NaNbO}_3$ solid solutions from BaCO_3 (99.8 %), Bi_2O_3 (99.9 %), Na_2CO_3 (99.5 %), ZnO (99.9 %), Nb_2O_5 (99.9 %), and TiO_2 (99.0 %) precursors. Aqueous mixtures of batched powders were mixed and ground using a vibratory milling machine for 6 h and then dried at 100 °C overnight. A calcination process was carried out at 900–1000 °C for 4 h in covered alumina crucibles. The calcined powders were ground again before pressing into disk shape by uniaxial pressing. Green pellets were sintered on top of powders of the same composition in covered alumina crucibles in static air at 1120–1220 °C for 4 h in a box furnace with ramp rates of ± 3 °C/min. Formation of a single perovskite phase was confirmed using a laboratory X-ray diffractometer (Bruker AXS D8 Discover, $\text{Cu K}\alpha$). The surfaces of the resulting dense ceramics were polished before applying silver electrodes which were fired on at 700 °C for 10 min. The sample dimensions for electrical measurements were ~ 0.80 mm in thickness and ~ 10.60 mm in diameter. The temperature dependence of dielectric properties was measured on cooling from 200 to -150 °C using an Agilent 4284A LCR meter and a custom sample holder. Polarization-field behavior was investigated at room temperature using a Precision Premier II ferroelectric test system (Radiant Technologies).

Results and discussion

The results presented in this section are divided into two groups according to the composition series previously mentioned. The ternary diagram shown in Fig. 1 identifies the specific composition in this study.

Group 1 consists of $x\text{BaTiO}_3\text{--}(100 - x)(0.5\text{Bi}(\text{Zn}_{1/2}\text{Ti}_{1/2})\text{O}_3\text{--}0.5\text{NaNbO}_3)$, where $x = 50\text{--}80$; in other words, the ratio of $\text{Bi}(\text{Zn}_{1/2}\text{Ti}_{1/2})\text{O}_3\text{:NaNbO}_3$ was fixed at 1:1, while the BaTiO_3 content was varied from 50 to 80 mol%. X-ray diffraction patterns from the group 1 compositions are shown in Fig. 2. All compositions were observed to exhibit a single perovskite phase. Careful investigation of the diffraction peaks showed no evidence of distortions or asymmetry, which suggests the existence of cubic symmetry. Lattice parameters were calculated by Cohen's method and showed an increasing trend with BaTiO_3 content. However, at $x = 60$ and 70, the lattice parameter remained nearly constant.

Figure 3a shows temperature-dependent dielectric data measured at 1 and 10 kHz. A strong frequency dependence of relative permittivity is clearly seen at temperatures below T_{max} for all compositions, but dispersion above T_{max} is negligible. This behavior is consistent with the relaxor-like behavior

Fig. 2 X-ray diffraction patterns and lattice parameter of the $x\text{BaTiO}_3-(100-x)(0.5\text{Bi}(\text{Zn}_{1/2}\text{Ti}_{1/2})\text{O}_3-0.5\text{NaNbO}_3)$ compositions, where $x = 50-80$

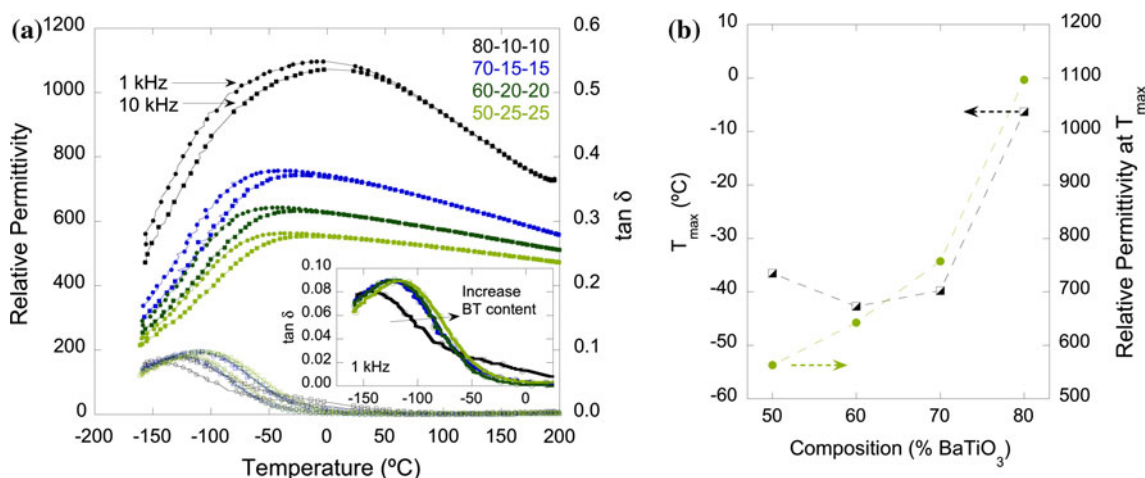
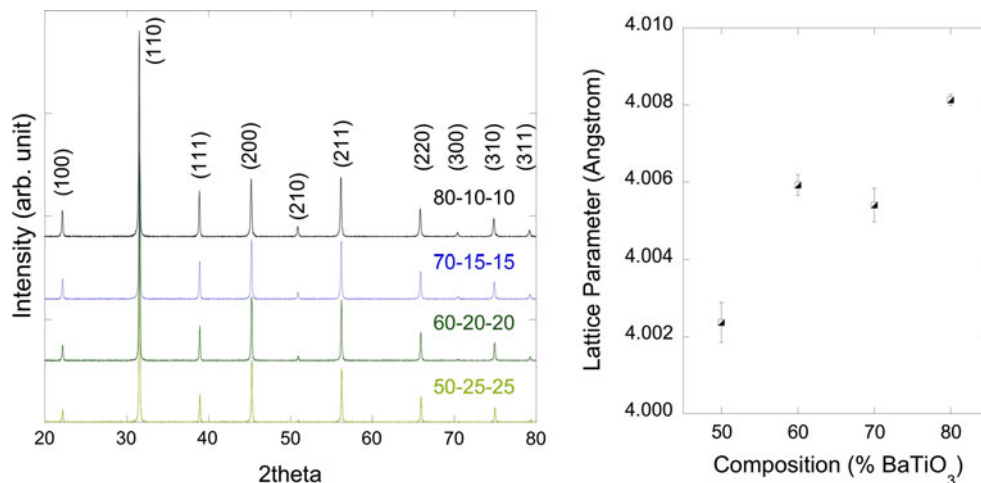


Fig. 3 **a** Temperature dependence of relative permittivity and $\tan \delta$ at 1 kHz and 10 kHz. **b** Composition dependence of T_{\max} and $\epsilon_{r,\max}$ (1 kHz) for $x\text{BaTiO}_3-(100-x)(0.5\text{Bi}(\text{Zn}_{1/2}\text{Ti}_{1/2})\text{O}_3-0.5\text{NaNbO}_3)$

compositions where $x = 50-80$. Inset shows “zoom-in” $\tan \delta$ data (1 kHz) at low temperature

observed in similar systems [6, 14, 15]. Interestingly, as shown in Fig. 3b, while the maximum relative permittivity ($\epsilon_{r,\max}$) decreased monotonically with decreasing BaTiO₃ content between $x = 80$ mol% and $x = 50$ mol%, T_{\max} initially decreased from -6 to -40 °C as the BaTiO₃ content decreased from 80 to 70 mol% but remained nearly unchanged for BaTiO₃ values from 50 to 70 mol%. It should be mentioned that $\tan \delta$ at 1 kHz remained below 0.05 over a wide temperature range from approximately -70 to 350 °C. Over the temperature range of 0–200 °C, where the relative permittivity is not frequency dispersive, a decreased BaTiO₃ content improved the temperature stability of relative permittivity. The temperature coefficients of permittivity ($\text{TC}\epsilon$) varied from -1960 ppm/K for the 80:10:10 composition to -790 ppm/K for the 50:25:25 composition.

Polarization hysteresis measurements collected at room temperature show slim hysteresis loops with negligible remanent polarization for all compositions (Fig. 4). While

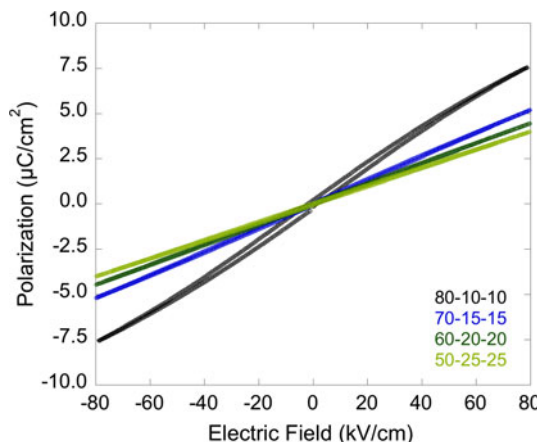


Fig. 4 Room temperature polarization hysteresis measured at 10 Hz of the $x\text{BaTiO}_3-(100-x)(0.5\text{Bi}(\text{Zn}_{1/2}\text{Ti}_{1/2})\text{O}_3-0.5\text{NaNbO}_3)$ compositions, where $x = 50-80$

the 80BT–10BZT–10NN composition showed pseudo-linear dielectric response with slight saturation at high fields, all other compositions remained linear across the measured fields. The maximum polarization (P_{\max}) exhibited a decreasing trend as the BaTiO_3 content decreased, similar to the relative permittivity values.

The results shown in this section demonstrate the role of BaTiO_3 content on dielectric properties of the BaTiO_3 – $\text{Bi}(\text{Zn}_{1/2}\text{Ti}_{1/2})\text{O}_3$ – NaNbO_3 ternary system. The compositions containing 50, 60, and 70 % of BaTiO_3 exhibited similar dielectric characteristics in which higher relative permittivity for the 70BT–15BZT–15NN composition was observed as compared to that of the 60BT–20BZT–20NN and the 50BT–25BZT–25NN compositions. Therefore, the composition containing 70 % of BaTiO_3 was selected to further examine the role of NaNbO_3 in this ternary system.

Group 2 compositions consist of 70BaTiO_3 – $(30 - y)\text{Bi}(\text{Zn}_{1/2}\text{Ti}_{1/2})\text{O}_3$ – $y\text{NaNbO}_3$ where $y = 5$ –25; in other words, the BaTiO_3 content was kept constant while the

BZT:NN ratio was varied (Fig. 1). As shown in Fig. 5, diffraction patterns from all compositions were consistent with a single cubic perovskite phase. A decrease in lattice parameter was observed as y increased (substitution of NaNbO_3 for $\text{Bi}(\text{Zn}_{1/2}\text{Ti}_{1/2})\text{O}_3$) which could be due to a decrease in the amount of the larger Zn^{2+} ions (0.740 Å) relative to Ti^{4+} (0.605 Å) and Nb^{5+} (0.640 Å).

The temperature dependence of the relative permittivity and $\tan \delta$ of all group 2 compositions are shown in Fig. 6. With a fixed concentration of BaTiO_3 (70 mol%), it could be seen that the dielectric characteristics were manipulated via substitution of NaNbO_3 for $\text{Bi}(\text{Zn}_{1/2}\text{Ti}_{1/2})\text{O}_3$. As the NaNbO_3 content increased from 5 to 25 mol%, the T_{\max} decreased from 42 to -103 °C, as shown in Fig. 6. The permittivity at T_{\max} , however, decreased with increasing NaNbO_3 substitution up to 20 mol%, and then increased when the NaNbO_3 further increased to 25 mol%. Previous studies of BaTiO_3 – NaNbO_3 solid solutions showed that a 70BaTiO_3 – 30NaNbO_3 composition exhibited a minimum

Fig. 5 X-ray diffraction patterns and lattice parameter of the 70BaTiO_3 – $(30 - y)\text{Bi}(\text{Zn}_{1/2}\text{Ti}_{1/2})\text{O}_3$ – $y\text{NaNbO}_3$ compositions, where $y = 5$ –25

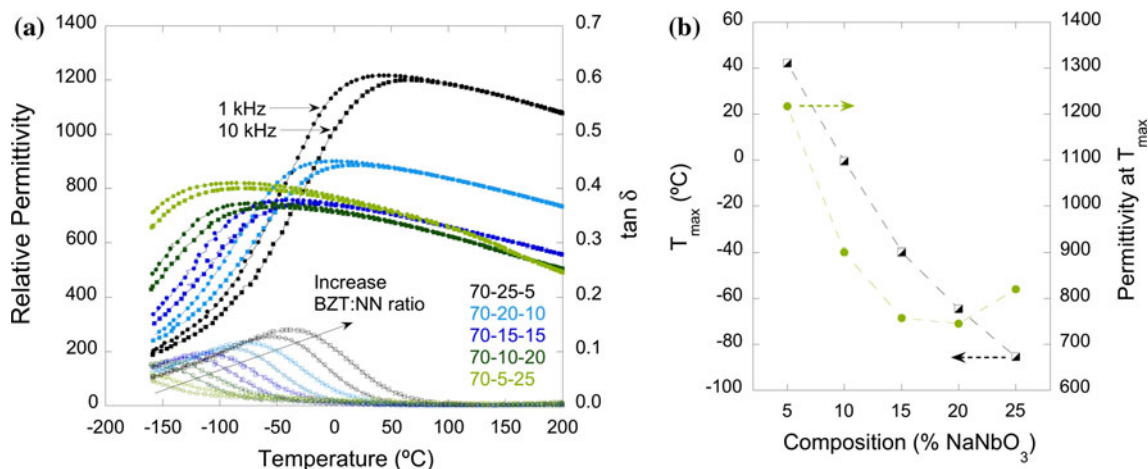
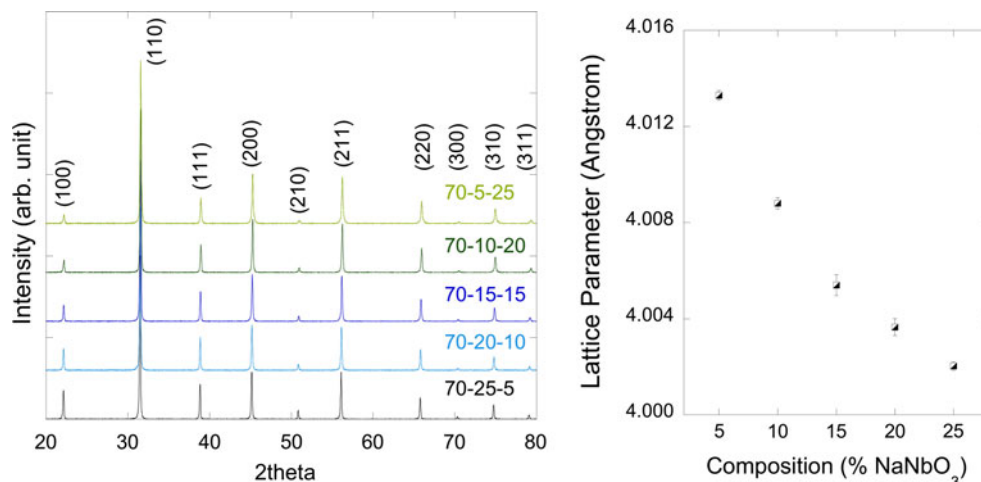


Fig. 6 **a** Temperature dependence of relative permittivity and $\tan \delta$ at 1 and 10 kHz. **b** Composition dependence of T_{\max} and $\epsilon_{r,\max}$ (1 kHz) for 70BaTiO_3 – $(30 - y)\text{Bi}(\text{Zn}_{1/2}\text{Ti}_{1/2})\text{O}_3$ – $y\text{NaNbO}_3$ compositions where $y = 5$ –25

Table 1 Summary of dielectric properties of BaTiO₃–Bi(Zn_{1/2}Ti_{1/2})O₃–NaNbO₃ solid solutions

Composition	T_{\max} (°C)	Room temperature		300 °C		TC ϵ (ppm/K)
		ϵ_r	$\tan \delta$	ϵ_r	$\tan \delta$	
Group 1						
80:10:10	–6	1080	0.008	580	0.014	–1960
70:15:15	–40	730	0.002	510	0.009	–1390
60:20:20	–43	620	0.001	490	0.008	–1010
50:25:25	–37	550	0.003	460	0.006	–790
Group 2						
70:25:5	42	1210	0.023	980	0.006	–930
70:20:10	–1	890	0.004	660	0.006	–990
70:15:15	–40	730	0.002	510	0.009	–1390
70:10:20	–65	700	0.004	450	0.013	–1700
70:5:25	–103	680	0.004	390	0.031	–2230

Data were taken at a measurement frequency of 1 kHz

T_{\max} (about –123 °C) with an associated permittivity of more than 3000 at 1 kHz [22]. A degradation of the temperature stability of the relative permittivity was observed as the NaNbO₃ content increased in which the temperature coefficient of relative permittivity (TC ϵ) increased from –990 ppm/K for the 70:25:5 composition to –2230 ppm/K for the 70:5:25 composition. A summary of the dielectric properties is listed in Table 1, which includes high-temperature dielectric data recorded at 300 °C.

Polarization versus electric field behavior measured at room temperature revealed a linear dielectric response with minimal hysteresis and remanence, as shown in Fig. 7. The maximum polarization (P_{\max}) showed a similar trend as seen in the temperature dependence of dielectric properties data. In other words, P_{\max} decreased with an increased of

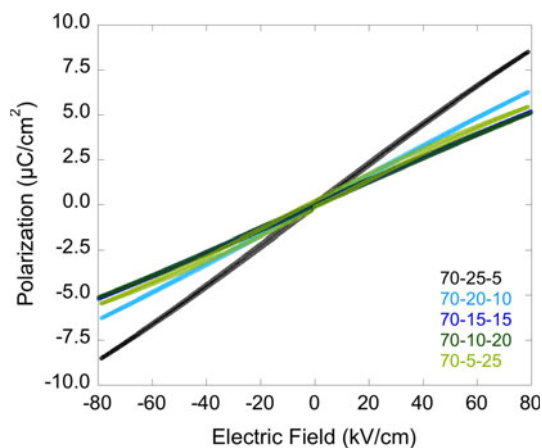


Fig. 7 Room temperature polarization hysteresis measured at 10 Hz of the 70BaTiO₃–(30 – y)Bi(Zn_{1/2}Ti_{1/2})O₃–yNaNbO₃ compositions, where y = 5–25

NaNbO₃ up to 20mol% then increased when the NaNbO₃ increased to 25 mol%.

Altogether, all compositions studied here exhibited broad permittivity peaks with temperature and significant frequency dispersion below T_{\max} , consistent with the previous study on BaTiO₃–BiMO₃ solid solutions, where M represents Zn_{1/2}Ti_{1/2} [6], Mg_{1/2}Ti_{1/2} [8, 9], Ni_{1/2}Ti_{1/2} [12], or Sc [14]. The addition of NaNbO₃ to BaTiO₃–Bi(Zn_{1/2}Ti_{1/2})O₃ further modified the dielectric characteristics, generally decreasing both T_{\max} and $\epsilon_{r,\max}$. Although the compositions with high NaNbO₃ content shifted T_{\max} to low temperature, the temperature stability of relative permittivity was not improved (group 2 compositions) but degraded. By having equal amounts of BZT and NN in the solid solutions (group 1 compositions), the temperature stability of relative permittivity could be improved while still maintaining T_{\max} values well below room temperature. Overall, the optimum composition, 50BT–25BZT–25NN, exhibited comparable performance to the temperature stable BT–BZT–BS composition previously mentioned in the introduction [17]. Compared to BT–BZT–BS, the BT–BZT–NN material exhibited a lower relative permittivity and had higher temperature coefficients. However, BT–BZT–NN exhibited a broad dielectric maximum that spanned from below room temperature and included less expensive constituents than BT–BZT–BS. In fact, the 50BT–25BZT–25NN composition also exceeds the EIA X9R standard ($\Delta C/C_{25} = \pm 15\%$ ranging from –55 to 200 °C) where capacitance variation remains within $\pm 15\%$ for temperatures ranging from –100 to +200 °C. These key advantages suggest that BT–BZT–NN is another promising candidate for advanced capacitor applications.

Conclusion

In this study, solid solutions in the BaTiO₃–Bi(Zn_{1/2}Ti_{1/2})O₃–NaNbO₃ ternary system were investigated. Compositions containing 50–80 mol% BaTiO₃ exhibit cubic symmetry based on X-ray diffraction data. By increasing the (Bi(Zn_{1/2}Ti_{1/2})O₃–NaNbO₃) concentration, the temperature stability of relative permittivity was improved, the relative permittivity decreased, and T_{\max} remained nearly unchanged. At a composition containing 70 % BaTiO₃, substitution of Bi(Zn_{1/2}Ti_{1/2})O₃ by NaNbO₃ shifted T_{\max} to values as low as –103 °C. The existence of highly broad permittivity peak (satisfy X9R standard), linear dielectric response with electric field, and T_{\max} well below room temperature make the material in this ternary system a promising candidate for a wide operational temperature range dielectrics for capacitor applications.

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