ORIGINAL RESEARCH

Transparent, mechanically robust, low‑temperature‑tolerant, and stretchable ionogels enhanced by konjac glucomannan toward wireless strain sensors

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Abstract Electronic skins (*E*-skins) can detect human health and movement, and have potential in the felds of human–machine interactions and artifcial intelligence. However, traditional hydrogel-based *E*-skins suffer from poor mechanical strength, low conductivity, and instability due to water evaporation. Herein, a semi-interpenetrating network developed by polysaccharide biomass konjac glucomannan (KGM) was introduced into a covalent-crosslinked network polyacrylamide-*co*- polyacrylic acid (PAM*co*-PAA) to advance the above dissatisfaction of *E*-skins. This synthesized a transparent, tough, nonvolatile, and highly stretchable ionogel with an ionic liquid named 1-ethyl-3-methylimidazolium dicyamide (EMIM:DCA) as conductive media. This ionogel exhibited extraordinary mechanical strength (tensile strength of 2.77 MPa), outstanding mechanical

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durability (100 stretching cycles of 250%), and elongation (elongation at break of 997%). More importantly, the ionogel demonstrated remarkable anti-freezing performance (high fexibility at -20℃) and high conductivity (3.94 mS/cm) in the absence of water. Besides, after assembling KGM-enhanced ionogel, the sensor exhibited comprehensive strain sensing performance, which could efectively and accurately monitor human motion via Bluetooth transmission. This strategy paves the way for a viable new generation of multifunctional biomimetic supersensitive sensors, which are promising for applications such as intelligent devices, health detection, and biomedical monitoring in harsh conditions.

Graphical abstract

Keywords Konjac Glucomannan · Ionogels · Strain sensor · Biomass · Wearable device

Introduction

With the rise of fexible electronic devices, synthesizing biomass-based fexible tensile sensors with high sensitivity, high tensile strength, and stable performance has become signifcant for physical health and smart wearable devices (Chen et al. [2021](#page-12-0); Wang et al. [2021](#page-13-0); Zhou et al. [2022](#page-14-0); Zhu et al. [2022\)](#page-14-1). Traditional conductive materials, such as carbon nanotubes and graphene, do not have the capacity for stretchability, and their complex internal structures are difficult to achieve, which also limits the transparency of the material (Hou et al. [2019](#page-12-1); Huang et al. [2023;](#page-12-2) Liao et al. [2017;](#page-13-1) Lipomi et al. [2011](#page-13-2); Pang et al. [2018](#page-13-3)). Conversely, conductive gel has already become a novel material for wearable sensors, with the advantages of easy manufacturing, excellent fexibility, and high biocompatibility.

At present, the conductive gel material substrate is mainly hydrogel, widely applied in health care, biosensors, and human–machine interactions (Chen et al. [2022\)](#page-12-3). Although its fantastic fexibility, biocompatibility, and conductivity are shown in recent research, under the infuence of water evaporation in the hydrogel, its sensitivity, mechanical robustness, stability, and conductivity would dramatically decrease, limiting its extensive use as wearable sensors. Therefore, it is signifcant to manufacture a wearable sensor with high conductivity, non-volatile properties, high stretchability, and high stability (Xue et al. [2023](#page-14-2)).

Ionogels are formed by the polymer network swelling process in the ionic liquid, which has heat stability, high ionic conductivity, electrochemical stability, and non-volatile properties (Huang et al. [2023](#page-12-2); Kim et al. [2023](#page-13-4); Li et al. [2023;](#page-13-5) Wang et al. [2022a,](#page-13-6) [b,](#page-13-7) [c;](#page-13-8) Jiang et al. [2021\)](#page-12-4). Chemical gelling is mainly used for preparation, and other conductive materials are also added to obtain better mechanical and electrochemical properties. Sun et al. ([2021\)](#page-13-9) synthesized a kind of dual-network (DN) ionogel with excellent mechanical properties, high transparency, wide operating temperature range, and strong adhesion prepared by photopolymerization method. The gel synthesis was based on an agar network and formed an interpenetrating network by introducing polyhydroxyethyl acrylate into an ionic solution. The excellent mechanical properties and sensing properties of this ionogel enabled it to be used as a fexible strain sensor to detect human joint movement and pulse. Xiang et al. [\(2022](#page-13-10)) prepared another kind of ionogel material (IL-PILs-IG) by copolymerizing 1-vinyl-3-butylimidazole tetrafuoroboric acid ([VBIm]) and 1-dodecyl-3-tetrafluoroboric acid $([C_{12}VIm][BF_4]$ in non-polymeric ionic liquid 1-butyl-3-methylimidazole ([BMIm] [fuoborate]) in the one-pot method with favorable tensile properties, high transparency (91%), high thermal stability $(>300 \text{ °C})$, and remarkable recovering property. Therefore, the important role of ion-hydrophobic "microdomains" composed of $[C_1, VIm]^+$ long alkyl chains in improving mechanical properties was confrmed. In this regard, ionic liquid (1-ethyl-3-methyl imidazole dihydro amine salt, EMIM:DCA) was applied as a solvent to prepare a biomass-based composite ionogel with high conductivity, non-volatile properties, and high mechanical properties.

Biodegradable biomass materials, such as collagen, cellulose, and chitosan, are receiving increasing attention and are widely used in biomedicine, adsorption and catalysis, energy storage, and fexible electronics (Hajiali et al. [2022;](#page-12-5) Rezaei et al. [2021](#page-13-11); Seabra et al. [2018;](#page-13-12) Strauss and Chmielewski [2016](#page-13-13)). Researchers have used many biopolymer materials to construct fexible strain sensors (Ahmed et al. [2016;](#page-12-6) Kalambate et al. [2020](#page-12-7); Rezaei et al. [2021](#page-13-11)). Konjac glucomannan (KGM) is a watersoluble, neutral, natural heteropolysaccharide with high molecular weight and hydrophilic groups such as active primary hydroxy $(-CH₂OH)$ and acetyl groups (Gao et al. [2022;](#page-12-8) Hu et al. [2019](#page-12-9); Wu et al. [2020](#page-13-14); Zhu [2018\)](#page-14-3). KGM consists of β-D-mannose and β-D-glucose. The molar ratio of glucose to mannose is 1.6:1 and they are connected through $β-1,4$ glycosidic bonds. Moreover, acetyl groups attach to the saccharide units and distribute randomly along the molecule, with an occurrence of 1 per 19 sugar residues at the C (6) position. It is popular in developing biomedical materials and functional food formulations (Cao et al. [2022;](#page-12-10) Zhang et al. [2023](#page-14-4); Wu et al. [2022a,](#page-13-15) [b\)](#page-13-16). Introducing KGM forms a semiinterpenetrating network in the original PAM-*co*-PAA network, and the functional groups enabled it to enhance the mechanical properties of the ionogel. Xu et al. ([2022](#page-14-5)) adopted polyvinylidene fuoride-*co*hexafuoropropylene (PVDF-HFP) as the linear non-cross-linked network in the ionogel, and strain strength increased with the addition of PVDF-HFP (3.67 to 8.76 MPa) due to an increase in cross-linking points. Wang et al. ([2021\)](#page-13-0) involved 2,2,6,6-tetramethylpiperidinyl-1-oxyl-oxidized cellulose nanofbril (TEMPO-CNF) in the hydrogel with covalently cross-linked polyacrylamide and formed a DN structure. Its elongation at break was 1100%, and its tensile strength was 710 kPa. However, there are few studies on applying KGM to fexible ionogelbased sensors. Hence, it was considered that KGM, as a kind of biomass-based polymer polysaccharide material, has the potential to enhance the toughness, stretchability, resilience and fexibility of ionogels. There are few studies on the application of KGM in ionogel-based fexible sensors.

Herein, we introduced KGM polysaccharide biomass into the ionic liquid to prepare a transparent, mechanically robust, and anti-freezing ionogel. First, KGM was dissolved in EMIM:DCA to obtain the KGM/ionic liquid mixture, and then acrylamide (AM) and acrylic acid (AA) were dispersed in the above mixture. After that, the mixture was exposed to ultraviolet (UV) light to form a semi-interpenetrating network-enhanced KGM/PAM-*co*-PAA ionogel. Simultaneously, EMIM:DCA provided high conductivity, non-volatile, and anti-freezing properties due to its various benefcial properties. Importantly, KGM is a biomass-based and environmentally friendly material. Therefore, after mixing and gelling in an EMIM:DCA solution, an ionogel-based *E*-skin was proposed, which had several attributes of being mechanically strong, highly conductive, biocompatible, and highly self-recovering. Notably, this enhanced hydrogen bond efect was proven to endow ionogels with tensile strength, self-recovering properties, and elongation at break. Furthermore, we demonstrated the potential value of the assembled ionogel sensors in diferent application scenarios (fnger bending, neck bending, running, and jumping). It confrmed that our study will help designers and manufacturers create multifunctional wearables with various use cases.

Experiment section

Materials

Konjac Glucomannan (KGM, viscosity≥15,000 mpa.s) was supplied by Shanghai Yuanye Bio-Technology Co., Ltd. (Shanghai China). 2-hydroxy-4'-(2-Hydroxyethoxy)- 2-methylpropiophenone (Irgacure 2959) and *N, N'*-Methylene-bis-acrylamide (MBAA Catalyst) was obtained from Shanghai Titan Science Co., Ltd. (Shanghai China). Acrylamide (AM) and acrylic acid (AA) were supplied from Aladdin Reagent Inc. (Shanghai, China). 1-ethyl-3-methylimidazolium dicyamide (EMIM:DCA, Ionic liquid) was obtained from Monils (Nanjing) Technology Co., Ltd.

Synthesis of PAM-co-PAA ionogel

Firstly, dewatering treatment of raw materials was given to AM and MBAA in a vacuum environment at 120 °C for 2 h. Secondly, 8.0 g EMIM:DCA, 2.0 g AM, 2.0 g AA, 12.0 mg MBAA were mixed and stirred at room temperature for 10 min to get a uniform mixture named IG (KGM content: 0 wt%). Then the photoinitiator was added to the IG solution. After that, the mixture was stirred for 10 min, and then placed in a vacuum to remove the bubbles for 20 min. Finally, it was gelled under UV light for 15 min and the ionogel sample was obtained which was named $IKG-I₀$.

Preparation of Konjac Glucomannan ionogel (Labeled IKG‑Ix ionogel)

The detailed process of synthesizing $IKG-I_x$ is shown in Scheme [1.](#page-4-0) Varied mass of IG solution (12.0 g) , 9.3 g, 6.7 g, 4.0 g) was mixed with varied mass of IKG (KGM content: 6 wt%) solution $(0 \text{ g}, 2.7 \text{ g})$, 5.3 g, 8.0 g), making the total mass of the mixture 12.0 g with diferent KGM contents (0 wt%, 2.0 wt%, 4.0 wt%, and 6.0 wt%), which were denoted as IKG_{0} , IKG_2 , IKG_4 , and IKG_6 . Then the photoinitiator was added to the IKG solutions. After that, the mixed solution was magnetically stirred for 20 min and then disposed of in a vacuum environment for 20 min with a preservative flm covering their bottlenecks to prevent the evaporation of AA. After pouring the solution into PTFE moulds for molding and using UV light to initiate free radical polymerization, ionogel samples were obtained and named IKG-I₀, IKG-I₂, IKG-I₄, and IKG-I₆, respectively.

Characterization

Fourier transform infrared spectroscopy (FT-IR) was carried out with a Nicolet 560 FTIR spectrometer (Nicolet, USA) in the range from 4000 cm^{-1} to 650 cm⁻¹ under a resolution of 4 cm⁻¹ at ambient temperature with a total accumulation of 16 scans to evaluate the chemical structures and elemental components of konjac glucomannan and IKG-I_x. Moreover, the ultraviolet–visible (UV–vis) spectra of $IKG-I_x$ were tested by the Analytik-Jena spectrophotometer S600 (Germany), ranging from 300 cm⁻¹ to 800 cm⁻¹.

Scanning electron microscopy (JEOL TSM-7500F, Japan) was carried out to investigate the morphologies of IKG-I₄ and IG. IKG-I₄ and IG were quenched in nitrogen and lyophilized for 2 days. The samples above were then sputtered with gold to get conductivity.

The mechanical properties of diferent samples were tested by Instron 5967 universal tensile testing machine (Instron Electron Instrument Co. Ltd., USA). To ensure the experiments' reproducibility, tensile tests (cyclic test and single test) were performed on the three fat rectangular dumbbell-shaped samples with 30 mm $(\text{length})\times 2 \text{ mm}$ (thickness) $\times 4 \text{ mm}$ (width), at room temperature, and the measurement rate was 50 mm min^{-1} , respectively. The results were expressed as the average of three readings taken for each measurement.

The rheological properties were recorded on a rheometer (MCR 302, Germany) at 25 ℃ using a parallel plate with a diameter of 25 mm at 1 mm gap. The frequency-sweep test was performed from 0.1 to 100 rad s^{-1} at a fixed strain of 0.1%. The strainsweep was assessed in the strain range of 0.01–100% at a fxed frequency of 10 Hz. Cyclic strain time scanning was measured at a constant frequency of 10 Hz, and time scanning of low strain 0.1% and high strain 100% was carried out alternately. Each test section lasted for 100 s, and a total of 10 test Sects. (5 cycles) were conducted. All of the testing was done at 25 ℃.

The electrochemical workstation (PARSTAT MC, USA) was used to test the electrochemical performance of IG and $IKG-I_x$. Besides, the conductivity of ionogel samples was calculated from Formula [\(1](#page-3-0)):

$$
\sigma = L/(R \times S) \tag{1}
$$

where σ , *L*, *R*, and *S* are the conductivity, thickness, resistance, and cross-sectional area (CSA) of the samples respectively. The experiments' reproducibility

Scheme 1 Schematic of the preparation and application of IKG ionogels

was ensured by testing the electrochemical performance (conductivity and response time) with three fat rectangular specimens.

To measure the real-time relative resistance changes of duplicated samples, they were clamped by copper sheets on both sides as contact electrodes, and the PARSTAT MC, which had already been paired with a cellphone through a Bluetooth device, was connected to both ends of the sample by wires.

The thermal properties of IG and $IKG-I_x$ were analyzed by a diferential scanning calorimeter (DSC, 204F1, NETZSCH, Germany). Samples were implanted in an aluminum pan under a heating–cooling cycle in the range of -50 \degree C to 100 \degree C at 10 \degree C per minute.

In order to study the stability of ionogels in an ambient environment, weight losses of ionogels and hydrogels were carried out by the traditional gravimetric method in triplicate, and the average of three tests is used as the result.

Result and discussion

Synthesis and characterization

In this work, KGM biomass material was frst dispersed in an EMIM:DCA solution to obtain a uniform mixture of KGM and ionic liquid. Thus, IKG-Ix ionogel was prepared by free radical copolymerizing AM and AA monomers in KGM-IG dispersion. At last, the PAM-*co*-PAA covalent cross-linked network bridged the KGM chains through physical entanglement interactions and hydrogen bonding (Wang et al. [2021;](#page-13-0) Wu et al. [2022a,](#page-13-15) [b](#page-13-16); Gao et al. [2023](#page-12-11); Zhu et al. [2020\)](#page-14-6). The snapshot in Fig. [1a](#page-5-0) shows the high transparency of $IKG-I₄$ when applied to a colorful picture beneath. As shown in Fig. [1](#page-5-0)b, the transmittance of IKG-I₀ reached 95%, ranging from 800 to 1000 nm. As the KGM content increased, the transmittance of IKG-I₂, IKG-I₄, and IKG-I₆ was 90%, 84%, and 75%,

respectively. The above results indicate that the KGM matrix had high miscibility and homogeneity with EMIM:DCA. In addition, IKG-I₄ showed excellent transparency (84%), greatly expanding the practical application range of IKG ionogels (Jiang et al. [2021](#page-12-4)).

Fourier-transform infrared spectroscopy (FT-IR) and UV–visible (UV–vis) spectra were recorded to study the chemical composition of IG and $IKG-I_x$. As shown in Fig. [1](#page-5-0)c, the peak at 1341 cm^{-1} corresponds to the stretching vibration of C-N belonging to PAM, while the peaks at 3431 and 1640 cm^{-1} are attributed to the N–H, O–H, and $C=O$ bending vibrations of PAA and PAM. The peak at 1456 cm^{-1} is assigned to the bending vibration of $-CH₂$, and the stretching vibration of $C=O$ can be related to 1722 and 1166 cm−1, illustrating the -COOH group in PAA. Additionally, the peak at 3431 cm^{-1} is attributed to the -OH group, and a series of peaks from 1370 to 1454 cm^{-1} are the bending vibration of -CH₂ in -CH₂OH, indicating the existence of KGM. Moreover, under the infuence of N on the imidazole ring, the asymmetric stretching vibrations of C-H were at 1460 and 1428 cm^{-1} , and the peaks at 3159 and 3105 cm⁻¹ are the stretching vibrations of the two adjacent C-H on the imidazole ring of EMIM:DCA. The peak at 2203 cm⁻¹ indicates C≡N of the negative ion in EMIM:DCA (Kabanda and Bahadur [2023](#page-12-12)). As a result, the PAM-*co*-PAA ionogel was successfully synthesized, and KGM biomass material was

Scanning electron microscopy (SEM) images of the freeze-dried IKG- I_0 surface in Fig. [1](#page-5-0)d–f exhibit an uneven structure with distributed large pores. On the

introduced in the covalently cross-linked networks to

obtain a PAM-*co*-PAA/KGM DN ionogel.

Fig. 1 a) Photograph demonstrating the transparency of IKG-I4. (**b**) Transmittance curves of ionogels. (**c**) FT-IR spectra of KGM, IKG-I₀, IKG-I₂, IKG-I₄, IKG-I₆. SEM images of (**d**)

IKG-I₀, (**e**) surface of IKG-I₄ and (**f**) fracture surface of IKG-I₄ at diferent magnifcations

contrary, IKG ionogels showed a dense pore structure (Fig. [1e](#page-5-0)) on their surface. The aperture size was much smaller than IKG- I_0 , showing a more robust honeycomb structure derived from hydrogen bonding between hydroxy groups of KGM chains and carboxyl/amidogen groups of PAM-*co-*PAA chains (Kim et al. [2020](#page-13-17)). This resulted in increased pore density and ensured the stability and stretchability of the IKG ionogel. Therefore, with its serried porous structure, $IKG-I₄$ could withstand high stress and elongation and recover rapidly from deformation.

Mechanical performances

Mechanical properties are one of the essential performances of fexible sensors. KGM, which introduced the semi-interpenetrating network into the IKG ionogels, endowed the IKG- I_x with excellent toughness, stretchability, resilience, and fex-ibility. Figure [2a](#page-6-0) shows that $IKG-I₄$ could resist 900% stretching of its initial length, enabling the ionogels to sustain all kinds of external stimuli without visible cracks. Ionogels with diferent KGM contents were tested by a universal tensile machine to investigate the tensile properties and

durability of ionogels. IKG- I_0 showed poor tensile strength (0.35 MPa) and elongation at break (559%). As the KGM content increased to 1.0 wt%, tensile strength and elongation at break increased to 1.16 MPa and 973%, respectively. Moreover, the toughness increased from 1.13 to 4.73 $MJ/m³$, and Young's modulus increased from 0.14 to 0.21 MPa.

Fig. 2 Mechanical properties of ionogels (**a**) Snapshots of IKG-I4 under external stimuli (stretch, tie, fold, twist). (**b**) Tensile stress–strain curves and (**c**) Toughness and Young's modulus of ionogels with diferent content of KGM. (**d**) Stress

data of 100 cycles of the IKG-I₄ at 250%. (**e**) Tensile loading– unloading tests of the IKG-I₄ at different setting strains $(50\%$, 100%, 150%, 200%, 250%, 300%, 350%, and 400%). (**f**) 1000 stretching cycles of stress-time curves

When the KGM content increased to 2.0 wt%, its tensile strength, toughness, and Young's modulus increased to 1.500 MPa , 6.15 MJ/m^3 , and 0.28 MPa , respectively, but elongation at break decreased to 914%. IKG- I_4 showed superior mechanical properties; tensile strength became 2.77 MPa, the elongation at break reached 997% (in Fig. [2b](#page-6-0)), the toughness of IKG- I_4 reached 11.04 MJ/m³, and Young's modulus was 0.60 MPa (in Fig. [2c](#page-6-0)). However, when KGM content was further increased to 6.0 wt%, tensile strength, elongation at break, toughness, and Young's modulus decreased to 1.392 MPa, 758%, 4.79 MJ/m^3 , and 0.31 MPa , respectively, due to the synergistic effect of the DN density and defects (Xie et al. 2021). In this case, IKG-I₄ (tensile strength of 2.77 MPa and elongation at break of 997%) was the optimal choice for the following tests.

As shown in Fig. [2](#page-6-0)d, when the tensile test was at 250%, it could complete 100 strain cycles without breaking or cracking. This indicates that increasing the cycle number caused a change in the dissipated energy. Moreover, the loading–unloading tensile curves of IKG- I_4 displayed remarkable overlap after fnishing the frst cycle, and its self-recovery rate reached 77.6% after 1000 cycles (Fig. [2](#page-6-0)f), indicating its excellent anti-fatigue properties and durability. The hysteresis during the stretching cycles illustrates that the hydrogen bonds in IKG-I could break to dissipate energy. The slight decrease in maximum stress in successive cycles is attributed to the requirement of time to recover dynamic hydrogen bonds. Therefore, the tensile strength tended to decrease during stretching cycles (Zhou et al. [2023](#page-14-7)). A tensile stress–strain test was also carried out at eight diferent strains (50%, 100%, 150%, 200%, 250%, 300%, 350%, and 400%). As seen in Fig. [2](#page-6-0)e, the curves coincided, proving the stability of IKG- I_4 . Therefore, the semi-interpenetrating network introduced by KGM improved the comprehensive mechanical properties and obtained favorable resilience. This allowed the IKG- I_x to build resilient, robust, and durable strain sensors.

Anti-freezing performance

Frost resistance plays an important role in broadening the applicable temperature range of fexible sensors. Diferential scanning calorimetry (DSC) was carried out to measure the glass transition temperature (T_g) of IKG-I₄ and hydrogel in the range from -55 to 100℃ to further study the anti-freezing performance of ionogels. As observed in Fig. [3](#page-8-0)a and b, when the temperature decreased (-20°C) , the flexibility and transparency of hydrogel decreased dramatically, but the IKG- I_4 maintained favorable flexibility and transparency. Still, the IKG- I_4 exhibited excellent transparency and could stand the twisting deformation and maintain flexibility. This illustrates that the IKG- $I₄$ could withstand -20℃ without a decrease in its fex-ibility. As shown in Fig. [3](#page-8-0)c, the weight loss of $IKG-I₄$ was 1.2 wt% after 20 days. However, the PAM-*co*-PAA hydrogel exhibited 60.0 wt% weight loss, indicating excellent stability of IKG- I_4 exposed to the ambient environment. Moreover, as seen in Fig. [3d](#page-8-0), there was a prominent exothermic peak at 0° C in the curve of the PAM-*co*-PAA hydrogel. On the contrary, the T_g was -43°C for the ionogels owing to the extremely low freezing point of the EMIM:DCA. Therefore, in this work, the ionogel had favorable anti-freezing and water retention properties when exposed to air. This completely complies with the requirements of the wearable sensors used in extreme environments.

Rheological properties

First, ionogels with diferent KGM contents were tested by frequency scanning. Figure [4a](#page-9-0) shows the apparent frequency-dependent behaviors of the storage modulus (*G'*) and loss modulus (*G''*) in the range from 0.1 to 100 rad s^{-1} . Every sample maintained ideal elasticity. This was found by the phenomenon that the G' values of IKG-I_x were all greater than G'' values in the whole frequency range. As the KGM content increased, the *G* curve had a higher value. However, when the KGM content exceeded 4.0 wt%, excessive cross-linking density and hydrogen bonding between PAM-*co*-PAA and KGM chains constrained the movement of the PAM-*co*-PAA network and the elastic properties of KGM chains. This led to an increase in the *G"* value and a decrease in the elastic nature of ionogels. Amplitude scanning was carried out (Fig. [4b](#page-9-0)) to investigate the efect of the KGM content on the mechanical properties. As the strain range was from 0.1% to 1.48%, G' and G" values maintained steady, indicating it was the linear viscoelastic region. When the strain was greater than 1.48%, *G'* and *G"* started to decrease. Furthermore, the increase

Fig. 3 Low temperature tolerance of (**b**) IKG-I4 comparing with (**a**) PAM-*co*-PAA hydrogel. (**c**) Weight change of IKG-I4 and PAM*co*-PAA hydrogel at room temperature. (**d**) DSC curves of IKG-I_x and PAM-*co*-PAA hydrogel

in the KGM content generated an increase in the *G'* value in the range of 0 to 4.0 wt%, but the G' value gradually decreased when the KGM content exceeded 4.0 wt%. This was because KGM chains could improve the toughness of ionogels. On the contrary, the over-linked structure formed by the hydrogen bonding between KGM chains and the PAM-*co*-PAA network could lead to the over-rigidity of materials (Wang et al. [2022a,](#page-13-6) [b](#page-13-7), [c](#page-13-8)), which reduces flexibility. Cyclic strain–time scanning was carried out (Fig. [4c](#page-9-0) and d) to further study the thixotropy infuenced by the KGM content. It can be observed that both IKG- I_0 and IKG-I₄ performed typical elastic states at the strain range of 0.1% to 100%, and the recovery rate of IKG- I_4 in six cycles (85.47%) was higher than that of IKG-I₀ (84.87%). The phenomenon above proved that adding KGM improved the structural stability to a certain extent, and KGM chains increased the cross-linking density, which led to the high stifness of $IKG-I₄$.

Electromechanical properties

The ionogels were installed as strain sensors to put them into use. So, they should be capable of transforming mechanical deformations into electrical signals through the change in resistance (ΔR) and possess excellent durable sensing ability. EMIM:DCA contained zwitterion, which could deliver electrons and give the ionogel conductivity. Besides, $IKG-I_x$ was constituted by a cross-linked DN and EMIM:DCA. Under the electric feld, the dipole ions of EMIM:DCA migrated toward diferent directions, giving the ionogel conductive properties (Wang, et al. [2022a](#page-13-6), [b](#page-13-7), [c](#page-13-8)). Additionally, the DN also gave IKG-I_x conductive pathways. When the sensor was deformed,

Fig. 4 The elastic modulus *G'* and loss modulus *G"* as a function of (**a**) frequency and (**b**) strain of ionogels with diferent content of KGM. (c) The cycle strain time sweep of IKG-I₄ and (d) IKG-I₀ under the action of low strain (0.1%) and high strain (100%)

it also narrowed the ion migration channels and caused a change in resistance (Scheme [2\)](#page-9-1). Figure [5a](#page-10-0) shows that as the KGM content increased, the conductivity of the ionogel sensor declined slightly; the conductivity of IKG-I₀ reached 4.77 mS/cm. When the KGM content increased to 2.0 wt%, its conductivity decreased to 4.44 mS/cm. Ionogels with 4.0 wt%

KGM still demonstrated excellent conductivity (3.94 mS/cm), and the conductivity of IKG- I_6 was 3.44 mS/ cm. Moreover, as presented in Fig. [5](#page-10-0)b, the change in resistance demonstrated diferent variations under various stretching circumstances. In diferent frequencies (0.344, 0.156, and 0.125 Hz), the sensors were stretched at a fxed strain of 100%. It is shown

Scheme 2 Schematic demonstration of the conductive mechanism of $IKG-I₄$

Fig. 5 a Conductivity of ionogels with diferent content of KGM. (**b**) The evolution of the $\Delta R/R_0$ signals under different strains (50%, 100%, 200%, 300%, and 400%). (**c**) Cyclic stretching-relaxing under diferent frequencies. (**d**) Relative resistance variation of the IKG-I₄ sensor under 100% strain for

in Fig. [5c](#page-10-0) that the electric signal increased as the tensile strain of the sample augmented. Noticeably, the sensor showed superior durability and stability during the continuous stretching-release test at 100% tensile strain for 1000 cycles (84.6% self-recovery ratio). The curves in the tests above changed steadily during the cycle (Fig. [5d](#page-10-0)). The sensitivity of the installed sensor was evaluated by the gauge factor (GF) (Xiang et al. [2022\)](#page-13-10), which is defned as the slope of the currentstrain curve. $GF = \Delta(\Delta R/R_0)/\Delta \varepsilon$, where $R = R_t - R_0$ and R_1 , R_0 , ε are real-time resistance, original state resistance, and mechanical strain $(\%)$ of IKG-I₄, respectively (Wang et al. [2022a](#page-13-6), [b](#page-13-7), [c](#page-13-8)). Regarding the use of IKG- I_4 to install a strain sensor, its strain sensi-tivity was studied in Fig. [5](#page-10-0)e. As seen, the GF of IKG- I_4

1000 cycles. (**e**) The gauge factor of the IKG- I_4 as strain sensors. (f) Response time of IKG-I₄ during stretching-relax process. (\bf{g}) Comparison of the IKG-I₄ and other ionogels in the literature in terms of conductivity, tensile strength, transmittance, response time and elongation at break

was 3.25 in the range of 0 to 500%. The responding time of the sensor was only 100 ms in the stretching process, and in the recovery process, the recovery time was 513 ms, which met the requirements of a strain sensor (in Fig. [5f](#page-10-0)). As depicted in Fig. [5](#page-10-0)g, the ionogel sensors exhibited extraordinary properties of conductivity, tensile strength, transmittance, response time, and elongation at break compared with ionogel sensors reported in the literature (Tie et al. [2022](#page-13-19); Sun et al. [2019](#page-13-20); Kim et al. [2023](#page-13-4); Huang et al. [2023;](#page-12-2) Klepić et al. [2020;](#page-13-21) Song et al. [2022;](#page-13-22) Wu et al. [2022a,](#page-13-15) [b](#page-13-16); Xu et al. [2021](#page-13-23)). Therefore, as a promising and fexible biomass-based functional material, IKG-I₄ ionogel could meet the high sensitivity and durability requirements of sensors to detect human motion.

Human body movement detection

With the application of the PARSTAT MC electrochemical workstation, the signal produced by the tensile strain of the ionogels can be transmitted to the Bluetooth device in a mobile phone. The movement was frst transformed into signals after immobilizing the sensor on diferent parts of the human body. As shown in Fig. [6](#page-11-0), diferent movements represented diverse curves, and the current increased when the movement range increased. Moreover, diferent angles of 30, 60, and 90° that the fnger bent were also detected by sensors fxed on the fnger joints in real-time, and the current gradually decreased simultaneously. Therefore, the change in the current could accurately refect the bending angles of fngers. Other movements were also tested, such as nodding, jumping, walking, elbow bending, and wrist bending. The sensor showed excellent sensitivity during the cycles without any apparent current loss. In short, the IKG- I_x is competent to be an ideal sensor for real-time monitoring and body movement detection with longevity and stability.

Conclusion

In this work, the KGM biomass polymer was introduced in the ionic solution, and a novel conductive ionogel was synthesized for the frst time. By adding new substrates, the resulting ionogels exhibited excellent tensile strength (the highest and lowest detection limits were 2.77 MPa,

Fig. 6 Variation of electrical signals detecting diferent human body movements. (Nodding, fnger bending, wrist bending, walking, running, and jumping)

997%, and 2.76 MPa, 955%, respectively), durability (1000 cycles), optical transparency (84%, 2 mm), high conductivity (3.94 mS/cm), and environmental tolerance. The ionogel was used as a fexible sensor to detect human movements, including neck bending, elbow bending, knee bending, or subtle movements such as fnger bending. Moreover, as a biomass subsistence, KGM is environmentally friendly and easily available, which decreases the cost of synthesizing fexible sensors with high mechanical properties. Based on the superior properties, assembling the ionogels as fexible sensors successfully monitored diferent ranges of human body movements with high stability and accuracy.

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Data availability The Data used in this study are available from corresponding authors on reasonable request.

Declarations

Consent for publication All authors revised the manuscript and agreed with the publication.

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