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Facile preparation of high dielectric flexible films based on titanium dioxide and cellulose nanofibrils

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Abstract A series of high dielectric composite films based on low-cost and eco-friendly titanium dioxide (TiO₂) and cellulose nanofibril (CNF) was prepared under a facile condition. The relative dielectric constants (ε_r) and dielectric loss (tan δ) were studied as the function of frequency and filler content. The ε_r of CNF/TiO₂ composite film was 19.51 (at 1 kHz) with a relatively low dielectric loss. Compared with pure CNF films ($\varepsilon_r = 6.92$ at 1 kHz), the ε_r of the composite film was improved about three times with

the dielectric loss increased slightly. The effects of TiO_2 addition and hot-press treatment on microstructure, thermal stability, and dynamic mechanical properties of the composite films were also analyzed. It was found that the addition of TiO_2 particles reduces the cellulose–cellulose bonding so generates more pores in the films, which has significant impacts on both dielectric and physical strength properties.

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Graphical abstract



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Introduction

With the fast-growing of advanced electronics and electric power system, soft film capacitors with both high-power and high-energy density at the same time have drawn more and more attention (Dang 2018). This film capacitors (TFCs) usually request high capacity density, large breakdown strength [typically exceeds 300 MV m⁻¹ (Paniagua et al. 2014)], highfrequency response, fast charge/discharge speed, low dissipation, and long lifetime. Compared with electrochemical capacitors and batteries, the charge/ discharge speed of capacitors depends on their polarizations, which are contributed by the orientation of dipoles and deformations of atoms and molecules and are much faster and less influenced by high frequency than charge carriers transport. With higher dielectrics, the capacitors with same capacitance can be smaller in size and lighter in weight, thus have higher capacity density. Polymer-based dielectrics surpass ceramics by its higher breakdown strength, lower dissipation, easier preparation and longer lifetime.

The most commonly used dielectric polymers for TFCs mainly are non-biodegradable and nonrenewable thermoplastic polymers, such as polypropylene (PP) (Kumari and Ghosh 2018; Lay et al. 2018), polyethylene terephthalate (PET) (Tang et al. 2018; Topala et al. 2007), polyethylene naphthalate (PEN), polyphenylene sulfide (PPS), polytetrafluoroethylene



(PTFE) (Thomas et al. 2008; Wang and Dang 2018; Yuan et al. 2014, 2018) and polystyrene (PS). Besides, other polymers include polyimide (PI) (Chang et al. 2009; Feng et al. 2013, 2014; Ishmael et al. 2014; Kizilkaya et al. 2012; Koytepe et al. 2008; Lay et al. 2018; Lee et al. 2009, 2012; Lee and Wang 2010; Lin et al. 2017; Lu et al. 2017; Meena et al. 2012; Olariu et al. 2017; Wang et al. 2018, 2010b; Wu et al. 2017; Yin et al. 2014; Zha et al. 2010a, b), polyamide (PA) (Novac et al. 2017; Qi et al. 2017, 2018), and polyvinylidene fluoride (PVDF) (Al-Saygh et al. 2017; Alam et al. 2017; Deshmukh et al. 2017; Dou et al. 2017; Gan and Abd Majid 2014; Park et al. 2013; Prabakaran et al. 2014; Rekik et al. 2013; Ribeiro et al. 2018; Su et al. 2016; Wang et al. 2010b; Yang et al. 2016) are also studied. PI-based nanocomposites are reported to have enhanced corona aging performance (Lin et al. 2017; Lu et al. 2017; Yin et al. 2014; Zha et al. 2010a), but their relative dielectric constant is generally below 6 (1 kHz) (Feng et al. 2013, 2014; Lay et al. 2018; Wang et al. 2018). Qi et al. prepared polyamide11 (PA11)/BaTiO₃/carbon nanotube (CNT) ternary nanocomposites with 3D segregated percolation routes, the relative dielectric constant is 16.2 with a dielectric loss of ~ 0.08 (1 kHz) (Qi et al. 2018). Alam et al. (2017) put titanium dioxide (TiO_2) nanoparticles into γ -phase containing PVDF, and the dielectric constant of the nanocomposite film reaches 32 with a dielectric loss of 0.25 (1 kHz). Recently, low-cost and eco-friendly cellulose nanofibrils are being increasingly explored as a candidate to replace some conventional dielectric materials (Abdel-karim et al. 2018; Al-Saygh et al. 2017; Bonardd et al. 2018; Chiang and Popielarz 2002; Gaspar et al. 2014; Inui et al. 2015; Jayaramudu et al. 2018b; Le Bras et al. 2015; Madusanka et al. 2016, 2017; Milinskii et al. 2018; Milovidova et al. 2014; Poyraz 2018; Poyraz et al. 2017b; Rajala et al. 2016; Shi et al. 2018; Yagyu et al. 2017; Zeng et al. 2016; Zhou et al. 2018).

Cellulose nanofibril (CNF) has a low density and coefficient of thermal expansion (CTE) $(12-28.5 \text{ ppm K}^{-1}),$ high mechanical strength (200-400 MPa) and Young's modulus (7.4-14 GPa), excellent thermal stability (> 180 °C) and chemical durability, and it is an almost inexhaustible green material (Du et al. 2017; Fujisaki et al. 2014). The relative dielectric constant (ε_r) of traditional paper prepared from micro-sized cellulose is in the low range of 1.3–4.0, resulting from the porous microstructure (Inui et al. 2014; Inui et al. 2015). With a densely packed nanostructure, the ε_r of nanocellulose paper reaches 5.3 (at 1.1 GHz) (Inui et al. 2014, 2015) with a breakdown strength of 613.8 kV cm^{-1} (Zeng et al. 2016), making it a promising candidate for the dielectric matrix. Comparing to regenerated cellulose films that usually need toxic and expensive solvent, nanocelluloses can be made from pure mechanical grinding so toxic solvent is no longer needed. Furthermore, the dispersibility of TiO₂ nanoparticles in regenerated cellulose/solvent is poor, and phase separation between nanoparticles and cellulose occurs during film preparation. However, CNF is a nanofibrils rather than soluble molecules so the phase separation between TiO₂ particles and nanofibril network could be effectively prevented. However, the hydroxyl-richen cellulose shows strong hydrophilicity, which inevitably results in high electric leakage, high dielectric loss, low breakdown strength and low energy densities in humid environments (Shimizu et al. 2016; Yang et al. 2018a, b). Many studies have tended to focus on further improving the dielectric constant of cellulose nanopapers by introducing conductive fillers (Inui et al. 2015; Ji et al. 2017; Kafy et al. 2015b; Milovidova et al. 2014) but little attention has been paid on reducing dielectric loss.

Herein, we prepared a high dielectric composite film based on TiO₂ and CNF by a solution casting method. TiO₂ has a high dielectric constant ($\varepsilon_r = 63.7$ at 1 MHz), low dielectric loss (tan $\delta < 0.051$) (Wypych et al. 2014) and is stable in a broad temperature range (< 1000 °C). Besides, the hydrophilic property of TiO₂ offers a way out for homogeneous mixing with CNF suspension. Thus, TiO₂ is a promising candidate for CNF based dielectrics. Homogenous composite films were made by mechanically mixing TiO₂ nanoparticles with CNF. The relative dielectric constant (ε_r) and dielectric loss (tan δ) were studied as the function of frequency and filler content. The effects of hot-press treatment on dielectric properties, microscopy, thermal stability, dynamic mechanical properties, and hydrophilicity of composite films were also studied.

Experimental

Materials

Both CNF slurry and 2,2,6,6-tetramethylpiperidinooxy (TEMPO)-oxidized CNF (TCNF) slurry were purchased from the University of Maine, with solid content of 3.4 wt% CNF and 1.1 wt% TCNF, respectively (Fukuzumi et al. 2010; Isogai et al. 2011; Kumar et al. 2014; Osong et al. 2016; Postek et al. 2013; Sacui et al. 2014; Saito et al. 2009; Stelte and Sanadi 2009). TiO₂ nanoparticles of diameter ~ 21 nm (P25) were purchased from Nippon Aerosil Co. Ltd.

Preparation of CNF/TiO₂ composite film

The purchased CNF and TCNF were diluted into 0.34 wt% and 0.30 wt% respectively by distilled water. According to the dry weight percentage, a certain amount of TiO₂ was added, then the solution was stirred homogeneously by a homogenizer for 10 min before poured into a petri dish. The dried film was obtained after being put in a fume for 2–4 days. After being hot-pressed under 80 °C and 1.1 MPa for 3 h, light yellow, flat and thin films were obtained. At least three samples were prepared for each composition. The thickness of the sample films was in the range of 30–100 μ m. Figure 1 shows the flow diagram of the preparation of the CNF/TiO₂ composite film.

Characterizations

Morphologies of both surface and cross-section of the composite films were analyzed by thermally assisted field emission scanning electron microscope (TFE-SEM, LEO 1530, Germany) at an accelerating voltage of 10 kV. The composite films were sputtered with



Fig. 1 The flow diagram illustrates the preparation of CNF/TiO₂ composite film

gold in vacuum to avoid accumulation of charge before observation.

Thermogravimetric analysis and differential scanning calorimetry (TGA–DTA) was performed on a simultaneous thermal analyzer (PerkinElmer STA6000, USA), ranging from 35 to 700 °C with a heating rate of 10 °C min⁻¹ under nitrogen atmosphere and hold at 105 °C for 10 min before heading to a higher temperature.

The dielectric properties of sample films were measured by an LCR meter (Keysight E4980 with a 16451B fixture, USA) in the frequency range of 20 Hz–2 MHz. The test for each sample film was repeated at least five times. The thickness of the sample film was measured by a micrometer and was averaged over seven measurements on each sample. Unless otherwise stated, the dielectric constant of a material refers to the relative dielectric constant, which is the ratio of its absolute dielectric constant to the dielectric constants (ε_r) of the sample films were calculated by Eq. (1)

$$C = \varepsilon_0 \varepsilon_r \frac{A}{d} \tag{1}$$

where C is the capacitance; ε_0 is the absolute dielectric constant of vacuum, $\varepsilon_0 = 8.854 \times 10^{-12}$ F/m; A is the electrode area, $A = 1.963 \times 10^{-12}$ m²; d is the thickness of the sample film.

Tensile strength and ultimate elongation were studied with a dynamic mechanical analyzer DMA (Q800, TA Instruments, New Castle, DE, USA) with a test rate of $10\% \text{ min}^{-1}$ at room temperature. The film specimens were 5 mm wide and 20 mm long. At least four specimens were tested for each sample.

The densities of the sample films were calculated by Eq. (2)

$$d = \frac{m}{A \times t} \tag{2}$$

where d is the density, m is the weight, A is the surface area, and t is the thickness.

Results and discussion

Microscopy

Figure 2 shows the surface and cross-section morphologies of pure CNF and TiO₂ (50 wt%)/CNF composite film. The CNF was typically dozens of micrometers long with a diameter lower than 0.3 μ m, and part of CNFs aggregated with each other (as shown in red circle). However, unlike normal paper which has a porous structure, no obvious pore was observed in pure CNF film. Figure 2b shows some level laminated structures with obvious layer gaps that might result from the peeling during the sample



Fig. 2 Cross-section and surface SEM images of pure CNF (a, c) and hot-pressed CNF/TiO₂ (50 wt%) (b, d)

preparation. A dense layer and much smoother surface were observed, and it might result from the hot-press treatment. Figure 2d illustrates the distribution of TiO_2 in CNF. Because of the poor capability between the inorganic filler and organic matrix, there were some small pores, which had great influence on the properties of the composite film.

Dielectric properties

Prior to study the dielectric properties, a series of dielectric tests with different levels of oscillation signal (OSC level) was conducted. As shown in Fig. 3, the OSC level was adjusted in the range of 0.1-2 V. At lower OSC level of 0.1-1.5 V, the dielectric constant increased with the increase of OSC level, and the growth rate was decreased. After OSC level reached 2.0 V, the dielectric constant started to decrease. The same results were obtained in other composite films with different TiO₂ content. It illustrated that the testing electric field starts to overpass the breakdown strength of the films. Higher OSC level brings a higher



Fig. 3 The influence of OSC level on the dielectric test (50 wt% TiO_2/TCNF)

risk of breakdown. Thus, the following dielectric tests were conducted with OSC level at 1.0 V.

Figure 4 shows the influence of frequency and filler content on relative dielectric constant (ϵ_r) and dielectric loss (tan δ). With the increase of frequency, both ϵ_r



Fig. 4 The influence of frequency and filler content on the dielectric constant and dielectric loss

and tan δ decreased. In the lower range of 20–100 kHz, sharp reductions were observed. It was caused by the electrode polarization which took place at the interface between metal electrode and samples, and Maxwell–Wagner–Sillars interfacial polarization which can be observed at the interface between CNF and TiO₂ in inhomogeneous materials (Emmert et al. 2011; Lu and Zhang 2006; Samet et al. 2015; Anju and Naraya-nankutty 2016; Mohiuddin et al. 2015). Resulted from the interfacial polarization, the test results of ε_r and tan δ at the point of 20 Hz were abnormally high (several hundred or even higher) and were not showed

in Fig. 4. In the higher frequency range of above 500 kHz, the reduction of both ε_r and tan δ tended to be flatter and showed much less dependence on frequency and filler content, which indicated that electronic and atomic polarization and orientation polarization started to play a predominant role.

The dielectric properties of the composite films can be explained by the multi-layered core model (Tanaka et al. 2005). According to this model, the interface of filler particle is chemically consisting of a bonded layer, a bound layer, a loose layer, and an electric double layer overlapping the above three layers. The addition of nano-sized TiO2 has contradicted effects on the dielectric properties. On the one hand, nanosized TiO₂ introduced large surface areas for interfacial polarization compared with micro-sized TiO₂ (Kafy et al. 2015a), resulting in an extraordinary increase of both ε_r and tan δ in the low frequency range. On the other hand, the bonded layer and bound layer of the nanoparticles impair the motion of dipoles, leading to a reduction of ε_r and tan δ . And the dipoles and ionic carriers in the loose layer may act inversely. The farfield effect caused by the double electric layer makes the neighbored nano-particles collaborate with each other. The imperfection of heterogeneous structures can increase ε_r and tan δ . With the interfacial polarization becoming weak in the high frequency range, TiO₂ content showed less effect. In the case of microsized TiO₂, the increased ε_r is usually explained in term of the Lichtenecker-Rother logarithmic law of mixing (Tanaka 2005).

The ε_r of CNF/TiO₂ increased with the increasing of TiO_2 content in the range of 0–50 wt%. The maximum dielectric constant was 19.51 (at 1 kHz) at 50 wt%. As the TiO_2 content continued to increase, the dielectric constant decreased. It was caused by the aggregation of TiO_2 and the appearance of pores, as shown in SEM images. In the range of 0-70 wt%, the dielectric loss showed less dependency on TiO₂ content and fluctuated in the range of 0.51-0.81 (at 1 kHz). The ε_r of TCNF/TiO₂ increased with TiO₂ content in the range of 10-70 wt% and reached the maximum value of 47.15 (at 1 kHz) at 70 wt%. The dielectric loss of TCNF/TiO₂ composite films also showed less dependency on TiO₂ content and was fluctuated in the range of 2.57-3.32 (at 1 kHz). Generally, the dielectric loss of TCNF/TiO₂ composite films was three times higher than CNF/TiO₂ composite films, which was caused by the residual ions after TEMPO oxidation treatment of CNF. Thus, from the perspective of reliability and energy saving, CNF/TiO₂ was better than TCNF/TiO₂. Compared with other reported CNF based dielectrics, our CNF/ TiO_2 showed a much lower dielectric loss.

Thermal properties

In order to study the thermal stability of the composite films, TGA–DTA measurements were conducted on a series of CNF/TiO₂ composite films with the TiO₂ content in the range of 10–50 wt%. Figure 5a, b shows

the typical TGA–DTA curves for pure CNF and CNF/ TiO_2 (50 wt%) composite film after hot-press treatment, respectively. In the low temperature region below 105 °C, weight losses of 5.66-2.95% on the TGA curve and an endothermic peak at 41 °C on the DTA curve were observed. It was caused by water evaporation (Chenampulli et al. 2019; Hassan et al. 2019; Jayaramudu et al. 2018a; Lizundia et al. 2016; Poyraz 2018; Poyraz et al. 2017a; Raghunathan et al. 2017; Zeng et al. 2016). The temperature was held at 105 °C for 10 min, and it was showed obviously on the DTA curve as a marked drop. The 5% decomposition temperature of the sample films was 291–302 °C, which indicated that the TiO₂/CNF has a low water absorption and good thermostability. The fluctuation of the DTA curve and sharply loss of weight around 400 °C showed the decomposition of CNF. At 700 °C, for the composite films, the residual weight percentage of sample films was per the TiO₂ content, which indicated that CNF had been completely decomposed at this temperature. However, for pure CNF, the weight percentage was 15%. It indicated that the addition of TiO₂ accelerates the thermal degradation of CNF.

Dynamic mechanical property

Figure 5c shows the DMA test results of CNF/TiO_2 composite films. All the stress-strain curves show "S" shape, suggesting that with the increase of strain, the stress first increased slowly, then sharply, and turned to be slow again, and finally the sample broke down, which is a typical property of flexible films. For pure CNF, hot-pressing treatment improved the strain at break from 7.65 to 10.92% with the stress slightly decreased from 76.91 to 76.59 MPa, which was resulted from better bounding between CNFs after hot-pressing treatment. However, CNF/TiO₂ (50 wt%) composite film has a contract result. After hot-pressing, the strain at break decreased from 5.45 to 2.63%, and the stress increased from 12.68 to 17.71 MPa, which may be caused by the weak bonding between TiO₂ particles and CNFs.

Density

Figure 5d shows the densities of untreated and hotpressed (HP) CNF films. Pure CNF films have higher densities than CNF/TiO₂ (50 wt%) although the



Fig. 5 a TGA–DTA curves of pure CNF film. b TGA–DTA curves of CNF/TiO₂ (50 wt%) composite film after hot-press treatment. c Dynamic mechanical properties of hot-pressed (HP)

density of TiO₂ is much higher than cellulose. Because the poor bonding between CNF and TiO₂, a looser structure of the films was obtained by adding TiO₂ particles, which is confirmed by SEM images shown in Fig. 2. Hot-pressing could improve the density of CNF films, and the rise of pure CNF is much higher than that of CNF/TiO₂ (50 wt%), suggesting that the pores in the CNF/TiO₂ films are hardly be removed by physical treatment.

and untreated CNF/TiO_2 composite films with different TiO_2 content. \boldsymbol{d} Densities of untreated and hot-pressed (HP) CNF films

Conclusions

In conclusion, a high dielectric film was prepared under facile condition. The relative dielectric constant of CNF/TiO₂ composite film reached 19.51 (at 1 kHz). Compared with pure CNF films ($\varepsilon_r = 6.92$ at 1 kHz), the ε_r of composite films was improved about three times. It was also illustrated that hot-pressed CNF/ TiO₂ had good flexibility and thermal stability. The addition of TiO₂ particles reduces the cellulose– cellulose bonding so generates more pores in the films, which has significant impacts on both dielectric and physical strength properties.

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