

Properties of hydroxypropyl guar/TEMPO-oxidized cellulose nanofibrils composite films

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Abstract Flexible hydroxypropyl guar/TEMPO-oxidized cellulose nanofibrils (HPG/TOCNs) composite films were prepared using a conventional solvent-casting technique. Their properties were investigated with a variety of techniques including tensile test, Fourier-transform infrared spectroscopy, scanning electron microscopy, oxygen permeability test, water vapor permeability test, thermogravimetric analysis and water contact angle measurement. The results indicate that HPG and TOCNs have an excellent miscibility and their blending mass ratio can significantly affect the physical, thermal, oxygen barrier and water vapor barrier properties of the composite films. Compared with pure HPG film, the composite films exhibit higher tensile strength and oxygen barrier properties. The water vapor resistance and thermal stability of the composite films are slightly lower than those of HPG film. However, the excellent flexibility,

transparency and gas-barrier properties of the environmentally friendly HPG/TOCNs composite film make it a promising packaging material for food and pharmaceutical industries.

Keywords Cellulose nanofibrils · Hydroxypropyl guar · Composite film · TEMPO · Oxygen barrier · Water vapor permeability

Introduction

Cellulose nanofibrils (CNs) have attracted considerable attention in recent years due to their biodegradability, large surface area, high strength and high film-forming capacity (Rodionova et al. 2012a). It is a highly reproducible and environmentally friendly nanomaterial (Fujisawa et al. 2011) and has been used as the nano-fillers (Iwatake et al. 2008), thin coating layers (Aulin and Strom 2013), and films (Song et al. 2014) for many generic and cutting-edge products. Cellulose nanofibrils are usually prepared from native cellulose fibers through 2,2,6,6-tetramethylpiperidine-1-oxyl (TEMPO)-mediated oxidation and successive mild shear mechanical treatment (Saito et al. 2006, 2007). Compared with native cellulose fibers, TEMPO-oxidized cellulose nanofibrils (TOCNs) is deemed as an innovative nano-sized biomaterial due to its crystalline and amorphous domains (Jradi et al. 2012). TOCNs prepared from wood cellulose are 3–4 nm wide nanofibers with high aspect ratios of >50

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and can be individually dispersed in water (Fujisawa et al. 2011). Self-standing films made by casting and drying aqueous TOCNs dispersions are very flexible and transparent (Fukuzumi et al. 2009). In addition, the oxygen permeability (OP) of TOCNs film is only $0.049 \text{ mL } \mu\text{m m}^{-2} \text{ day}^{-1} \text{ kPa}^{-1}$ (Fujisawa et al. 2011), which is lower than that of poly(vinyl alcohol) (PVA) (Kato et al. 2005).

TOCNs can be blended with water-soluble polymers such as PVA (Zhou et al. 2012), poly(ethylene oxide) (Fukuya et al. 2014) and poly(acrylamide) (Kurihara and Isogai 2014) to prepare highly flexible and strong composites since TOCNs have high dispersibility in water (Sirvio et al. 2014; Wu et al. 2012; Kurihara and Isogai 2014). The “nano-effects”, referring to the phenomenon that the composites properties improve significantly with the incorporation of nano-element, occurred during the blending can improve the performance of the produced TOCNs composite films. In addition, layer-by-layer coated film of chitin nanofibrils/TOCNs (Qi et al. 2012) and oxidized starch/TOCNs composite film (Kurihara and Isogai 2014) prepared with water as a dispersion medium, show excellent mechanical properties and significantly different properties from those of the composites without TOCNs. Therefore, the incorporation of TOCNs into the modified composite films can expand their applications.

Hydroxypropyl guar (HPG) is a water-soluble non-ionic polysaccharide consisting of a linear backbone of β -(1-4) D-mannose units with α -(1-6) D-galactose units randomly attached as side chains (Nayak and Singh 2001). It has better water solubility and film forming property than guar gum (Lu et al. 2005; Xiao et al. 2003) and plays a large role in the various products of chemically modified guar gum (Cheng et al. 2002). HPG has been widely used in many industries such as oil recovery, food industry, paints and the formulation of cartridge explosives (Wu et al. 2010; Xiao et al. 2003). The large amount of $-\text{OH}$ groups in HPG can interact with sodium carboxylate groups and hydroxyl groups on the surfaces of TOCNs. These attractive interactions may result in a well-blended composite material with improved properties.

In the present work, composite films with various blending mass ratios of HPG to TOCNs were prepared through solution casting method. The miscibility, physical properties and thermal properties of the composite films were determined with tensile

measurements, Fourier transform infrared spectroscopy (FTIR), scanning electron microscopy (SEM) and thermogravimetric analysis (TGA). The oxygen and water vapor barrier properties were also examined. The effects of the blending ratio between TOCNs and HPG on the properties of the composite films were investigated. The results reported in this article may contribute to finding further applications of these composite films in package materials.

Experimental

Materials

A commercial softwood bleached kraft pulp (SBKP) was pretreated with 5 % (v/v) formic acid solution to improve the efficiency of TEMPO-mediated oxidation (Dai et al. 2014). Hydroxypropyl guar (HPG) with a degree of substitution (DS) of 0.15 was kindly provided by Wuxi JinXin Group Co., Ltd. (Wuxi, China). All other reagents were purchased from Sinopharm Chemical Reagent Co., Ltd. (Shanghai, China) and used as received.

TEMPO-oxidized cellulose nanofibrils (TOCNs) production

The formic acid pretreated pulp (5 g) was suspended in a 500 mL solution containing 0.1 g TEMPO, 1.0 g NaBr, 3.71 g Na_2CO_3 and 1.26 g NaHCO_3 . Thereafter, NaClO (4.5 mmol/g pulp) was added to the suspension under ambient condition to prepare TEMPO-oxidized cellulose. The oxidation reaction was quenched by adding 10 mL ethanol after 4 h. The carboxylate content of TEMPO-oxidized cellulose was 1.34 mmol g^{-1} . An aqueous 1 wt% TEMPO-oxidized cellulose slurry was homogenized with an IKA T25 digital ULTRA-TURRAX homogenizer at 20,000 rpm for 30 min and subsequently sonicated with an ultrasonic probe (25 mm probe tip diameter, Scientz, China) at 19.5 kHz and 400 W for 20 min. The transparent TOCNs aqueous dispersion was stored at 4 °C before use.

Preparation of HPG/TOCNs composite films

Ten grams HPG powder was completely dissolved in 1000 mL deionized water at room temperature by

stirring at 300 rpm for 8 h. Then, the HPG solution was mixed with TOCNs aqueous dispersion with the dry mass ratios of 100:0, 70:30, 50:50, 30:70 and 0:100. The mixtures were stirred for 3 h and poured into polytetrafluoroethylene petri dishes (150 mm diameter), respectively. The films were oven-dried at 50 °C for 3 days, peeled off and marked as HPG (100:0), CF1 (70:30), CF2 (50:50), CF3 (30:70) and TOCNs (0:100), respectively. All these films were kept at 23 °C and a relative humidity (RH) of $50 \pm 1 \%$ for 2 days before characterization.

Thickness/density measurement

Film thickness was determined as an average of ten measurements using a thickness meter (MP0, Fischer, Germany). Density of individual film was calculated by dividing grammage by film thickness.

UV–Vis transmittance

The light transmittance of the films was characterized with a Shimadzu UV-1800 UV–Vis spectrometer, and was measured from 200 to 800 nm.

Mechanical test

The films were cut into 10 mm × 100 mm rectangular strips and their tensile strength and elongation at break were measured at room temperature with a BZ2.5/TNIS Zwick Material Tester (Zwick, Germany). The initial grip separation was set at 50 mm and specimens were loaded at a constant cross-head speed of 50 mm/min. At least 5 specimens of each sample were tested.

Fourier-transform infrared (FTIR) spectroscopy

FTIR spectra of all films were recorded on a Nicolet is 10 spectrometer with an attenuated total reflectance (ATR) accessory (Thermo Fisher Scientific Inc., USA) at a resolution of 4 cm^{-1} in the range of 550–4000 cm^{-1} . An average of 16 scans was reported.

Scanning electron microscopy (SEM)

The cross-sections of the films were affixed to a vertical brass specimen holder for the SEM imaging. Samples were coated with a gold layer and their

surface morphologies were examined with a HITACHI SU1510 SEM at 5 kV acceleration voltage.

Contact angle (CA) measurement

The water contact angle of each film was measured with a Drop Shape Analyzer-DSA 100 (KRÜSS GmbH, Germany) at ambient condition. A drop of water (4 μL) was deposited on the specimen surface and a series of water droplet images were captured and analyzed.

Oxygen permeability

The oxygen permeability of the films was determined with a Labthink VAC-V1 apparatus (Labthink, China) at 23 °C and $40 \pm 1 \%$ RH. The sample size was 38.48 cm^2 and the partial pressure of oxygen was 0.1 MPa.

Water vapor permeability (WVP)

The WVP of the films was determined according to the previous reports (Das et al. 2011; Sharma et al. 2014). Briefly, a composite film with an area of 19.64 cm^2 was sealed on a cup containing 10 g of oven-dried CaCl_2 as a desiccant. The cup covered with sample was accurately weighted and put in a desiccator cabinet containing saturated NaCl solution that was used to generate 75 % RH gradient. The cup covered with film was reweighted every 24 h until a constant weight was obtained. The WVP was calculated using the following equation:

$$\text{WVP} = \frac{\Delta m \times \delta}{\Delta t \times A \times \Delta P}$$

where $\Delta m/\Delta t$ is the moisture weight of the film gained per time unit, δ is the average film thickness, A is the exposed surface area of the specimen, ΔP is the difference between the water vapor pressures on the two sides of the specimen. All measurements were repeated three times and only the mean value was reported.

Thermogravimetric analysis (TGA)

TGA of films was conducted on a TGA/SDTA 851e Thermoanalyzer (METTLER TOLEDO, Switzerland). The samples were heated from 25 to 600 °C at

a heating rate of 10 °C/min. A nitrogen gas stream was continuously passed into the furnace at 10 mL/min and the gradual weight loss of the sample was recorded.

Results and discussion

Optical and mechanical properties

All films were translucent, smooth, flexible and relatively tough. Figure 1 shows the photographs of HPG, CF1, CF2, CF3 and TOCNs films. Obviously, there exists visual difference between the films with different composition ratios. In addition, light transmittance through the films as a function of wavelength was obtained and is shown in Fig. 2. The transmittance at 600 nm was around 85 % for TOCNs film, while it was about 52 % for the HPG film. The transparency of the composite films was increased with a higher amount of TOCNs. It is well known that the particles with diameters less than one-tenth of the visible-light wavelength do not scatter light (Yang et al. 2012; Yano et al. 2005). Therefore, the high transparency of the TOCNs film can be attributed to the nano-sized fibers of TOCNs (Rodionova et al. 2012c) while the lower transmittance of HPG film is

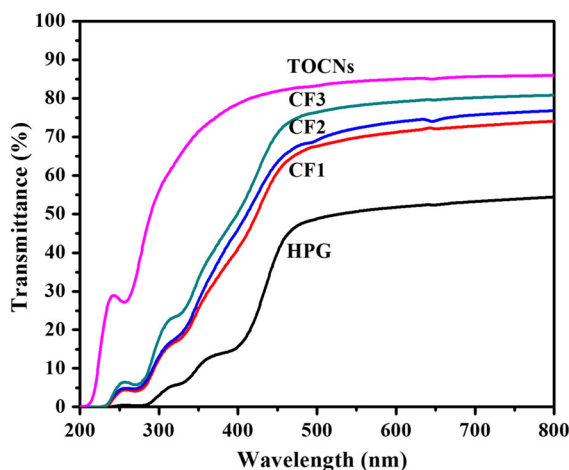


Fig. 2 UV-Vis transmittance of films made from HPG and TOCNs

probably due to the remaining cell wall fragments. Otherwise, the introduction of HPG into the composite films increased the particle size, leading to increased light scattering.

The moisture content, thickness, density and tensile properties of the films with various HPG/TOCNs ratios are listed in Table 1. Moisture content of the films was determined by drying the samples in an oven at 105 °C for 5 h. The moisture content of the



Fig. 1 Photographs of control (nothing covered) and films made from HPG and TOCNs

Table 1 Physical and tensile properties of films prepared from TOCNs and HPG

	Moisture content (%)	Thickness (μm)	Density (g/cm^3)	Tensile strength (MPa)	Elongation (%)
HPG film	15.63	106.8 ± 1.4	1.06 ± 0.01	19.1 ± 1.0	7.23 ± 0.50
CF1	14.10	96.4 ± 1.4	1.17 ± 0.02	39.8 ± 4.5	6.64 ± 0.42
CF2	13.57	82.5 ± 0.5	1.37 ± 0.01	44.6 ± 4.3	5.62 ± 0.24
CF3	12.92	70.2 ± 2.9	1.45 ± 0.06	51.6 ± 1.6	4.13 ± 0.30
TOCNs film	12.72	61.1 ± 0.7	1.52 ± 0.02	149.9 ± 13.8	2.21 ± 0.47

composite films was decreased with the increase of TOCNs content, indicating that cellulose fibers might have lower affinity to water than HPG. The density of the film was increased and thickness was decreased with the increase of TOCNs content, suggesting that the TOCNs content affected the pore size and cracks in the composite film. Pure HPG film showed the lowest tensile strength of around 20.0 MPa and the tensile strength of the composite films was increased with the increase of TOCNs content. Meanwhile, the elongation at break of the films was decreased dramatically with the increase of TOCNs content. The highest tensile strength of 149.9 MPa was found in pure TOCNs film. The high tensile strength of the TOCNs film may be attributed to its high crystallinity, large aspect ratio of nanofibrils (Fukuzumi et al. 2009; Wu et al. 2012) and the numerous O \cdots HO hydrogen bonds formed between its elements (Shimizu et al. 2014). The high crystallinity, large aspect ratio and uniform distribution of TOCNs can also significantly improve the mechanical properties of its composite film (Jradi et al. 2012; Wu et al. 2012; Zhou et al. 2012). The denser structure caused by the addition of TOCNs may also contribute to the increased tensile strength of the composite films (Kong et al. 2014; Takagi and Asano 2008; Qing et al. 2012). The enhanced tensile strength of the composite films might indicate the good interaction between HPG and TOCNs. Meanwhile, the incorporation of TOCNs could diminish the elongation at break of composite films. The stiffness of TOCNs as well as the low moisture contents in the composite films with higher TOCNs contents may lead to their low elongation (Fujisawa et al. 2011).

Miscibility

The IR spectrum of a blend of two compatible polymers should be significantly different from their individual IR spectra. And the differences, including

band shifts, intensity changes and peak broadening, are derived from chemical interactions (Dong and Ozaki 1997; Xiao et al. 2003). Figure 3 shows the IR spectra of HPG, TOCNs and CF2 films. The broad absorption band of HPG at 3650–3000 cm^{-1} could be assigned to the stretching of O–H. The bands at 2920 cm^{-1} was ascribed to C–H stretching. The peak at 3340 cm^{-1} in the spectrum of pure TOCNs film was attributed to the characteristic peak of hydroxyl groups. However, the intensity of O–H stretching vibration band was lower due to the oxidation of the primary hydroxyl at C6 (Wang et al. 2014b). The peak at $\sim 1600 \text{ cm}^{-1}$ was attributed to the C=O stretching vibration of sodium carboxylate groups (Homma et al. 2013; Fukuzumi et al. 2013a). The corresponding absorption peaks of composite film CF2, including O–H stretching, C–H stretching and C=O stretching appeared at 3343, 2900 and 1602 cm^{-1} , respectively. Based on the increased intensity of O–H absorption peak and the peak shift of sodium carboxylate group, it could be concluded that HPG and TOCNs were cross-linked.

Figure 4 shows the SEM images of the cross-sections of all films. Numerous pores and cracks were observed in HPG film. TOCNs film showed a dense and uniform compact structure. No big aggregates were observed in composite films. Their homogeneous morphology suggested that high miscibility occurred during the compositing and HPG had good adhesion to TOCNs. The interaction between HPG and TOCNs also led to the gradually decreased film thickness as the content of TOCNs increased. A rigid hydrogen-bond network of cellulose nanofibrils governed by percolation might also form in the composite films (Zhou et al. 2012).

Contact angle

Figure 5 shows the contact angle (CA) change of a water droplet on the films with the increase of

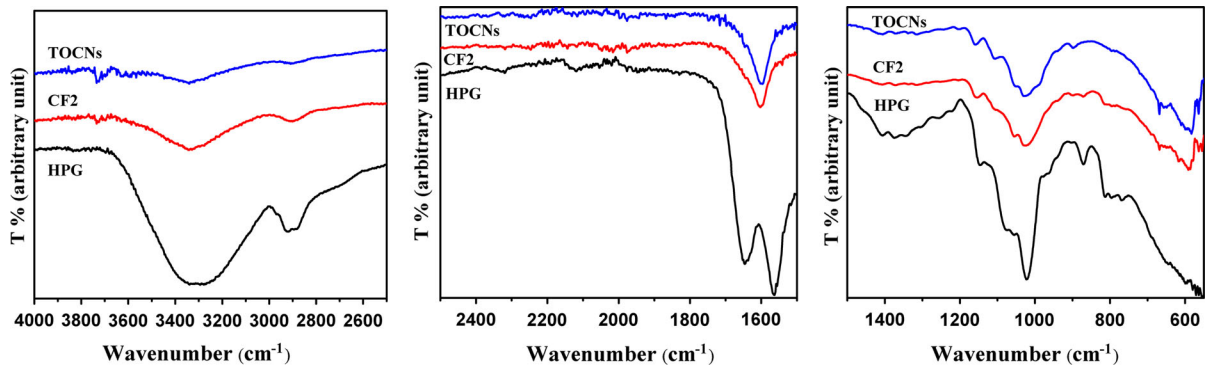


Fig. 3 FT-IR spectra of HPG, CF2, and TOCNs films

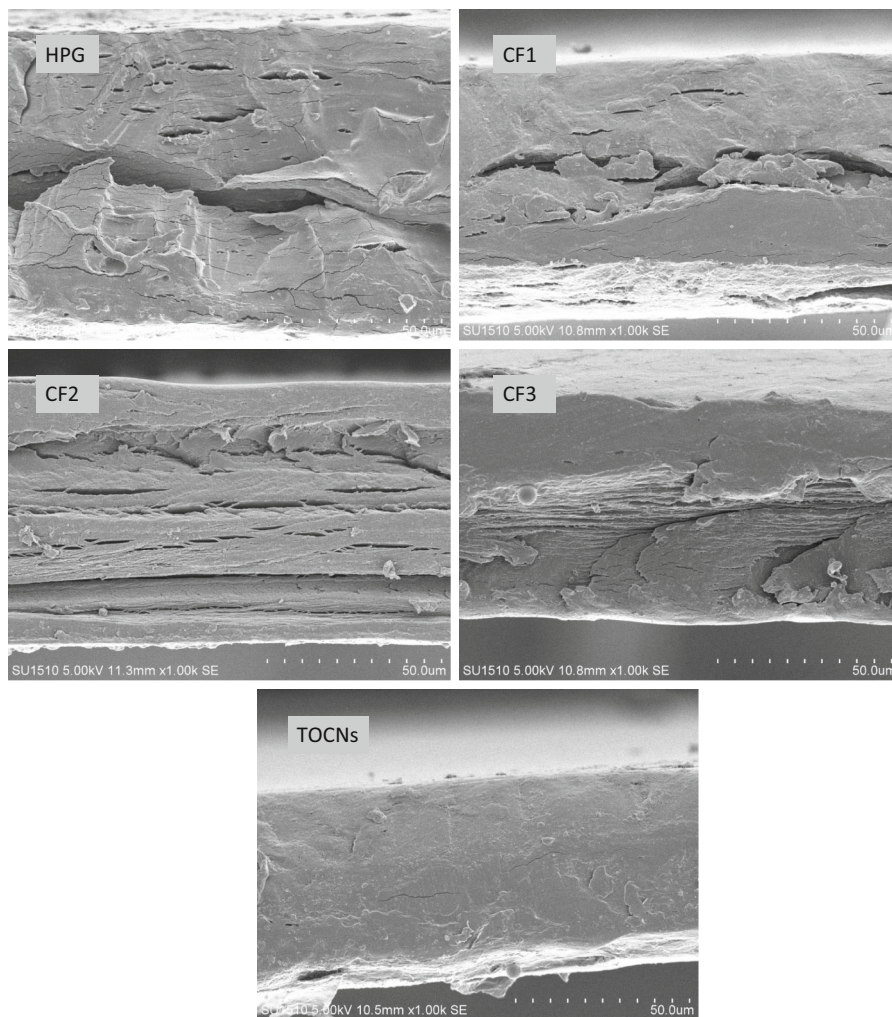


Fig. 4 SEM images of the cross-sections of pure HPG, pure TOCN, and composite films

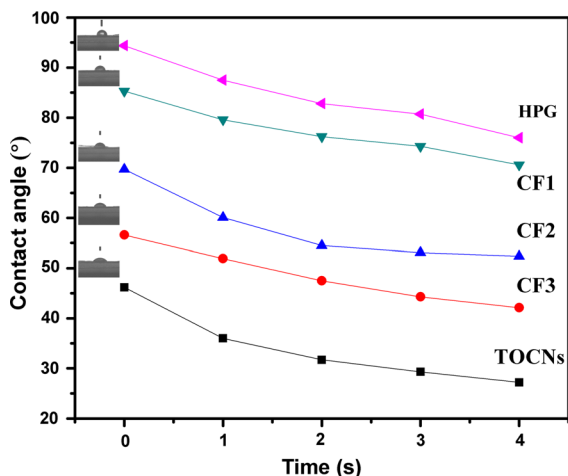


Fig. 5 Change in the contact angle with time of a water droplet on films

contacting time. The surface roughness and chemical composition of a substrate can significantly affect its surface wettability (Rodionova et al. 2012a). The initial water contact angle of pure TOCNs film was 46.2° , consistent with the high hydrophilicity of the TOCNs film reported elsewhere (Fukuzumi et al. 2009). Otherwise, it was noticed that the pure HPG film had an initial water-contact angle of 94.4° . The water CAs of the composite films indicated that the incorporation of HPG significantly improved the surface water resistance of the composite films. Nevertheless, the different CAs of the composite films might also be partially attributed to their different surface roughness (Rodionova et al. 2012a; Wu et al. 2014). The CAs of the composite films decreased with the increase of contacting time due to the partial penetration of water into the films. For example, the CA of CF2 dropped to about 52.4° in 4 s. High water/moisture resistance is highly required for the application of TOCNs film as an oxygen-barrier packaging material (Kato et al. 2005). Therefore, the introduction of HPG can expand the application of TOCNs.

Oxygen-barrier and water vapor-barrier properties

All films including HPG, CF1, CF2, CF3 and TOCNs films were subjected to oxygen permeability measurements at 23°C and $40 \pm 1\%$ RH. The results are present in Fig. 6. The TOCNs film showed the lowest

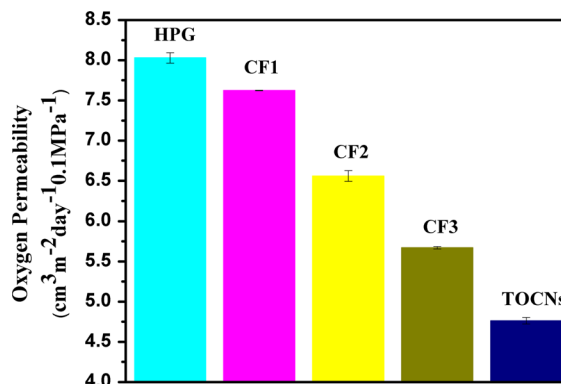


Fig. 6 Oxygen permeability of the films

oxygen permeability of $4.76\text{ cm}^3\text{ m}^{-2}\text{ day}^{-1}\cdot 0.1\text{ MPa}^{-1}$. The regular chemical structures of the cellouronic acids in TOCNs film without bulky substituents or additives could contribute to its high oxygen-barrier level (Kato et al. 2005). Moreover, the high crystallinity and highly self-aligned elements in TOCNs film were also partially contribute to its superb oxygen-barrier property (Fukuzumi et al. 2013b; Rodionova et al. 2012c; Wu et al. 2012). The loose and cracked structure of pure HPG film diminished its oxygen barrier property. However, the oxygen permeability of TOCNs/HPG composite films was in the range of $5.67\text{--}7.62\text{ cm}^3\text{ m}^{-2}\text{ day}^{-1}\cdot 0.1\text{ MPa}^{-1}$ (Fig. 6). The large amount of sodium carboxyl and hydroxyl groups in TOCNs could form denser aggregates alone or with HPG through strong intra- and inter-molecular hydrogen bond, which increased the oxygen barrier of the composite films (Kato et al. 2005). Moreover, water can act as a permeation medium for gases. Therefore, the reduced moisture content caused by the introduction of TOCNs to the composite films also improved their oxygen barrier (Sharma et al. 2014). In all, the HPG/TOCNs composite films still showed excellent oxygen-barrier properties and could be used as an oxygen sensitive food packaging.

The WVP of all films is given in Fig. 7. The WVP values of TOCNs and HPG films were $1.17 \times 10^{-5}\text{ g m}/(\text{m}^2\text{ day Pa})$ and $0.53 \times 10^{-5}\text{ g m}/(\text{m}^2\text{ day Pa})$, respectively. As discussed above, HPG has higher water affinity than cellulose fibers. However, it has been reported that guar gum can effectively trap and retain the water molecules within a matrix (Rosiaux et al. 2013; Poinot et al. 2014). Guar gum film shows

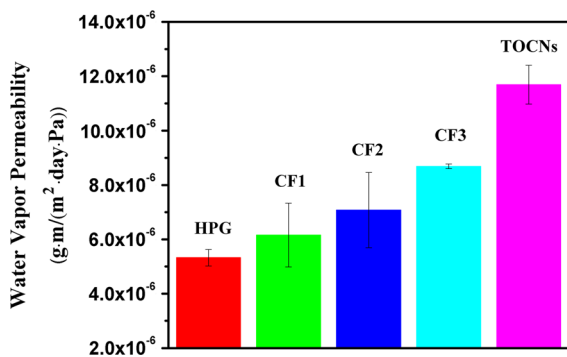


Fig. 7 Water vapor permeability of films

pronounced desorption hysteresis in equilibrium moisture content and can act as barrier to air and moisture (Keating et al. 2013; Shetty et al. 1996). The composite films showed lower WVP than TOCNs film, indicating that the introduction of HPG effectively reduced the WVP of the composite films. Mueller et al. (2009) found that the water vapor barrier of starch/cellulose films was improved as the film density reduced from 2.41×10^6 to 1.31×10^6 g/m³. Therefore, the lower density of the composite film might also partially contribute to its decreased WVP.

Thermal stability of the films

The thermal behaviors of HPG, TOCNs and HPG/TOCNs composite films were investigated by thermogravimetric analysis (TGA). All films subjected to a weight loss in the temperature range of 50–110 °C, which was attributed to their moisture contents (Fig. 8). The second weight loss of HPG film was observed at 240 °C where the polymer decomposition occurred (Nayak and Singh 2001). Further increasing the temperature to 330 °C led to increased weight loss rate. Thermal degradation of TOCNs film appeared at 200 °C where the crystalline phase of TOCNs was destroyed and the amorphous phase was decomposed into mono-D-glucopyranose (Rodionova et al. 2012b; Jradi et al. 2012). The third weight loss at 360–600 °C was attributed to the carbonation (Wang et al. 2014a). The decomposition temperature for the composite films, i.e., CF1, CF2 and CF3 were 223, 217 and 210 °C, respectively, indicating that the thermal stability of the composite films was increased with the increase of HPG content. The high thermal stability of composites could be attributed to the

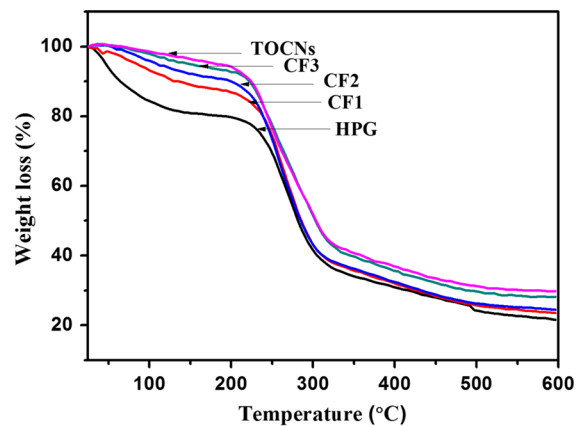


Fig. 8 TGA curves of neat HPG film, TOCNs film and composite films

hydrogen bonds formed between the –OH and –COO[−] groups of TOCNs and the –OH groups of HPG (Xiao et al. 2003).

Conclusions

A new transparent high gas-barrier composite film was prepared from hydroxypropyl guar (HPG) and TEMPO-oxidized cellulose nanofibrils (TOCNs) by a solution casting method. The morphologies, tensile strength, water resistance, oxygen permeability, water vapor permeability and thermal stability were investigated. The results indicate that the composite films have excellent mechanical property, as well as high oxygen and water vapor barrier properties. The bending mass ratio of HPG to TOCNs can significantly affect their properties. The HPG/TOCNs composite films can be used as wholly biomass-sourced films for the food and medicine packaging.

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