Effects of temperature and molecular weight on dissolution of cellulose in NaOH/urea aqueous solution

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Abstract Dissolution of cellulose having different viscosity-average molecular weight (M_n) in 7 wt% NaOH/12 wt%urea aqueous solution at temperature from 60 to -12.6° C was investigated with optical microscope, viscosity measurements and wide X-ray diffraction (WXRD). The solubility (S_a) of cellulose in NaOH/urea aqueous solution strongly depended on the temperature, and molecular weight. Their Sa values increased with a decrease in temperature, and cellulose having M_{η} below 10.0×10^4 could be dissolved completely in NaOH/urea aqueous solution pre-cooled to -12.6° C. The activation energy of dissolution (E_{a.s}) of the cellulose dissolution was a negative value, suggesting that the cellulose solution state had lower enthalpy than the solid cellulose. The cellulose concentration in this system increased with a decrease of M_n to achieve about 8 wt% for M_n of 3.1×10^4 . Moreover, cellulose having 12.7×10^4 could be dissolved completely in the solvent precooled to -12.6° C as its crystallinity (χ_c) decreased from 0.62 to 0.53. We could improve the solubility of cellulose in NaOH/urea aqueous system by changing M_{η} , χ_{c} and temperature. In addition, the zero-shear viscosity (η_0) at 0°C for the 4 wt% cellulose solution increased rapidly with an increase of M_n , as a result of

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the enhancement of the aggregation and entanglement for the relatively long chains.

Keywords Cellulose · Solubility · Temperature · Molecular weight · Crystallinity

Introduction

As well known, the cellulose is the most abundant material resource in the nature, especially the petrolic resource do not newborn and the crude oil prices fly to \$100/per barrel. However, cellulose forms unique microcrystal structures through strong hydrogen bonding networks (Zhao et al. 2007), leading to the difficulty of its dissolution in common solvent. In traditional viscose route for production regenerated cellulose fibers, CS2 (toxic gas) leads to the serious environmental pollution and the poor health of the human body. Therefore, several new solvent systems have been developed to dissolve cellulose at high temperature. The dissolution of cellulose in the N-methyl-morpholine-N-oxide (NMMO) occurs at 85–130°C (Heinze and Liebert 2001; Michael et al. 2000). By heating or refluxing cellulose in N,Ndimethylacetamide (DMAc), or in DMAc containing LiCl at about 150°C, transparent cellulose solution can be obtained (Tosh et al. 2000; Potthast et al. 2002). Recently, various ionic liquids (ILs) have been found to dissolve cellulose (Zhu et al. 2006), such as cellulose can be dissolved in the 1-N-butyl-3methylimidazolium chloride at about 85-100°C (Heinze et al. 2005; Kosan et al. 2008). At high temperature, ion pairs in 1-allyl-3-methylimidazolium chloride (AMIMCl) can be dissociated to individual Cl⁻ and AMIM⁺ ions, and then free Cl⁻ ions associated with the cellulose hydroxyl proton and the free cations disrupte the hydrogen bonding in cellulose, leading to its dissolution (Zhang et al. 2005). In addition, solution of cellulose in heavy metal-amine complex solutions system (such as Cuoxam, Ni-tren, Cd-tren, and so on) is usually prepared at room temperature (Saalwaechter et al. 2000). Microcrystalline cellulose (MCC) having low degree of polymerization (DP < 250) can be dissolved in 8-10%NaOH aqueous system via a subsequent freeze and thaw thermal cycling (Yamashiki et al. 1988; Kuo and Hong 2005; Isogai and Atalla 1998).

In our laboratory, new solvents, such as aqueous NaOH/urea, NaOH/thiourea and LiOH/urea aqueous solutions have been used to dissolve cellulose at low temperature (Cai and Zhang 2006; Cai et al. 2007a; Ruan et al. 2004; Cai and Zhang 2005). These solvents are attractable because cellulose can be easily and quickly dissolved them and produce stable cellulose solutions (Yan et al. 2007). It is worth noting that NaOH/urea aqueous solution is an economical and environmentally friendly solvent of cellulose. Cellulose could be rapidly dissolved in 7 wt% NaOH/12 wt%urea aqueous solution pre-cooled to -12° C (Cai and Zhang 2005). Moreover, the regenerated cellulose membranes and fibers have been prepared successfully (Cai et al. 2007b; Mao et al. 2006). However, native cellulose (cellulose I) having high viscositymolecular weight (M_n) $(M_n > 14 \times 10^4)$ could not be dissolved completely in 7 wt% NaOH/12 wt% urea aqueous solution pre-cooled to -12° C. Namely, the dissolution of cellulose is related to its molecular weight and conditions. A basic understanding of the effects of temperature and molecular weight of cellulose on its dissolution is essential for the successful development and application of cellulose. In this work, the cellulose samples having different M_{η} were dissolved in 7 wt% NaOH/12 wt% urea aqueous solution at temperature from 60 to -12.6°C, and their solubility and viscosity of the cellulose in the solution were investigated and discussed.

Experimental

Materials

Cellulose (cotton linters pulp) with an α -cellulose content of about 92% was supplied by Hubei Chemical Fiber Co. Ltd. (Xiangfan, China). The viscosity-average molecular weight (M_n) of the cellulose samples was 13.1×10^4 , 11.1×10^4 , 9.2×10^4 , 8.3×10^4 , 7.8×10^4 , 6.3×10^4 , 4.8×10^4 and 3.1×10^4 , and coded as C13, C11, C9, C8, C7, C6, C5 and C3, respectively. The cellulose samples were shredded and dried at 60°C for 24 h in a vacuum oven, stored in a desiccator until used. Two cellulose samples (C13, C9) were beated circularly in distilled water at 20°C by Valley beating machine (ZQS2-23, Shaanxi University of Science and Technology Machine Works, Xianyang, China) for 72 h to reduce their crystallinity, and then were dried upon bronze-net. All of the chemical reagents were purchased from commercial resources in China, and were of analytical grade.

Solubility test

7 wt% NaOH/12 wt% urea aqueous solution was prepared by mixing of NaOH, urea and distilled water (7:12:81 by weight). After 192 g solvent was prepared to a desired temperature, 8 g cellulose sample was added immediately into it with stirring vigorously for 10 min. The dispersed solution was then centrifuged at 8,000 rpm at 5°C for 2 min. The remaining undissolved fractions were washed using water and acetone, respectively, and then dried at 60°C for 24 h in a vacuum oven. Thus, the solubility of cellulose (S_a) in NaOH/urea aqueous solution was calculated by

$$S_{a} = (W_{0} - W_{i})/W_{0}$$
(1)

where W_0 is weight of original cellulose, and W_i is weight of the undissolved fractions.

Characterization

Intrinsic viscosity ([η]) of the dilute cellulose solution in LiOH/urea system was determined using a Ubbelohde capillary viscometer at 25 ± 0.1°C, and its M_{η} value was calculated according to [η] = $3.72 \times 10^{-2} M_{\rm w}^{0.77}$ (mL g⁻¹) (Cai et al. 2006). The zero-shear viscosity (η_0) of the cellulose solution was determined using the steady shear tests on an ARES-RFS III rheometer (TA Instrument, USA). Double-concentric cylinder geometry with a gap of 2 mm was used to measure shear viscosity as function of shear rate. Temperature control was established by connection with a julabo FS18 cooling/heating bath kept within 0.2°C over an extended time.

Optical microscope (Axiovert 200 M, ZEISS, Germany) was used to compare the morphological change and solubility of cellulose in the 7 wt% NaOH/12 wt% urea aqueous solution. The cellulose solutions were pressed between two glass slides directly after being prepared, and then sealed by paraffin to be observed and photographed. Wideangle X-Ray diffraction (WAXD) measurement was carried out with an X-Ray diffractometry (D/MAX-1200, Rigaku Denki Co. Ltd., Japan). The X-ray radiation used was Ni-filtered $\text{Cu-}K_{\alpha}$ with a wavelength of 1.5406 Å. The voltage was set at 40 kV and the current was set at 30 mA. The samples were mounted on a solid circular holder, and the proportional counter detector was set to collect data at a rate of $2\theta = 1^{\circ}$ /min over the 2θ range from 4° to 40° . All samples were cut into particle-like size to erase the influence of the crystalline orientation. The crystallinity (χ_c) of each sample was calculated according to the usual method (Rabek 1980).

Results and discussion

Influence of temperature on dissolution

The dissolution state of the cellulose in 7 wt% NaOH/12 wt% urea aqueous solution at different temperature has been investigated. Figure 1 shows the pictures of the cellulose solution dissolved at different temperature for 2 min. The results reveal that, at high temperature from 0°C to 25°C, cellulose changes hardly, and at 60°C it only swells in the solvent, showing fiber with diameter of 20-30 µm. By decreasing temperature from 0° C to -10° C, the swelling degree of the cellulose fiber increased. Transparent cellulose solution has occurred at -12.6° C, indicating that the cellulose has been dissolved completely. To clarify the change of solubility, the dependence of the S_a value on temperature for C9 in solvent is shown in Fig. 2. At temperature from 60°C to 10°C, the $S_{\rm a}$ value maintains in a constant (about 7.6-8.3%), indicating that the few fractions with the low molecular weight cellulose could be dissolved. The average molecular weight of undissolved fractions is close to that of its initial cellulose (9.2×10^4) as shown in Table 1. Clearly, the cellulose could not be dissolved at temperature above 10°C in NaOH/urea aqueous solution. However, with a decrease of the temperature from 10°C to -12°C, the S_a value of cellulose



Fig. 1 Optical microscope images of the C9 cellulose (4 wt%) dissolved in 7%NaOH/12urea aqueous solution at different temperature for 2 min



Fig. 2 Dependence of S_a value of cellulose (4 wt%, C9) on temperature from $-12.6^\circ C$ to $60^\circ C$

increased rapidly, and it could be dissolved in the range from -10 to -12.6° C. Ice will form in the solvent below -12.6° C (freezing point), which is a critical temperature (T_c) of this mixture solution. Although the solvent is a liquid-solid two-phase below T_c, cellulose could also be dissolved completely in the solvent system.

The relationship between S_a value and temperature can be described by the Arrhenius equation as

$$\ln S_a = \ln A - E_{a,s}/RT \tag{2}$$

where $E_{a,s}$ is the apparent activation energy of dissolution (kJ/mol), R is the molar gas constant (0.008314 kJ mol⁻¹ K⁻¹), T is the absolute temperature (K), and A is a preexponential factor (Su and

Table 1 Experimental results of solubility (S_a), viscosityaverage molecular weight (M_η) and crystallinity index (χ_c) of dissolved and undissolved fractions of cellulose (C9) in the solvent with different temperature

T, ℃	Dissolved fra	actions	Undissolved t	$S_a \ \%$		
	$\overline{M_\eta \times 10^{-4}}$	χc	$M_\eta \times 10^{-4}$	χc		
60	_	_	9.2	0.68	7.8	
25	-	-	9.3	0.67	7.8	
10	-	-	9.4	0.64	8.3	
0	4.5	_	9.6	0.55	17.7	
-6	6.0	_	10.6	0.55	49.6	
-10	8.2	_	10.7	0.49	88.9	
-12	8.7	_	10.0	0.51	97.7	
-12.6	8.9	0.62	_	_	100	
<-12.6	8.9	0.62	_	-	100	

Puls 1999). The $E_{a,s}$ could be obtained from the slope of a plot of lnS_a vs 1/T using linear least-squares analysis (Fig. 3). Interestingly, the calculated E_a value is -101 kJ/mol in the range from 5 to -10° C, as a result of the dissolution at low temperature. Usually, increasing the temperature can promote the dissolution of normal polymers, because of the solvation being enhanced by heat movement. However, the solubility of a little of aqueous soluble polymers decreases when temperature increases (Spelzini et al. 2005). Negative activation energies have been observed for dimer formation in selfcomplementary oligonucleotides (Craig et al. 1971) and in β -hairpin and α -helical formation in polypeptides (Muñoz et al. 1997; Lednev et al. 1999), which imply that the transition "state" along the effective reaction coordinate has a lower enthalpy than the random coil state, and that the free energy barrier arises from a significant loss of entropy (Ansari et al. 2001). The negative apparent activation energy in our system suggests that the cellulose solution state has a lower enthalpy than the solid cellulose, and the dissolution of cellulose in aqueous NaOH/urea solution pre-cooled to low temperature could be described as an entropy-driven process. When solid cellulose was immersed in NaOH/urea aqueous solution pre-cooled to -12° C, the hydrogen-bonded network structure between the cellulose macromolecules and small molecules in the solvent was created rapidly to form a inclusion complex, bringing cellulose into the aqueous solution (Cai et al. 2007a, b).

Figure 4 shows X-Ray diffractograms of the undissolved fractions at different temperature. The



Fig. 3 Arrhenius plots of temperature dependence of S_a value for 4 wt% cellulose solution (C9) in the range from 0 to -12° C

diffraction peaks at $2\theta = 14.8^{\circ}$, 16.3°, and 22.6° for $(1\overline{1}0)$, (110), and (200) planes are characteristic for cellulose I crystal, and those at $2\theta = 12.1^{\circ}$, 19.8° , and 22.0° for (110), (110), and (200) planes are signed to cellulose II crystal. The initial cellulose (C9) has typical crystalline peaks of cellulose I, whereas the dissolved fractions in cellulose solution exhibit peaks of the cellulose II. At 25°C the X-Ray diffractograms of undissolved fractions is similar to that of cellulose I, this further confirms that the solvent cannot destroy the structure of cellulose at above 25°C. However, with the decreasing of temperature, the diffraction peaks at $2\theta = 14.8^{\circ}$, 16.3° , and 22.6° corresponding to cellulose I in undissolved fractions gradually disappear, and those at $2\theta = 12.1^{\circ}$, 19.8°, and 22.0° corresponding to cellulose II increase. It is worth noting that the diffraction peaks corresponding to cellulose I disappear almost completely below -10° C, suggesting that they have changed to cellulose II. The results strongly indicate



Fig. 4 X-Ray diffractograms of undissolved fractions of cellulose (4 wt%, C9) at different temperature

that the structure of native cellulose could be destroyed in the present solvent system at low temperature. On the basis of the data, the degree of mercerization of the undissolved fractions increases with a decrease of temperature, indicating that the dissolution power of NaOH/urea aqueous solution on cellulose increases. The experimental results of M_n and χ_c of dissolved and undissolved fractions of C9 at different temperature are summarized in Table 1. The χ_c values of undissolved fractions decrease with the decreasing of temperature, which suggests that the crystalline structure of cellulose can be destroyed more easily by NaOH at low temperature. Furthermore, M_{η} of regenerated cellulose from the dissolved fractions is close to that of initial cellulose, indicating that no obvious degradation of cellulose occurred in the dissolution processes.

Influence of M_{η} on dissolution

Figure 5 shows the dependence of the S_a value on temperature in the range from -12.6° C to 15°C. The S_a values increase with a decrease of M_{η} of cellulose. At 15°C, the S_a value is about 8% and there is no obvious difference for the cellulose samples with different M_{η} , indicating that cellulose cannot be dissolved at this temperature. The C13 cellulose sample $(M_{\eta} = 13.1 \times 10^4)$ could not be dissolved completely even at -12.6° C, whereas C3 $(M_{\eta} = 3.1 \times 10^4)$ can be dissolved completely at about -9° C. The results indicate that solubility depends strongly on molecular weight of cellulose and temperature.



Fig. 5 Dependence of S_a value of cellulose (4 wt%) on temperature (a, C3; b, C5; c, C6; d, C7; e, C9; f, C11; and g, C13)



Fig. 6 Dependence of $E_{a,s}$ of cellulose dissolution on M_{η} in NaOH/urea aqueous solution

From Arrhenius plots of temperature dependence of S_a , the activation energy of dissolution ($E_{a,s}$) of the cellulose samples having different M_{η} has been calculated, and the result is shown in Fig. 6. Interestingly, the E_{a,s} values decrease rapidly with an increase of molecular weight of cellulose, which differ significantly from normal polymers. For normal polymer, with their long chains, the dissolution is slightly relative to the whole polymer molecular weight. This suggests that every glucose unit in cellulose contributes to the formation of their interand intra-molecular hydrogen bonds to remain the super-molecular structure. Thus, the high sensitivity of the $E_{a,s}$ value on M_{η} indicates that the cellulose dissolution has been controlled by their molecular weight. The E_{a,s} value for the C13 cellulose sample having $M_n = 3.1 \times 10^4$, which could not be dissolved completely even at -12.6°C, is -134 kJ/mol. This suggests that cellulose could not be dissolved completely in the solvent when E_{a.s} value was as low as -134 kJ/mol. Therefore, the cellulose samples having M_{η} below 10.0 × 10⁴ (E_{a, s} > -120 kJ/mol) could be dissolved completely in NaOH/urea aqueous solution pre-cooled to -12° C.

Influence of χ_c on dissolution

The solubility of the cellulose samples are summarized in Table 2. The S_a value at $-6^{\circ}C$ increases with the decreasing of M_n , supporting above conclusion that cellulose with lower M_n is more easily to be dissolved in this solvent. Interestingly, the solvent can dissolve cellulose (C13) with M_n of 9.8 \times 10⁴ and value of 0.67 at -6° C, whereas it cannot dissolve cellulose (C3) with M_n of 4.0×10^4 and value of 0.76 at the same condition. This can be explained that the relatively high χ_c could prevent the dissolution of cellulose. Figure 7 shows X-Ray diffractograms of undissolved fractions at -6° C, which indicates that there is both cellulose I and cellulose II. Although there is slightly decreasing with the decreasing of M_n , the χ_c value of undissolved fractions is almost the same.

Two cellulose samples (C13 and C9) are selected to be treated by Valley beating machine to change their crystallinity. Table 3 shows that M_n of the treated cellulose decrease slightly from 13.1×10^4 to 12.7×10^4 for C13 and from 9.2×10^4 to 9.0×10^4 for C9. However, the χ_c value of the treated cellulose significantly decreases from 0.62 to 0.53 for C13 and from 0.66 to 0.55 for C9. Figure 8 shows the dependence of the Sa value of the treated and untreated cellulose samples on temperature. The solubility of the treated cellulose samples increase obviously. It is noted that the treated C13 having 12.7×10^4 of M_η can be dissolved in the solvent pre-cooled to -12.6° C completely, as a result of the decreased crystallinity. We also obtained $E_{a,s}$ value of

42.3
45.2
49.6
63.0
75.0
85.1
94.2

Table 2 viscosity-



Fig. 7 X-Ray diffractograms of undissolved fractions of different cellulose at $-6^{\circ}C$

the cellulose dissolution from the slope of a plot of $\ln S_a$ vs 1/T using linear least-squares analysis (Table 3). The $E_{a,s}$ value of the treated cellulose is decreased remarkably from 134 kJ/mol to 85 kJ/mol for C13 and from 101 kJ/mol to 76 kJ/mol for C9. Obviously, the solubility of the cellulose can be enhanced by the decreasing of the χ_c value of cellulose. In view of the above results, we can improve the solubility of the cellulose in this solvent obviously through the physical treating of cellulose.

Solution behavior

It is difficult to disperse cellulose molecules in this solvent at the molecular level. The cellulose



Fig. 8 Dependence of the S_a value of cellulose on temperature: \Box , C13 (untreated); \blacksquare , C13 (treated); \bigcirc , C9 (untreated); \bullet , C9 (treated)

molecules in the NaOH/urea aqueous solution could form aggregates demonstrated by combined static and dynamic laser light scattering (Chen et al. 2007). The cellulose solution in NaOH/urea aqueous system could remain in a liquid state for a long time period at about 0 to 5°C (Cai and Zhang 2006). Figure 9 shows the M_η dependence of the zero-shear viscosity (η_0) at 0°C for the 4 wt% cellulose solution. We can obtain an equation following from the slope of a plot of log η_0 vs log M_η using linear least-squares analysis:

$$\eta_0 = 10^{-18.8} \times M_n^{4.01} \tag{3}$$

In view of the exponent ($\alpha = 4.01$), the η_0 of the cellulose solution increases more rapidly with the increasing of M_η than that of normal polymers. It indicates that the aggregation and entanglement exist among the cellulose chains, and increase rapidly with an increase of M_η in the cellulose solution at 0°C.

We need to control the viscosity of cellulose solution in producing regenerated cellulose products such as regenerated fibers and films. Thus, the cellulose concentration in the solution is important. Figure 10 shows dependence of the cellulose concentration (c) on M_{η} in the solvent system pre-cooled

Table 3 Comparison of solubility of cellulose being treated and untreated in solvents

Cellulose	Untreated	Untreated				Treated			
	$\overline{M_\eta \times 10^4}$	χc	E _{a,s} (kJ/mol)	S _a (%)	$M_\eta \times 10^4$	χc	E _{a,s} (kJ/mol)	S _a (%)	
С9	9.2	0.66	101	100	9.0	0.55	76	100	
C13	13.1	0.62	134	93	12.7	0.53	85	99	



Fig. 9 M_{η} dependence of the viscosity for the 4 wt% cellulose solution at 0°C



Fig. 10 M_{η} dependence of the proper concentration (c) for the cellulose dissolution in solvent pre-cooled to -12.6° C

to -12.6° C. It indicates that the polymer concentration increases with a decrease of M_{η} to achieve about 8 wt% for M_{η} of 3.1 × 10⁴. Therefore, we can reduce the molecular weight of cellulose to increase its concentration in the solution.

Conclusion

The solubility of cellulose in NaOH/urea aqueous solution strongly depended on temperature of solvent, molecular weight and crystallinity. All of the cellulose samples having below M_{η} below 10.0×10^4 had good solubility in aqueous NaOH/urea pre-cooled to -12.6° C. Interestingly, the E_{a,s} was negative value

and decreased rapidly with an increase of molecular weight of cellulose, indicating a high sensitivity of $E_{a,s}$ on M_{η} , which significantly differed from normal polymers. Moreover, the solubility of the cellulose could be enhanced by decreasing the χ_c value of cellulose. We could reduce M_{η} , χ_c to appropriate extent to obtain high concentration of the cellulose solution. The zero-shear viscosity of the cellulose solution increased more rapidly with the increasing of M_{η} than that of the normal polymers.

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