

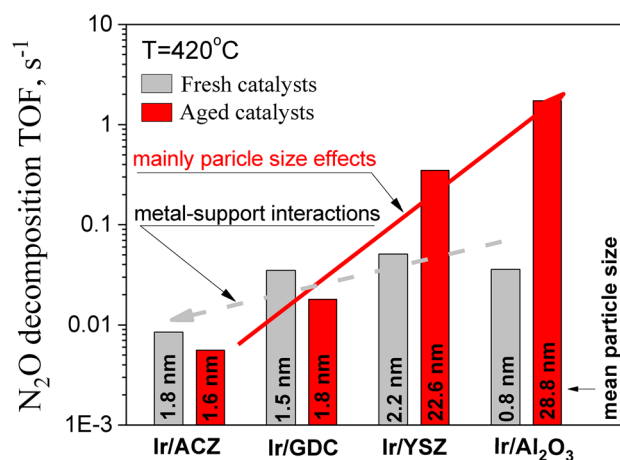
# Ir-Catalysed Nitrous oxide (N<sub>2</sub>O) Decomposition: Effect of Ir Particle Size and Metal–Support Interactions

I. V. Yentekakis<sup>1</sup> · G. Goula<sup>1</sup> · S. Kampouri<sup>1</sup> · I. Betsi-Argyropoulou<sup>1</sup> · P. Panagiotopoulou<sup>1</sup> · M. J. Taylor<sup>2</sup> · G. Kyriakou<sup>2</sup> · R. M. Lambert<sup>3,4</sup>

Received: 14 July 2017 / Accepted: 25 October 2017 / Published online: 13 November 2017  
© Springer Science+Business Media, LLC 2017

**Abstract** The effect of the morphology of Ir particles supported on  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>, 8 mol%Y<sub>2</sub>O<sub>3</sub>-stabilized ZrO<sub>2</sub> (YSZ), 10 mol%Gd<sub>2</sub>O<sub>3</sub>-doped CeO<sub>2</sub> (GDC) and 80 wt%Al<sub>2</sub>O<sub>3</sub>–10 wt%CeO<sub>2</sub>–10 wt%ZrO<sub>2</sub> (ACZ) on their stability on oxidative conditions, the associated metal–support interactions and activity for catalytic decomposition of N<sub>2</sub>O has been studied. Supports with intermediate or high oxygen ion lability (GDC and ACZ) effectively stabilized Ir nanoparticles against sintering, in striking contrast to supports offering negligible or low oxygen ion lability ( $\gamma$ -Al<sub>2</sub>O<sub>3</sub> and YSZ). Turnover frequency studies using size-controlled Ir particles showed strong structure sensitivity, de-N<sub>2</sub>O catalysis being favoured on large catalyst particles. Although metallic Ir showed some de-N<sub>2</sub>O activity, IrO<sub>2</sub> was more active, possibly present as a superficial overlayer on the iridium particles under reaction conditions. Support-induced turnover rate modifications, resulted from an effective double layer [O<sup>δ-</sup>–δ<sup>+</sup>](Ir) on the surface of iridium nanoparticles, via O<sup>2-</sup> backspillover from the support, were significant in the case of GDC and ACZ.

## Graphical Abstract



**Keywords** Nitrous oxide decomposition · Thermal aging · Nanoparticles sintering · Ostwald ripening · Oxygen storage capacity · Ceria

## 1 Introduction

N<sub>2</sub>O is a powerful greenhouse gas (global warming potential ~ 300) and also the dominant ozone-depleting gas of the twenty-first century [1, 2]. Stationary or mobile combustion sources and certain chemical processes, are the main sources of anthropogenic N<sub>2</sub>O emissions, whose abatement is therefore of global importance. In this context, the catalytic reduction of N<sub>2</sub>O to N<sub>2</sub> is a most promising approach [3].

A wide range of materials have been investigated as catalysts, including noble metals, transition metals, mixed oxides, perovskites and zeolites [4–8]. Among them, noble

✉ I. V. Yentekakis  
yyentek@isc.tuc.gr

<sup>1</sup> School of Environmental Engineering, Technical University of Crete, 73100 Chania, Crete, Greece

<sup>2</sup> European Bioenergy Research Institute, Chemical Engineering & Applied Chemistry, Aston University, Aston Triangle, Birmingham B4 7ET, UK

<sup>3</sup> Chemistry Department, Cambridge University, Cambridge, UK

<sup>4</sup> Instituto de Ciencia de Materiales de Sevilla (CSIC), Americo Vespucio 49, 41092 Seville, Spain

metals (NMs) exhibit good activity at intermediate temperatures so despite their high cost they are promising candidates for practical implementation [3, 4]. Rh exhibits the best catalytic activity, although Ru and Ir, significantly cheaper, are almost as good [7, 9]. However, a drawback of both Ir and Ru nanoparticle catalysts is their well known inferior thermal stability compared to Rh and Pt [10].

We have recently shown that the thermal sintering characteristics of Ir are strongly correlated with the lability of lattice oxygen in the oxide supports [11] and that metal–support interactions between Ir and oxide supports having high oxygen ion lability can strongly stabilize Ir nanoparticles at high temperatures under oxidizing conditions. It is well known that such supports can also strongly modify the electronic state of the active phase interfaced with them, and, depending on the electro-philic or –phobic character of the reactants, promote or inhibit the catalytic chemistry [12, 13]. Here we clarify the influence of these phenomena on the direct decomposition of  $N_2O$  by means of well characterized supported Ir catalysts whose particle size and morphology were controlled by aging. Specifically, we extend the general applicability of our model for the inhibition of sintering to include the important case of Ostwald Ripening. Furthermore, we report the first observation of a, starting from low temperature, counter-clockwise hysteresis phenomenon in *light-off* curves of Ir-catalysed  $N_2O$  decomposition and discuss its origin.

## 2 Experimental

### 2.1 Materials Preparation

The support materials were  $\gamma$ - $Al_2O_3$  (Engelhard), 8 mol% yttria stabilized zirconia, YSZ (Zirconia sales, UK Ltd.), 10 mol% gadolinia doped ceria, GDC (Anan Kasei Co, Ltd.) and laboratory made by co-precipitation ACZ (80 wt%  $Al_2O_3$ -10 wt%  $CeO_2$ -10 wt%  $ZrO_2$  mixed oxide). All were pre-treated in air at 800 °C for 2 h. Ir-dispersed catalysts, Ir/ $\gamma$ - $Al_2O_3$ , Ir/YSZ, Ir/GDC, and Ir/ACZ, were prepared by wet impregnation of the supports at 75 °C in a solution of  $IrCl_3 \cdot H_2O$  (Aber GmbH & Co.KG) followed by reduction in  $H_2/He$  flow to yield 1.0 wt% nominal Ir loading. The actual metal loading was determined by induced coupled plasma optical emission spectroscopy (ICP-OES). After water evaporation, suspensions were dried at 110 °C for 12 h and the resulting powders heated at 400 °C for 2 h under 50%  $H_2/He$  flow and then at 850 °C for 1 h under 1%  $H_2/He$  flow (hereafter denoted as *fresh* catalysts). This produced a very narrow Ir particle size distribution (1–2 nm), unlike other methods in which an oxidation step (instead of a reduction) is used at this stage leading to a broad Ir size distribution (from ~ 1 to > 150 nm) [9, 14]. In addition, this prolonged  $H_2$  treatment

allows the effective removal of residual chlorine resulting from the  $IrCl_3$  precursor used. In order to study the susceptibility of our catalysts toward sintering and the consequent influence on their de- $N_2O$  activity, pre-aged counterparts of the four catalysts were prepared from the fresh samples using the following protocol: two consecutive cycles of 1 h oxidation (20%  $O_2/He$ , 50  $cm^3 \text{ min}^{-1}$ , 750 °C) followed by 0.5 h reduction (2%  $H_2/He$ , 50  $cm^3 \text{ min}^{-1}$ , 750 °C) (hereafter referred to as *aged* catalysts).

### 2.2 Materials Characterization and Catalytic Evaluation

$N_2$  adsorption–desorption (BET), isothermal hydrogen chemisorption ( $H_2$ -chem.), hydrogen Temperature-Programmed Reduction ( $H_2$ -TPR), inductively coupled plasma optical emission spectroscopy (ICP-OES), high resolution transmission electron microscopy (HRTEM) and powder X-ray diffraction (PXRD) were used to determine the textural, structural and morphological characteristics of the fresh and aged catalysts. BET surface areas ( $S_{BET}$ ) and total pore volumes of degassed samples were obtained using a Nova 2200e Quantachrome with  $N_2$  as adsorbate. Isothermal  $H_2$  pulse-chemisorption measurements using  $H_2$  pre-reduced samples were acquired with a Quantachrome/ChemBet Pulsar TPR/TPD chemisorption analyzer with simultaneous use of a TCD and an Omnistar/Pfeiffer Vacuum mass spectrometer. To avoid  $H_2$  spillover effects the chemisorption experiments were conducted at 0 °C. This allowed determination of the H/Ir ratio (i.e. iridium dispersion) and the mean Ir crystallite size. Oxygen storage capacities (OSC) of the oxygen-saturated supports were determined by means of  $H_2$ -TPR measurements using the same apparatus. Iridium contents were determined by ICP-OES (Thermo Scientific iCAP 7400 duo) after microwave digestion (CEM–MARS microwave reactor) of the samples in 5 ml 70%  $HNO_3$  and 100 mg  $NH_4F$  (Sigma Aldrich,  $\geq 98.0\%$ ) at 190 °C followed by dilution in 10% aqueous  $HNO_3$ . Powder X-ray diffraction (PXRD) was performed on a Bruker D8 Advance diffractometer using monochromated  $Cu K_{\alpha 1}$  radiation ( $\lambda = 0.1542 \text{ nm}$ ). Electron microscopy experiments were performed with a JEOL 2100F High Resolution Transmission Electron Microscope (HRTEM), operated at 200 kV. Samples were dispersed in methanol and deposited on 300-mesh carbon supported copper grids and dried under ambient conditions.

Catalytic testing and stability measurements were carried out in a continuous flow apparatus equipped with a single-pass, tubular, quartz, fixed-bed reactor operated in both differential and integral modes. Catalyst bed temperature was measured via a centrally located K-type thermocouple. Reactor inlet and outlet gases were analysed by on-line gas chromatography (Shimadzu GC-2014, TC detector, He carrier, MS 5A and HayeSep D columns). Nitrogen and oxygen

mass balances on the basis of the reactor inlet N<sub>2</sub>O concentration and the outlet N<sub>2</sub> and O<sub>2</sub> concentrations measured by on-line GC were found to be better than 98% in all cases. The feed composition was always 1000 ppm(v) N<sub>2</sub>O balanced with He. Evaluation of catalyst performance as a function of temperature (*light-off* profiles) was carried out at 300–550 °C with a total feed flow rate of  $F_t = 150 \text{ cm}^3 \text{ min}^{-1}$ ; differential reactor operation for intrinsic activity evaluation (i.e., turnover frequency), was performed maintaining N<sub>2</sub>O conversion below ~ 15%.

### 3 Results and Discussion

#### 3.1 Textural, Structural and Other Characteristics of the Materials

The OSCs of the four supports and associated characteristics, are shown in Table 1.

In accord with the literature [15, 16],  $\gamma\text{-Al}_2\text{O}_3$  and YSZ showed negligible and very small OSC of 0 and 5.7, respectively, whereas ACZ and GDC exhibited substantial OSC of 116 and 185.5  $\mu\text{mol O}_2/\text{g}$ , respectively (Table 1). The higher capacity of GDC was expected as it contains a higher proportion of CeO<sub>2</sub> (~90%) than does ACZ (~10%). Both GDC and ACZ showed two broad peaks in their H<sub>2</sub>-TPR profiles [17]. For ACZ these peaks are located at ~630 °C (larger) and ~900 °C (smaller); For GDC the peaks appeared at the same temperatures but the high temperature peak was much larger than the low temperature one (Table 1). Note that the redox (H<sub>2</sub>-consumption) behavior of ACZ was activated at a significantly lower

temperature (~300 °C) than for GDC (~500 °C). The low temperature peak of CeO<sub>2</sub>-based materials is attributed to the reduction of small surface particles whereas the high temperature peak is assigned to the reduction of bulk CeO<sub>2</sub> [17]. We thus conclude that in the ACZ composite the CeO<sub>2</sub>-ZrO<sub>2</sub> species are preferentially located at the surface of the  $\gamma\text{-Al}_2\text{O}_3$  particles as a capping layer. As we shall see, both the magnitude of the OSC and its detailed characteristics have important implications for N<sub>2</sub>O decomposition catalysed by Ir particles supported on ACZ and GDC.

Textural characteristics of the supporting materials and of the fresh and aged catalysts are summarized in Table 2. Only marginal differences in the total surface areas, S<sub>BET</sub>, resulted from incorporation of Ir onto the supports. ACZ was an exception; the pronounced sensitivity of its surface area to high temperature treatments is well known [16, 18]. The supports span a wide range of S<sub>BET</sub> values, with YSZ lowest (ca. 5 m<sup>2</sup>/g) and  $\gamma\text{-Al}_2\text{O}_3$  highest (178 m<sup>2</sup>/g). High temperature aging resulted in only marginal changes (Table 2).

Table 3 summarizes the H<sub>2</sub>-chemisorption, PXRD and HRTEM data. All the fresh catalysts exhibited small Ir particles (~1–2 nm), regardless of the nature and surface area of the support, despite the large variation in the latter (Table 3). This most probably results from the low Ir loading and the reduction used in the preparation protocol. In contrast, the aged samples showed a very strong dependence of Ir catalyst particle size on the identity of the support. Ir/ $\gamma\text{-Al}_2\text{O}_3$  underwent extensive growth of the Ir particles from 0.8 to 28.8 nm; Ir/YSZ significantly less (2.2 → 22.6 nm); GDC and ACZ were almost unaffected (1.5 → 1.8 and 1.8 → 1.6, respectively).

**Table 1** Total oxygen storage capacity (OSC) of the supports and their characteristics

Support	Total OSC ( $\mu\text{mol O}_2/\text{g}$ )	Temperature of redox properties activation (°C)	Low temperature peak (°C)	High temperature peak (°C)
$\gamma\text{-Al}_2\text{O}_3$	0.0	–	–	–
YSZ	5.7	425	505	–
ACZ	116.0	300	630 (big)	900 (small)
GDC	185.5	500	630 (small)	900 (big)

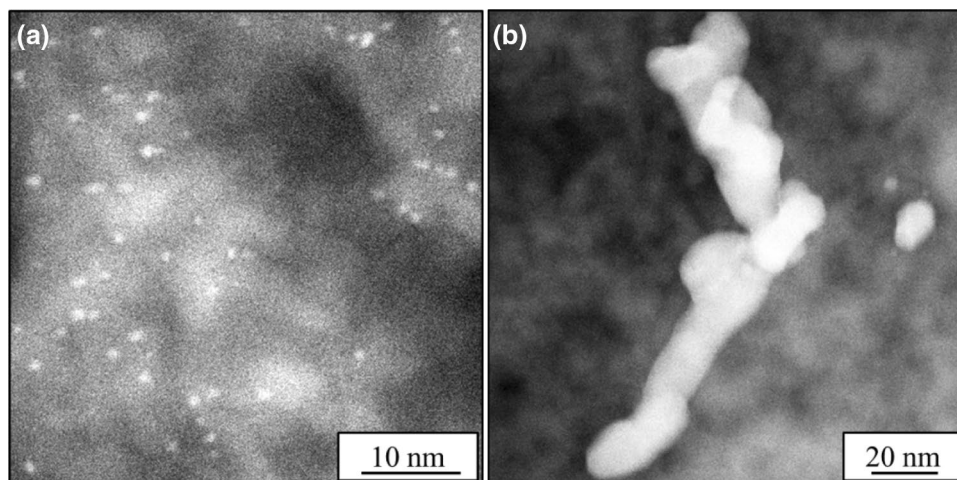
**Table 2** Textural properties of the supports and fresh and aged catalysts

Catalyst	wt%Ir <sup>a</sup>	S <sub>BET</sub> (m <sup>2</sup> g <sup>-1</sup> )			Total pore volume (cm <sup>3</sup> g <sup>-1</sup> )			Average pore size diameter (nm)		
		Support	Fresh cat.	Aged cat.	Support	Fresh cat.	Aged cat.	Support	Fresh cat.	Aged cat.
Ir/ $\gamma\text{-Al}_2\text{O}_3$	0.7	178	159	140	0.600	0.550	0.310	13.5	13.9	8.9
Ir/YSZ	0.7	5	4	4	0.009	0.008	0.009	6.7	8.2	7.8
Ir/GDC	0.4	10	10	10	0.026	0.036	0.016	9.9	14.1	6.1
Ir/ACZ	0.4	149	73	64	0.290	0.216	0.203	7.9	11.9	12.7

<sup>a</sup>Iridium content determined by ICP-OES

**Table 3** Ir particle size characteristics of the fresh and aged catalysts

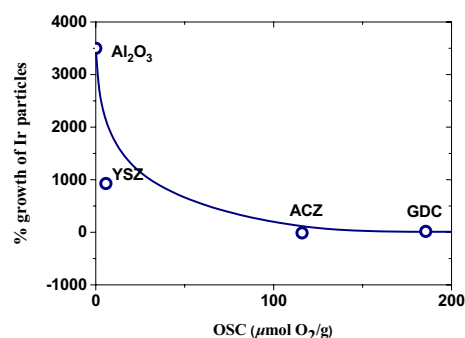
Catalyst	H <sub>2</sub> -uptake (m <sup>3</sup> g <sup>-1</sup> )		Mean Ir particle size (nm)			Dispersion (H/Ir)	
	Fresh cat.	Aged cat.	H <sub>2</sub> -Chem fresh/aged	PXRD fresh/aged	HRTEM fresh/aged	Fresh cat.	Aged cat.
Ir/ $\gamma$ -Al <sub>2</sub> O <sub>3</sub>	0.392	0.011	0.8/28.8	n.a./28.4	1.2 ± 0.4/14.5 ± 6.5 (max. 150)	0.88	0.03
Ir/YSZ	0.145	0.014	2.2/22.6	n.a./28.7	1.2 ± 0.4/10.5 ± 5.5 (max. 34)	0.33	0.03
Ir/GDC	0.121	0.103	1.5/1.8	n.a./n.a	1.7 ± 0.5/2.0 ± 1.0	0.48	0.41
Ir/ACZ	0.103	0.114	1.8/1.6	n.a./n.a	1.6 ± 0.5/1.9 ± 0.4	0.41	0.45

**Fig. 1** HRTEM dark field images of the *fresh* and *aged* Ir/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> samples

The particle sizes derived from H<sub>2</sub>-chemisorption were very close to those estimated from PXRD data Scherrer analysis, although in the case of very small particles Ir were not detectable by PXRD. On the other hand, pre- and post aging particle sizes obtained from HRTEM are in broad agreement with those obtained from the other two methods; the apparent discrepancy in the cases of aged Ir/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> and Ir/YSZ is due the presence of individual particles together with large aggregates of particles with maximum sizes of about 150 and 34 nm, respectively (for Ir/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> see Fig. 1a, b). Similarly, in aged Ir/YSZ, most of the Ir particles had undergone significant changes in shape and substantial increases in size (10.5 nm) compared to the fresh sample. *In contrast, the nanoparticles supported on the aged Ir/GDC and aged Ir/ACZ retained their original small size and uniform spatial distribution.*

The susceptibility to sintering of Ir particles on  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> under oxidizing conditions due to the high volatility of IrO<sub>2</sub>, is well known [10]. In contrast, metallic Ir particles are very stable during high temperature treatments in reducing atmospheres [10, 11].

The key point arising from our results is that the strong resistance to sintering in an oxidising environment of Ir particles on supports with high oxygen ion lability (GDC and ACZ) is dramatically different from their vulnerable behaviour on supports with negligible or low oxygen ion lability ( $\gamma$ -Al<sub>2</sub>O<sub>3</sub> and YSZ). This is clearly demonstrated in Fig. 2,

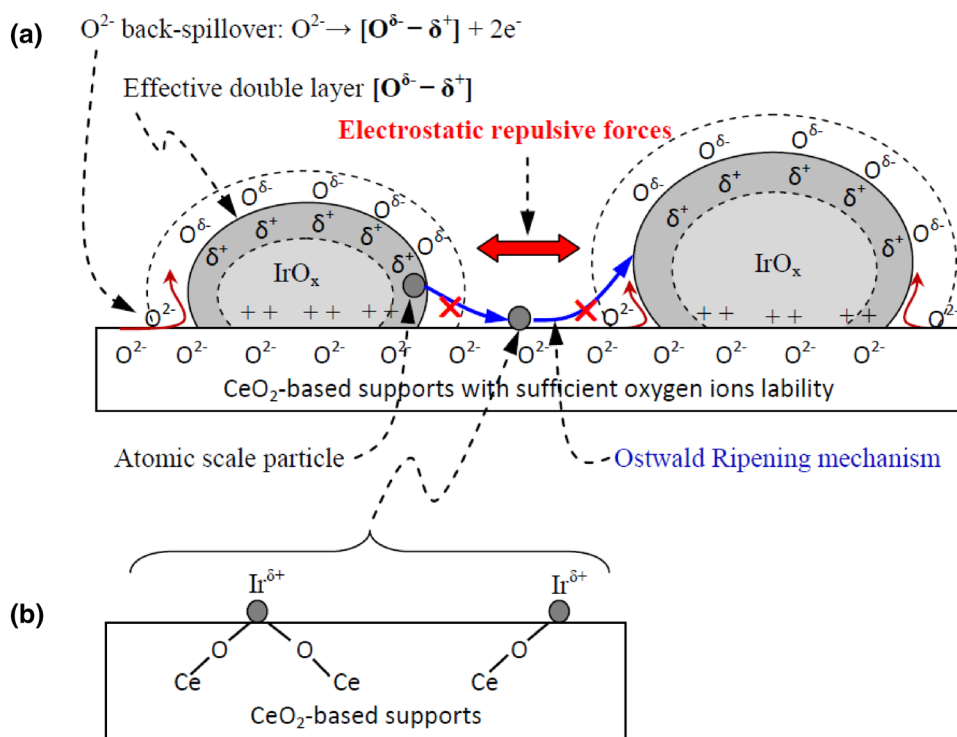
**Fig. 2** Correlation of the OSC of the support with the sintering behaviour of supported iridium particles

which depicts the percentage of sintering (i.e., % growth of Ir particles =  $100 \times (PS_{\text{aged}} - PS_{\text{fresh}}) / PS_{\text{fresh}}$ ; PS: particle size) resulting from oxidative aging of the catalysts versus the OSC of the supports. Clearly, there is a strong correlation between the thermal sintering of Ir nanoparticles and the OSC of the support.

We have previously proposed a plausible model to explain the effect of the support on catalyst sintering characteristics [11] summarized briefly below (Fig. 3). The model is further expanded in the context of the present work. An effective electrical double layer,  $[O^{\delta-} - \delta^+](M)$ , forms on the catalyst nanoparticle (M) due to lattice oxygen spillover



**Fig. 3** **a** Creation of the effective double layer on Ir particles and its influence on the particles sintering susceptibility through the created electrostatic repulsive forces as well as through the inhibition of the Ostwald Ripening mechanism, **b** labile lattice oxygen acts as a trap for diffusing atomic Ir species



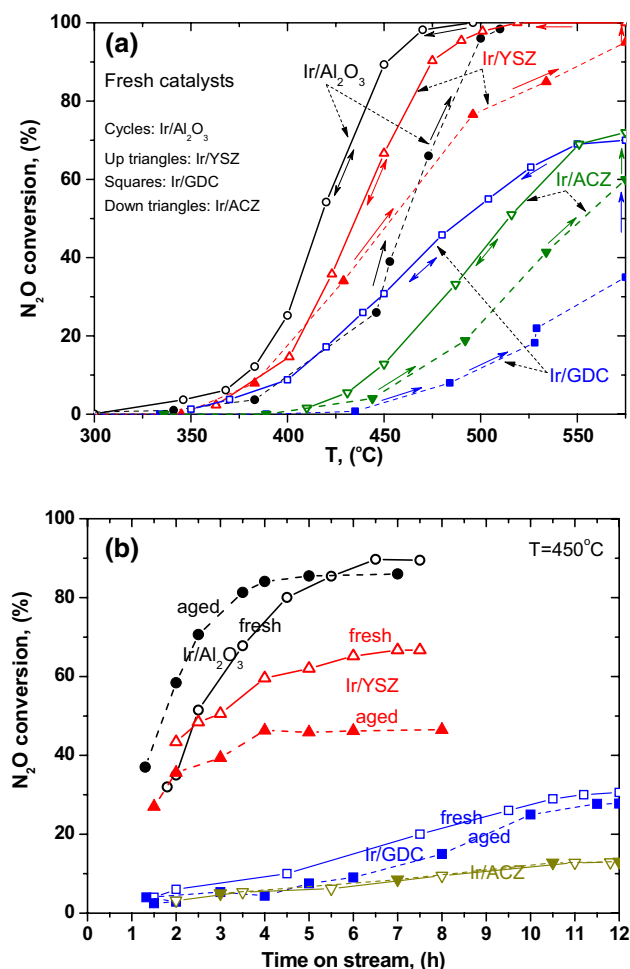
from supports with substantial oxygen ion lability [12, 13]. It should be noted here that the aging protocol followed involves temperatures much greater to the Tamman temperature of  $IrO_2$ , which is  $\sim 413$  °C [19], and is associated with the onset of mobility of metal particles on the surface [20]. Thus small iridium nanoparticle possessing a surface oxide overlayer is expected to be, in principle, highly mobile (therefore prone to sintering) on the surface of all oxide supports investigated in this study, as indeed confirmed experimentally for iridium nanoparticles dispersed on  $\gamma$ - $Al_2O_3$  [10].

However, the effective double layer  $[O^{\delta-} - \delta^+]$ , created on supports with high oxygen ions lability (Fig. 3a), endows the particle with a net negative charge at its gas-exposed surface, leading to inter-particle repulsion at short range, thus preventing particle-particle encounters followed by coalescence. Such spillover effects are expected to be strongest for supports with the highest oxygen ion lability. In the present case this effect could operate with both metallic  $Ir^0$  and  $IrO_2$  nanoparticles or composites of the two, since  $IrO_2$  is an electronic conductor. In the case of very small Ir particles, labile lattice oxygen creates vacancies that act as traps which inhibit diffusion and agglomeration (Fig. 3b) under a concept consistent to that described in [21, 22]. It is also expected that the  $O^{\delta-}$  modified surface barrier increases the activation energies for detachment (diffusion) and re-attached of the entities whose transport would otherwise lead to continuous growth of large particles at the expense of smaller particles according to the Ostwald Ripening model

(see Fig. 3). All aforementioned effects will be negligible for low-lability supports (e.g.  $Al_2O_3$  and YSZ), which therefore do not prevent particles agglomeration whatever the sintering mechanism is (large particles diffusion and coalescence or Ostwald Ripening). It is worth noting that pure  $CeO_2$  has limited lattice defects concentration in comparison to  $Gd_2O_3$ - or  $ZrO_2$ -doped  $CeO_2$ . Therefore the model suggests that optimal supports against sintering are those with sufficiently high  $O^{2-}$  ions mobility i.e.  $CeO_2$  modified with cations such as gadolinium, zirconium and others, that enhance ceria lattice defects, increasing  $O^{2-}$  vacancies and consequently  $O^{2-}$  mobility [16, 18].

### 3.2 N<sub>2</sub>O Decomposition

Figure 4a shows the *light-off* behaviour of all fresh catalysts ( $Ir/\gamma$ - $Al_2O_3$ ,  $Ir/YSZ$ ,  $Ir/ACZ$  and  $Ir/GDC$ ).  $N_2O$  conversion profiles correspond to Ir-free supporting materials are not included in the figure; in accord to literature  $\gamma$ - $Al_2O_3$ ,  $CeO_2$  and modified  $CeO_2$  oxide supports similar to that of the present study are almost inactive in direct  $N_2O$  decomposition ( $N_2O$  conversions  $< 10\%$ ) at the temperature range under investigation (300–550 °C) [23, 24], whilst Ir is one of the most active catalysts for  $N_2O$  decomposition from the Pt-group metals family [9]. Starting with a pre-reduced catalyst a significant counter-clockwise hysteresis during heating-cooling is clear for all samples. Holding the sample at maximum temperature until a true steady state was reached caused the system to follow a higher activity, and



**Fig. 4** **a** The *light-off* behavior of the fresh Ir/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>, Ir/YSZ, Ir/ACZ and Ir/GDC catalysts showing *counter-clockwise* hysteresis. **b** Time-on-stream stability of all fresh and aged catalysts starting with pre-reduced samples. Other conditions: catalyst mass,  $w_{\text{cat}} = 50$  mg; total inlet flow rate,  $F_1 = 150$  cm<sup>3</sup> min<sup>-1</sup>; N<sub>2</sub>O concentration, 1000 ppm(v) N<sub>2</sub>O/He

subsequently reproducible, curve upon cooling (Fig. 4a). This hysteresis indicates changes in the catalyst state, further demonstrated in Fig. 4b, which depicts the time-on-stream stability of all fresh and aged catalysts at 450 °C. Starting with pre-reduced catalysts, in every case N<sub>2</sub>O conversion increased progressively with time-on-stream, stabilizing at maximum activity after ~5–7 h (for Ir/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> and Ir/YSZ)

and ~10–12 h (for Ir/GDC and Ir/ACZ). Since at ~450 °C particle growth is not favoured [10], the increase in catalytic activity may be attributed to progressive oxidation of the initially metallic particles under the mildly oxidizing conditions (1000 ppm N<sub>2</sub>O), leading to formation of IrO<sub>2</sub> or a superficial overlayer of IrO<sub>2</sub>, likely the catalytically active phase, as proposed by Ohnishi et al. [14].

To avoid pseudo steady-state, all subsequent catalytic measurements for both differential and integral modes of reactor operation were carried out on catalysts that had been stabilized after sufficient time-on-stream. Table 4 lists the temperatures for 20, 50 and 80% conversion ( $T_{20}$ ,  $T_{50}$  and  $T_{80}$ , respectively) by the four catalyst pairs, obtained from the corresponding steady state *light-off* profiles. Note that  $T_{50}$  values of aged samples are close to those of fresh samples; and unexpectedly in the case of Ir/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> the  $T_{50}$  value of the aged sample is *lower* than that of the fresh sample—i.e. aged Ir/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> was more active than its fresh counterpart. As we shall see, this behaviour can be rationalized by taking account of the significant particle size dependence of the intrinsic turnover activity of Ir-catalysed N<sub>2</sub>O decomposition.

Note also that the activities of the fresh catalysts, which have quite similar particle sizes, are in the order Ir/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> > Ir/YSZ > Ir/GDC > Ir/ACZ, as judged by their  $T_{50}$  values (Table 4). This sequence is almost the exact opposite to the OSC sequence of the supports ( $\gamma$ -Al<sub>2</sub>O<sub>3</sub> < YSZ < ACZ < GDC) pointing to the presence of metal–support interactions: the higher the oxygen ion lability of the support the higher the inhibition of the de-N<sub>2</sub>O activity of the catalyst. In passing, we note that the inverted order between GDC and ACZ is understandable, because although ACZ has lower oxygen storage capacity than GDC, its redox properties are activated at significantly lower temperature (300 vs. 500 °C; Table 1). Accordingly, in the temperature interval 350–550 °C, within which the N<sub>2</sub>O decomposition reaction was investigated, the oxygen lability of ACZ actually exceeds that of GDC. In passing, we note that in the temperature range used all the supports were essentially inactive towards N<sub>2</sub>O decomposition. In accord with this, there was no correlation between the BET areas and catalyst activity.

Differential reactor operation (< 15% N<sub>2</sub>O conversion) at 420 °C provided turnover frequency (TOF) values for

**Table 4**  $T_{20}$ ,  $T_{50}$ ,  $T_{80}$  and TOF values for *fresh* and *aged* catalysts at 420 °C

Catalyst	$T_{20}/T_{50}/T_{80}$ (°C)		TOF (s <sup>-1</sup> )		TOF <sub>aged</sub> /TOF <sub>fresh</sub>
	Fresh cat.	Aged cat.	Fresh cat.	Aged cat.	
Ir/ $\gamma$ -Al <sub>2</sub> O <sub>3</sub>	394/417/442	378/407/441	$3.6 \times 10^{-2}$	$173 \times 10^{-2}$	48.0
Ir/YSZ	408/435/465	422/453/483	$5.1 \times 10^{-2}$	$35 \times 10^{-2}$	6.9
Ir/GDC	427/490/>575	437/490/>575	$3.5 \times 10^{-2}$	$1.8 \times 10^{-2}$	0.5
Ir/ACZ	466/515/>575	464/522/>575	$0.85 \times 10^{-2}$	$0.56 \times 10^{-2}$	0.7

the fresh and aged catalysts as shown in Table 4. Striking differences between the low and high oxygen ion lability supports is obvious (Table 4). In particular, aged Ir/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> exhibits a much higher TOF than its fresh counterpart (ca. 48 times higher). TOF<sub>aged</sub>/TOF<sub>fresh</sub> is also substantial for Ir/YSZ (~7), whereas only small differences were observed for the Ir/GDC and Ir/ACZ catalyst pairs. This is consistent with the aforementioned sintering behaviour of the catalysts and the known particle size dependence of Ir catalysts in N<sub>2</sub>O decomposition: large particles are more active than small ones [14] as we also find here (Table 4). This is also consistent with the view that the influence of the particle size on the de-N<sub>2</sub>O activity of Ir-based catalysts is far more important than effects due to metal–support interactions. This is apparent from comparison of the TOF<sub>aged</sub>/TOF<sub>fresh</sub> ratio values of Ir/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> and Ir/YSZ with that of Ir/GDC and Ir/ACZ, suggesting that the O<sup>δ-</sup> layer created on catalyst particles interfaced with intermediate or high oxygen ion lability supports acts to inhibit de-N<sub>2</sub>O activity. According to Vayenas [12] this effective O<sup>δ-</sup> layer (Fig. 3) is expected to weaken the chemisorption bond of electron acceptor adsorbates such as N<sub>2</sub>O<sub>ads</sub>, and O<sub>ads</sub> arising from N<sub>2</sub>O dissociation. To a first approximation, weakening the chemisorption bonds of both N<sub>2</sub>O<sub>ads</sub> and O<sub>ads</sub> should have opposite effects on the reaction rate in terms of the well-established mechanism for N<sub>2</sub>O dissociation over noble metals:



Weakening both the N<sub>2</sub>O<sub>ads</sub> and O<sub>ads</sub> chemisorption bonds by oxygen spillover should inhibit reaction (1) whilst also promoting reaction (2). In the present case, the first effect predominates.

## 4 Conclusions

Ir nanoparticles on a variety of supports show thermal sintering characteristics that are very strongly correlated with the lability of lattice oxygen in the supporting oxide material: the higher the lability of oxygen in the support, the lower the vulnerability of the nanoparticles to sintering under oxidative conditions. Oxygen back-spillover from the support to iridium particles causes this resistance to sintering. The same species are responsible for the appearance of metal–support interactions, which cause inhibiting effects on the de-N<sub>2</sub>O activity of the iridium particles. These effects

are however of significantly less importance than particle size effects on the structure sensitive de-N<sub>2</sub>O reaction. The catalytically active phase in direct N<sub>2</sub>O decomposition by Ir-based catalysts is IrO<sub>2</sub> or a superficial overlayer of IrO<sub>2</sub>.

**Acknowledgements** IVY acknowledges Technical University of Crete for partial financial support. MJT acknowledges Aston University for a PhD scholarship. GK acknowledges funding from the Royal Society and EPSRC. The authors would like to thank Dr C.M.A. Parlett and Dr M.A. Isaacs for their kind assistance with the TEM measurements.

## References

1. Ravishankara AR, Daniel JS, Portmann RW (2009) *Science* 326:123
2. Li L, Xu J, Hu J, Han J (2014) *Environ Sci Technol* 48:5290
3. Centi G, Perathoner S, Vazzana F, Marella M, Tomaselli M, Mantegazza M (2000) *Adv Environ Res* 4:325
4. Liu Z, He F, Ma L, Peng S (2016) *Catal Serv Asia* 20:121
5. Tolman WB (2010) *Angew Chem Int Ed* 49:1018
6. Kapteijn FJ, Rodriguez-Mirasol JA, Moulijn JA (1996) *Appl Catal B* 9:25
7. Li Y, Armor JN (1992) *Appl Catal B* 1:L21
8. Russo N, Mescia D, Fino D, Saracco G, Specchia V (2007) *Ind Eng Chem Res* 46:4226
9. Pachatouridou E, Papista E, Iliopoulou EF, Delimitis A, Goula G, Yentekakis IV, Marnellos GE, Konsolakis M (2015) *J Environ Chem Eng* 3:815
10. Fiedorow RMJ, Chahar BS, Wanke SE (1978) *J Catal* 51:193
11. Yentekakis IV, Goula G, Panagiotopoulou P, Kampouri S, Taylor MJ, Kyriakou G, Lambert RM (2016) *Appl Catal B* 192:357
12. Vagenas CG (2013) *Catal Lett* 143, 1085
13. Vernoux P, Lizarraga L, Tsampas MN, Sapountzi FM, De Lucas-Consuegra A, Valverde J-L, Souentie S, Vayenas CG, Tsiplakides D, Balomenou S, Baranova EA (2013) *Chem Rev* 113:8192
14. Ohnishi C, Iwamoto S, Inoue M (2008) *Chem Eng Sci* 63:5076
15. Duprez D, Descorme C, Birchem T, Rohart E (2001) *Top Catal* 16/17:49
16. Ivanova AS (2009) *Kinet Catal* 50:797
17. Damyanova S, Bueno JMC (2003) *Appl Catal A* 253:135
18. Montini T, Melchionna M, Monai M, Foriasiero P (2016) *Chem Rev* 116:5987
19. Shi H, Li X, Haller GL, Gutierrez OY, Lercher JA (2012) *J Catal* 295:133
20. Hansen TW, Delariva AT, Challa SR, Datye AK (2013) *Acc Chem Res* 46:1720
21. Hatanaka M, Takahashi N, Tanabe T, Nagai Y, Dohmae K, Aoki Y, Yoshida T, Shinjoh H (2010) *Appl Catal B* 99:336
22. Dvorak F, Camellone MF, Tovt A, Tran N-D, Negreiros FR, Vorokhta M, Skala T, Matolinova I, Myslivecek J, Matolin V, Fabris S (2016) *Nature Commun* 7:10801
23. Parres-Esclapez S, Illan-Gomez MJ, Salinas-Martinez de Lecea C, Bueno-Lopez A (2010) *Appl Catal B* 96:370
24. Esteves P, Wu Y, Dujardin C, Dongare MK, Granger P (2011) *Catal Today* 176:453