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Excellent quality microchannels for rapid microdevice prototyping: direct CO₂ laser writing with efficient chemical postprocessing

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Abstract

Rapid, simple microchannel prototyping is critical for the development of modern microfuidic devices and platforms. Laser cutting (ablation) using a commercially available continuous wave $(CW) CO₂$ laser followed by thermal bonding is one of the most common approaches for prototyping in thermoplastics such as polymethyl methacrylate (PMMA). However, this technique sufers from poorly controlled channel quality, inconsistent results from solvent-based post-processing, and inconsistency of thermal bonding. We have overcome these challenges through a systematic study of channel ablation in PMMA using a CW $CO₂$ laser. A new solvent treatment approach results in clearly improved microchannel quality and processing consistency, with negligible residual solvent. Thermal bonding of the processed material showed fourfold increase in bonding strength with full retention of PMMA's favourable optical clarity. As proof of concept, a high-quality three-layered microfuidic prototype is fabricated with this new method and its performance demonstrated.

1 Introduction

The rapid pace in development of micro-total analysis systems (µTAS) has led to the rise in availability of point-ofcare and point-of-site microfuidic diagnostic tools. These

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devices are able to integrate and perform several diferent functions simultaneously, such as precise metering of reagents and samples, sample preparation, sorting and processing, and detection of biomarkers (Bhagat et al. [2010](#page-11-0); Strohmeier et al. [2015](#page-12-0); Tang et al. [2016\)](#page-12-1). These microfuidic devices offer many advantages, including automation of analytical procedures, cost efectiveness, ease of use, repeatable accuracy and reduced processing time, making them perfect candidates to dominate the diagnostic test industry in the next decade (Wlodkowic and Darzynkiewicz [2011;](#page-12-2) Gomez [2013](#page-11-1); Streets and Huang [2013](#page-12-3); Sackmann et al. [2014](#page-11-2)). The main type of material used in these types of diagnostic μ TAS devices is thermoplastics, largely due to their relatively low cost and ability to readily translate smoothly into industrial processes (Sackmann et al. [2014](#page-11-2)).

Several methods such as hot embossing, injection molding and micro-thermoforming that have been applied to microfluidic chip manufacture can be transferred to industry, as they allow for fast and inexpensive fabrication on an industrial scale (Boone et al. [2002](#page-11-3); Attia et al. [2009;](#page-11-4) Focke et al. [2011\)](#page-11-5). However, these methods become too expensive and time consuming when used for initial development and fast-prototyping of new designs of microfuidic circuits. The use of thermoplastics for research development in microfuidics is thus often avoided despite the overwhelming advantages in technology transfer (Aghvami et al. [2017\)](#page-10-0). However,

direct laser writing on polymers can speed up the technological transfer of these devices from initial development to the fnal point-of-care products seen in clinical settings, thus is a good candidate for rapid prototyping in the initial research stage (Klank et al. [2002;](#page-11-6) Cheng et al. [2004;](#page-11-7) Malek [2006](#page-11-8); Hong et al. [2010](#page-11-9)).

Several low-cost commercial continuous wave $(CW)CO₂$ laser engravers are commercially available (for example, Epilog®, Gravograph®, Mactron®, Trotec®, Universal® and Xenetech®, among others) and do not require high levels of user training for efficient operation (Moreira Gabriel et al. [2014](#page-11-10)). These lasers are quite suitable for laser micromanufacturing in inexpensive thermoplastics. The use of such substrates allows for easy and direct technology transfer from research laboratories to industry which has a manufacturing and economical preference for these materials (Chin et al. [2012\)](#page-11-11). In particular, poly(methyl methacrylate) (PMMA) has become one of the main polymers for production of µTAS chips due to its high mechanical and chemical stability, excellent dielectric properties, optical transparency and low cost (Chen et al. [2008](#page-11-12); Urech and Lippert [2010](#page-12-4)).

When $CO₂$ laser machining is applied to a polymer, photochemical and photo-thermal processes occur that result in thermal, thermo-oxidative or mechanical breakdown (Chen et al. [2008](#page-11-12)). The impinging gas or plasma jet vaporizes fragments of the macromolecules (down to oligomers) or causes breakaway of particles from the polymer and fller.

PMMA is well suited for $CO₂$ laser machining since it is possible to achieve 100% monomers as the by-products (Urech and Lippert [2010](#page-12-4); Said-Galiev and Nikitin [1992](#page-12-5)). However, a number of quality issues can arise in direct writing of channels in PMMA using laser machining, particularly when using inexpensive CW $CO₂$ laser machines. The main problems contributing to poor quality microchannels are laser power instability and low translational resolution of the laser platform and mechanical backbone in these machines, besides mechanisms inherent in the laser ablation itself. Debris can be ejected the instant the laser ablates the material before re-melting can occur. Other problems are poor edge quality of the walls and borders (with formation of cracks) and poor smoothness of the channel walls. As such, the quality of the channels generated solely by laser machining is unsatisfactory for reliable prototyping. Several studies have attempted to optimize settings to improve $CW CO₂$ laser machining process but have used a modelling rather than an empirical approach, or lacked detailed consideration of reproducibility of the channels and fuid flow, or have limited the focus to the channel dimensions and not the quality (Klank et al. [2002](#page-11-6); Nayak et al. [2008](#page-11-13); Hong et al. [2010;](#page-11-9) Romoli et al. [2011](#page-11-14); Kant et al. [2015;](#page-11-15) Prakash and Kumar [2015\)](#page-11-16).

Thus, laser ablation of PMMA and other thermoplastics in isolation generates relatively poor quality channels, and it is unfortunately unsatisfactory for reliable prototyping when used in the absence of post processing steps. A better understanding of the thermodynamics and the efect of diferent power settings and scan speeds could be used to generate efective post-treatment methods for improving the roughness and quality of the channels. To this end, we constructed an extensive database from empirical information, not just of the dimensions of the channels, but also on their quality regarding different aspects related to $CO₂$ laser machining.

PMMA is an amorphous polymer, hence it does not melt at a single, well-defned temperature, but instead softens at elevated temperatures, having a glass transition temperature (T_o) between 100 and 122 °C depending on grade or supplier (Zhu et al. [2007\)](#page-12-6). Various studies have explored the T_g of PMMA to optimize the thermal bonding or have used temperature as a variable to improve the quality of the channels (Chen et al. [2003;](#page-11-17) Lin [2003](#page-11-18); Nikcevic et al. [2007;](#page-11-19) Li et al. [2008;](#page-11-20) Ogończyk et al. [2010\)](#page-11-21). The laser cutting process relies on efective use of the energy delivered as heat to allow the flow of the polymer and to decrease the roughness of the microchannel area on the polymer surface. In addition to a smooth microchannel, the transparency of polymer is important, especially around the detection area of microfuidic devices used in colorimetric, absorbance and fuorescence analytical applications (Gómez-de Pedro et al. [2017](#page-11-22); Lefèvre et al. [2015](#page-11-23)).

The importance of creating smoother microchannels cannot be underestimated for the future of microfuidics and nanofuidics felds. The smoothness of microchannels has been shown to highly affect the shear viscosity and the hydrodynamic boundary slip during fuid fow (Asproulis and Drikakis [2011](#page-11-24); Papanikolaou et al. [2016,](#page-11-25) [2017\)](#page-11-26). Furthermore, models on the infuence of the microchannel wall stifness during fuid fow have shown to afect both the slip process and the thermal equilibrium at the solid–liquid interface (Asproulis and Drikakis [2010\)](#page-11-27). Hence, diferent polymer stifness and the roughness of the micro or nanomicrochannel walls are important factors for a controlled fluid flow. These studies indicate that the use of channel smoothing techniques in stif polymers that have an easier transition to industry, such as PMMA, can give rise to fastprototyping nanofuidic chips using laser direct writing.

Whilst there is scope for optimization of the laser parameters, additional post-machining treatments are necessary to produce channels of quality suitable for prototyping. Solvent treatments show promise for post-ablation treatment. In contrast to thermal treatments, solvents act only on the surface of the polymer and therefore do not afect the bulk physical or chemical properties of the polymer. However, controlling the rate of evaporation and distribution of the solvents on the channels in both solvent vapour and liquid solvent treatments is challenging. Here we developed a reproducible solvent treatment enabling careful control of the solvent–polymer dissolution, by total immersion of the polymer in a solvent bath at controlled temperature. This method delivers reproducible results because it does not depend on external factors such as distance from solvent vapour to the area to be treated, external temperature, or rate of evaporation.

In this paper we present an extensive database of $CO₂$ laser settings in terms of laser scan speed and power, with evaluation of the resulting microchannel quality and dimensions in PMMA. We additionally describe a novel chemical (solvent) surface treatment that both improves the quality of the channels and increases the bond strength of three-layerbased microfuidic chips, while keeping the detection area of the device transparent.

2 Materials and methods

2.1 Sample manufacturing

The substrates used for laser ablation were clear PMMA 2 mm thick sheets (PSP Ltd., Auckland, NZ). The laser system used was a CW $CO₂$ laser (Universal Laser Systems, VLS3.50, 30 W, 10.6 µm) with a lens working distance of 50.8 mm, generating a 127 µm spot size. These conditions were maintained throughout the experiment. Using this system, 20 mm straight lines were machined using variable power settings and scan speeds that ranged from 1.2 to 10.2 W (16 steps in total) and $20-175$ mm/s (13 steps in total), resulting in a total of 208 diferent settings. Three samples were used for each setting. The overlap of fuences is presented in this experiment. 35 sets of fuence measurements (in duplicate) were made to study the effects of the speed and power. In addition, a set of square chambers with

surface area 10 mm^2 were machined using the raster mode of the laser system at 10.2 W and 175 mm/s to be used for chemical analysis. The microfuidic chips that were produced to verify channel quality and performance consisted of three diferent layers of 2 mm thick PMMA. The top layer contained the outlets, the second layer a detection window (chamber) and channels (Fig. [1c](#page-2-0)) and the third layer was the base layer. The laser power setting used for these was 7.8 W and the scan speed was 62.5 mm/s. The three layers were attached and sealed by thermal bonding under steel (5 mm) plates at 140 °C for 41 min and 2.2 MPa, then cooled to room temperature inside the steel plates.

2.2 Chemical treatment

For post-ablation chemical treatment, a boiling bath (82.3 °C) of isopropyl alcohol (EMPARTA[®] ACS, Merck) was used, with submersion of the samples for 75 s. By keeping the solvent at boiling temperature, we maintain a homogenous and reproducible solvent treatment without the need for high precision heating plates. The samples were left to quickly air dry and then heated at 90 °C for 12 h to remove the excess solvent.

2.3 Characterization and tests

The width of the channels, edge quality, bottom surface quality and start of the channel (Fig. [1a](#page-2-0)), were qualitatively characterized using a SEM (JCM-6000PLUS NeoScope Benchtop). A gold sputter coater (Quorum Technologies— Q150R) was used to create a conductive layer on the PMMA samples prior to SEM analysis.

Fig. 1 a The four parameters measured: channel width, edge quality, bottom surface quality and start of the channel. **b** Proof-of-concept microfluidic test device with 80×40 mm, 10 mm² central optical window and input window of 3 mm radius. **c** Cross-section of the

proof-of-concept test device showing the three layers (L1, L2 and L3) and the channel, inlet and outlet layer and the base layer. The radius of the inlet and outlet ports are 0.6 mm radius and the red layers represent the machined and solvent treated surfaces

The depth and profles of the channels were measured using a custom made spectral domain optical coherence tomography (SD-OCT) system to acquire two-dimensional cross-section of the sample. The system uses a broadband super-luminescent diode (SLD) (Superlum, Ireland), with a central wavelength of 840 nm, a bandwidth of 100 nm giving a theoretical depth resolution of 3.1 µm in air (Lippok et al. [2012\)](#page-11-28). The signal-to-noise ratio of the system was approximately 95 dB. The lateral resolution was measured to be 21 µm. Only one sample for each parameter was images by OCT.

Fourier transform infrared spectroscopy (FTIR) spectra were recorded of the polymer surfaces in attenuated total refectance (ATR) mode using a PerkinElmer Spectrum 400 FTIR spectrometer with KRS/diamond composite crystal window and single bounce at 45°, allowing interrogation of the surface to a depth of 4–6 µm, with averaging of eight scans at spectral resolution of 4 cm−1. Spectra were recorded of both the machined chamber surfaces and adjacent surfaces to identify any changes in composition due to the laser machining, as well as any changes resulting from diferent treatments (untreated polymer, untreated polymer followed by treatment, ablated polymer, ablated polymer followed by treatment, and the evaporation of solvent residue by heating). The ethyl C–H deformation mode, –CH₂, of the polymer chains at 1435 cm⁻¹ was used to normalize the spectra for comparison of the other vibrational bands for any changes resulting from the treatment.

A goniometer (KSV Instruments, CAM100) was used to measure the static contact angle and to calculate the surface energy of the treated and untreated samples. An Electroplus E3000 unit (Instron, High Wycombe, UK) was used to measure the bond strength of solvent treated and untreated PMMA bonded in a 600 mm^2 area.

The performance of the microfuidic chip constructed for proof-of-concept was tested using distilled water and blue food dye. Initially the channels in the microcircuit were flled using a syringe pump (Dolomite Mitos Syringe Pump XS) at 100 µl/min. A blue food dye was added to ultra-purifed water and the pump rate was increased to 600 µl/min. The entering and mixing of the dye with the water in the detection chamber werer analysed using an optical microscope (Leica, DF450C-M205FA components).

The surface roughness of the treated and untreated PMMA was analysed using optical proflometry (Contour GT-K, Bruker USA). Finally, the transparency of the detection windows of the solvent-treated and untreated microfuidic chips was analysed using a UV–Vis spectrophotometer (Ocean Optics S2000-UV–Vis) and a rhodamine B (Sigma-Aldrich) 8.35 µM solution in ultra-purifed water.

3 Results and discussion

3.1 Channel dimensions

The dimensions resulting from the diferent settings of the variables were correlated with scan speed and power of the laser, which can also be translated into fuence. The relationship between channel width and variations in speed and power is given in Fig. [2](#page-4-0)a, b. Speeds above 180 mm/s and below 20 mm/s resulted in very poor quality channels that did not allow for proper measurements. The channel width increases logarithmically with power, as we increase the laser power while keeping the speed constant. This is consistent with what we expect from the Gaussian profile of the commercial $CO₂$ lasers and with previous reports in the literature (Snakenborg et al. [2004](#page-12-7); Chung et al. [2005\)](#page-11-29). The width of the channels varied from 58 to 264 µm, depending on laser power and at diferent speed settings, with standard deviation (SD) between 3 and 22 μ m and an average of 11 μ m. The variation in SD for the channel width measured using the SEM is mostly due to variation in location where channels were measured and imperfections in some of the channels due to errors in the laser oscillation. However, the standard deviations using speeds around 80–100 mm/s at powers between 10 and 6 W were lower, with lowest standard deviations for lower powers between 1.8 and 3 W at scanning speeds of 40–100 mm/s.

The depths of the channels measured using OCT varied from 14 to 1360 µm. The channel depth measurements were afected similarly to the SEM measurements.

The scan speed and laser power were varied while keeping the fluences the same, to determine the effect of these parameters on the quality of the channels. The correlation between the fuence (calculated using scan speed and laser power) and width and depth of the channels is displayed in Fig. [2c](#page-4-0). It was found that lower scan speed and laser power are correlated with lower standard deviations $(p<0.05)$ at fluence below 13 J/cm² in contrast with using higher scan speed and higher laser power to achieve the same fuence. At higher fuences, there was no correlation between standard deviation and scan speed and laser power. This result indicates that lowering both laser power and scan speeds allow greater reproducibility in channel dimensions.

The ftted depth profles of the channels measured by SD-OCT are shown in Fig. [2d](#page-4-0) for five different fluences. A Gaussian profle for the channel can be identifed, as expected from the $CO₂$ laser beam profile. After the solvent treatment there was no signifcant diference in the measurements of width and depth as the treated sample values were within the standard deviation.

Fig. 2 a Channel width/μm as measured by SEM and **b** depth/μm (right) as measured by OCT as a function of laser power and scan speed as contour plots. **c** Width (crosses) and depth (open circles) of

channels as a function of incident laser fuence. **d** Channel profles ft from data acquired by OCT

3.2 Qualitative analysis

SEM is a useful tool to compare morphologies of samples (Golding et al. [2016\)](#page-11-30). Here we used the SEM images of the channels to assess and compare the quality of channels over the whole range of powers, scanning speeds and treatments used. This analysis of the samples confrmed the need to optimize these settings to avoid formation of inferior quality channels. Channel quality was graded qualitatively based on the observations of morphology, on a scale from 0.5 to 5, with 5 denoting the highest quality and 0.5 the lowest. The parameters assessed were edge/wall quality, bottom quality and shape of the start of the channel. Figure [3](#page-5-0) shows the relationships between these parameters in the untreated samples. The bottom quality was improved at lower scan speeds between 20 and 40 mm/s and lower laser powers between 4.2 and 7.8 W (fluence 10.52 and 39.10 mJ/cm²). The edge/wall quality in these samples followed a similar trend (Supplementary information), but it was also improved using high scan speeds and high laser powers, between 40–60 mm/s and 8–10.2 W (13.03–25.57 mJ/cm²), respectively. The graphical results for the quality parameters of the edges and start of channel are given in the Supplementary information.

It is important to notice that in this section, we are dealing with qualitative or categorical results. These results were generated by the evaluation of multiple samples and diferent researchers classifying the data in the experiment according to a set scale. To verify the uncertainty related to the visual quality classifcation, we have created ordinal variables (0.5 is "poor" and 5 is "very good"), used 0.5 units steps when evaluating the channel and averaged the sample values to a fnal quality classifcation. During the experiment, the standard deviation between the qualitative classifcation of the same sample was approximately ± 0.25 units of the qualitative scale,

Fig. 3 (Left) Qualitative colour map of the quality of the untreated bottom (control) of the channels (colour scale on right: 0.5–5, lowest to highest quality). The images on the right illustrate 'poor' (quality value 0.5), 'medium' (quality value 3) and 'good' (quality value

5) channels. The map shows that the areas where the solvent is most efective are where the channels are not too deep, allowing for solvent treatment (colour fgure online)

which represented a coefficient of variation of 9.5% for the analysis of the bottom of the channels, 8.1% for the debris at the end of the channels and 10.2% for the edge of the channels.

Solvent treatment was performed to smooth the channels and remove the debris using Isopropyl alcohol at its boiling point. A solvent is most efective when the forces of attraction between the solvent molecules are smaller than the forces of attraction between the polymer and solvent (Brydson and Ebrary [1999\)](#page-11-31), and can be characterized by the Hildebrand solubility parameter (δ) , which describes the enthalpy of mixing (Kwei [1992](#page-11-32)). Polymers dissolve in solvents for which the solubility parameter is similar (Miller-Chou and Koenig [2003](#page-11-33)) and for which the Gibbs free energy on mixing is below zero (Hansen [2000\)](#page-11-34). The Gibbs free energy on mixing (ΔG_m) during the dissolution of an amorphous polymer in a solvent is given by:

$$
0 > \Delta G_{\rm m} = \Delta H_{\rm m} - T\Delta S_{\rm m},\tag{1}
$$

where (ΔH_m) is the enthalpy change on mixing, (*T*) is temperature and (ΔS_m) is the entropy change on mixing. The dissolution of high molecular weight polymers is associated with a small positive entropy change. The Hildebrand parameter also provides a measure of the cohesive molecular forces of each of the solvent and solute. The energy required to break the intermolecular physical links in a unit volume of a material, cohesive energy density (CED), can be related to the solubility parameter as follows:

$$
\delta = (\text{CED})^{1/2} = \left(\frac{\Delta H_{\text{vap}} - RT}{V}\right)^{1/2},\tag{2}
$$

where ΔH_{van} is the enthalpy of vaporization, (*R*) is the gas constant, and (*V*) is the molar volume of the solvent. Several studies have demonstrated the use of diferent solvents to bond PMMA (Ogilvie et al. [2010](#page-11-35); Steigert et al. [2007](#page-12-8); Tsao and DeVoe [2009\)](#page-12-9), but these did not explore the efect of such treatments on the quality of the channels nor on the efect on the bond strength between the PMMA layers in the microfluidic device.

The solubility parameter of PMMA is $20.18 \text{ MPa}^{1/2}$, and solvents such as dichloromethane, acetone and isopropyl alcohol and water are, respectively, 20.1 $MPa^{1/2}$, 19.9 MPa^{1/2} and 23.4 MPa^{1/2} and 47.9 MPa^{1/2} at room temperature (Tsao and DeVoe [2009](#page-12-9); Ng et al. [2008\)](#page-11-36) PMMA is not afected by water or by isopropyl alcohol at room temperature, but is attacked by dichloromethane and acetone. Therefore, using solvents with similar solubility parameters would only work when using a solvent vapour to smooth and clean the channels. The solubility parameter can be manipulated by changing the temperature as seen in Eq. ([2](#page-5-1)). If we increase the temperature of isopropanol we can decrease the solubility parameter to bring its value closer to that of PMMA. Table [1](#page-6-0) shows the change in solubility parameter as we change the temperature and therefore the enthalpy of vaporization.

This solvent treatment demonstrates a signifcant increase in the overall quality of the channels and a visible decrease or complete removal of debris from inside of the channels (Fig. [4\)](#page-6-1) as well as debris located at the end of the machined lines (Fig. [5\)](#page-6-2).

Fig. 4 (Left) Channel foor surface after direct laser writing. (Right) Decrease or removal of debris from inside the microchannel after solvent treatment

Fig. 5 Debris at the edges and at the end of the line being removed by the solvent bath technique

The qualitatively assessed quality parameters for the edge and bottom of the channels following the solvent treatment follow a similar trend to the untreated, control sample related to which settings give the best outcomes (Supplementary information). The surface quality of the channel foors created using lower fuences is greatly increased after boiling isopropyl submersion. For surfaces made using higher fuences, improvement is either slight or none, even worsened. This is due to difficulties for the solvent to reach the bottom of the channels (Fig. [6\)](#page-7-0). The dimensions of the

Fig. 6 Deeper channels before (top/left) and after (top/right) solvent treatment showing unafected interior dimensions (parameters: $30 \text{ mm/s}, 9.6 \text{ W}, 80.20 \text{ mJ/cm}^2, 819 \text{ µm deep}.$ The untreated channel

(bottom/left) and solvent smoothed (bottom/right) shallow channel (parameters: 30 mm/s, 4.2 W, 35.09 mJ/cm2 , 478 µm deep)

channels treated with boiling isopropanol were within the standard deviation found in the control samples, showing that our treatment process did not signifcantly afect the microchannel dimensions. However, as the solvent treatment is still an etching technique care must be taken to not immerse the samples for too long, even for deeper channels, placing a limitation on the immersion time.

Another change in morphology resulting from the solvent treatment and not previously reported, were steps or grooves created in the channel as the laser raster scans the surface (Fig. [7\)](#page-8-0). A possible cause for this is an intrinsic property of the laser equipment that originates from small, rapid power oscillations from the stepper motor belt that moves the laser head. These step defects can be improved using scan speeds below 100 mm/s and laser power below 6 W (up to 15.03 mJ/cm^2). Lower scanning speeds and laser powers result in poorer quality or even unformed channels (Supplementary information). Furthermore, Fig. [3](#page-5-0) shows an optimal range for the machining parameters for which post-treatment of the channels was efective in retaining a good quality.

3.3 Surface chemistry analysis

An important consideration in microfabrication incorporating chemical treatment is characterization of any chemical changes on the polymer surface or chemical residues that may be present post-treatment. PMMA surfaces from two treatments were compared with those of the control; the frst was the laser machining and the second was solvent treatment of the laser machined area with the boiling isopropyl alcohol procedure described earlier. FTIR spectra of the plain PMMA surface (green), laser machined surface (red) and laser machined surface followed by solvent treatment (black) are given in Fig. [8](#page-8-1). All spectra were peak height normalized to the ethyl C–H deformation mode at 1451 cm^{-1} and overlaid for comparison, with an inset showing an expanded view of the fngerprint region. The decrease in intensity of the strong bands at 1723 cm⁻¹ (C=O stretch mode) and 1188 and 1143 cm−1 (C–O–C stretching modes) after laser ablation (red spectrum) clearly shows that laser ablation reduces the relative amount of carbonyl groups on the PMMA surface.

Fig. 7 Periodicity found in channels (parameter shown: 60 mm/s, 9 W, 15.03 mJ/cm³) and evident through the solvent treatment method. (Left) Untreated and (right) treated sample

Fig. 8 ATR FTIR spectra of untreated PMMA surfaces (green), after laser ablating (red) and after treatment with IPA followed by curing (black). Inset shows expanded view of the reduction in C–O (1143 and 1186 cm⁻¹) and C=O bonds (1723 cm^{-1}) (colour figure online)

The results confrm the observation by Klank et al. ([2002\)](#page-11-6) and Nayak et al. [\(2008\)](#page-11-13) of thermal degradation of the surface by the laser. Other studies on the decomposition of PMMA using 10.6 µm lasers demonstrate that the full degradation pathway due to laser-induced heating can result in (up to 100%) formation of methyl methacrylate (MMA) monomers (Srinivasan [1993\)](#page-12-10). In addition, generation of CO and $CO₂$ can occur and is likely responsible for the observed decrease in relative amounts of C=O and C–O–C. The use of boiling isopropyl alcohol resulted in adsorption of the solvent on the polymer surface in both the ablated and non-ablated part of the polymer: solvent residue was detected on all the treated samples and was removed only by curing them at 90 °C for 12 h, which is above the boiling point of isopropyl alcohol but well below the transition glass temperature of PMMA.

Figure [9](#page-9-0) shows the decrease in the C–O band at 952.5 cm^{-1} as the isopropyl alcohol was eliminated after the heat treatment.

After the solvent treatment was applied to the machined area, further, but smaller, decreases in the relative intensities of the C–O–C and C=O modes were observed. These changes suggest that the boiling solvent treatment was also working as a means to wash off by-products of PMMA monomers resulting from the laser machining. In addition, the byproduct dissolved in the isopropyl alcohol wash was analysed and found to comprise largely acrylic groups.

An additional comparison of the surface for any differences resulting from pre- and post-treatment was performed by measuring the contact angle of each surface. The untreated and treated polymers showed contact angles of $70.8^{\circ} \pm 1.7^{\circ}$ and $71.0^{\circ} \pm 2.0^{\circ}$, respectively. These results

Fig. 9 FTIR spectra of the C–O stretch mode of IPA at 950 cm−1 during solvent treatment: control (red), solvent applied to control (PMMA sheet) (blue), solvent applied to control followed by temper-

Fig. 10 Visible detection window created with the presented method. Images taken (left) after 5 s and (right) after 60 s of blue food dye as it began to enter the microfuidic chamber (colour fgure online)

indicated that no signifcant changes were likely to occur in the polymer wettability when using this technique for microfabrication.

3.4 Proof‑of‑concept: a microfuidic device

Microfuidic devices often incorporate an analytical tool or sensor on at least one segment of the microfuidic circuit. Some detection is performed using spectral, fuorometric or colorimetric measurements. These techniques have in common the requirement of a clear and reproducible optical detection window. One of the efects of the solvent treatment is a slight cloudiness on the surface of the polymer due to adsorption of acrylic particles from the solution as well as increase in surface roughness (see next section); these affect the transparency. In this study, we used thermal bonding

ature curing (light green), solvent on the laser ablated PMMA (purple) and solvent treatment on the laser ablated surface followed by temperature curing (dark green) (colour fgure online)

after solvent treatment of the channels to create a microfuidic chip setup that allow us to keep the transparency of the detection chamber, while achieving smoothing of the channel surfaces as previously described. Mixing of water and blue dye solution and vortex formation inside the detection chamber could clearly be observed in the microfuidic chamber through the transparent optical window (Fig. [10](#page-9-1)). When using colorimetric analysis in microfuidics as well as verifying the turbulence, mixing, and miscibility of

Fig. 11 Main rhodamine B peak at 556 nm showing highly similar absorbance peak for treated and untreated samples, hence the efectiveness of using a three-layer sandwich chip when using the solvent treatment

Fig. 12 Surface profle acquired using a optical proflometer colour scale/um) of the same area analysed before and after solvent treatment. Left: untreated PMMA showing some depressions (blue) that

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may trap air. Right: increase in surface roughness seen after treatment (colour fgure online)

fuids, transparency of the detection window transparency is critical.

PMMA absorbs strongly in the UV region, therefore rhodamine B was used to test the transparency of the detection window as it absorbs strongly at the visible region. The similarity of absorbance intensity for treated and untreated samples showed that no efective change in absorption occurred (Fig. [11\)](#page-9-2).

The bonding technique in microfuidic prototypes needs to be inexpensive, easy and fast. Thermal bonding fulfls these requirements. The strength of thermal bonding in our system was tested by bonding a treated area with an untreated area to simulate the bond between the smoothed channel surface and optically transparent PMMA. The bond strength of the untreated PMMA was 185.0 ± 33.3 kPa (111 N per 600 mm⁻²), and 808.0 ± 80 kPa (545 N per 600 mm−2) for the thermally bonded treated and untreated samples.

The increase in strength for the treated polymer bonding is likely due to slight swelling of the polymer, which creates a micro roughness (Fig. [12\)](#page-10-1). The optical profle results also showed some air traps that were not seen in treated samples. The extra roughness on the surface generated by the solvent treatment allowed air to be released from the trapped areas during the thermal bonding process and so increase the contact area. Consequently, the bond strength was increased.

To calculate and compare the surface roughness of the samples we used the standard measurement for roughness parameters of arithmetical mean roughness value (R_a) and the root-mean-square of the roughness (*R*q) (Petropoulos et al. 2010). The R_a and R_a found for the untreated sample were, respectively, 6 and 7 nm, whereas for the treated samples these values increased to 81 and 115 nm. We accredit this roughness difference to increasing the effective bonding area by decreasing the air bubbles trapped,

and by increasing the effective bonding depth when the treated sample is used.

4 Conclusion

We have developed a protocol for making microfuidic circuits using a rapid prototyping technique while still achieving good quality channels. This protocol utilizes variation in laser power and scan speeds to generate diferent microchannels dimensions. The use of boiling isopropanol is a rapid means to reproducibly smooth the rough microchannels and debris that result when using commercially available $CO₂$ lasers, and can be applied and removed from the sample surfaces without altering their properties. A proofof-concept microfuidic chip confrmed that we could retain the optical transparency of the detection window, while an approximately fourfold increase was achieved in the bonding strength for binding between an untreated and a treated sample. This protocol enables rapid prototyping of new microfluidic designs without the need for expensive equipment, providing greater opportunities for laboratories with limited resources.

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