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Higher photocatalytic removal of organic pollutants using pangolin‑like composites made of 3–4 atomic layers of MoS2 nanosheets deposited on tourmaline

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Abstract

Environmental pollution by organic pollutants is a serious concern which may be solved by photocatalytic degradation of pollutants, yet the efficiency of actual photocatalytic materials is limited. For instance, conventional $MoS₂$ nanosheets tend to agglomerate, which hinders the access of pollutants to active sites. To overcome this challenge, we hypothesized that assembling MoS₂ nanosheets on a mineral support would improve access to active sites. We synthesized a catalyst made of 3–4 atomic layers of $MoS₂$ nanosheets deposited on tourmaline using a microwave hydrothermal method. Results show that tourmaline occurs as a polyhedron single crystal that supports the epitaxial growth of $2H-MoS₂$ layers on tourmaline (773) facets, while an intrinsic rolling up behavior of $MoS₂$ layer from [002] to [106] on the mineral surface accounts for the curly morphology. The pangolin-like MoS₂/tourmaline composite degrades rhodamine B much better than the pure MoS₂ nanosheets assembled microspheres. This is explained by the reduced thickness of $MoS₂$ nanosheets according to the density functional theory. Overall, our fndings represent a new tactic for the cost-efective batch preparation of two-dimensional materials with high catalytic performance.

Keywords Natural minerals · Tourmaline · Molybdenum disulfde · Microwave hydrothermal · Epitaxial growth · Photocatalysis

Introduction

With the development of industry, excessive discharge of organic wastewater has become one of the most serious environmental problems around the world, which poses a

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signifcant threat to environment and human health due to its persistence and toxicity (Saravanan et al. [2021](#page-9-0); Madima et al. [2020](#page-8-0); Mecha. et al. [2020](#page-8-1)). At present, photocatalysis as a simple and economical method has been widely used to remove organic pollutants from wastewater (de Lima et al. [2020](#page-8-2); Saeed et al. [2018](#page-9-1)). Compared with the common pho- $\overline{\text{Ming Hao and Hao Li are contributed equally to this study.}}$ tocatalyst containing TiO_2 and CdS, molybdenum sulfide

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 $(MoS₂)$, a typical layered two-dimensional semiconductor material, has received considerable attention in the photodegradation of organic pollutants due to its strong visible light response, layered structure, and environment-friendly nature (Gan et al. [2020](#page-8-3); Wang and Mi [2017](#page-9-2)). However, due to its high specific surface energy, nanostructured $MoS₂$ suffers from aggregation inevitably which causes rapid recombination of electron–hole pairs and thus poor catalytic performance (Qu et al. [2013\)](#page-9-3). One efective solution is to disperse the $MoS₂$ on support materials, such as graphene (Kumar et al. [2019](#page-8-4)), graphitic carbon nitride (Liu et al. [2018](#page-8-5)), reduced graphene oxide (Sun et al. [2017](#page-9-4)), titanium dioxide (Zhao et al. [2019\)](#page-9-5), and $Bi_2MoO₆$ (Li et al. [2018](#page-8-6)). However, the high preparation cost and the problematic environmental load limit these fabrication approaches from industrial applications. It becomes practically important that the $MoS₂$ nanostructures with few layers and full of active sites could be stably assembled on a support of highly natural abundance, e.g., minerals. Mineral carriers own merits of low-cost and environmental benignity (Wang et al. [2021](#page-9-6); Hao et al. [2021;](#page-8-7) Saja et al. [2020](#page-9-7)), making them as promising candidates as a functional support. However, only few studies have focused on the fabrication of $MoS₂$ nanostructures using mineral supports, especially tourmaline. Tourmalinesupported $MoS₂$ is believed to be structurally beneficial to provide more active sites via good assembly fexibility based on its unique characteristics including spontaneous polarization and far infrared radiation.

Tourmaline is a borosilicate mineral with the R_{3m} space group (Guerra et al. [2012\)](#page-8-8) and the chemical formula of $XY_3Z_6Si_6O_{18}(BO_3)_3W_4$, where X is Na⁺, Ca²⁺, K⁺, or vacancies; Y is Mg^{2+} , Mn^{2+} , Fe²⁺, Al³⁺, Fe³⁺, Mn³⁺, or Li⁺; Z is Al³⁺, Fe³⁺, Cr³⁺, or Mg²⁺; and W is OH⁻, F⁻, or O²⁻. It has been extensively studied due to the spontaneous and permanent poles which can produce an electric dipole (Zhou et al. [2018\)](#page-9-8). Moreover, the strong electric feld formed on the tourmaline surface can accelerate the separation/transportation of photoexcited charge carriers (Li et al. [2015](#page-8-9)). Besides, other unique properties such as pyroelectric, piezoelectric, and far infrared radiation have made tourmaline flexible in various fields (Zhang et al. [2014](#page-9-9)). Therefore, tourmaline owns not only the merits of the low cost but also several unique properties compared with some common carbon materials including carbon powder, biochar, and carbon nanotubes (Xu et al. [2020](#page-9-10), [2019,](#page-9-11) [2016\)](#page-9-12). In our previous studies, tourmaline has been found to efectively decrease agglomeration of silica hollow microspheres (Wang et al. [2017\)](#page-9-13), as well as the nanoparticle's size (Zhang et al. [2017a](#page-9-14)). In addition, our previous study was successful in fabricating a FeMnTiO_x catalyst synergized with tourmaline support for low-temperature $NH₃-SCR$ (Wang et al. [2019](#page-9-15)). These works serve as direct evidence that tourmaline can be used to improve catalytic activity. In this work, using

a one-step microwave hydrothermal method, we report the first preparation of an ingenious $MoS₂/t$ ourmaline composite composed of atomically few layered $MoS₂$ nanosheets growing along the surface of tourmaline particles. This structure significantly increases the dispersion of $MoS₂$ nanosheets and provides more active sites for catalysis. The spontaneous polarization and far infrared radiation of tourmaline support ensure even adsorption and nucleation of $MoS₂$ species through promoting the thermal vibration and reducing the difusion activation energy of Mo and S atoms. Meanwhile, strong electric feld on the surface of tourmaline can restrain the recombination of photo-generated electron–hole pairs (Tzeng et al. [2019](#page-9-16)). This work provides an inspiring scheme for low-cost batch preparation of high-quality twodimensional materials via assembly on tourmaline mineral functional materials.

Materials and methods

Materials

The tourmaline was provided by HY Technology Co., Ltd. The ammonium molybdate $(NH_4)_6Mo_7O_{24} \cdot 4H_2O$, thiourea (CH_4N_2S) and rhodamine B (RhB) were supplied by Kewei Chemical Group Co., Ltd. These materials were used as received. Deionized (DI) water was used in all experiments.

Synthesis of MoS₂/tourmaline composites

2.4 g (NH₄)₆Mo₇O₂₄·4H₂O and 2.28 g CH₄N₂S were dissolved in 70 mL deionized water and stirred for 0.5 h. Then, 0.84 g of tourmaline powder was added into the above solution and mechanically stirred for another 0.5 h. Next, the suspension was sonicated for 0.5 h. After that, 50 mL of the above suspension was transferred into a 100 mL Tefon-lined stainless steel autoclave and kept at diferent hydrothermal temperatures including 190, 200, 210, 220, and 230 °C for 3 h. After the samples were cooled to room temperature, the fnal products were obtained by fltration, washed thoroughly with deionized water, and dried at 80 °C for 12 h. The samples synthesized at 190 °C, 200 °C, 210 °C, 220 °C, and 230 °C were named as T-190, T-200, T-210, T-220, and T-230, respectively.

Characterizations

X-ray difraction was performed on a D8 ADVANCE X-ray difractometer with nickel-fltered Cu Kα radiation as the X-ray source, and $V = 40$ kV, $I = 40$ mA, and $\lambda = 1.54$ Å. Fourier-transform infrared spectroscopy (FTIR) spectra were recorded in transmission mode from 400 to 4000 cm−1 on a Tensor II Fourier transform infrared spectrometer purchased

from Bruker in Germany. The morphology of the samples was observed using a Quanta FEG450 scanning electron microscope working at 200 kV. Transmission electron microscope (TEM) and high-resolution TEM (HRTEM) images were acquired using a JEM 2100F transmission electron microscope from JEOL in Japan at an accelerating voltage of 200 kV. X-ray photoelectron spectroscopy (XPS) measurements were performed on an ESCALAB250 spectrometer.

The atomic resolution scanning transmission electron microscope (STEM) was realized on a JEM-ARM200F equipped with EDS analyzer (200 kV, probe Cs-corrector, 68–174 mrad collection angle for high-angle annular dark field imaging, ~ 0.8 Å spatial resolution). The samples were ground into fne powders, dispersed in ethanol, dropped onto a lacey carbon flm-Cu TEM ready grid from Ted-Pella Co., and then dried on a hot plate at 70 °C. STEM images were taken using the high-angle annular dark feld detector. Energy-dispersive X-ray spectroscopy (EDS) was measured on an EDS analyzer (JEOL, EX-230 100 m² detector) equipped on a JEM-ARM200F microscope. The elemental mapping was performed at a sampling resolution of 128×128 pixels and a dwell time of 0.1 ms/pixel under multiple frame integration.

Computational methodology

Density functional theory (DFT) calculations were conducted using the Vienna Ab-initio Simulation Package (VASP). The core and valence electrons were described by the projector augmented wave method (Blöchl [1994\)](#page-8-10) and the Kohn–Sham wave functions being expanded in plane wave basis (Kohn and Sham 1965). The energy cutoff was set as 400 eV for all of the calculations. The electronic exchange and correlation were described by a generalized gradient approximation; the functional method developed by Perdew, Burke, and Ernzerhof was chosen for the calculations (Perdew et al. [1996](#page-9-17)). The Brillouin zone was sampled within a $(3 \times 3 \times 1)$ *k*-point mesh using the method developed by Monkhorst and Pack (Monkhorst and Pack [1976\)](#page-9-18). Structures were considered as being optimized after all the forces fell below 0.05 eV per Å.

Photocatalytic tests

Photocatalytic activity of the as-prepared samples was assessed by the photocatalytic degradation of rhodamine B under visible light irradiation, where optical absorption of rhodamine B at 553 nm was used as a major wavelength. In a typical experiment, 20 mg of sample was added into 100 mL of rhodamine B solution, yielding the suspension under stirring in the dark for 0.5 h to reach the adsorption/desorption equilibrium. The suspension was subjected to irradiation by

a 500 W Xe lamp at *λ*>420 nm under stirring at ambient conditions. At intervals of 0.5 h, 6 mL of the suspension was extracted and centrifuged to remove the photocatalysts. The fltrates were analyzed by recording the UV–vis spectra using a Shimadzu UV-1800 spectrophotometer (Osman et al. [2017\)](#page-9-19). The residual percentage of the rhodamine B is calculated through C_t/C_0 , where C_0 represents the concentration of rhodamine B before the irradiation, C_t represents the concentration of rhodamine B after irradiating time *t* was utilized to represent the photocatalytic performance of the catalyst.

Results and discussion

Here we studied the phase, morphology, structure, vibrational bands and the specifc interactions, surface elemental composition and chemical status of the as-prepared $M_0S_2/$ tourmaline composites, and the growth mechanism of the composite was raised. Next, the photocatalytic test was carried out and both density functional theory calculations and proposed synergic effects between tourmaline and $MoS₂$ for rhodamine B degradation were used to explain the high catalytic efficiency of the composite.

Morphology observation of the MoS₂/tourmaline composite

Scanning electron microscopy images of the as-prepared $MoS₂/tourmaline composites delineate the overall architecture$ tures. For the $MoS_2/tourmaline$ samples, MoS_2 nanosheets are evenly distributed on the surface of tourmaline particles, displaying a gripping pangolin-like structure (Fig. [1](#page-3-0)a and Fig. S1). As the synthesis temperature ramps, the coverage of the $MoS₂$ nanosheets over the tourmaline particles increases. As revealed by Fig. $1a$, MoS₂ nanosheets are consistently grown on the surface of tourmaline particles in the sample prepared at 210 °C. High-resolution transmission electron microscope image (Fig. S2) further shows direct evidence that $MoS₂$ phase has been uniformly loaded on the tourmaline particles, which benefts charge separation during electron transfer (Yin et al. [2017\)](#page-9-20) and catalytic efficiency.

Phase study of the MoS₂/tourmaline composite

X-ray difraction patterns further testifed the phases compositions of tourmaline (JCPDS Card No. 43–1464) and $2H-MoS₂$ (JCPDS Card No.37–1492). As shown in Fig. [1](#page-3-0)c, the peak of the $MoS₂$ diffraction (002) shifts slightly to a smaller angle, which is probably due to the inhalation of ammonium ions into the $MoS₂$ layers (Wang et al. [2020](#page-9-21)). Meanwhile, the specifc peaks of the tourmaline difraction shift to larger angles indicating the shrinkage of tourmaline unit cell (Table S1), which is caused by the oxidation of Fe^{2+} to Fe^{3+} inside the tourmaline. (Guo [2019](#page-8-12)). The diffraction peak becomes sharper as the temperature increases, indicating the crystallinity of $MoS₂$ is enhanced (Wang et al. [2014](#page-9-22)), and the various fabrication times demonstrate a consistent tendency with the temperature change (Figs. S3, S4).

FTIR and X-ray spectroscopy analysis of the MoS₂/ tourmaline composite

FTIR results (Fig. [1d](#page-3-0)) identifed the functional groups of the tourmaline skeleton, such as the stretching vibration at 3562 cm^{-1} for O–H, 1266 cm⁻¹ and 1350 cm⁻¹ for antisymmetric B-O, 980 cm⁻¹ for Si–O-Al, 712 cm⁻¹, 780 cm⁻¹ and 1030 cm⁻¹ for Si-O-Si, 491 cm⁻¹, 573 cm⁻¹, and 655 cm⁻¹ for octahedral cation M–O, along with the 470 cm^{-1} for Mo-S stretching vibration (Luo et al. [2020](#page-8-13); Yu et al. [2020](#page-9-23); Pramoda et al. [2016\)](#page-9-24). Furthermore, XPS measurements show a significant overall blue shift for $MoS₂$ when assembling on the tourmaline (Fig. S5), indicating the fact of electron transfer from $MoS₂$ to the tourmaline support with the surface electric feld (Liu et al. [2015a\)](#page-8-14).

Atomic structural details for tourmaline, MoS₂ **and their interface**

To study the atomic structures of the synthesized $MoS₂/$ tourmaline composites, probe aberration-corrected scanning transmission electron microscopy (AC-STEM) was performed (Fig. [2\)](#page-4-0). High-angle annular dark feld imaging combined with X-ray EDS was also used to further characterize the morphology of $MoS₂/tourmaline composite$ (Fig. [2a](#page-4-0)), atomic structure details of tourmaline (Fig. [2](#page-4-0)b, c), and $MoS₂$ -tourmaline interface (Fig. [2](#page-4-0)d). It can be seen that tourmaline structures facilitate a good dispersion of the twodimensional $MoS₂$ nanosheets (Figs. [1](#page-3-0)b, [2a](#page-4-0)). Particularly, this turns to be the frst time to directly image the atomic structure of a tourmaline crystal (Fig. [2](#page-4-0)b), where the atomcoordination details are in good agreement with the ideal structure (Bosi et al. [2005](#page-8-15)). It proves that tourmaline belongs to the R_{3m} space group with six-membered Si rings, Fe–Mg

Fig. 1 Morphology and phase composition of the as-fabricated MoS_2 / tourmaline composites. **a** Scanning electron microscopy images of the $MoS₂/tourmaline composite obtained via microwave hydrother$ mal at 210 °C; **b** Energy-dispersive X-ray spectroscopy (EDS) mappings for elemental distributions of the individual $MoS₂/tourmaline$

structure; **c** X-ray difraction patterns of the as-prepared samples synthesized at various temperatures; **d** Fourier-transform infrared (FTIR) analysis of functional groups on the surface of as-fabricated samples, where T-190, T-200, T-210, T-220, T-230 correspond to the sample prepared at 190 °C, 200 °C, 210 °C, 220 °C, 230 °C, respectively

Fig. 2 Scanning transmission electron microscopy (STEM) analysis on the atomic structure of $MoS₂/tourmaline$ (T-210) composite, where T represents tourmaline. **a** A low-magnifed STEM image shows two-dimensional $MoS₂$ nanosheets assembled along the tourmaline particles; **b** Enlarged high-angle annular dark feld images: the

atomic structure details of tourmaline, the further details observed by tourmaline <217>axis (**c**) and MoS_2 -tourmaline interface (**d**); **e** Growth mechanism of $MoS₂/tourmaline$ composites. I: Even distribution and adsorption process of Mo and S atoms and clusters on the tourmaline surface; II: Microwave hydrothermal synthesis process

octahedra, B triangles; structural units linked together by Al octahedra (Bosi and Lucchesi [2007\)](#page-8-16). This unique structure is noncentrosymmetric and polar, the c axis is threefold symmetry, and there is no axis plane of symmetry perpendicular to c and no center of symmetry (Nakamura et al. [1992](#page-9-25); Wang et al. [2011\)](#page-9-26). Therefore, tourmaline is usually in the shapes of trilateral and hexagonal column, with properties of far infrared radiation, spontaneous polarization, pyroelectricity, and piezoelectricity (Wang et al. [2019](#page-9-15); Chen et al. [2020;](#page-8-17) Dietrich [1985\)](#page-8-18). Moreover, close examination of tourmaline/ $MoS₂$ interface reveals it was the uniform anchor of MoS_x clusters on the mineral surface of tourmaline (773) facets (Fig. [2](#page-4-0)c) that benefits the uniform growth of $MoS₂$ nanosheets, while these $MoS₂$ sheets are in atomic thickness of only 3–4 layers epitaxially stacking (Zhang et al. [2017b\)](#page-9-27) in a 2H sequence on the mineral surface in relationship of tourmaline (773) $|M_0S_2(002)(Fig. 2d)$ $|M_0S_2(002)(Fig. 2d)$ $|M_0S_2(002)(Fig. 2d)$. Intriguingly, the M_0S_2 nanosheets evolve into curly and intercross morphology, which according to the microscopy evidence, herein, should be attributed to the intrinsic behaviors of $MoS₂$ layer rolling up during

Fig. 3 Note the increase in the calculated bandgap (**a**) and *d*-band center (**b**) of MoS_2 with the number of layer varying from 4 to 1. Insets show the optimized structures considered for calculations

stacking at the tourmaline/ $MoS₂$ interface, e.g., in Fig. [2d](#page-4-0), the initial $MoS₂(002)$ termination rolls up to be $MoS₂(106)$.

Proposed growth mechanism of MoS₂/tourmaline composite

According to the above structural analysis, the growth mechanism of $MoS₂/tourmaline composites can be illustrated$ schematically in Fig. [2](#page-4-0)e, where Mo and S atoms gradually adhere to the tourmaline followed by the epitaxial growth of $MoS₂$ nanosheets with curly edges on the surface of tourmaline particles. More surprisingly, using the naturally abundant mineral to support highly dispersed $MoS₂$, the fabrication cost has been reduced to be less than one-thousandth of the commercial counterparts (Fig. S6).

Density functional theory calculations

To understand the potential structure–activity relation of $MoS₂/tourmaline, DFT calculations were performed to$ study the bandgap and *d*-band center, i.e., average energy of d -electrons of MoS₂ with the number of layer varying from 1 to 4 (Fig. [3](#page-5-0)).

Interestingly, it can be seen that reducing the thickness leads to a wider bandgap (Fig. [3](#page-5-0)a), while a monolayer of $MoS₂$ owns the bandgap close to that of optimum for photocatalysis (Liu et al. $2015b$). Meanwhile, with a thinner layer of MoS₂, there is a signifcant upshift of the *d*-band center toward the Fermi level, indicating stronger bond activation capacity for degradation reaction (Li et al [2019](#page-8-20)). Both these theoretical calculations indicate a signifcant advantage of using tourmaline for the preparation of $MoS₂$: The use of tourmaline results in a controllable thickness of $MoS₂$, while a thin layer of $MoS₂$ guarantees a better photocatalytic activity.

Photocatalytic performance of the composite toward rhodamine B degradation

Photocatalysis is an efective technology which has been frequently utilized for water splitting and degradation of organic pollutants (Li et al. [2020](#page-8-21); Liao et al. [2019;](#page-8-22) Zhong et al. 2018). The photocatalytic activity of the MoS₂/tourmaline composites was evaluated using the rhodamine B degradation as a probe reaction, as shown in Fig. [4](#page-6-0). The degradation processes of all the $MoS₂/tourmaline$ composites exhibit much higher efficiencies than that, \sim 41.2%, of the pure MoS₂ after photocatalytic reaction for 150 min (Fig. [4](#page-6-0)a). Notably, the T-210 sample reaches the highest degradation efficiency of \sim 93%, roughly 2.3 times the degradation efficiency of the pure $MoS₂$. These numbers suggest that the distinct architecture of the $MoS₂/tourmaline composite produces much$ greater photocatalytic degradation efficiency. In addition, the

Fig. 4 Photocatalytic per formance and mechanism of rhodamine B degradation. Photocatalytic activity of the MoS 2/tourmaline composites prepared at diferent tem peratures (**a**) and times (**b**). It is worth noting the sample prepared at 210 °C and 3 h exhibits higher photocatalytic activity than those prepared at other synthetic parameters and much higher than pure $MoS₂$, where T-190, T-200, T-210, T-220, T-230 correspond to the sample prepared at 190 °C, 200 °C, 210 °C, 220 °C, 230 °C, respectively; **c** Comparison of the photocatalytic sites between MoS 2/tourmaline composite and pure MoS 2 microsphere; **d** Illustration of the reaction steps involved in the $MoS_2/tourma$ line composite, in which the synergic efects between tour maline and $MoS₂$ are worthy of paying attention especially

T-220 and T-230 samples exhibit a slightly lower degradation rate than the T-210, which may be due to the agglomeration of $MoS₂$ nanosheets at higher temperature. For all the samples, the changes in the fabrication time (Fig. [4a](#page-6-0)) and fabrication temperature (Fig. [4](#page-6-0)b) exhibit similar tendencies.

Comparison of pristine MoS₂ and MoS₂/tourmaline **composite during photocatalysis**

Evidently, although the as-synthesized $MoS₂$ microspheres are assembled by $MoS₂$ nanosheets (Fig. [4c](#page-6-0)), most of them are encapsulated inside of the microspheres, which have little-to-zero access to the visible light irradiation and thus contribute much less to the photocatalytic degradation of rhodamine B compared with the outmost $MoS₂$ nanosheets in the microspheres. In contrast, $MoS₂$ nanosheets in the $MoS₂/tourmaline composite are highly dispersed, offering$ much more exposed 3–4 layered ultrathin $MoS₂$ nanosheets with the epitaxial and curly stacking for the photocatalysis process, which enhanced the absorption of photons and the efficiency of charge transfer (Fig. [4c](#page-6-0)) (Zhou et al. [2019;](#page-9-29) Dai et al. [2016](#page-8-23)).

Proposed synergic efects between tourmaline and MoS₂ in photocatalytic rhodamine B **degradation**

Figure [4](#page-6-0)d further illustrates the synergic effects between the tourmaline and the grown $MoS₂$ nanosheets in the $MoS₂$ / tourmaline composite. First, OH is the dominant oxidant for photocatalytic degradation in wastewater treatment, tourmaline can polarize water (Wang et al. [2012](#page-9-30)) to produce OH− (Eq. [1\)](#page-7-0) and decrease the water association degree on the tourmaline surface $(Eq. 2)$ $(Eq. 2)$ (Tzeng et al. [2019](#page-9-16)), thus it can form more •OH species from the oxidation of OH⁻ or H₂O via photogenerated holes (Eq. [3](#page-7-2)). Second, tourmaline, schorl herein, is a natural mineral bearing iron ions, which can catalyze H_2O_2 that formed in the photocatalytic reaction to produce •OH and •O^{2−} (Eq. [4\)](#page-7-3) (Xu et al. [2010;](#page-9-31) Wang et al. [2013\)](#page-9-32), the details of the formation of H_2O_2 , as well as its catalytic mechanism, are shown in Fig. S7. Third, the holes show high affinities for the electrons and the high degree of the recombination of the electron–hole pairs is a major limiting factor controlling the photocatalytic efficiency. O_2 acts primarily as an efficient electron trap to prevent the recombination of the photogenerated electron–hole pairs, while tourmaline can augment the content of dissolved O_2 in water by ionizing water (Eq. [5\)](#page-7-4) (Wang et al. [2013\)](#page-9-32). Fourth, the photo-induced electrons from the $MoS₂$ are adsorbed tightly on the anode of tourmaline due to the 'electrostatic poles' on the surface of tourmaline, which prolongs the charge lifetime from MoS_2 (Eq. [6\)](#page-7-5) (Yu et al. [2016](#page-9-33)). Such retardation effect of charge not only results in trapping the charge to suppress

the recombination of $h⁺-e⁻$ pairs, but also transferring the trapped charges to contact with the reactants (Eq. [7\)](#page-7-6). In the mean time, the remaining holes also migrate outward to form reactive species (Eq. 8). Therefore, the MoS₂/tourmaline composite demonstrates much higher photocatalytic activity than that of the pure $MoS₂$ microspheres (Eq. [9](#page-7-8)). The main steps of the synergistic efects are shown as follows:

$$
H_2O \to H^+ + OH^- \tag{1}
$$

$$
H_2O clusters \rightarrow H_2O molecules (2)
$$

$$
h^{+} + OH^{-}/H_{2}O \rightarrow \cdot OH
$$
 (3)

$$
Tournaline + H2O2 \rightarrow \cdot OH + \cdot O2-
$$
\n(4)

Tourmaline + H₂O + O₂
$$
\rightarrow
$$
 O₂ dissolved in water (5)

$$
Tournaline + e^{-}/ h^{+} \rightarrow Tournaline (e^{-}/ h^{+})
$$
\n(6)

Tourmaline
$$
(h^+) + OH^- / H_2O \rightarrow \cdot OH
$$
 (7)

$$
Tournaline (e^-) + O_2 \rightarrow O^{2-}
$$
\n(8)

$$
\cdot \text{O}^{2-}/\cdot \text{OH} + \text{rhodamine B} \rightarrow \text{CO}_2 + \text{H}_2\text{O}
$$
 (9)

In conclusion, the unique $MoS₂/tourmaline structure with$ the excellent synergism of tourmaline signifcantly facilitates photocatalytic degradation compared with the pure $MoS₂$ microspheres.

Conclusion

 $MoS₂$ nanosheets with 3–4 atomic layers are uniformly assembled on the tourmaline particles via a facile microwave hydrothermal method. In this new composite architecture, tourmaline carrier is in a form of polyhedron single crystal which supports the epitaxial growth of $2H-MoS₂$ layers at the tourmaline (773) facets, while an intrinsic rolling up behavior of $MoS₂$ layer along the support surface from direction [002] to [106] accounts for the structural origin of the overall curly morphology. In a typical photocatalytic application, the as-prepared $MoS₂/$ tourmaline has demonstrated considerably enhanced photocatalytic rhodamine B degradation. To the best of our knowledge, this is the frst report on the combination of atomically thin two-dimensional $MoS₂$ nanosheets and tourmaline mineral. Density functional theory calculations further showed that thinner $MoS₂$ nanosheets lead

to wider bandgap and the upshift of *d*-band centers, which in turn enhances the photocatalystic activity. This work provides new insights for the low-cost mass production of high-quality, atomically thin two-dimensional materials supported on tourmaline minerals.

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Authors' contribution FW and JL conceived and supervised experiments. MH, LC, XX and DW performed experiments. WL and FW performed and analyzed the STEM characterization. HL performed the DFT simulations. FW, WL, BF, MH and LC prepared and revised the manuscript. All authors contributed to the experiments and manuscript preparation.

Data availability The data that support the fndings of this study are available from the corresponding author upon reasonable request.

Declarations

Conflict of interests The authors declared that they have no confict of interest.

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