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Incorporating fber recruitment in hyperelastic modeling of vascular tissues by means of kinematic average

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Abstract

We present a framework for considering the gradual recruitment of collagen fibers in hyperelastic constitutive modeling. An efective stretch, which is a response variable representing the true stretch at the tissue-scale, is introduced. Properties of the efective stretch are discussed in detail. The efective stretch and strain invariants derived from it are used in selected hyperelastic constitutive models to describe the tissue response. This construction is investigated in conjunction with Holzapfel-Gasser-Ogden family strain energy functions. The ensuing models are validated against a large body of uniaxial and bi-axial stress–strain response data from human aortic aneurysm tissues. Both the descriptive and the predictive capabilities are examined. The former is evaluated by the quality of constitutive ftting, and the latter is assessed using fnite element simulation. The models signifcantly improve the quality of ftting, and reproduce the experiment displacement, stress, and strain distributions with high fdelity in the fnite element simulation.

Keywords Fiber recruitment · Waviness distribution · Kinematic average · Vascular tissue · Structurally motivated model

1 Introduction

Elastin and collagen fbers are major determinants of passive mechanical properties of vascular tissues (Dobrin [1978](#page-16-0)). Near the ground state, collagen fbers are mostly crimped and contribute little to the mechanical response (Clark and Glagov [1985](#page-16-1); Dingemans et al. [2000](#page-16-2)). As the load increases, the collagen fbers are gradually uncrimped and recruited into load bearing. At higher pressures, the collagen fbers are fully straightened; the tissue becomes stifer to reinforce the vessel wall. The gradual recruitment of wavy collagen fbers is an important feature of the vascular response. It ensures that the vascular vessels are sufficiently compliant in the physiological pressure range for them to function as a conduit and at the same time, prevents them from overstretching at elevated pressures (Humphrey [2002\)](#page-16-3). Studies also suggested that disease conditions such as aneurysm can

 \boxtimes Jia Lu jia-lu@uiowa.edu Xuehuan He xuehuan-he@uiowa.edu alter the recruitment response. Aneurysm formation is intimately connected with loss of elastin implicated for dilation (Dobrin [1989\)](#page-16-4). As such, aortic aneurysms are collagen-rich structures (Choudhury et al. [2009](#page-16-5); Dobrin et al. [1984;](#page-16-6) Fonck et al. [2007;](#page-16-7) Iliopoulos et al. [2009;](#page-17-0) Martyn and Greenwald [1997;](#page-17-1) Sokolis et al. [2012](#page-17-2)); their mechanical response is often associated with a quick collagen recruitment (Sugita et al. [2011](#page-17-3); Sugita and Matsumoto [2013\)](#page-17-4). A recent study by the authors' group indicated that patterns in the recruitment phase of response can be exploited to evaluate the rupture risk in ascending thoracic aortic aneurysms (Luo et al. [2018](#page-17-5); He et al. [2021a](#page-16-8), [b\)](#page-16-9). Thus, the phase of response where the collagen fbers are gradually recruited merits attention.

Since the seminal work of Fung (Fung et al. [1979](#page-16-10); Chuong and Fung [1983\)](#page-16-11), hyperelastic models have been routinely used to describe the elastic behavior of soft tissues. For the vascular system, perhaps the most popular model is the Holzapfel-Gasser-Ogden (HGO) strain energy function (SEF) and its variants (Holzapfel et al. [2004a,](#page-16-12) [b](#page-16-13); Gasser et al. [2006](#page-16-14)). See Holzapfel and Ogden [\(2010](#page-16-15)) for a comprehensive review and references therein. The HGO family is structurally motivated models; they are phenomenological in nature and yet they take into consideration of prominent microstructural features of the fber network. The fber energy is represented by functions of strain invariants

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that are associated with distinct modes of fber deformation. A prominent advantage of the invariant-based formulations is that they are efficient in numerical computation. As such, the HGO family has been widely used in organ-level fnite element analysis. The HGO family SEFs describe the vascular tissues generally well when the collagen fbers are sufficiently stretched. However, they are less apt in capturing the transitional response associated with the graduate recruitment of the collagen fbers.

In the literature, the gradual recruitment of collagen fbers has been considered largely within the framework of *structural constitutive model* pioneered by Lanir ([1979,](#page-17-6) [1983](#page-17-7)). In this approach, the strain energy of the tissue was assumed to be the sum of the individual fber contributions. Lanir recognized that fbers have non-uniform waviness, and proposed to model the gradual straightening by probability distribution. Orientation dispersion of the fbers was described by either the mass fractions in discrete directions or a continuous density function. Following Lanir, a number of structural models have been developed for arterial tissues; see (Wuyts et al. [1995;](#page-17-8) Hurschler et al. [1997](#page-16-16); Zulliger et al. [2004;](#page-17-9) Roy et al. [2010;](#page-17-10) Agianniotis et al. [2011](#page-16-17); Zeinali-Davarani et al. [2015;](#page-17-11) Sacks [2003;](#page-17-12) Fan and Sacks [2014;](#page-16-18) Cacho et al. [2007](#page-16-19); Weisbecker et al. [2015;](#page-17-13) Wulandana and Robertson [2005;](#page-17-14) Hill et al. [2012](#page-16-20); Hamedzadeh et al. [2018;](#page-16-21) Rachev and Shazly [2019](#page-17-15)) for a selected few. These models difer in their descriptions of the waviness probability and the forms of the single fber energy function, but they share a distinct feature in common: the fber stress at the tissue level is formulated by a convolution or convolution-type integral over the probability space. In some special cases the fber stress can be obtained in closed form (Roy et al. [2010;](#page-17-10) Martufi and Gasser [2011](#page-17-16)), but in general the integral needs to be evaluated numerically. The angular dispersion is incorporated by either an integration over the unit sphere, or a direct sum over discrete directions. A full structural model considering both the recruitment and the continuous fber dispersion, therefore, can involve a triple integral. In numerical computation, structural models could invoke numerical integrations over the probability space and angular domains at the Gauss point level. In contrast, Gauss point computation in invariant-based models entails only function evaluation. To date the structural models have been used mostly at specimen level. Only lately some structural models have been tested in 3D deformation of real structures (Fan and Sacks [2014\)](#page-16-18).

In this work, we present a diferent framework for incorporating the fber recruitment. The point of departure lies in that the recruitment is dealt with at *kinematics* level, separated from kinetic considerations. An *efective stretch* is introduced. The efective stretch is a tissue-scale kinematic variable that represents, in an average sense, the true stretch of the tissue after factoring out the strengthening efect. It turns out to be a smooth function of the deformation. The notion of efective stretch or strain may be implied in some structural constitutive descriptions (Roy et al. [2010;](#page-17-10) Martuf and Gasser [2011\)](#page-17-16); however, it has never been elucidated in the literature, to the best knowledge of the authors. The efective stretch is then used in selected hyperelastic constitutive models to describe the tissue behavior. In this manner, the recruitment response is considered within the hyperelastic framework. Evaluation of stress follows from a systematic application of chain rule. In the present work, we focus on the HGO family SEFs. We demonstrate in detail how the Gasser-Ogden-Holzapfel (GOH) model (Gasser et al. [2006\)](#page-16-14) is transformed in light of the efective stretch.

The method of characterizing the waviness distribution advocated by Rezakhaniha et al. ([2012\)](#page-17-17) plays a pivoting role in the present development. Rezakhaniha et al investigated collagen waviness and orientation in the arterial adventitia using confocal laser scanning microscopy. They introduced a straightness parameter, which is a physical variable that can be directly measured from experiment, to quantify the waviness of a single fber. They found that the waviness distribution over the fber population was well described by beta distribution. Zeinali-Davaran et al used a recruitment stretch, which is reciprocal to the straightness, to parameterize the collagen undulation and benchmarked the distribution to normal and gamma functions (Zeinali-Davarani et al. [2015\)](#page-17-11). The beta distribution fts nicely into the kinematic framework adapted in this work; it allows the effective stretch to be expressed in closed form, by special functions that are directly available in common programing languages (FORTRAN, C, Matlab, Python, Mathematica, R, to name a few). This is signifcant for theoretical development and numerical computation.

The validity of the constitutive construction is investigated using two groups of experimental data. The frst group consists of 157 sets of uniaxial stress–stretch data harvested from human ascending thoracic aortic aneurysm (ATAA) tissues (Ferrara et al. [2016](#page-16-22), [2018](#page-16-23)). The aim of this test is to examine the descriptive capability. The data are fitted to a 1D HGO strain energy function. R^2 values from with or without the recruitment are compared. The second group contains bi-axial tension–strain data extracted from bulge-infation test of ATAA tissue specimens. Over each specimen, 2304 curves were garnered (Davis et al. [2015,](#page-16-24) [2016](#page-16-25); Luo et al. [2016\)](#page-17-18). A subset of these data are revisited in the current study. As before, the frst aim is to evaluate the descriptive capability. The tension-stretch data are ftted to a 2D GOH model, with and without the fber recruitment. Again the ftting quality is evaluated by comparing the *R*² values. The second aim is to evaluate the predictive capability as well as the robustness in fnite element simulation. To this end, fnite element models are developed to simulate the infation tests. The nodal displacements and the Gauss

point tension strain values from the fnite element analysis are compared with the experimental results. Finally, a synthetic model of an aorta segment undergoing pressure-driven expansion is introduced. The goal is to evaluate the performance in 3D simulation, especially for motions where the material is compressed in some directions. As will be seen in Sect. [3,](#page-4-0) we introduce two slightly different approximations for a strain invariant in the modifed 3D GOH function. In theory, they should be close. A side goal of this third test is to check whether they yield close results.

2 Theory of kinematic average

Theoretical underpinnings of the proposed kinematic average approach are presented in this section. The cornerstone is the concept of efective stretch, which is a macro-scale

$$
\psi(p; \alpha, \beta) = \frac{p^{\alpha - 1}(1 - p)^{\beta - 1}}{B(\alpha, \beta)}, \quad p \in (0, 1)
$$
\n(2.2)

where (α, β) are shape parameters, both positive and $B(\alpha, \beta) = \int_0^1 x^{\alpha-1}(1-x)^{\beta-1} dx$ is the beta function. If α < 1 or β < 1, the density function is unbounded when *p* approaches 0 or 1. These cases are not of physical relevance for the problem in hand, and hence, we impose $\alpha \geq 1, \beta \geq 1$. The mean straightness attains a simple expression

$$
E(p) = \frac{\int_0^1 p^{\alpha} (1 - p)^{\beta - 1} dp}{B(\alpha, \beta)} = \frac{\alpha}{\alpha + \beta}
$$
 (2.3)

If $\alpha > \beta$, the mean straightness is greater than 0.5 and the distribution is right-skewed (left-tailed). If $\alpha < \beta$, the mean straightness is less than 0.5. and the function skews left (right-tailed). The cumulative distribution function and its tail

$$
F(x; \alpha, \beta) = \frac{\int_0^x p^{\alpha - 1} (1 - p)^{\beta - 1} dp}{B(\alpha, \beta)}, F(x; \alpha, \beta, \text{ upper}) = \frac{\int_x^1 p^{\alpha - 1} (1 - p)^{\beta - 1} dp}{B(\alpha, \beta)}
$$
(2.4)

variable representing the true stretch of the tissue. This concept is developed in a multiscale context. At macro-scale (i.e., the tissue-scale), the deformation of a material line element is described by a macroscopic stretch. At the microscale (i.e., the fber-scale), a material element is associated with a family of fibers. Each fiber has its own waviness, and thus, the true stretches that the fbers experience are diferent. The efective stretch is the ensemble average of the fber-scale true stretches over the waviness distribution in the fber family.

2.1 Waviness distribution

We use the distribution model by Rezakhaniha et al. ([2012\)](#page-17-17) to characterize the fber waviness. They measured the undulation of a single fber by a straightness parameter

$$
p = \frac{L}{L_f} \tag{2.1}
$$

where *L* is end-to-end distance of the fiber and L_f is the fber's arclength in the ground state. The parameter takes value in (0, 1). The larger the value, the lesser the waviness. The reciprocal of p is the stretch at which the fiber becomes taut. Fibers have heterogeneous waviness. Rezakhaniha et al reported that the probability density of *p* was well described by beta distribution

will be referred in the sequel. Clearly, $F(x; \alpha, \beta) + F(x; \alpha, \beta, \text{upper}) = 1.$

2.2 True stretch of a single fber

Consider a single fiber whose curved length is L_f and the end-to-end length is *L* in the reference state. When it is subjected to a stretch which makes a deformed end-to-end length *l*, the kinematic stretch is

$$
\lambda = \frac{l}{L} \tag{2.5}
$$

Considering the fber waviness, the kinematic stretch can be decomposed into the product of a true stretch λ_t and a recruitment stretch λ_i :

$$
\lambda = \lambda_t \lambda_r
$$

The true stretch is a portion of stretch responsible for stress generation, while the recruitment stretch is the portion spent to uncrimp the fber. Their values depend on whether the fber is fully straightened or not. When $l > L_f$ (fully stretched), the effective stretch is the ratio of *l* over the arclength L_f , and thus

$$
\lambda_t := \frac{l}{L_f} = \frac{l}{L} \frac{L}{L_f} = \lambda \cdot p \tag{2.6}
$$

And the recruitment stretch is $\frac{L_f}{L} = \frac{1}{p}$. When $l \le L_f$, the total stretch is entirely spent to un-crimp the fber but merely

Fig. 1 a Schematic of fiber deformation. (A) The reference state; (B) partially uncrimped ($\lambda < \frac{1}{p}$); and (C) fully stretched ($\lambda > \frac{1}{p}$). **b** The fiber scale true stretch and the recruitment stretch plotted against λ in the case of $p = 0.8$

reduces its waviness. In this case $\lambda_r = \lambda$, and consequently $\lambda_t = 1$. These scenarios are illustrated in Fig. [1.](#page-3-0)

2.3 Efective stretch

Consider now the family of fbers contained in a material line element. When the element is subjected to a macroscopic stretch, it is natural to assume that all fibers undergo an affine deformation. Since the fibers have different waviness, they bear diferent values of true stretch. The efective stretch is defned as the ensemble average of the fber-scale true stretches over the stochastic waviness distribution:

$$
\bar{\lambda} = \int_0^1 \lambda_t \psi(p) \, dp \tag{2.7}
$$

According to the discussion in Sect. [2.2](#page-2-0), the true stretches in the fber family follow the distribution:

$$
\lambda_t = \begin{cases} 1 & \text{For fibers with } p \in (0, \lambda^{-1}) \\ \lambda \cdot p & \text{For fibers with } p \in (\lambda^{-1}, 1) \end{cases}
$$
 (2.8)

Taking into account this distribution, the efective stretch becomes

$$
\bar{\lambda} = \int_0^{\frac{1}{\lambda}} \psi(p) \, dp + \int_{\frac{1}{\lambda}}^1 \lambda p \psi(p) \, dp \tag{2.9}
$$

Upon using the beta distribution, the effective stretch reduces to a closed-form expression

$$
(2.8) \t\t\t\t\t\t1.8
$$

where we have used the transformation $\int_{\frac{1}{\lambda}}^{1}$ $\frac{p^{\alpha}(1-p)^{\beta-1}}{B(\alpha,\beta)}$ *dp* = $F(\lambda^{-1}; \alpha + 1, \beta, \text{upper}) \frac{B(\alpha+1,\beta)}{B(\alpha,\beta)}$ and the iden- $\lim_{B(\alpha,\beta)}$ = $\frac{\alpha}{\alpha+\beta}$.

The effective stretch is defined for $\lambda \geq 1$. As long as fber kinetics is concerned, we do not need to consider λ < 1 because a crimped fiber does not generate axial resistance when compressed further if we ignore efects from bending and lateral contact. For computational purpose, however, it is necessary to defne the efective

Fig. 2 Tissue-scale recruitment response in the case of $(\alpha, \beta) = (6.4, 1.8)$. The limiting slope of $\overline{\lambda}$ is $\frac{\alpha}{\alpha+\beta} = 0.78$. The asymptotic value of the recruitment stretch is $\frac{\alpha+\beta}{\alpha} = 1.28$

stretch for $\lambda < 1$. We set $\overline{\lambda} = 1$ in this case. Unless stated otherwise, subsequent discussions about $\bar{\lambda}$ are restricted to $\lambda \geq 1$.

Once $\bar{\lambda}$ is determined, the tissue-scale recruitment stretch can be defined by $\bar{\lambda}_r = \frac{\lambda}{\lambda}$. This is a postprocess variable; it complements $\bar{\lambda}$ in describing the recruitment response. An example of the recruitment response is shown in Fig. [2.](#page-3-1) It is worth noting that both $\bar{\lambda}$ and $\bar{\lambda}_r$ are *response variables*. This is in contrast with some studies, e.g., Li and Holzapfel. ([2019\)](#page-17-19), which employs a tissue scale recruitment stretch as a primitive constitutive parameter.

2.4 Efective stretch in 3D deformation

In three-dimensional deformation, material elements undergo affine deformation prescribed by the local deformation gradient **F**. At the continuum scale, a reference line element $d\mathbf{X}$ is mapped into $d\mathbf{x}$ via $d\mathbf{x} = \mathbf{F}d\mathbf{X}$. Let dS and \mathbf{N} be the length and direction of $d\mathbf{X}$, ds and **n** be the length and direction of $d\mathbf{x}$, the mapping is written as $\mathbf{n}d\mathbf{s} = \mathbf{F} \mathbf{N}d\mathbf{S}$. The kinematic stretch of the element is

$$
\lambda := \frac{ds}{dS} = \sqrt{\mathbf{N} \cdot \mathbf{C} \mathbf{N}}
$$
\n(2.11)

where $\mathbf{C} = \mathbf{F}^T \mathbf{F}$ is the Green–Lagrangian tensor. Note that λ is a function of **F** and **N** and when necessary it will be denoted as $\lambda(\mathbf{F}, \mathbf{N})$. At the fiber-scale, a material line element is associated with a subfamily of fbers whose endto-end vectors fall in a small cone around the direction N. The ensemble average over this subfamily yields the efective stretch in that direction. Thus, in 3D case, the efective stretch is a function of **and** $**N**$ **; it is defined by**

$$
\bar{\lambda}(\mathbf{F}, \mathbf{N}; \alpha, \beta) = f(\lambda(\mathbf{F}, \mathbf{N}); \alpha, \beta) \tag{2.12}
$$

When the context is clear, we will drop either \mathbf{F} , or \mathbf{N} , or (α, β) , or all, from the argument list. It is worth noting that in general, the waviness distribution can vary with direction (anisotropic waviness). In this work, we assume that it is the same in all directions (isotropic waviness). This assumption is consistent with the fndings in Rezakhaniha et al. ([2012\)](#page-17-17) which observed that the fber straightness parameters in the arterial adventitia do not depend on the mean orientation of the fbers.

2.5 Properties of *̄*

- 1. $\bar{\lambda}(1) = 1, 1 < \bar{\lambda} < \lambda \forall \lambda > 1$.
- 2. $\bar{\lambda}$ is a strictly increasing function;

3.
$$
\lim_{\lambda \to \infty} \bar{\lambda}'(\lambda) = \frac{\alpha}{\alpha + \beta};
$$

4. $\bar{\lambda}'(1) = 0;$

5. In 3D deformation, $\overline{\lambda}$ is a convex function of **F**.

Proofs are presented in Appendix [1.](#page-14-0) These properties are essential for modeling the gradual recruitment of fbers. The frst property ensures that there is no spurious compression. The second property guarantees that there is no spurious unloading. The third property is intuitive; it indicates that at large strains the efective stretch is the total stretch scaled by the mean straightness. The fourth property ensures that the transition from the undulated state to load bearing is smooth. The ffth property is critical for constructing polyconvex strain energy functions (Hartmann and Neff [2003](#page-16-26)).

We conclude this section by a remark on the choice of waviness parameters and distribution functions. Existing studies in the literature often used the recruitment stretch $\lambda_r = \frac{L_f}{L}$, which is the reciprocal of straightness parameter *p*, or the straightening strain $E_s = \frac{1}{2}(\lambda_r^2 - 1) = \frac{1}{2}(p^{-2} - 1)$, to describe the undulation of a single fber. To cite a few works, Cacho et al. [\(2007](#page-16-19)) and Weisbecker et al. ([2015\)](#page-17-13) used λ_r and beta distributions. Sacks ([2003](#page-17-12)) used E_s with a Gamma distribution. Zulliger et al. ([2004](#page-17-9), [2007\)](#page-17-20) and Fonck et al. [\(2007\)](#page-16-7) used *Es* with log-logistic distributions. Rodriguez et al. [\(2006,](#page-17-21) [2008](#page-17-22)) employed the maximum fber length with beta distributions. The present approach (*p* with the beta distribution) is motivated, on the one hand, by the experimental evidence that this combination best described the experimental data (Rezakhaniha et al. [2012](#page-17-17)), and on the other hand, by the algorithmic convenience it offers. If other distribution functions were used, it would be impossible to obtain a closed-form efective stretch. In addition, the straightness parameter is defined over the canonical domain $(0, 1)$. If λ_r or *Es* were used, these parameters have to be bounded in subjectively selected intervals. However, the choice of bounds would afect the shape parameters. In some cases, the shape parameters could be very sensitive to the bounds.

3 Constitutive framework

In structure-motivated hyperelastic models for the arterial tissues, the strain energy functions are constructed as the sum of two terms: a function W_g which accounts for the contribution of ground substances, and an energy function W_f which represents the contribution of fibers

$$
W = W_g(\mathbf{C}) + W_f(\mathbf{C})
$$
\n(3.1)

Here, $W_{a}(\mathbf{C})$ is an isotropic function and is represented by principal invariants of C:

$$
I_1 = \text{tr} \mathbf{C}, \ I_2 = \frac{1}{2} (I_1^2 - \text{tr} \mathbf{C}^2) \tag{3.2}
$$

The third invariant $I_3 = \det C$ does not present because of incompressibility. The Cauchy stress is determined to within a hydrostatic pressure. In numerical computation, a volumertic term $U(J)$ with $J = \det \mathbf{F}$, or a penalty function $p(J - 1)$, would be augmented. Note that *p* is the hydrostatic pressure and it should not be confused with the straightness parameter used earlier. The fber energy function is anisotropic. Vascular tissues typically contain several distinct families of fibers, each identified by a preferred direction M_i . To reflect the fber structure, most structurally motivated models chose to represent the fber energy function using some of the following invariants

$$
I_1, I_2, J_1, J_2, \cdots, J_N, K_1 K_2, \cdots, K_N
$$
\n(3.3)

where $\{J_i := \mathbf{C} : \mathbf{M}_i \otimes \mathbf{M}_i\}_{i=1}^N$ are squared stretches in each preferred direction, and ${K_i^- := J^2 \mathbf{C}^{-1} : \mathbf{M}_i \otimes \mathbf{M}_i}_{i=1}^N}$ are the (squared) area stretches in planes transverse to the fber directions. The subgroup

$$
I_1, J_1, J_2, \cdots, J_N \tag{3.4}
$$

are line stretch-based measures; J_I 's are squared stretches in the fiber directions and I_1 is the average squared stretches (modulo a factor of 3) over all directions. It can be formulated by the spherical integral

$$
I_1 = \frac{3}{4\pi} \int_{S^2} \lambda^2(\mathbf{M}) \, da \tag{3.5}
$$

where S^2 is the surface of the unit sphere and da in an infinitesimal area element. In spherical coordinates, $da = \sin \phi d\phi d\theta$, **, and** $\lambda^2(\mathbf{M}) =$ **𝐂** ∶ **𝐌** *⊗* **𝐌**. The invariants

$$
I_2, K_1, K_2, \cdots, K_N
$$
\n(3.6)

are area stretch-based measures; I_2 (to a factor of 3) is the average of squared area stretch over all directions. Most models used today are based on the line stretch measures. For this reason, we will focus on functions represented by the subgroup (3.4) (3.4) (3.4) . Strain energy functions that involve the subgroup ([3.6\)](#page-5-1), such as ones considered in (Balzani et al. [2006\)](#page-16-27), will be dealt in a more general setting where an efective deformation tensor will be introduced.

Generally speaking, the proposed framework for considering the fber recruitment amounts to replacing the invariants (3.4) in W_f with their counterparts

$$
\bar{I}_1, \bar{J}_1, \bar{J}_2, \cdots, \bar{J}_N \tag{3.7}
$$

while keeping W_g intact. Here, ${\{\bar{J}_I = f^2(\sqrt{J_I})\}}_{I=1}^N$ are the squared effective stretches of the corresponding fiber

stretches, and \bar{I}_1 is the spherical average of squared effective stretches

$$
\bar{I}_1 = \frac{3}{4\pi} \int_{S^2} \bar{\lambda}^2(\mathbf{M}) \, da \tag{3.8}
$$

In what follows we demonstrate the modifcation to the energy function of the Gasser-Ogden-Holzapfel model.

Gasser-Ogden-Holzapfel model. The GOH model (Gasser et al. [2006](#page-16-14)) has been broadly adapted to describe the vascular tissue response. It is an extension of the HGO function that takes into account of the directional distribution of fbers. In the GOH model, it is assumed that the tissue contains several distinct families of fbers, as marked by their mean orientations, and fbers in each family possess certain dispersion. A particular energy function for a two fber family tissue, wherein the fber stifnesses are the same while the dispersions difer, with a neo-Hookean function for the ground substance, takes the form

$$
W = \underbrace{\frac{\mu_1}{2}(I_1 - 3)}_{W_g} + \underbrace{\sum_{i=1}^{2} \frac{\mu_2}{4\gamma} \left(e^{\gamma(I_{ki} - 1)^2} - 1\right)}_{W_f}
$$
(3.9)

Here, I_{ki} is a weighted average of squared stretches in the *i*-th family considering the angular dispersion:

$$
I_{\kappa i} = \frac{1}{4\pi} \int_{S^2} \rho(\phi, \theta) \lambda^2(\mathbf{M}) \, da \tag{3.10}
$$

where ρ is the orientation density function, normalized such that $\frac{1}{4\pi}$ $\int_{S^2} \rho \, da = 1$. Under the assumption that ρ depends only on ϕ , and letting $\kappa_i = \frac{1}{4} \int_0^{\pi} \rho(\phi) \sin^3 \phi d\phi$, the invariant reduces to

$$
I_{ki} = \kappa_i I_1 + (1 - 3\kappa_i) J_i, \quad J_i = \mathbf{M}_i \cdot \mathbf{C} \mathbf{M}_i
$$
\n(3.11)

where M_i is the preferred direction of the *i*-th family. The parameter κ_i thus represents the dispersion. When $\kappa_i = 0$, all fibers are aligned in the preferred direction. When $\kappa_i = \frac{1}{3}$, the fbers are randomly distributed, giving an isotropic response. When $\kappa_i = \frac{1}{2}$, the fibers are concentrated on the transverse plane and randomly distributed thereon.

Considering the fber recruitment, we defne

$$
\bar{I}_{\kappa i} = \frac{1}{4\pi} \int_{S^2} \rho(\phi, \theta) \bar{\lambda}^2 da \quad \text{with} \quad \frac{1}{4\pi} \int_{S^2} \rho(\phi, \theta) da = 1
$$
\n(3.12)

To facilitate numerical computation, we approximate the integral using Lebedev quadratures, which are designed to exactly integrate polynomials up to a certain order over the unit sphere (Lebedev [1976](#page-17-23)). A 6-point and a 14-point quadratures are recoded in Appendix [2](#page-15-0). Some higher-order formulas can be found in (Lebedev [1975,](#page-17-24) [1977](#page-17-25)). To this

end we choose a local basis ${\{\mathbf{E}'_j\}}_{l=1}^3$ (called the fiber basis hereafter) such that \mathbf{E}'_3 aligns with the fiber direction \mathbf{M}_i . The other two basis vectors can be chosen with the aid of anatomic information. For a two-family model, \mathbf{E}'_1 is set to lie in the fiber plane. In this case \mathbf{E}'_2 naturally points to the thickness direction. To start with, we employ the six point quadrature. The integration points are taken to be the vertices of the local basis vectors, $\{\pm \mathbf{E}'_1, \pm \mathbf{E}'_2, \pm \mathbf{M}_i\}$. The weights are uniform and equal to $\frac{4\pi}{6}$. This gives

$$
\bar{I}_{\kappa i} = \frac{1}{3} \left(\rho(\mathbf{E}'_1) \bar{\lambda}^2(\mathbf{E}'_1) + \rho(\mathbf{E}'_2) \bar{\lambda}^2(\mathbf{E}'_2) + \rho(\mathbf{M}_i) \bar{\lambda}^2(\mathbf{M}_i) \right)
$$
(3.13)

The normalization condition, discretized by the same scheme, yields

$$
\frac{1}{3} \left(\rho(\mathbf{E}'_1) + \rho(\mathbf{E}'_2) + \rho(\mathbf{M}_i) \right) = 1 \tag{3.14}
$$

Under the assumption that ρ depends only on the meridional angle ϕ , $\rho(\mathbf{E}'_1) = \rho(\mathbf{E}'_2)$. Let $\frac{\rho(\mathbf{E}'_1)}{3} = \kappa$, we obtain

$$
\overline{I}_{\kappa i} = \kappa (\overline{\lambda}^2 (\mathbf{E}'_1) + \overline{\lambda}^2 (\mathbf{E}'_2)) + (1 - 2\kappa) \overline{\lambda}^2 (\mathbf{M}_i)
$$
(3.15)

The parameter κ retains its original caliber. When the fibers are randomly distributed in all directions ($\rho = 1$), $\kappa = \frac{1}{3}$. When the fibers align perfectly with M_i , $\rho(E'_1) = \rho(E'_2) = 0$ and thus $\kappa = 0$. When the fibers fall on the transverse plane, $\rho(\mathbf{M}_i) = 0$; upon using the discretized normalization condition and the randomness assumption we conclude $\kappa = \frac{1}{2}$.

With the new invariant, the energy function becomes

$$
W = \frac{\mu_1}{2}(I_1 - 3) + \sum_{i=1}^{2} \frac{\mu_2}{4\gamma} \left(e^{\gamma(\bar{I}_{ki} - 1)^2} - 1 \right)
$$
 (3.16)

Since the ground substance term remains intact, we will omit it in the discussion and focus on the fber response only.

The fiber stress $S_f := 2 \frac{dW_f}{dC}$ is

 $\bar{\lambda}(\pm N_I) = f(\lambda_I) := \bar{\lambda}_I$ where λ_I 's are the principal stretches. This gives

$$
\bar{I}_1 = \sum_{I=1}^3 \bar{\lambda}_I^2 \tag{3.19}
$$

The ensuing stress functions are recorded in Appendix [3](#page-15-1).

Note that the invariant \overline{I}_k in [\(3.15](#page-6-0)) can be written as

$$
\bar{I}_\kappa = \kappa (\bar{\lambda}^2 (\mathbf{E}'_1) + \bar{\lambda}^2 (\mathbf{E}'_2) + \bar{\lambda}^2 (\mathbf{M})) + (1 - 3\kappa) \bar{\lambda}^2 (\mathbf{M}) \quad (3.20)
$$

Comparing with (3.18) (3.18) (3.18) , the difference is that the invariant $\bar{\lambda}_1^2 + \bar{\lambda}_2^2 + \bar{\lambda}_3^2$ is replaced by $\bar{\lambda}^2(\mathbf{E}'_1) + \bar{\lambda}^2(\mathbf{E}'_2) + \bar{\lambda}^2(\mathbf{M})$. They are two different approximations of \overline{I}_1 . Without fiber recruitment they are exactly the same. It is expected that $\bar{\lambda}^2(\mathbf{E}'_2) + \bar{\lambda}^2(\mathbf{E}'_3) + \bar{\lambda}^2(\mathbf{M}) \approx \bar{\lambda}_1^2 + \bar{\lambda}_2^2 + \bar{\lambda}_3^2$

Note that \overline{I}_k defined in ([3.15\)](#page-6-0) is no longer an invariant *per se* because it depends on the choice of fber basis. Nonetheless, it leads to a simpler computation. These two formulations are compared in a 3D fnite element simulation reported later.

4 Validation and verifcation

4.1 Uniaxial response of ATAA tissue

By courtesy of Dr. Auricchio, uniaxial stress–stretch data of ATAA tissues collected at his laboratory were made available to this study. The experimental protocols were documented in Ferrara et al. [\(2016,](#page-16-22) [2018](#page-16-23)). A total of 187 specimens were excised from 68 patients. From each specimen, a stretch-stress curve was obtained. Through pre-screening, 30 curves deemed too noisy were excluded from this study. We elect to describe the response using a 1D HGO energy function

$$
W = \frac{\mu_1}{2} \left(\lambda^2 + \frac{2}{\lambda} - 3 \right) + \frac{\mu_2}{4\gamma} \left(e^{\gamma (\lambda^2 - 1)^2} - 1 \right)
$$
 (4.1)

$$
\mathbf{S}_{f} = \sum_{i=1}^{2} \mu_{2} e^{\gamma(\bar{I}_{ki}-1)^{2}} (\bar{I}_{ki}-1) \left(\kappa_{i} \left(\sum_{I=1}^{2} \frac{d(\bar{\lambda}^{2}(\mathbf{E}_{I}^{\prime}))}{d(\lambda^{2}(\mathbf{E}_{I}^{\prime}))} \mathbf{E}_{I}^{\prime} \otimes \mathbf{E}_{I}^{\prime} \right) + (1 - 2\kappa_{i}) \frac{d\bar{J}_{i}}{dJ_{i}} \mathbf{M}_{i} \otimes \mathbf{M}_{i} \right) \tag{3.17}
$$

The material tangent tensor and Cauchy response functions are recorded in Appendix[3.](#page-2-1)

Alternative representation of \overline{I}_k . Alternatively, \overline{I}_k can be constructed from the reduced form (3.11) by replacing I_1 with \bar{I}_1 and J_i with \bar{J}_i :

$$
\bar{I}_{ki} = \kappa_i \bar{I}_1 + (1 - 3\kappa_i)\bar{J}_i, \quad \bar{J}_i = f^2(\sqrt{J_i})
$$
\n(3.18)

where \bar{I}_1 is defined in ([3.8](#page-5-3)). In numerical computation, the integral is approximated by a six-point Lebedev quadrature. The integration points are taken to be the vertices of the principal vectors, $\{\pm \mathbf{N}_I\}_{I=1}^3$. At these points The frst term, which represents the ground substance contribution, is reduced from the neo-Hookean energy function μ_1 $\frac{u_1}{2}(I_1 - 3)$ upon imposing the condition $\lambda_2 = \lambda_3 = \frac{1}{\sqrt{\lambda}}$, where λ_2 and λ_3 are the principal stretches occurring in the transverse plane. It is assumed that the tissue strip contracted isotropically in lateral directions. The second term is the fiber energy, and is defined for $\lambda \geq 1$ only. The Cauchy stress follows from the formula $\sigma = \lambda \frac{dW}{d\lambda}$. Without considering the fber recruitment,

$$
\sigma = \mu_1(\lambda^2 - \lambda^{-1}) + \mu_2 e^{\gamma(\lambda^2 - 1)^2} (\lambda^4 - \lambda^2)
$$
 (4.2)

Fig. 3 Comparison of R^2 values with and without fiber recruitment in all 157 samples. Each dot represents a sample. The abscissa is the $R²$ value without fiber recruitment; the ordinate is the corresponding value with fber recruitment. PP: percentage pf points which show improvement in the R^2 value

With the fiber recruitment, the stretch λ in the second term is replaced by $\bar{\lambda}$. This gives

$$
\sigma = \mu_1(\lambda^2 - \lambda^{-1}) + \mu_2 e^{\gamma(\bar{\lambda}^2 - 1)^2} (\bar{\lambda}^3 - \bar{\lambda}) \lambda \bar{\lambda}'(\lambda) \tag{4.3}
$$

The stress–strain data were ftted to both the recruitment model (Eq. (4.3) (4.3)) and the original model (Eq. (4.2)). Optimized material parameters were determined by minimizing the sum of square diference between the experimental stress values and the model predictions. For the recruitment model, the parameters include the elastic properties (μ_1, μ_2, γ) and the waviness parameters (α, β) . The latter are absent in the original model. These parameters were constrained such that $(\mu_1, \mu_2, \gamma) > 0$, $1 \leq (\alpha, \beta) \leq 100$. Since the goal is not to obtain the constitutive parameters, the parameters are not generally reported.

Improvement in fitting quality. The R^2 values of all 157 samples are compared in Fig. [3.](#page-7-1) The percentage of points (PP) which show improvement in the R^2 value reaches 98.7%. Clearly, the ftting quality are universally improved when the recruitment is considered.

Figure [4](#page-7-2) presents three examples of fitted curves; the corresponding points are marked with red circles in Fig. [3](#page-7-1). The fiber waviness parameters (α, β) , and the R^2 values from both models, are annotated on the fgure. The frst example is a representation of the majority of cases. The improvement is garnered in the toe region. This is the region where the graduate fber recruitment takes place. The original model categorically over-predicts the stress, resulting a curve that lies above the experimental data points. The over-stress is rectifed by using the efective stretch. The second example is an extreme case in this category. For this sample, the fbers are predicted to be nearly

Fig. 4 Fitted response curves from three samples. (α, β) : Fitted waviness parameters. (R_1^2, R_2^2) : R^2 values without and with considering the fber recruitment, respectively

straight $(E(p) = 0.98)$. In this case, the energy functions with or without recruitment would perform almost identically. It so happens that the original model already fts the data very well and thus little is gained from the recruitment consideration. The third example reveals another

avenue the improvement is reaped. In this example, the original exponential function fails to ft well in both the low strain and the higher strain regions. When the recruitment is considered, the parameter γ comes out practically zero ($\gamma = 1.24e - 08$, as opposed to $\gamma = 2.49$ without fber recruitment). This suggests that the strain energy is efectively a quadratic function of the efective strain. In other words, when the straightening efect is considered, the fber stress (in terms of the second Piola–Kirchhof stress) is a linear function of the efective Green–Lagrangian strain. This linear relation is nicely captured by the recruitment model.

4.2 Bulge‑infation tests of ATAA tissue

In collaboration with Dr. Avril, our teams collected bi-axial response data of ATAA tissues through bulge inflation test (Davis et al. [2015](#page-16-24), [2016](#page-16-25); Luo et al. [2016;](#page-17-18) Romo et al. [2014\)](#page-17-26). Briefly, ATAA specimens of approximated $4 \times 4 \text{ cm}^2$ were subjected to infation test. The specimens were loaded incrementally at 3 kPa pressure increment until rupture. The deformed confguration was recorded using Digital Image Correlation (DIC), and deforming NURBS meshes were constructed for the middle regions of the specimens surfaces. Figure [5](#page-8-0) illustrates an example. The authors used the inverse stress analysis (Lu et al. [2007](#page-17-27); Zhou et al. [2010](#page-17-28); Lu et al. [2013](#page-17-29); Lu and Luo [2016](#page-17-30)) to compute the tension distribution in each deformed state. The predicted tensions were taken to be the experimental data. The surface strains were extracted from the DIC images. In this manner, 2304 tension–strain curves were collected from each specimen. The data from two specimens are used in the present study.

Since the specimens were generally thin compared to the lateral dimensions and they underwent bulge expansion, it is reasonable to treat the material as planer. Also, it has been reported that the collage structure in ATAAs is typically

disrupted (de Figueiredo Borges et al. [2008\)](#page-16-28), and thus, there is no compelling reason to assume the existence of directionally distinct families of fbers. For these reasons, a 2D material model reduced from the GOH function, which considers only one fber family, is employed:

$$
W = \frac{1}{2}(I_1 + I_2^{-1} - 3) + \frac{\mu_2}{4\gamma} \left(e^{\gamma(I_k - 1)^2} - 1\right)
$$
 (4.4)

The frst term is associated with the ground substance and is reduced from the neo-Hookean function $\frac{\mu_1}{2}(\lambda_1^2 + \lambda_2^2 + \lambda_3^2 - 3)$ upon substituting $\lambda_3 = \lambda_1^{-1} \lambda_2^{-1}$ from the incompressibility condition. Note that the deformation tensor C is 2×2 , and $I_1 = \text{tr} \mathbf{C} = \lambda_1^2 + \lambda_2^2$ is the first principal invariant, where (λ_1, λ_2) λ_2) are the in-plane principal stretches. $I_2 = \lambda_1^2 \lambda_2^2$ is the second principal invariant. The second term accounts for the fber contribution. It is assumed that the fbers are distributed symmetrically around a preferred direction **M**, with a certain degree of angular dispersion. The invariant I_k is the planer average of squared fber stretches weighted by the directional density. Without considering the recruitment, I_k has the form

$$
I_{\kappa} = \kappa I_1 + (1 - 2\kappa)J_1, \quad J_1 = \mathbf{M} \cdot \mathbf{CM}
$$
 (4.5)

where $\kappa \in [0, 0.5]$ is the dispersion parameter. When $\kappa = 0.5$, the fibers are randomly distributed, giving an isotropic response. When $\kappa = 0$, all fibers are perfectly aligned in the direction **M**. When $0 < \kappa < 0.5$, the response is planar orthotropic. See Holzapfel and Ogden [\(2010\)](#page-16-15). The second Piola–Kirchhoff stress is

$$
\mathbf{S} = \mu_1 (\mathbf{I} - I_2^{-1} \mathbf{C}^{-1}) + \mu_2 e^{\gamma (I_k - 1)^2} (I_k - 1)(\kappa \mathbf{I} + (1 - 2\kappa) \mathbf{M} \otimes \mathbf{M})
$$
\n(4.6)

where **I** is the 2 \times 2 identity tensor. With the recruitment, I_k is replaced by

$$
\bar{I}_k = \kappa \bar{J}_{\perp} + (1 - \kappa) \bar{J}_1, \quad \bar{J}_{\perp} = \bar{\lambda}^2 (\mathbf{M}_{\perp}), \ \bar{J}_1 = \bar{\lambda}^2 (\mathbf{M}) \tag{4.7}
$$

where M_{\perp} the perpendicular direction. Consequently,

$$
\mathbf{S} = \mu_1 (\mathbf{I} - I_2^{-1} \mathbf{C}^{-1}) + \mu_2 e^{\gamma (\bar{I}_k - 1)^2} (\bar{I}_k - 1) \left(\kappa \frac{d \bar{J}_\perp}{d J_\perp} \mathbf{M}_\perp \otimes \mathbf{M}_\perp + (1 - \kappa) \frac{d \bar{J}_1}{d J_1} \mathbf{M} \otimes \mathbf{M} \right)
$$
(4.8)

Fig. 5 NURBS mesh superimposed on the DIC point cloud at pressures 12 KPa (**left**) and 48 KPa (**right**)

Fig. 6 Pointwise R^2 values with and without considering the fiber recruitment. The abscissa is the value without fber recruitment, and the ordinate is the value with the recruitment. The PP values are the

Table 1 Statistics of the $E(p)$ distributions: means \pm standard deviations (SD), medians, and inter-quartile ranges [*Q*1, *Q*3] with *Q*1 and *Q*3 being the 25th and the 75th percentiles, respectively

Variable	$Mean + SD$	Median	[01, 03]
Specimen	$0.891 + 0.074$	0.912	[0.851, 0.951]
	$0.928 + 0.103$	0.976	[0.854, 0.984]

The material tensor can be readily obtained by reduction from the 3D case.

4.2.1 Pointwise identifcation of material properties

The constitutive parameters are determined by minimizing the sum of the squares diference between the tension

Fig. 7 Tension-stretch curves at a Gauss point, obtained with (left) and without (right) considering the recruitment. (t_{11}, t_{22}, t_{12}) : components of tension in the principal strain axes. Note that t_{12} is nonzero,

percentage of points at which the fitting quality improves. (R_1^2, R_2^2) : R^2 values without and with fber recruitment, respectively

components computed from the inverse stress analysis (the *experiment values*) and the model predictions. All three components are included. The data are ftted to stress functions (4.6) and (4.8) (4.8) (4.8) . For (4.8) (4.8) , the constitutive parameters include $(\alpha, \beta, \mu_1, \mu_2, \gamma, \theta, \kappa)$, while for ([4.6](#page-8-1)) the waviness parameters are absent. The following constraints are imposed: $(\mu 1, \mu 2, \gamma) > 0$, $0 \le \theta \le \pi$, $0 \le \kappa \le 0.5$, and $1 \leq (\alpha, \beta) \leq 100$. The fitting is carried out *individually at every Gauss point*, to obtain the heterogeneous property distributions. Again, the parameters are not generally reported as the goal is to compare the ftting qualities.

Improvement in fitting quality. Figure [6](#page-9-0) depicts the pointwise R^2 values associated with the stress functions (4.6) (4.6) (4.6) and ([4.8](#page-8-2)) in Specimens 1 and 2. Each subfgure corresponds to a specimen and contains 2304 data points. The mean *R*² values (R_2^2 : with fiber recruitment; R_1^2 : without) and the

although small. Without considering the recruitment, t_{11} and t_{22} curves lie above the experimental points in the low strain region

PP values are also presented. Overall the ftting quality is improved when taking into account the fber recruitment, with PP values greater than 93% for both specimens. Without recruitment, the fitting quality in Specimen 1 ($R_1^2 = 0.922$) is poorer than in Specimen 2 ($R_1^2 = 0.961$). With the recruitment, the R_2^2 values are similar. Specimen 1 shows a larger margin of improvement. We submit that the diference in the margin relates directly to the waviness; the wavier the specimen, the greater the improvement. To verify this, statistics of *E*(*p*) distribution in each specimen are summarized in Table [1.](#page-9-1) Clearly, fbers in Specimen 1 are wavier, in congruence with the conjecture.

Figure [7](#page-9-2) presents ftted tension components at a Gauss point in Specimen 1 for visual inspection. To facilitate plotting, the tension components are rotated to the principal strain axes, and plotted against the principal stretches. As in the 1D case, the ftted curves without recruitment fail to accurately capture the response in the low strain region. The components t_{11} and t_{22} lie above the experimental points. In contrast, the recruitment model fits the experimental data very well over the entire stretch range.

4.2.2 Forward validation

Finite element models are developed to simulate the infation tests. The specimens are modeled as membranes; the identifed pointwise constitutive parameters are mapped back to defne the heterogeneous anisotropic properties. The mesh boundary is displacement-driven, prescribed by the motion recorded by the DIC. A transverse follower pressure is applied on the specimen surface.

Contours of total displacement, total tension (i.e., the norm) and total stain are presented in Figs. [8](#page-11-0) and [9.](#page-12-0) Figure [8](#page-11-0) shows the fnite element predictions with and without fber recruitment at 48 KPa pressure, along with the experimental data, for Specimen 1. Figure [9](#page-12-0) does the same for Specimen 2 at 33 KPa. It is evident that the model with the fber recruitment reproduces the tension and strain distributions with excellent fdelity. Intricate local features are replicated. The model without recruitment captures the global patterns generally well, but fails to accurately match the local features.

To further assess the fdelity of the fnite element solutions, the mean absolute percentage errors (MAPEs) of the displacement, the tension and the strain over the domain are computed at every pressure steps. The MAPE is computed according to

$$
MAPE(\cdot) = \frac{1}{n} \sum_{t=1}^{n} \left| \frac{E_t(\cdot) - P_t(\cdot)}{E_i(\cdot)} \right|,
$$
\n(4.9)

where E_t stands for the experimental value, P_t is the finite element prediction, and *n* represents the number of Gauss points or nodes. The evolution of the MAPEs with pressure

steps is shown in Fig. [10,](#page-13-0) where the MAPE values at 7 pressure steps were plotted. Evidently, the MAPESs from the fnite element model with the fber recruitment are far below their counterparts without the fber recruitment, in all categories. For Specimen 1, the MAPEs of tension and strain fall below 2.5% after 40 KPa. For Specimen 2, the values are below 1% after 20 KPa. In these pressure ranges the displacement errors are near zero. The fnite element models with the fber recruitment clearly outperform the ones without.

4.3 Pressure‑driven expansion of an artery segment

As alluded before, this is a synthetic example introduced to test the performance of the 3D recruitment GOH function (3.16) (3.16) (3.16) in finite element simulation. In Sect. [3](#page-4-0), we introduced two approximations for the strain invariant \bar{I}_k . The purpose of this test is to check the influence of the approximations. The constitutive parameters are obtained by fitting stress–strain data selected from the uniaxial response database employed in Sect. [4.1.](#page-6-4) A hypothetical 3D setting is assumed. The loading direction is taken to be a principal stretch (λ_i) direction. The tissue is assumed to have two families of fibers symmetrically oriented with respect to the loading direction. Considering the incompressibility, the deformation gradient is then given as $$ fiber plane is numerically identified together with constitutive parameters by minimizing the sum of the squared difference between full 3D stress components, wherein the zero stress conditions along the transverse and the thickness directions are imposed. Two sets of parameters are acquired, and both are used in the simulation. The obtained constitutive parameters are listed in Table [2.](#page-13-1)

The finite element model is shown in Fig. [11](#page-13-2). The inner radius, wall thickness, and length are taken to be $R = 4.745$ mm, $H = 0.43$ mm, and $L = 5$ mm, respectively. Due to symmetry, only a quarter of the structure is considered. The top and bottom ends are constrained in the axial direction, and symmetry conditions are applied on the symmetry planes.

Quasi-static deformation is simulated over the pressure range of (0, 25)KPa. The solution converged smoothly at each time step, without any numerical issue. Pressure–radius (PR) curves are presented in Fig. [12.](#page-14-1) Two forms of stress functions, arising from the different approximations of \bar{I}_k , are used. The PR curves are very close for both materials.

Fig. 8 Contours of the total displacement (mm, **frst row**), total tension (N/mm, **second row**) and total strain (**third row**) from the experiment (**a, d, m**), fnite element simulation with the recruitment

(**b, e, n**), and fnite element results without recruitment (**c, f, k**), at 48 KPa pressure, in Specimen 1. The experimental tension (**d**) was obtained from the inverse stress analysis

5 Concluding remarks

We presented a framework for considering the graduate recruitment of collagen fbers in hyperelastic modeling of vascular tissues. The salient feature of this approach is that the recruitment is considered at the kinematics level. The effective stretch, which is a smooth response function embedding the gradual recruitment, is the cornerstone of the construction. The efective stretch is invoked to modify strain invariants, which are then used in selected hyperelastic SEFs to model the tissue behavior. We have implemented the recruitment model in selected HGO strain energy functions. The ensuing models

Fig. 9 Contours of the total displacement (mm, **frst row**), total tension (N/mm, **second row**) and total strain (**third row**) from the experiment (**a, d, m**), fnite element simulation with the recruitment

(**b, e, n**), and fnite element results without recruitment (**c, f, k**), at 33 KPa pressure, in Specimen 2. The experimental tension (**d**) was obtained from the inverse stress analysis

are essentially extensions of HGO family SEFs with two more parameters to take into account of the random fber waviness. We validated this constitutive construction against experimental data. Models incorporated the fber recruitment signifcantly improve the ftting quality, and overwhelmingly outperform the original models in reproducing the experimental results.

This approach permits us to consider the fber recruitment within the hyperelastic framework. Constitutive models constructed in this way preserve the invariant formulation. They are thus efficient in numerical computation. Owing to the beta distribution, the efective stretch is expressed by incomplete beta functions which are directly available in major programing languages. We have shown that, for the modifed HGO family SEFs, the stress functions entail only minor modifcations to the original formulation. In addition, the construction preserves the polyconvexity of the HGO

Fig. 10 Mean absolute percentage error (MAPE) of displacement (**black**), tension (**red**) and strain (**green**) at diferent pressure steps, from the models with the fber recruitment (**solid line**) and without (**dash line**)

family SEFs, because the effective stretch is a convex function of **F**. This feature is not discussed in the main text, but it is a key factor underlying the excellent numerical stability observed in the fnite element simulations.

On the other hand, since the response of a material element is formulated in terms of the average deformation of fbers, this approach is not fexible for taking into account of individual fber behavior. The limitation should be kept in

Fig. 12 Pressure–radius curves showing the infuence of diferent definitions of \overline{I}_k in the global response. Solid curves: responses arising from representation of \overline{I}_k using the principal basis. Dashed curves: using the fber basis

mind. Furthermore, a model constructed in this framework may inherit limitations embedded in the parent SEF. For example, the invariant I_r in the GOH model does not differentiate extended and compressed fbers in a dispersed family, and consequently, its validity in the situations where the fbers are partially in compression becomes questionable (Lanir and Namani [2015\)](#page-17-31). Several remedies have been proposed within the framework of invariant formulation (Holzapfel and Ogden [2017](#page-16-29); Li et al. [2018\)](#page-17-32). Presumably, the recruitment treatment can also be incorporated into the modifed SEFs. Issues of this nature should be addressed case by case.

A straightforward computation shows that the second and third terms cancel out, leaving

$$
\bar{\lambda}' = \frac{\alpha}{\alpha + \beta} F(\lambda^{-1}; \alpha + 1, \beta, \text{upper})
$$
\n(4.11)

Clearly,

$$
\bar{\lambda}'(1) = 0, \quad \lim_{\lambda \to \infty} \bar{\lambda}'(\lambda) = \frac{\alpha}{\alpha + \beta} \tag{4.12}
$$

Properties 3 and 4 are proved. Further, it is evident that

$$
0 < \bar{\lambda}' < 1 \quad \forall \lambda \in (1, \infty) \tag{4.13}
$$

and

$$
\bar{\lambda}(1) = 1\tag{4.14}
$$

Properties 1 and 2 are the consequences of these results. First, $\bar{\lambda}' > 0$ indicates that $\bar{\lambda}$ is a strictly increasing function in $(1, \infty)$. Consequently,

$$
\bar{\lambda}(\lambda) > \bar{\lambda}(1) = 1 \quad \forall \lambda \in (1, \infty)
$$
\n(4.15)

Similarly, $\bar{\lambda}'$ < 1 implies that $\bar{\lambda}(\lambda) - \lambda$ is a strictly decreasing function in $(1, \infty)$, and thus

$$
\bar{\lambda}(\lambda) - \lambda < \bar{\lambda}(1) - 1 = 0 \quad \forall \lambda \in (1, \infty) \tag{4.16}
$$

We thus conclude that

$$
1 < \bar{\lambda}(\lambda) < \lambda \quad \forall \lambda \in (1, \infty) \tag{4.17}
$$

To show that $\bar{\lambda}(\mathbf{F})$ is a convex function of **F**, first observe that $\bar{\lambda}' = \int_{\frac{1}{\lambda}}^{1}$ $\frac{p^{\alpha}(1-p)^{\beta-1}}{B(\alpha,\beta)}dp$, and hence

$$
\bar{\lambda}'' = \frac{d}{d\lambda} \int_{\frac{1}{\lambda}}^{1} \frac{p^{\alpha}(1-p)^{\beta-1}}{B(\alpha,\beta)} dp = \frac{\left(\frac{1}{\lambda}\right)^{\alpha+2} \left(1 - \left(\frac{1}{\lambda}\right)\right)^{\beta-1}}{B(\alpha,\beta)} > 0 \quad \text{for} \quad \lambda \in (1,\infty)
$$
\n(4.18)

Lastly, it is assumed that the waviness distribution is isotropic. This assumption could be reasonable for artery adventitia (Rezakhaniha et al. [2012\)](#page-17-17), but could be questionable for other vascular tissues. Anisotropic waviness distributions will be considered in future work.

Appendix 1 Proof of properties of *̄*

Properties 1, 2, 3, and 4 hinge on the derivative of $\bar{\lambda}$:

By construction, $\bar{\lambda} = 1$ for $\lambda \leq 1$, which is convex. Thus, $\bar{\lambda}$ is a convex function of λ over $(0, \infty)$. It follows that

$$
\bar{\lambda}(\lambda_2) - \bar{\lambda}(\lambda_1) \ge \frac{d\bar{\lambda}}{d\lambda}\bigg|_{\lambda = \lambda_1} (\lambda_2 - \lambda_1) \quad \text{for } \lambda \in (0, \infty) \quad (4.19)
$$

At the same time, $\lambda = \sqrt{FN \cdot FN}$ is a convex function of **F** for any fixed N (Hartmann and Neff [2003](#page-16-26)), so that

$$
\lambda(\mathbf{F}_2) - \lambda(\mathbf{F}_1) \ge \frac{d\lambda(\mathbf{F})}{d\mathbf{F}}\Big|_{\mathbf{F} = \mathbf{F}_1} : (\mathbf{F}_2 - \mathbf{F}_1)
$$
(4.20)

$$
\bar{\lambda}' = \frac{\alpha}{\alpha + \beta} F(\lambda^{-1}, \alpha + 1, \beta, \text{upper}) + \frac{d}{d\lambda} \int_0^{\frac{1}{\lambda}} \frac{p^{\alpha - 1} (1 - p)^{\beta - 1}}{B(\alpha, \beta)} dp + \lambda \frac{d}{d\lambda} \int_{\frac{1}{\lambda}}^1 \frac{p^{\alpha} (1 - p)^{\beta - 1}}{B(\alpha, \beta)} dp
$$
\n(4.10)

Combining these two inequalities and noticing $\frac{d\bar{\lambda}}{d\lambda} \ge 0$, we conclude

$$
\bar{\lambda}(\mathbf{F}_2) - \bar{\lambda}(\mathbf{F}_1) \ge \frac{d\bar{\lambda}}{d\lambda} \frac{d\lambda(\mathbf{F})}{d\mathbf{F}}\Big|_{\mathbf{F} = \mathbf{F}_1} :
$$
\n
$$
(\mathbf{F}_2 - \mathbf{F}_1) = \frac{d\bar{\lambda}(\mathbf{F})}{d\mathbf{F}}\Big|_{\mathbf{F} = \mathbf{F}_1} : (\mathbf{F}_2 - \mathbf{F}_1)
$$
\n(4.21)

which proves the convexity.

Appendix 2 Lebedev quadrature

Six-point quadrature

$$
\frac{1}{4\pi} \int_{S^2} f \, da = \frac{1}{6} \sum_{i=1}^6 f(\mathbf{a}_i^{(1)}), \quad \mathbf{a}_i^{(1)} = \{\pm 1, 0, 0\}, \{0 \pm 1, 0\}, \{0, 0, \pm 1\}
$$
\n(4.22)

14-point quadrature

$$
\frac{1}{4\pi} \int_{S^2} f \, da = \frac{1}{15} \sum_{i=1}^6 f(\mathbf{a}_i^{(1)}) + \frac{3}{40} \sum_{i=1}^8 f(\mathbf{a}_i^{(3)}) \tag{4.23}
$$

$$
\mathbf{a}_{i}^{(3)} = \{ \pm \frac{\sqrt{3}}{3}, \pm \frac{\sqrt{3}}{3}, \pm \frac{\sqrt{3}}{3} \}, \{ \mp \frac{\sqrt{3}}{3}, \pm \frac{\sqrt{3}}{3}, \pm \frac{\sqrt{3}}{3} \},
$$

$$
\{ \pm \frac{\sqrt{3}}{3}, \mp \frac{\sqrt{3}}{3}, \pm \frac{\sqrt{3}}{3} \}, \{ \pm \frac{\sqrt{3}}{3}, \pm \frac{\sqrt{3}}{3}, \mp \frac{\sqrt{3}}{3} \}
$$
(4.24)

Appendix 3 Response function of the GOH model with recruitment

The derivative of S_f in ([3.17\)](#page-6-5) to $\frac{C}{2}$ yields the material tangent tensor

$$
\mathbb{D}_{f} = \sum_{i=1}^{2} \left[2\mu_{2} \frac{d\left(e^{\gamma(\bar{l}_{k}-1)^{2}}(\bar{l}_{k}-1)\right)}{d\bar{l}_{k}} \mathbf{H}_{i} \otimes \mathbf{H}_{i} + 2\mu_{2} e^{\gamma(\bar{l}_{k}-1)^{2}} (\bar{l}_{k}-1) \left(\kappa \mathbb{H} + (1-2\kappa) \frac{d^{2} \bar{J}_{i}}{dJ_{i}^{2}} \mathbf{M}_{i} \otimes \mathbf{M}_{i} \otimes \mathbf{M}_{i} \otimes \mathbf{M}_{i}\right) \right]
$$
\n(4.25)

where

$$
\mathbf{S}_{f} = \sum_{i=1}^{2} \mu_{2} e^{\gamma(\bar{l}_{ki}-1)^{2}} (\bar{l}_{ki}-1) \left(\kappa_{i} \mathbf{\bar{I}} + (1-3\kappa_{i}) \frac{d\bar{J}_{i}}{dJ_{i}} \mathbf{M}_{i} \otimes \mathbf{M}_{i} \right)
$$
(4.27)

where $\bar{\mathbf{I}} := \frac{d\bar{I}_1}{d\mathbf{C}} = \sum_{I=1}^3 \frac{f(\lambda_I)f'(\lambda_I)}{\lambda_I}$ $\frac{\partial f''(\lambda_I)}{\lambda_I} \mathbf{N}_I \otimes \mathbf{N}_I$, and $\{\mathbf{N}_I\}_{I=1}^3$ are the principal vectors. The material tangent is

$$
\mathbb{D}_{f} = \sum_{i=1}^{2} \left[2\mu_{2} \frac{d\left(e^{\gamma(\bar{l}_{k}-1)^{2}}(\bar{l}_{k}-1)\right)}{d\bar{l}_{k}} \mathbf{H}_{i} \otimes \mathbf{H}_{i} + 2\mu_{2} e^{\gamma(\bar{l}_{k}-1)^{2}} (\bar{l}_{k}-1) \left(\kappa \frac{d\bar{\mathbf{I}}}{d\mathbf{C}} + (1-3\kappa) \frac{d^{2} \bar{J}_{i}}{dJ_{i}^{2}} \mathbf{M}_{i} \otimes \mathbf{M}_{i} \otimes \mathbf{M}_{i} \otimes \mathbf{M}_{i}\right) \right]
$$
\n(4.28)

where $\mathbf{H}_i = \kappa \mathbf{\overline{I}} + (1 - 3\kappa) \frac{dJ_i}{dJ_i} \mathbf{M}_i \otimes \mathbf{M}_i$. The fourth-order tensor $\frac{d\mathbf{I}}{d\mathbf{C}}$ can be obtained by a straightforward computation.

$$
\frac{d\overline{\mathbf{I}}}{d\mathbf{C}} = \sum_{I=1}^{3} \frac{1}{2\lambda_{I}} \frac{d}{d\lambda_{I}} \left(\frac{f(\overline{\lambda}_{I})f'(\overline{\lambda}_{I})}{\lambda_{I}} \right) \mathbf{N}_{I} \otimes \mathbf{N}_{I} \otimes \mathbf{N}_{I} \otimes \mathbf{N}_{I} + \sum_{I=1}^{3} \sum_{J>I} \frac{1}{2(\lambda_{I}^{2} - \lambda_{J}^{2})} \left(\frac{f(\lambda_{I})f'(\lambda_{I})}{\lambda_{I}} - \frac{f(\lambda_{J})f'(\lambda_{J})}{\lambda_{J}} \right) \Psi_{IJ} \otimes \Psi_{IJ}
$$
\n(4.29)

where $\Psi_{IJ} = \mathbf{N}_I \otimes \mathbf{N}_J + \mathbf{N}_J \otimes \mathbf{N}_I$. In the limiting case of $\lambda_I \rightarrow \lambda_I$:

$$
\frac{1}{\lambda_I^2 - \lambda_I^2} \left(\frac{\bar{\lambda}_I}{\lambda_I} \frac{d\bar{\lambda}_I}{d\lambda_I} - \frac{\bar{\lambda}_J}{\lambda_J} \frac{d\bar{\lambda}_J}{d\lambda_J} \right) = \frac{1}{2\lambda_I} \frac{d}{d\lambda_I} \left(\frac{f(\bar{\lambda}_I) f'(\bar{\lambda}_I)}{\lambda_I} \right)
$$
(4.30)

Volumetric-deviatoric splitting. In fnite element implementation, a multiplicative volumetric-deviatoric splitting is applied to the deformation gradient. The isochoric factor $\tilde{\mathbf{F}} = \mathbf{J}^{-\frac{1}{3}} \mathbf{F}$ is passed to the strain energy function. A volumetric term $U(J)$ is added to energy function, giving

$$
W(\mathbf{F}) = \frac{\mu_1}{2} (\text{tr}\tilde{\mathbf{C}} - 3) + W_f(\tilde{\mathbf{C}}) + U(J), \quad \tilde{\mathbf{C}} = J^{-\frac{2}{3}} \mathbf{C} \quad (4.31)
$$

Let $\tilde{\sigma}$ and the $\tilde{\sigma}$ be the *algorithm* Cauchy stress and material tangent tensor arising from $W_f + W_g$, namely the response evaluating at $\mathbf{F} = \mathbf{\tilde{F}}$. The actual Cauchy stress and spatial material tensor are obtained through deviatoric projections (Simo et al. [1985](#page-17-33)) of the algorithm response, adding the contributions form *U*(*J*):

$$
\mathbf{H}_{i} = \kappa \left(\sum_{I=1}^{2} \frac{d(\bar{\lambda}^{2}(\mathbf{E}_{I}'))}{d(\lambda^{2}(\mathbf{E}_{I}'))} \mathbf{E}_{I}' \otimes \mathbf{E}_{I}' \right) + (1 - 2\kappa) \frac{d\bar{J}_{i}}{dJ_{i}} \mathbf{M}_{i} \otimes \mathbf{M}_{i}, \ \mathbb{H} = \sum_{I=1}^{2} \frac{d^{2}(\bar{\lambda}^{2}(\mathbf{E}_{I}'))}{d(\lambda^{2}(\mathbf{E}_{I}'))^{2}} \mathbf{E}_{I}' \otimes \mathbf{E}_{I}' \otimes \mathbf{E}_{I}' \otimes \mathbf{E}_{I}' \tag{4.26}
$$

Response function from the alternative approximation of \overline{I}_{κ} . The stress function arising from the representation (3.18) (3.18) reads

$$
\sigma = \tilde{\sigma}_{dev} + U'(J)\mathbf{1}, \quad \text{where } \tilde{\sigma}_{dev} = \mathbb{I}_{dev}\tilde{\sigma}
$$
\n
$$
\mathbb{C} = \mathbb{I}_{dev}\tilde{\mathbb{C}}\mathbb{I}_{dev} - \frac{2}{3}\tilde{\sigma}_{dev}\otimes \mathbf{1} - \frac{2}{3}\mathbf{1}\otimes \tilde{\sigma}_{dev} + \frac{2\text{tr}\tilde{\sigma}}{3}\mathbb{I}_{dev} - 2U'(J)\mathbb{I} + (JU'(J))'\mathbf{1}\otimes \mathbf{1}
$$
\n(4.32)

where $\mathbb I$ and 1 are the fourth- and second-order identity tensors, $\mathbb{I}_{\text{dev}} = \mathbb{I} - \frac{1}{3} \mathbf{1} \otimes \mathbf{1}$ is the deviatoric projector.

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