



Zn-based metal–organic framework with intramolecular hydrogen bond for the electroreduction of CO₂ to formate

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Abstract

Metal–organic frameworks (MOFs) have important research value in the field of electrochemical CO₂ reduction reaction because of their rational design. Here, a new MOF-CH₃ was prepared via a simple solvothermal method by using Zn as the metal center and 1,2,4-triazole and 2-methyl-terephthalic acid as ligands for electrocatalytic CO₂ reduction. The single-crystal X-ray diffraction shows that MOF-CH₃ is N, O-coordinated 3D columnar layer framework with intramolecular hydrogen-bonding interactions. The powder X-ray diffraction for MOF-CH₃ displays the good crystallinity of 24 h in 0.5 mol L⁻¹ KHCO₃ electrolyte solution. The electrochemical CO₂ reduction reaction tests indicate that the MOF could effectively convert CO₂ to formate, and the highest Faradaic efficiency of formate (FE_{formate}) is 76.5% at -1.37 V (vs. reversible hydrogen electrode) with a partial current density of formate of -12.1 mA cm⁻². The performance of MOF-CH₃ is better than that of the reported other two structural analogues MOF-NH₂ with FE_{formate} of 55.7% at -1.57 V or MOF-H with FE_{formate} of 73.5% at -1.37 V in aqueous CO₂-saturated electrolyte solution. The work shows that the performance could be improved by regulating the microenvironment of MOF catalysts.

Keywords Metal – organic frameworks · Electrochemical reduction of CO₂ · Formate · Intramolecular hydrogen bond

Introduction

With the rapid development of the global economy, the consumption of diverse fossil energy sources has been increasing day by day [1–3]. CO₂, as the final product of the burning of fossil fuels, its excessive emission has caused the global warming and other environmental issues [4, 5]. To address the issues, it is an inevitable trend of reducing the emissions of CO₂. Electrochemical CO₂ reduction, which could effectively convert CO₂ to valuable petrochemicals, has been considered as one of the more attractive and promising strategies due to its simple process and mild reaction conditions [6–8]. However, it is still a challenge to achieve a high-efficiency CO₂ reduction with a suitable electrocatalyst which contains high activity and good selectivity, due to the large thermodynamic energy barrier and difficulty

of activating stable CO₂ molecules as well as competitive hydrogen evolution reaction [9, 10].

In recent years, various catalysts, such as nanomaterials [11–13], metal [14, 15], transition-metal oxide [16, 17], metal-free materials carbon [18, 19], and metal–organic framework (MOFs) [20, 21], have been explored for electrocatalytic CO₂ conversion [22–24]. Compared with other electrocatalysts, MOFs, as a highly porous material with a periodic network structure formed by the self-assembly of organic ligands and metal ions or clusters, exhibit a hopeful electrocatalyst for CO₂ reduction due to their high surface areas, single active sites, and potential to capture CO₂ [25–27]. However, they often suffer from poor chemical stability in the electrolyte solution and are sensitive to water, severely hindering their industrial use. Recently, balanced hydrogen-bonding interactions between the cationic frameworks and anions have been proposed to enhance the stability of MOFs [28]. And intramolecular hydrogen-bonding interactions (IHBI) play fundamentally important roles in the fabrication of stable MOFs. To date, improving MOF stability for electrochemical CO₂ reduction through IHBI has rarely been considered. Because of Zn-based catalysts with high cost-effectiveness and low-activity hydrogen

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evolution reaction [29], Zn-MOF-based electrocatalysts of CO₂ reduction have attracted great interest [30–32]. Presently, most reported Zn-MOFs show high selectivity and activity to CO or CH₄ [31, 33, 34], only a few Zn-MOFs [35] exhibit high selectivity of formate which is economically feasible in view of the market value and the energy input. Investigating its reason, the coordination environment of the metal as an active site greatly affects the product selectivity of CO₂ reduction. Therefore, it is significant to design and synthesize new Zn-MOFs with IHBI for electrocatalytic conversion of CO₂ to formate.

In this work, a new three-dimensional N, O-co-coordinated MOF-CH₃ was obtained using Zn as the metal center and 1,2,4-triazole and 2-methyl terephthalic acid as organic ligands by a simple solvothermal method. The MOF-CH₃ is isostructural with the reported FJU-40-NH₂ (denoted by MOF-NH₂ here) containing IHBI and FJU-40-H (denoted by MOF-H here) without IHBI [28]. MOF-CH₃ and MOF-NH₂ show far better chemical stability than MOF-H in an electrolyte solution. The electrochemical CO₂ reduction test displays that all the N,O-co-coordinated Zn-based MOFs exhibit specific selectivity for conversion CO₂ to formate with the faradaic efficiency (FE) more than 55% at the optimal potential, which is consistent with the reported N,O-coordinated Zn-based MOFs [35], but different from the reported N- or O-coordinated Zn-based MOFs catalysts with the primary product of CO or CH₄ [31, 33, 34]. The above results indicate that the microenvironment of the catalyst could affect the electroreduction of CO₂. The FE_{formate} of 76.5% for MOF-CH₃ is lower than that for MOF materials such as Zn-MOF [35], Bi-BTC-D-3.75 [36], and MIL-68(In)-NH₂ [37], but higher than that for (Me₂NH₂⁺)[InIII-(TTFTB)]·0.7C₂H₅OH·DMF [38], Cu-MOF/GO [39], and MFM-300(In)-t/CP [40]. Although there are relatively poor CO₂ reaction activities for MOF-CH₃ in comparison with the reported advanced catalysts, this work provided a novel Zn-based MOF catalyst with IHBI for electrochemical conversion of CO₂ to formate.

Experiment

Materials

All reagents and chemicals were obtained commercially and used without further purification. 1,2,4-triazole, terephthalic acid, 2-amino-terephthalic acid, 2-methyl-terephthalic acid, zinc nitrate hexahydrate, and *N,N*-dimethylformamide (DMF) were purchased from Sinopharm Chemical Reagent Co., Ltd. Nafion 117 proton exchange membrane and NafionD-521 dispersion (5 wt% in lower aliphatic alcohols and water) were purchased from Shanghai Hesun Electric Co., Ltd.

Instrumentation

Powder X-ray diffraction (PXRD) was carried out with a PANalytical X'Pert³ diffractometer equipped with a Cu-sealed tube ($\lambda = 1.541874 \text{ \AA}$) at 40 kV and 40 mA over the 2θ range of 5–30°. The simulated PXRD pattern was produced by using the Mercury V1.4 program and single-crystal diffraction data. Thermal analysis was carried out on a METTLER TGA/SDTA 851 thermal analyzer from 30 to 600 °C at a heating rate of 10 °C min⁻¹ under N₂ flow. Fourier transform infrared spectra (FT-IR, KBr pellet) were measured in the range of 400–4000 cm⁻¹ by a Nicolet 5700 FT-IR. Elemental (C, H, N) analytical data were obtained on a Perkin-Elmer model 240C elemental analyzer. Electrochemical test data were recorded with a Versa STAT 3F electrochemical workstation (Princeton Instruments, USA). All the gas products (CO and H₂) were quantified with a gas chromatography (PANNA-A60), which was equipped with a thermal conductivity detector and flame ionization detector, and N₂ was used as carrier gas. Liquid products were analyzed by quantitative ¹H-nuclear magnetic resonance (NMR, Bruker AVNANCE-400) using deuterium oxide as an internal standard.

Synthesis of MOF-R (R = CH₃, H₂, H) for the electroreduction of CO₂

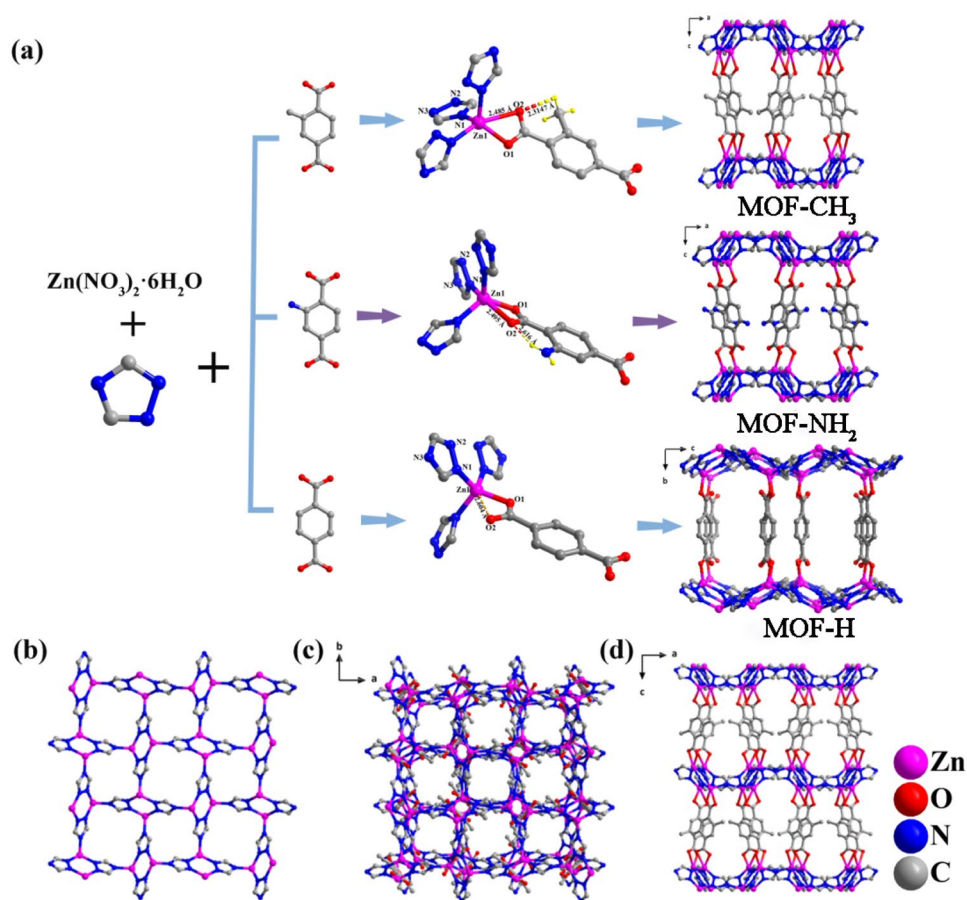
MOF-R (R = CH₃, H₂, H) were prepared via solvothermal reaction. The detailed information was shown in Supplementary Information, including synthesis, SCXRD analysis, electrochemical measurements, and product analysis.

Results and discussion

Structures and characterization of MOF-R (R = CH₃, NH₂, H)

As illustrated in Fig. 1a, a series of Zn-based MOFs (MOF-R, R = CH₃, NH₂, H) were prepared by a facile one-step solvothermal procedure with zinc nitrate and organic ligands. The images of scanning electron microscopy (SEM) show that all the MOFs are relatively regular block structures (Fig. S1), wherein MOF-CH₃ is a new MOF material. SCXRD patterns of the obtained MOF-CH₃ could be ascribed to a structural type with a tetragonal P4/nnc space group which is consistent with MOF-NH₂ (Table S1). Each Zn^{II} atom with tetrahedron geometry (Table S2 and S3) is coordinated with three N atoms from three 1,2,4-triazole ligands molecules and two O atoms from a 2-methyl-terephthalic acid ligand (Fig. 1a). And

Fig. 1 **a** Synthesis and structure illustration of MOF-R (R = CH₃, NH₂, H), **b** [Zn₂(1,2,4-triazole)₂] units connect to each other to form a two-dimensional layer of MOF-CH₃, and **c, d** three-dimensional open framework pillared by 2-methyl-terephthalic acid of MOF-CH₃



the two Zn^{II} atoms are joined by two 1,2,4-triazole ligands to form a [Zn₂(1,2,4-triazole)₂] unit (Fig. 1b), which connect to each other to extend further into a waved two-dimensional layer. These layers are further supported by 2-methyl-terephthalic acid ligands via Zn–O coordinated bonds to form a three-dimensional columnar layer framework (Fig. 1c, d). These Zn-MOFs have almost the same physical structure as MOF-NH₂ and MOF-H, which are all columnar structures constructed with [Zn₂(1,2,4-triazole)₂]_n as layers and 2-methyl-terephthalic acid as column (Fig. 1a). Nevertheless, their structures are slightly different due to the differences in the functional groups of terephthalic acid-R linker (Fig. 1a). For MOF-CH₃ or MOF-NH₂, the IHBI with d[O(–COO)⋯H(–C H₃)] = 2.3147 Å or d[O(–COO)⋯H(–NH₂)] = 2.616 Å are formed between hydrogen atoms of –CH₃ or NH₂ and carboxyl oxygen atoms (Fig. 1a), which could shorten the distance of Zn1–O2. Specifically, the Zn1–O2 bond lengths in MOF-CH₃, MOF-NH₂, and MOF-H are 2.485 Å, 2.495 Å, and 2.604 Å, respectively (Table S2). A shorter Zn1–O2 bond length could make MOF-CH₃ and MOF-NH₂ more stable than MOF-H [28], which was confirmed by the subsequent stability test in KHCO₃ electrolyte solutions via powder X-ray diffraction (PXRD).

The PXRD data for the as-prepared MOF-CH₃ or MOF-NH₂ materials (Fig. 2a, b) has typical diffraction peaks at the 2θ of 6.387°, 9.128°, 13.120°, and 15.17°, which match well with the simulated PXRD pattern of the main crystal faces (002), (102), (200), and (211), respectively. MOF-H also shows the crystal faces (020), (011), (101), and (111) at the typical diffraction peaks of 2θ = 6.611°, 9.735°, 12.763°, and 13.117°, respectively (Fig. 2c). These results indicate the successful synthesis of MOF-R. The thermal stability of MOF-R was measured by thermogravimetric (TG) analysis. TG curve results display that MOF-R has pretty similar weightlessness tracks up before 600 °C (Fig. 2d). For MOF-CH₃, the first weight loss at 123 °C is the loss of DMF molecules. Then, the plateau was maintained until the framework began to decompose at around 369 °C. The MOF-NH₂ and MOF-H lose the first weightlessness before 100 °C, which could be ascribed to the loss of water molecules in the pore channel. For MOF-NH₂, the second weight loss at 123 °C is the loss of DMF molecules. The last weight loss over 322 °C corresponds to the decomposition of the framework. For MOF-H, the molecules of DMF evaporate at ~112 °C, and when heated above 345 °C, the organic ligands start the decomposition, and the framework collapse. These results suggest that the MOFs possess excellent

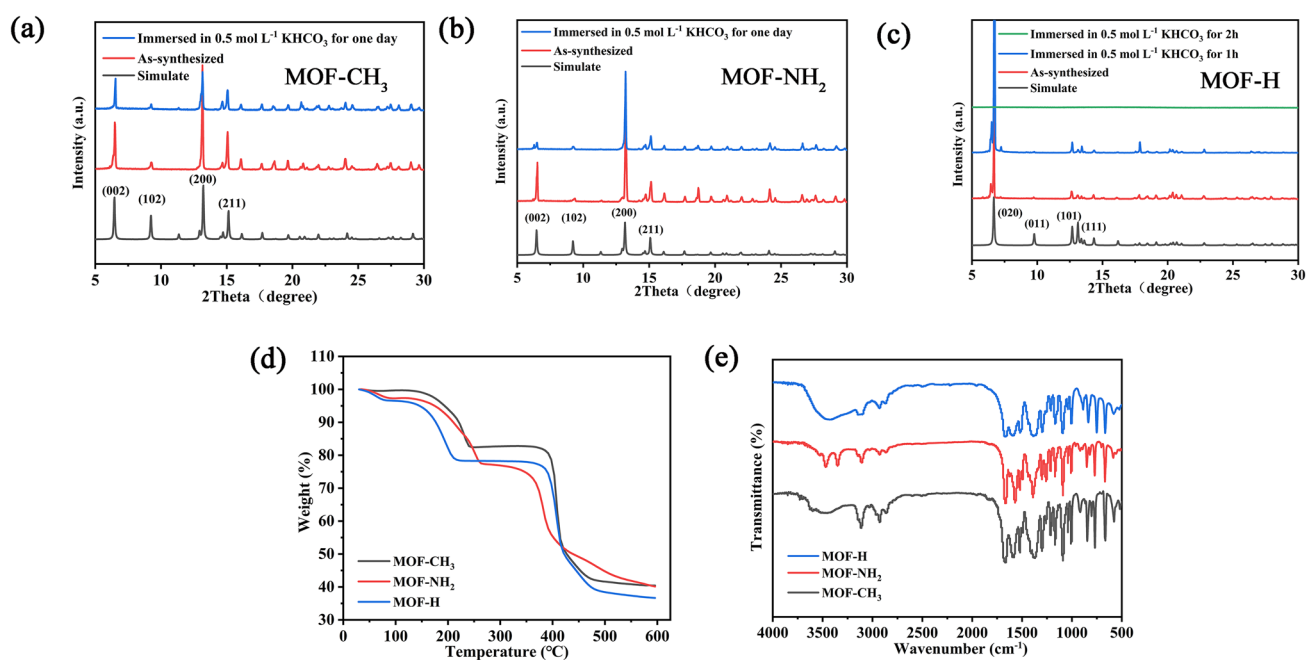


Fig. 2 a–c Powder X-ray diffraction patterns, **d** thermogravimetric curves, and **e** Fourier transform infrared spectroscopy spectra of MOF-R (R=CH₃, NH₂, H)

thermal stability. The characteristic temperatures and final weight losses of all samples are also stated in Table S4. The Fourier transform infrared spectroscopy (FT-IR) spectrums of MOF-R are displayed in Fig. 2e. All the MOFs show C–N peak at about 1130 cm⁻¹ for 1,2,4-triazole ligand and the characteristic peaks of C=O/C–O, respectively, at about 1720/1100 cm⁻¹ for terephthalic acid ligand. For MOF-CH₃, the characteristic peaks of –CH₃ group are at 2940 cm⁻¹, 2860 cm⁻¹, and 1380 cm⁻¹. There are two peaks at 3470 cm⁻¹ and 3340 cm⁻¹ with moderate intensity, which correspond to N–H antisymmetric stretching vibration and symmetric stretching vibration on –NH₂ functional group of MOF-NH₂, respectively. In addition, we found that MOF-CH₃ and MOF-NH₂ have blunt peaks of O–H stretching vibration intramolecular hydrogen bonds at 3610 cm⁻¹ and 3530 cm⁻¹[28], while MOF-H does not, which is consistent with the crystal structure analysis described in Fig. 1a.

Electrochemical measurements and CO₂RR performance

The stability of the three MOFs immersed in 0.5 mol mL⁻¹ KHCO₃ electrolyte solution was analyzed, as shown in Fig. 2a–c. The results show that MOF-CH₃ and MOF-NH₂ in the electrolyte solution could maintain good crystallinity for 24 h, while there are basically no diffraction peaks for MOF-H after soaking for 2 h. The stability differences of the MOFs in the electrolyte solutions should be closely related to the presence of IHBIs in their structures. For

MOF-CH₃ or MOF-NH₂, the distance between Zn1-O2 is shortened due to the presence of IHBIs. A shorter Zn1-O2 distance could protect Zn atoms from water attack, resulting in better stability of MOF-CH₃ or MOF-NH₂ than MOF-H in electrolyte solutions.

Electrochemical CO₂ reduction tests were conducted in 0.5 mol mL⁻¹ KHCO₃ electrolyte solution (pH = 7.2) via a liquid H-type electrochemical cell with successive CO₂ bubbling at a flow rate of 20 mL min⁻¹ [41]. The gaseous and liquid reduction products were respectively detected and quantified by gas chromatography and ¹H-nuclear magnetic resonance (¹H-NMR), and details were provided in supporting information. The cyclic voltammograms curves for MOF-CH₃ and MOF-H (Fig. 3a, b) with the redox current peaks of Zn(II) reduced to Zn(I) [31, 35, 42] show that the onset potentials of MOF-R under CO₂ atmosphere are more positive than that in Ar condition, which indicates their activities are originated from the CO₂ conversion [43, 44]. Although MOF-NH₂ has similar CV curves in Ar and CO₂ condition, the current density values in the CO₂ condition is higher than that in the Ar condition when the potential values surpass 1 V (Fig. 3c), indicating CO₂ reduction reaction still is the superior reaction. Moreover, MOF-CH₃ exhibits slightly larger current densities at high potential than MOF-NH₂ or MOF-H from linear sweep voltammetry curves (Fig. S2). In order to study CO₂ reduction activity and selectivity, chronoamperometry tests were conducted at different potentials (Fig. S3). Formate, CO, and H₂ are the products of CO₂ reduction over the MOF-R materials at each

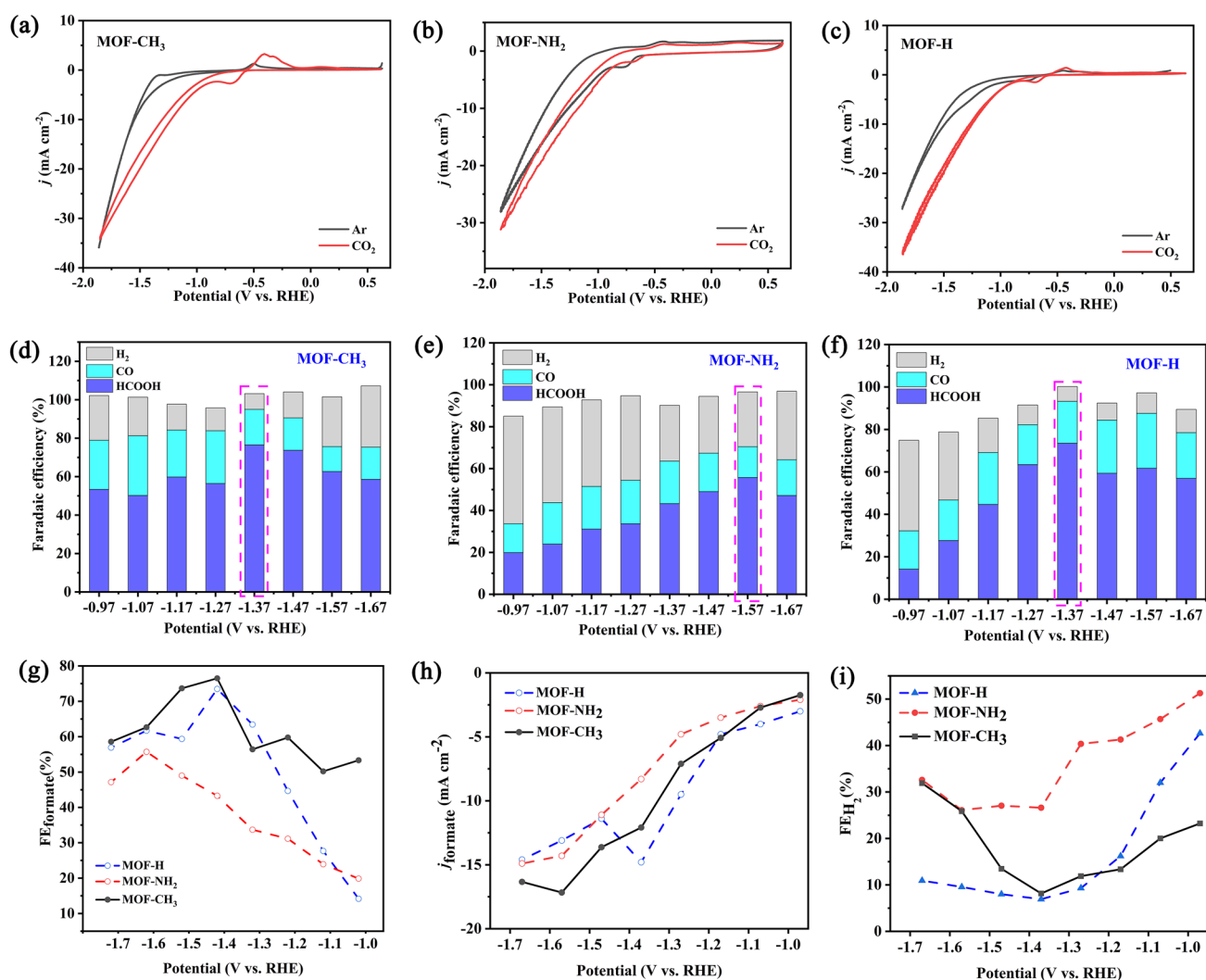


Fig. 3 a–c Cyclic voltammograms curves in Ar- (black line) or CO₂- (red line) saturated 0.5 mol L⁻¹ KHCO₃ solutions with a scan rate of 50 mV s⁻¹, d–f faradaic efficiency of different products, g

faradaic efficiency of formate, h partial current densities of formate, and i faradaic efficiency of H₂ for MOF-R (R = CH₃, NH₂, H) at given potentials

potential from -0.97 to -1.67 V vs. RHE (reversible hydrogen electrode, the same below) (Fig. 3d–f), and formate is the primary product at the most given potential, especially MOF-CH₃ (Fig. 3d). The detailed data in Fig. 3d–f has also been shown in Table S5. In addition, the gas chromatograms of the gaseous products formed at the optimum potential were provided in Fig. S4 for explicitness, and ¹H-NMR of the liquid phase products was shown in Fig. S5a. To confirm the formate from the electroreduction of CO₂, isotopic labeling tests were further performed by using the mixture of ¹³CO₂/¹²CO₂ as the feedstock. The ¹H-NMR signals of H¹³COOH (Fig. S5b) illustrate that the formate originated from the CO₂ reduction.

As shown in Fig. 3g, h, the highest FE_{formate} for MOF-CH₃ and MOF-H is, respectively, 76.5% and 73.5% at -1.37 V, with the partial current density of formate (j_{formate})

of -12.1 mA cm⁻² and -14.0 mA cm⁻², respectively. The maximum FE_{formate} for MOF-NH₂ is 55.7% at -1.57 V with j_{formate} of -14.5 mA cm⁻². As shown in Fig. 3g, the MOF-CH₃ shows good selectivity of formate with FE_{formate} of more than 50% at all applied potentials, while poorer formate selectivity of MOF-NH₂ was obtained than that of MOF-CH₃ or MOF-H. Also, the FE_{H₂} for MOF-NH₂ is up to 40–50% in the low potential range of -0.97 to -1.27 V, and MOF-H shows high FE_{H₂} at -0.97 V (Fig. 3i). The formate concentration of MOF-CH₃ is higher than MOF-NH₂ or MOF-H in a wide potential ranging from -0.97 to -1.67 V except -1.27 V (Fig. S6). The concentration of formate for MOF-CH₃ is 11.1 mmol L⁻¹ at -1.37 V, while that for MOF-NH₂ and MOF-H is 7.0 mmol L⁻¹ and 9.6 mmol L⁻¹, respectively. Taken overall, MOF-CH₃ is more efficient electrocatalysts for CO₂ reduction than MOF-NH₂ or MOF-H.

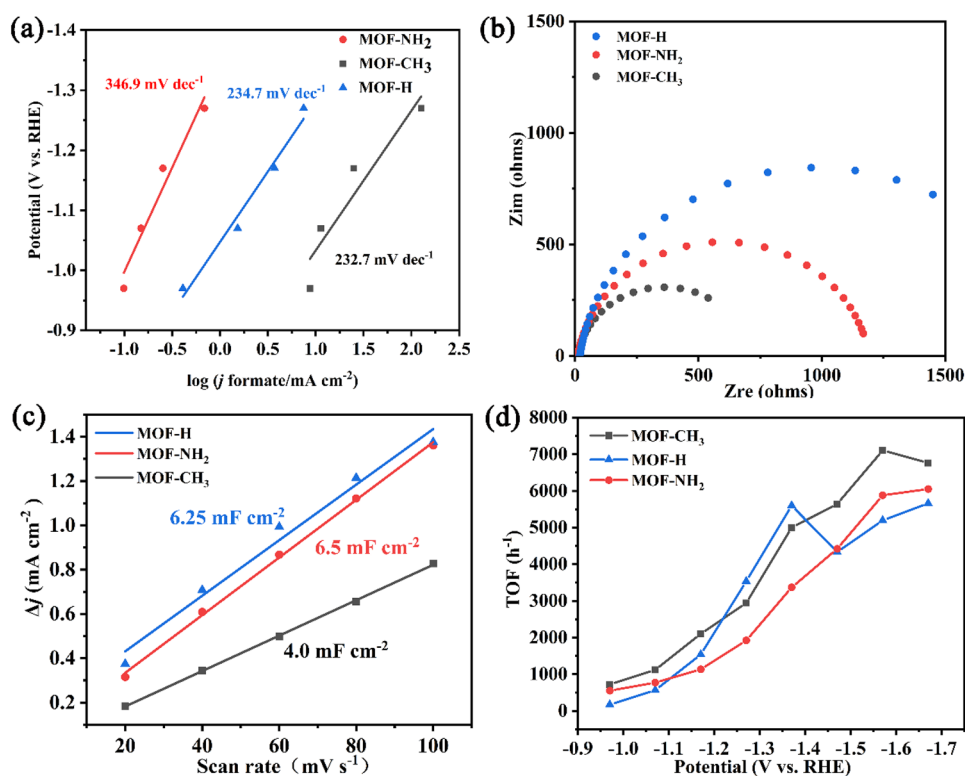
A comparison of Zn-based MOFs and complex electrocatalysts for CO₂ reduction was summarized in Table S6. These N,O-co-coordinated Zn-based MOFs exhibit specific selectivity for formate at the optimum potential ($FE > 55\%$), different from the reported N- or O-coordinated Zn-based MOF catalysts with the primary product of CO or CH₄ [31, 33, 34]. Although there is a relatively poor CO₂ reduction selectivity for MOF-CH₃ in comparison with the reported N, O-coordinated Zn-based MOFs [35], this work provided a novel example of Zn-MOF catalysts with IHBI for electrochemical conversion of CO₂ to formate.

To further probe the catalytic activity of Zn-MOFs during CO₂ reduction, the Tafel plots, electrochemical impedance spectroscopy (EIS), the electrochemical active surface area (ECSA), and turnover frequency (TOF) [45] were studied. The reaction kinetics of the CO₂ reduction process was explored by the Tafel analysis. The low Tafel slope means fast dynamics and better electrocatalytic performance. As illustrated in Fig. 4a, MOF-CH₃ displayed a Tafel slope of 232.7 mV dec⁻¹, slightly smaller than MOF-NH₂ (346.9 mV dec⁻¹) and MOF-H (234.7 mV dec⁻¹). This indicates that MOF-CH₃ has a relatively faster kinetics and excellent electrocatalytic performance for CO₂ reduction. In addition, the electron transfer behavior of MOF-R during CO₂ reduction was studied via EIS measurement, which is displayed in Fig. 4b. MOF-CH₃ has a lower semicircular radius (R) of $\sim 700 \Omega$ than MOF-NH₂ or MOF-H at -0.37 V. The smaller R reflects the much favorable charge transfer

kinetics, which also reveals that MOF-CH₃ has better electrochemical capabilities [46, 47]. Additionally, the electrochemical double-layer capacitance (C_{dl}) measured by CV at different scan rates (Fig. S7) was used to evaluate the ECSA of the Zn-MOF electrocatalysts. A higher C_{dl} generally means a higher ECSA which illustrates more exposed active sites [48]. The result showed that C_{dl} value for MOF-CH₃, MOF-NH₂, and MOF-H is 4.0, 6.5, and 6.25 mF cm⁻², respectively (Fig. 4c). Although MOF-NH₂ or MOF-H has an obviously higher ECSA than MOF-CH₃, the two electrocatalysts do not exhibit better conversion performance of CO₂ to formate, as could be explained by more reaction active sites given to HER or the conversion of CO₂ to CO at most applied potentials, based on the FE values of the products. Besides, TOF was used to evaluate the intrinsic activity of electrocatalysts. These TOF results for MOF-CH₃ are better than that for MOF-NH₂ or MOF-H at most given potentials (Fig. 4d), which could verify that MOF-CH₃ has higher intrinsic activities.

Although the metal center for these MOFs is the same and their structure is similar, these MOFs show different effects of CO₂ reduction. Apart from the presence or absence of intramolecular hydrogen bonds which mainly affect the stability, they also possess different functional groups ($-\text{CH}_3$, $-\text{NH}_2$, $-\text{H}$) with regard to microenvironment. Thus, we further explore the effect of functional groups. And contact angle tests were performed to analyze the hydrophilicity of the Zn-MOFs. As shown in Fig. S8, the contact

Fig. 4 **a** Tafel plots, **b** electrochemical impedance spectroscopy spectra at -0.37 V, **c** charging current density at different scan rates, and **d** turnover frequency plots for the generation of formate for MOF-R ($R = \text{CH}_3, \text{NH}_2, \text{H}$)



angle of 44.2° for MOF-CH₃ is larger than that of the other two MOFs, indicating that the hydrophilicity of MOF-CH₃ is poorer than the others. The introduction of hydrophobic groups (–CH₃) could make MOFs show good chemical stability and avoid the attack from water to generate H₂ to some extent, which is helpful for CO₂ reduction [49]. FE_{H₂} of MOF-CH₃, that is lower than that of MOF-NH₂ at most given different potentials (Fig. 3i), also illustrated that MOF-CH₃ could effectively inhibit the hydrogen evolution reaction. To sum up, favorable charge transfer kinetics and intrinsic activity as well as hydrophobicity let MOF-CH₃ improve electrocatalytic CO₂ reduction in comparison with the other two structural analogues. It is clear that the performance could be improved by regulating the coordination microenvironment of MOF catalysts.

Conclusions

In summary, a series of N, O-coordinated Zn-based isostructuralism MOF catalysts were obtained by a simple solvothermal method for selective electroconversion of CO₂ to formate under ambient conditions. Among them, MOF-CH₃ is a new MOF with IHBI distance of 2.3147 Å from the SCXRD analyses. MOF-NH₂ also exists as an intramolecular hydrogen bond with the distance of 2.616 Å, while MOF-H does not. PXRD results indicate that the presence of IHBI in MOFs is helpful for keeping the good crystallinity in electrolyte solution. The related electrochemical tests for CO₂ reduction show that the catalytic performance of MOF-CH₃ with FE_{formate} 76.5% at –1.37 V is better than that of MOF-NH₂ with FE_{formate} of 55.7% at –1.57 V or MOF-H with FE_{formate} of 73.5% at –1.37 V. According to the analyses of EIS, TOF, and contact angle, it was found that MOF-CH₃ possesses the lowest *R* of ~700 Ω, high intrinsic actives, and the largest contact angle of 44.2° . The introduction of IHBI and hydrophobic groups (–CH₃) may be helpful for CO₂ reduction. This work provides a novel Zn-MOF catalyst for the electroreduction of CO₂ to formate.

Supplementary Information The online version contains supplementary material available at <https://doi.org/10.1007/s10008-023-05616-5>.

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