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The in situ grown of activated Fe-N-C nanofibers derived from polypyrrole on carbon paper and its electro-catalytic activity for oxygen reduction reaction

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Abstract

The Fe-N-C nanofibers (NFs) exhibit excellent electro-catalytic activity for oxygen reduction reaction (ORR), which is essential for energy conversion system. Compared with the traditional ink-based electrode, we synthesized the structured Fe-N-C ANFs/CP (activated NFs derived from polypyrrole(PPy) doped with iron atoms in situ grown on carbon paper) by electrochemical polymerization. Herein, we investigate the effect of the iron concentration on the electro-catalytic activity for ORR, revealing that the as-prepared 0.05-Fe-N-C ANFs exhibit superior electro-catalytic activity than 0.01-Fe-N-C ANFs, 0.03-Fe-N-C ANFs as well as 0.10-Fe-N-C ANFs and excellent durability in alkaline solution. It is shown that the better electro-catalytic activity of 0.05-Fe-N-C ANFs for ORR could be ascribing to the fluffy fibrous structure, higher Fe-N_x concentration, and well-dispersed catalytic active sites.

Keywords Polypyrrole · Activated nanofibers · Catalyst · Doping · Oxygen reduction reaction

Introduction

Fuel cells have attracted tremendous attentions as an ideal and clean energy-converting devices with their excellent performances, such as high power density, high-efficiency, environmental friendliness and low emission [1–3]. However, the widely commercialized applications of the fuel cells have always been hindered due to their prohibitive-cost Pt-based catalysts and poor durability [4–7]. Thus, researchers have devoted themselves to exploit novel catalysts, namely noble-metal catalysts, and non-noble metal catalysts. For noble catalysts, alloy catalysts [5, 8],

catalyst structure and catalyst support innovation [5, 7, 9] are important aspects to use less (or no) precious metal Pt and increase Pt utilization, resulting in lowering the cost of fuel cells. As for non-noble metal catalysts, metal-N-C [6, 10–12] have drawn tremendous attentions from many researchers, holding great promise to substitute precious Pt-based catalysts used in fuel cells.

In fact, incorporating heteroatoms, such as nitrogen, boron, sulfur into carbon lattice seems to be an effective approach to improve the physical and chemical properties of carbon materials [13–17]. Reports show that the doped nitrogen atoms can significantly change the electronic structure of the carbon-nitrogen materials due to the electronegativity difference, resulting into the charge transfer, oxygen molecule adsorption mode transforming and the electro-catalytic activity enhancing, compared to intrinsic carbon materials [18, 19]. Among various kinds of carbon nanomaterials, one-dimensional nanofibers (NFs) hold great promise to serve as catalyst support as well as catalysts in fuel cells, and electrode materials in supercapacitor due to their large specific area, high electro-catalytic activity and relatively feasible preparation process [20, 21]. Traditionally, one-dimensional nanofibers can be synthesized by chemical vapour deposition (CVD),

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electrospinning, template method, electrochemical method and so on, among which electrochemical deposition is an easy and facile method with low-cost, precisely controllable process and without supplementary catalyst precursor as well as removing the templates [22–25].

In recent years, polypyrrole (PPy), as a kind of conductive macromolecule with nitrogen and carbon atoms co-existing, has attracted lots of attentions ascribing to their good electrical conductivity, ion exchange capacity and easy preparation [26–28]. These excellent attributes make Fe-N-C NFs derived from PPy a very promising nanomaterial used in various application fields, such as catalysts, catalyst supports, sensor, supercapacitor, and so on [26, 29-32]. The transition metals doped within N-C NFs are reported to enhance the electro-catalytic activity of non-precious catalysts at the two following sections: on the one hand, the metal could combine with the other nitrogen atoms, forming Me-N₄, Me-N₂ or Me-N_x active sites [10, 12, 33]; on the other hand, some researchers proposed that the metal itself is not the composition of active sites, but can facilitate the formation of active sites [34, 35]. Based on these factors, Khomenko et al. reported the electro-catalytic activity of PPy for oxygen reduction reaction (ORR) and discussed the possible catalytic mechanism based on a quantum-chemical modeling [36]. And in recent years, the catalytic activity of N-C NFs doped with metallic elements (Fe, Co et al.) and non-metallic elements (such as N and S) have attracted tremendous attentions due to their excellent electrochemical performance, cost-effective and environment-friendly attributes [11, 12, 27, 28, 37, 38].

Fig. 1 The schematic images of Fe-N-C ANFs synthesis techniques

Traditionally, the catalysts primarily come in the form of powders, and the tedious painting process can inevitably result into catalyst aggregation and catalytic active sites dilution. In our previous paper, the structured CP/Fe-N-CNFs, which was used as electro-chemical catalyst for ORR, was in situ synthesized by CVD [10]. And the high purity pyridinic-N doped platelet CNFs were also synthesized by CVD and heat-treatment under NH₃ atmosphere [39]. In this paper, the N-C NFs derived from PPy were in situ synthesized on the carbon paper (CP) via electro-chemical polymerization, synthesizing the structured Fe-N-C ANFs/CP. And the effects of iron concentration on the electro-catalytic activity were also investigated.

Experimental section

Samples preparation

Synthesis of PPy NFs in situ grown on carbon paper

In this paper, the polypyrrole nanofibers (PPy NFs) in situ grown on carbon paper (CP) are synthesized according to the following specific procedures: chemical etching, and electrochemical polymerization. Firstly, the homemade CP [40, 41] (the preparation process of CP was reported in our previous paper) was used as PPy NFs in situ grown substrate, which was chemical-etched by 1 M HNO₃ solution, forming chemical-etched CP with



coarse surface; Secondly, 3.18 g Na₂CO₃ and 2.52 g NaHCO₃ were dissolved into distilled water to form homogenous solution, followed by being added 3.19 g LiClO₄ and 3.56 g pyrrole monomer, then ultrasonic processing for 1 h to ensure the solute evenly dispersing in the solvent, forming the electrochemical polymerization solution with 0.15 M pyrrole monomer, 0.1 M LiClO₄ and 0.2 M Na₂CO₃/NaHCO₃; thirdly, the pyrrole monomer was in situ electrochemical polymerization on CP using a potentiostat appliance with a double-electrode operating in the synthesized electrochemical polymerization solution. During the electrochemical polymerization process, the graphite electrode was employed as negative electrode and the etched CP was used as the positive electrode, the two electrodes were parallel and 2 cm apart. The electrochemical polymerization current was 0.1 A and time was set as 1, 2, 5, and 10 min, respectively.

At last, the synthesized PPy NFs were washed in deionized water to remove the residual dissociative anions.

Synthesis of activated N-C NFs derived from PPy on CP

The as-prepared PPy NFs were impregnated in 5 M KOH aqueous solution, followed by heat-treatment in tube furnace and the detailed steps are as follows: firstly, the temperature was heated up to 600 $^{\circ}$ C with a heating rate of 5 $^{\circ}$ C/min; secondly, the temperature was kept at 600 $^{\circ}$ C for 1 h to activate the nanofibers, producing the activated N-C NFs, and the as-synthesized activated N-C NFs were denoted as N-C ANFs.

Synthesis of N-C ANFs doped with iron on CP

Firstly, the homogenous $Fe(NO_3)_3$ aqueous solution was prepared with different concentration, 0.01 M, 0.03 M, 0.05 M, and 0.10 M; Secondly, the four pieces of CP with N-C

Fig. 2 The SEM images of **a** the home-made CP and electrochemical polymerized PPy NFs with different time, **b** 1 min, **c** 2 min, **d** 5 min, **e** and **f** 10 min

ANFs were impregnated in the prepared $Fe(NO_3)_3$ aqueous solution for 5 h, respectively; Thirdly, the impregnated samples were dried at 80 °C for 1 h and heat-treated as follows: the temperature was heated up to 500 °C with 3 °C/min to make $Fe(NO_3)_3$ decompose completely and then the temperature was elevated to 700 °C, in order to make ferric oxide be reduced under H₂ atmosphere.

At last, the synthesized N-C ANFs doped with iron were washed in 0.1 M HCl to remove the residual metal and then washed in deionized water till neutral. The as-prepared N-C ANFs doped with iron were abbreviated as Fe-N-C ANFs and Fe-N-C ANFs prepared with 0.01 M, 0.03 M, 0.05 M, 0.10 M Fe(NO₃)₃ aqueous solution were denoted as 0.01-Fe-N-C ANFs, 0.03-Fe-N-C ANFs, 0.05-Fe-N-C ANFs, and 0.10-Fe-N-C ANFs, respectively. The synthesis process of Fe-N-C ANFs is visually described in Fig. 1.

Characterization

Morphology and microstructure characterization

The morphology and element content analysis are represented by scanning electron microscope (SEM, JEOL JSM-7600F) and energy dispersive spectrometer (EDS). The microstructures are observed by transmission electron microscopy (TEM, JEM-2100F). The chemical states and element content are characterized by an X-ray photoelectron spectroscopy (XPS) apparatus (Thermo Scientific K-Alpha) equipped with a monochromatic Al K_{α} X-ray source of 1486.6 eV.

Electro-chemical characterization

The electrochemical activity of the samples are characterized by an electrochemical workstation (CHI 660e, Shanghai Chenhua Instrument Co. Ltd., China). The Cyclic



voltammetry (CV) and Linear Sweep Voltammetry (LSV) are carried out at room temperature (25 °C) in a conventional three-electrode electro-chemical cell with platinum electrode as counter electrode, Hg/HgO electrode as reference electrode and the prepared Fe-N-C ANFs as working electrode. The CV and LSV spectra are tested in O2 or N₂ saturated 0.1 M KOH aqueous solution. Rotating disk electrode (RDE) (AFMSRCE) voltammetry measurements and amperometric *i*-t curves were carried out at the same condition. Before the experiments, the glassy carbon (GC) electrode was carefully polished, then washed with distilled water for double times, ethanol solution once and dried in vacuum. The Fe-N-C ANFs were scraped gently from CP substrate, then 3.5 mg scraped Fe-N-C ANFs powders were mixed with 1.0 ml solution (the ratio of distill water with nafion is 9:1) under ultrasonic process. A certain amount of the Fe-N-C ANFs (15 μ l) is dropped onto the GC electrode. For comparison, a commercial catalyst of 3.5 mg/ml 20% Pt/C suspension is synthesized according to the same procedure as above.

Results and discussions

SEM images of PPy NFs with different electrochemical polymerization time (1, 2, 5, and 10 min, respectively) are exhibited in Fig. 2. Figure 2a shows that home-made CP [40–42] is composed of the carbon fibers with a diameter of 7 μ m and matrix carbon, which was fabricated via curing, pressing, carbonization, and graphitization process, so that the home-made CP possesses good electrical conductivity. In this paper, we chose the home-made CP as substrate for in situ electrochemical polymerizing PPy NFs. It can be seen that the nucleation of smallish PPy NFs have occurred (Fig. 2b) in the initial period of electrochemical polymerization and then the nucleus turn to grow up mainly along one-dimensional direction as time extends, as shown in Fig. 2c–e. The images show that the morphology differences of PPy NFs are not obvious except for the length, which can also confirm that the PPy NFs have only grown along with the axial direction. At last, the in situ electrochemical polymerized PPy NFs have covered the CP substrate almost completely and homogeneously with length of tens of microns, as shown in Fig. 2f.

Figure 3 shows the morphology and microstructure differences between PPy NFs before and after activation, namely, the PPy NFs and N-C ANFs (Fig. 3). It is shown that the specific surface area increases obviously from PPy NFs (68.52 m² g⁻¹) to N-C ANFs (82.16 m² g⁻¹). Meantime, the diameter decreases from PPy NFs (Fig. 3a and a inset) to N-C ANFs (Fig. 3b and b inset), which could also be confirmed by TEM images of PPy NFs and N-C ANFs (Fig. 3c, d). It can be seen some defects and



Fig. 3 SEM, TEM and HRTEM images of PPy NFs and N-C ANFs. SEM images of **a** before and **b** after KOH activation, the corresponding insets are the normal distribution of nanofiber diameter; TEM images of **c** before and **d** after KOH activation, and the corresponding HRTEM images (the red dashed rectangles in **c** and **d**) of **e** before and **f** after KOH activation, the **e** inset is the diffraction pattern of the amorphous carbon in PPy NFs

pores at the boundary sites of N-C ANFs ascribing to the KOH activation (red dashed circles in Fig. 3f) compared to boundary sites of PPy NFs (Fig. 3e), which is in an accordance with the the specific surface area results. Moreover, it is shown obviously that the PPy NFs and N-C ANFs both exhibit an amorphous type (Fig. 3e, and f), which could also be proved by the diffraction pattern (Fig. 3e inset).

SEM images of Fe-N-C ANFs pre-treated with different $Fe(NO_3)_3$ precursor concentration are shown in Fig. 4. Seen from SEM images, there are no significant morphology differences between 0.01-Fe-N-C ANFs and 0.03-Fe-N-C ANFs (Fig. 4a, b), in which only fibrous Fe-N-C ANFs could be seen on CP substrate without iron-containing particles. While, as $Fe(NO_3)_3$ precursor concentration increases, it is observed that few iron-containing nanoparticles could be observed in 0.05-Fe-N-C ANFs, as labeled by the red dashed circles in Fig. 4c. When

а

С



Fig. 4 SEM, TEM, and HRTEM images of the Fe-N-C ANFs. SEM images of a 0.01-Fe-N-C ANFs, b 0.03-Fe-N-C ANFs, c 0.05-Fe-N-C ANFs, and d 0.10-Fe-N-C ANFs; e TEM images of 0.05-Fe-N-C ANFs; f HRTEM images of the red dashed rectangle in e and the f inset is diffraction pattern of the iron-containing nanoparticle

 $Fe(NO_3)_3$ precursor concentration reaches up to 0.10 M, many iron-containing nanoparticles can be seen in 0.10-Fe-N-C ANFs (Fig. 4d red dashed circles). The reason could be speculated that the precursor component distributes nonuniformly on N-C ANFs when Fe(NO₃)₃ precursor concentration is high, as a result obtaining many visible iron-containing nanoparticles (Fig. 4d). TEM image shows that the iron-containing nanoparticles are combined well with the surrounding nanofiber (Fig. 4e), which could be also proved by the HRTEM image in Fig. 4f, namely, the iron-containing nanoparticle has bonding combining with the surrounding amorphous structure. Furthermore, the HRTEM image (Fig. 4f) and the diffraction pattern (Fig. 4f inset) of the iron-containing nanoparticles exhibit that it shows good crystallization characteristic of iron-containing nanoparticles, which could change the electron structure of the surrounding carbon and nitrogen atoms and then improve the electro-chemical activity of the N-C ANFs.

The electrochemical performances of 0.01-Fe-N-C ANFs, 0.03-Fe-N-C ANFs, 0.05-Fe-N-C ANFs, and 0.10-Fe-N-C ANFs are investigated in 0.1 M O₂/N₂-saturated KOH solution via cyclic voltammetry (CV), rotating disk electrode (RDE) voltammetry curves and amperometric *i-t* curves (Fig. 5). The obvious oxygen reduction peak could be observed in N-C ANFs (Fig. 5a) and Fe-N-C ANFs (Fig. 5b), indicating their electro-catalytic activity for oxygen reduction. As seen in Fig. 5b, Fe-N-C ANFs show significant positive shifts in onset-potential and peak potential, compared to those of N-C ANFs. Also, it is indicated that no obvious reduction peaks are observed in CV spectra measured in N₂-saturated 0.1 M KOH solution, appearing a tilted rectangular shape. This is due to the influence of the ohmic resistance from electrolyte motion in carbon pores on the mechanism of double-layer formation [43, 44]. Considering the effect of double-layer capacitance on the electrochemical performance of the electrocatalysts, the LSV spectra in Fig. 5c, d have been removed the capacitance effect. It shows that 0.05-Fe-N-C ANFs and 0.10-Fe-N-C ANFs reveal much higher onset-potential (0.89 V) than that of 0.01-Fe-N-C ANFs and 0.03-Fe-N-C ANFs (Fig. 5c). Fig. 5d shows the RDE curves of 0.05-Fe-N-C ANFs with different rotation speeds in 0.1 M O₂/N₂-saturated KOH solution and Fig. 5e shows four linear K-L plots near parallelism at different potentials, suggesting first-order reaction kinetics towards the concentration of dissolved oxygen on 0.05-Fe-N-C ANFs. The numbers of transferred electron can be calculated by the following K-L equation [13, 45]:

$$\frac{1}{J} = \frac{1}{B\omega^{0.5}} + \frac{1}{J_k}$$
(1)

$$B = 0.62nFC_o D_o^{2/3} v^{-1/6}$$
⁽²⁾

Where J is the observed current density, J_k denotes the kinetic-limit current density, B is the Levich slope, ω is the rotating rate; n is the number of electrons transferred per O₂, F = 96485 C mol⁻¹, $C_o = 1.2 \times 10^{-6}$ mol cm⁻³, $D_o = 1.9 \times 10^{-5}$ cm² s⁻¹, and v = 0.01 cm² s⁻¹. The numbers of transferred electrons of 0.05-Fe-N-C ANFs calculated via the slopes of the plots of J^{-1} versus $\omega^{-1/2}$ are 3.34, 3.33, 3.35, and 3.37 at 0.50, 0.55, 0.60, and 0.65 V, respectively. It reveals a mixed two electron and four electron transfer process and the steadier electron transfer numbers indicate a smoother and more electrochemically stable ORR process over 0.05-Fe-N-C ANFs. To some extent, introducing metal into carbon-nitrogen materials could enhance the electro-catalytic activity for ORR, which is ascribing to the generation of active sites $Fe-N_x$ with the introduction of iron element. Basically, if we observe n is larger than 3, which means that the material supports 4-electron pathway with slow final steps. As a result, n should be 4 when we calculate J_k using the Eq. 1, and

Fig. 5 The electro-chemical activity of the as-prepared samples in 0.1 M KOH electrolyte. a CV of N-C ANFs tested in O2/N2 saturated 0.1 M KOH solution with scan rate of 10 mV s⁻¹; **b** CV of 0.01-Fe-N-C ANFs, 0.03-Fe-N-C ANFs, 0.05-Fe-N-C ANFs and 0.10-Fe-N-C ANFs in O₂/N₂ saturated 0.1 M KOH solution; c LSV curves of Fe-N-C ANFs and Pt/C at rotation speed of 1600 rpm; d RDE voltammetry curves of the 0.05-Fe-N-C ANFs for ORR at rotation speeds ranging from 400 to 2025 rpm; e the K-L plots of 0.05-Fe-N-C ANFs at 0.50 V, 0.55 V, 0.60 V, and 0.65 V; f The Amperometric i-t curves of 0.05-Fe-N-C ANFs and Pt/C at 0.6 V in O2 saturated 0.1 M KOH solution



in this way, the calculated J_k of the 0.05-Fe-N-C ANFs and 0.10-Fe-N-C ANFs at 0.8 V (vs RHE)(0.68 mA cm⁻² and 1.03 mA cm⁻², respectively) are much larger than that of 0.01-Fe-N-C ANFs (0.12 mA cm⁻²) and 0.03-Fe-N-C ANFs (0.22 mA cm⁻²). Additionally, the mass J_k at 0.8 V (vs RHE) of 0.01-Fe-N-C ANFs, 0.03-Fe-N-C ANFs, 0.05-Fe-N-C ANFs and 0.10-Fe-N-C ANFs are 0.46 A g⁻¹,

0.82 A g⁻¹, 2.55 A g⁻¹ and 3.84 A g⁻¹, respectively. The mass J_k of 0.05-Fe-N-C ANFs and 0.10-Fe-N-C ANFs at 0.8 V are comparable with that of Fe-N/C catalysts in literatures [46]. Moreover, the durability of 0.05-Fe-N-C ANFs was measured at a constant voltage for 40,000 s in O₂ saturated 0.1 M KOH electrolyte, and the *i*-t curves reveal that 0.05-Fe-N-C ANFs present a small fluctuation

Fig. 6 The SEM images of 0.05-Fe-N-C ANFs **a** before and **b** after the electro-chemical tests



Fig. 7 The SEM mapping images of a 0.10-Fe-N-C ANFs, b 0.05-Fe-N-C ANFs, c 0.03-Fe-N-C ANFs, d 0.01-Fe-N-C ANFs, and e N-C ANFs. The corresponding C element, N element, O element, and Fe element mapping images are also shown in the figure



(95.56%) after 40,000 s, while the normalized current of Pt/C decreases to approximately 79.93% after 40,000 s (Fig. 5f).

As a result, on the one hand, the catalytic activity of Fe-N-C ANFs enhances with iron concentration increases, namely 0.01-Fe-N-C ANFs to 0.05-Fe-N-C ANFs, which can be explained that the Fe-N_x active sites increase correspondingly; on the other hand, with iron concentration continuously increasing in 0.10-Fe-N-C ANFs sample,

its catalytic activity does not increase monotonously, which could be ascribing to the bulk agglomeration of iron-containing nanoparticles, resulting in the active sites aggregation and inhomogeneous.

The SEM morphologies of 0.05-Fe-N-C ANFs before and after the electro-chemical tests are shown in Fig. 6. It is observed that the structure of the one-dimensional Fe-N-C ANFs can be basically maintained, that is to say, the microstructure of the 0.05-Fe-N-C ANFs after ORR in 0.1 M



Table 1 C, N, Fe, and O content in the as-prepared samples, and No. 1, 2, 3, and 4 represent the 0.01-Fe-N-C ANFs, 0.03-Fe-N-C ANFs, 0.05-Fe-N-C ANFs, and 0.10-Fe-N-C ANFs, respectively	No.	C (at.%)	N (at.%)	N _{pyr} (at.%)	N _{Fe-Nx} (at.%)	N _{N-H} (at.%)	N _g (at.%)	Fe (at.%)	O (at.%)
	1	78.11	16.98	11.48	1.09	2.92	1.49	0.36	4.55
	2	77.36	16.07	10.19	2.41	2.24	1.23	0.58	5.99
	3	75.15	15.30	7.66	3.85	1.46	2.33	1.63	7.92
	4	74.58	15.70	10.20	3.15	1.14	1.22	2.08	7.64

KOH solution was not destroyed, suggesting good durability in another point of view.

SEM mapping images of the 0.01-Fe-N-C ANFs, 0.03-Fe-N-C ANFs, 0.05-Fe-N-C ANFs, and 0.10-Fe-N-C ANFs (Fig. 7) show that the carbon, nitrogen, oxygen, and iron elements distribute on N-C ANFs homogeneously and almost non-aggregation. Moreover, it could be seen that the Fe concentration gradually decreases from 0.10-Fe-N-C ANFs to 0.01-Fe-N-C ANFs, and only a few agglomerated iron-containing nanoparticles appear in 0.10-Fe-N-C ANFs (Fig. 7a), while there are no aggregations in the other as-prepared samples.

To figure out the N, and Fe bonding configurations in 0.01-Fe-N-C ANFs, 0.03-Fe-N-C ANFs, 0.05-Fe-N-C ANFs, and 0.10-Fe-N-C ANFs, we investigated their X-ray photoelectron spectroscopy (XPS), as shown in Fig. 8. It can be seen that carbon, nitrogen, oxygen, and iron element characteristic-peaks appear in the XPS total spectra, as shown in Fig. 8. The corresponding element concentration is listed in Table 1, from which it could be found that the changes of carbon and nitrogen element concentration are ruleless, while iron element concentration monotonously increasing from 0.01 to 0.10 M. In addition, the C 1s and N 1s XPS spectra are also analyzed to investigate the changes of the bonding configurations and it could be observed that N 1s XPS spectra of the Fe-N-C ANFs can be deconvoluted into four types according to the binding energy: pyridinic-N (N_{pyr}, 398.4 eV), Fe-N_x (N_{Fe-Nx}, 399.3 eV), pyrrolic-N (N_{N-H}, 400 eV), and graphitic-N $(N_g, 401 \text{ eV})$ [47]. It is exhibited that the concentration of Fe-N_x group first increases, then decreases along with the iron element concentration increases from 0.01-Fe-N-C ANFs to 0.10-Fe-N-C ANFs, which is in agreement with the electro-catalytic activity for ORR of the as-prepared samples. As a result, the Fe- N_x groups could be the electrocatalytic active sites for ORR instead of Fe.

Conclusions

To sum it up, the structured Fe-N-C ANFs/CP (The activated polypyrrole nanofibers doped with iron element in-situ electro-chemical polymerization on the home-made carbon paper (CP)) was synthesized by electrochemical

polymerization. And the 0.05-Fe-N-C ANFs exhibit more positive onset-potential and higher peak current density than those of 0.01-Fe-N-C ANFs, 0.03-Fe-N-C ANFs, and 0.10-Fe-N-C ANFs, revealing superior electro-catalytic activity for ORR, which could be ascribing to higher Fe- N_x concentration, and well-dispersed as well as undiluted catalytic active sites. Meanwhile, 0.05-Fe-N-C ANFs show better alkaline tolerance than the commercial Pt/C catalyst. However, the catalytic activity of the Fe-N-C ANFs is much poorer than that of the commercial Pt/C catalyst, which could be ascribing to the low catalytic active-site concentration.

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