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# Highly porous disk-like shape of  $Co<sub>3</sub>O<sub>4</sub>$  as an anode material for lithium ion batteries

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Abstract A novel disk-like shape of  $Co<sub>3</sub>O<sub>4</sub>$  with high porosity was synthesized by a facile hydrothermal approach followed by calcination at 485 °C for 2 h. In order to further confirm the crystal structure, morphology, particle size, surface area, and porosity of the sample, a series of corresponding characterization techniques were used. The disk-like shape of  $Co<sub>3</sub>O<sub>4</sub>$  as an anode delivered excellent rate capability such as  $510.5$  mAh g<sup>-1</sup> at 4.0 C, which is much higher than the theoretical capacity of commercial graphite anode (372 mAh  $g^{-1}$ ). However, the electrode could not recover the high capacity during the long-term cycling at various higher current rates due to the deformation of the structure as confirmed by the ex situ studies. It is believed that the obtained remarkable structural feature with numerous void pores within the structure may be helpful for short-term cycling due to the large contact areas between the electrode and the electrolyte and a shorter diffusion length for lithium ion insertion but unable to act as a buffer to relax the volume expansion/contraction and alleviate the structural damage of the electrode during long-term cycling.

Keywords  $Co<sub>3</sub>O<sub>4</sub>$  Hydrothermal Porous Anode Lithium ion battery

## Introduction

To design new anode materials with higher safety, excellent rate performance, and great power densities, transition metal

 $\boxtimes$  Alok Kumar Rai alokkumarrai1@gmail.com oxides are considered as potential anodes for next-generation lithium ion battery (LIBs) [\[1](#page-6-0)]. Cobalt oxide  $(Co<sub>3</sub>O<sub>4</sub>)$  has been widely studied among all the transition metal oxides due to its high theoretical capacity (890 mAh  $g^{-1}$ ), low cost, facile synthesis, and good chemical/thermal stability [[2\]](#page-6-0). Since the theoretical capacity of  $Co<sub>3</sub>O<sub>4</sub>$  is more than twice higher than that of commercial graphite anode (372 mAh  $g^{-1}$ ), it is believed that  $Co<sub>3</sub>O<sub>4</sub>$  would satisfy the requirements of future energy storage systems. However, the practical application of  $Co<sub>3</sub>O<sub>4</sub>$ is still limited due to various reasons such as large volume expansion/contraction and pulverization of the electrode material during Li<sup>+</sup> insertion/deinsertion, which leads to cracking, fracture, and electrical contact loss of the electrode material [[3\]](#page-6-0). In order to address these issues, various strategies have been attempted such as synthesis of a variety of  $Co<sub>3</sub>O<sub>4</sub>$  nanostructures to increase the electrochemical activity [[4](#page-6-0)], fabrication of nanocomposites with more efficient conductive components to enhance the conductivity [[5\]](#page-6-0), and direct growth of  $Co<sub>3</sub>O<sub>4</sub>$  nanomaterials on binder-free and freestanding substrate to avoid the usage of inert components in the electrode [\[6](#page-6-0)]. However, in spite of all the above attempts, the issues with  $Co<sub>3</sub>O<sub>4</sub>$  as an anode material are still unsolved. Therefore, in the past few years, several considerable efforts have also been made to synthesize highly porous  $Co<sub>3</sub>O<sub>4</sub>$  with rationally designed novel morphologies [\[7](#page-6-0), [8](#page-6-0)]. It is believed that the porous structure endows the materials with more electrochemical active sites, a large accessible surface area for the interaction with the electrolyte, and short mass/ion pathways, which were considered as potential parameters for high-performance electrode materials. However, the synthesis of the porous  $Co<sub>3</sub>O<sub>4</sub>$ anode with extra accessible space for  $Li<sup>+</sup>$  ion insertion/ extraction via a facile synthetic approach still remains a challenge.

Herein, for the first time, a facile hydrothermal approach is used to synthesize the highly porous novel disk-like shape of

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 $Co<sub>3</sub>O<sub>4</sub>$ . The obtained porous disk-like shape of the  $Co<sub>3</sub>O<sub>4</sub>$ anode delivered an excellent capacity of 510.5 mAh  $g^{-1}$  at a high current rate of 4.0 C (1 C = 890 mA  $g^{-1}$ ) after 50 cycles, which is still higher than the theoretical capacity of commercial graphite anode. The enhanced C-rate performance can be attributed to the highly porous unique architecture of the material in the electrode. It is believed that the porosity could increase the electrolyte/ $Co<sub>3</sub>O<sub>4</sub>$  interfacial area, shorten the Li<sup>+</sup> diffusion path, and improve the electrochemical kinetics. Moreover, the obtained porous structure not only provides a sufficient space to buffer against the local volume change and mitigate the pulverization problem but also promotes the easy access of the electrolyte into the bulk electrode material and offers fast lithium ion transports, which are necessary for highrate performance.

### Experimental

#### Material synthesis

Typically, 0.01 mol of cobalt acetate (99.999%, Sigma-Aldrich) was dissolved in a mixture of deionized water (15 ml) and ethanol (15 ml) under continuous stirring. Then, 2.5 ml of ammonium hydroxide (Nice Chemicals, India, 25%) was added dropwise to the solution to maintain the pH at 9. After continuous stirring for 1 h, the resultant solution was sealed into a Teflon-lined stainless steel autoclave and then heated at 170 °C for 3 h. Subsequently, the obtained material was filtered, washed with distilled water and ethanol several times, and then dried at 70 °C for 12 h in an oven. The asprepared powder was eventually calcined in a muffle furnace at 485 °C for 2 h with a 0.25° ramp rate.

#### Material characterization

The crystal structure of the sample was characterized using powder X-ray diffraction (XRD, Shimadzu XRD-6000) with Cu-K<sub>α</sub> radiation ( $\lambda = 1.5406$  Å). Field-emission scanning electron microscopy (FE-SEM; S-4800 Hitachi) and fieldemission transmission electron microscopy (FE-TEM; FEI 20 FEG) were used to examine the morphology of the sample.  $N_2$  adsorption–desorption isotherms were measured using a BELSORP-mini II instrument. X-ray photoelectron spectroscopy (XPS) was performed using a Kratos Analytical, Ultra axis XPS instrument with a monochromatic Al  $K_{\alpha}$  X-ray source. Raman spectroscopy was performed on Witec Alpha 300 RA with a diode laser as the excitation source.

#### Electrochemical measurements

The electrochemical tests were carried out with a CR2032 type coin cell. Lithium foil was used as the counter electrode.

The electrode slurry was made by mixing the active material, carbon black, and PVDF in a weight ratio of 70:20:10 in Nmethyl pyrrolidone, and the resultant slurry was then uniformly pasted on a Cu foil (current collector). The electrolyte was 1 M LiPF $_6$  dissolved in ethylene carbonate/dimethyl carbonate mixture (1:1 in volume). The electrochemical tests were performed using BioLogic Science Instruments (BCS-805) within the voltage range of 0.01–3.0 V.

### Results and discussion

#### Structural and morphological characterization

Figure [1](#page-2-0)a shows the XRD pattern of the annealed  $Co<sub>3</sub>O<sub>4</sub>$  sample. The diffraction peaks are well-matched with that of the standard cubic spinel  $Co<sub>3</sub>O<sub>4</sub>$  (JCPDS Card No. 421467). There were no other impurity peaks in the sample, indicating the phase purity of the  $Co<sub>3</sub>O<sub>4</sub>$ . XPS analysis was also performed to investigate the oxidation state and surface composition of the sample. As shown in Fig. [1b](#page-2-0), the wide spectrum of  $Co<sub>3</sub>O<sub>4</sub>$  clearly shows sharp peaks of C-1s, O-1s, and Co-2p, indicating the existence of carbon, oxygen, and cobalt elements in the sample. More precisely, the high-resolution XPS spectrum of Co-2p (Fig. [1c](#page-2-0)) exhibited two peaks located at 779.7 eV for  $Co-2p_{3/2}$  and 795.1 eV for  $Co-2p_{1/2}$  with a binding energy difference of 15.4 eV, corresponding to the  $Co<sub>3</sub>O<sub>4</sub>$  phase reported in the literature [\[4](#page-6-0)]. The existence of  $Co<sub>3</sub>O<sub>4</sub>$  can be further confirmed by the O-[1](#page-2-0)s peak (Fig. 1d) located at 530.3 eV, which corresponds to the lattice oxygen species of  $Co<sub>3</sub>O<sub>4</sub>$  [\[4](#page-6-0)]. Figure [1](#page-2-0)e shows the Raman spectrum of the obtained  $Co<sub>3</sub>O<sub>4</sub>$  sample. The sharp peaks at 184.3 and 458.3 cm−<sup>1</sup> and a wide peak in the range of 550.1 to  $680.2 \text{ cm}^{-1}$  are observed, which can be attributed to the various modes of  $Co<sub>3</sub>O<sub>4</sub>$  [[9,](#page-6-0) [10\]](#page-6-0).

Figure [2](#page-3-0)a–c shows the typical FE-SEM images of the porous disk-like shape of  $Co<sub>3</sub>O<sub>4</sub>$  at different magnifications. The FE-SEM image clearly shows the disk-like shape of the sample with multi-porous structures. The dimensions of the porous  $Co<sub>3</sub>O<sub>4</sub>$  structures range from  $~500$  to 1000 nm in length and ~700 to 900 nm in width, respectively. A detailed structural investigation of the obtained  $Co<sub>3</sub>O<sub>4</sub>$  is done by HR-TEM. A high-magnification FE-TEM image of an individual  $Co<sub>3</sub>O<sub>4</sub>$ structure is shown in Fig. [2d](#page-3-0). The presence of numerous irregular pores can be observed in the structure, which is well consistent with the results of FE-SEM. The clear contrasts inside the structure also imply the presence of pores. It can be observed from the FE-TEM images (Fig. [2e](#page-3-0), f) that the nanosized pores are uniformly distributed throughout the disk-like structures. It is reasonable to suggest that the presence of porous architecture and hollow cavities in the structure will increase the surface area of  $Co<sub>3</sub>O<sub>4</sub>$  and may lead to a great enhancement of the electrochemical activities. A lattice-resolved HR-TEM study

<span id="page-2-0"></span>

 $Co<sub>3</sub>O<sub>4</sub>$ 



of the disk-like shape of  $Co<sub>3</sub>O<sub>4</sub>$  shows a well-defined crystalline structure with lattice spacing of 2.01 Å, corresponding to the (400) plane of the cubic  $Co<sub>3</sub>O<sub>4</sub>$  phase. The formation of pores in the structure can be attributed to the combined effects generated by the decomposition of the large amount of organic species and the simultaneous gas evacuation during heat treatment [[11,](#page-6-0) [12](#page-6-0)].

The material was further characterized by BET surface area measurement. The  $N_2$  adsorption/desorption isotherms (Fig. [3](#page-3-0)) exhibited type IV isotherm with a H3-type hysteresis loop. The surface area of the sample was measured to be 40.4  $\text{m}^2$  g<sup>-1</sup>. The average pore diameters and pore volumes calculated by the corresponding Barrett–Joyner–Halenda (BJH) pore size distribution curve (inset of Fig. [3](#page-3-0)) were found to be 42.5 nm and 0.43 cm<sup>3</sup>  $g^{-1}$ , respectively, which clearly show the mesoporous nature of the disk-like shape of the Co3O4 sample. Obviously, the porous structure can not only provide more locations and channels for fast diffusion of Li+ ion into the electrode material but also accommodate the volume changes of the charge/discharge process to maintain the structural integrity.

#### Electrochemical studies

Figure [4](#page-4-0)a displays the cyclic voltammograms (CV) for the first five cycles at a scan rate of 0.1 mV s<sup> $^{-1}$ </sup> under the voltage

<span id="page-3-0"></span>Fig. 2 a–c FE-SEM images and d–f FE-TEM images of the porous disk-like shape of  $Co<sub>3</sub>O<sub>4</sub>$ at different regions. The inset of f shows the HR-TEM image of the same disk-like shape of  $Co<sub>3</sub>O<sub>4</sub>$ 



window of 0–3.0 V. The first CV profile shows a strong cathodic peak at 0.89 V and anodic peak at 2.05 V, which can be assigned to the reversible reduction and oxidation reaction of  $Co<sub>3</sub>O<sub>4</sub>(Co<sub>3</sub>O<sub>4</sub> + 8Li<sup>+</sup> + 8e<sup>-</sup> \leftrightarrow 3Co + 4Li<sub>2</sub>O)$  along with the formation of a solid electrolyte interphase (SEI) layer [\[7](#page-6-0)]. The formation of the SEI layer is an irreversible process, which will result in the initial irreversible capacity loss. Afterwards, it can be seen that the anodic peak position remains the same in the subsequent cycles, whereas the cathodic peak shifts to a higher potential of 1.1 V with less intensity, which is caused by the polarization of the electrode material. After the first cycle, the overlapping of the CV curves can be seen, indicating the good reversibility of the porous disk-like shape of  $Co<sub>3</sub>O<sub>4</sub>$  electrode.

Figure [4b](#page-4-0) shows the 1st, 2nd, 5th, 10th, 15th, and 20th charge/discharge profiles of the porous disk-like shape of the  $Co<sub>3</sub>O<sub>4</sub>$  electrode at the current rate of 0.25 C. It can be seen



Fig. 3 Nitrogen adsorption/desorption isotherm and BJH pore size distribution curve of the porous disk-like shape of  $Co<sub>3</sub>O<sub>4</sub>$ 

that the potential value in the first discharge curve quickly falls to 1.1 V with an extended plateau and then gradually declines to the cutoff voltage of 0.01 V, which can be assigned to the conversion reaction of  $Co<sub>3</sub>O<sub>4</sub>$  to Co along with the formation of a polymer gel-like SEI layer on the surface of the  $Co_3O_4$ particles. This finding is consistent with the above CV results. In the first cycle, the electrode delivered a discharge capacity of 1260 mAh  $g^{-1}$  and a charge capacity of 981.4 mAh  $g^{-1}$  with an initial irreversible capacity loss of 22%, which can be attributed to the inevitable formation of the SEI layer and decomposition of the electrolyte. The subsequent charge/ discharge curves in the various cycles tend to be stable with high Coulombic efficiency and exhibit similar electrochemical behavior. The electrode delivered discharge and charge capacities of 1000 and 977.5 mAh  $g^{-1}$  for the 2nd cycle and 1070 and 1052.6 mAh  $g^{-1}$  for the 20th cycle, respectively, with an average irreversible capacity loss of only 2%, which indicates high stability and excellent capacity retention of the electrode. It can also be noticed that the obtained capacities in the following cycles are quite higher than the theoretical capacity of  $Co<sub>3</sub>O<sub>4</sub>$  (890 mAh  $g<sup>-1</sup>$ ), which is probably caused by the reversible formation/dissolution of the gel-like film originating from the kinetic activation in the electrode and contributing to an additional reversible capacity [\[4](#page-6-0), [11\]](#page-6-0).

Figure [4c](#page-4-0) shows the cycling performance of the electrode at different current rates of 0.25 and 1.0 C for 100 cycles. High reversible capacities of 851.2 mAh  $g^{-1}$  at 0.25 C and 448.4 mAh  $g^{-1}$  at 1.0 C are achieved after 100 charge/ discharge cycles with almost 100% Coulombic efficiency. Moreover, in order to further investigate the effect of high porosity on the electrochemical performance of the disk-like shape of the  $Co<sub>3</sub>O<sub>4</sub>$  electrode, the cycling performance was again examined at high current rates of 2.0 and 3.0 C for 100 cycles, as shown in Fig. [4](#page-4-0)d. The electrode exhibits reversible capacities of only 304 mAh g<sup>-1</sup> at 2.0 C and 260.9 mAh g<sup>-1</sup> at 3.0 C after 100 cycles. It can be clearly seen that the obtained

<span id="page-4-0"></span>

Fig. 4 a CV profile, b discharge–charge curves at 0.25 C, c cycling performance at a lower current rate of 0.25 and 1.0 C, d cycling performance at higher current rates of 2.0 and 3.0 C, e rate performance between 0.5 to 4.0 C of the porous disk-like shape of the  $Co<sub>3</sub>O<sub>4</sub>$  electrode

capacities after long-term cycling on various high current rates such as 1.0, 2.0, and 3.0 C are not very promising towards battery application. It can also be seen that the electrode could not maintain the highest reversible capacity as obtained in the initial few cyclings during the long-term cycling. It is believed that the surface of the active material only participates in the electrochemical reaction at higher current rates due to the kinetic limitations, which leads to the decrease in the capacity [\[13](#page-6-0), [14](#page-6-0)]. In addition, it is also possible to suggest that the decrease in capacity may be due to the low electronic conductivity of  $Co<sub>3</sub>O<sub>4</sub>$  since it is a semiconductor.

The rate capability of the porous disk-like shape of the  $Co<sub>3</sub>O<sub>4</sub>$  electrode is also tested at various current rates between 0.5 to 4.0 C for every ten successive cycles. The C-rate profile vs. cycle number is shown in Fig. 4e, indicating that the capacity values are fairly stable at each C rate. With the

Materials	Synthesis method	Current density/rate	Cycle number	Specific capacity Ref. $(mAh g^{-1})$	
$Co3O4$ nanocages	Thermal decomposition	2000 mA $g^{-1}$ = 2.2 C 25		252 mAh $g^{-1}$	$[15]$
Ultrathin mesoporous $Co3O4$ nanosheet	Hydrothermal synthesis	$2 \text{ A } g^{-1} = 2.2 \text{ C}$	50	479 mAh $g^{-1}$	[16]
Porous $Co3O4$ microspheres	Ultrasonic spray pyrolysis technique	400 mA $g^{-1}$ = 0.45 C 40		$654$ mAh g <sup>-1</sup>	$[17]$
$Co3O4$ cuboids	Thermolysis of metal organic framework	1600 mA $g^{-1}$ = 1.8 C 50		163 mAh $g^{-1}$	[18]
$Co3O4$ mesoporous microdisks	Solvothermal synthesis	800 mA $g^{-1}$ = 0.9 C 50		$260 \text{ mAh g}^{-1}$	[19]
Porous $Co3O4$ nanorod	Hydrothermal synthesis	2000 mA $g^{-1}$ = 2.2 C 25		516 mAh $g^{-1}$	$\lceil 2 \rceil$
Plate-like $Co3O4$ mesocrystal	Solution-phase route	2.0 C	40	$>500$ mAh g <sup>-1</sup>	[20]
Hierarchical network like $Co3O4$	Hydrothermal method	1000 mA $g^{-1}$ = 1.1 C 40		31 mAh $g^{-1}$	[21]
Porous disk-like shape of $Co3O4$	Hydrothermal synthesis	4.0 C	50	510.5 mAh $g^{-1}$	Current work

<span id="page-5-0"></span>**Table 1** Comparison of specific capacities of the current porous disk-like shape of the Co<sub>3</sub>O<sub>4</sub> electrode with previous Co<sub>3</sub>O<sub>4</sub> electrodes reported in the literatures

increase of the current rates, the corresponding capacities were measured and found to be 964.3 mAh  $g^{-1}$  at 0.5 C, 911.2 mAh  $g^{-1}$  at 1.0 C, 788.9 mAh  $g^{-1}$  at 2.0 C, 655.2 mAh g<sup>-1</sup> at 3.0 C, and 510.5 mAh g<sup>-1</sup> at 4.0 C with excellent reversibility. It is worth noticing that the capacity of the electrode obtained at a high current rate of 4.0 C is still higher than the theoretical capacity of commercial graphite anode (372 mAh  $g^{-1}$ ). In addition, in order to make it more clear, a comparison between the current work and various  $Co<sub>3</sub>O<sub>4</sub>$  nanostructures reported in the literatures is provided in Table 1. It can be clearly seen that the porous disk-like shape of  $Co<sub>3</sub>O<sub>4</sub>$  exhibits the highest capacity at 4.0 C [[2,](#page-6-0) [15](#page-6-0)–[21](#page-6-0)]. In addition, when the current rate returns to the initial C rate of 0.5 C after 50 cycles of charge/discharge on various high current rates, the electrode recovered almost 100% initial

capacity, which indicates the high-capacity retention capability of the electrode. It is believed that the porous architecture of the electrode could be trapping more  $Li<sup>+</sup>$  ions during electrochemical cycling, which leads to the improved electrochemical performance.

Ex situ studies were carried out to further understand the mechanism behind the deterioration of the capacity during long-term cycling at various higher current rates of 1.0, 2.0, and 3.0 C. In order to do the ex situ studies, the cycled electrodes were initially detached from the cell in an argon-filled glove box. Dimethyl carbonate was used to wash the electrode to remove the electrolyte. After drying, the electrode material was scraped off of the Cu substrate to do the ex situ SEM and TEM analyses. Figure 5a–d displays a representative ex situ SEM, EDX, and TEM of an electrode cycled at 1.0 C for 100

porous disk-like shape of the  $Co<sub>3</sub>O<sub>4</sub>$  electrode cycled at 1.0 C for 100 cycles. a FE-SEM image, b EDX spectrum, c and d FE-TEM images



<span id="page-6-0"></span>cycles. It can be easily observed that the porous disk-like shape of  $Co<sub>3</sub>O<sub>4</sub>$  is completely deformed during cycling, indicating that the obtained highly porous disk-like shape is not capable enough to relax the volume expansion/contraction and alleviates the structural damage of the electrode during the long term cycling. However, the EDX image (Fig. [5b](#page-5-0)) clearly shows the presence of Co and O which confirmed the chemical integrity of the sample.

### **Conclusion**

In summary, a novel disk-like shape of  $Co<sub>3</sub>O<sub>4</sub>$  with high porosity was fabricated via a facile hydrothermal method. The microstructure analysis shows that the obtained  $Co<sub>3</sub>O<sub>4</sub>$  is full of voids. Benefiting from this unique morphology, the disklike shape of the  $Co<sub>3</sub>O<sub>4</sub>$  anode delivered an excellent rate capability of 510.5 mAh  $g^{-1}$  at 4.0 C after 50 charge/ discharge cycles with excellent reversibility. However, the electrode could not deliver the high capacity during the long-term cycling at higher current rates of 1.0, 2.0, and 3.0 C due to the deformation of the structure as confirmed by the ex situ SEM and TEM studies. The high rate capability of the electrode can be ascribed to the unique architecture of the electrode, which offers a large contact area for the electrode/electrolyte interface and short pathways for Li<sup>+</sup> diffusion, as well as can provide more active sites to facilitate the electrochemical reactions and increase the tolerance towards volume change.

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