

# Facile synthesis of Fe@Fe<sub>2</sub>O<sub>3</sub> core-shell nanowires as O<sub>2</sub> electrode for high-energy Li-O<sub>2</sub> batteries

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**Abstract** Fe@Fe<sub>2</sub>O<sub>3</sub> core-shell nanowires were synthesized via the reduction of Fe<sup>3+</sup> ions by sodium borohydride in an aqueous solution with a subsequent heat treatment to form Fe<sub>2</sub>O<sub>3</sub> shell and employed as a cathode catalyst for non aqueous Li-air batteries. The synthesized core-shell nanowires with an average diameter of 50–100 nm manifest superior catalytic activity for oxygen evolution reaction (OER) in Li-O<sub>2</sub> batteries with the charge voltage plateau reduced to ~3.8 V. An outstanding performance of cycling stability was also achieved with a cutoff specific capacity of 1000 milliampere hour per gram over 40 cycles at a current density of 100 mA g<sup>-1</sup>. The excellent electrochemical properties of Fe@Fe<sub>2</sub>O<sub>3</sub> as an O<sub>2</sub> electrode are ascribed to the high surface area of the nanowires' structure and high electron conductivity. This study indicates that the resulting iron-containing nanostructures are promising catalyst in Li-O<sub>2</sub> batteries.

**Keywords** Lithium-oxygen batteries; Fe@Fe<sub>2</sub>O<sub>3</sub> · Core-shell nanowires · Catalyst · Reduction reaction

## Introduction

In recent years, the electric vehicles which have a fast-growing market are faced with serious problems in their power systems which have led to increasing demand for high energy density storage systems [1]. Since the specific energy

density of traditional lithium ion batteries are getting close to the theoretical limit, breakthroughs in a new battery system with ultra-high energy density are strongly expected [2]. Rechargeable Li-O<sub>2</sub> batteries with significantly high theoretical energy density have gained wide attention [3, 4]. Compared with current Li-ion batteries, Li-O<sub>2</sub> batteries have 10 times higher theoretical energy density owing to its different electrochemical reaction mechanism [5, 6]. Hence, much interest has been aroused among research workers since the first report by Abraham et al. [7] and the first experimental confirmation by Bruce et al. [8]. However, Li-O<sub>2</sub> batteries still suffer many challenges for practical applications, such as high overpotentials, poor cycle stability, and low rate capacity [9–11]. The main reason may be due to the sluggish kinetics of the oxygen evolution reaction (OER) and oxygen reduction reaction (ORR). So it is necessary to develop highly efficient cathode catalysts which can promote the OER and ORR, thus leading to a reduced overpotential and an improved cycle performance [12–16].

To date, various electrocatalysts, including carbons, metal oxides, precious metals, have been investigated as the cathode catalysts in Li-O<sub>2</sub> batteries to address the sluggish kinetics of ORR and OER in Li-O<sub>2</sub> batteries and lower the overpotentials [17–21]. Among these, transition metal oxides are of great interest because of the low price, relative stability, good catalytic performance, etc. MnO<sub>2</sub> and Co<sub>3</sub>O<sub>4</sub> and their composites are the most common transition metal oxides that have been studied by many researchers [19, 22–24]. Up to now, Fe-based catalysts have not been widely studied in Li-O<sub>2</sub> batteries, but they exhibit efficient electrocatalytic activities for the ORR in fuel cells [25]. Recently, some work on iron oxides as catalyst in Li-O<sub>2</sub> batteries was reported. Wenyu Zhang et al. [26] synthesized Fe<sub>2</sub>O<sub>3</sub> decorated graphene through an electrochemical deposition method to investigate its catalytic performance in Li-air batteries. Zhang et al. [27] synthesized Fe<sub>2</sub>O<sub>3</sub>

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nanoflakes as cathode catalysts in Li-O<sub>2</sub> batteries. Jun Lu et al. [28] fabricated Fe<sub>3</sub>O<sub>4</sub>/Fe/carbon composite for Li-O<sub>2</sub> batteries. All these works showed an obviously improved electrochemical performance, including lower overpotentials, higher capacity, and better cycling performance, but the polarization performance still should be improved. These studies inspired us to continue to explore the electrocatalytic performance of Fe-based catalysts for Li-air batteries. Owing to the advantages of low price, environment friendliness, and good performance, Fe-based catalysts should have a better potential in Li-O<sub>2</sub> batteries.

Here, in this work, we synthesized a kind of Fe/Fe<sub>2</sub>O<sub>3</sub> core-shell nanowires through the reduction of Fe<sup>3+</sup> ions by sodium borohydride in aqueous solution at ambient atmosphere. With a further heat treatment, the Fe nanowires were coated with a layer of Fe<sub>2</sub>O<sub>3</sub>. This kind of composite was used as a cathode in Li-O<sub>2</sub> batteries. The charge overpotential was reduced and a good cycling performance was achieved. The Li-O<sub>2</sub> cell with Fe@Fe<sub>2</sub>O<sub>3</sub> as a cathode can cycle more than 40 times with a cutoff specific capacity of 1000 mAh g<sup>-1</sup>.

## Experimental

### Preparation of Fe@Fe<sub>2</sub>O<sub>3</sub> composites

All chemical reagents were of analytical grade and were used without further purification. In a typical synthesis, 0.3 g of FeCl<sub>3</sub>·6H<sub>2</sub>O was added to 60 mL of deionized water and 0.6 g of NaBH<sub>4</sub> was dissolved in 40 mL deionized water. NaBH<sub>4</sub> solution was then poured into FeCl<sub>3</sub> solution at the rate of 0.8 mL/s. During the addition, the solutions were shaken by hand. The solution was bubbled with plenty of gas with the addition of NaBH<sub>4</sub> solution, accompanied by fluffy black precipitates that appeared on the surface of the solution. The fluffy black precipitates were washed with deionized water and ethanol and finally dried under vacuum at 60 °C overnight. Finally, the Fe@Fe<sub>2</sub>O<sub>3</sub> was obtained by heat treatment in an oven at 200 °C for 2 h.

### Material characterization

X-ray powder diffraction patterns were obtained on a Rigaku Ultima diffractometer with Cu-Kα radiation (λ = 1.542 Å). Morphologies of the synthesized samples were characterized with field emission scanning electron microscope (FESEM JSM-4800F) and a transmission electron microscope (TEM JEM-2100F). The surface area was determined by BET (Brunauer–Emmett–Teller) measurements using a Tristar 3000 surface area analyzer. To confirm the surface component, the X-ray photoelectron spectrum (XPS, Thermo Fisher Scientific ESCALab250) was also recorded. Cyclic voltammetry measurements in the potential window from

2.0 to 4.3 V (vs. Li<sup>+</sup>/Li) with a scan rate of 0.1 mV s<sup>-1</sup> was tested on an Autolab PGSTAT302 potentiostat-galvanostat electrochemical work station (Metrohm).

### Preparation of Li-O<sub>2</sub> cell and electrochemical investigation

The electrocatalytic activities of the as-prepared Fe@Fe<sub>2</sub>O<sub>3</sub> for the discharge and charge steps were tested in Li-O<sub>2</sub> cells. The electrodes were formed by mixing the Fe@Fe<sub>2</sub>O<sub>3</sub>, Ketjen black carbon (KB), and polytetrafluoroethylene (PTFE) with a weight ratio of 7:1:2 then rolling it into a film which was punched into small disks and pressed onto carbon paper as cathode. The electrochemical cells used were based on a Swagelok Cell design composed of a Li metal anode, an electrolyte of 1 M LiTFSI in TEGDME solvent, the Celgard 2400 separator, and the as-prepared porous cathode. To avoid the influence of H<sub>2</sub>O and CO<sub>2</sub> or any other impurities, the cells were operated at 1.5 mbar of pure O<sub>2</sub>.

The galvanostatic charge and discharge tests were conducted on a LAND CT2001A battery test system at different current densities, at ambient temperature after a 3-h rest period. All the specific capacities were calculated by normalizing with the mass loading of catalyst in the air electrodes.

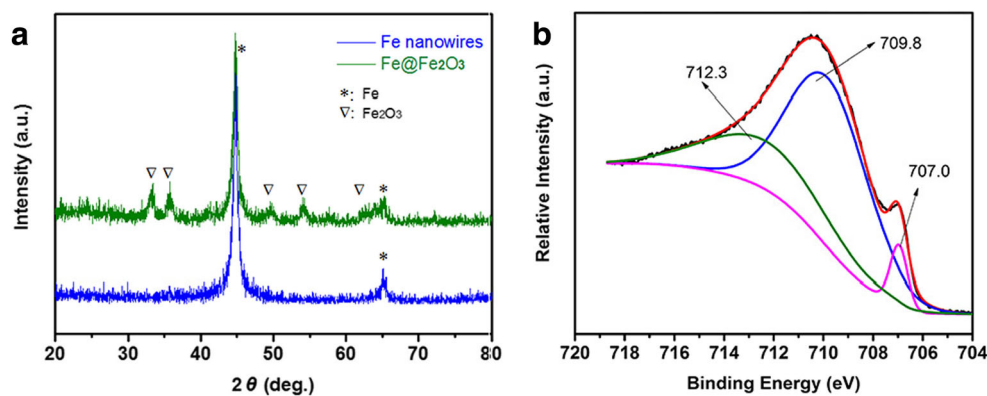
## Results and discussion

Figure 1a shows the XRD pattern of the as-prepared Fe nanowires and Fe@Fe<sub>2</sub>O<sub>3</sub> nanowires. The peaks located at 2θ ≈ 44.7° and 65.0° correlated to the (110) and (200) peaks of body-centered cubic iron (JCPDS card No. 06–0696). The peaks at 2θ ≈ 24.1°, 33.1°, 35.6°, 49.5°, 54.1°, and 62.4° correlated to the (012), (104), (110), (024), (116), and (214) peaks of α-Fe<sub>2</sub>O<sub>3</sub> (JCPDS card 33–0664). No other peaks have been observed, illustrating that the samples were comparatively pure with less impurities. It is obvious that the peak intensity of Fe is much stronger than Fe<sub>2</sub>O<sub>3</sub>, indicating the formation of Fe<sub>2</sub>O<sub>3</sub> on the surface of Fe nanowires.

The chemical composition of Fe@Fe<sub>2</sub>O<sub>3</sub> core-shell nanowires were further investigated by X-ray photoelectron spectroscopy (XPS). The XPS results of Fe 2p<sub>3/2</sub> spectra are shown in Fig. 1b. The Fe 2p<sub>3/2</sub> core level peak appeared at the binding energies of 709.8 eV corresponding to Fe<sub>2</sub>O<sub>3</sub> [29]. It indicated that the surface of the Fe nanowires was coated with Fe<sub>2</sub>O<sub>3</sub> layer, which was further confirmed by the TEM image. The peak at a lower binding energy of 707.0 eV is attributed to Fe 2p<sub>3/2</sub> in pure iron [30]. A higher binding energy peak at about 712.3 eV may be attributed to an interaction between the Fe core and the Fe<sub>2</sub>O<sub>3</sub> shell [29, 31].

Figure 2a shows the SEM images of the as-prepared sample. It shows that the morphology of the Fe@Fe<sub>2</sub>O<sub>3</sub> was nanowire structures and crisscross-like branches. The nanowires

**Fig. 1** **a** XRD pattern of Fe nanowires and Fe@Fe<sub>2</sub>O<sub>3</sub> nanowires; **b** XPS spectra of Fe 2p<sub>3/2</sub> peaks of the Fe@Fe<sub>2</sub>O<sub>3</sub> nanowires

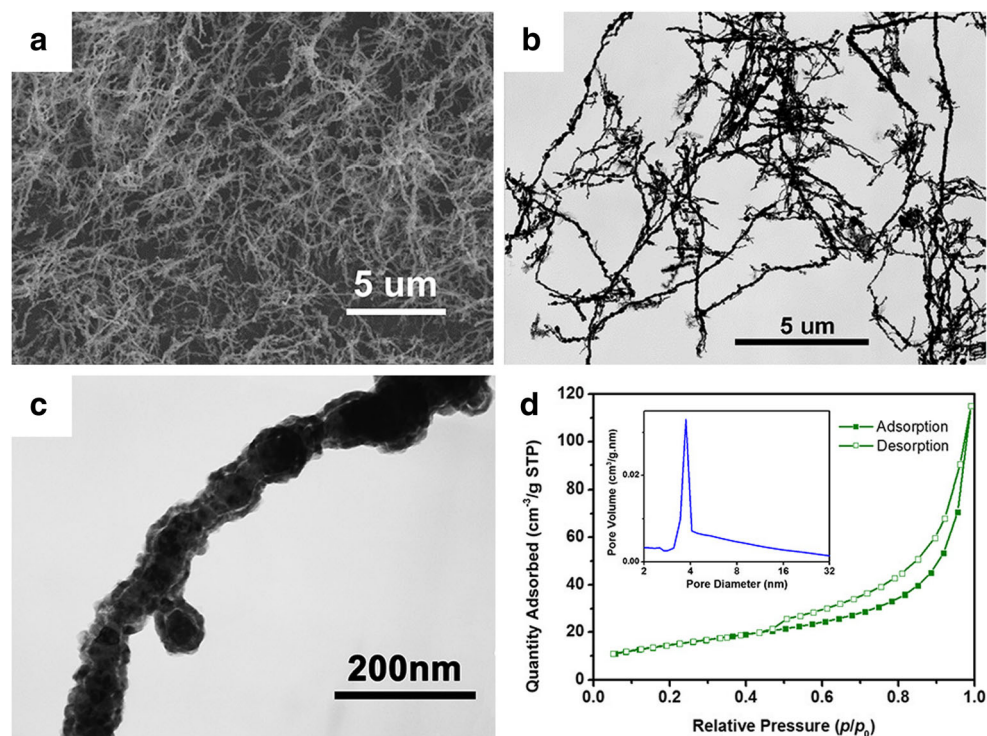


are several microns in length and 50–100 nm in diameter. The resulting nanostructures were further investigated by TEM. The structure of nanowires was further confirmed by the TEM images (Fig. 2b, c). The contrast between the edge and the center of the TEM image in Fig. 2c shows the core-shell structure of the Fe@Fe<sub>2</sub>O<sub>3</sub> nanowires and also confirms that the Fe<sub>2</sub>O<sub>3</sub> layer is about 15 nm in thickness.

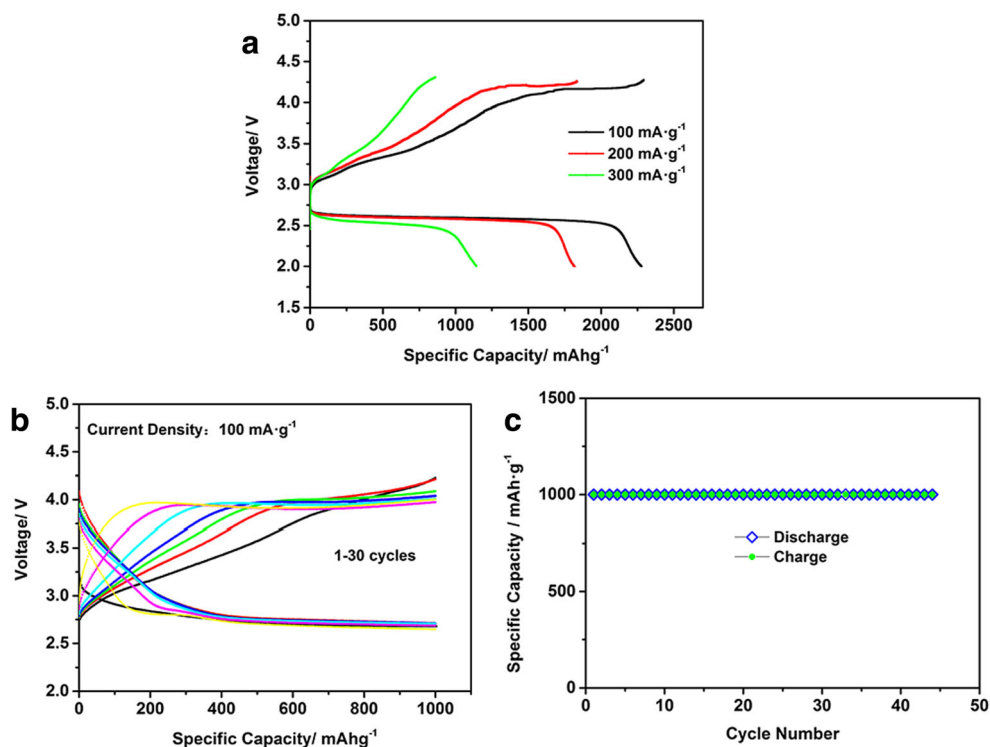
The BET-specific surface area detected by nitrogen adsorption-desorption measurements of Fe@Fe<sub>2</sub>O<sub>3</sub> was about 53.2 m<sup>2</sup> g<sup>-1</sup>. The pore size distributions of Fe@Fe<sub>2</sub>O<sub>3</sub> are shown in the inset of Fig. 2d. The strong peak at ~4 nm indicates the mesoporous nature of the prepared powder. The high specific surface area would be beneficial for the deposition of discharge products during ORR process which could increase the specific capacity of the Li-O<sub>2</sub> batteries.

The electrocatalytic activity of the as-prepared Fe@Fe<sub>2</sub>O<sub>3</sub> catalyst for ORR and OER reactions in Li-O<sub>2</sub> cells was investigated under galvanostatic cycling conditions. The initial discharge-specific capacity was measured at different current densities to characterize the rate capability of the cell (Fig. 3a). The battery with the Fe@Fe<sub>2</sub>O<sub>3</sub> cathode exhibits a high discharge capacity of 2270 mAh g<sup>-1</sup> at the current density of 100 mA g<sup>-1</sup>. The specific capacity decreased with the progressively increasing current from 100 to 300 mA g<sup>-1</sup>. Remarkably, the discharge capacity can still reach 1140 mAh g<sup>-1</sup> at a high current density of 300 mA g<sup>-1</sup>. Another important point is that the medium voltage during the first charge is only about 3.8 V at the current density of 100 mA g<sup>-1</sup>. The high specific capacity may be attributed to the high specific surface area and mesoporous nature of the

**Fig. 2** SEM image (a) and TEM images (b, c) of Fe@Fe<sub>2</sub>O<sub>3</sub> nanowires; (d) typical N<sub>2</sub> gas adsorption-desorption isotherm of the Fe@Fe<sub>2</sub>O<sub>3</sub> powder



**Fig. 3** First galvanostatic discharge/charge curves at different current density (a); discharging and charging profiles (b); cycle performance (c) of Li-O<sub>2</sub> batteries with Fe@Fe<sub>2</sub>O<sub>3</sub> cathode



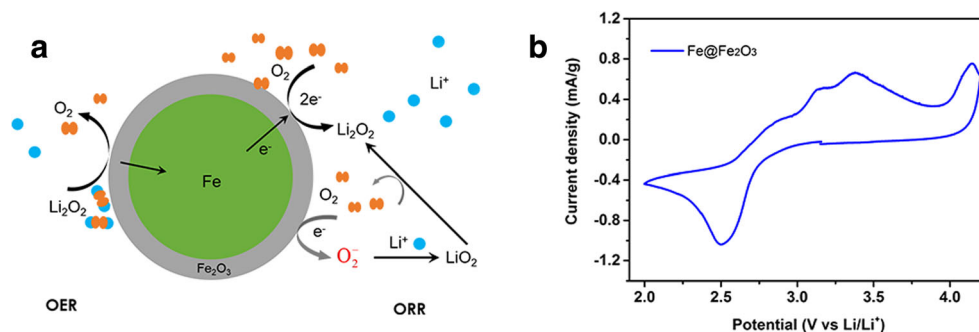
Fe@Fe<sub>2</sub>O<sub>3</sub> nanowires, and the core of Fe in the nanowire could increase the electron conductivity, which may lead to the good rate capability.

The cyclic performance of the Fe@Fe<sub>2</sub>O<sub>3</sub> cathode in Li-O<sub>2</sub> battery was shown in Fig. 3b. In order to avoid a large depth of discharge and achieve improved cyclic performance of the batteries, the discharge capacities of the electrodes were limited to 1000 mAh g<sup>-1</sup>. The battery exhibited excellent cycle stability over 45 cycles without pronounced variation in the very stable discharge/charge curves (Fig. 3c). After 30 cycles, the cutoff discharge voltage is still above 2.6 V and the cutoff charge voltage below 4.0 V, showing the good stability of Li-O<sub>2</sub> battery with the Fe@Fe<sub>2</sub>O<sub>3</sub> cathode.

One possible factor for this good catalytic performance is the core-shell Fe@Fe<sub>2</sub>O<sub>3</sub> structure. Based on some published references [32, 33], we presented the possible catalytic process of the Fe@Fe<sub>2</sub>O<sub>3</sub> core-shell nanowires in Fig. 4a. Molecular oxygen could be adsorbed on the surface of iron oxide and could be

reduced to peroxide iron or superoxide iron via the two-electron or one-electron transfer from iron core to the iron oxide shell surface to produce Li<sub>2</sub>O<sub>2</sub> or LiO<sub>2</sub> during the ORR process. The electrocatalytic activity of the Fe@Fe<sub>2</sub>O<sub>3</sub> electrode was further investigated by cyclic voltammetry (CV) (Fig. 4b). The Fe@Fe<sub>2</sub>O<sub>3</sub> electrode exhibited a high ORR onset potential (2.9 V), which implied a low ORR kinetic overpotential. The OER process started at the potential of 3.0 V, and the two obvious OER peaks between 3.0 and 3.5 V should be attributed to the oxidation of LiO<sub>2</sub> and Li<sub>2</sub>O<sub>2</sub> [32, 34]. The peak above 4.15 V should be due to the deposition of the carbon-containing substance such as electrolyte and carbon [35]. Also the high surface area of the nanowire structure could be beneficial for the discharge products' deposition, and it is also helpful for the transmission of oxygen. This result suggests that Fe@Fe<sub>2</sub>O<sub>3</sub> has considerable ORR/OER catalytic activity in Li-O<sub>2</sub> batteries. Since the catalytic reaction in Li-O<sub>2</sub> batteries is quite complicated, the reaction process still needs further study.

**Fig. 4** a Possible reaction process in Li-O<sub>2</sub> batteries with Fe@Fe<sub>2</sub>O<sub>3</sub> nanowires as catalyst; b cyclic voltammetry curves between 2.0 and 4.3 V at 0.1 mV s<sup>-1</sup>



## Conclusions

Fe@Fe<sub>2</sub>O<sub>3</sub> core-shell nanowires were synthesized via the reduction of Fe<sup>3+</sup> ions by sodium borohydride in aqueous solution with subsequent heat treatment to form Fe<sub>2</sub>O<sub>3</sub> shell and employed as a cathode catalyst for non-aqueous Li-air batteries. The resulted electrodes could promote the cycle stability of Li-O<sub>2</sub> batteries. The good performance may be due to the high specific surface area which provides more space for discharge products to deposit and the core-shell structure's good catalytic performance. The Fe core not only increased the conductivity of the cathode but also improved the catalytic activity. These data show that the Fe@Fe<sub>2</sub>O<sub>3</sub> core-shell nanowires could be a promising catalyst in Li-O<sub>2</sub> batteries.

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## References

- Bruce PG, Freunberger SA, Hardwick LJ, Tarascon J-M (2011) Li-O<sub>2</sub> and Li-S batteries with high energy storage. *Nat Mater* 11(1):19–29
- Armand M, Tarascon JM (2008) Building better batteries. *Nature* 451(7179):652–657
- Cao R, Lee JS, Liu ML, Cho J (2012) Recent progress in non-precious catalysts for metal-air batteries. *Adv Energy Mater* 2(7):816–829
- Kraytsberg A, Ein-Eli Y (2011) Review on Li-air batteries-opportunities, limitations and perspective. *J Power Sources* 196(3):886–893
- Zhao YL, Xu L, Mai LQ, Han CH, An QY, Xu X, Liu X, Zhang QJ (2012) Hierarchical mesoporous perovskite La<sub>0.5</sub>Sr<sub>0.5</sub>Co<sub>0.91</sub> nanowires with ultrahigh capacity for Li-air batteries. *P Natl Acad Sci USA* 109(48):19569–19574
- Xu JJ, Wang ZL, Xu D, Zhang LL, Zhang XB (2013) Tailoring deposition and morphology of discharge products towards high-rate and long-life lithium-oxygen batteries. *Nat Commun* 4:2438
- Abraham K, Jiang Z (1996) A polymer electrolyte-based rechargeable lithium/oxygen battery. *J Electrochem Soc* 143(1):1–5
- Ogasawara T, Débart A, Holzapfel M, Novák P, Bruce PG (2006) Rechargeable Li<sub>2</sub>O<sub>2</sub> electrode for lithium batteries. *J Am Chem Soc* 128(4):1390–1393
- Girishkumar G, McCloskey B, Luntz AC, Swanson S, Wilcke W (2010) Lithium—air battery: promise and challenges. *J Phys Chem Lett* 1(14):2193–2203
- Shao YY, Park S, Xiao J, Zhang JG, Wang Y, Liu J (2012) Electrocatalysts for nonaqueous lithium-air batteries: status, challenges, and perspective. *ACS Catal* 2(5):844–857
- Li FJ, Zhang T, Zhou HS (2013) Challenges of non-aqueous Li-O<sub>2</sub> batteries: electrolytes, catalysts, and anodes. *Energy Environ Sci* 6(4):1125–1141
- Li FJ, Chen Y, Tang DM, Jian ZL, Liu C, Golberg D, Yamada A, Zhou HS (2014) Performance-improved Li-O<sub>2</sub> battery with Ru nanoparticles supported on binder-free multi-walled carbon nanotube paper as cathode. *Energy Environ Sci* 7(5):1648–1652
- Liao K, Wang X, Sun Y, Tang D, Han M, He P, Jiang X, Zhang T, Zhou H (2015) An oxygen cathode with stable full discharge-charge capability based on 2D conducting oxide. *Energy Environ Sci* 8(7):1992–1997
- Black R, Lee JH, Adams B, Mims CA, Nazar LF (2013) The role of catalysts and peroxide oxidation in lithium-oxygen batteries. *Angew Chem Int Edit* 52(1):392–396
- Wang F, Wen ZY, Shen C, Rui K, Wu XW, Chen CH (2015) Open mesoporous spherical shell structured Co<sub>3</sub>O<sub>4</sub> with highly efficient catalytic performance in Li-O<sub>2</sub> batteries. *J Mater Chem A* 3(14):7600–7606
- Kim BG, Kim HJ, Back S, Nam KW, Jung Y, Han YK, Choi JW (2014) Improved reversibility in lithium-oxygen battery: understanding elementary reactions and surface charge engineering of metal alloy catalyst. *Sci Rep* 4:9
- Cui YM, Wen ZY, Liu Y (2011) A free-standing-type design for cathodes of rechargeable Li-O<sub>2</sub> batteries. *Energy Environ Sci* 4(11):4727–4734
- Zhang JK, Li PF, Wang ZH, Qiao JS, Rooney D, Sun W, Sun KN (2015) Three-dimensional graphene-Co<sub>3</sub>O<sub>4</sub> cathodes for rechargeable Li-O<sub>2</sub> batteries. *J Mater Chem A* 3(4):1504–1510
- Li M, Han C, Zhang YF, Bo XJ, Guo LP (2015) Facile synthesis of ultrafine Co<sub>3</sub>O<sub>4</sub> nanocrystals embedded carbon matrices with specific skeletal structures as efficient non-enzymatic glucose sensors. *Anal Chim Acta* 861:25–35
- Zhu D, Zhang L, Song M, Wang XF, Chen YG (2013) An in situ formed Pd nanolayer as a bifunctional catalyst for Li-air batteries in ambient or simulated air. *Chem Commun* 49(83):9573–9575
- Zhai XM, Yang W, Li MY, Lu GQ, Liu JP, Zhang XL (2013) Noncovalent hybrid of CoMn<sub>2</sub>O<sub>4</sub> spinel nanocrystals and poly(diallyldimethylammonium chloride) functionalized carbon nanotubes as efficient electrocatalysts for oxygen reduction reaction. *Carbon* 65:277–286
- Su D, Dou S, Wang G (2014) Single crystalline Co<sub>3</sub>O<sub>4</sub> nanocrystals exposed with different crystal planes for Li-O<sub>2</sub> batteries. *Sci Rep* 4:5767–5775
- Zhang P, He M, Xu S, Yan XB (2015) The controlled growth of porous delta-MnO<sub>2</sub> nanosheets on carbon fibers as a bi-functional catalyst for rechargeable lithium-oxygen batteries. *J Mater Chem A* 3(20):10811–10818
- Cao Y, Wei Z, He J, Zang J, Zhang Q, Zheng M, Dong Q (2012) α-MnO<sub>2</sub> nanorods grown in situ on graphene as catalysts for Li-O<sub>2</sub> batteries with excellent electrochemical performance. *Energy Environ Sci* 5(12):9765–9768
- Lefevre M, Proietti E, Jaouen F, Dodelet JP (2009) Iron-based catalysts with improved oxygen reduction activity in polymer electrolyte fuel cells. *Science* 324(5923):71–74
- Zhang WY, Zeng Y, Xu C, Tan HT, Liu WL, Zhu JX, Xiao N, Hng HH, Ma J, Hoster HE, Yazami R, Yan QY (2012) Fe<sub>2</sub>O<sub>3</sub> nanocluster-decorated graphene as O<sub>2</sub> electrode for high energy Li-O<sub>2</sub> batteries. *Rsc Adv* 2(22):8508–8514
- Zhang ZA, Zhou G, Chen W, Lai YQ, Li J (2014) Facile synthesis of Fe<sub>2</sub>O<sub>3</sub> nanoflakes and their electrochemical properties for Li-air batteries. *Ecs Electrochem Lett* 3(1):A8–A10
- Lu J, Qin Y, Du P, Luo XY, Wu TP, Ren Y, Wen JG, Miller DJ, Miller JT, Amine K (2013) Synthesis and characterization of uniformly dispersed Fe<sub>3</sub>O<sub>4</sub>/Fe nanocomposite on porous carbon: application for rechargeable Li-O<sub>2</sub> batteries. *Rsc Adv* 3(22):8276–8285
- Lu LR, Ai ZH, Li JP, Zheng Z, Li Q, Zhang LZ (2007) Synthesis and characterization of Fe-Fe<sub>2</sub>O<sub>3</sub> core-shell nanowires and nanonecklaces. *Cryst Growth Des* 7(2):459–464
- Mills P, Sullivan J (1983) A study of the core level electrons in iron and its three oxides by means of X-ray photoelectron spectroscopy. *J Phys D Appl Phys* 16(5):723

31. Poonjarernsilp C, Sano N, Sawangpanich N, Charinpanitkul T, Tamon H (2014) Effect of Fe/Fe<sub>2</sub>O<sub>3</sub> loading on the catalytic activity of sulfonated single-walled carbon nanohorns for the esterification of palmitic acid. *Green Chem* 16(12):4936–4943
32. Laoire CO, Mukerjee S, Abraham KM, Plichta EJ, Hendrickson MA (2010) Influence of nonaqueous solvents on the electrochemistry of oxygen in the rechargeable lithium-air battery. *J Phys Chem C* 114(19):9178–9186
33. Liu W, Ai Z, Cao M, Zhang L (2014) Ferrous ions promoted aerobic simazine degradation with Fe@Fe<sub>2</sub>O<sub>3</sub> core-shell nanowires. *Appl Catal B-Environ* 150:1–11
34. Shui JL, Karan NK, Balasubramanian M, Li SY, Liu DJ (2012) Fe/N/C composite in Li-O<sub>2</sub> battery: studies of catalytic structure and activity toward oxygen evolution reaction. *J Am Chem Soc* 134(40):16654–16661
35. Kang SJ, Mori T, Narizuka S, Wilcke W, Kim HC (2014) Deactivation of carbon electrode for elimination of carbon dioxide evolution from rechargeable lithium-oxygen cells. *Nat Commun* 5: 3937