

# Modified tricalcium silicate cement formulations with added zirconium oxide

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## Abstract

**Objectives** This study aims to investigate the effect of modifying tricalcium silicate (TCS) cements on three key properties by adding ZrO<sub>2</sub>.

**Materials and methods** TCS powders were prepared by adding ZrO<sub>2</sub> at six different concentrations. The powders were mixed with 1 M CaCl<sub>2</sub> solution at a 3:1 weight ratio. Biodentine (contains 5 wt.% ZrO<sub>2</sub>) served as control. To evaluate the potential effect on mechanical properties, the mini-fracture toughness (mini-FT) was measured. Regarding bioactivity, Ca release was assessed using ICP-AES. The component distribution within the cement matrix was evaluated by Feg-SEM/EPMA. Cytotoxicity was assessed using an XTT assay.

**Results** Adding ZrO<sub>2</sub> to TCS did not alter the mini-FT ( $p = 0.52$ ), which remained in range of that of Biodentine ( $p = 0.31$ ). Ca release from TSC cements was slightly lower than that from Biodentine at 1 day ( $p > 0.05$ ). After 1 week, Ca release from TCS 30 and TCS 50 increased to a level that was significantly higher than that from Biodentine ( $p < 0.05$ ).

After 1 month, Ca release all decreased ( $p < 0.05$ ), yet TCS 0 and TCS 50 released comparable amounts of Ca as at 1 day ( $p > 0.05$ ). EPMA revealed a more even distribution of ZrO<sub>2</sub> within the TCS cements. Particles with an un-reacted core were surrounded by a hydration zone. The 24-, 48-, and 72-h extracts of TCS 50 were the least cytotoxic.

**Conclusions** ZrO<sub>2</sub> can be added to TCS without affecting the mini-FT; Ca release was reduced initially, to reach a prolonged release thereafter; adding ZrO<sub>2</sub> made TCS cements more biocompatible.

**Clinical relevance** TCS 50 is a promising cement formulation to serve as a biocompatible hydraulic calcium silicate cement.

**Keywords** Tricalcium silicate · Zirconium oxide · Mini-fracture toughness · Ca release · Component distribution · Cytotoxicity

## Introduction

Hydraulic calcium silicate cements have widely been used not only in various endodontic treatments, including pulp capping, pulpotomy, apexogenesis, repair of root perforation, treatment of root resorption, root-end filling, and apexification [1, 2], but also as material in replacement of dentin [3]. A great amount of studies reported the production of an apatite-rich surface layer on the cement upon contact with phosphate buffered saline (PBS) or simulated body fluid (SBF) [4–6]. The mechanism of action involves the production of calcium silicate hydrate upon hydrolysis of the cement, which in turn induces nucleation and crystallization of calcium-phosphate minerals [7, 8].

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The deposition of such calcium-phosphate minerals was also observed in caries-like demineralized dentin [3, 9, 10]. In our previous study, calcium-phosphate minerals were formed within caries-like demineralized dentin upon 1-week SBF storage however, the re-mineralization zone stopped to become thicker when a thickness of approximately 40  $\mu\text{m}$  was reached. Even upon 6-month SBF storage, dentin remained demineralized at larger depths. A very plausible explanation is that this phenomenon is caused by a fast and densely formed re-mineralization zone that hinders further infiltration of calcium deep into the remaining demineralized dentin; hence, re-mineralization remains incomplete. In an attempt to overcome this re-mineralization block and so to improve the re-mineralization potential of hydraulic calcium silicate cements, the release of calcium from the hydrated cement should be controlled better, i.e., slowed down and prolonged.

Hydraulic calcium silicate cements are mainly composed of Portland cement, of which the main component is tricalcium silicate (TCS) [11]. The percentage of TCS in Portland cement varies due to the use of different raw materials and differences in the manufacturing process. This variation is likely to induce different cement reactivity and related properties. In addition, the release of heavy metal elements from Portland cement, like arsenic and lead, remains a concern [12, 13]. To maintain stable performance and reliable quality, Portland cement in hydraulic calcium silicate cements can better be replaced with pure TCS since TCS can be manufactured from pure raw materials using a sol-gel process under controlled conditions [14–16]. In vitro bioactivity testing of TCS revealed generation of hydroxyapatite (HAp) upon contact with SBF [17]; dentinal tubules were occluded by HAp crystals [16]. The commercial TCS-based dentin-replacement cement Biodentine (Septodont, Saint-Maur-des-fosses, France) was shown to possess improved handling characteristics, reduced discoloration potential, as well as comparable or even superior physical and biological properties than those of the Portland cement-containing mineral trioxide aggregate (ProRoot MTA; Dentsply, Tulsa, OK, USA) [18–20]. As such, TCS has great potential as biomaterial to induce dentin re-mineralization and pulpal repair.

Bismuth oxide, as radiopacifier, is added to hydraulic calcium silicate cements, such as ProRoot MTA (Dentsply) and MTA Angelus (Angelus, Londrina, PR, Brazil) [21]. Several studies have, however, shown that bismuth oxide decreased compressive strength [22, 23], increased the relative porosity [24], and negatively affected the biocompatibility of ProRoot MTA cement (Dentsply) [25, 26]. In addition, bismuth oxide adversely affected the setting time of Portland cement; with 20 wt.% bismuth oxide, the setting time increased by 20 % [23]. Bismuth oxide was also documented to interact with

collagen within dentin, thereby causing tooth discoloration, as was shown for MTA Angelus (Angelus) [27].

To avoid the abovementioned shortcomings, zirconium oxide ( $\text{ZrO}_2$ ) [23, 28–32], niobium oxide [33], barium sulfate [30], and zinc oxide [30] have been recommended as alternative radiopacifiers. Portland cement containing 30 wt.%  $\text{ZrO}_2$ , added either at micro- or nano-particle size, revealed a radiopacity of above 3-mm thickness of aluminum, which corresponds to the recommended ISO standard 6876 for an endodontic material [23, 29]; 30–50 wt.%  $\text{ZrO}_2$  resulted in a radiopacity comparable to that of ProRoot MTA (Dentsply) [31].  $\text{ZrO}_2$  is a bio-inert filler that does not affect the hydration reaction of Portland cement [28, 32]. On the contrary,  $\text{ZrO}_2$  added to Portland cement improved physical properties, like compressive strength, and triggered a better biological response than hydraulic calcium silicate cement to which bismuth oxide was added [32, 34]. In comparison with pure Portland cement, the Ca release from Portland cement with added  $\text{ZrO}_2$  was reduced [24]. This addition of  $\text{ZrO}_2$  to TCS may be a promising alternative to control the Ca release. Furthermore, the addition of  $\text{ZrO}_2$  reduces the relative concentration of TCS in hydraulic calcium silicate cements, thereby diminishing the high production cost.

The objective of this laboratory research was to investigate the effect of replacing TCS with varying percentages of  $\text{ZrO}_2$  on three properties of high relevance for hydraulic calcium silicate cements, which are strength, bioactivity, and biocompatibility. The null hypothesis tested was that adding  $\text{ZrO}_2$  does not affect the three key properties of TCS cements.

## Materials and methods

### Cement preparation

The materials used in present study included TCS powder (diameter of  $\pm 10 \mu\text{m}$ ; Mineral Research Processing, Meyzieu, France) and  $\text{ZrO}_2$  (diameter of  $\pm 200 \text{nm}$ ; Tosoh, Tokyo, Japan). Six different experimental cement powders were prepared by replacing TCS in part by  $\text{ZrO}_2$  at various concentrations: 0, 5, 10, 20, 30, and 50 wt.% (Table 1). Since calcium chloride ( $\text{CaCl}_2$ ) accelerates the setting time of TCS cement, 1 M  $\text{CaCl}_2$  aqueous solution was employed as the experimental cement liquid. The same powder to solution weight ratio of 3:1 was selected to achieve cements with a similar cement consistency, and thus similar handling and also similar surface-wetting characteristics. The different powders were mixed with 1 M  $\text{CaCl}_2$  aqueous solution for 30 s using a capsule mixer (RotoMix Capsule Mixer; 3M ESPE, Seefeld, Germany). Biodentine (Septodont), which contains 5 wt.%  $\text{ZrO}_2$ , served as control. The powder and liquid

**Table 1** List of the experimental cements investigated and their composition

Cement	TCS <sup>a</sup> (wt.%)	ZrO <sub>2</sub> (wt.%)	Powder/liquid weight ratio
TCS 0	100	0	3
TCS 5	95	5	3
TCS 10	90	10	3
TCS 20	80	20	3
TCS 30	70	30	3
TCS 50	50	50	3

<sup>a</sup> TCS tricalcium silicate

of Biodentine were mixed according to the manufacturer’s instruction.

**Mini-fracture toughness (mini-FT)**

Ten rectangular bars (16 × 2 × 1.5 mm) were prepared per experimental material and Biodentine by filling the cement into custom-made silicon molds. Proper adaptation was achieved ultra-sonically using a dental ultrasonic scaler equipped with a flat tip (MiniMaster Ultrasonic Scaler; Electro Medical Systems, Chemin de

la Vuarpilliere, Switzerland). Specimens were stored at 37 °C and 100 % humidity for 1 week. A single mini-FT notch tip was positioned at the middle of each specimen under a stereo-microscope (Leica M715, Wetzlar, Germany); the notch was prepared under water-cooling using a 150-µm ultra-thin diamond blade (M1D08; Struers, Ballerup, Denmark) at a feed speed of 0.015 mm/s and a wheel speed of 1000 rpm (Fig. 1a). The specimen was then transferred to the universal testing machine (Instron 5848 Micro Tester; Instron, Norwood, MA, USA) facing the specimen tip down in the test fixture. The so-called mini-FT ( $K_{Ic}$ ) was determined using a four-point bending test setup with a cross-head speed of 0.05 mm/min (Fig. 1b).

After testing, all fractured specimens were air dried and gold-sputter coated, and the exact dimensions of the mini-FT notch were measured using a measuring microscope at a magnification of ×250, after which  $K_{Ic}$  was calculated (MPa m<sup>1/2</sup>), according to the following equation from the ISO 24370:2005 standard:

$$K_{Ic} = \frac{F(S_o - S_i)}{BW^{3/2}} \times \frac{Y*min}{\sqrt{1000}}$$

with the minimum stress intensity factor coefficient Y\*min being calculated from the following function:

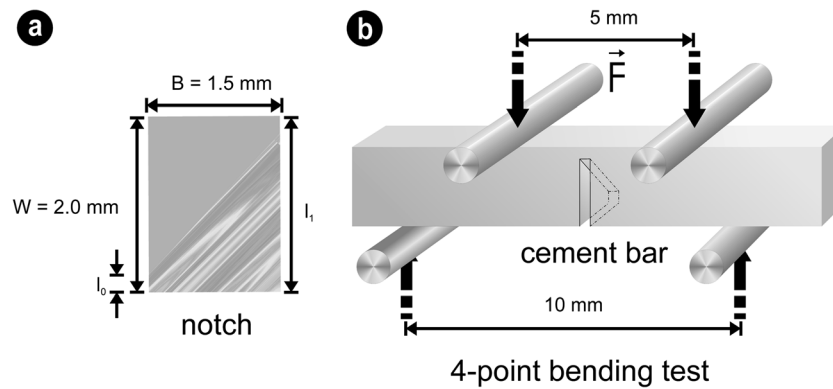
**Table 2** List of the cements selected for cytotoxicity testing

Cement	Manufacturer	Lot number	Composition <sup>a</sup>
TCS 0	Experimental	NA <sup>b</sup>	Powder: 100 wt.% tricalcium silicate Liquid: distilled water, calcium chloride Powder/liquid weight ratio: 3
TCS 50	Experimental	NA	Powder: 50 wt.% tricalcium silicate, 50 wt.% zirconium oxide Liquid: distilled water, calcium chloride Powder/liquid weight ratio: 3
Alganol	Kemdent	160350	Powder: zinc oxide Liquid: eugenol, olive oil
Biodentine	Septodont	B11172	Powder: tricalcium silicate, dicalcium silicate, calcium carbonate and oxide, iron oxide, zirconium oxide Liquid: distilled water, calcium chloride, hydrosoluble polymer
Dycal	Dentsply	140603	Base paste: 1,3-butylene glycol disalicylate, zinc oxide, calcium phosphate, calcium tungstate, iron oxide pigments Catalyst paste: calcium hydroxide, <i>N</i> -ethyl- <i>o</i> / <i>p</i> -toluene sulfonamide, zinc oxide, titanium dioxide, zinc stearate, iron oxide pigments
TheraCal LC <sup>c</sup>	Bisco	1400003407	Portland cement type III, polyethylene glycol dimethacrylate, barium zirconate

<sup>a</sup> For the commercial materials, based on technical information provided by the respective manufacturer

<sup>b</sup> NA not applicable

<sup>c</sup> A light-curing resin-modified calcium silicate cement



**Fig. 1** Schematic diagram explaining the mini-FT setup. **a** The single notch was prepared at the middle of each rectangular bar (1.5 × 2.0 mm). The notch width was smaller than 0.3 mm and the tip angle was 45°. The tip of the notch was located on the long surface at 0.24–0.48 mm ( $l_0$ ) from the left bottom corner. The opposite part of the notch cut ( $l_1$ ) did not touch

$$Y^*_{min} = [2.92 + 4.52(l_0/W) + 10.14(l_0/W)^2] \sqrt{\frac{(l_1/W) - (l_0/W)}{1 - (l_0/W)}}$$

where  $K_{Ic}$  is the fracture toughness value (MPa m<sup>1/2</sup>),  $F$  is the total force (N) (maximum force,  $F_{max}$ , plus tare force,  $F_{Tare}$ ),  $S_0$  is the outer span (mm) or 10.0,  $S_1$  is the inner span (mm) or 5.0,  $B$  is the specimen thickness (mm) or 1.50 ± 0.1,  $W$  is the specimen width (mm) or 2.00 ± 0.1,  $Y^*_{min}$  is the stress intensity factor coefficient,  $l_0$  is the position of the tip (mm) with 0.12 <  $l_0/W$  < 0.24, and  $l_1$  is the position of the bottom part (mm) with 0.90 <  $l_1/W$  < 1.00 (Fig. 1a). The effect of adding ZrO<sub>2</sub> on the  $K_{Ic}$  value of the experimental cements was evaluated by calculating the Pearson's correlation coefficient. As no correlation was found, the data from the experimental cements and Biodentine (Septodont) were compared by a  $t$  test. All statistical analyses were performed at a significance level of 0.05 using a software package (R3.01; R Foundation for Statistical Computing, Vienna, Austria).

### Ca release

The experimental cements TCS 0, TCS 30, and TCS 50 were chosen to investigate the effect of adding ZrO<sub>2</sub> on the Ca release from TCS cements. Biodentine (Septodont) and Dycal (Dentsply, Konstanz, Germany) served as controls. For each cement, five cylindrical cement blocks of 5-mm diameter and 1-mm thickness were prepared. Each specimen was stored at 37 °C for 24 h under 100 % humidity. A NaCl solution (133 mmol/L) was prepared and buffered to pH 7 using 50 mmol/L 4-(2-hydroxyethyl)-1-piperazineethanesulfonic acid (HEPES). Each cylindrical cement block was immersed in 5-mL solution; the solution was collected and refreshed after 1 day, 1 week, and 1 month. Ca release was analyzed using Inductively Coupled Plasma Mass Spectrometry (ICP-AES; VISTA

the top part, but ended less than 0.2 mm from the top side.  $W$  specimen width,  $B$  specimen thickness. **b** The mini-FT of the specimen was measured using a four-point bending test. The inner and outer spans were 5 and 10 mm, respectively

Pro, SII, Chiba, Japan). Prior to analysis, the solution was filtered and acidified using 5 mL of 10 % HNO<sub>3</sub> solution. The data were statistically analyzed using a linear mixed effects model taking into account the time intervals and cements tested as fixed effects and the specimen origin as a random effect. The values were assessed by a Tukey multiple comparisons test.

### Electron probe micro-analysis (EPMA)

The fractured mini-FT cement bars were air dried for 1 day and subsequently polished using an argon-ion-beam polisher (IB-09010CP Cross Section Polisher; JEOL, Tokyo, Japan) at 5.0 kV for 7 h to achieve an ion-beam polished cement surface of approximately 1 mm<sup>2</sup>. A 2-nm-thick Pt-Pd coating was applied to the cement surface by means of a turbomolecular-pumped coater (Q150T S; Quorum, East Sussex, UK). The hydration characteristics and the distribution of Ca, Si, and Zr along the cross-sections were investigated using Feg-SEM/EPMA (JXA-8530F; JEOL, Tokyo, Japan) at a high spatial resolution (±0.05 μm). X-ray profiles and quantification were performed at 15 kV and 15 μA (probe current) under high vacuum.

### Cement cytotoxicity

#### Cell culture

Healthy human third molars (from patients at 15–25 years of age), extracted for orthodontic reasons, were gathered as approved by the Commission for Medical Ethics of KU Leuven under the file number S57622. The teeth were rinsed in PBS (Gibco, Carlsbad, CA, USA) with 200 U/mL penicillin and 200 μg/mL streptomycin (Invitrogen, Carlsbad, CA, USA); the periodontal ligament was removed with a blade, and the teeth were next mechanically split. From each tooth, pulpal

tissue was gently separated using a forceps; the isolated pulpal tissue was cut with a scissor into approximately 1-mm<sup>3</sup> fragments. The pulp-tissue fragments were seeded in six-well plates (Costar, Cambridge, MA, USA) filled with culture medium consisting of Dulbecco modified Eagle medium (Gibco) supplemented with 10 % fetal bovine serum (FBS, Gibco). When human pulp fibroblast cells (HPFs) reached 70–80 % confluence, the cells were harvested using 0.25 % Trypsin/EDTA (Sigma-Aldrich, St. Louis, MO, USA) and observed as passage 0. The HPFs (passage 4–10) were cultured in 175 cm<sup>2</sup> cell culture flasks at 37 °C, 5 % CO<sub>2</sub>, and 100 % humidity. Experiments were performed with confluent cell monolayers at a density of 80–90 %.

#### Cement extracts

The experimental cements TCS 0 and TCS 50 were chosen to test the effect of adding ZrO<sub>2</sub> on the cytotoxicity of TCS cements. To correctly estimate the level of biocompatibility of the experimental cement formulations, commercial products Biodentine (Septodont), TheraCal LC (Bisco, Schaumburg, IL, USA), Dycal (Dentsply), and Alganol (Kemdent, Swindon, UK) were selected as controls (Table 2). The experimental cements were prepared by mixing the powder with 1 M CaCl<sub>2</sub> at a weight ratio of 3:1; the other cements were mixed according to the respective manufacturer's instructions. The cements were applied to the bottom of each well in a 24-well plate; the thickness of the cylindrical cement disks was 1 mm. The resin-based light-curing calcium silicate cement TheraCal LC (Bisco) was light-cured for 20 s (Bluephase 20i, Ivoclar Vivadent, Schaan, Liechtenstein). A 300-μL culture medium was immediately added to the cement disk in each well. The 24-, 48-, and 72-h extracts were collected; each extract concentrate was diluted with fresh culture medium to 1:1, 1:3, 1:9, 1:27, and 1:81 of the original concentration.

#### XTT assay

A Cell Proliferation Kit II (Roche Diagnostic, Penzberg, Germany) was used to assess the metabolic activity of cells on the basis of cleavage of yellow tetrazolium salt 2, 3-bis-(2-methoxy-4-nitro-5-sulphophenyl)-2H-tetrazolium-5-carboxanilide (XTT) by mitochondrial enzymes in metabolically active cells to form a soluble orange formazan product. The production of formazan is directly related to the number of vital cells. HPFs were seeded at a concentration of 20,000 cells/well in 100-μL culture medium in 96-well plates. The cells were incubated at 37 °C and 5 % CO<sub>2</sub> for 24 h. The culture medium was next removed and replaced by 50 μL of undiluted or diluted extracts. After 20 h of incubation and 4 h before photospectrometric analysis, 50 μL XTT labeling agent (in RPMI [Roswell Park Memorial Institute] without phenol red) and electron-coupling reagent (PMS [*N*-

methylidibenzopyrazine methyl sulfate] in PBS) was added to each well. Quantification of the formazan production was performed photospectrometrically at a wavelength of 450 nm using a microplate reader (Multiskan Ascent 96/384; Thermo Scientific, Waltham, MA, USA). The metabolic activities of cells, which were exposed to fresh culture medium or 1 % Triton X-100, were used as positive and negative controls, respectively. Each extract was tested in triplicate per test and the XTT test was repeated four times. Absorbance values of the positive and negative controls were adjusted to 100 and 0 % and the relative formazan production was calculated. Subsequently, concentration/dose-effect models were fitted for each cement at each extraction time separately; the estimated median effective dose (ED<sub>50</sub>) was calculated. This data processing was conducted using a software package (R3.01, and drc, package version 2.3-96).

## RESULTS

### Mini-FT (Fig. 2)

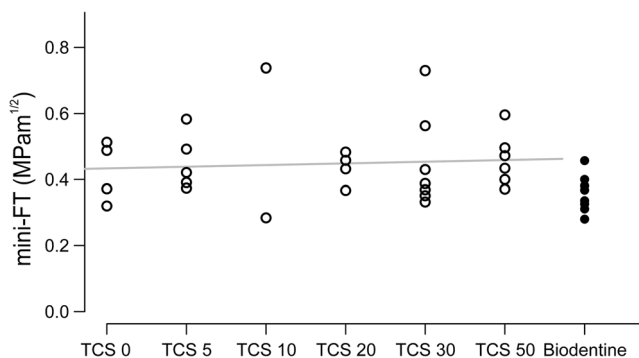
Adding ZrO<sub>2</sub> to TCS did not affect the mini-FT ( $K_{IC}$ ) of the experimental cements (estimated effect of 0.0005/wt.%;  $p = 0.52$ ). Overall, the experimental TCS cements exhibited a slightly but not significantly higher mean  $K_{IC}$  than Biodentine ( $0.45 \pm 0.11$  vs.  $0.36 \pm 0.06$ , respectively,  $p = 0.31$ ).

### Ca release (Table 3)

The concentrations of Ca ions released from each cement at 1 day, 1 week, and 1 month in distilled water are summarized in Table 3. At 1 day, Ca release among all experimental TCS cements did not differ statistically ( $p > 0.05$ ). The initial Ca release at 1 day was slightly higher for Biodentine, while at 1 week Ca release from TCS 30 and TCS 50 was increased to a level that was significantly higher than that from Biodentine ( $p < 0.05$ ). Ca release from all TCS cements and Biodentine significantly decreased after a 1-month storage period ( $p < 0.05$ ), yet TCS 0 and TCS 50 remained to release comparable amounts of Ca as at 1 day ( $p > 0.05$ ). The Ca release from Biodentine at 1 month was, however, significantly lower than the release at 1 day ( $p < 0.05$ ). At all time intervals, Dycal revealed the lowest Ca release, and its Ca release did not statistically significantly differ for the different time intervals ( $p > 0.05$ ).

### Component distribution and hydration characteristics (Fig. 3)

ZrO<sub>2</sub> particles with a small particle size (diameter of  $\pm 200$  nm) were rather evenly distributed within the matrix of the



**Fig. 2** Mini-FT ( $K_{Ic}$ ) of the experimental TCS cements (open circles) and Biodentine (Septodont; closed circles). The mini-FT did not significantly increase ( $p = 0.52$ ) with a higher concentration of  $ZrO_2$  (estimated effect of 0.0005/wt.%). Overall, the experimental TCSs revealed a slightly higher but not significantly different ( $p = 0.31$ ) mean  $K_{Ic}$  of  $0.45 \pm 0.11$  MPa  $m^{1/2}$  vs.  $0.36 \pm 0.06$  MPa  $m^{1/2}$  recorded for Biodentine (Septodont)

experimental cements TCS 30 and TCS 50 (Fig. 3e, i, h, l). In contrast,  $ZrO_2$  particles were less homogeneously dispersed in the Biodentine matrix, in which clearly some highly concentrated Zr islands could be detected (Fig. 3m, p, marked with #). Back-scattered electron (BSE) image of Biodentine showed particles with a thin rim, representing a light-gray hydration zone around the dark-gray un-reacted core (Fig. 3m, marked with ☆). Compared with the un-reacted core, the hydration zone was less rich in Ca (Fig. 3n) and Si (Fig. 3o), while the core was free of Zr (Fig. 3p). Similarly reacted hydration zones were observed in all experimental TCS-cement matrices (Fig. 3a–l, marked with \*).

**Cytotoxicity**

The metabolic activities of HPFs after exposure to the 24-, 48-, and 72-h cement extracts are summarized in Fig. 4. The 24-h undiluted extracts of all tested cements were cytotoxic, while TCS 50 exhibited the least cytotoxicity (Fig. 4a).  $ED_{50}$  of the 24-h undiluted extracts could be ranked in decreasing order as follows: TCS 50 > TCS 0 > Dycal > TheraCal LC > Biodentine > Alganol (Table. 4). Accordingly, the level of cytotoxicity of the 24 h undiluted cement extracts followed a reverse order. The 48- and 72-h undiluted experimental cements TCS 0 and TCS 50 appeared non-cytotoxic (Fig. 4b, c), and consequently the related  $ED_{50}$  could not be generated (Table 4). The cytotoxicity of 48- and 72-h undiluted Biodentine extract was lower in comparison to that of the

24-h undiluted cement extract; however, it was still more cytotoxic than the experimental TCS cements (Fig. 4). The 48- and 72-h undiluted extracts of the light-curing resin-based cement TheraCal LC were found to be cytotoxic, although the cytotoxicity decreased for the 72-h undiluted extract compared with the 48-h undiluted extract (Table 4). The undiluted Alganol extract, employed as negative control, demonstrated the highest cytotoxicity among all cements tested (Table 4).

**Discussion**

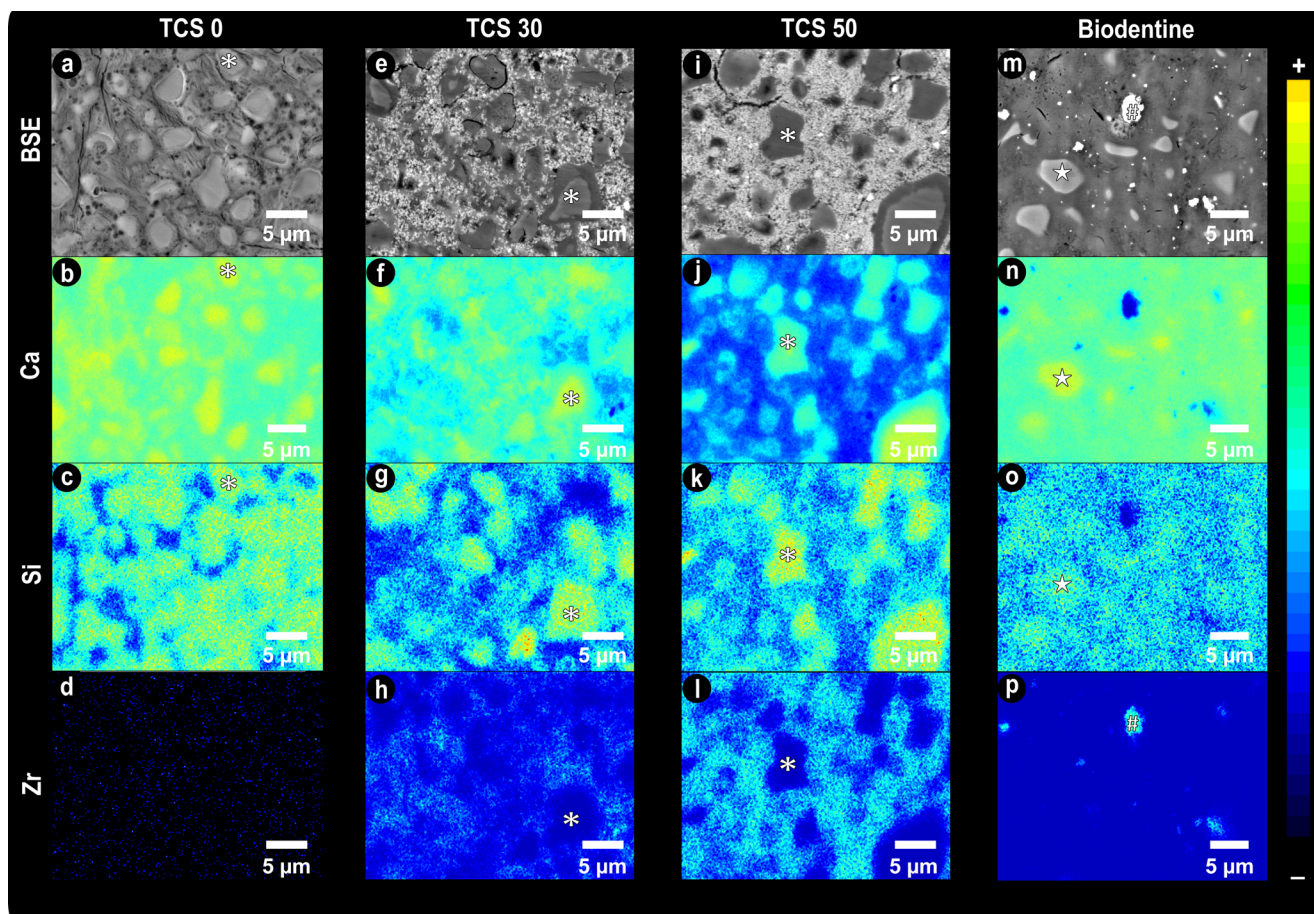
The hypothesis that the addition of  $ZrO_2$  does not affect strength, bioactivity, and biocompatibility of TCS cements was partially rejected. The mini-FT was not statistically altered by adding  $ZrO_2$ . Regarding Ca release, all experimental TCS cements released less Ca at 1 day as compared to Biodentine, and in particular the Ca release from TCS 50 was prolonged at a higher rate than that of Biodentine at 1 week and 1 month. Adding  $ZrO_2$  appeared an effective means to control Ca release with a slower early and longer higher rate. No effect of adding  $ZrO_2$  on the chemical hydration characteristics of TCS cements could be documented, this basically because  $ZrO_2$  was not involved in the hydration reaction of TCS. Finally, adding  $ZrO_2$  reduced the cytotoxicity of TCS cements, this as compared to four control Ca-releasing materials and with TCS 50 being least cytotoxic.

To assess mechanical strength of hydraulic calcium silicate cements, most commonly compressive strength is measured [19, 23, 24]. However, (ultimate) compressive strength is the capacity of a material to withstand compressive loads until complete failure; it thus also depends on the number and size of intrinsic flaws, e.g., pores, which hydraulic calcium silicate cements as powder-liquid mixtures certainly incorporate [35]. We therefore opted to measure fracture toughness, as it describes an intrinsic material property to resist crack propagation and eventually fracture. We employed a single-gradient notch beam (SGNB) FT method [36, 37]; compared with other FT methods, specimen preparation of SGNB FT is simpler and easier to miniaturize [37]. Special care was taken to prepare defect-poor specimen bars by ultra-sonically filling the silicon molds with cement. Upon complete setting for 1 week, a single notch was prepared as standardized defect according to an innovative method introduced by Pongprueksa et al. [37]; the notch area was carefully checked for absence of

**Table 3** Calcium ion release from cements at 1 day, 1 week, and 1 month

Ca release (ppm)	TCS 0	TCS 30	TCS 50	Biodentine	Dycal
1 day	133.554 <sup>e,f,g</sup>	173.570 <sup>c,d,e</sup>	156.899 <sup>d,e,f</sup>	225.013 <sup>b,c,d</sup>	17.366 <sup>h</sup>
1 week	258.323 <sup>b</sup>	382.574 <sup>a</sup>	357.166 <sup>a</sup>	241.182 <sup>b,c</sup>	25.056 <sup>h</sup>
1 month	124.028 <sup>e,f,g</sup>	58.673 <sup>g,h</sup>	142.183 <sup>e,f</sup>	84.486 <sup>f,g,h</sup>	21.994 <sup>h</sup>

Values with the same superscript letter are not significantly different (Tukey multiple comparisons test,  $p > 0.05$ )



**Fig. 3** Back-scattered electron (BSE) SEM photomicrographs of the argon-ion-beam polished cement cross-sections along with EPMA chemical elemental mappings of calcium (Ca), silicon (Si), and zirconium (Zr), this for the experimental TCS cements “TCS 0” (a–d), “TCS 30” (e–h), and “TCS 50” (i–l), as well as for Biodentine (Septodont; m–p). The shallow dark-gray rim (marked with \*) at the particle boundary represents a zone of hydration surrounding the

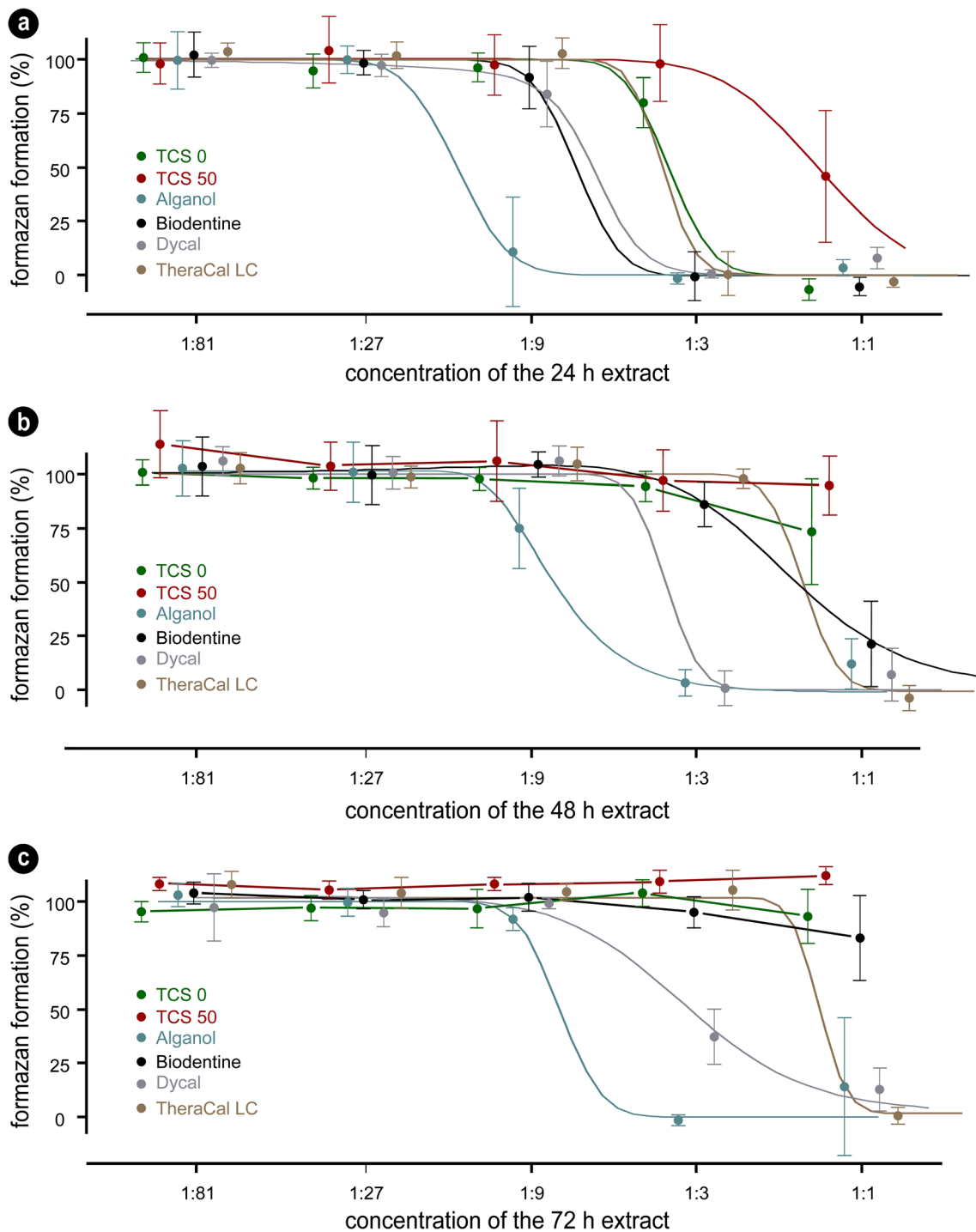
particle core. The hydration zones have lower concentrations of Ca and Si than the un-reacted particle cores. No Zr was found within the particle core. Similarly reacted hydration zones were observed in Biodentine (marked with ☆).  $ZrO_2$  particles with a smaller particle size were more evenly distributed within the matrix of the experimental cements TCS 30 (h) and TCS 50 (l) compared to those  $ZrO_2$  particles (marked with #) in Biodentine (Septodont) matrix

voids and other defects, upon which the notched specimen bars were loaded until failure in a four-point bending test to determine the mini-FT, thereby representing a reliable test to assess the cements’ mechanical performance [35]. Despite the addition of  $ZrO_2$  and irrespective of its concentration, the mechanical properties in terms of FT were not affected. The relative small size of the  $ZrO_2$  particles (diameter of  $\pm 200$  nm) may have resulted in a superior filler packing with the  $ZrO_2$  particles filling up the spaces in between the TCS particles, thereby having resulted a refined microstructure. The fact that adding  $ZrO_2$  did not affect cement strength correlated well with previous studies in which the addition of  $ZrO_2$  to Portland cement did not harm its compressive strength [23, 31].

In terms of bioactivity, adding  $ZrO_2$  to TCS enabled to control Ca release with a lower early but a prolonged higher release, this as compared to the Ca release of the commercial cement Biodentine. This controlled Ca release may

speculatively be ascribed to the refined microstructure of the experimental TCS cements, potentially preventing Ca to be initially resealed massively; the actual working mechanism behind this controlled Ca release, however, remains to be determined. Overall, the experimental TCS 50 cement demonstrated a continuous and constant release of Ca during a period of 1 month. Our Ca release assessment appeared valid, as the Ca release data for Dycal were in agreement with a previous study that showed a stable and low Ca release at neutral pH [38].

The distribution of the cement ingredients and the hydration characteristics of the different experimental TCS cements were evaluated by high-resolution Feg-SEM/EPMA chemical-element mapping and corresponding back-scattered electron (BSE) imaging of the ion-beam polished cement cross-sections. The  $ZrO_2$  particles were smaller and more evenly distributed within the experimental cements, as compared to the structure of Biodentine. All the experimental



**Fig. 4** Evaluation of relative formazan formation of human pulp fibroblasts (HPFs) after exposure to the 24 h (a), 48 h (b), and 72 h (c) undiluted (1:1) and diluted (1:3, 1:9, 1:27, 1:81) cement extracts using an XTT assay. **a** The 24-h undiluted (1:1) extracts of all cements tested appeared cytotoxic. The ranking of the 24-h cement extracts in order of highest to lowest cytotoxicity is Alganol (Kemdent) > Biodentine (Septodont) > Dycal (Dentsply) > TheraCal LC (Bisco) > TCS 0 > TCS 50. **b** The 48-h undiluted (1:1) TCS 50 and TCS 0 cement extracts were

relatively non-cytotoxic. The ranking of the 48-h cement extracts in order of highest to lowest cytotoxicity changed slightly: Alganol (Kemdent) > Dycal (Dentsply) > TheraCal LC (Bisco) > Biodentine (Septodont) > TCS 0 > TCS 50. **c** The 72-h undiluted (1:1) TCS 50, TCS 0, and Biodentine (Septodont) cement extracts were relatively non-cytotoxic. The ranking of the 72-h cement extracts for cytotoxicity was the same as for the 48-h cement extracts in **b**

and commercial cements investigated exhibited similar hydration features, basically containing particles with an un-reacted

core surrounded by a hydration zone, in which no precipitation of zirconium could be detected; these structural hydration



**Table 4** Median effective dose (ED<sub>50</sub>) of the undiluted cement extracts at 24, 48, and 72 h

Cement	TCS 0			TCS 50			Alganol			Biodentine			Dycal			TheraCal LC		
	24 h	48 h	72 h	24 h	48 h	72 h	24 h	48 h	72 h	24 h	48 h	72 h	24 h	48 h	72 h	24 h	48 h	72 h
ED <sub>50</sub>	0.366	NA <sup>a</sup>	NA	0.931	NA	NA	0.079	0.140	0.127	0.130	0.582	NA	0.131	0.178	0.265	0.173	0.411	0.456
ED <sub>50</sub> 2.5 %	0.276	NA	NA	0.795	NA	NA	-0.023	0.124	0.079	0.073	0.495	NA	0.013	-0.085	0.233	-0.692	0.241	0.063
ED <sub>50</sub> 97.5 %	0.456	NA	NA	1.068	NA	NA	0.180	0.156	0.176	0.188	0.670	NA	0.249	0.442	0.296	1.039	0.580	0.850

<sup>a</sup> NA not available as the experimental cement is no longer cytotoxic

effects were also identified in previous research [28]. Hence, besides not affecting mechanical properties, adding ZrO<sub>2</sub> to TCS appeared also not to participate in/interfere with the hydration reaction of TCS.

Finally, the effect of adding ZrO<sub>2</sub> to TCS was evaluated with respect to biocompatibility by subjecting the cements to a cytotoxicity test that involved the assessment of cell viability of HPFs after being exposed to extracts of cement blocks. The results from the mini-FT and Ca release tests revealed that TCS 50 is the most suitable cement formulation. Therefore, only TCS 0 and TCS 50 were selected for the cytotoxicity test; TCS 50 contains the largest amount of ZrO<sub>2</sub>, and TCS 0 contains pure TCS. It is therefore assumed that the other experimental cements would behave regarding biocompatibility somewhere in between that of TCS 50 and TCS 0. A series of diluted extracts were prepared to observe a possible dose-response relationship. HPFs were selected as test cells based on the clinical indication of using TCS cements for pulp capping. XTT assay showed that the viability of HPFs exposed to the cement extracts was highly dependent on the dilution rate. All the undiluted cement extracts exhibited initial (24 h) cytotoxicity, which decreased with a longer extraction time. Among all cements, TCS 50 appeared least cytotoxic regardless of the extraction time. Our results confirmed previous findings that showed that ZrO<sub>2</sub> improved the *in vitro* biocompatibility of TCS cements [32, 34]. The difference in cytotoxicity found among the cements investigated in this study may be due to the specific chemical composition as well as solubility of each individual cement. TCS 0, which consists of pure TCS, was more biocompatible than Biodentine, indicating that other components contained in Biodentine, like calcium carbonate or iron oxide, may affect cell vitality of HPFs. A previous study using ICP-MS revealed a high level of iron release (711.7 ± 267.9 g/L) from Biodentine upon 7 days of storage in distilled water [19]. Moreover, traces of arsenic, lead, and chromium were detected in elutes from Biodentine [39]. The biocompatibility of the light-curing resin-based calcium silicate cement TheraCal LC was sparsely reported on in literature. Only one study reported that TheraCal LC presented with a low cytocompatibility in direct contact with rat odontoblast-like cells (MDPC-23), this as compared to Biodentine [40]. In that study, the 24-h undiluted TheraCal

LC extracts were slightly less cytotoxic than the 24-h undiluted Biodentine. This difference may be due to the different evaluation method used; we investigated cement extracts versus cement blocks in the former study [40]. The cytotoxicity of resin-based materials is commonly attributed to leaching out of unpolymerized monomers due to incomplete polymerization [41, 42]. Polyethylene glycol dimethacrylate (PEG-DMA) is the organic component present in TheraCal LC. Calcium hydroxide has traditionally been the standard pulp-capping material [43, 44]. Its initial cytotoxic effect has been shown previously [40, 45–47] and should be ascribed to its high alkalinity (pH ≈ 12). Along with calcium hydroxide's disinfection potential, the simultaneous induction of necrosis at the superficial pulp is thought to initiate an inflammatory response that, in the absence of bacteria, will heal by forming a “dentin bridge.”

## Conclusion

In view of the above, ZrO<sub>2</sub> can be added in relatively large quantity to TCS without affecting the mechanical properties in terms of mini-FT; Ca release was reduced initially, to reach a higher and longer release rate thereafter; adding ZrO<sub>2</sub> made TCS cements more biocompatible. It is concluded that adding ZrO<sub>2</sub> to TCS improved some key properties in function of its indication as dentin-replacement and/or pulp-repair material.

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## Compliance with ethical standards

**Conflict of interest** Author Xin Li declares that she has no conflict of interest. Author Kumiko Yoshihara declares that she has no conflict of interest. Author Jan De Munck declares that he has no conflict of interest. Author Stevan Cokic declares that he has no conflict of interest. Author Pong Pongruksa declares that he has no conflict of interest. Author Eveline Putzeys declares that she has no conflict of interest. Author Mariano Pedano declares that he has no conflict of interest. Author Zhi Chen declares that he has no conflict of interest. Author Kirsten Van Landuyt declares that she has no conflict of interest. Author Bart Van Meerbeek declares that he has no conflict of interest.

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**Ethical approval** This article does not contain any studies with human participants or animals performed by any of the authors.

**Informed consent** For this type of study, formal consent is not required.

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