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Constitutive modeling and algorithmic implementation of a plasticity-like model for trabecular bone structures

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Abstract Trabecular bone is a highly porous orthotropic cellular solid material present inside human bones such as the femur (hip bone) and vertebra (spine). In this study, an infinitesimal plasticity-like model with isotropic/kinematic hard-

ening is developed to describe yielding of trabecular bone at the continuum level. One of the unique features of this formulation is the development of the plasticity-like model in strain space for a yield envelope expressed in terms of principal strains having asymmetric yield behavior. An implicit return-mapping approach is adopted to obtain a symmetric algorithmic tangent modulus and a step-by-step procedure of algorithmic implementation is derived. To investigate the performance of this approach in a full-scale finite element simulation, the model is implemented in a non-linear finite element analysis program and several test problems including the simulation of loading of the human femur structures are analyzed. The results show good agreement with the experimental data.

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1 Introduction

Osteoporotic hip fractures are the most common serious injury for the elderly [1, 2], associated with substantial socio-economic consequences. According to the American Academy of Orthopaedic Surgeons, over 350,000 hip fractures occur every year with an estimated cost of over \$10 billion; almost one in four hip fracture patients die within one year, and this problem is expected to worsen as the size of the elderly population increases. The finite element technique has been widely used to elucidate the mechanisms of these fractures through the study of the mechanical behavior of the proximal femur under non-habitual (i.e., traumatic) loading conditions, such as those caused by sideways falls [3–5].

Trabecular bone (Fig. 1), a highly porous biological tissue with an open-celled cellular structure, is a major load-carrying component in the femur as well as in other whole bones. It is essentially an orthotropic material which exhibits tension-compression asymmetry in the yield strength [6]. It has been experimentally established that the yield stress of

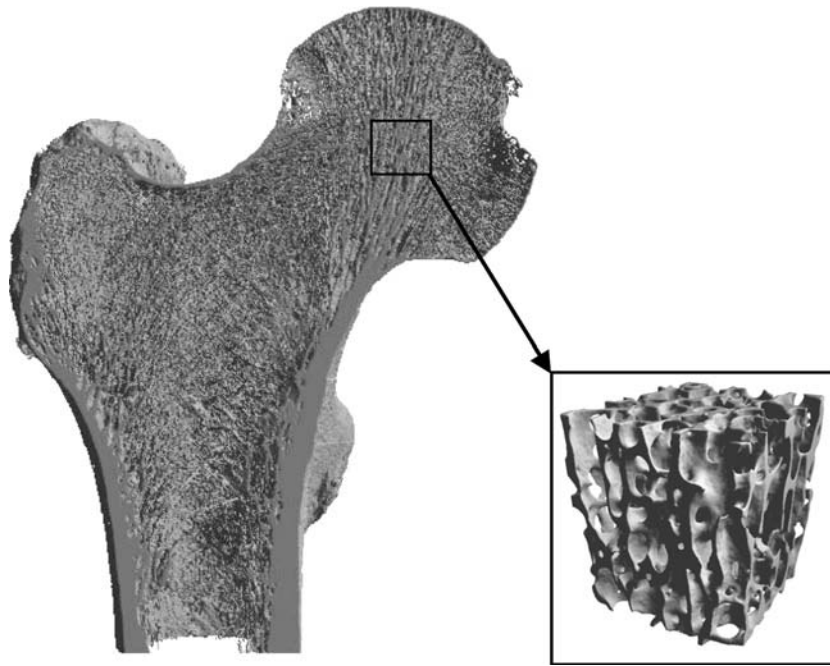


Fig. 1 Three-dimensional rendering of a human proximal femur. The *through section cut* shows the variation in the density and the orientation of trabecular bone inside the femur. This image is obtained using micro-computed tomography (micro-CT) at $89 \times 89 \times 93 \mu$ resolution (SCANCO, Bassersdorf, Switzerland). The *inset* shows a representative 5 mm cubic trabecular bone specimen from the femoral head

trabecular bone is highly heterogeneous within and across different anatomic sites, while the yield strain is uniform within any given anatomic site [7, 8]. To properly predict the failure behavior of the femur and other whole bones using continuum finite element models, it is necessary to accurately capture the nonlinear mechanical behavior of trabecular bone at the continuum level [5, 9–12]. Such modeling may also be helpful in orthopaedic implant design, because implant loosening is associated with multiaxial interface stresses on the trabecular bone [13].

In the biomechanics literature, it is common to use a plasticity-like yield function to model the envelope of bone failure, although it is clear that the relevant inelastic process is different from that of the classical metal plasticity. While a number of approaches have been adopted for the constitutive modeling of the trabecular bone within the whole bone finite element models [3, 5, 14], to date no multiaxial constitutive model has been implemented that captures the anisotropy, asymmetry, and heterogeneity of trabecular bone strength. Several studies conducted on failure load prediction of the femur using nonlinear models have employed stress-based failure theories assuming isotropic behavior for the trabecular bone [3, 5, 14], whereas trabecular bone is an orthotropic material [15–17]. Although trabecular bone is stronger in compression than in tension [18, 19], some of these studies have used the von Mises failure criterion that assumes equal compressive and tensile strength [14, 20]. Further, it has been demonstrated that the use of the von Mises criterion for the trabecular bone yielding results in overestimation of stresses when high shear stresses are present [10, 21]. Recently, a

multiaxial yield criterion (Fig. 2) has been developed for the femoral neck trabecular bone [22]. This “Modified Super-Ellipsoid” (MSE) criterion is formulated in terms of principal strains because it is found experimentally that the strain-space formulation eliminates the heterogeneity effect from the failure behavior of trabecular bone within an anatomic site [7, 8], thus can be applied to specimens of different porosity. The criterion also exploits the fact that the trabecular bone yield strains are uniform in all three principal material directions. While this yield envelope can be used in a plasticity-like formulation of trabecular bone at the continuum level, limited work has been done in the area of strain-based plasticity-like formulations for a yield envelope expressed in the principal space. Furthermore, trabecular bone is known to fail in an uncoupled fashion under multiaxial loads, i.e., despite yielding in one direction, near intact properties are preserved in other directions [22, 23].

The overall goal of this study is to develop a rate-independent infinitesimal plasticity-like model for the yielding of trabecular bone in strain space incorporating anisotropy and the MSE multiaxial yield criterion. Specifically, the objectives are to: (1) formulate a rate-independent plasticity-like model in strain space using the MSE multiaxial yield envelope with general parameters and including both kinematic and isotropic hardening; (2) implement an integration algorithm using an implicit return mapping scheme in a finite element analysis program; and (3) use the plasticity-like material model for whole bone analyses of the proximal femur and evaluate its performance with respect to the experimental results. The proposed constitutive model is novel

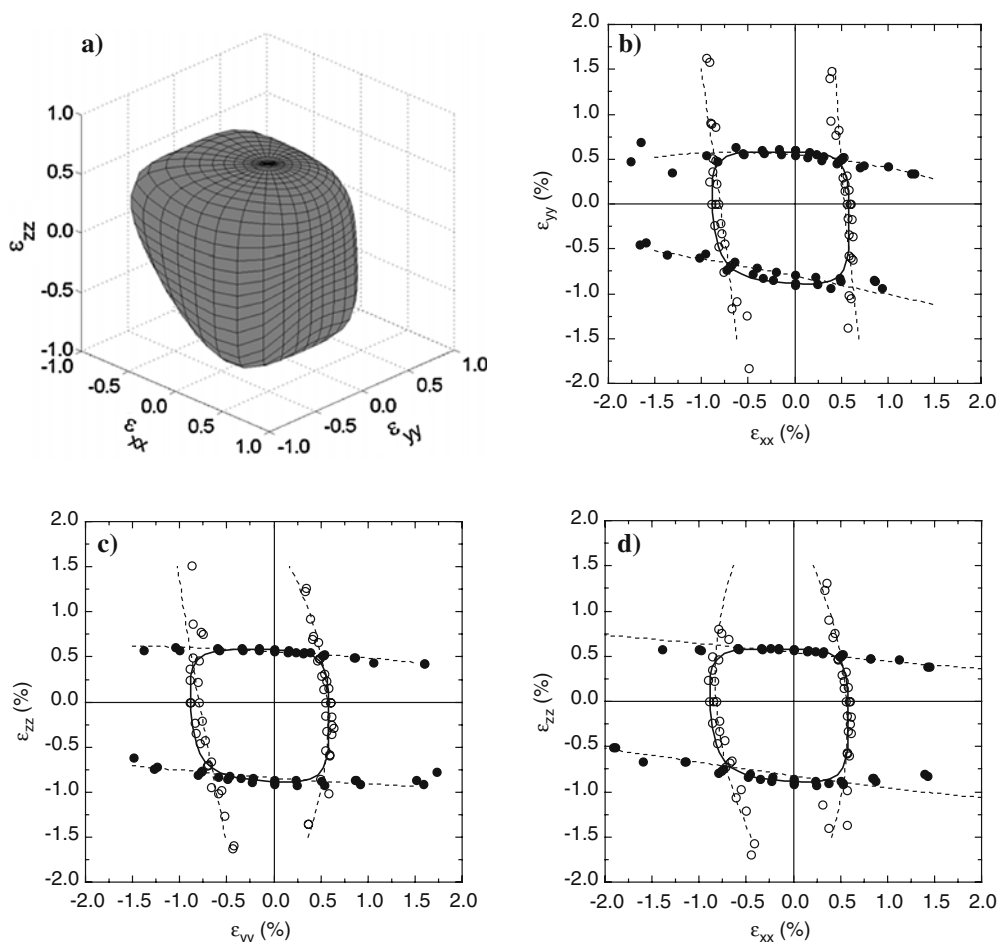


Fig. 2 **a** Complete 3D modified super-ellipsoid yield surface for the femoral trabecular bone based on Eq. (1) and its cross-section in three mutually perpendicular strain planes: **b** ε_{xx} - ε_{yy} , **c** ε_{yy} - ε_{zz} , and **d** ε_{xx} - ε_{zz} . In **(b)**, **(c)** and **(d)** *circles* indicate the FE computed yield data; *solid symbols* indicate yielding along the *vertical axis*; *empty symbols* indicate yielding along the *horizontal axis*. *Dashed lines* shown are quadratic fits to the yield points along *each axis*. This figure is based on a combination of figures from [22], with permission from ASME

in its formulation of infinitesimal strain-based plasticity for a yield envelope expressed in terms of principal strains that departs from the conventional von Mises model (J2-plasticity).

2 Theory

Trabecular bone (Fig. 1) is a heterogeneous material that has a plate and rod-like cellular solid-type structure with a tensile strength lower than its compressive strength. In a previous study [22], a yield envelope (Fig. 2) was obtained in strain space for 5 mm cube specimens of human femoral trabecular bone using high-resolution finite element analyses as a surrogate for experiments. Further, a super-ellipsoid equation [24] was modified and fit to this yield envelope to mathematically express it in terms of certain experimentally measured parameters in the following form [22]

$$g(\varepsilon_1, \varepsilon_2, \varepsilon_3) = \sum_{i=1}^3 \left| \frac{\varepsilon_i - c}{r} \right|^{2/n} + \left| t \frac{\text{tr}(\boldsymbol{\varepsilon})}{3r} \right|^{2/n} - 1. \quad (1)$$

Here, ε_i ($i = 1, 2, 3$) are the principal values of the complete strain tensor $\boldsymbol{\varepsilon}$, r is the radius of the super-ellipsoid, c is the shift in the center coordinates with respect to the origin, n is a “squareness” parameter, and t is a “flattening” parameter. In this model, the radius and shift in center directly correspond to the yield strains and their asymmetry in tension and compression (Fig. 2). This yield envelope captures the micromechanics of trabecular bone at the tissue-level and departs from the Mises-like ellipsoidal form. Further, it exhibits squareness at the corner of tension-shear quadrant (Fig. 2). It is observed from this envelope that the yield behavior of trabecular bone is isotropic when expressed in terms of principal strains with tension-compression yield asymmetry. This observation is supported by various experimental and computational studies [6, 22, 25, 26], which included the detailed tissue-level properties and trabeculae architecture to capture the exact mechanical behavior of trabecular bone at the microstructural level.

In this work, the infinitesimal plasticity-like model in strain space is first developed for a generalized yield envelope.

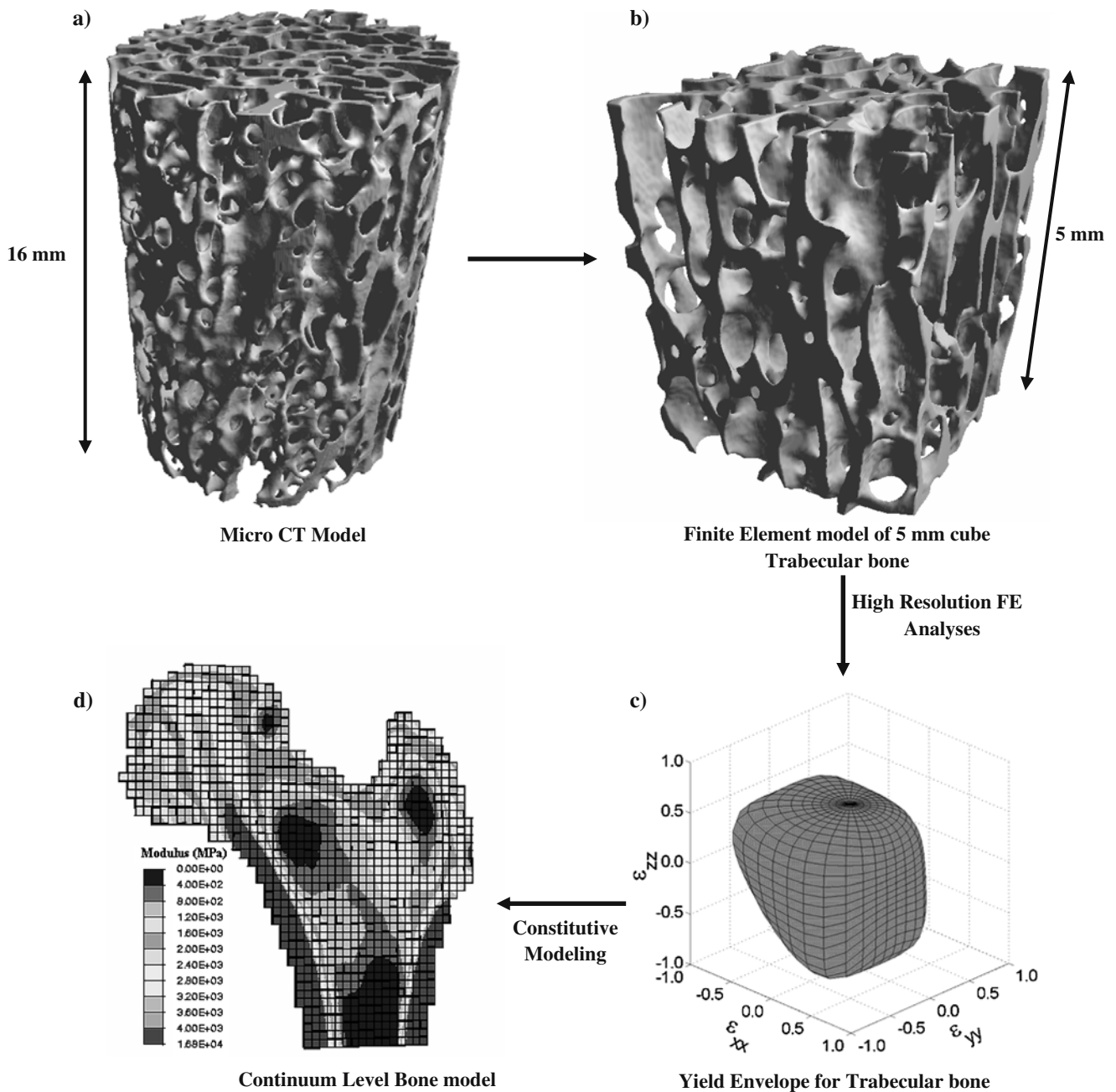


Fig. 3 A flowchart of the finite element modeling of the proximal femur: **a** trabecular bone cylindrical cores (8 mm diameter) obtained from the proximal femur and scanned using micro-CT (μ CT) scanner (spatial resolution: $22 \mu\text{m}$), **b** 5 mm cube finite element models generated from these cylinders (element size: $66 \times 66 \times 66 \mu\text{m}$), **c** yield envelope obtained from a series of finite element analyses and optimization using these trabecular bone cube specimens and, **d** continuum-level models of the whole bone (element size: $1.9335 \times 1.9335 \times 2.0 \text{ mm}$) developed and FE analyses performed in which the yield envelope of trabecular bone is included in the material constitutive model. A frontal cross-section of femur with primary elastic modulus distribution is shown here

Subsequently, a specific form of this model is derived for the yield envelope formulated in principal strain space based on Eq. (1), which may be used for whole bone analyses at the continuum level (Fig. 3). It is assumed that the yield envelope specified in Eq. (1) contains all the microstructural information of trabecular bone and the continuum-level model developed here would be employed for the purpose of nonlinear

analyses at the whole-bone-level. Since this is the first step toward the development of a complete constitutive model for the trabecular bone, damage in bone and effects of bone remodeling at the tissue-level are not included. In addition, post-yield behavior is modeled using hardening. While the latter does not capture the stress reduction that can occur in trabecular bone after the ultimate yield point, it is a commonly

used strategy in whole bone mechanics and is considered acceptable at this phase of the constitutive model development. Further, this study is motivated towards capturing the yield behavior and not the post-ultimate fracture behavior of trabecular bone.

In what follows, the notation that all the boldface lower-case letters represent second-order tensors and boldface upper-case letters represent fourth-order tensors is used.

It is observed in the experiments that the trabecular bone behaves nonlinearly under tensile/compressive loading [27] exhibiting minor increase in load carrying capacity after yield and then reduction in stress beyond the ultimate point. After the slight stress reduction trabecular bone continues to sustain appreciable load due to its cellular solid nature. To capture the negligible increase in load carrying capacity and to ensure completeness, hardening has been included in this model. Softening behavior has not been included at this juncture due to lack of detailed post-failure three-dimensional stress response of trabecular bone. Therefore, accounting for hardening effects a generalized yield surface in strain space can be written as

$$g = g(\boldsymbol{\varepsilon}^e, \boldsymbol{\alpha}, \beta), \quad (2)$$

where $\boldsymbol{\varepsilon}^e$ is the elastic strain tensor, and $\boldsymbol{\alpha}$ and β are strain-like kinematic and isotropic hardening variables, respectively. The constitutive equation for stress ($\boldsymbol{\sigma}$) assuming additive decomposition of strains into elastic and plastic (inelastic) parts, can be written using the Hooke's law as

$$\boldsymbol{\sigma} = \mathbf{C}^e : (\boldsymbol{\varepsilon} - \boldsymbol{\varepsilon}^p) = \mathbf{C}^e : \boldsymbol{\varepsilon}^e, \quad (3)$$

where \mathbf{C}^e is the fourth-order elasticity tensor and $\boldsymbol{\varepsilon}^e$, $\boldsymbol{\varepsilon}^p$ are the elastic and plastic strain tensors, respectively. Due to lack of any experimental evidence, the flow rule is assumed to be associative owing to its inherent mathematical simplicity and ease of implementation. Its general form in stress and strain space is written as

$$\dot{\boldsymbol{\varepsilon}}^p = \mu \frac{\partial f}{\partial \boldsymbol{\sigma}} = \mu \mathbf{D}^e : \frac{\partial g}{\partial \boldsymbol{\varepsilon}^e}, \quad (4)$$

where $\dot{\boldsymbol{\varepsilon}}^p$ is the plastic strain rate, $\mathbf{D}^e = (\mathbf{C}^e)^{-1}$, f is the yield surface in stress space, and μ is the plastic consistency parameter.

Similarly, the hardening rules can be written in the following form

$$\begin{aligned} \dot{\boldsymbol{\alpha}} &= \mu H_{\text{kin}} \mathbf{D}^e : \frac{\partial g}{\partial \boldsymbol{\alpha}}, \\ \dot{\beta} &= \mu H_{\text{iso}} D_{\text{norm}}^e \frac{\partial g}{\partial \beta}. \end{aligned} \quad (5)$$

Here, H_{kin} and H_{iso} are the kinematic and isotropic hardening parameters, respectively, and D_{norm}^e is a scalar multiplication factor introduced to maintain consistency with the units of μ . The associativity of the hardening rules ensures symmetry of the algorithmic tangent moduli [28].

3 Return-mapping algorithm

The return-mapping algorithm is widely used to numerically integrate the differential equations in rate independent plasticity. This approach is well-established and, under certain conditions, ensures a stable and accurate integration of the constitutive equation [28]. An implicit return-mapping approach is adopted which preserves the symmetry of the algorithmic tangent moduli (Appendix A). In the return-mapping algorithm, the state variable values ($\boldsymbol{\varepsilon}^p$, $\boldsymbol{\alpha}$, β) from the previous converged step are used in determining those values for the current iteration. A step-by-step procedure is derived as shown below for the proposed plasticity-like model.

The yield envelope for the step $n + 1$ can be written as

$$g_{n+1} = g(\boldsymbol{\varepsilon}_{n+1}^e, \boldsymbol{\alpha}_{n+1}, \beta_{n+1}). \quad (6)$$

Equations (4) and (5) can be cast in residual form as

$$\begin{aligned} \mathbf{R}_{n+1} &= -\boldsymbol{\varepsilon}_{n+1}^p + \boldsymbol{\varepsilon}_n^p + \Delta\mu_{n+1} \mathbf{D}^e : \partial_{\boldsymbol{\varepsilon}^e} g(\boldsymbol{\varepsilon}_{n+1}^e, \boldsymbol{\alpha}_{n+1}, \beta_{n+1}) \\ &= \mathbf{0}, \\ \mathbf{S}_{n+1} &= -\boldsymbol{\alpha}_{n+1} + \boldsymbol{\alpha}_n + \Delta\mu_{n+1} H_{\text{kin}} \mathbf{D}^e : \partial_{\boldsymbol{\alpha}} g(\boldsymbol{\varepsilon}_{n+1}^e, \boldsymbol{\alpha}_{n+1}, \beta_{n+1}) \\ &= \mathbf{0}, \\ T_{n+1} &= -\beta_{n+1} + \beta_n + \Delta\mu_{n+1} H_{\text{iso}} D_{\text{norm}}^e \partial_{\beta} g(\boldsymbol{\varepsilon}_{n+1}^e, \boldsymbol{\alpha}_{n+1}, \beta_{n+1}) \\ &= 0, \end{aligned} \quad (7)$$

where \mathbf{R}_{n+1} , \mathbf{S}_{n+1} and T_{n+1} are residuals of plastic strain, kinematic hardening and isotropic hardening variables, respectively, at the step $n + 1$. Also, during the plastic corrector phase the total strain is fixed therefore,

$$\Delta\boldsymbol{\varepsilon}_{n+1}^e = -\Delta\boldsymbol{\varepsilon}_{n+1}^p. \quad (8)$$

Upon linearizing and combining Eqs. (6) to (8) for k th iteration, it is readily found that

$$\begin{aligned} \mathbf{R}^{(k)} - \Delta\boldsymbol{\varepsilon}^{p(k)} + \Delta\mu^{(k)} \mathbf{D}^e : (-\partial_{\boldsymbol{\varepsilon}^e} g^{(k)} : \Delta\boldsymbol{\varepsilon}^{p(k)} + \partial_{\boldsymbol{\varepsilon}^e} \boldsymbol{\alpha} g^{(k)} : \\ \Delta\boldsymbol{\alpha}^{(k)} + \partial_{\boldsymbol{\varepsilon}^e \beta} g^{(k)} \Delta\beta^{(k)}) + \delta\mu^{(k)} \mathbf{D}^e : \partial_{\boldsymbol{\varepsilon}^e} g^{(k)} = \mathbf{0}, \\ \mathbf{S}^{(k)} - \Delta\boldsymbol{\alpha}^{(k)} + \Delta\mu^{(k)} H_{\text{kin}} \mathbf{D}^e : (-\partial_{\boldsymbol{\alpha}} g^{(k)} : \Delta\boldsymbol{\varepsilon}^{p(k)} + \partial_{\boldsymbol{\alpha}} \boldsymbol{\alpha} g^{(k)} : \\ \Delta\boldsymbol{\alpha}^{(k)} + \partial_{\boldsymbol{\alpha} \beta} g^{(k)} \Delta\beta^{(k)}) + \delta\mu^{(k)} H_{\text{kin}} \mathbf{D}^e : \partial_{\boldsymbol{\alpha}} g^{(k)} = \mathbf{0}, \\ T^{(k)} - \Delta\beta^{(k)} + \Delta\mu^{(k)} H_{\text{iso}} D_{\text{norm}}^e (-\partial_{\beta} g^{(k)} : \Delta\boldsymbol{\varepsilon}^{p(k)} + \partial_{\beta} \boldsymbol{\alpha} g^{(k)} : \\ \Delta\boldsymbol{\alpha}^{(k)} + \partial_{\beta \beta} g^{(k)} \Delta\beta^{(k)}) + \delta\mu^{(k)} H_{\text{iso}} D_{\text{norm}}^e \partial_{\beta} g^{(k)} = 0, \\ g^{(k)} - \partial_{\boldsymbol{\varepsilon}^e} g^{(k)} : \Delta\boldsymbol{\varepsilon}^{p(k)} + \partial_{\boldsymbol{\alpha}} g^{(k)} : \\ \Delta\boldsymbol{\alpha}^{(k)} + \partial_{\beta} g^{(k)} \Delta\beta^{(k)} = 0, \end{aligned} \quad (9)$$

where, for notational simplicity, the subscripts $n + 1$ are dropped. Solving for the unknowns, $\delta\mu^{(k)}$, $\Delta\boldsymbol{\varepsilon}^{p(k)}$, $\Delta\boldsymbol{\alpha}^{(k)}$ and $\Delta\beta^{(k)}$ leads to

$$\begin{aligned} \delta\mu^{(k)} &= \frac{-g^{(k)} - \{\partial \mathbf{g}\}^{(k)} \mathbf{A}^{(k)} \{\tilde{\mathbf{a}}\}^{(k)}}{\{\partial \mathbf{g}\}^{(k)} \mathbf{A}^{(k)} \{\tilde{\mathbf{r}}\}^{(k)}}, \\ \begin{pmatrix} \Delta\boldsymbol{\varepsilon}^{p(k)} \\ \Delta\boldsymbol{\alpha}^{(k)} \\ \Delta\beta^{(k)} \end{pmatrix} &= \mathbf{A}^{(k)} \{\tilde{\mathbf{a}}\}^{(k)} + \delta\mu^{(k)} \mathbf{A}^{(k)} \{\tilde{\mathbf{r}}\}^{(k)}, \end{aligned} \quad (10)$$

1. Initialization:

$$k = 0 ; \boldsymbol{\varepsilon}^{p(0)} = \boldsymbol{\varepsilon}_n^p ; \boldsymbol{\alpha}^{(0)} = \boldsymbol{\alpha}_n ; \boldsymbol{\beta}^{(0)} = \boldsymbol{\beta}_n ; \Delta\mu^{(0)} = 0$$

2. Check yield condition and convergence at k^{th} iteration:

$$\mathbf{g}^{(k)} = \mathbf{g}(\boldsymbol{\varepsilon}^{e(k)}, \boldsymbol{\alpha}^{(k)}, \boldsymbol{\beta}^{(k)}) ; \{\tilde{\mathbf{a}}\}^{(k)} = \begin{pmatrix} \mathbf{R}^{(k)} \\ \mathbf{S}^{(k)} \\ \mathbf{T}^{(k)} \end{pmatrix}$$

Condition: If $\mathbf{g}^{(k)} < \text{tol}_1$ and $\|\{\tilde{\mathbf{a}}\}^{(k)}\| < \text{tol}_2$ then converged.

Else,

3. Compute increment in plasticity parameter:

$$\left[\mathbf{A}^{(k)} \right]^{-1}, \{\tilde{\mathbf{r}}\}^{(k)}, \{\partial \mathbf{g}\}^{(k)} \text{ and}$$

$$\delta\mu^{(k)} = \frac{-\mathbf{g}^{(k)} - \{\partial \mathbf{g}\}^{(k)} \mathbf{A}^{(k)} \{\tilde{\mathbf{a}}\}^{(k)}}{\{\partial \mathbf{g}\}^{(k)} \mathbf{A}^{(k)} \{\tilde{\mathbf{r}}\}^{(k)}}$$

4. Obtain increment in plastic strain and internal variables:

$$\begin{pmatrix} \Delta \boldsymbol{\varepsilon}^{p(k)} \\ \Delta \boldsymbol{\alpha}^{(k)} \\ \Delta \boldsymbol{\beta}^{(k)} \end{pmatrix} = \mathbf{A}^{(k)} \{\tilde{\mathbf{a}}\}^{(k)} + \delta\mu^{(k)} \mathbf{A}^{(k)} \{\tilde{\mathbf{r}}\}^{(k)}$$

5. Update plastic strain and internal variables:

$$\begin{aligned} \boldsymbol{\varepsilon}^{p(k+1)} &= \boldsymbol{\varepsilon}^{p(k)} + \Delta \boldsymbol{\varepsilon}^{p(k)} \\ \boldsymbol{\alpha}^{(k+1)} &= \boldsymbol{\alpha}^{(k)} + \Delta \boldsymbol{\alpha}^{(k)} \\ \boldsymbol{\beta}^{(k+1)} &= \boldsymbol{\beta}^{(k)} + \Delta \boldsymbol{\beta}^{(k)} \\ \Delta\mu^{(k+1)} &= \Delta\mu^{(k)} + \delta\mu^{(k)} \end{aligned}$$

$k \leftarrow k + 1$, goto 2.

Box. 1 Steps of the implicit return-mapping algorithm

where

$$\left[\mathbf{A}^{(k)} \right]^{-1} = \begin{bmatrix} \mathbf{I} + \Delta\mu^{(k)} \mathbf{D}^e \partial_{\boldsymbol{\varepsilon}^e} \mathbf{g}^{(k)} & -\Delta\mu^{(k)} \mathbf{D}^e \partial_{\boldsymbol{\varepsilon}^e} \boldsymbol{\alpha} \mathbf{g}^{(k)} & -\Delta\mu^{(k)} \mathbf{D}^e \partial_{\boldsymbol{\varepsilon}^e} \boldsymbol{\beta} \mathbf{g}^{(k)} \\ \Delta\mu^{(k)} H_{\text{kin}} \mathbf{D}^e \partial_{\boldsymbol{\alpha}} \mathbf{g}^{(k)} & \mathbf{I} - \Delta\mu^{(k)} H_{\text{kin}} \mathbf{D}^e \partial_{\boldsymbol{\alpha} \boldsymbol{\alpha}} \mathbf{g}^{(k)} & -\Delta\mu^{(k)} H_{\text{kin}} \mathbf{D}^e \partial_{\boldsymbol{\alpha} \boldsymbol{\beta}} \mathbf{g}^{(k)} \\ \Delta\mu^{(k)} H_{\text{iso}} D_{\text{norm}}^e \partial_{\boldsymbol{\beta}} \mathbf{g}^{(k)} & -\Delta\mu^{(k)} H_{\text{iso}} D_{\text{norm}}^e \partial_{\boldsymbol{\beta} \boldsymbol{\alpha}} \mathbf{g}^{(k)} & 1 - \Delta\mu^{(k)} H_{\text{iso}} D_{\text{norm}}^e \partial_{\boldsymbol{\beta} \boldsymbol{\beta}} \mathbf{g}^{(k)} \end{bmatrix}$$

and

$$\{\tilde{\mathbf{a}}\}^{(k)} = \begin{pmatrix} \mathbf{R}^{(k)} \\ \mathbf{S}^{(k)} \\ \mathbf{T}^{(k)} \end{pmatrix}, \{\tilde{\mathbf{r}}\}^{(k)} = \begin{pmatrix} \mathbf{D}^e : \partial_{\boldsymbol{\varepsilon}^e} \mathbf{g}^{(k)} \\ H_{\text{kin}} \mathbf{D}^e : \partial_{\boldsymbol{\alpha}} \mathbf{g}^{(k)} \\ H_{\text{iso}} D_{\text{norm}}^e \partial_{\boldsymbol{\beta}} \mathbf{g}^{(k)} \end{pmatrix},$$

$$\{\partial \mathbf{g}\}^{(k)} = [-\partial_{\boldsymbol{\varepsilon}^e} \mathbf{g}^{(k)} \quad \partial_{\boldsymbol{\alpha}} \mathbf{g}^{(k)} \quad \partial_{\boldsymbol{\beta}} \mathbf{g}^{(k)}].$$

Using these expressions the state variables in incremental form are updated as

$$\begin{aligned} \boldsymbol{\varepsilon}^{p(k+1)} &= \boldsymbol{\varepsilon}^{p(k)} + \Delta \boldsymbol{\varepsilon}^{p(k)}, \\ \boldsymbol{\alpha}^{(k+1)} &= \boldsymbol{\alpha}^{(k)} + \Delta \boldsymbol{\alpha}^{(k)}, \\ \boldsymbol{\beta}^{(k+1)} &= \boldsymbol{\beta}^{(k)} + \Delta \boldsymbol{\beta}^{(k)}, \\ \Delta\mu^{(k+1)} &= \Delta\mu^{(k)} + \delta\mu^{(k)}. \end{aligned} \quad (11)$$

The detailed steps of the return-mapping algorithm are presented in Box 1.

4 Application to trabecular bone modeling

The yield Eq. (1) including hardening parameters can be rewritten as

$$g = \sum_{i=1}^3 \sqrt{(\gamma_i - c)^2} + \sqrt{\left(\frac{\text{tr}(\boldsymbol{\gamma})}{3} \right)^2} - \sqrt{(r + \beta)^2}, \quad (12)$$

where $\boldsymbol{\gamma} = \boldsymbol{\varepsilon}^e - \boldsymbol{\alpha}$ and γ_i are the principal values of $\boldsymbol{\gamma}$. Using the above relations and the chain rule leads to

$$\begin{aligned}
\partial_{\mathbf{e}^e \mathbf{e}^e} g &= \partial_{\boldsymbol{\gamma} \boldsymbol{\gamma}} g, \\
\partial_{\mathbf{e}^e \boldsymbol{\alpha}} g &= \partial_{\boldsymbol{\alpha} \mathbf{e}^e} g = -\partial_{\boldsymbol{\gamma} \boldsymbol{\gamma}} g, \\
\partial_{\mathbf{e}^e \boldsymbol{\beta}} g &= \partial_{\boldsymbol{\beta} \mathbf{e}^e} g = \mathbf{0}, \\
\partial_{\boldsymbol{\alpha} \boldsymbol{\alpha}} g &= \partial_{\boldsymbol{\gamma} \boldsymbol{\gamma}} g, \\
\partial_{\boldsymbol{\alpha} \boldsymbol{\beta}} g &= \partial_{\boldsymbol{\beta} \boldsymbol{\alpha}} g = \mathbf{0}.
\end{aligned} \tag{13}$$

Taking into account Eqs. (10) and (13), \mathbf{A}^{-1} may be rewritten as

$$\mathbf{A}^{-1} = \begin{bmatrix} \mathbf{I} + \Delta\mu \mathbf{D}^e \partial_{\boldsymbol{\gamma} \boldsymbol{\gamma}} g & \Delta\mu \mathbf{D}^e \partial_{\boldsymbol{\gamma} \boldsymbol{\gamma}} g & \mathbf{0} \\ -\Delta\mu H_{\text{kin}} \mathbf{D}^e \partial_{\boldsymbol{\gamma} \boldsymbol{\gamma}} g & \mathbf{I} - \Delta\mu H_{\text{kin}} \mathbf{D}^e \partial_{\boldsymbol{\gamma} \boldsymbol{\gamma}} g & \mathbf{0} \\ \mathbf{0} & \mathbf{0} & 1 - \Delta\mu H_{\text{iso}} D_{\text{norm}}^e \partial_{\boldsymbol{\beta} \boldsymbol{\beta}} g \end{bmatrix}, \tag{14}$$

where the superscript (k) is omitted for brevity. Therefore, \mathbf{A} can be deduced from the above equation as

$$\mathbf{A} = \begin{bmatrix} \mathbf{A}_2 \mathbf{A}_1 & -\mathbf{A}_3 \mathbf{A}_1 & \mathbf{0} \\ \mathbf{A}_4 \mathbf{A}_1 & \mathbf{A}_5 \mathbf{A}_1 & \mathbf{0} \\ \mathbf{0} & \mathbf{0} & A_6 \end{bmatrix}, \tag{15}$$

where $\mathbf{A}_1 = [\mathbf{D}^e + \Delta\mu(1 - H_{\text{kin}}) \mathbf{D}^e \partial_{\boldsymbol{\gamma} \boldsymbol{\gamma}} g \mathbf{D}^e]^{-1}$, $\mathbf{A}_2 = \mathbf{D}^e - \Delta\mu H_{\text{kin}} \mathbf{D}^e \partial_{\boldsymbol{\gamma} \boldsymbol{\gamma}} g \mathbf{D}^e$, $\mathbf{A}_3 = \Delta\mu \mathbf{D}^e \partial_{\boldsymbol{\gamma} \boldsymbol{\gamma}} g \mathbf{D}^e$, $\mathbf{A}_4 = \Delta\mu H_{\text{kin}} \mathbf{D}^e \partial_{\boldsymbol{\gamma} \boldsymbol{\gamma}} g \mathbf{D}^e$, $\mathbf{A}_5 = \mathbf{D}^e + \Delta\mu \mathbf{D}^e \partial_{\boldsymbol{\gamma} \boldsymbol{\gamma}} g \mathbf{D}^e$, $A_6 = (1 - \Delta\mu H_{\text{iso}} D_{\text{norm}}^e \partial_{\boldsymbol{\beta} \boldsymbol{\beta}} g)^{-1}$. The algorithmic tangent modulus for this model is derived in Appendix A.

This derivation also involves the calculation of the first and second derivatives of the yield function g with respect to the strain-like tensor $\boldsymbol{\gamma}$, where g is an isotropic function of the principal values of $\boldsymbol{\gamma}$. The first derivative can be written as

$$\frac{\partial g}{\partial \boldsymbol{\gamma}} = \sum_{A=1}^3 \frac{\partial g}{\partial \gamma_A} \mathbf{m}^{(A)}, \tag{16}$$

see [29], where γ_A are the principal values and $\mathbf{m}^{(A)}$ are the eigenbases of the tensor $\boldsymbol{\gamma}$. The second derivatives are computed for distinct and non-zero principal values of $\boldsymbol{\gamma}$ as

$$\frac{\partial^2 g}{\partial \boldsymbol{\gamma}^2} = \sum_{A=1}^3 \sum_{B=1}^3 \frac{\partial^2 g}{\partial \gamma_A \partial \gamma_B} \mathbf{m}^{(A)} \otimes \mathbf{m}^{(B)} + \sum_{A=1}^3 \frac{\partial g}{\partial \gamma_A} \frac{\partial \mathbf{m}^{(A)}}{\partial \boldsymbol{\gamma}}, \tag{17}$$

see [29–31], where an explicit expression for $\partial \mathbf{m}^{(A)} / \partial \boldsymbol{\gamma}$ is obtained as

$$\begin{aligned}
\frac{\partial \mathbf{m}^{(A)}}{\partial \boldsymbol{\gamma}} &= \left[\mathbf{I}_e - (I_1 - \gamma_A) \mathbf{I} - \boldsymbol{\gamma} \otimes \mathbf{1} + \boldsymbol{\gamma} \otimes \mathbf{m}^{(A)} + I_3 \gamma_A^{-1} \mathbf{1} \otimes \boldsymbol{\gamma}^{-1} \right. \\
&\quad \left. - I_3 \gamma_A^{-2} \mathbf{1} \otimes \mathbf{m}^{(A)} + \gamma_A \mathbf{m}^{(A)} \otimes \mathbf{1} - I_3 \gamma_A^{-1} \mathbf{m}^{(A)} \otimes \boldsymbol{\gamma}^{-1} \right. \\
&\quad \left. + \psi \mathbf{m}^{(A)} \otimes \mathbf{m}^{(A)} \right] / D_A, \tag{18}
\end{aligned}$$

in terms of the three invariants I_1, I_2, I_3 of $\boldsymbol{\gamma}$, the second-order identity $\mathbf{1}$ and the fourth-order identity \mathbf{I} . In addition, Eq. (18) makes use of the following quantities

$$D_A = 2\gamma_A^2 - I_1 \gamma_A + I_3 \gamma_A^{-1} \neq 0,$$

$$\mathbb{I}_{(e)ijkl} = \frac{1}{2} (\delta_{ik} \gamma_{jl} + \delta_{il} \gamma_{jk}) + \frac{1}{2} (\gamma_{ik} \delta_{jl} + \gamma_{il} \delta_{jk}),$$

$$\psi = I_1 + I_3 \gamma_A^{-2} - 4\gamma_A.$$

This equation becomes indeterminate for equal or zero principal values and special forms of it for these cases are discussed in Appendix B.

5 Algorithmic implementation

The plasticity-like model derived in the previous section was implemented in FEAP, a fully nonlinear finite element code documented in [32]. To include material strength anisotropy, the fourth-order elasticity tensor \mathbf{C}^e in the principal material coordinate system was rotated to the global mesh coordinate system using known material orientations and then used in the plasticity algorithm to update state variables for each integration point. Representative simulations were conducted to test the accuracy and behavior of the model and to ensure numerical convergence and applicability to the finite element modeling of human proximal femur. Eight-node hexahedral elements were used for all the simulations. The Newton–Raphson scheme was employed to solve the nonlinear system emanating from the weak form of the equilibrium equations.

5.1 Homogeneous strain cycle

This problem consisted of a single brick element subjected to cyclic pure tension–compression displacement boundary conditions. Generic elastic material properties were assigned to this element, in which the material was assumed to be isotropic with Young’s modulus $E = 1000$ MPa and Poisson’s ratio $\nu = 0.3$. The parameters r, c, n and t of the yield envelope (Table 1) were assumed to be the same as that for the trabecular bone previously determined by Bayraktar et al. [22]. Four test cases were considered assuming: (1) the elastic perfectly plastic material behavior; (2) the kinematic hardening 0.1 times the elastic modulus ($H_{\text{kin}} = 0.1$); (3) the isotropic hardening 0.1 times the elastic modulus ($H_{\text{iso}} = 0.1$); and (4) both kinematic and isotropic hardening ($H_{\text{kin}} = 0.05, H_{\text{iso}} = 0.05$). The material behavior (stresses and strains) at one of the integration points is shown in Fig. 4.

This plasticity-like model under cyclic loading captured the expected behavior with the kinematic and isotropic hardening, as well as the tension-compression yield strength asymmetry (Fig. 4). The yield envelope was stationary in the elastic perfectly plastic case. A shift was observed in the yield envelope in the pure kinematic hardening case. The size of the yield envelope increased under pure isotropic hardening. The yield values were also in agreement with the analytical predictions.

Table 1 Coefficients of the modified super-ellipsoid yield surface given in Eq. (1) for femoral trabecular bone

Coefficient list	
r	0.738
c	-0.157
n	0.414
t	1.417

The radius and center has units of % strain; n and t are dimensionless

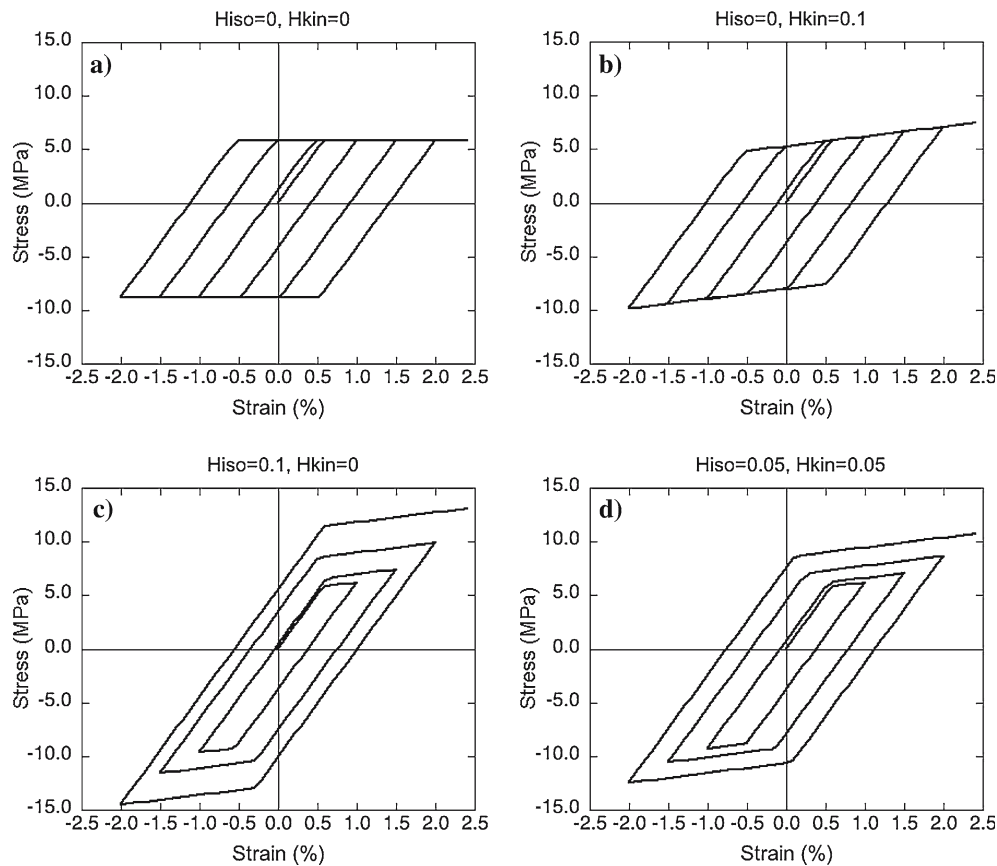


Fig. 4 The stress–strain behavior at one of the integration points of an eight-node brick subjected to cyclic loading. Four test cases were considered assuming material behavior as **a** elastic perfectly plastic; **b** kinematic hardening 0.1 times the elastic modulus; **c** isotropic hardening 0.1 times the elastic modulus; and **d** both kinematic and isotropic hardening

5.2 Solid cube under triaxial strain

In this problem, a $4 \times 4 \times 4$ mm solid cube with $1 \times 1 \times 1$ mm brick elements was subjected to uniform triaxial compression displacement boundary condition. Orthotropic material properties: $E_1 = 2,376$ MPa, $E_2 = 1,377$ MPa, $E_3 = 3,645$ MPa, $\nu_{12} = 0.28$, $\nu_{23} = 0.14$, $\nu_{13} = 0.15$, $G_{12} = 616$ MPa, $G_{23} = 784$ MPa, and $G_{13} = 1,193$ MPa, similar to trabecular bone were assigned to each element and the parameters r , c , and t were taken from Table 1. This problem was solved in FEAP with $H_{kin} = 0.05$ and $H_{iso} = 0.05$.

The stress–strain curves obtained in the three directions are shown in Fig. 5. In all three directions, the model failed at the same strain because of the isotropy of the yield envelope in strain space but at different stresses due to material orthotropy (Fig. 5). This result is in agreement with the behavior of the plasticity-like model presented here.

5.3 Nonlinear analysis of a human proximal femur

A proximal femur obtained from an 86-year-old female human cadaver was scanned with quantitative computed tomography (QCT, Somatom Plus 4, Siemens Medical,

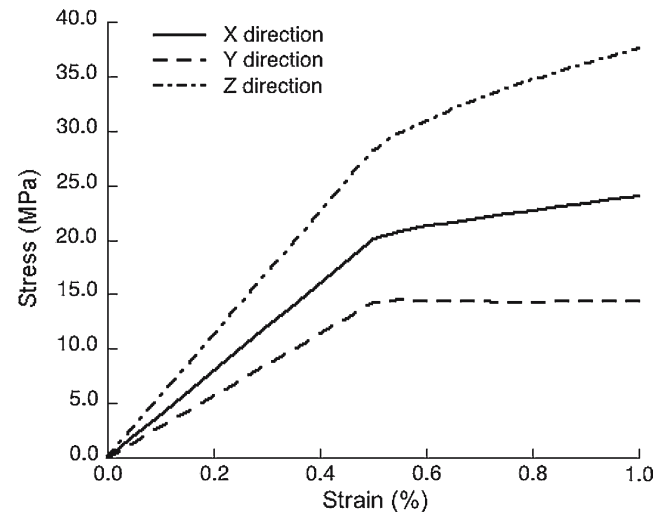


Fig. 5 Stress–strain curves for a $4 \times 4 \times 4$ mm cube model subjected to triaxial compression boundary condition. The model yielded at the same strain in three directions but at different stresses

Erlangen, Germany) at 120 kVp, 240 mA, (0.2 mm in-plane and 1 mm out-of-plane voxel size), and again with micro-Computed Tomography (micro-CT) using a cubic voxel size

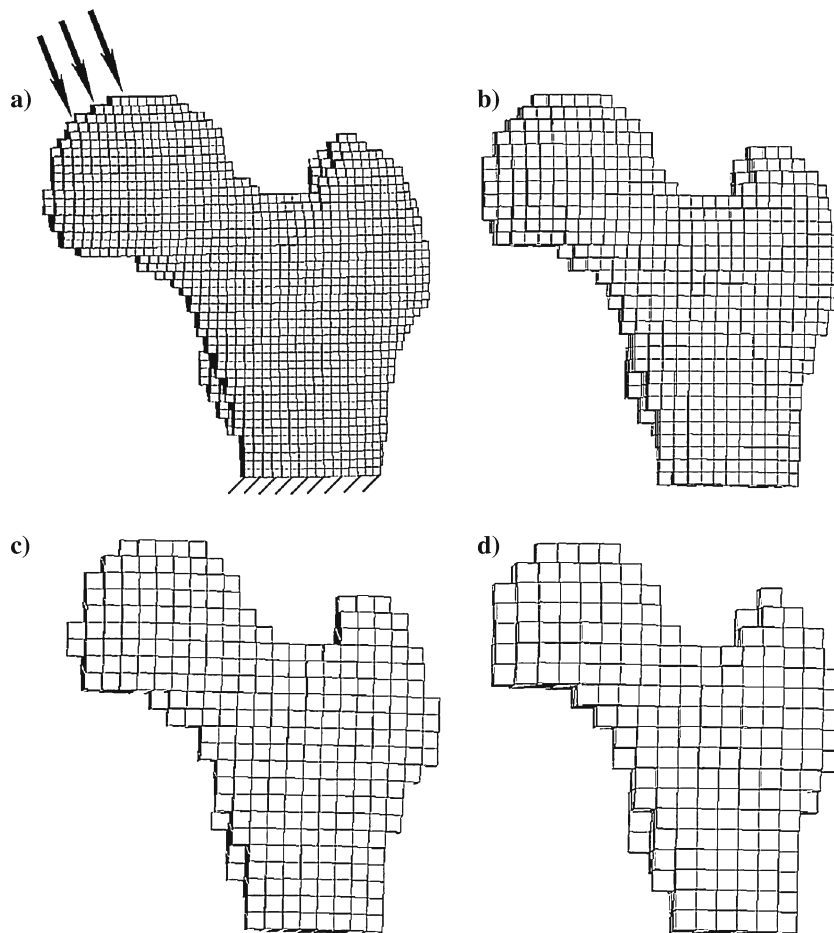


Fig. 6 Finite element models of a human femur generated from the QCT scans of 86-year-old female human cadaver with element sizes **a** $1.9335 \times 1.9335 \times 2.0$ mm, **b** $3.094 \times 3.094 \times 3.0$ mm, **c** $4.0604 \times 4.0604 \times 4.0$ mm and, **d** $5.0271 \times 5.0271 \times 5.0$ mm having 17,516, 4,540, 1,962, and 1,028 elements, respectively. Stance type loading condition, similar to loading of femur in the experiments, is shown for the 2 mm model in which a uniform displacement was applied at the femoral head and the distal end was fixed

of approximately $90 \mu\text{m}$ (Radios, SCANCO Medical AG, Bassersdorf, Switzerland) (Fig. 1). Three-dimensional voxel-based finite element models (Fig. 6) were generated from the QCT scans by coarsening these images, i.e., collapsing the image voxels in all directions, and converting these coarsened voxels directly into 8-noded brick elements with element size ranging from $5.0271 \times 5.0271 \times 5.0$ to $1.9335 \times 1.9335 \times 2.0$ mm having 1,028 to 17,516 elements.

The bone apparent density, ρ (in g/cm^3) of each element was determined from the QCT scans using a regression between the known apparent density values and the corresponding pixel intensities in hounsfield units (HU). The regression was created by linearly correlating mean HU values in the trochanteric and femoral neck regions of the proximal femur to the mean apparent density values for these regions, as calculated in a large cross-sectional study [7]. The primary elastic modulus (E_1 , in MPa) was assigned to each element using apparent density-modulus relationship determined from the on-axis mechanical tests of trabecular bone cores from the greater trochanter and femoral neck [7, 26] as

$$E_1 = \begin{cases} c_1 \rho^{\alpha_1}, & \text{for } \rho \leq 0.32 \text{g}/\text{cm}^3, \\ c_2 \rho^{\alpha_2}, & \text{for } \rho > 0.32 \text{g}/\text{cm}^3. \end{cases}$$

Here $\alpha_1 = 2.18$ and $\alpha_2 = 1.49$ are dimensionless constants, while $c_1 = 15,010$ and $c_2 = 6,850$ have the units necessary to render the preceding relations dimensionally consistent. The principal values and directions that characterize the microstructural anisotropic orientation of trabecular bone as well as the volume fraction were determined uniquely for each element in the finite element mesh using the mean intercept length tensor [33] measured from the micro-CT scans in 4 mm cube regions throughout the proximal femur [34]. Element-specific orthotropic elastic constants were then assigned to each element using published relations between volume fraction and the elastic modulus in the principal material direction [35].

A stance-type displacement boundary condition (Fig. 6) was applied to these models by fixing the distal end and applying a distributed compressive load at an angle of 20° to nodes

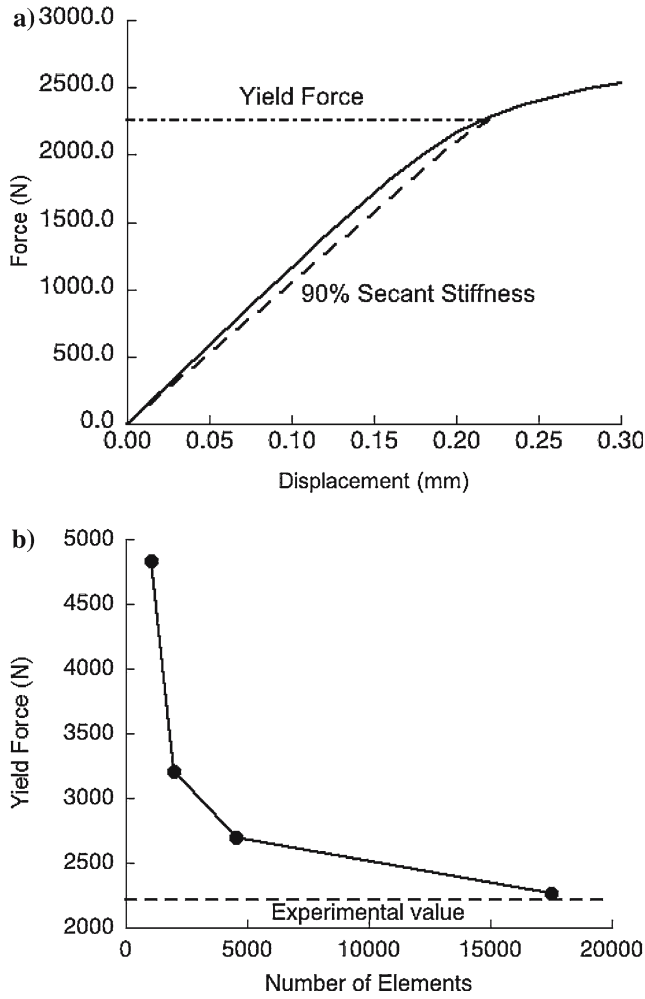


Fig. 7 **a** Force–deformation curve for 2 mm element size femur model. Yield force was calculated from the force–deformation curve using 90% secant method. **b** Convergence behavior of the yield force obtained from FE analyses of femur models where FE results approach actual solution as the mesh resolution was increased from 5 mm to 2 mm. For comparison experimental yield value is also shown in this figure, which was obtained from the destructive testing of this femur

on the superior aspect of the femoral head [36]. Muscle forces were avoided to match the FE simulations with the simplified boundary conditions used in the experiments. Nonlinear analyses were performed on these models using FEAP with both isotropic and kinematic hardening ($H_{\text{kin}} = 0.005$, $H_{\text{iso}} = 0.005$).

The yield force values of the femur were calculated from the force–deformation curve using a 90% secant method. This method displayed convergence, as the mesh was refined (Fig. 7). The yield force for the finest mesh was in close agreement with the experimental result obtained from the destructive testing of this femur. The convergence of the residual norm for this problem was quadratic and the force–deformation curve (Fig. 7) showed that the solution at each step was stable.

6 Conclusions

This article presents a constitutive and computational framework for the analysis of a cellular solid-type material with a yield envelope expressed in terms of principal strains. The material anisotropy in stress space is also incorporated in our model by rotation of the anisotropic fourth-order elasticity tensor from the principal material coordinate system to the global mesh coordinate system. A detailed procedure is derived for this plasticity-like model to facilitate algorithmic implementation. From a computational standpoint, the method developed here preserves the structure of return-mapping algorithm used widely for the stress- and strain-based plasticity formulations. A numerically stable implicit approach resulted in a robust formulation ensuring quadratic convergence for the Newton–Raphson iterative solution strategy. In future studies, the present formulation can be extended to include geometric nonlinearities that might have an effect on lower density trabecular bone found in many sites, and to other applications involving cellular solid materials with different yield envelope and hardening rules. The model could be further refined to include bone remodeling based on mechanical stimuli and damage behavior, which might play a significant role in the post-yield and reloading behavior of trabecular bone.

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Appendix A: algorithmic tangent moduli

In nonlinear finite element analysis, a consistent tangent operator is used for the algorithmic implementation of the solution procedure with iterative strategies such as the Newton–Raphson method. Assuming iterative solution procedure the algorithmic modulus can be defined for the step $n + 1$ as

$$\mathbf{C}^{(\text{alg})} = \left(\frac{d\boldsymbol{\sigma}}{d\boldsymbol{\varepsilon}} \right)_{n+1}. \quad (19)$$

Writing Eqs. (3), (4) and (5) in differential form and substituting $d\boldsymbol{\varepsilon}^e = \mathbf{D}^e : d\boldsymbol{\sigma}$ leads to

$$d\boldsymbol{\sigma} = \mathbf{C}^e : (d\boldsymbol{\varepsilon} - d\boldsymbol{\varepsilon}^p) \Rightarrow d\boldsymbol{\varepsilon}^p = -\mathbf{D}^e : d\boldsymbol{\sigma} + d\boldsymbol{\varepsilon},$$

$$d\boldsymbol{\varepsilon}^p = d(\Delta\mu)\mathbf{D}^e : \partial_{\boldsymbol{\varepsilon}^e} g + \Delta\mu \mathbf{D}^e :$$

$$(\partial_{\boldsymbol{\varepsilon}^e \boldsymbol{\varepsilon}^e} g : \mathbf{D}^e : d\boldsymbol{\sigma} + \partial_{\boldsymbol{\varepsilon}^e \boldsymbol{\alpha}} g : d\boldsymbol{\alpha} + \partial_{\boldsymbol{\varepsilon}^e \beta} g d\beta),$$

$$d\boldsymbol{\alpha} = d(\Delta\mu)H_{\text{kin}}\mathbf{D}^e : \partial_{\boldsymbol{\alpha}} g + \Delta\mu H_{\text{kin}}\mathbf{D}^e : \quad (20)$$

$$(\partial_{\boldsymbol{\alpha} \boldsymbol{\varepsilon}^e} g : \mathbf{D}^e : d\boldsymbol{\sigma} + \partial_{\boldsymbol{\alpha} \boldsymbol{\alpha}} g : d\boldsymbol{\alpha} + \partial_{\boldsymbol{\alpha} \beta} g d\beta),$$

$$d\beta = d(\Delta\mu)H_{\text{iso}}D_{\text{norm}}^e \partial_{\beta} g + \Delta\mu H_{\text{iso}}D_{\text{norm}}^e$$

$$(\partial_{\beta \boldsymbol{\varepsilon}^e} g : \mathbf{D}^e : d\boldsymbol{\sigma} + \partial_{\beta \boldsymbol{\alpha}} g : d\boldsymbol{\alpha} + \partial_{\beta \beta} g d\beta).$$

Also, from Eq. (2) it follows that

$$d\mathbf{g} = \mathbf{D}^e : \partial_{\boldsymbol{\varepsilon}^e} g : d\boldsymbol{\sigma} + \partial_{\boldsymbol{\alpha}} g : d\boldsymbol{\alpha} + \partial_{\beta} g d\beta = 0, \quad (21)$$

or, in matrix form,

$$dg = \underbrace{\left[\mathbf{D}^e : \partial_{\boldsymbol{\varepsilon}^e} g \quad \partial_{\boldsymbol{\alpha}} g \quad \partial_{\beta} g \right]}_{\{\partial \mathbf{g}\}} \begin{pmatrix} d\boldsymbol{\sigma} \\ d\boldsymbol{\alpha} \\ d\beta \end{pmatrix} = 0. \quad (22)$$

Likewise, Eq. (20) can be written in the matrix form as

$$\begin{pmatrix} d\boldsymbol{\varepsilon} \\ \mathbf{0} \\ 0 \end{pmatrix} - d(\Delta\mu) \underbrace{\begin{pmatrix} \mathbf{D}^e : \partial_{\boldsymbol{\varepsilon}^e} g \\ \partial_{\boldsymbol{\alpha}} g \\ \partial_{\beta} g \end{pmatrix}}_{\{\tilde{\mathbf{r}}\}} = \underbrace{\begin{bmatrix} \mathbf{D}^e + \Delta\mu \mathbf{D}^e \partial_{\boldsymbol{\varepsilon}^e} g \mathbf{D}^e & \Delta\mu \mathbf{D}^e \partial_{\boldsymbol{\alpha}} g & \Delta\mu \mathbf{D}^e \partial_{\beta} g \\ \Delta\mu \partial_{\boldsymbol{\alpha}} g \mathbf{D}^e & -\frac{C^e}{H_{\text{kin}}} + \Delta\mu \partial_{\boldsymbol{\alpha}} g & \Delta\mu \partial_{\beta} g \\ \Delta\mu \partial_{\beta} g \mathbf{D}^e & \Delta\mu \partial_{\beta} g & -\frac{1}{H_{\text{iso}} D_{\text{norm}}^e} + \Delta\mu \partial_{\beta} g \end{bmatrix}}_{\mathbf{B}^{-1}} \times \begin{pmatrix} d\boldsymbol{\sigma} \\ d\boldsymbol{\alpha} \\ d\beta \end{pmatrix}, \quad (23)$$

or

$$\begin{pmatrix} d\boldsymbol{\sigma} \\ d\boldsymbol{\alpha} \\ d\beta \end{pmatrix} = \mathbf{B} : \begin{pmatrix} d\boldsymbol{\varepsilon} \\ \mathbf{0} \\ 0 \end{pmatrix} - d(\Delta\mu) \mathbf{B} : \{\tilde{\mathbf{r}}\}. \quad (24)$$

Combining Eqs. (22) and (24) to solve for the unknown $d(\Delta\mu)$ results in

$$d(\Delta\mu) = \frac{\{\partial \mathbf{g}\} : \mathbf{B} : \begin{pmatrix} d\boldsymbol{\varepsilon} \\ \mathbf{0} \\ 0 \end{pmatrix}}{\{\partial \mathbf{g}\} : \mathbf{B} : \{\tilde{\mathbf{r}}\}}. \quad (25)$$

Substitution of the value of $d(\Delta\mu)$ in Eq. (24) results in

$$\begin{pmatrix} d\boldsymbol{\sigma} \\ d\boldsymbol{\alpha} \\ d\beta \end{pmatrix} = \left[\mathbf{B} - \frac{(\mathbf{B} : \{\tilde{\mathbf{r}}\}) \otimes (\{\partial \mathbf{g}\} : \mathbf{B})}{\{\partial \mathbf{g}\} : \mathbf{B} : \{\tilde{\mathbf{r}}\}} \right] \begin{pmatrix} d\boldsymbol{\varepsilon} \\ \mathbf{0} \\ 0 \end{pmatrix}. \quad (26)$$

From the above derivation, the algorithmic modulus (consistent tangent operator) can be written as

$$\mathbf{C}^{(\text{alg})} = \left[\mathbf{B} - \frac{(\mathbf{B} : \{\tilde{\mathbf{r}}\}) \otimes (\{\partial \mathbf{g}\} : \mathbf{B})}{\{\partial \mathbf{g}\} : \mathbf{B} : \{\tilde{\mathbf{r}}\}} \right]_{\text{upper } 6 \times 6 \text{ part}}. \quad (27)$$

Using Eq. (13), \mathbf{B}^{-1} from Eq. (23) can be simplified as

$$\mathbf{B}^{-1} = \begin{bmatrix} \mathbf{D}^e + \Delta\mu \mathbf{D}^e \partial_{\boldsymbol{\gamma}\boldsymbol{\gamma}} g \mathbf{D}^e & -\Delta\mu \mathbf{D}^e \partial_{\boldsymbol{\gamma}\boldsymbol{\gamma}} g & \mathbf{0} \\ -\Delta\mu \partial_{\boldsymbol{\gamma}\boldsymbol{\gamma}} g \mathbf{D}^e & -\frac{C^e}{H_{\text{kin}}} + \Delta\mu \partial_{\boldsymbol{\gamma}\boldsymbol{\gamma}} g & \mathbf{0} \\ \mathbf{0} & \mathbf{0} & -\frac{1}{H_{\text{iso}} D_{\text{norm}}^e} + \Delta\mu \partial_{\beta\beta} g \end{bmatrix}. \quad (28)$$

After inverting, \mathbf{B} is obtained as

$$\mathbf{B} = \begin{bmatrix} \mathbf{B}_1 - \mathbf{B}_1 \mathbf{B}_2 & -\mathbf{B}_1 \mathbf{B}_3 & \mathbf{0} \\ -\mathbf{B}_3 \mathbf{B}_1 & -\mathbf{D}^e \mathbf{B}_1 \mathbf{B}_4 & \mathbf{0} \\ \mathbf{0} & \mathbf{0} & B_5 \end{bmatrix}, \quad (29)$$

where $\mathbf{B}_1 = [\mathbf{D}^e + \Delta\mu(1 - H_{\text{kin}})\mathbf{D}^e \partial_{\boldsymbol{\gamma}\boldsymbol{\gamma}} g \mathbf{D}^e]^{-1}$, $\mathbf{B}_2 = \Delta\mu H_{\text{kin}} \mathbf{D}^e \partial_{\boldsymbol{\gamma}\boldsymbol{\gamma}} g$, $\mathbf{B}_3 = \Delta\mu H_{\text{kin}} \mathbf{D}^e \partial_{\boldsymbol{\gamma}\boldsymbol{\gamma}} g \mathbf{D}^e$, $\mathbf{B}_4 = H_{\text{kin}} \mathbf{D}^e + \Delta\mu H_{\text{kin}} \mathbf{D}^e \partial_{\boldsymbol{\gamma}\boldsymbol{\gamma}} g \mathbf{D}^e$, $B_5 = H_{\text{iso}} D_{\text{norm}}^e (\Delta\mu H_{\text{iso}} D_{\text{norm}}^e \partial_{\beta\beta} g - 1)^{-1}$.

Appendix B: second derivatives

The second derivative of the yield envelope g with respect to $\boldsymbol{\gamma}$ [Eq. (17)] becomes indeterminate for the case of equal or zero roots of tensor $\boldsymbol{\gamma}$. To avoid the indeterminacy, particular expressions of these derivatives are deduced. In what follows, γ_A are the principal values, $\mathbf{n}^{(A)}$ are the eigenvectors and $\mathbf{m}^{(A)}$ are the eigenbases of tensor $\boldsymbol{\gamma}$.

1. If $\gamma_1 = \gamma_2 \neq \gamma_3$,

$$\begin{aligned} \frac{\partial^2 g}{\partial \boldsymbol{\gamma}^2} &= \frac{\partial \gamma_1 g}{\partial \gamma_1} \mathbf{1} \otimes \mathbf{1} + \left(\frac{\partial \gamma_1 g}{\partial \gamma_3} - \frac{\partial \gamma_1 g}{\partial \gamma_1} \right) \mathbf{1} \otimes \mathbf{m}^{(3)} \\ &\quad + \left(\frac{\partial \gamma_3 g}{\partial \gamma_1} - \frac{\partial \gamma_1 g}{\partial \gamma_1} \right) \mathbf{m}^{(3)} \otimes \mathbf{1} \\ &\quad + \left(\frac{\partial \gamma_3 g}{\partial \gamma_3} - \frac{\partial \gamma_1 g}{\partial \gamma_3} - \frac{\partial \gamma_3 g}{\partial \gamma_1} + \frac{\partial \gamma_1 g}{\partial \gamma_1} \right) \mathbf{m}^{(3)} \otimes \mathbf{m}^{(3)} \\ &\quad + \left(\frac{\partial g}{\partial \gamma_3} - \frac{\partial g}{\partial \gamma_1} \right) \frac{\partial \mathbf{m}^{(3)}}{\partial \boldsymbol{\gamma}}, \end{aligned} \quad (30)$$

where $\partial \mathbf{m}^{(3)} / \partial \boldsymbol{\gamma}$ can be calculated based on Eq. (18).

2. If $\gamma_1 = \gamma_2 = \gamma_3$,

$$\frac{\partial^2 g}{\partial \boldsymbol{\gamma}^2} = \frac{\partial \gamma_1 g}{\partial \gamma_1} \mathbf{1} \otimes \mathbf{1}. \quad (31)$$

3. If $\gamma_1 \neq \gamma_2$ and $\gamma_3 = 0$,

$$\begin{aligned} \frac{\partial^2 g}{\partial \boldsymbol{\gamma}^2} &= \sum_{A=1}^3 \sum_{B=1}^3 \frac{\partial \gamma_A g}{\partial \gamma_B} \mathbf{m}^{(A)} \otimes \mathbf{m}^{(B)} \\ &\quad + \frac{1}{2} \sum_{A=1}^3 \sum_{B \neq A}^3 \frac{(\partial g / \partial \gamma_B) - (\partial g / \partial \gamma_A)}{\gamma_B - \gamma_A} \\ &\quad \times \left(\mathbf{m}^{(AB)} \otimes \mathbf{m}^{(AB)} + \mathbf{m}^{(AB)} \otimes \mathbf{m}^{(BA)} \right), \end{aligned} \quad (32)$$

where $\mathbf{m}^{(A)} = \mathbf{n}^{(A)} \otimes \mathbf{n}^{(A)}$, $\mathbf{m}^{(AB)} = \mathbf{n}^{(A)} \otimes \mathbf{n}^{(B)}$, $A \neq B$.

4. If $\gamma_1 = \gamma_2 = 0$ and $\gamma_3 \neq 0$,

$$\begin{aligned} \frac{\partial^2 g}{\partial \boldsymbol{\gamma}^2} &= \frac{\partial \gamma_1 g}{\partial \gamma_1} \mathbf{1} \otimes \mathbf{1} \\ &\quad + \left(\frac{\partial \gamma_1 g}{\partial \gamma_3} - \frac{\partial \gamma_1 g}{\partial \gamma_1} \right) \mathbf{1} \otimes \mathbf{m}^{(3)} \\ &\quad + \left(\frac{\partial \gamma_3 g}{\partial \gamma_1} - \frac{\partial \gamma_1 g}{\partial \gamma_1} \right) \mathbf{m}^{(3)} \otimes \mathbf{1} \\ &\quad + \left(\frac{\partial \gamma_3 g}{\partial \gamma_3} - \frac{\partial \gamma_1 g}{\partial \gamma_3} - \frac{\partial \gamma_3 g}{\partial \gamma_1} + \frac{\partial \gamma_1 g}{\partial \gamma_1} \right) \mathbf{m}^{(3)} \otimes \mathbf{m}^{(3)} \\ &\quad + \left(\frac{\partial g}{\partial \gamma_3} - \frac{\partial g}{\partial \gamma_1} \right) \frac{\partial \mathbf{m}^{(3)}}{\partial \boldsymbol{\gamma}}, \end{aligned} \quad (33)$$

where $(\partial \mathbf{m}^{(3)} / \partial \boldsymbol{\gamma}) = (1/\gamma_3)(\mathbf{I} - \mathbf{m}^{(3)} \otimes \mathbf{m}^{(3)})$.

5. If $\gamma_1 = \gamma_2$ and $\gamma_3 = 0$,

$$\begin{aligned} \frac{\partial^2 g}{\partial \boldsymbol{\gamma}^2} &= \frac{\partial \gamma_3 g}{\partial \gamma_3} \mathbf{1} \otimes \mathbf{1} \\ &+ \left(\frac{\partial \gamma_3 g}{\partial \gamma_1} - \frac{\partial \gamma_3 g}{\partial \gamma_3} \right) \mathbf{1} \otimes (\mathbf{m}^{(1)} + \mathbf{m}^{(2)}) \\ &+ \left(\frac{\partial \gamma_1 g}{\partial \gamma_3} - \frac{\partial \gamma_3 g}{\partial \gamma_3} \right) (\mathbf{m}^{(1)} + \mathbf{m}^{(2)}) \otimes \mathbf{1} \\ &+ \left(\frac{\partial \gamma_1 g}{\partial \gamma_1} - \frac{\partial \gamma_3 g}{\partial \gamma_1} - \frac{\partial \gamma_1 g}{\partial \gamma_3} + \frac{\partial \gamma_3 g}{\partial \gamma_3} \right) \\ &\times (\mathbf{m}^{(1)} + \mathbf{m}^{(2)}) \otimes (\mathbf{m}^{(1)} + \mathbf{m}^{(2)}) \\ &+ \left(\frac{\partial g}{\partial \gamma_1} - \frac{\partial g}{\partial \gamma_3} \right) \frac{\partial (\mathbf{m}^{(1)} + \mathbf{m}^{(2)})}{\partial \boldsymbol{\gamma}}, \end{aligned} \quad (34)$$

where $(\partial (\mathbf{m}^{(1)} + \mathbf{m}^{(2)}) / \partial \boldsymbol{\gamma}) = (1/\gamma_1) (\mathbf{I} - (\mathbf{m}^{(1)} + \mathbf{m}^{(2)}) \otimes (\mathbf{m}^{(1)} + \mathbf{m}^{(2)}))$.

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