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## The dependence of the partitioning of iron and europium between plagioclase and hydrous tonalitic melt on oxygen fugacity

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**Abstract** The dependence of iron and europium partitioning between plagioclase and melt on oxygen fugacity was studied in the system  $\text{SiO}_2(\text{Qz})\text{—NaAlSi}_3\text{O}_8(\text{Ab})\text{—CaAl}_2\text{Si}_2\text{O}_8(\text{An})\text{—H}_2\text{O}$ . Experiments were performed at 500 MPa and 850 °C/750 °C under water saturated conditions. The oxygen fugacity was varied in the  $\log f_{\text{O}_2}$ -range from  $-7.27$  to  $-15.78$ . To work at the most reducing conditions the classical double-capsule technique was modified. The sample and a C—O—H bearing sensor capsule were placed next to each other within a BN jacket to minimise loss of hydrogen to the vessel atmosphere. By this setup redox conditions slightly more reducing than the FeO—Fe<sub>3</sub>O<sub>4</sub> buffer could be maintained even in 96 h runs. Raman spectra showed that the BN was modified by reaction with hydrogen resulting in a low hydrogen permeability. The partition coefficients determined for Eu at 850 °C and 500 MPa vary from 0.095 at conditions of the Cu—Cu<sub>2</sub>O buffer to 1.81 at the most reducing conditions (C—O—H sensor). In the same  $f_{\text{O}_2}$  interval the partition coefficient for Fe varies from 0.55 at oxidising conditions to 0.08 at the most reducing conditions. The partitioning of Sm, which was added as a reference for a trivalent REE, does not vary with the oxygen fugacity, yielding an average value for  $D = 0.07$ . Lowering the temperature to 750 °C for a given  $f_{\text{O}_2}$  decreases the partition coefficient of Eu and increases that of Fe. Comparison with published data at 1 atm and at higher temperatures shows that both temperature and composition of the melt have strong effects on the partitioning behaviour. As the change of the partition coefficients in the geologically relevant  $f_{\text{O}_2}$  range is quite strong, element partitioning of Eu and Fe might

be used to estimate redox conditions for the genesis of igneous rocks. Furthermore, by modelling the partitioning data it is possible to extract information about the redox state of the melt. Resulting ferric-ferrous ratios show significant differences from those predicted by empirical models.

### Introduction

The redox conditions during the formation of igneous rocks can have a strong effect on the resulting phase assemblages (Scaillet 1995; Rutherford and Devine 1996; Martel et al. 1998). Hence there is a need to quantify the oxygen fugacity prevailing during the genesis of these rocks. Most of the changes in phase assemblages are due to the changing redox state of heterovalent elements like Fe and Eu. The partitioning of these elements between coexisting phases is strongly dependent on the oxygen fugacity. If pressure and temperature are known, the redox conditions during the formation of igneous rocks can be quantified using well calibrated redox sensors, e.g. coexisting iron-titanium oxides (Spencer and Lindsley 1981; Andersen and Lindsley 1988). Estimates by this sensor at rather oxidising conditions (NNO + 1.3), however, must be treated with caution since extra components in ilmenite expand the solvus in a way that is not accommodated for in the current model of the thermobarometer (Lindsley and Frost 1992; Evans and Scaillet 1997).

However, numerous igneous rocks do not contain ilmenite and thus this thermobarometer cannot be applied. Another approach is to bracket the redox conditions during the formation of igneous rocks by using the partitioning of iron and europium between plagioclase and melt. Both elements have two stable oxidation states,  $\text{M}^{2+}$  and  $\text{M}^{3+}$  and thus their partitioning behaviour is dependent on the oxygen fugacity. Different incorporation schemes in plagioclase are observed for  $\text{M}^{2+}$  and  $\text{M}^{3+}$ .

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In the case of Eu, two substitution mechanisms have been verified experimentally: (i)  $\text{Eu}^{2+} \leftrightarrow \text{Ca}^{2+}$  and (ii)  $3 \text{Ca}^{2+} \leftrightarrow 2 \text{Eu}^{3+} + \text{V}$  where V denotes a vacancy (Kimata 1988). Incorporation of a trivalent cation on the Ca site of plagioclase is rather unfavourable and the first substitution scheme is strongly preferred. This also is supported by the Eu anomaly found in igneous rocks with residual plagioclase. In contrast to the other rare earth elements (REE), Eu also exists in the divalent state under geological conditions leading to enrichment of Eu in plagioclase.

Incorporation of Fe into the feldspar structure is more complex and at least three substitution mechanisms can occur: (i)  $\text{Fe}^{3+} \leftrightarrow \text{Al}^{3+}$ , (ii)  $\text{Fe}^{2+} \leftrightarrow \text{Ca}^{2+}$ , (iii)  $\text{Fe}^{2+} + \text{Si}^{4+} \leftrightarrow 2 \text{Al}^{3+}$ . In alkali feldspars only the first substitution scheme has been observed (Coombs 1954; Hofmeister and Rossman 1984). Plagioclase can also incorporate  $\text{Fe}^{2+}$  and there is experimental evidence for substitution schemes (ii) and (iii) (Longhi et al. 1976; Hofmeister and Rossman 1984, Behrens et al. 1990). Hofmeister and Rossman (1984) have found by spectroscopic investigation of terrestrial plagioclase a positive correlation of the  $\text{Fe}^{2+}/\text{Fe}^{3+}$  to the An content of the plagioclase with  $\text{Fe}^{2+}$  becoming dominant only in pure anorthite.

Several attempts have been made to elucidate the oxygen fugacity dependence for the partitioning of europium (Drake 1975; Weill and McKay 1975) and iron (Phinney 1992; Sato 1989) between basaltic melts and plagioclase. All experiments were performed at 1 atm and temperatures around 1200 °C. These studies have shown a variation of the partition coefficients ( $D = c_{\text{crystal}}/c_{\text{melt}}$ ,  $c$  in weight percent) of about 1.5 log units for Eu and of about 0.6 log units for Fe in the log  $f_{\text{O}_2}$ -range -6 to 0.7.

As natural magmas often contain water it is important to know the influence of water dissolved in the melt on the partitioning. Furthermore, the effect of anhydrous melt composition is not known. Extrapolation of the data obtained for basaltic melts at high temperature to granitic and tonalitic melts at lower temperatures is highly uncertain. In this paper we present data on the partitioning of Eu and Fe between melt and plagioclase in the haplotonalitic system  $\text{SiO}_2(\text{Qz})\text{—NaAlSi}_3\text{O}_8(\text{Ab})\text{—CaAl}_2\text{Si}_2\text{O}_8(\text{An})\text{—H}_2\text{O}$  at 500 MPa pressure and 850 °C/750 °C at water saturation. A central part of this paper is the presentation of the technical improvements we have applied to the solid-buffer technique to maintain highly reducing conditions in internally heated pressure vessels (IHPV) at high temperatures. Furthermore, we give a model to describe the  $f_{\text{O}_2}$ -dependence of the element partitioning and show how information about the redox state of the melt can be extracted from the partitioning data.

## Experimental techniques

### Starting materials and experimental apparatus

The experiments were performed in the system  $\text{SiO}_2(\text{Qz})\text{—NaAlSi}_3\text{O}_8(\text{Ab})\text{—CaAl}_2\text{Si}_2\text{O}_8(\text{An})\text{—H}_2\text{O}$ . A plagioclase-rich composition was chosen for the 850 °C experiments and a silica-rich composition for the 750 °C experiments so that high melt/plagioclase ratios were obtained for both temperatures. The synthetic glasses contained 0.5 wt%  $\text{Eu}_2\text{O}_3$  and  $\text{Sm}_2\text{O}_3$ . As Sm does not change its redox state it is a reference for the partitioning of a trivalent REE. In experiments at 750 °C (500 MPa) using glasses doped with Sm and Eu, many tiny needles of a REE-rich phase were formed and no crystal-free area in the glass could be found sufficiently large to perform microprobe analyses. Hence the experiments at 750 °C had to be done with Eu only. Glasses for Fe partitioning experiments were doped with 1.5 wt%  $\text{Fe}_2\text{O}_3$ . In order to establish identical run conditions it would have been preferable to use Fe, Eu and Sm in the same experiment, but doping the glasses with both REE and Fe lead to precipitation of allanite. Therefore, Fe and Eu partitioning have to be studied separately. The anhydrous glasses were synthesised from oxides and carbonates at 1600 °C in platinum crucibles and checked by electron microprobe for homogeneity. The compositions of the anhydrous starting glasses are shown in Table 1.

Disequilibrium growth of plagioclase was observed in runs using anhydrous or water poor glass as starting material. We attribute this to the rather slow diffusion of the water (Behrens and Nowak 1997; Nowak and Behrens 1997) into the initially anhydrous glass and to slow relaxation of the highly viscous melt (Schulze et al. 1996). Therefore glasses containing about 10 wt%  $\text{H}_2\text{O}$  which were pre-saturated at the same pressure but at higher temperature, were used as starting material. To synthesise the Eu-bearing glasses the crushed anhydrous glass was sealed together with 12 wt%  $\text{H}_2\text{O}$  in Pt capsules and annealed at 1200 °C and 500 MPa for 48 h. In order to minimise iron loss to the capsule glasses containing Fe were synthesised at lower temperature (1000 °C) in shorter runs (30 h) using Au capsules. The water contents of the glasses were analysed by Karl-Fischer Titration (Behrens 1995). In the partitioning experiments single pieces of the hydrous glasses were enclosed in Pt capsules together with an appropriate amount of water to ensure saturated conditions. For Fe bearing glasses Ag capsules were used except for runs using the Cu— $\text{Cu}_2\text{O}$  buffer because alloying of Cu and Ag lead to melting of the capsules. Pt was used for these runs as dissolution of Fe in Pt is small at such oxidising conditions. To encourage growth of few but rather big plagioclase crystals all samples were annealed for 30 min at conditions above the liquidus and then cooled at 1–0.5 °C/min to the run temperature (retrograde temperature path).

**Table 1** Electron microprobe analyses and CIPW normative compositions of anhydrous starting glasses. All data given in wt%

	To1(Eu)	To2(Eu)	To1(Fe)	To2(Fe)
$\text{SiO}_2$	70.18	76.77	70.57	76.08
$\text{Al}_2\text{O}_3$	19.03	15.05	17.85	14.49
CaO	7.35	6.65	7.32	6.00
$\text{Na}_2\text{O}$	3.46	2.10	3.41	2.56
$\text{Fe}_2\text{O}_3$	—	—	1.57	1.58
$\text{Eu}_2\text{O}_3$	0.51	0.47	—	—
$\text{Sm}_2\text{O}_3$	0.52	—	—	—
Total	101.05	101.04	100.89	100.71
Qz	34.3	49.8	35.0	48.5
Ab	29.2	17.6	28.8	21.7
An	36.5	32.6	36.2	29.8
C	0.0	0.0	0.0	0.0

The experiments were performed in two different types of pressure vessels: (1) horizontally orientated cold-seal pressure vessels (CSPV) pressurised with water, and (2) vertically orientated internally heated pressure vessels (IHPV) pressurised with argon.

Type (1): The CSPV used for experiments at 750 °C are made of a superalloy (Vakumelt 2.4888) with an inner diameter of 6 mm. To minimise temperature gradients the free volume was filled by inconel filler rods. The temperature was controlled by an external, sheathed type-K thermocouple. The temperature at the sample position was calibrated to the external thermocouple using a calibration vessel. The recorded temperature is believed to be accurate within  $\pm 8$  °C. Pressure was measured with a strain gauge manometer with an uncertainty of  $\pm 50$  bar (for details on the CSPV see Puziewicz and Johannes 1988). A temperature of 700 °C was reached after 18 min and 750 °C after 30 min. Samples were quenched isobarically in a flux of compressed air resulting in an initial quench rate of 250 °C/min.

Type (2): The IHPV used for all experiments above 800 °C has a working volume of 15 mm in diameter and 50 mm in length within the hot spot zone. Pressure was measured with a strain gauge manometer with an uncertainty of  $\pm 50$  bar. The temperature in the hot spot zone was monitored using three sheathed type-K thermocouples spaced 20 mm apart. The IHPV was heated by a two-loop furnace made of molybdenum wire. The temperature variation in the whole working volume was below  $\pm 10$  °C in all experiments. The samples were heated to superliquidus temperature at a constant rate of 30 °C/min using a programmable controller (Eurotherm 900 Series). The samples were quenched at an average rate of 150 °C/min to below 400 °C. Constant pressure was maintained during cooling by automatic pumping.

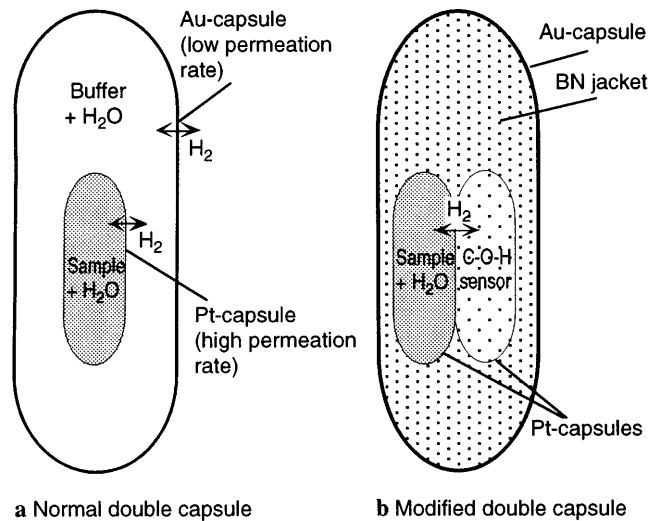
### Buffering technique

In high pressure experiments the oxygen fugacity is often controlled using the double-capsule technique described by Eugster and Wones (1962), in which the hydrogen fugacity is set in an outer capsule by the redox equilibrium of a metal and a corresponding metal oxide (e.g.  $\text{Ni} + \text{H}_2\text{O} = \text{NiO} + \text{H}_2$  for the NNO buffer, Fig. 1). The hydrogen fugacity is buffered by the reaction of metal with water and equilibrates with the atmosphere of the inner sample capsule, setting the oxygen fugacity via the water equilibrium ( $\text{H}_2 + 1/2 \text{O}_2 = \text{H}_2\text{O}$ ). For the inner capsule normally a metal with a high  $\text{H}_2$ -permeation rate is used and for the outer capsule a metal with a low  $\text{H}_2$ -permeation rate. At high temperatures, however, the differences in the permeability constants of the metals become smaller (Chou 1986) and only short run durations can be achieved under strongly reducing or oxidising conditions due to rapid hydrogen permeation through the capsule walls.

Using the CSPV at conditions of NNO or CCO the buffer material was enclosed directly into the vessel. The CCO buffer had to be renewed at least every third day due to hydrogen loss from the vessel (Grochau 1996) and the experiments were interrupted once by quenching and reheating. Morphology and major element composition of the plagioclase showed no difference compared to the runs using the NNO buffer which were performed without interruption at the same  $P$ - $T$  conditions. Thus we infer that the interruption is of minor importance for the resulting partition coefficients.

To quantify  $f_{\text{O}_2}$  of experiments under intrinsic conditions of the IHPV the redox conditions at 5 MPa and 850 °C were determined using NiPd-solid sensors after Taylor et al. (1992). The obtained value of  $\log f_{\text{O}_2} = -10.5$  (corresponding to  $f_{\text{H}_2} = 2$  bar in the vessel) is in agreement with previous determinations for IHPV (Popp et al. 1984; Linnen et al. 1995).

The conventional double-capsule technique was used at  $f_{\text{O}_2}$  of NNO and higher. At more reducing conditions hydrogen is rapidly lost from the buffer capsule at the high temperatures used in this study and run durations of several days are not possible. This problem was circumvented by modifying the double-capsule technique. Fig. 1 shows a schematic sketch of the classical double-capsule technique together with the setup used here. Our novel



**Fig. 1a, b** Schematic sketch of normal double-capsule setup and the modified version (outer capsule 8 mm diameter  $\times$  50 mm, inner capsule 3 mm diameter  $\times$  20 mm, wall thickness 0.2 mm). In the modified setup sensor capsule and sample capsule are in direct contact. Both capsules are enclosed in a BN jacket. Equilibration in  $f_{\text{H}_2}$  only takes place between the sample and the sensor capsule. The BN prevents rapid loss of hydrogen to the vessel atmosphere

setup is based on the technique used by Truckenbrodt et al. (1997) for piston cylinder apparatus. An oxygen fugacity controlling capsule is placed next to the sample capsule within a boron nitride (BN) jacket which is enclosed in an outer Au capsule. Instead of a buffer assemblage, a C—O—H fluid is used both as hydrogen source and  $f_{\text{H}_2}$  sensor. During the experiment the assemblage continuously loses hydrogen. However, the use of a BN jacket strongly reduces the loss of hydrogen so that the variation of oxygen fugacity is small during the experiment. As starting material for the sensors organic substances with various H/O ratios are used ( $\text{C}_9\text{H}_{10}\text{O}_2$ , H/O=5;  $\text{C}_{14}\text{H}_{22}\text{O}$ , H/O=22 or  $\text{C}_{14}\text{H}_{10}$ , H/O= $\infty$ ). These substances decompose to graphite and a C—O—H fluid (a mixture of CO,  $\text{CO}_2$ ,  $\text{H}_2\text{O}$ ,  $\text{CH}_4$ ,  $\text{C}_2\text{H}_6$ ), which is analysed by gas chromatography after the run. By summing the amount of H and O in all components of the fluid the total H/O ratio was determined and the hydrogen fugacity calculated using a modified, compensated Redlich Kwong model (Krautheim et al. 1992). The calculation is based on the compensated Redlich Kwong equation of state of Holland and Powell (1991) combined with the mixing rules of De Santis (1974) and the correction of Flowers (1979).

### Analytical procedure

Polished sections of the samples were analysed using the Cameca Camebax electron microprobe controlled by the Xmas<sup>®</sup> software package. The major elements were analysed using standard electron beam conditions, 15 kV, 18 nA. Counting time for each element was 10 s except for Na (5 s). For analyses of the hydrous glass the beam was defocused to 20  $\mu\text{m}$  to avoid Na loss. Elevated exciting conditions (25 kV and 100 nA) were used to obtain accurate analyses of the minor components Eu, Sm and Fe. The concentration of these were determined in separate analyses measuring only these components and using a fixed matrix correction based on the average of the analyses of the major components. Using counting time of 30–60 s the detection limits were in the range of 100 ppm. In both crystals and glasses, the signal intensities for Eu, Sm and Fe show no variation with counting time when monitored for 120 s. Standards were: wollastonite for Si and Ca, albite for Na, corundum for Al, hematite for Fe, europiumphosphate for Eu and samariumphosphate for Sm.

## Results and discussion

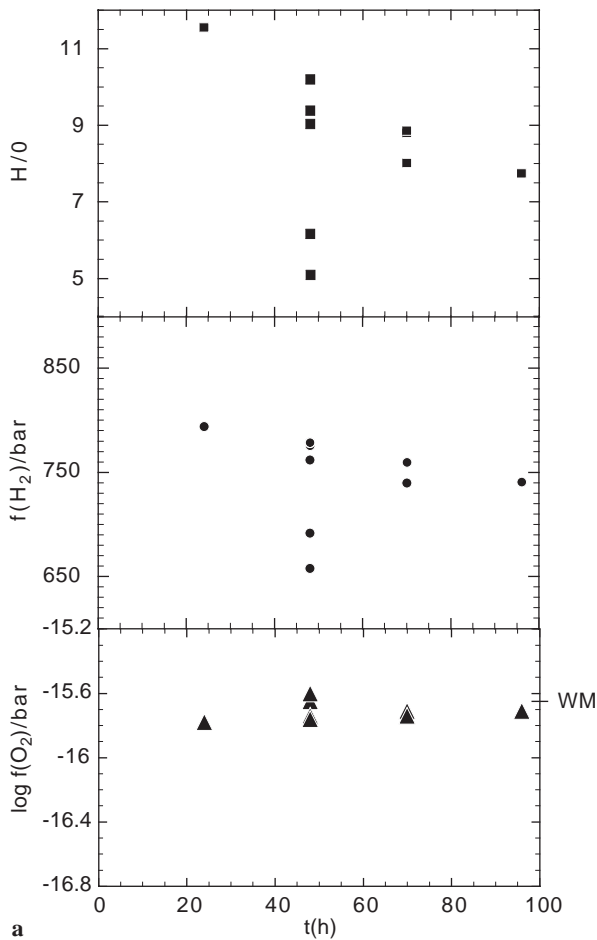
### Improvement of buffering techniques

The stability of the hydrogen fugacity in the modified double-capsule assemblage was investigated by analysing H/O ratios in the sensor capsules and Eu partitioning in the sample capsule as a function of run duration. Figure 2a shows the development of the H/O ratio measured in the sensor capsules. With the exception of two runs, a smooth decrease of the H/O ratio with run duration is observed. In one of the outlying runs the content of organic material in the sensor capsule was only one half of that used in the other runs. In the second outlying run a different temperature path was cho-

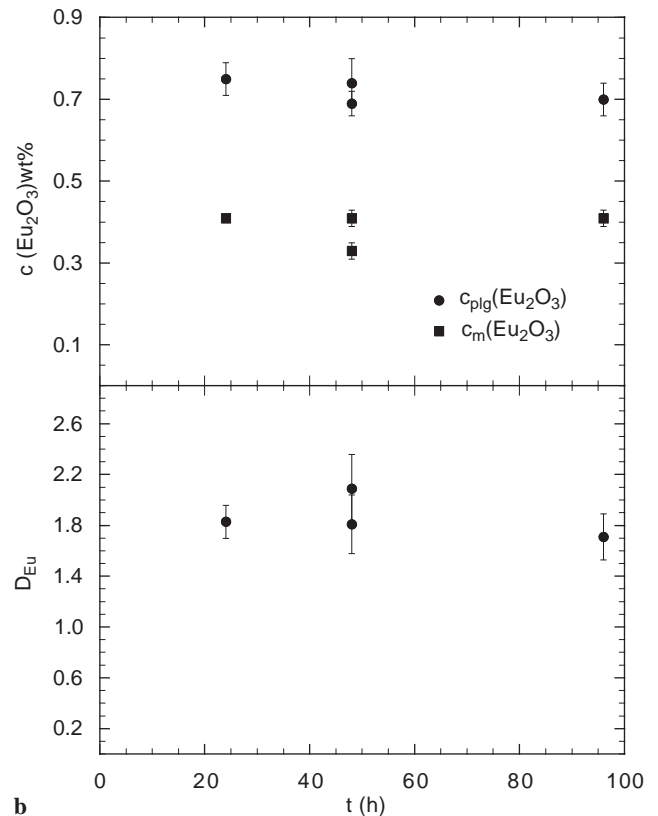
sen. The capsule was directly heated to the run temperature whereas the others were held at higher temperature at the beginning of the experiment. This implies that the temperature has large influence on the compaction of the BN and thus on its permeability for hydrogen. Although there is a considerable variation in the H/O ratio and hydrogen fugacity, the resulting oxygen fugacity is quite similar for all run duration. The  $\text{Eu}_2\text{O}_3$  contents of the coexisting phases and the resulting partition coefficients do not vary significantly with run duration. This indicates that the  $f_{\text{O}_2}$  was similar for all runs and osmotic equilibrium is reached even for the shortest run duration.

A single run was conducted using anthracene ( $\text{C}_{14}\text{H}_{10}$ ) as sensor material. As this material is virtually free of oxygen the only oxygen source available is the air enclosed during capsule preparation and water adsorbed on the powder and the capsule wall. In a 48 h experiment at 500 MPa and 850 °C only  $\text{H}_2\text{O}$  was detected as an oxygen bearing fluid species, however, the amount of  $\text{H}_2\text{O}$  was near the detection limit. The H/O-ratio of 798 corresponds to a  $f_{\text{H}_2}$  of 933 bar for the sensor capsule and  $\log f_{\text{O}_2} = -15.93$  for the sample capsule which is only slightly more reducing than by using  $\text{C}_{14}\text{H}_{22}\text{O}$ . The lowest hydrogen fugacity which can be achieved theoretically by using organic sensor materials is defined by the graphite-methane buffer.

The mechanism by which the permeability of BN to hydrogen is reduced is not known. In runs using

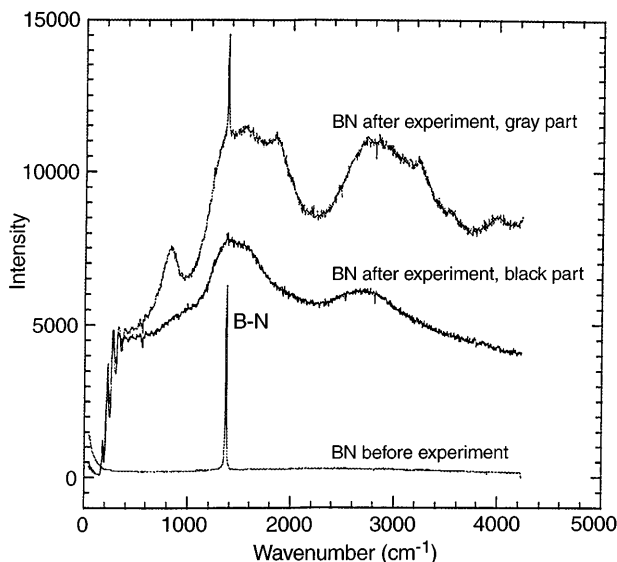


**Fig. 2a, b** Kinetic experiments at 850 °C and 500 MPa using the BN-jacket technique. **a** Development of H/O ratio,  $f_{\text{H}_2}$  in the sensor capsule and calculated  $f_{\text{O}_2}$  in the sample capsule with run duration. The oxygen fugacity in the sample capsule is calculated assuming osmotic equilibrium between sample and sensor capsule and Lewis–Randall mixing of the fluid components in the sample capsule. **b**  $\text{Eu}_2\text{O}_3$  contents of melt and plagioclase and the resulting partition coefficients versus run duration. Note that both Eu concentrations in the coexisting phases and the partition coefficient show very little variation implying osmotic equilibrium is reached even for the shortest run duration



$C_9H_{10}O_2$  having a low initial H/O we could not detect any hydrogen in the sensor capsule after the run, indicating a high hydrogen permeability of the BN. This implies that the structure of BN is modified by chemical reaction with hydrogen. To investigate the nature of this reaction we used X-ray powder diffraction and Raman spectroscopy. X-ray diffraction analysis on the bulk of the used BN shows no changes compared to the starting material. However, by Raman microspectroscopy the used BN jacket was found to be heterogeneous (Fig. 3). Raman spectra collected by J.M. Beny at the CNRS-CSCM in Orleans (France) using a Dilor XY confocal micro-Raman spectrometer display different features in the centre of the BN jacket, which was light grey, and on the outer rim, which was black. The peak at  $1365\text{ cm}^{-1}$  of crystalline BN is still visible for the inner part of the BN but is absent for the outer part. In both parts new bands appear at  $\approx 1600\text{ cm}^{-1}$  and  $\approx 2800\text{ cm}^{-1}$ . Additional bands at  $\approx 830\text{ cm}^{-1}$  and  $\approx 1800\text{ cm}^{-1}$  are visible in the spectrum of the inner part. Despite the origin of the new bands is not well known, the spectra clearly demonstrate alteration of the BN during the experiments. We suggest that the formation of new phases by reaction with hydrogen favours the compaction of BN resulting in a low hydrogen permeability.

It might be expected that the compaction of the BN results in pressure shielding of the sample capsule. Thus the effective pressure on the sample might be lower than the external pressure. However, water contents of glasses, estimated by the totals of microprobe analyses, are in good agreement for experiments using BN jackets and other experiments (mean totals:  $89.11 \pm 1.31$  for BN runs compared to  $88.12 \pm 0.74$  for the other runs). The estimated water contents are close to water solu-



**Fig. 3** Raman Spectra recorded on the central, grey part and the outer, black part of an used BN jacket. A spectrum of the original BN is shown for comparison. The sharp peak at  $1365\text{ cm}^{-1}$  of crystalline BN vanishes and several broad bands appear. For details see text

bilities determined at  $1200\text{ }^\circ\text{C}$  and  $500\text{ MPa}$  for tonalitic melts by Grams and Behrens (1996). Therefore we infer that the differences in pressure of the experiments are below  $50\text{ MPa}$ .

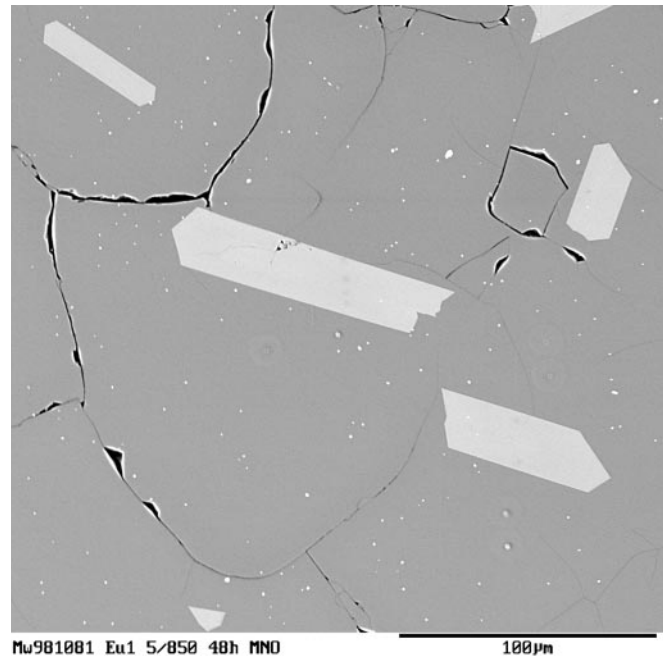
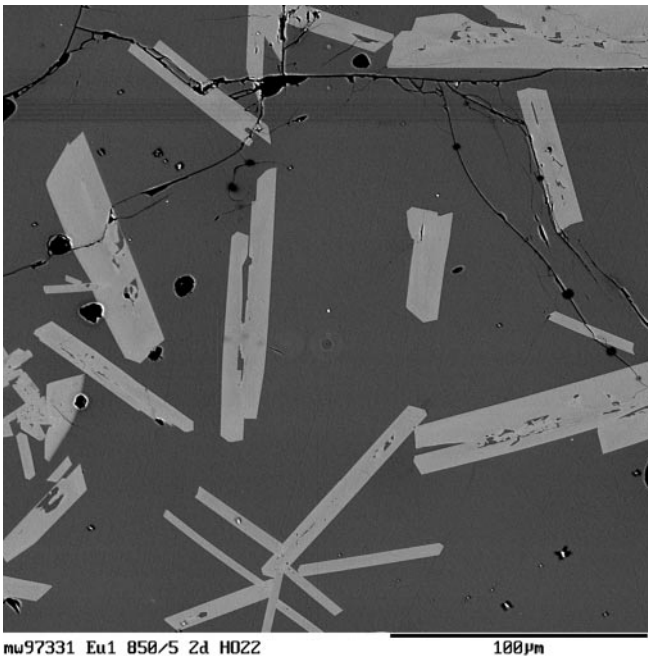
#### Reaction kinetics and achievement of chemical equilibrium

Figure 4a shows a BSE image of a charge run at  $850\text{ }^\circ\text{C}/500\text{ MPa}$  using a rate of  $1^\circ/\text{min}$  for cooling from liquidus to crystallisation temperature ( $900^\circ \leftrightarrow 850^\circ\text{C}$ ). The crystals contain some melt inclusions and have skeletal shapes, indicating rapid, disequilibrium growth. Formation of melt inclusions in the plagioclase and skeletal crystal growth are negligible with a cooling rate of  $0.5^\circ/\text{min}$  (Fig. 4b), so this cooling rate was used in the subsequent runs. The phases show no systematic variations in either major or minor components, neither the crystals (rim and core) nor the glass (Table 2). From the euhedral crystal shape and the absence of zoning in the crystals we conclude that the plagioclase was formed by near equilibrium growth.

Kinetic parameters which might influence the partitioning of the heterovalent cations are (i) redox equilibration of the sample with the buffer, (ii) nucleation and growth of crystals and (iii) diffusion of components in the melt.

Permeation of hydrogen through the capsule wall is fast even at  $750\text{ }^\circ\text{C}$ . The times to transport the amount of hydrogen necessary to set up the redox conditions in the sample capsule calculated after Chou (1986) are in the range of  $30\text{--}70\text{ s}$  for Pt capsules and NNO buffer. Even for Au the times are in the range of only  $300\text{--}700\text{ s}$ . Transport rates are not well known for hydrogen and oxygen in hydrous silicate melts. Experimental observations on  $H_2$  and  $H_2O$  bearing glasses indicate that the diffusivity of hydrogen is orders of magnitude higher than the diffusivity of water (Schmidt 1996; Gaillard et al. 1998). This suggests that the redox conditions are established throughout the sample within a few minutes.

Time dependent experiments have shown continuous growth of plagioclase over several days at  $750\text{ }^\circ\text{C}$  and  $850\text{ }^\circ\text{C}$ . At  $850\text{ }^\circ\text{C}$  using the retrograde temperature path with a slower cooling rate ( $0.5\text{ }^\circ\text{C}/\text{min}$ ) plagioclase crystals with a size of  $10\text{ }\mu\text{m}$  are found even in runs quenched directly after cooling to the crystallisation temperature. These crystals have nucleated and grown during the cooling period (100 min) preferentially close to the surface of the melt body. The An content is similar to those of the 48 h runs ( $An_{73}$ ). However, the partition coefficient for Eu is significantly lower in short term experiments and increases with run duration (for intrinsic conditions:  $D_{Eu} = 0.38$  directly after reaching run temperature,  $D_{Eu} = 0.50$  after 4 h,  $D_{Eu} = 0.77$  after 48 h). The partition coefficients at the beginning are too small to be explained by initially higher temperature during crystallisation. We suggest that the low partition coeffi-



**Fig. 4a, b** Back scattered electron images of run products at 850 °C and 500 MPa. *Dark areas* melt, *bright areas* plagioclase. **a** (left) Temperature path: 30 °C/min to 900 °C – 30 min at 900 °C – 1 °C/min to 850 °C – 48 h at 850 °C. **b** (right) 30 °C/min to 900 °C – 30 min at 900 °C – 0.5 °C/min to 850 °C – 48 h at 850 °C

cients result from more oxidising conditions prevailing in the melt at the beginning of the experiment. This is in accordance with observations made by Gaillard et al. (1998), who found that the change in the redox state is not as fast as expected from the rapid hydrogen diffusion in the melt. A possible explanation is that the solubility of hydrogen in the melt is small and thus the transport rate of hydrogen (which is controlling the redox reaction) is small. The time dependent experiments at the most reducing conditions demonstrate that equilibrium partitioning is achieved for run durations  $\geq 24$  h. No relicts of the initially formed plagioclase with low Eu contents could be found in the long duration runs.

At 750 °C plagioclase is firstly observed after 60 min using a retrograde temperature path (30 min at 830 °C – 3 °C/min to 750 °C). Even after 4 h the crystals were too small ( $< 2 \mu\text{m}$ ) for microprobe analysis. The maximum size after 4 days was about  $80 \times 30 \mu\text{m}$ . No systematic investigations were performed to prove equilibrium partitioning for this temperature because of the slow crystallisation kinetics. We assume that a run duration of 4 days is sufficient to achieve near equilibrium conditions.

#### Partitioning of Eu and Sm

Compositions of plagioclase and melt and the calculated partition coefficients and major component compositions are shown in Table 2 and 3. At 850 °C the  $D$  value for Eu (Fig. 5a) varies in the investigated  $f_{\text{O}_2}$  range from

0.095 for oxidising conditions (Cu—Cu<sub>2</sub>O buffer) to 1.81 for the most reducing conditions (C—O—H sensor). The  $D_{\text{Sm}}$  does not vary with oxygen fugacity and has an average value of 0.07. Using the Cu—Cu<sub>2</sub>O buffer Eu and Sm have similar partition coefficients. It can be expected that more oxidising conditions give no significant changes for the Eu partitioning. Decreasing the temperature to 750 °C for a given  $f_{\text{O}_2}$  the partition coefficient of Eu is strongly decreased at intermediate oxygen fugacities (Fig. 5a), e.g. by a factor of 2.0 for  $\log f_{\text{O}_2} = -11$ . It is emphasised that for a given buffer condition the effect of temperature is small as the equilibrium  $f_{\text{O}_2}$  of the buffer also changes with temperature. At MnO—Mn<sub>3</sub>O<sub>4</sub> the partition coefficient is even independent on temperature. This insensitivity of  $D_{\text{Eu}}$  to temperature on a relative  $f_{\text{O}_2}$  scale (e.g.  $\Delta\text{NNO} = \log(f_{\text{O}_2}/f_{\text{O}_2\text{NHO}})$ ) is important for the geological application of the experimental data since the temperature of formation of natural rocks is often not known very precisely.

$D_{\text{Eu}}$  and  $D_{\text{Sm}}$  determined by Weill and McKay (1975) in 1 atm experiments at 1200 °C in lunar basaltic rocks are similar to our data for a given  $f_{\text{O}_2}$  (Fig. 5a). For terrestrial basaltic and andesitic compositions Drake and Weill (1975) found higher partition coefficients for REE than for lunar basalts (e.g. for Sm by a factor 2.5 at 1200 °C). The differences are attributed by Weill and McKay (1975) to the composition of the melt. Partitioning of Eu in terrestrial basalts and andesites at 1200 °C and 1 atm (Drake and Weill 1975) is for a given  $f_{\text{O}_2}$  higher than for hydrous tonalitic melts at 850 °C and 500 MPa. Using the Arrhenius equations determined by Drake and Weill (1975) for the  $T$  range 1150–1400 °C to calculate the partition coefficients of Sm and Eu at 850 °C gives  $D_{\text{Sm}} = 0.13$  and  $D_{\text{Eu}} = 0.10$  (in air). The extrapolated value for Sm is higher by approximately a

**Table 2** Compositions of run products from representative experiments. In parentheses 1s standard deviation is given. Equilibrium-oxygen fugacities of buffer are calculated using data of Robie and Hemingway (1995). Low sums of glass analyzes are due to approx. 11 wt% water dissolved

To1 (Eu) 500 MPa, 850 °C												
Sample	68	136	108	13	86	57						
Buffer	Cu—Cu <sub>2</sub> O	Fe <sub>2</sub> O <sub>3</sub> —Fe <sub>3</sub> O <sub>4</sub>	MnO—Mn <sub>3</sub> O <sub>4</sub>	Intrinsic	Ni—NiO	C—O—H Sensor						
log( <i>f</i> <sub>O<sub>2</sub></sub> /bar)	-7.27	-8.54	-9.00	-10.5	-12.74	-15.72						
Phase	Plg	Gls	Plg	Gls	Plg	Gls	Plg	Gls	Plg	Gls		
SiO <sub>2</sub>	49.40 (0.35)	65.90 (0.56)	50.40 (0.50)	65.21 (0.29)	50.64 (0.47)	64.49 (0.14)	50.84 (0.53)	65.15 (0.27)	49.20 (0.71)	64.61 (0.34)	50.77 (0.64)	68.06 (0.47)
Al <sub>2</sub> O <sub>3</sub>	32.36 (0.25)	14.40 (0.16)	31.43 (0.32)	14.45 (0.13)	31.27 (0.24)	14.86 (0.19)	31.88 (0.44)	14.59 (0.20)	31.95 (0.51)	14.32 (0.11)	31.67 (0.44)	14.22 (0.16)
CaO	15.52 (0.27)	5.11 (0.10)	14.99 (0.43)	5.15 (0.07)	14.45 (0.32)	5.66 (0.04)	15.08 (0.32)	5.41 (0.07)	15.07 (0.59)	5.04 (0.06)	14.55 (0.46)	4.92 (0.08)
Na <sub>2</sub> O	2.91 (0.23)	2.88 (0.06)	3.12 (0.24)	2.90 (0.15)	3.34 (0.21)	2.88 (0.02)	3.15 (0.21)	2.60 (0.14)	2.76 (0.22)	2.83 (0.14)	3.23 (0.18)	2.74 (0.11)
Eu <sub>2</sub> O <sub>3</sub>	0.05 (0.01)	0.36 (0.04)	0.079 (0.007)	0.35 (0.01)	0.11 (0.01)	0.37 (0.01)	0.37 (0.02)	0.48 (0.01)	0.36 (0.02)	0.39 (0.01)	0.70 (0.04)	0.41 (0.02)
Sm <sub>2</sub> O <sub>3</sub>	0.04 (0.01)	0.36 (0.07)	0.020 (0.005)	0.33 (0.01)	0.02 (0.01)	0.35 (0.02)	0.03 (0.01)	0.53 (0.03)	0.03 (0.01)	0.36 (0.01)	0.04 (0.01)	0.48 (0.02)
Total	100.28	89.01	100.04	88.39	99.83	88.61	101.35	88.76	99.37	87.55	100.96	90.83
Qz		43		43		40		44		43		46
Ab <sup>a</sup>	25	28	27	28	29	28	27	25	25	28	29	26
An <sup>a</sup>	75	29	73	29	71	32	73	31	75	29	71	27
C												
<i>D</i> <sub>Eu</sub>	0.116 ± 0.028		0.226 ± 0.026		0.297 ± 0.035		0.771 ± 0.058		0.92 ± 0.075		1.71 ± 0.18	
<i>D</i> <sub>Sm</sub>	0.098 ± 0.017		0.061 ± 0.017		0.057 ± 0.032		0.057 ± 0.022		0.08 ± 0.030		0.083 ± 0.024	
To1 (Fe) 500 MPa, 850 °C												
Sample	77	92	109	98	107	66						
Buffer	Cu—Cu <sub>2</sub> O	Fe <sub>2</sub> O <sub>3</sub> —Fe <sub>3</sub> O <sub>4</sub>	MnO—Mn <sub>3</sub> O <sub>4</sub>	Intrinsic	Ni—NiO	C—O—H Sensor						
log( <i>f</i> <sub>O<sub>2</sub></sub> /bar)	-7.27	-8.54	-9.00	-10.5	-12.74	-15.88						
Phase	Plg	Gls	Plg	Gls	Plg	Gls	Plg	Gls	Plg	Gls		
SiO <sub>2</sub>	51.02 (1.21)	66.77 (0.64)	50.50 (0.24)	65.04 (0.15)	50.73 (0.37)	62.45 (0.43)	49.98 (0.41)	64.32 (0.17)	50.35 (0.20)	65.35 (0.30)	49.53 (0.48)	66.40 (0.36)
Al <sub>2</sub> O <sub>3</sub>	31.05 (0.69)	13.92 (0.12)	31.47 (0.19)	14.74 (0.24)	30.90 (0.27)	15.06 (0.05)	31.72 (0.43)	14.30 (0.14)	31.36 (0.15)	15.11 (0.08)	32.25 (0.28)	14.12 (0.18)
CaO	14.40 (0.69)	4.96 (0.05)	14.83 (0.16)	5.39 (0.08)	14.53 (0.20)	6.59 (0.04)	15.32 (0.43)	5.44 (0.12)	14.64 (0.17)	5.60 (0.11)	15.11 (0.37)	5.16 (0.17)
Na <sub>2</sub> O	3.42 (0.45)	2.39 (0.23)	2.95 (0.08)	2.43 (0.10)	3.32 (0.05)	2.51 (0.17)	3.38 (0.29)	2.63 (0.08)	3.31 (0.06)	2.32 (0.10)	2.86 (0.19)	2.22 (0.19)
Fe <sub>2</sub> O <sub>3,tot</sub>	0.41 (0.08)	1.00 (0.10)	0.39 (0.05)	0.84 (0.04)	0.32 (0.05)	0.92 (0.04)	0.17 (0.01)	0.94 (0.06)	0.10 (0.01)	0.87 (0.04)	0.08 (0.01)	0.94 (0.05)
Total	100.31	89.04	100.14	88.44	99.80	87.53	100.57	87.63	99.76	89.25	99.83	88.84
Qz		48		45		39		43		45		48
Ab <sup>a</sup>	30	23	27	23	29	24	29	26	29	22	26	21
An <sup>a</sup>	70	28	73	31	71	37	71	31	71	31	74	29
C												
<i>D</i> <sub>Fe</sub>	0.600 ± 0.138		0.46 ± 0.08		0.380 ± 0.027		0.181 ± 0.022		0.118 ± 0.015		0.085 ± 0.013	

To2 (Eu) 500 MPa, 750 °C											
Sample	156	120	111	151	85						
Buffer	Cu—Cu <sub>2</sub> O	Fe <sub>2</sub> O <sub>3</sub> —Fe <sub>3</sub> O <sub>4</sub>	MnO—Mn <sub>3</sub> O <sub>5</sub>	Ni—NiO	Co—CoO						
log( <i>f</i> <sub>O<sub>2</sub></sub> /bar)	-8.74	-10.90	-10.96	-14.84	-16.15						
Phase	Plg	Gls	Plg	Gls	Plg	Gls	Plg	Gls	Plg	Gls	
SiO <sub>2</sub>	51.20 (0.49)	67.33 (0.72)	51.03 (0.23)	67.13 (0.20)	50.75 (0.26)	66.44 (0.22)	52.36 (0.48)	67.80 (0.21)	52.64 (0.64)	68.40 (0.35)	
Al <sub>2</sub> O <sub>3</sub>	31.08 (0.09)	12.97 (0.18)	31.43 (0.28)	13.08 (0.11)	30.84 (0.26)	12.56 (0.13)	30.29 (0.32)	12.43 (0.05)	29.60 (0.47)	12.38 (0.14)	
CuO	14.31 (0.08)	4.89 (0.07)	14.55 (0.19)	5.41 (0.04)	14.48 (0.15)	5.37 (0.05)	13.31 (0.35)	4.56 (0.06)	12.47 (0.44)	4.96 (0.05)	
Na <sub>2</sub> O	3.44 (0.10)	1.96 (0.08)	3.43 (0.13)	1.91 (0.09)	3.22 (0.10)	1.54 (0.16)	3.80 (0.17)	2.34 (0.09)	4.01 (0.28)	1.87 (0.12)	
Eu <sub>2</sub> O <sub>3</sub>	0.09 (0.02)	0.57 (0.01)	0.09 (0.01)	0.30 (0.01)	0.13 (0.01)	0.45 (0.02)	0.73 (0.03)	0.64 (0.02)	0.85 (0.08)	0.64 (0.01)	
Total	100.12	87.73	100.44	87.83	99.78	86.36	100.49	87.77	99.57	88.25	
Qz		52		51		54		51		54	
Ab <sup>a</sup>	30	19	30	18	29	15	34	23	37	18	
An <sup>a</sup>	70	28	70	31	71	31	66	26	63	28	
C		1									
<i>D</i> <sub>Eu</sub>	0.158 ± 0.038		0.300 ± 0.043		0.289 ± 0.035		1.18 ± 0.09		1.44 ± 0.16		

To2 (Fe) 500 MPa, 750 °C											
Sample	157	137	114	150	83						
Buffer	Cu—Cu <sub>2</sub> O	Fe <sub>2</sub> O <sub>3</sub> —Fe <sub>3</sub> O <sub>4</sub>	MnO—Mn <sub>3</sub> O <sub>5</sub>	Ni—NiO	Co—CoO						
log( <i>f</i> <sub>O<sub>2</sub></sub> /bar)	-8.74	-10.90	-10.96	-14.84	-16.15						
Phase	Plg	Gls	Plg	Gls	Plg	Gls	Plg	Gls	Plg	Gls	
SiO <sub>2</sub>	52.33 (0.51)	67.83 (0.38)	51.75 (0.60)	67.45 (0.11)	53.58 (0.20)	66.84 (0.31)	51.45 (0.30)	67.94 (0.20)	52.56 (0.28)	67.18 (0.17)	
Al <sub>2</sub> O <sub>3</sub>	30.38 (0.23)	12.72 (0.18)	30.09 (0.19)	12.54 (0.15)	28.96 (0.13)	12.68 (0.11)	31.05 (0.32)	12.73 (0.07)	30.17 (0.18)	12.88 (0.08)	
CaO	14.05 (0.23)	5.25 (0.09)	13.68 (0.25)	4.88 (0.05)	12.51 (0.18)	4.93 (0.04)	14.32 (0.23)	4.91 (0.08)	13.15 (0.15)	4.89 (0.07)	
Na <sub>2</sub> O	3.75 (0.12)	1.89 (0.21)	3.74 (0.20)	1.85 (0.17)	4.31 (0.04)	1.34 (0.29)	3.53 (0.16)	1.57 (0.29)	4.09 (0.10)	1.98 (0.16)	
Fe <sub>2</sub> O <sub>3,tot</sub>	0.53 (0.03)	0.67 (0.08)	0.46 (0.04)	0.88 (0.03)	0.42 (0.04)	1.00 (0.07)	0.23 (0.04)	1.61 (0.06)	0.14 (0.01)	1.41 (0.07)	
Total	101.04	88.37	99.72	87.60	99.78	86.79	100.58	88.76	100.11	88.34	
Qz		52		53		56		56		52	
Ab <sup>a</sup>	33	18	33	18	38	13	31	15	36	18	
An <sup>a</sup>	67	30	67	28	62	29	69	28	64	29	
C		1		1		2		1		1	
<i>D</i> <sub>Fe</sub>	0.791 ± 0.139		0.523 ± 0.063		0.420 ± 0.069		0.143 ± 0.030		0.099 ± 0.012	0.100 ± 0.018	

<sup>a</sup> Feldspar components are in mol% for crystals and in wt% for glass.



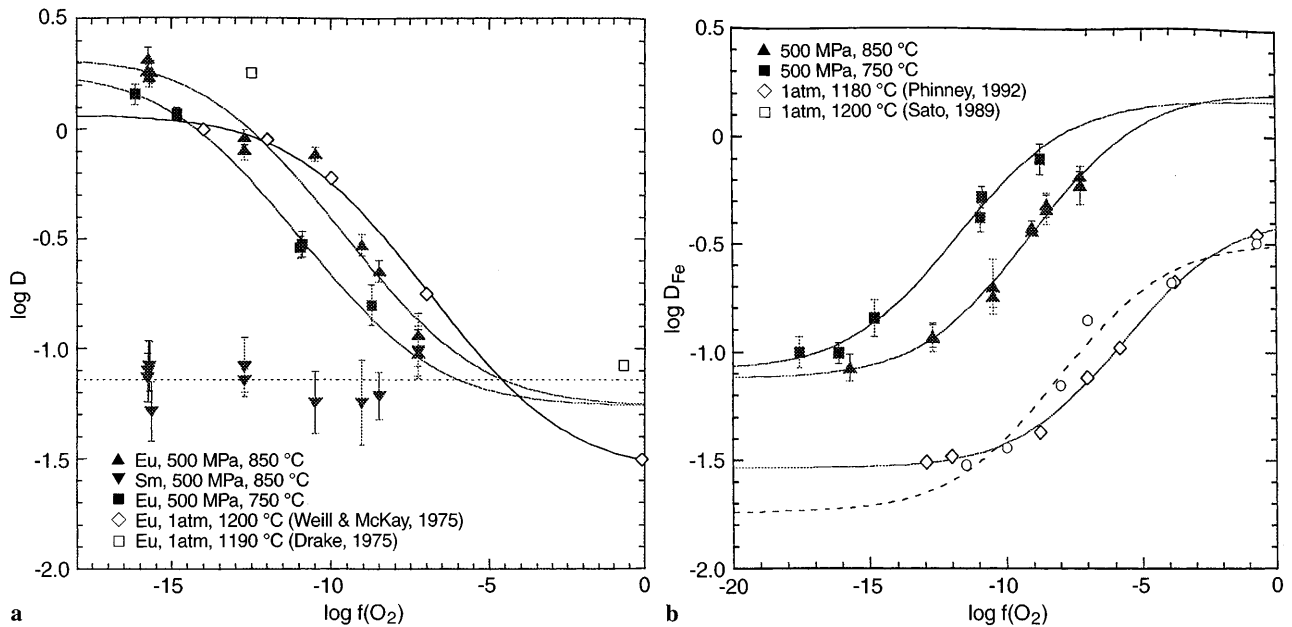
factor of 2 than our experimental value. For Eu, however, the predicted value is quite similar to our data measured at most oxidising conditions.

### Partitioning of Fe

In the investigated  $f_{O_2}$  range the variation of element partitioning with  $f_{O_2}$  is significantly smaller for Fe than for Eu (Fig. 5b). At 850 °C and 500 MPa  $D_{Fe}$  for partitioning between plagioclase and hydrous tonalitic melt varies from 0.08 at C—O—H sensor to 0.55 at Cu—Cu<sub>2</sub>O buffer. In contrast to the case for Eu, lowering the temperature to 750 °C increases the partition coefficient of Fe for a given oxygen fugacity, e.g. by a factor of 2.2 for  $\log f_{O_2} = -11$ . As observed for Eu, in the experimental range the effect of temperature is more pronounced at intermediate oxygen fugacities. Furthermore for a given buffer the effect of temperature is relatively small also for  $D_{Fe}$ .

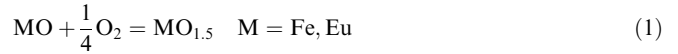
Compared to high temperature experiments in basaltic systems of Sato (1989) and Phinney (1992) our data are shifted to higher  $D$  values for a given  $f_{O_2}$  (Fig. 5b). From the experimental data available so far we cannot clarify whether these differences are due to temperature or melt composition.

**Fig. 5a, b** Variation of the partition coefficients of Eu and Sm (a, left) and Fe (b, right) with  $\log f_{O_2}$ . For comparison data on Eu partitioning (Weill and McKay 1975; Drake 1975) and Fe partitioning (Sato 1989; Phinney 1992) for basaltic compositions at 1180–1200 °C and 1 atm are shown. Curves were calculated using Eq. 7 and the fit parameters in Table 4 (for details see text)



### Modelling the partitioning data

The partitioning data can be modelled using the redox equilibrium of the heterovalent cations in the melt and assuming independent partitioning for divalent and trivalent cations. The redox reaction between divalent and trivalent cations in the melt can be written as:



The equilibrium constant  $K_m$  for this reaction is,

$$K_m = \frac{a_{MO_{1.5,m}}}{a_{MO,m} \cdot f_{O_2}^{1/4}} \quad (2)$$

where  $a$  denotes the activity of the respective components in the melt. Using  $a_i = \gamma_i x_i$  ( $\gamma_i$  = activity coefficient;  $x_i$  = mole fraction of component  $i$ ) and assuming  $\gamma_{MO} = \gamma_{MO_{1.5}}$  we can rewrite the equilibrium reaction on the basis of mole fractions:

$$K_m = \frac{(x_{MO_{1.5}}/x_{MO})_m}{f_{O_2}^{1/4}} \quad (3)$$

Partition coefficients usually are given on the basis of weight percent. Rewriting Eq. 3 in terms of mass fractions of the heterovalent cations (given in brackets) results in:

$$K_m = \frac{[M^{3+}]_m/[M^{2+}]_m}{f_{O_2}^{1/4}} \quad (4)$$

The partition coefficient can be written as follows:

$$D_M = \frac{[M^{2+}]_{plg} + [M^{3+}]_{plg}}{[M^{2+}]_m + [M^{3+}]_m} = \frac{[M^{2+}]_{plg} + [M^{3+}]_{plg}}{1 + \frac{[M^{3+}]_m}{[M^{2+}]_m}} \quad (5)$$

Combining Eqs. 5 and 4 gives

**Table 3** Average concentrations of Eu, Sm and Fe for plagioclase and glass and the derived partition coefficients of all experiments shown in the figures

Exp. No.	$T$ (°C)	$\log f_{O_2}$	Eu <sub>2</sub> O <sub>3</sub> Plg	Eu <sub>2</sub> O <sub>3</sub> Gls	Sm <sub>2</sub> O <sub>3</sub> Plg	Sm <sub>2</sub> O <sub>3</sub> Gls	$D_{Eu}$	$D_{Sm}$
68	850	-7.27	0.05 (0.01)	0.36 (0.04)	0.04 (0.01)	0.36 (0.07)	0.116 ± 0.028	0.098 ± 0.029
59	850	-7.27	0.04 (0.01)	0.42 (0.02)	0.040 (0.005)	0.41 (0.01)	0.095 ± 0.028	0.098 ± 0.017
136	850	-8.54	0.079 (0.007)	0.35 (0.01)	0.020 (0.005)	0.33 (0.01)	0.226 ± 0.026	0.061 ± 0.017
108	850	-9.00	0.11 (0.01)	0.37 (0.01)	0.02 (0.01)	0.35 (0.02)	0.297 ± 0.035	0.057 ± 0.032
13	850	-10.50	0.37 (0.02)	0.48 (0.01)	0.03 (0.01)	0.53 (0.03)	0.771 ± 0.058	0.057 ± 0.022
86	850	-12.74	0.36 (0.02)	0.39 (0.01)	0.03 (0.01)	0.36 (0.01)	0.920 ± 0.075	0.083 ± 0.030
87	850	-12.74	0.36 (0.02)	0.45 (0.02)	0.036 (0.003)	0.50 (0.02)	0.810 ± 0.089	0.072 ± 0.009
57	850	-15.71	0.70 (0.04)	0.41 (0.02)	0.04 (0.01)	0.48 (0.02)	1.71 ± 0.18	0.083 ± 0.024
33	850	-15.65	0.74 (0.06)	0.41 (0.02)	0.03 (0.01)	0.58 (0.02)	1.81 ± 0.23	0.052 ± 0.019
48	850	-15.75	0.69 (0.03)	0.33 (0.02)	0.03 (0.01)	0.38 (0.02)	2.09 ± 0.27	0.079 ± 0.030
58	850	-15.78	0.75 (0.04)	0.41 (0.01)	0.04 (0.01)	0.54 (0.02)	1.83 ± 0.13	0.074 ± 0.021
156	750	-8.74	0.09 (0.02)	0.57 (0.01)			0.158 ± 0.038	
120	750	-10.90	0.09 (0.01)	0.30 (0.01)			0.300 ± 0.043	
111	750	-10.96	0.13 (0.01)	0.45 (0.02)			0.289 ± 0.035	
151	750	-14.84	0.73 (0.03)	0.64 (0.02)			1.18 ± 0.09	
85	750	-16.15	0.64 (0.01)	0.85 (0.08)			1.44 ± 0.16	

Exp. No.	$T$ (°C)	$\log f_{O_2}$	Fe <sub>2</sub> O <sub>3,tot</sub> Plg	Fe <sub>2</sub> O <sub>3,tot</sub> Gls	$D_{Fe}$
60	850	-7.27	0.46 (0.02)	0.70 (0.01)	0.660 ± 0.038
77	850	-7.27	0.41 (0.08)	1.00 (0.10)	0.600 ± 0.138
92	850	-8.54	0.39 (0.05)	0.84 (0.04)	0.460 ± 0.080
110	850	-8.54	0.82 (0.07)	1.69 (0.09)	0.485 ± 0.067
123	850	-9.00	0.62 (0.02)	1.78 (0.05)	0.365 ± 0.018
109	850	-9.00	0.32 (0.05)	0.92 (0.04)	0.380 ± 0.027
90	850	-10.50	0.18 (0.04)	0.88 (0.11)	0.200 ± 0.068
98	850	-10.50	0.17 (0.01)	0.94 (0.06)	0.181 ± 0.022
107	850	-12.74	0.10 (0.01)	0.87 (0.04)	0.118 ± 0.015
99	850	-12.74	0.11 (0.01)	0.94 (0.07)	0.117 ± 0.019
66	850	-15.74	0.08 (0.01)	0.94 (0.05)	0.085 ± 0.013
157	750	-8.74	0.53 (0.03)	0.67 (0.08)	0.791 ± 0.139
137	750	-10.90	0.46 (0.04)	0.88 (0.03)	0.523 ± 0.063
114	750	-10.96	0.42 (0.04)	1.00 (0.07)	0.420 ± 0.069
150	750	-14.84	0.23 (0.04)	1.61 (0.06)	0.143 ± 0.030
83	750	-16.15	0.14 (0.01)	1.41 (0.07)	0.099 ± 0.012
148	750	-17.61	0.14 (0.02)	1.40 (0.05)	0.100 ± 0.018

$$D_M = \frac{D_{M^{2+}} + D_{M^{3+}} \cdot K_m \cdot f_{O_2}^{1/4}}{1 + K_m \cdot f_{O_2}^{1/4}} \quad (6)$$

where  $D_{M^{3+}}$ , and  $D_{M^{2+}}$  denote the partition coefficients of the two cations.

Equation 6 describes the dependence of the partition coefficient on oxygen fugacity. The three parameters  $D_{M^{2+}}$ ,  $D_{M^{3+}}$  and  $K_m$  were determined by fitting the partitioning data using a plot of  $\log D$  versus  $\log f_{O_2}$ .

$$\log D_M = \log \frac{D_{M^{2+}} + D_{M^{3+}} \cdot K_m \cdot 10^{1/4 \log f_{O_2}}}{1 + K_m \cdot 10^{1/4 \log f_{O_2}}} \quad (7)$$

It is emphasised that Eqs. 5 and 6 only are valid if the activity coefficients of the two melt components MO and MO<sub>1.5</sub> are identical but does not necessarily require ideal mixing of components in the melt. As the composition of coexisting melt and plagioclase does not change in the investigated  $f_{O_2}$ -range the activities of the components in melt and crystals are constant, and therefore  $D_{M^{2+}}$  and  $D_{M^{3+}}$  do not vary with  $f_{O_2}$ . Thus, the  $f_{O_2}$ -dependence of Eu and Fe partition coefficients are expected to be

controlled by the redox ratio of the elements in the melt, only.

In Figs. 5a and 5b the experimentally derived data are shown together with the fitted functions for Eu and Fe at 850 °C, respectively. For both elements the fit is poorly constrained for very oxidising conditions. Without extra constraints the fit of the Eu data extends to negative values for  $D_{Eu^{3+}}$ , a nonsensical result. Unfortunately, all attempts to perform experiments at very oxidising conditions (Mn<sub>3</sub>O<sub>4</sub>—Mn<sub>2</sub>O<sub>3</sub>) failed due to rapid consumption of the buffer material. Therefore the  $D_{Eu^{3+}}$  value was estimated using the measured  $D$  for the Sm partitioning. The ratio of the partition coefficients for Sm and Eu was derived by extrapolating the data given by Drake and Weill (1975) ( $D_{Sm}/D_{Eu} = 1.3$  at 850 °C). Thus for the fit of the Eu data  $D_{Eu^{3+}}$  was set to 0.055. At 750 °C the partition coefficient for Sm was not determined experimentally. But as the partition coefficient of Sm at 950 °C does not differ from the one at 850 °C within the error of the analyses (Wilke 1999) the data were fitted assuming a  $T$  independent  $D$  value for

$\text{Eu}^{3+}$ . With these constraints the experimental data of Eu are well reproduced except for one outlying value at 850 °C and  $\log f_{\text{O}_2} = -10.5$ , probably because  $f_{\text{O}_2}$  was not well defined in this unbuffered experiment. The results of modelling the partitioning data imply that  $D_{\text{Eu}^{2+}}$  increases whereas  $K_{\text{m, Eu}}$  decreases with increasing temperature (Table 4).

Experimental Fe partitioning data are well fitted without any constraint on  $D_{\text{Fe}^{3+}}$  and  $D_{\text{Fe}^{2+}}$ . The values for  $D_{\text{Fe}^{3+}}$  and  $D_{\text{Fe}^{2+}}$  derived from the Fe partitioning data at 750 °C agree, within experimental error, with the values obtained at 850 °C indicating that the observed change in Fe partitioning with temperature results mainly from the decreasing  $K_{\text{m, Fe}}$  with increasing temperature. The observed  $T$  dependence of  $\text{M}^{3+}/\text{M}^{2+}$  is in accordance with results of previous studies on silicate melts in that the divalent cation is favoured at high temperature (e.g. Fudali 1965; Morris et al. 1974; Sack et al. 1980).

The same model was applied to the data for Fe partitioning (Sato 1989; Phinney 1992) and Eu partitioning (Weill and McKay 1975) between plagioclase and basaltic melts. Slightly different parameters are obtained for Fe partitioning in basaltic melts using the data of Sato (1989) and Phinney (1992), which might be attributed to different bulk compositions used in both studies. The starting glass used by Phinney has higher FeO but lower MgO and CaO contents. For both, Fe and Eu partitioning the resulting values for  $D_{\text{M}^{3+}}$  and  $D_{\text{M}^{2+}}$  as well as  $K_{\text{m}}$  are lower for basaltic melts at  $\approx 1200$  °C and 1 atm than for the tonalitic melts at 850 °C/500 MPa data. With the available data it is not possible to decide whether the differences of the parameters are due to melt composition, temperature, pressure or combinations of these.

### Redox state of the melt

$\text{M}^{3+}/\text{M}^{2+}$  ratios calculated from the partitioning data for the tonalitic melts at 750 and 850 °C and at 500 MPa are plotted in Fig. 6a. Due to larger scattering of the experimental data and to the constraints made for fitting, the redox ratios for Eu are less reliable than those for Fe. Nevertheless, it is evident that for a given  $f_{\text{O}_2}$  and temperature  $\text{Eu}^{3+}/\text{Eu}^{2+}$  is more than one order of magnitude larger than  $\text{Fe}^{3+}/\text{Fe}^{2+}$ .

When we compare the ferric-ferrous ratios determined from the partitioning data with those predicted by the empirical calculation model of Kress and Carmichael (1991), the absolute values are different (up to an order of magnitude at reducing conditions) as well as the  $f_{\text{O}_2}$  dependence (Fig. 6a). The ferric-ferrous ratios in the melts calculated from Fe partitioning are supported by preliminary results of  $^{57}\text{Fe}$ -Moessbauer spectroscopy on tonalitic glasses equilibrated at the same experimental conditions (Wilke and Behrens 1998). This implies that the calculation model of Kress and Carmichael (1991) must be revised for application to low temperature magmatic systems. On the other hand, a much better agreement of ferric-ferrous ratios extracted from partitioning data with those calculated from the Kress and Carmichael model is obtained using the high temperature data of Phinney (1992) and Sato (1989) (Fig. 6b). This suggests that the Kress and Carmichael model is more suitable for estimating ferric-ferrous ratios of magmatic melts at high temperature, however, our results imply that the  $f_{\text{O}_2}$  dependence might be larger than predicted by the empirical model.

### Application: an oxygen barometer for magmatic systems?

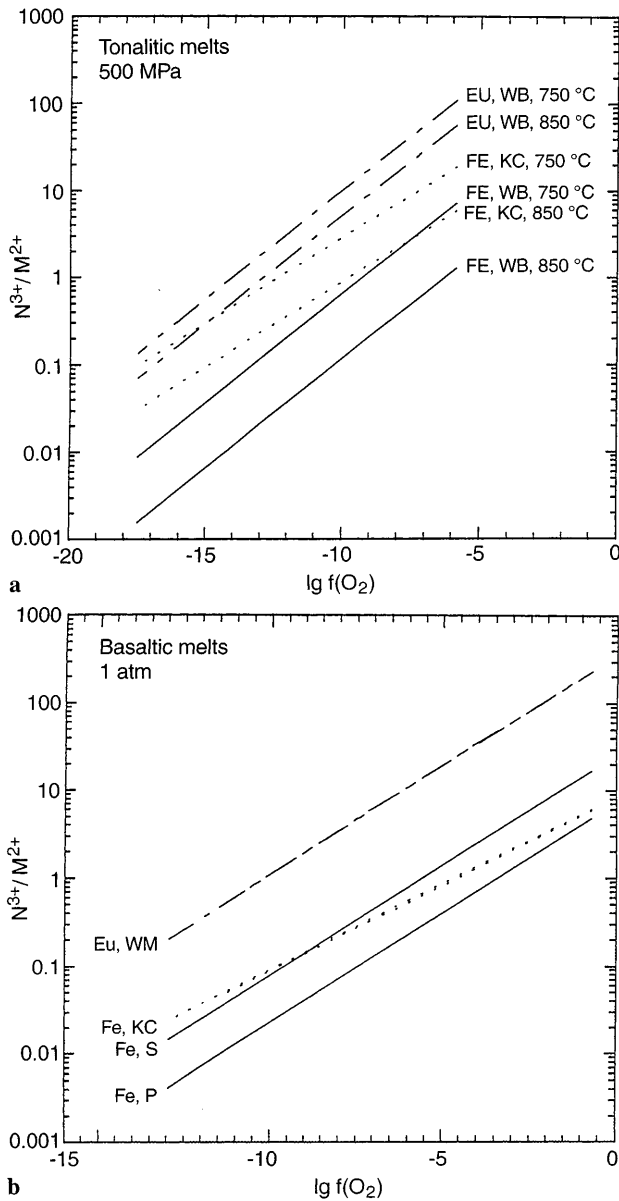
A strong dependence of Fe and Eu partitioning between plagioclase and hydrous tonalitic melts on  $f_{\text{O}_2}$  is observed between NNO and NNO+4. In this  $f_{\text{O}_2}$  range  $D_{\text{Fe}}$  increases by around 0.10 log units and  $D_{\text{Eu}}$  decreases by  $\sim 0.13$  log units per log unit of  $f_{\text{O}_2}$  at 850 °C and 500 MPa. This implies that Fe and Eu partitioning may be used as a sensitive oxygen barometer for magmatic processes. Pre-eruptive redox conditions in magma chambers have been estimated from Fe-Ti oxides and phase relationships between NNO+0.4 and NNO+0.8 for Mt. Pelee (Martel et al. 1998) and close to NNO+1.7 for Pinatubo (Evans and Scaillet 1997), i.e. in the range where the partitioning of Eu and Fe is very sensitive to  $f_{\text{O}_2}$ . Thus measurements on plagioclase and on glass inclusions in the plagioclase might further constrain the oxygen fugacity during the differentiation in the magma chamber.

Oxygen fugacities can be calculated from partition coefficients of Fe and Eu after rearrangement of Eq. 7:

**Table 4** Fit parameter and standard error (given in brackets) obtained for Eq. 7 using the datasets shown

	Eu this study 500 MPa/750 °C Tonalite	Eu this study 500 MPa/850 °C Tonalite	Eu Weill and McKay (1975) 1 atm, 1200 °C Basalt	Fe this study 500 MPa/750 °C Tonalite	Fe this study 500 MPa/850 °C Tonalite	Fe Phinney (1992) 1 atm, 1180 °C Basalt	Fe Sato (1989) 1 atm, 1200 °C Basalt
$D_{\text{M}^{2+}}$	1.85 (0.17)	2.14 (0.31)	1.16 (0.07)	0.083 (0.009)	0.078 (0.006)	0.029 (0.001)	0.0018 (0.006)
$D_{\text{M}^{3+}}$	0.055 <sup>a</sup>	0.055 <sup>a</sup>	0.028 (0.002)	1.47 (0.41)	1.7 (0.5)	0.42 (0.03)	0.32 (0.06)
$K_{\text{m}}$	3198 (489)	1650 (420)	349 (44)	207 (99)	37 (16)	7.20 (0.95)	25.0 (9.9)
$R^2$	0.993	0.961	0.999	0.992	0.991	0.998	0.978

<sup>a</sup> Fixed value.



**Fig. 6a**  $M^{2+}/M^{3+}$  ratios versus  $\log f_{O_2}$  for tonalitic melts at 500 MPa water pressure. Straight lines for Fe are calculated using the partitioning data (WB) and dotted lines using the empirical model of Kress and Carmichael (1991) (KC); dashed-dotted lines for Eu are calculated using partitioning data (WB).  $M^{2+}/M^{3+}$  ratios versus  $\log f_{O_2}$  for basaltic melts at 1 atm. Straight lines are calculated using Fe partitioning data of Phinney (1992) (P) and Sato (1989) (S). Redox ratios of Fe calculated using the empirical model of Kress and Carmichael (1991) for the two datasets (KC, dotted lines) are virtually identical for both basaltic compositions. Dashed-dotted lines are calculated from Eu partitioning data of Weill and McKay (1975) (WM)

$$\log f_{O_2} = 4 \cdot \log \frac{D_M - D_{M^{2+}}}{K_m \cdot (D_{M^{3+}} - D_M)} \quad (8)$$

In the case of tonalitic melts and temperatures between 750 and 850 °C the values of  $D_{M^{2+}}$ ,  $D_{M^{3+}}$  and  $K_m$  given in Table 3 may be directly used to calculate  $f_{O_2}$ . It is emphasised that the effect of temperature on Eu and Fe partitioning is small if oxygen fugacities relative to a

buffer are considered ( $\Delta NNO = \log(f_{O_2}/f_{O_2,NHO})$ ;  $\Delta FMQ = \log(f_{O_2}/f_{O_2,FMQ})$ ) and thus an exact knowledge of the temperature in the magma chamber is not required. In contrast melt composition may have a strong influence on the partitioning parameters. For both Fe and Eu,  $K_m$  is dependent on the Al/Si ratio and degree of polymerisation of the melt (Mysen 1991; Moeller and Muecke 1984). Thus for application of the potential oxygen barometer to other magmatic systems  $K_m$  must be determined for the melts of interest (e.g. dacite or rhyodacite).  $D_{M^{2+}}$  and  $D_{M^{3+}}$  are dependant on the activities of the components involved in the exchange reactions. For instance, substitution of  $Eu^{2+}$  for  $Ca^{2+}$  in the plagioclase depends not only on the EuO activity in the melt but also the activities of CaO in the melt and in the plagioclase. Therefore, a more detailed knowledge on the influence of chemical parameters on  $D_{M^{2+}}$ ,  $D_{M^{3+}}$  and  $K_m$  is required to use of Fe and Eu partitioning as an independent oxygen barometer. However, in combination with other methods (experimental determination of phase equilibria, analysis of compositions of Fe-Ti oxides in natural samples), element partitioning can give further constraints on the development of oxygen fugacity during magma genesis.

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