Interband and free charge carrier absorption in silicon at 800 nm: experiments and model calculations

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Abstract The average transmitted power of a Ti:sapphire femtosecond (fs) laser beam through thin, oxidized silicon (Si) membranes in the 10–30 μ m thickness range at 45 \degree external angle of incidence at frst increases with the incident average laser power and then decreases considerably. This stationary state behavior is quantitatively reproduced by a recently derived model using an efective absorption coefficient $\alpha_{\text{eff}} = \alpha_1 + \alpha_{\text{FCA}}$, adding the coefficients of two independent linear absorption processes, α_1 of the indirect interband transition and α_{ECA} of free charge carrier absorption (FCA). The relations between the empirical parameter α_{FCA} and the model parameters n_{eh} (density of free charge carriers) and σ_{eh} (absorption cross section of free charge carriers) are found to depend on the Si membrane thickness and are briefy discussed. Using diferent sample thicknesses, experimental evidence has been found for a nonlinear intensity dependence of $\alpha_{\text{FCA}} \sim I_0^n$, where $1 \le n \le 3$ shows

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Leibniz Institute of Photonic Technology, Albert-Einstein-Str. 9, 07745 Jena, Germany systematic variation. The FCA model in combination with its empirical ft parameters allowed to fully calculate the transmission behavior of the Si membranes within the investigated sample thickness and laser power regimes.

1 Introduction

Silicon (Si) continues to be a prominent material in microelectronics, optoelectronics, micromechanics, solar cells, and increasingly in photonics (cf., e.g., [[1\]](#page-8-0)). Delicate Si devices can be shaped and/or modifed by laser technology providing a manifold of remote, contactless, spatially confned and time controlled methods for, e.g., doping, annealing, crystallization, and ablation. Proper processing control requires, however, a detailed understanding of the linear and nonlinear optical phenomena in Si. Some of the nonlinear phenomena may occur simultaneously and be difficult to discriminate like, e.g., coherent two-photon absorption (TPA), free charge carrier absorption (FCA), and thermally induced absorption enhancement (TAE) [[2\]](#page-8-1).

FCA has a long history of more than half a century (cf., e.g., [\[3](#page-8-2)]), and was at frst investigated by the standard spectroscopy and conventional free charge carrier generation. With the advent of lasers, the contactless and well-confned photo generation of free charge carriers has become amenable, convenient, and increasingly important. The bulk part of the FCA studies is, however, related to FCA in the wavelength range $\lambda > 0.9 \,\mu$ m with small (negligible) interference from the indirect interband transition of Si at $\lambda \leq 1.13 \,\text{\textmu m}$ (1.1 eV optical band gap). Only a few papers address FCA in the spectral overlap region with the indirect and direct optical transitions (cf. $[2, 4-6]$ $[2, 4-6]$ $[2, 4-6]$ $[2, 4-6]$). They differ fundamentally from the studies at long wavelengths: the generation of free charge carriers is dominated by 1-photon and 2-photon absorption at short and long wavelengths, respectively.

In this paper, a femtosecond (fs) Ti:sapphire laser tuned to $\lambda = 800$ nm ($hv = 1.55$ eV) is applied to investigate the stationary state optical behavior of thin Si samples in the 10–30 μm thickness range by measuring the average laser power of the fs pulse train transmitted through the membranes as a function of the incident laser power (Fig. [1\)](#page-1-0). The main experimental fndings provide an essential extension of our recent results [[6](#page-8-4)] and address the membrane thickness *d*, the linear absorption coefficient α_1 , as well as the FCA coefficient α_{FCA} and its relations to the model parameters n_{eh} (density of free charge carriers) and σ_{eh} (absorption cross section of free charge carriers).

2 Experimental

For the transmission measurements through the Si membranes, a Ti:sapphire laser beam tuned to $\lambda = 800$ nm is applied to irradiate the Si samples at an external angle of incidence $\theta = 45^\circ$ (Fig. [1](#page-1-0)). The commercial fs oscillator (Spectra Physics, 3941-M3S, Tsunami) is pumped by a frequency doubled Nd:YVO₄ laser (Spectra Physics, Millenia V) with 5.5 W c.w. maximum output power and provides an average power up to 1 W at 80 MHz repetition rate. The pulse duration of 90 fs (fwhm) on the sample is determined by a custom built autocorrelator. The beam p-polarization is defned by a polarizing beam splitter (Thorlabs, Glan-laser polarizer, 10 mm, CA, AR coating: 650–1050 nm) combined with two half-wave plates (Thorlabs, zero-order half-wave

Fig. 1 Typical power dependences of the fs-laser beam refection from and transmission through a thin silicon membrane with the contributions of linear absorption (broken line) and free charge carrier absorption (FCA, indicated by arrow) obtained for a pulse train at 800 nm; inset: scheme of the experimental setup with external angle of incidence $\theta = 45^\circ$ and a propagation angle $\gamma = 11^\circ$ within silicon

plate, 780 nm) for power adjustment and polarization control. The beam is shaped by two lenses $(f = -50 \text{ mm and})$ $f = 35$ mm) and focused to a $1/e^2$ focal beam diameter of (11.2 ± 2) μm as determined by a z-scan measurement. The Rayleigh length amounts to $z_R = 0.12$ mm. The beam transmitted through the Si membrane is collected by a lens $f = 125$ mm. A power meter (Thorlabs, PMD100D) is used to measure the refected and transmitted powers.

The samples are prepared from commercial (100)-Si wafers $(370 \,\mu m)$ thick) by chemical etching of one wafer side using tetramethylammonium hydroxide (TMAH) to produce thin membranes of 3×3 mm² area and a thickness *d* of $10 \le d \le 30 \,\text{\mu m}$ (details cf. Sect. [3.1\)](#page-3-0). The wafers are slightly p-doped $(3 \dots 6 \times 10^{14} \text{ cm}^{-3})$. Prior to the investigations, the membranes are cleaned using acetone and 40% hydrofuoric acid to remove dirt and old oxide layers, respectively. In contact with the air, the clean surfaces oxidize in the dark under ambient conditions reaching within 48 h an equilibrium thickness of (2.5 ± 0.5) nm (cf., e.g., [[7\]](#page-8-5)). In addition to the membrane thickness determination in Sect. [3.1,](#page-3-0) independent optical thickness measurements were performed using a Zeiss microscope Axio Imager.M2m in two ways. In case of intact membranes, the membrane thickness is derived from the diference between the wafer thickness (370 μm specifed and measured) and the measured depth of the etched square (\approx 360 µm) with an uncertainty of about ± 5 µm. The thickness homogeneity of the individual Si membranes is determined to ± 1 µm. In case of membrane fragments, the thickness measurement by direct inspection of the membrane cross section is relatively precise around ± 0.2 µm, but does not refer to the membranes used for the transmission measurements. It is assumed that preferably thin membranes are broken. Overall, the microscopic thickness measurements are in agreement with the results obtained in Sect. [3.1](#page-3-0) (Fig. [3;](#page-2-0) Table [1](#page-3-1)).

3 Results and discussion

Laser beam refection always displays a linear dependence on the incident laser power P_{in} with $R = 0.15 \pm 0.01$ for the applied Si membranes without indicating a systematic thickness dependence (Fig. [2a](#page-2-1)). This fnding is in agreement with previous reports (cf., e.g., [\[8](#page-8-6), [9\]](#page-8-7)) and confrms the validity of the Fresnel equations using the optical constants of the sample under ambient conditions, i.e., refractive indices $n_{\text{air}} = 1$, $n_{SiO_2} = 1.45$ [[10](#page-8-8)], and $n_{Si} = 3.69$ [[11](#page-8-9), [12\]](#page-8-10) yielding $R = 0.155$.

Figure [2](#page-2-1)b shows the measured laser beam transmission through the membranes as a function of the incident average laser power P_{in} up to 0.8 W. It is pointed out that the same transmission values are measured for rising or decreasing laser power P_{in} (i.e., no hysteresis effects). The transmitted laser power P_{trans} deviates considerably from linearity

Fig. 2 a Reflected laser power P_{ref} with linear fit curves yielding reflectivity $R = 0.15 \pm 0.01$ and **b** average transmitted laser power P_{trans} of a fs-laser pulse train ($\lambda = 800$ nm, $\tau_{pulse} = 90$ fs, repetition rate 80 MHz, external angle of incidence $\theta = 45^{\circ}$ using thin Si membranes measured as a function of the incident average laser power *P*in. The ft curves in part b originate from the FCA model described in Sect. 3.2 using a fixed exponent $n = 2$ (solid lines) and a variable exponent *n* (broken lines). Each Si membrane is labeled by a number and thickness $(d_{\text{mean}}$ from Table [1](#page-3-1)). The uncertainty of the measured power is 0.0005 W, which yields error bars that are smaller than the symbols used. c Absorption coefficients α_{FCA} derived from Fig. [2b](#page-2-1) following the scheme of Fig. [1](#page-1-0) by use of Eq. [\(3b\)](#page-4-1), the value α_1 = 785 cm⁻¹ and the *d* values from Table [1](#page-3-1) (also cf. Sect. [3.2](#page-4-0))

Fig. 3 Natural logarithm ln ($I(z_m)/I_0$) = ln (P_{trans}/P_{in}) − ln $T^2 = -\alpha_1 z_m$ (Eq. [2](#page-3-2)) with $I(z = z_m)$ and $I_0 = I_{z=0}$ being the laser beam intensities at the rear and at the front sides of the Si membrane as a function of the natural logarithm $\ln P_{\text{in}}$ of the average laser power P_{in} incident onto the Si membranes. The boxes (broken lines) indicate the data points used to calculate $(\alpha_1 z_m)^{lin}$ $(\alpha_1 z_m)^{lin}$ $(\alpha_1 z_m)^{lin}$ and d_i^{lin} for Table 1

(Figs. [1,](#page-1-0) [2](#page-2-1)b): starting with an initial linear increase of P_{trans} for low power P_{in} (< 0.2 W in Fig. [1](#page-1-0)), the transmitted power reaches an absolute maximum value at $P_{\text{in}} \approx 0.6 \pm 0.1 \text{ W}$ (Fig. [1\)](#page-1-0). For higher laser power P_{in} , the transmitted power *P*_{trans} decreases again (Fig. [2b](#page-2-1)). Contributions from internally reflected beams to the transmitted laser power P_{trans} are neglected $(<0.5\%)$. Interference between the incoming and internally refected beams is excluded, because the time for a pulse path forth and back through the membrane amounts to \geq 250 fs being much longer than the 90 fs laser pulse duration.

Following the essence of our previous article [[6](#page-8-4)], the discussion of the fndings in Fig. [2](#page-2-1)b is focused on linear absorption (Beer's law, indirect interband transition) in the low power region and FCA as the dominant effect controlling the experimental fndings in the high power range (cf. Figs. [1,](#page-1-0) [2b](#page-2-1); e.g., [[2,](#page-8-1) [5,](#page-8-11) [6\]](#page-8-4)).

The lifetime of thermalized electrons in the conduction band (CB) of Si, or equivalently, of thermalized electron–hole (eh) pairs ranges around $\tau_{eh} \approx 10^{-5}$ s for intrinsic and weakly doped Si samples [[13](#page-8-12)[–15](#page-8-13)]. This relatively long lifetime is much longer than the typical duration of ultrashort (ps to fs) or even short $(1 \text{ ns to } 1 \mu s)$ laser pulses. Therefore, eh pairs can accumulate during long laser pulses and/or during laser pulse trains used for the excitation of Si and thus generate additional absorption of the laser beam by free charge carriers. At our repetition rate of 80 MHz $\tau_{\text{ren}} = 1.25 \times 10^{-8}$ s << τ_{eh} . The accumulation of eh pairs during repetitively pulsed irradiation leads to a power dependent stationary density of eh pairs *n*eh which is determined by the ratio between the power dependent eh pair generation and its losses. Therefore, the laser beam transmission is approximated by Beer's law (linear absorption) at low incident power P_{in} and by additional linear absorption by free charge carriers (FCA) at high power, as shown in Fig. [1.](#page-1-0)

3.1 Linear laser beam absorption (Beer's law)

In this section, the low P_{in} power range of the linear absorption is considered and its validity region estimated. Beer's law in its logarithmic form reads

$$
\ln(I(z_m)/I_0) = -\alpha_1 z_m,\tag{1}
$$

with α_1 and $z_m = d / \cos \gamma$ being the linear absorption coefficient of Si at λ_{laser} and the absorption pathlength within the Si membrane of thickness *d* under propagation angle γ (Fig. [1](#page-1-0)), respectively. The intensities $I(z_m)$ and I_0 in Eq. [\(1](#page-3-3)) refer to the laser beam intensities at the rear side in the Si membrane $I(z = z_m)$ and the front side $I(z = 0)$ directly before the rear $Si/SiO₂$ interface and behind the front $Si/$ $SiO₂$ interface, respectively. The relations between these intensities and the measured powers P_{trans} and P_{in} in Fig. [2](#page-2-1) are established by assuming the proportionalities $I_0 = \eta_{in} P_{in} T_{air/Si}$ and $I(z_m) = \eta_{trans} P_{trans}/T_{Si/air}$ with $T_{air/Si}$ and $T_{Si/air}$ being the intensity transmission factors ($T = 1 - R$, *R* = reflectivity). η_{in} and η_{trans} are the factors to convert the average laser power to the laser pulse intensity (averaged over the pulse duration and the beam cross section). With the assumption $\eta_{\text{in}} = \eta_{\text{trans}}$, i.e., assuming identical spatial and temporal beam profiles, both η terms cancel. Using, furthermore, the relation $T_{\text{air/Si}} = T_{\text{air/Si}} = T = 1 - R = 0.845$, the intensity terms I_0 and $I(z_m)$ in Eq. ([1](#page-3-3)) can be replaced by the measured values P_{in} and P_{trans} (Fig. [2\)](#page-2-1), respectively, and yield:

$$
\ln (I(z_{\rm m})/I_0) = \ln (P_{\rm trans}/P_{\rm in}) - \ln T^2 = -\alpha_1 z_{\rm m}.
$$
 (2)

Equation ([2\)](#page-3-2) provides the basis of Fig. [3](#page-2-0) and the option to empirically determine the product $\alpha_1 z_m$ for each investigated Si membrane. As far as Beer's law is applicable, the product $\alpha_1 z_m$ should stay constant independent of P_{in} and have a specific value for each membrane thickness $d = z_m \cos \gamma$. Consequently, the P_{in} range of linear absorption (validity of Beer's law) is characterized by the horizontal part of each trace in Fig. [3](#page-2-0) as indicated by the broken lines. The abscissa scale is given in logarithmic units ($\ln (P_{in}/W)$) for convenient presentation only.

The *T* value used in calculating the ordinate scale value in Fig. [3](#page-2-0) is uncertain because the used refractive index of bulk amorphous $SiO₂$ need not necessarily apply to the thin layer of the natural oxide. $\Delta T = \pm 0.05$ corresponds to an uncertainty $\Delta n_{SiO_2} = \pm 0.033$ in the refractive index of SiO₂. These uncertainties have been taken into account, together with the uncertainty in the measured P_{trans} values of ΔP_{trans} $= \pm 0.0005$ W to calculate the error bars in Fig. [3.](#page-2-0)

For high power P_{in} , the traces $\ln (P_{\text{trans}}/P_{\text{in}}) - \ln T^2$ turn strongly negative due to free charge carrier absorption (FCA, cf. Sect. [3.2](#page-4-0)). For thick membranes and small *P*in values, on the other hand, the turn of $\ln (P_{trans}/P_{in}) - \ln T^2$ to negative values is attributed to measurement errors (baseline imperfections). The values of $(\alpha_1 z_m)^{lin}$ listed in Table [1](#page-3-1) were obtained by averaging over the data points in the horizontal part of each curve, with the standard deviation taken as the error. The literature values for α_1 cover a broad range from, e.g., 604 cm⁻¹ [[16\]](#page-8-14) to 1100 cm⁻¹ [[17\]](#page-8-15). The value $\alpha_1 = 604 \text{ cm}^{-1}$ was obtained by a z-scan measurement using a fs-laser pulse train at high repetition rate. Knowing now about considerable FCA under these conditions [[6\]](#page-8-4), this experimental result has to be reconsidered. From a recent paper $\alpha_1 = 785 \text{ cm}^{-1}$ is obtained [\[18\]](#page-8-16), which appears to be most reliable. The Si membrane thicknesses d_i^{lin} listed in Table [1](#page-3-1) were calculated with $\alpha_1 = 785 \pm 100 \text{ cm}^{-1}$, using the standard error propagation techniques.

In the paragraph above, the membrane thickness values d_i were derived from the experimentally determined $(\alpha_1 z_m)_i$ product of Fig. [3](#page-2-0) using the relation $d_i = (\alpha_1 z_m)_i (\cos \gamma)/\alpha_1$. A second source of $(\alpha_1 z_m)_i$ values is, however, available from the $\exp(-\alpha_1 z_m)$ term in the FCA model (Eqs. [5](#page-5-0), [7\)](#page-5-1) in Sect. [3.2](#page-4-0) and the related ft curve parameter *F* (Eq. [8\)](#page-5-2). For clarity of the presentation, the above values d_i and $(\alpha_1 z_m)_i$ derived from

Table 1 Parameters $(\alpha_1 z_m)$ and thickness d_i of the Si membranes obtained from the linear absorption regime (Beer's law, Fig. [3](#page-2-0), index: lin) and the FCA model (Eqs. [5](#page-5-0) and [7\)](#page-5-1) used to fit the meas-

urement points in Fig. [2b](#page-2-1) with exponent $n = 2$ (index: FCA, fix) and *n* = variable (index: FCA, var); d_{mean} is the average and Δd_{SD} is the standard deviation of the three *d* values

Membrane	$(\alpha_1 z_m)^{lin}$	d_i^{lin} [µm]	$d_i^{\text{FCA,fix}}$ [µm] $n=2$	$d_i^{\text{FCA, var}}$ [µm] $n =$ variable	n	$d_{\text{mean}} \pm \Delta d_{\text{SD}}$ [µm]
number						
	0.84 ± 0.04	10.5 ± 1.4	9.5 ± 1.9	10.7 ± 2.0	2.60 ± 0.05	10.2 ± 0.7
2	1.02 ± 0.01	12.8 ± 1.6	12.6 ± 2.1	13.5 ± 2.3	2.38 ± 0.06	$12.9 + 0.5$
3	1.51 ± 0.02	18.8 ± 2.4	20.5 ± 3.0	20.0 ± 3.0	1.77 ± 0.04	19.8 ± 0.9
4	1.66 ± 0.01	20.8 ± 2.7	21.2 ± 3.1	20.5 ± 3.0	1.71 ± 0.05	20.8 ± 0.4
5	$1.94 + 0.01$	24.2 ± 3.1	24.9 ± 3.5	24.0 ± 3.4	1.77 ± 0.03	24.3 ± 0.5
6	2.51 ± 0.02	$31.3 + 4.0$	33.0 ± 4.5	$32.3 + 4.4$	1.68 ± 0.07	32.2 ± 0.8

Fig. 4 a Square root of the deviation $\Delta I_{\text{FCA}}/n$ [W] (cf. text) of the measured transmission values from the ideal behavior of Beer's law (Eq. [9\)](#page-5-3) plotted versus the average laser power P_{in} [W] incident onto the Si membranes (thickness *d* given in fgure). **b** Stationary density n_{eh} of the free charge carriers obtained from the α_{FCA} values of Fig. [2](#page-2-1)c as a function of P_{in} for the Si membranes of thickness *d* using Eq. ([10](#page-6-0)) with fixed $n = 2$. **c** Stationary density $n_{\text{eh}}^{\text{var}}$ of free charge carriers obtained from the α_{FCA} values of Fig. [2](#page-2-1)c in the same way as before but using Eq. [10](#page-6-0) with the variable values *n* given in Table [2](#page-5-4)

the linear absorption regime (Beer's law) are in the following denominated d_i^{lin} and $(\alpha_1 z_m)^{\text{lin}}_i$, while d_i^{FCA} describe the thickness values extracted from the FCA model. The $d_{i}^{\text{lin}}, d_{\text{fix}}^{\text{FCA}},$ and $d_{\text{var}}^{\text{FCA}}$ values for a certain membrane in Table [1](#page-3-1) agree with one another with a standard deviation $\pm \Delta d_{SD} < 1 \,\mu$ m, meaning that the agreement of the *d* values is better than their individual errors. The averages of the three values are listed in the last column and used as labels in Figs. [2](#page-2-1), [3](#page-2-0), and [4](#page-4-2).

3.2 Free charge carrier absorption (FCA)

Recently, it was empirically found that the transmission of 590 nm dye laser and 800 nm fs-laser pulses through thin Si samples [\[6](#page-8-4)] can be modelled by use of the expressions:

$$
I(z_m, I_0) = I_0 \exp(-\alpha_1 z_m) \quad \text{for } I_0 < I_c \tag{3a}
$$

$$
I(z_m, I_0) = (I_0 - I_c) \exp(-\alpha_1 z_m) \exp(-\alpha_{\text{FCA}} z_m)
$$

+
$$
I_c \exp(-\alpha_1 z_m) \quad \text{for } I_0 > I_c
$$
 (3b)

with α_{FCA} representing the FCA contribution. In the low intensity regime $I_0 \leq I_c$, Beer's law is applicable due to neg-ligible FCA. Expression ([3b\)](#page-4-1) is valid for $I_0 > I_c$ and allows to fit, e.g., a constant level of transmitted laser intensity $I(z_m, I_0)$ for high I_0 values well above I_0^{max} as observed at 590 nm with I_0^{max} being the I_0 value for which $I(z_m, I_0)$ reaches its maximum $[6]$. The physical meaning of I_c can be interpreted as an $I₀$ threshold value below which electron/hole (eh) pairs are generated, but do not contribute to FCA, as shown, e.g., in Fig. [4](#page-4-2), where $\Delta I_{\text{FCA}} \approx 0$ near the origin of the P_{in} axis, with ΔI_{FCA} being the deviation of the experimental $I(z_{\text{m}})$ values from Beer's law (also cf. Fig. [1\)](#page-1-0). At least two reasons are conceivable for that: (1) part of the eh pairs are out of resonance with the incident laser beam photons or (2) the eh pairs built up in the frst and fnal parts of (relatively short) laser pulses cannot fully contribute to FCA, i.e., part of the laser pulse will pass through the membrane without FCA. In the frst case the I_c value is expected to depend on the laser wavelength. In the second case-probably applicable to the dye laser pulse transmission at 590 nm and 250 ns duration $[4]-I_c$ $[4]-I_c$ will depend on the laser pulse duration relative to the (intensity dependent) eh pair build-up time.

The absorption coefficient α_{FCA} was previously shown [\[6\]](#page-8-4) to follow the relation:

$$
\alpha_{FCA} \approx n_{\rm eh} \sigma_{\rm eh}(n_{\rm eh}) \approx \sigma_0 n_{\rm eh}^2 \approx \sigma_0 (\kappa I_0)^2 = \kappa_2 I_0^2 \tag{4}
$$

with n_{eh} being the effective stationary density of eh pairs in the laser-irradiated volume. The absorption cross section $\sigma_{eh}(n_{eh})$ is proportional to the density n_{eh} as could be derived from the findings in [[19\]](#page-8-17), i.e., $\sigma_{eh} \approx \sigma_0 n_{eh}$ showing a nonlinear absorption response. Furthermore, it was assumed that n_{eh} is homogeneous in the laser-irradiated volume and proportional to I_0 due to eh pair generation by 1-photon

Table 2 Fit parameters for the $P_{trans}(P_{in})$ traces in Fig. [2](#page-2-1)b obtained by the application of Eq. ([8\)](#page-5-2). The parameter values for variable *n* are graphed in Fig. [5](#page-6-1)

absorption. As a consequence, the FCA contribution in Eq. ($3b$) implies a quadratic dependence on I_0 in its exponent:

$$
I(z_{\rm m}) = (I_0 - I_{\rm c}) \exp(-\alpha_1 z_{\rm m}) \exp(-\kappa_2 I_0^2 z_{\rm m}) + I_{\rm c} \exp(-\alpha_1 z_{\rm m})
$$
 (5)

or more generally

$$
I(z_m) = (I_0 - I_c) \exp(-\alpha_1 z_m) \exp(-\kappa_n I_0^n z_m)
$$

+
$$
I_c \exp(-\alpha_1 z_m),
$$
 (6)

with $\kappa_{2,n}$ being the empirical constants and *n* the empirical variable parameter. Equation (6) (6) , i.e., $I(z_m)$ = $f(I_0, I_c, \alpha_1, \kappa_n, n, z_m)$ together with the above relations (Sect. [3.1\)](#page-3-0) between I_0 and P_{in} as well as $I(z_m)$ and P_{trans} is transformed to the function of the measured values P_{trans} = $f(P_{\text{in}}, P_{\text{c}}, \alpha_1, \alpha_{\text{FCA}}, z_{\text{m}})$ with $P_{\text{c}} = I_{\text{c}}/(\eta T)$:

$$
P_{\text{trans}}(z_{\text{m}}) = T^2 \exp(-\alpha_1 z_{\text{m}})(P_{\text{in}} - P_{\text{c}})
$$

$$
\times \exp(-\alpha_{\text{FCA}} z_{\text{m}}) + T^2 \exp(-\alpha_1 z_{\text{m}})P_{\text{c}}
$$
 (7)

or in an abbreviated form for convenient ftting

$$
P_{\text{trans}}(z_{\text{m}}) = F_n \left[\left(P_{\text{in}} - P_{\text{c}} \right) \exp \left(-E_n P_{\text{in}}^n \right) + P_{\text{c}} \right],\tag{8}
$$

with $F_n = T^2 \exp(-\alpha_1 z_m)$ and $E_n = \alpha_{\text{FCA}} z_m = \kappa_n (\eta T)^n z_m$. Application of Eq. [\(8](#page-5-2)) is performed in two ways, i.e., with fixed $n = 2$ to test the above approximation for α_{FCA} and with variable *n*. Applying these two procedures to the $P_{trans}(P_{in})$ traces in Fig. [2](#page-2-1)b obtained for the Si membrane thickness series yields the parameters summarized in Table [2](#page-5-4).

The fit curves obtained by Eq. [\(8](#page-5-2)) with fixed $n = 2$ or *n* as variable parameter are practically identical for thick membranes but differ in case of the two thinnest membranes (Fig. [2](#page-2-1)b). The ftting improvement by use of a variable exponent *n* is trivial from the mathematical point of view as there is one more free parameter available for optimization. On the other hand, the obtained $1.7 \le n \le 2.6$ $1.7 \le n \le 2.6$ $1.7 \le n \le 2.6$ values (Table 2) appear to be relevant for the physical meaning of the present fndings (cf. Fig. [5](#page-6-1)b). The parameters F_n and F_2 obtained with the variable exponent *n* and fixed $n = 2$, respectively (Table [2\)](#page-5-4) are very similar (maximum deviation: 10 %, membrane 1). The

obtained parameters E_n (or E_2) and $P_{c,n}$ (or $P_{c,2}$) appear to depend on the membrane thickness (cf. discussion below).

The deviation of the laser intensity transmitted through the thin Si membrane from the small signal absorption behavior (Beer's law) is essentially attributed to FCA (Fig. [1\)](#page-1-0) and is mathematically described by

$$
\Delta I_{\text{FCA}} = I^0(z_{\text{m}}, I_0) - I(z_{\text{m}}, I_0)
$$

= $\eta P_{\text{in}} T \exp(-\alpha_1 z_{\text{m}}) - \eta P_{\text{trans}} / T$
= $\eta(P_{\text{in}} T \exp(-\alpha_1 z_{\text{m}}) - P_{\text{trans}} / T)$
= $\eta(0.845 P_{\text{in}} \exp(-785 \text{ cm}^{-1} d_i / \cos 11^\circ)$
- $P_{\text{trans}} / 0.845$). (9)

In our previous paper [\[6](#page-8-4)], $(\Delta I_{\text{FCA}})^{0.5}$ was found to increase approximately linearly with the input power P_{in} (or I_0), as shown in Fig. [4,](#page-4-2) of [[6\]](#page-8-4). The calculated FCA contribution depends, however, on the assumed linear absorption behavior (Beer's law) which has been revised in this paper with respect to the α_1 and z_m values (Sect. [3.1](#page-3-0)). Therefore, it appears interesting to check the FCA behavior of ΔI_{FCA} for the membrane thickness series of this paper under the revised conditions, i.e., with $\alpha_1 = 785 \pm 100 \text{ cm}^{-1}$ and the newly determined membrane thickness values from Table [1.](#page-3-1) For simplicity, Eq. [\(9](#page-5-3)) is divided by the constant scaling factor $\eta = 1.0 \times 10^{11}$ [pulse \cdot cm²]⁻¹ for the power law test of ΔI_{FCA} which is depicted in Fig. [4a](#page-4-2).

At first sight, the square root of $\Delta I_{\text{FCA}}/\eta$ appears to increase (almost) linearly with the input power P_{in} above a P_{in} threshold P_{c} which depends on the membrane thickness. This fnding essentially confrms the previous results in [\[6\]](#page-8-4). The slope of the linear parts $d\sqrt{\Delta I_{\text{FCA}}/n}/dP_{\text{in}} =$ $(1/\sqrt{\eta})d\sqrt{\Delta I_{\text{FCA}}}$ / $d(I_0/\eta T) = (T\sqrt{\eta})d\sqrt{\Delta I_{\text{FCA}}}$ / $d(I_0)$ on the average decreases with increasing membrane thick-ness (Figs. [4](#page-4-2)a, [5](#page-6-1)a). The threshold P_c on the P_{in} axis for the linear behavior is difficult to determine and the values given in Fig. [4\(](#page-4-2)a) are rough estimates only.

The physical meaning of parameter $F_n = T^2 \exp(-\alpha_1 z_m)$ in Eq. ([8](#page-5-2)) is straightforward and its values trivially show an exponential thickness dependence (ftted line in Fig. [5](#page-6-1)a) as expected, since the *d* values on the abscissa scale have

been obtained from the F_n parameter representing the linear interband absorption (Beer's law).

The FCA effect, on the other hand, is represented by the absorption coefficient α_{FCA} (Fig. [2c](#page-2-1)) and the ΔI_{FCA} value (Fig. [4](#page-4-2)a). For its physical understanding the density n_{eh} of free charge carriers, their absorption cross section σ_{eh} , and the relation $\sigma_{eh} = \sigma_0 n_{eh}^{n-1}$ appear to be of utmost importance. Therefore, a selection of model values are displayed in Fig. [5](#page-6-1) as a function of the membrane thickness *d*: value *n* from Table [2,](#page-5-4) the slope of Fig. [4](#page-4-2)a and the values n_{eh} obtained from α_{FCA} in three ways using Eq. ([10\)](#page-6-0) with $n = 1, 2$ and $n =$ variable (Table [2\)](#page-5-4). In Fig. [5a](#page-6-1) the values *n* and the slope of Fig. [4a](#page-4-2) both decrease steadily with increasing membrane thickness.

The physical meaning of the parameter *n* is found in the relation

$$
\alpha_{\text{FCA}} \approx n_{\text{eh}} \sigma_{\text{eh}} (n_{\text{eh}}) \approx \sigma_0 n_{\text{eh}}^n \approx \sigma_0 (\kappa I_0)^n = \kappa_n I_0^n \tag{10}
$$

that is the generalized form of Eq. (4) , as was used in Eq. (6) (6) . Analysing the sequence of steps in Eq. ([10\)](#page-6-0), the value of *n* actually comes from the dependence of the absorption cross section σ_{eh} on the density n_{eh} of the eh pairs:

$$
\sigma_{\rm eh} = \sigma_0 n_{\rm eh}^{n-1}.
$$
\n(11)

Therefore, a change in the value of *n* corresponds to a change in the absorption cross section of the eh pairs as a function of the eh pair density. Evidently, the values $n_{eh}(d)$ obtained for $n = 1$ and 2 (Fig. [5](#page-6-1)b) show similar dependences from the membrane thickness, i.e., a steady decrease of n_{eh} with thickness *d*. This attenuation results from a strongly decreasing rate of the average optical eh pair generation with increasing *d* values. The value $n = 1$ cannot *a priori* be excluded as it appears applicable for low n_{eh} densities (Fig. 4 of Refs. $[19, 19]$ $[19, 19]$ $[19, 19]$ [20](#page-8-18)]). Assuming $3 \times 10^{-18} \le \sigma_{eh} \le 2 \times 10^{-17}$ cm² (Fig. 3b) of Ref. [[5](#page-8-11)]) and applying it to the highest α_{FCA} values of this study yields, however, n_{eh} values in the 1×10^{20} cm⁻³ to 5×10^{20} cm⁻³ range in which the n_{eh} dependence of σ_{eh} is no longer negligible (Fig. 4 of Ref. [[19](#page-8-17)]). The values $n_{\text{eh}}(d)$ obtained for $n =$ variable, on the other hand, display a maximum n_{eh} value around $d \approx 20 \,\mu$ m, indicating at least two counteracting effects. Taking into account that the *P*_{trans} values in Fig. [2b](#page-2-1) are optimally fitted by $n =$ variable, the related n_{eh} values are assumed to be the most reliable ones. The preference of these n_{eh} values is supported by additional fndings:

In Table [2](#page-5-4), the fit parameters $P_{c,n}$ and E_n as well as the threshold values from Fig. [4](#page-4-2) appear to have a minimum around membrane thickness $d \approx 20 \,\mu m$ indicating a fundamental change in the FCA behavior. $P_{c,n}$ and the threshold value of Fig. [4](#page-4-2) both have the physical meaning of the constant part of the incident power ($P_c = I_c / \eta T$) that undergoes linear absorption, but no FCA.

Fig. 5 Fit parameters of the FCA model (Eq. [8\)](#page-5-2) namely a F_i (rhombes), *n* (triangles) from (Tab. [2\)](#page-5-4) and the slopes of Fig. [4](#page-4-2) (squares) and **b** relative density values n_{eh} of free charge carriers as a function of the membrane thickness d ; the values n_{eh} were derived from the maximum α_{FCA} values in Fig. [2c](#page-2-1) recorded at $P_{\text{in}} = 0.8$ W in three ways using in Eq. ([10](#page-6-0)) the values $n = 1, 2$ and $n =$ variable (Table [2\)](#page-5-4). Proportionality factors are used for convenience of presentation $(F_n: \times 8$ and slope $\times 3$). The solid line in part **a** is an exponential ft curve (cf. text), whereas the dotted lines are guides for the eye only

Looking at Eqs. ([10\)](#page-6-0) and ([11\)](#page-6-2) the observed $n_{eh}(d)$ dependences may be traced back to the value n_{eh} itself and/or to the n_{eh} dependence of the absorption cross section σ_{eh} via variations of the exponent *n*. The density n_{eh} is controlled dynamically and represents a kinetic equilibrium between the laser induced generation of eh pairs and their losses. Their generation by the interband absorption in Si is determined by the α_1 value and the incident laser power P_{in} . The losses of n_{eh} pairs in the laser-irradiated Si volume, which is at the same time the probe volume for transmission, presumably occur mainly via (1) eh recombination at the front and rear $Si/SiO₂$ interfaces and (2) transverse diffusion of eh pairs out of the irradiated Si volume into the non-irradiated Si environment. Difusion (2) should dominantly depend on the transverse n_{eh} density gradient across the interface between the irradiated and non-irradiated parts of the Si membrane and show a weak membrane thickness dependence only.

The eh pair losses (1) should, however, increase with the thickness dependent surface-to-volume ratio of the membranes and thus be highest for the thinnest membranes. As a preliminary result the n_{eh} maximum at $d \approx 20 \,\mu\text{m}$ for $n =$ variable in Fig. [5b](#page-6-1) can qualitatively be rationalized by the balance between the counteracting efects of the laser beam attenuation (reduction of the eh pair generation rate) and the eh pair recombination at the front and rear $Si/SiO₂$ interfaces. These fndings suggest to consider additionally physical effects next to the $Si/SiO₂$ interfaces which may be negligible in case of thick membranes, but are important for thin membranes.

At the rear $Si/SiO₂$ interface there is a region of spatial and temporal overlap of the incoming and the refected laser beams. For 90 fs duration and refractive index $n_{\rm Si} = 3.69$ [[11,](#page-8-9) [12](#page-8-10)] the laser pulse in Si corresponds to a light disc of $d_{\text{las}} \approx 7.3 \,\mu\text{m}$ thickness. This results in a beam overlap sheet of $d_{ov} \approx d_{las}/2 \approx 3.6 \,\mu\text{m}$ thickness showing locally enhanced or reduced intensity for positive or negative interference between the incoming and the refected beam parts, respectively. The sign of the interference depends on the phase change during laser beam refection at the rear Si/ SiO₂/air interfaces. The ratio d_{ov}/d of the overlap sheet to the membrane thickness is inversely proportional to *d* and rises steeply for small *d* values.

Furthermore charge carrier injection from Si into $SiO₂$ is well known under the present laser irradiation conditions (cf., e.g., [\[21\]](#page-8-19)). The pulse peak intensities incident on the sample amount up to 80 GW/cm² averaged over the beam cross section and the pulse duration. This leads to quasistatic electric fields E_{dc} across both $Si/SiO₂$ interfaces. The feld strengths and their penetration depth into the neighbouring Si membrane regions depend on the local laser intensities and the shielding of the electric charges intermediately fixed in the $SiO₂$ layers. The thickness of the depletion layers (depth of shielding) within the Si membrane next to the interfaces is estimated to lie in the $1-1.5 \,\mu m$ region for the given p-dopant concentration (cf., e.g., [\[22](#page-8-20)]). This estimate assumes, however, an infnitely thick Si sample without laser excitation, i.e., free charge carriers originating from the dopant only. The eh pairs located in the electric field affected regions, that exist on both sides of the membrane [[23,](#page-8-21) [24](#page-8-22)], are subject to diferent electric feld strengths depending on their distance from the $Si/SiO₂$ interface. Therefore, it can be expected that features in the absorption spectrum of the afected eh pairs should be shifted due to the perturbation by the varying feld, leading to a shifting and broadening of the FCA features of the eh pair population as a whole and, therefore, a change in the average σ_{eh} value.

In our thickest sample $(\approx 30 \,\mu\text{m}$ thickness) the total depth of the E_{dc} field and/or the laser beam overlap regions is small relative to the total thickness, and therefore, only a small fraction of the total number of eh pairs are perturbed.

The remaining majority of the eh pairs are unperturbed and should show behavior characteristics similar to those found in thick (bulk) Si samples. In our thinnest sample of which the thickness (\approx 10 µm) is not much larger than that of the afected regions near the interfaces, a large fraction of the eh pairs are perturbed. Thus the FCA behavior becomes modified. The n_{eh} dependence of the absorption cross section σ_{eh} shows up, e.g., in the variable *n* values (Table [2](#page-5-4)) and the independently derived variable slopes in Fig. [4a](#page-4-2).

A more detailed analysis of the above fndings requires extended modelling of the discussed efects beyond the scope of the present study. In addition a more extensive experimental data basis will be helpful in order to disentangle the mutual interference between the model parameters n_{eh} , σ_{eh} , σ_0 and the exponent *n*, which all are enclosed in the single experimental value $\alpha_{\text{FCA}} = \alpha_{\text{FCA}}(n_{\text{eh}}, \sigma_{\text{eh}}, \sigma_0, n)$. One experimental approach to extend the data basis might use laser pulse trains at variable pulse repetition rates to change the relation between the eh pair generation rate (average input power P_{in}) and the average eh pair loss rate (available time for eh pair recombination between the laser pulses).

4 Summary and conclusion

The incidence of a Ti:sapphire fs-laser beam ($\lambda = 800$ nm, τ = 90 fs, 80 MHz) onto thin naturally oxidized, weakly B-doped Si membranes at $\theta = 45^\circ$ external angle of incidence and $\gamma = 11^\circ$ propagation angle in Si was investigated with respect to its refection as well as its transmission. The reflectivity $R = 0.15 \pm 0.01$ is found to be independent of the incident average laser power (≤ 0.8 W) and pulse peak intensity (≤ 80 GW/cm²).

The linear absorption (interband transition) is used to determine the membrane thickness *d* in the 10–30 μm range by two ways, using either the linear absorption regime at low laser power (Beer's law, sect. [3.1\)](#page-3-0) or the related ft parameters of the applied FCA model (Sect. [3.2](#page-4-0)). Both routes lead to very similar results with a standard deviation Δd_{SD} < 1 µm for a given value of the linear absorption coefficient α_1 , for which $\alpha_1 = 785 \text{ cm}^{-1}$ [[18](#page-8-16)] appears most reliable within the $600 \text{ cm}^{-1} \le \alpha_1 \le 1200 \text{ cm}^{-1}$ literature value range.

The transmission behavior at high laser power is mainly attributed to free charge carrier absorption (FCA) and quantitatively reproduced by a recently derived FCA model [[6\]](#page-8-4) (Sect. [3.2\)](#page-4-0). This model uses an efective absorption coefficient $\alpha_{\text{eff}} = \alpha_1 + \alpha_{\text{FCA}}$ with α_{FCA} increasing with the average laser intensity (input power): $\alpha_{\text{FCA}} = \kappa_n I_0^n$ with $n \approx 2$. The FCA model with its empirical ft parameters allows to calculate quantitatively the transmission behavior of Si

membranes within the investigated sample thickness, doping and laser power (intensity) regimes.

Some of the FCA ft parameters display extrema around the membrane thickness $d \approx 20 \,\mu m$ indicating a fundamental change in the FCA behavior. This fnding is attributed to both thickness dependent counteracting efects and to physical effects next to the $Si/SiO₂$ interfaces which are negligible in case of thick Si membranes, but important for thin membranes.

Two counteracting effects are considered most important in controlling the dynamical equilibrium between the generation of free charge carriers and their losses by eh pair recombination: The laser beam power strongly attenuates (exponentially) along its path through the Si membrane by linear interband absorption and the power dependent FCA. Thus the average generation of eh pairs decreases drastically with increasing Si membrane thickness *d*. Increasing *d* values, on the other hand, reduce the relative loss rate of free charge carriers which is expected to be mainly controlled by eh pair recombination at the $Si/SiO₂$ interfaces.

Furthermore charge carrier injection from Si into $SiO₂$ builds up a quasi-static electric field E_{dc} across the $Si/SiO₂$ interface which affects the neighboring Si region $(1 \mu m)$ order of magnitude) according, e.g., to the dopant dependent shielding of E_{dc} . The depth dependent E_{dc} affects the spectral properties of the electron/hole pairs in Si and thus the FCA. At the rear interface, on the other hand, interference between the incoming laser beam and its part refected from the Si/SiO₂/air interfaces modifies the laser impact onto the Si membrane within a region of about 3.5 μm thickness.

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References

- 1. L. Vivien, L. Pavesi, *Handbook of Silicon Photonics* (CRC Press, Boca Raton, 2013)
- 2. J.M. Moison, F. Barthe, M. Bensoussan, Phys. Rev. B **27**, 3611 (1983)
- 3. H.B. Briggs, R.C. Fletcher, Phys. Rev. **91**, 1342 (1953)
- 4. M. Yamada, H. Kotani, K. Yamamoto, K. Abe, Phys. Lett. **85A**, 191 (1981)
- 5. R.D. Kekatpure, M.L. Brongersma, Nano Lett. **8**, 3787 (2008)
- 6. P.-C. Heisel, W.I. Ndebeka, P.H. Neethling, W. Paa, E.G. Rohwer, C.M. Steenkamp, H. Stafast, Appl. Phys. B **122**, 60 (2016)
- 7. C. Logofatu, C.C. Negrila, R.V. Ghita, F. Ungureanu, C. Cotirlan, C.G.A.S. Manea, M.F. Lazarescu, Study of SiO₂/Si Interface by Surface Techniques, pp. 23– 42 (2011). [http://www.intechopen.](http://www.intechopen.com) [com](http://www.intechopen.com)
- 8. K. Sokolowski-Tinten, J. Bialkowski, D. von der Linde, Phys. Rev. B **51**, 14186 (1995)
- 9. J. Bogdanowicz, F. Dortu, T. Clarysse, W. Vandervorst, A. Salnik, J. Appl. Phys. **108**, 104908 (2010)
- 10. J. Martels, H. Borchers, H. Hansen, K.-H. Hellwege, K. Schäfer, E. Schmidt, *Landolt-Börnstein - Zahlenwerte und Funktionen aus Physik, Chemie, Astronomie, Geophysik und Technik. Eigenschaften der Materie in ihren Aggregatzuständen, 8. Teil: Optische Konstanten (Springer, Heidelberg, 1962)*
- 11. G. Vuye, S. Fisson, V. Van Nguyen, Y. Wang, J. Rivory, F. Abeles, Thin Solid Films **233**, 166 (1993)
- 12. E.D. Palik, G. Ghosh, *Handbook of Optical Constants of Solids* (Academic Press, San Diego, 1998)
- 13. D.J. Sandiford, Proc. Phys. Soc. **71**, 1002 (1958)
- 14. J. Dziewior, W. Schmidt, Appl. Phys. Lett. **31**, 346 (1977)
- 15. C.M. Horwitz, R.M. Swanson, Solid-State Electron. **23**, 1191 (1980)
- 16. G.P. Nyamuda, E.G. Rohwer, C.M. Steenkamp, H. Stafast, Appl. Phys. B **104**, 735 (2011)
- 17. R. Braunstein, A.R. Moore, F. Herman, Phys. Rev. **109**, 695 (1958)
- 18. M.A. Green, Sol. Energy Mat. Solar Cells **92**, 1305 (2008)
- 19. J. Meitzner, F.G. Moore, B.M. Tillotson, S.D. Kevan, G.L. Richmond, Appl. Phys. Lett. **103**, 092101 (2013)
- 20. D.K. Schroder, R.N. Thomas, J.C. Swartz, IEEE Trans. Electr. Dev. **ED-25,** 254 (1978)
- 21. T. Scheidt, E.G. Rohwer, H.M. von Bergmann, H. Stafast, Phys. Rev. B **69**, 165314–165321 (2004)
- 22. S.M. Sze, *Semiconductor Devices-Physics and Technology*, 2nd edn. (Wiley, New York, 2002)
- 23. W.I. Ndebeka, PhD Thesis, University of Stellenbosch, South Africa (2016)
- 24. P.-C. Heisel, PhD Thesis, University of Jena, Germany (2016)