

Two-color thermosensors based on $[Y_{1-x}Dy_x(\text{acetylacetonate})_3(1,10\text{-phenanthroline})]$ molecular crystals

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Abstract We develop a two-color thermometry (TCT) phosphor based on $[Y_{1-x}Dy_x(\text{acetylacetonate})_3(1,10\text{-phenanthroline})]$ ($[Y_{1-x}Dy_x(\text{acac})_3(\text{phen})]$) molecular crystals for use in heterogeneous materials. We characterize the optical properties of $[Y_{1-x}Dy_x(\text{acac})_3(\text{phen})]$ crystals at different temperatures and Dy concentrations, and find that the emission is strongly quenched by increasing temperature and concentration. We also observe a broad background emission (due to the ligands) and find that $[Y_{1-x}Dy_x(\text{acac})_3(\text{phen})]$ photodegrades under 355 nm illumination with the photodegradation resulting in decreased luminescence intensity. However, while decreasing the overall emission intensity, photodegradation is not found to influence the integrated intensity ratio of the ${}^4I_{15/2} \rightarrow {}^6H_{15/2}$ and ${}^4F_{9/2} \rightarrow {}^6H_{15/2}$ transitions. This ratio allows us to compute the temperature of the complex. Based on the temperature dependence of these ratios; we calculate that $[Y_{1-x}Dy_x(\text{acac})_3(\text{phen})]$ has a maximum sensitivity of $1.5\% \text{ K}^{-1}$ and our TCT system has a minimum temperature resolution of 1.8 K. Finally, we demonstrate the use of $[Y_{1-x}Dy_x(\text{acac})_3(\text{phen})]$ as a TCT phosphor by determining a dynamic temperature profile using the emission from $[Y_{1-x}Dy_x(\text{acac})_3(\text{phen})]$.

1 Introduction

Plastic-bonded explosives (PBXs) are heterogeneous materials consisting of energetic organic molecular crystals (e.g. RDX, PETN, and TNT) embedded in a polymer matrix, with possible additional additives being plasticizers, antioxidants, taggants/markers, and friction-generating grit. These energetic organic molecular crystals can be initiated by a variety of stimuli, including thermal, mechanical or electrical. In the case of non-thermal initiation, it is believed that these stimuli first heat the energetic material, which then causes thermally induced chemical decomposition to occur.

For mechanical shock initiation, this heating is due to the shock-induced formation of hotspots within the material. Much research has been devoted to understanding shock-induced hotspot formation [1–14] with the main mechanisms being pore collapse [14], plastic deformation [3, 12, 13], micro-fractures, binder/crystal interface friction, and crystal/crystal interface friction. While these mechanisms have been identified to form hotspots in shocked heterogeneous materials, it is currently unknown which of these mechanisms are responsible for chemical energy release for a given material composition and loading condition [12].

Determining this dependence requires performing simultaneous time-resolved microstructural and thermal imaging of heterogeneous sample under shock compression. Performance of these measurements is difficult, as shock-compression events typically occur on the ns- μ s time scales and microstructural imaging typically utilizes synchrotron X-Ray phase contrast imaging [15–19]. One approach to performing in situ microstructural imaging is the recently inaugurated NNSA-sponsored Dynamic Compression Sector (DCS) at the Advanced Photon Source (Argonne National Laboratory) [20, 21]. The DCS provides the opportunity to perform time

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resolved X-ray phase contrast imaging of shock-compressed heterogeneous materials, which provides the first of the two required measurements.

The second required measurement is rapid thermal imaging with ns- μ s resolution. To this end, we are developing temperature sensors based on two-color thermometry (TCT) phosphors, which can be used in heterogeneous materials to provide spatially resolved thermal imaging. Previously, TCT phosphors based on inorganic crystals have been used to perform temperature imaging in a wide variety of contexts [22, 23, 23–34], but for our application—inert PBX analogues—we desire to have organic TCT phosphors that have similar mechanical properties to the energetic materials.

The first step, toward this goal, is to develop appropriate organic molecular crystals that can be used for two-color thermometry. To achieve this goal, we choose to use Dy³⁺-doped yttrium ternary complexes that can be grown into molecular crystals with the first such material being [Y_{1-x}Dy_x(acetylacetonate)₃(1,10-phenanthroline)] ([Y_{1-x}Dy_x(acac)₃(phen)]). In this study, we report on the optical properties of [Y_{1-x}Dy_x(acac)₃(phen)] at different temperatures and concentrations.

2 Background

2.1 Two-color thermometry

Dy³⁺ is a well-known lanthanide ion used for a wide variety of optical applications [22, 23, 23–30, 34–42], with a common application being two-color thermometry (TCT) [22, 23, 23–30, 34]. TCT using Dy relies on two closely spaced Dy energy levels (⁴I_{15/2} and ⁴F_{9/2}) to act as a probe of temperature. When a Dy-doped material is optically pumped (with an appropriate wavelength), the ions are excited into the ⁴F_{9/2} energy level, from which they can also populate the ⁴I_{15/2} energy level due to thermal effects. The ratio of the populations in the ⁴I_{15/2} energy level n_2 , and the population in the ⁴F_{9/2} energy level n_1 , is determined by the Boltzmann distribution, such that their ratio is:

$$\frac{n_2}{n_1} = e^{-\Delta E/k_B T}, \quad (1)$$

where ΔE is the energy difference between the ⁴I_{15/2} and ⁴F_{9/2} energy levels (approximately 0.115 eV for aqua ions [43–46]), k_B is Boltzmann's constant, and T is the temperature.

This ratio between the ⁴I_{15/2} and ⁴F_{9/2} energy levels can be determined using emission spectroscopy; as the ratio of the ⁴I_{15/2} → ⁶H_{15/2} and ⁴F_{9/2} → ⁶H_{15/2} transition is proportional to the ratio of populations:

$$\begin{aligned} \frac{I_2}{I_1} &\propto \frac{n_2}{n_1} \\ &= A e^{-\Delta E/k_B T}, \end{aligned} \quad (2)$$

where A is a proportionality constant.

2.2 Ligand enhanced luminescence

While Dy³⁺ has advantageous optical properties for TCT, on its own Dy³⁺ (and the other lanthanide ions) has a small molar absorption coefficient, making it an inefficient phosphor. To overcome this limitation, much work has been done on embedding different lanthanide ions in ligands such that the ligands (which have large absorption coefficients) absorb the pump light and subsequently transfer the absorbed energy to the lanthanide ion [45, 47–61]. This energy transfer is typically radiationless and can occur via several possible mechanisms: electron exchange [48, 53], dipole–dipole interactions (both allowed [62–65] and unallowed transitions [48, 66]), dipole–quadrupole interactions [48], and excitons [67, 68].

This energy transfer has been empirically found to be most efficient if the ligand's lowest triplet energy level is between 0.23 and 0.62 eV greater than the lanthanide's primary luminescent state, as energy back-transfer is more likely to occur for energy differences of 0.23 eV [51, 69]. One such material, satisfying this criterion, is [Dy(acetylacetonate)₃(1,10-phenanthroline)] ([Dy(acac)₃(phen)] [52, 70–73], which is the subject material of this study. To demonstrate the energy level spacing of the ligands and Dy ion, we provide a simplified energy level diagram in Figure 1 where the energy levels are

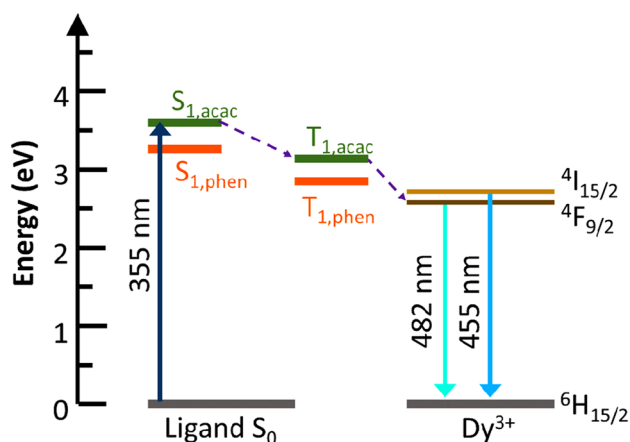


Fig. 1 Energy level schematic of [Dy(acac)₃(phen)] displaying energy transfer from the ligands' triplet state to Dy³⁺'s lowest excited state

taken from the literature [45, 47, 52, 74, 75]. Note that in general there are many more energy levels for the ligands and Dy than displayed in Fig. 1, but we simplify the diagram to highlight the relevant transitions.

From a review of the literature [45, 47, 52, 74, 75], we find that the acac ligand has an $S_0 \rightarrow S_1$ transition corresponding to ≈ 3.596 eV (345 nm), phen has an $S_0 \rightarrow S_1$ transition corresponding to ≈ 3.263 eV (380 nm), and Dy^{3+} 's primary luminescent state ($^4F_{9/2}$) is at ≈ 2.583 eV (480 nm). In our study, we use a frequency-tripled Nd:YAG laser with a wavelength of 355 nm for our excitation source, which places the excitation wavelength nearly resonant with acac's $S_0 \rightarrow S_1$ transition. This means that during excitation the primary absorption is due to the acac ligand, which can then nonradiatively transition to the lowest triplet state T_1 (energy ≈ 3.134 eV) via intersystem crossing. Once in the triplet state, the energy can be transferred from the ligand to the Dy^{3+} ion with the approximate energy difference between acac's T_1 state and the $^4F_{9/2}$ state of Dy^{3+} being 0.554 eV. Since this energy difference is greater than 0.23 eV, backwards energy transfer will be minimal.

3 Method

3.1 Materials

The primary component of our two-color thermosensors is [Dy(acetylacetonate)₃(1,10-phenanthroline)] ([Dy(acac)₃(phen)]). [Dy(acac)₃(phen)] has previously been investigated for use as a single molecule magnet [70, 71], a phosphor for white OLEDs [72, 73], and an NIR optical phosphor for use in telecommunications [52]. In these previous studies, the authors prepared the material with a Dy concentration of 100 mol%, while in our study we dilute [Dy(acac)₃(phen)] with [Y(acac)₃(phen)]. We perform this dilution to characterize the influence of Dy concentration on the material's optical properties. This combination of [Dy(acac)₃(phen)] and [Y(acac)₃(phen)] results in a hybrid molecular crystal, which we denote [Y_{1-x}Dy_x(acac)₃(phen)].

To prepare [Y_{1-x}Dy_x(acac)₃(phen)], we use the following procedure [71]: First, we prepare a solution of acetylacetonate (0.06 M) and 1,10-phenanthroline (0.02 M) in a DMF/methanol mixture (1:1 vol./vol. ratio). To this solution, we add an aqueous solution of potassium tert-butoxide (KOtBu) (0.06 M) at an equal volume ratio. Once mixed, we add another equal volume aqueous solution containing Dy(NO₃)₃·5H₂O and Y(NO₃)₃·6H₂O in different Dy/Y mole ratios such that the total concentration is 0.02 M. For this study, we make [Y_{1-x}Dy_x(acac)₃(phen)] with Dy concentrations of 1, 5, 10, and 15 mol%.

Gradually white precipitates appear and the suspension is allowed to age under stirring at room temperature for 4–6 h. The precipitates are isolated via centrifugation (6000 rpm for 3 min) and finally dried in vacuum at 80 °C for ≈ 12 h. The resulting powder consists of [Y_{1-x}Dy_x(acac)₃(phen)] molecular crystals (MCs).

To characterize the structure of the [Y_{1-x}Dy_x(acac)₃(phen)] MCs, we use X-ray diffraction (XRD) with a Cu-K α radiation source ($\lambda=1.5418$ Å) operated at 40 kV and 40 mA. The X-ray beam is monochromated using an X'Celerator monochromator (PANalytical B.V.) and collimated using a fixed divergence slit with 0.04 radian Soler slits, 0.5° divergence slit and 10 mm mask. The diffracted X-rays are detected using a PIXcel3D detector (PANalytical B.V.) with the resulting diffraction patterns being analyzed using X'Pert HighScore Plus, Version 3.0 (PANalytical B.V.). Additional characterization is performed by optical microscopy of the MC powder.

Figure 2 shows a representative XRD pattern for [Dy(acac)₃(phen)], with the pattern displaying sharp peaks consistent with a polycrystalline structure. This structure is also confirmed by optical microscopy, with the inset of Fig. 2 being an optical image displaying multiple crystals of varying shapes and sizes. To determine the crystal parameters (structure, space group, lattice size, etc.) of the material, we analyze the XRD pattern for each Dy concentration and find that the crystal parameters are within uncertainty of each other for all Dy concentrations. We therefore average the parameters and tabulate their average value in Table 1. These parameters are found to be consistent with previous XRD measurements of [Dy(acac)₃(phen)] MCs [71].

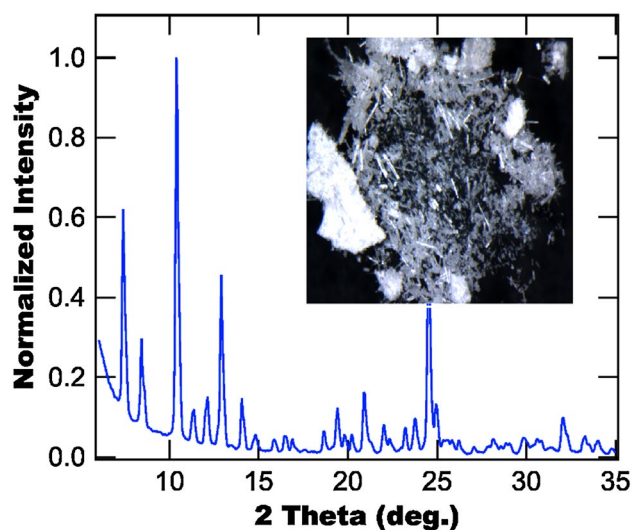


Fig. 2 XRD pattern for [Dy(acac)₃(phen)] with multiple sharp peaks visible. Inset: Optical microscope image of crystals

Table 1 Average crystal parameters for $[Y_{1-x}Dy_x(acac)_3(phen)]$ determined from XRD. These parameters are found to be invariant with Dy concentration (within experimental uncertainty)

| Parameter | Value |
|-----------------------|-------------|
| Crystal system | Monoclinic |
| Space group | $P2_1/c$ |
| a (Å) | 16.1 |
| b (Å) | 21 |
| c (Å) | 9.4 |
| α | 90° |
| β | 116° |
| γ | 90° |
| V (Å ³) | 2841.82 |

3.2 Spectroscopy

To measure the luminescence spectra of $[Y_{1-x}Dy_x(acac)_3(phen)]$ at different temperatures, we use a custom fluorescence spectroscopy system and powder heater. The spectroscopy system consists of a frequency-tripled Nd:YAG laser (Continuum Powerlite II 8000, 355 nm, 8 ns), focusing and collection optics, and two different spectrometers (one for temporally gated spectral measurements and the other for ungated measurements). The spectrometer for ungated measurements consists of an Acton SpectraPro 2750i monochromator, attached current PMT, and a Spectrahub interface, while the spectrometer for gated measurements consists of a Princeton Instruments PI-Max 3 ICCD connected to an Acton SpectraPro 2500i spectrometer. In addition to spectral measurements, the ungated system is also used to measure fluorescence lifetimes with the PMT output connected to a Tektronix

DPO 4104 oscilloscope. Figure 3 shows a schematic of the spectroscopy system.

To evenly heat our $[Y_{1-x}Dy_x(acac)_3(phen)]$ molecular crystal powder, we use a custom powder heater. The heater consists of a block of Aluminum (2" D × 1" H) containing a 1 cm diameter indentation in which the powder is placed. This configuration allows for heating the powder from both the bottom and sides. For heating, we use a 120 W canister heater which is controlled by an Omega CN32PT-220 PID controller with a K-type thermocouple providing temperature feedback.

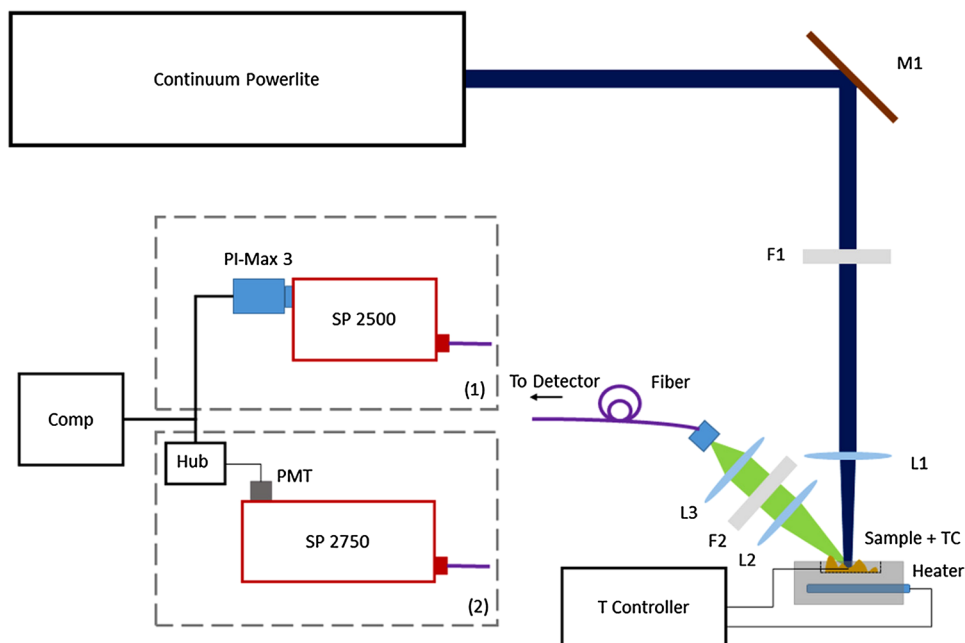
4 Results and discussion

4.1 Ungated spectra

We begin our study of $[Dy(acac)_3(phen)]$ by considering its ungated emission spectra at different Dy concentrations as shown in Fig. 4. Note that the intensity in Fig. 4 is scaled such that the intensity at 520 nm is normalized. From Fig. 4, we find that the emission spectra consist of four broad peaks corresponding to emission from Dy with the peak centers being 453 nm ($^4I_{15/2} \rightarrow ^6H_{15/2}$), 482 nm ($^4F_{9/2} \rightarrow ^6H_{15/2}$), 577 nm ($^4F_{9/2} \rightarrow ^6H_{13/2}$), and 660 nm ($^4F_{9/2} \rightarrow ^6H_{11/2}$).

In addition to the four Dy peaks in Fig. 4, we also observe a broad background emission centered at 520 nm (which is found to originate from the organic ligand) and a peak at 532 nm (corresponding to residual second harmonic in the pump). The ligand emission is isolated from

Fig. 3 Schematic of experimental setup with both the gated (1) and ungated (2) spectrometers. $F1$ 355 nm Laser line filter, $F2$ long-pass filter (400 nm cut-on), $L1$, $L2$, $L3$ Lenses, $M1$ mirror, Hub Acton Spectra Hub



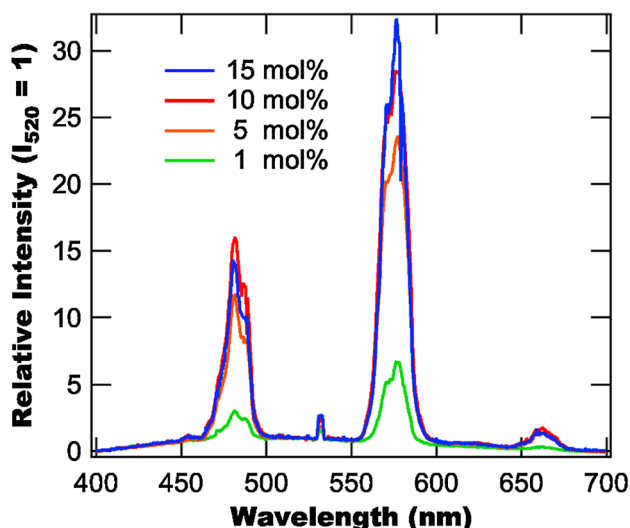


Fig. 4 Scaled emission spectra of $[Y_{1-x}Dy_x(acac)_3(phen)]$ MCs for different Dy concentrations. The spectra consist of four broad peaks from the Dy ions superimposed on a broad background emission from the organic ligand. The *small peak* at 532 nm is residual second harmonic from the pump laser

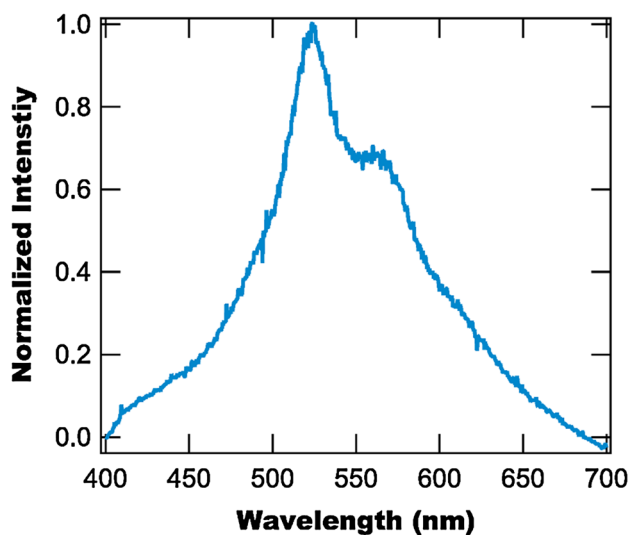


Fig. 5 Normalized emission spectrum from a $[Y(acac)_3(phen)]$ sample with 355 nm excitation. The emission is broad and spans the whole spectral region of interest

the Dy emission by measuring a pure $[Y(acac)_3(phen)]$ sample, as shown in Fig. 5.

4.2 Gated spectra

Given the proposed application of $[Y_{1-x}Dy_x(acac)_3(phen)]$ as a TCT phosphor it is necessary to remove the ligand's emission from the measured spectrum. This removal allows for an accurate probe of the Dy

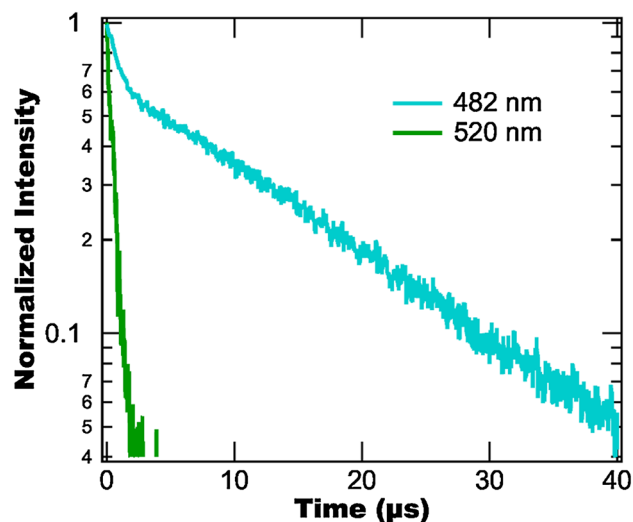


Fig. 6 Normalized 482 and 520 nm emission intensity from a $[Y_{0.99}Dy_{0.01}(acac)_3(phen)]$ sample as a function of time. The emission at 482 nm is found to be bi-exponential with lifetimes of 16 μ s and 0.6 μ s, while the 520 nm emission is found to follow a single exponential with a lifetime of 0.6 μ s. The bi-exponential behavior of the 482 nm intensity is due to emission from both the lanthanide and ligands, while the 520 nm intensity follows a single exponential as it consists of emission from the ligands only

populations in the $^4I_{15/2}$ and $^4F_{9/2}$ energy levels. While the ligand's emission can easily be subtracted using multi-peak fitting when performing spectral measurements, this technique is not an option when performing time-resolved TCT imaging. Instead, for time-resolved TCT imaging, we rely on the different luminescence lifetimes of the Dy^{3+} ions and the ligand.

To demonstrate the different lifetimes, we plot the normalized emission intensity (for a 1 mol% sample) at 482 nm (primarily due to Dy) and 520 nm (due to ligand) as a function of time in Fig. 6. From Fig. 6, we find that the 520 nm intensity follows a single exponential—with lifetime 0.6 μ s—while the 482 nm intensity is bi-exponential—with lifetimes of 16 μ s and 0.6 μ s. Given the value of the short lifetime (for the 482 nm intensity) and the observation that $[Y(acac)_3(phen)]$ has emission at 482 nm, we conclude that the fast component of the bi-exponential is due to the ligand, while the long lifetime component is due Dy's $^4F_{9/2} \rightarrow ^6H_{15/2}$ transition. The order of magnitude difference in emission lifetime between the ligands and the Dy ion means that we can remove the ligands' emission using gated measurements with a delay greater than the emission lifetime of the ligands.

Using a gate delay of 1 μ s, we re-perform the spectral measurements on the different $[Y_{1-x}Dy_x(acac)_3(phen)]$ samples (scaled spectra are shown in Fig. 7) and obtain

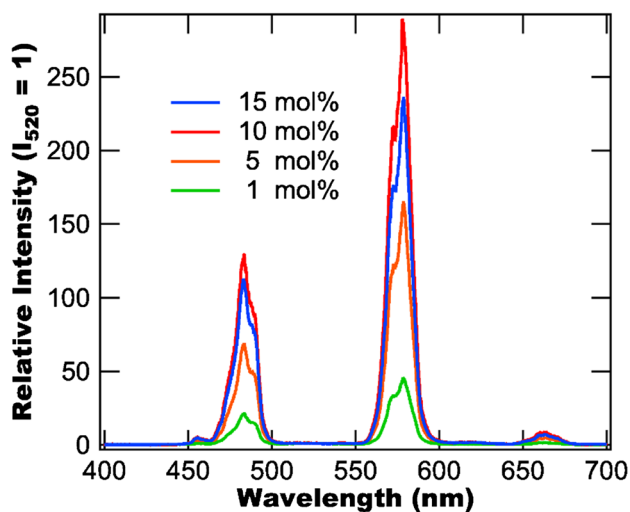


Fig. 7 Scaled gated emission spectra of $[Y_{1-x}Dy_x(acac)_3(phen)]$ MCs for different Dy concentrations

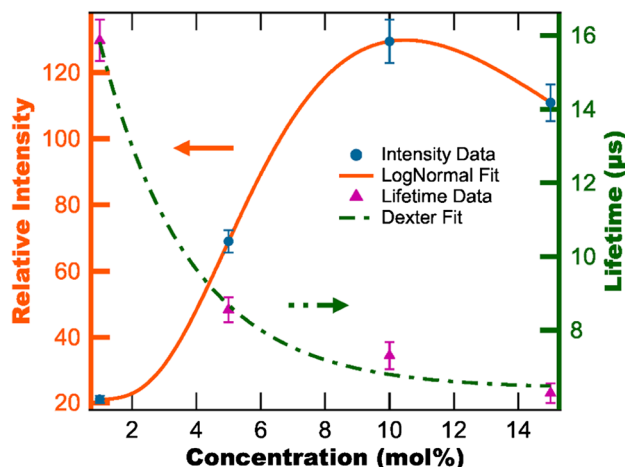


Fig. 8 Relative intensity and lifetime of the emission at 482 nm as a function of Dy concentration. The relative intensity is fit to a lognormal function and the lifetime is fit to Eq. 4.

improved contrast between the Dy and ligand emission as the ligand's emission is mostly removed. From Fig. 7, we find that the contrast between the emission at 520 and 482 nm is approximately 120, which is an order of magnitude larger than for the ungated spectral measurements (Fig. 4). This increased contrast minimizes the influence of the ligand's emission and allows for more accurate determination of the intensity ratios used in Eq. 2.

4.3 Concentration dependence

From Fig. 7, we observe that while the relative intensity of the Dy emission increases as the concentration increases

from 1 to 10 mol%, it begins to decrease as the concentration is further increased. This behavior arises as the increase in concentration initially is beneficial (providing more ions for luminescence), but as the concentration is further increased quenching becomes to dominate and limits the emission from the material. Figure 8 demonstrates this effect in the relative intensity at 482 nm as a function of concentration, with a fit to a lognormal function as a guide for the eye. From Fig. 8, we find that the 10 mol% sample provides the most emission intensity.

To further characterize concentration quenching in $[Y_{1-x}Dy_x(acac)_3(phen)]$, we measure the luminescence intensity as a function of time after excitation to determine the luminescence lifetime at different Dy concentrations. Figure 9a, b shows the emission intensity as a function of time after excitation for the different concentrations. From Fig. 9a, b we find that the luminescence decays as a bi-exponential function with the fast decay occurring with a concentration-independent lifetime of $\approx 0.6 \mu s$ and the slower decay having a concentration-dependent lifetime, which decreases with increasing concentration. The observation of the long lifetime depending on the Dy concentration and the short lifetime being invariant with concentration is additional evidence that the short lifetime component is due to the ligands, while the long lifetime component is due to the Dy ion.

Using simple exponential fits to the data in Fig. 9a, b, we determine the lifetime as a function of Dy concentration, which is shown in Fig. 8. From Fig. 8, we find that as the concentration increases the lifetime becomes shorter, with the largest change occurring when going from 1 to 5 mol%.

The influence of concentration quenching on lanthanide luminescence lifetimes has been studied in depth [48, 62–66, 76–80] with the two main mechanisms of concentration quenching in solid-state materials being long-range Energy Transfer (LRET) [62–65, 80], and short-range Dexter electron exchange [66]. These mechanisms result in different functional behavior of the lifetime as a function of concentration. Assuming that the ion has an unquenched lifetime of τ_0 , the lifetime τ (as a function of concentration C) is:

$$\tau = \frac{\tau_0}{1 + \left(\frac{C}{C_H}\right)^\alpha}, \quad (3)$$

for LRET with C_H being the half-quenching concentration and α being an exponent that depends on the energy transfer mechanism (e.g., dipole–dipole interactions and quadrupole–dipole interactions). For Dexter electron exchange, the lifetime as a function of concentration is given by:

$$\tau = \frac{\tau_0}{1 + k_D e^{-(C/C_0)^{-1/3}}}, \quad (4)$$

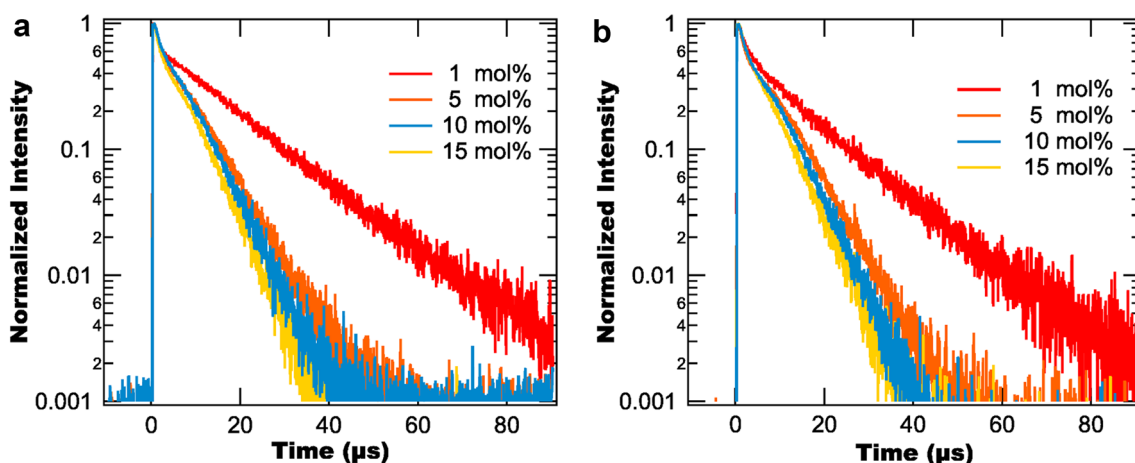


Fig. 9 Normalized luminescence intensity as a function of time for different Dy concentrations at wavelengths of 482 nm (a) and 575 nm (b). Note that the fast decay seen at early times has a concentration-

independent lifetime of $\approx 0.6\mu\text{s}$, which is consistent with emission from $[\text{Y}(\text{acac})_3(\text{phen})]$

where c_0 is the critical concentration [81] and k_D is a material dependent constant. From the functional forms of Eqs. 3 and 4, we note that LRET results in the lifetime asymptotically approaching zero, while the Dexter mechanism has the lifetime asymptotically approaching a nonzero value. Given these differences in functionality of Eqs. 3 and 4, and the observed behavior of the lifetime in Fig. 8, we conclude that the Dexter mechanism's concentration dependence best matches the observed experimental data. Therefore, we fit the lifetime data in Fig. 8 to Eq. 4 and observe good agreement.

The result of the experimental lifetimes following Eq. 4 is surprising, as typically the other quenching mechanisms are more dominant [80] as they are longer range interactions than Dexter electron exchange. Typically Dexter electron exchange-based quenching doesn't occur until the average spacing of ions is on the order of 10's of angstroms, as the Dexter mechanism is proportional to the overlap of the acceptor and donor wavefunctions [66].

Given this spatial dependence, it is necessary to estimate the ion spacing to determine if Dexter electron exchange is a valid hypothesis. To do so, we first estimate the particle density using the lattice volume (see Table 1), which gives a particle density of $n_p = 3.52 \times 10^{26} \text{ m}^{-3}$. This density represents the number of molecular complexes in a given volume, but not the density of Dy ions. To get the density of Dy ions, we need to scale the particle density by the doping percentage, which gives the ion density to be $n_i = cn_p$, where c is the doping percentage. While a precise calculation of the mean interion distance $\langle r \rangle$, from the concentration is difficult, we can get a rough estimate using the formula $\langle r \rangle \sim 1/n_i^{1/3}$. Using this formula, we get an inter-ion distance of 65.7 Å for the 1 mol% sample and 26.7 Å for the 15 mol% sample.

These spacings suggest that the Dexter mechanism is a valid hypothesis for our observed concentration dependence. However, further study (using more concentrations and a calculation of the exchange radius) is required to verify this result.

4.4 Temperature dependence

4.4.1 Emission spectra

With the concentration dependence of the emission of $[\text{Y}_{1-x}\text{Dy}_x(\text{acac})_3(\text{phen})]$ characterized, we now turn to

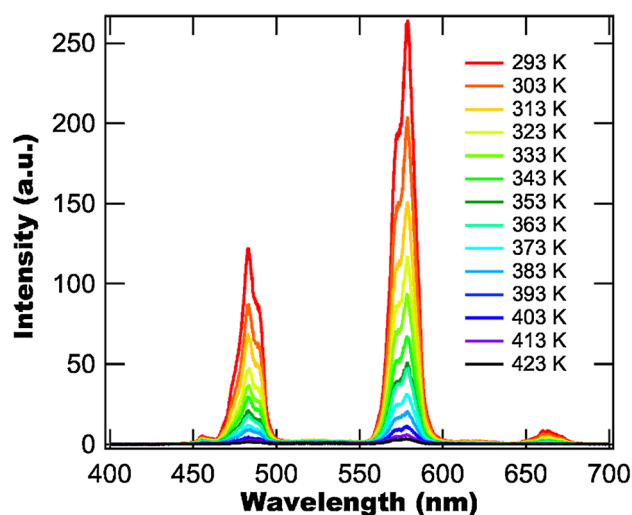


Fig. 10 Emission spectra of a $[\text{Y}_{1-x}\text{Dy}_x(\text{acac})_3(\text{phen})]$ (Dy concentration of 10 mol%) at 14 different temperatures. The emission displays strong thermal quenching

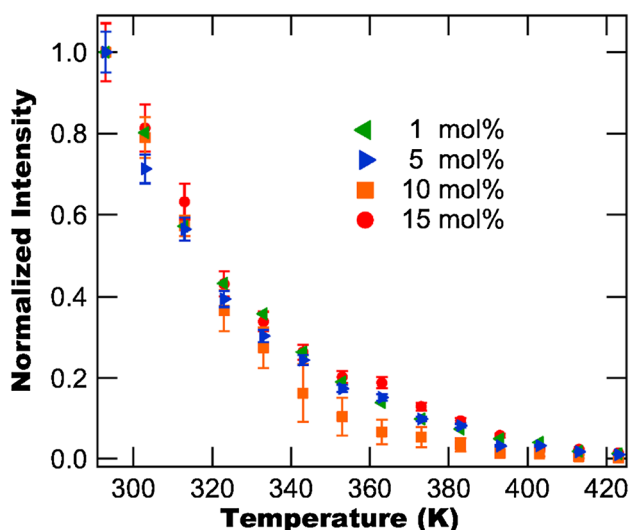


Fig. 11 Normalized emission intensity at 482 nm as a function of temperature for different Dy concentrations, with the intensity normalized to the room temperature value. The emission shows strong thermal quenching

consider the influence of temperature on its emission. To do so, we use the gated spectroscopy system to measure the temperature-dependent emission of $[Y_{1-x}Dy_x(acac)_3(phen)]$ at 14 different temperatures ranging from 293 to 423 K. Figure 10 shows an example set of emission spectra for the 10 mol% sample. From Fig. 10, we find that the emission spectra display strong thermal quenching, which is seen in the emission from all concentrations tested.

We characterize the material's thermal quenching by plotting the emission intensity at 482 nm as a function of temperature for all four concentrations in Fig. 11, with the intensity normalized such that the room temperature intensity is unity. From Fig. 11, we find that the intensity as a function of temperature decays exponentially with each concentration's curve being within uncertainty of each other. Fitting each curve to a simple exponential and taking the weighted average of the decay constants over all four concentrations, we find that the average decay constant is 36.89 ± 0.84 K.

To consider what this temperature dependence means practically, we assume that we have a detection system capable of accurately measuring down to 1% of the room temperature signal. Using the decay constant determined above, this dynamic range gives a functional maximum temperature of approximately 463 K. However, for our current detection system, we find that for temperatures above 423 K the signal to noise ratio is unacceptably small (≈ 2 for the 455 nm peak and ≈ 16 for the 482 nm peak). Thus, our current maximum temperature limit is ≈ 423 K.

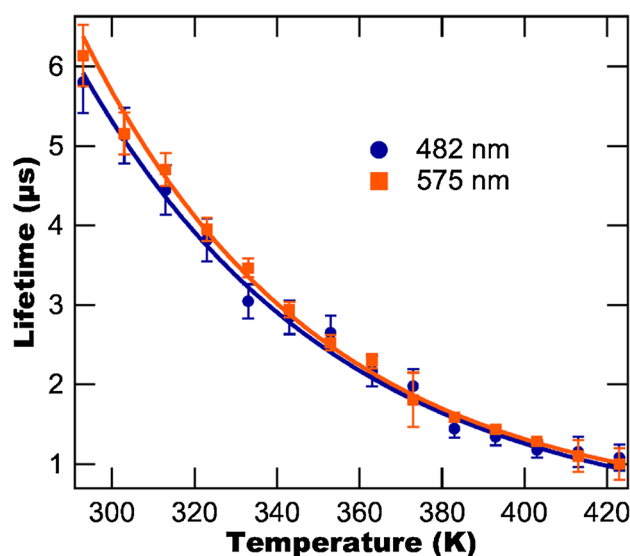


Fig. 12 Lifetimes of the luminescence emission at 482 and 575 nm as a function of temperature for $[Y_{0.9}Dy_{0.1}(acac)_3(phen)]$. As the temperature increases, the lifetimes decrease due to thermal quenching

4.4.2 Luminescence lifetime

In addition to the thermal quenching affecting the emission intensity, we also find that it affects emission lifetime, which is consistent with previous measurements of other lanthanide-doped complexes [77, 81–83]. We characterize the influence of thermal quenching on the lifetime of $[Y_{1-x}Dy_x(acac)_3(phen)]$ by measuring the lifetime of the emission at 482 and 575 nm for the 10 mol% sample, as shown in Fig. 12. From Fig. 12, we find that thermal quenching in $[Y_{0.9}Dy_{0.1}(acac)_3(phen)]$ results in the lifetime decreasing with increasing temperature.

This decrease in lifetime with increasing temperature is typically modeled as a barrier process with the rate of energy loss from the excited state k , being

$$\begin{aligned} k &= k_0 + k_{NR}(T), \\ &= k_0 + k_1 e^{-\Delta E_T/kT}, \end{aligned} \quad (5)$$

where k_0 is the excited state's natural decay rate, $k_{NR}(T) = k_1 e^{-\Delta E_T/kT}$ is the rate of nonradiative energy transfer as a function of temperature T , with ΔE_T being the energy transfer barrier and k_1 being a rate related to non-radiative energy transfer from the excited state. Recalling that the lifetime of the excited state is given by $\tau = 1/k$, we use Eq. 5 to give the lifetime as a function of temperature:

$$\begin{aligned} \tau &= \frac{1}{k_0 + k_1 e^{-\Delta E_T/kT}}, \\ &= \frac{\tau_0}{1 + \frac{\tau_0}{\tau_1} e^{-\Delta E_T/kT}}, \end{aligned} \quad (6)$$

Table 2 Fit parameters determined from lifetimes as a function of temperature

| Parameter | 482 nm | 575 nm |
|------------------------------|----------------|----------------|
| τ_0 (μ s) | 11.6 ± 4.5 | 13.8 ± 3.9 |
| τ_1 (ns) | 4.2 ± 2.5 | 5.3 ± 1.7 |
| ΔE_T (10^{-3} eV) | 200 ± 28 | 193 ± 13 |

The parameters for the lifetime at 482 and 575 nm are found to be within uncertainty of each other

where $\tau_i = 1/k_i$.

Fitting the lifetime data in Fig. 12 to Eq. 6, we determine the different parameters for both the 482 and 575 nm peaks with the results tabulated in Table 2. From Table 2, we find that the parameters for both the 482 and 575 nm peak are within uncertainty of each other. This is expected as both peaks correspond to transitions from the $^4F_{9/2}$ energy level.

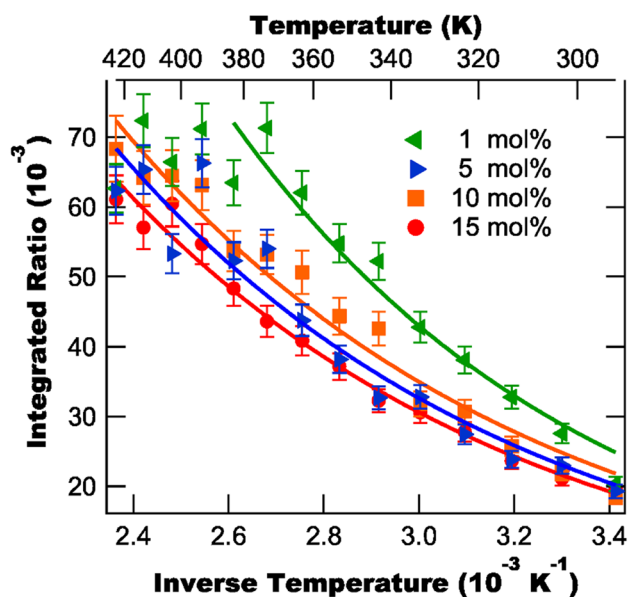
4.4.3 Intensity ratios

At this point, we have characterized the emission properties of $[Y_{1-x}Dy_x(acac)_3(phen)]$ for different temperatures and concentrations. The next step toward using $[Y_{1-x}Dy_x(acac)_3(phen)]$ as a TCT phosphor is to characterize the excited state ratios between the $^4I_{15/2}$ and $^4F_{9/2}$ states as a function of temperature. This ratio is probed by measuring the ratio between the emission peak at 455 nm ($^4I_{15/2} \rightarrow ^6H_{15/2}$) and 482 nm ($^4F_{9/2} \rightarrow ^6H_{15/2}$). In our study—to improve the signal to noise ratio—we use the spectrally integrated peak intensities to calculate the peak ratios with the 455 nm peak having an integration range of 450–460 nm and the 482 nm peak having an integration range of 465–500 nm.

Using the gated spectra measured at different temperatures for each concentration, we integrate the spectral peaks and calculate their ratios. The ratios of the integrated peaks are shown in Fig. 13 as a function of inverse temperature. Note that for the 1:99 sample, the intensity at temperatures above 383 K becomes so weak as to make the ratios unreliable. These ratios are therefore not used when performing fitting.

From Fig. 13, we find that for all four concentrations the integrated ratio behaves as an exponential with inverse temperature, which is consistent with Eq. 2. Fitting the ratios in Fig. 13 to Eq. 2, we determine both the amplitude parameter A and the energy barrier ΔE , which are tabulated in Table 3.

From Table 3, we find that the energy differences are all within uncertainty of each other and have a weighted mean of $100.2(\pm 3.2) \times 10^{-3}$ eV. This energy difference is significantly less than the expected energy difference of ≈ 0.15 eV (energy difference between photons with wavelengths of 482 and 455 nm). While the measured

**Fig. 13** Integrated peak ratio (455 nm/482 nm) as a function of temperature for four different Dy concentrations

difference is less than the difference expected due to the peak wavelengths, it is actually consistent with the energy difference determined for Dy^{3+} in a water solution: $\Delta E \approx 0.1147$ eV (energy difference between photons with wavelengths of 475 and 455 nm) [43–46].

To understand why the energy difference measured in this study (and seen in the literature [24, 28, 43–46, 84]) differs from the value determined using the peak wavelengths, we note that the peak spacing (455 vs. 482 nm) does not directly represent the spacing between the $^4I_{15/2}$ and $^4F_{9/2}$ energy levels. In reality, each broad spectral peak consists of multiple overlapping smaller peaks that arise due to transitions from the excited state into the Stark-split $^6H_{15/2}$ ground state [45, 85].

Therefore, to accurately determine the spacing between the $^4I_{15/2}$ and $^4F_{9/2}$ energy levels using peak positions, we need to use transitions from the excited states into the same $^6H_{15/2}$ sub-level. This, however, is difficult to accomplish when the transitions to the different sub levels are overlapping from thermal and/or inhomogeneous broadening. This broadening can be

Table 3 Exponential fit parameters determined from Fig. 13 for different Dy concentrations

| Dy concentration (mol%) | A | ΔE (10^{-3} eV) |
|-------------------------|-----------------|----------------------------|
| 1 | 2.30 ± 0.87 | 114 ± 14 |
| 5 | 1.06 ± 0.31 | 100.1 ± 9.4 |
| 10 | 1.08 ± 0.20 | 98.4 ± 6.2 |
| 15 | 0.98 ± 0.13 | 99.7 ± 4.3 |

minimized either using low temperatures [85] or a host material with a strong crystal field [45]. Alternatively, by integrating over the full peak (as we do in this study), we can account for all transitions from the ${}^4I_{15/2}$ and ${}^4F_{9/2}$ energy levels into the ${}^6H_{15/2}$ state and obtain an accurate energy difference.

Having determined the temperature response of the intensity ratio, we can now calculate the sensitivity of $[Y_{1-x}Dy_x(acac)_3(phen)]$ and the temperature resolution of our TCT technique. The sensitivity S of a TCT phosphor is given by [86]

$$S = \frac{100}{r} \frac{\partial r}{\partial T}, \quad (7)$$

where r is the intensity ratio and the sensitivity is given in units of $\% K^{-1}$. To calculate the sensitivity, we first take the derivative of the intensity ratio function (Eq. 2) and substitute in the average amplitude and energy difference from Table 1. We then divide the calculated derivative by the experimentally obtained ratio (for the 10 mol% samples) and compute Eq. 7 with Fig. 14 showing the computed sensitivity. From Fig. 14, we find that the sensitivity has a maximum value of $1.5\% K^{-1}$, which is comparable to many other temperature-sensing phosphors [86].

In addition to calculating the sensitivity of our phosphor, we also consider the temperature resolution ΔT , of our technique, which not only depends on the sensitivity of the phosphor, but also the experimental uncertainty of the optical detection system. The temperature resolution can be defined as the temperature change required to change the ratio by an amount equal to the experimental uncertainty. Mathematically this takes the form of

$$\Delta T = \sigma_r \left(\frac{\partial r}{\partial T} \right)^{-1}, \quad (8)$$

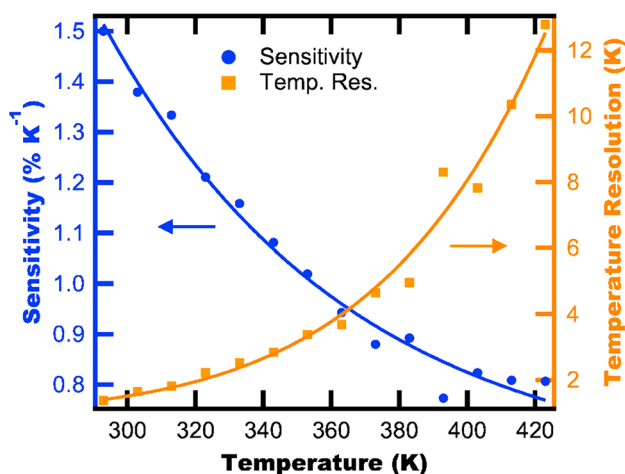


Fig. 14 Sensitivity and temperature resolution of our TCT method as a function of temperature

where σ_r is the experimental uncertainty in the ratio.

Using the calculated derivative of the ratio and the ratio uncertainty for the 10 mol% sample, we calculate the temperature resolution and show the results in Fig. 14. From Fig. 14, we find that at room temperature the temperature resolution is approximately 1.8 K and that it increases with temperature. This behavior is primarily due to the effects of thermal quenching, which is found to increase the relative uncertainty of the measured ratio as the temperature is increased.

4.5 Photodegradation

In the previous section, we measured the 455 nm/482 nm intensity ratios of $[Y_{1-x}Dy_x(acac)_3(phen)]$ at different temperatures and determined the calibration parameters for the material's temperature response. While this analysis is the main purpose of our study, we also observe significant yellowing of the $[Y_{1-x}Dy_x(acac)_3(phen)]$ powder during extended UV exposure, which suggests that the material undergoes significant photodegradation.

Given that $[Y_{1-x}Dy_x(acac)_3(phen)]$ consists of either a Dy or Y ion connected to organic ligands, photodegradation will occur solely due to the ligands, implying that the Dy spectral ratios should be unaffected (assuming the ligand stays transparent in the 450–500 nm range). However, while the ratios should be unaffected, the overall intensity will be decreased as the ligand acts as an “antenna” for the Dy ion [52] and damage to the ligand will decrease the efficiency of energy transfer between the ligand and Dy ion.

To characterize this decrease in energy transfer efficiency (and confirm that the ratios are unaffected), we measure the emission spectra between 450 and 500 nm for a 10 mol% sample as a function of time during constant UV illumination at two different temperatures (293 and 393 K). For both degradation and excitation, we use the same 355 nm pulsed laser with the fluence per pulse being 89 mJ/cm^2 (average intensity of 0.891 W/cm^2). Figure 15 shows the scaled integrated intensity of the 482 nm peak (integration range 465–500 nm, smoothed for clarity) as a function of time for both 293 and 393 K, with the intensity scaled such that the initial intensity is 1.

From Fig. 15, we find that the scaled integrated intensity of the 482 nm peak decays exponentially with time, with the higher temperature being found to decay more quickly. Fitting the decay curves in Fig. 15 to a simple exponential function, we find decay rates of $\gamma = 1.40(\pm 0.26) \times 10^{-3} \text{ s}^{-1}$ for the 293 K curve and $\gamma = 3.29(\pm 0.56) \times 10^{-3} \text{ s}^{-1}$ for the 393 K curve.

Comparing the decay rates between the two temperatures, we find that by increasing the temperature by 100 K the ligand decays 2.35× more quickly than at room temperature. This temperature dependence suggests that photodegradation

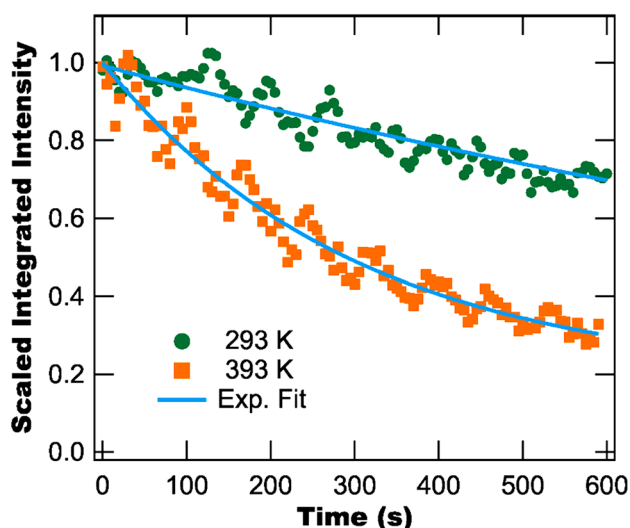


Fig. 15 Scaled integrated intensity (465–500 nm) as a function of time during 355 nm exposure for two different temperatures. The material is found to photodegrade under 355 nm illumination with the decay rate increasing with temperature. Note that the data is smoothed using a binomial algorithm for clarity

is due to an energy barrier process. To estimate this energy barrier, we first assume that the decay rate depends on temperature as an Arrhenius function:

$$\gamma = \gamma_0 e^{-\Delta E_D/kT}, \quad (9)$$

where γ_0 is the asymptotic decay rate and ΔE_D is the photodegradation energy barrier. With this assumption, we calculate the ratio of decay rates to be,

$$\frac{\gamma_1}{\gamma_2} = \exp \left[-\frac{\Delta E_D}{k} \left(\frac{1}{T_1} - \frac{1}{T_2} \right) \right], \quad (10)$$

where γ_1 is the decay rate at temperature T_1 and γ_2 is the decay rate at temperature T_2 . Equation 10 can be rearranged to give the energy barrier to be

$$\Delta E_D = -k \frac{T_1 T_2}{T_2 - T_1} \ln \left(\frac{\gamma_1}{\gamma_2} \right). \quad (11)$$

Substituting in our experimental values, we can estimate the energy barrier to be $\Delta E_D = 0.084 \pm 0.011$ eV. Note that a more accurate determination—using degradation at many different temperatures—of this energy barrier is beyond the scope of this current study.

In addition to determining the temperature dependence of (acac)₃(phen)'s photodegradation, we also compare its photostability to other organic materials by computing the commonly used figure of merit for organic dyes [87]:

$$\text{FoM} = \frac{B}{\sigma} \quad (12)$$

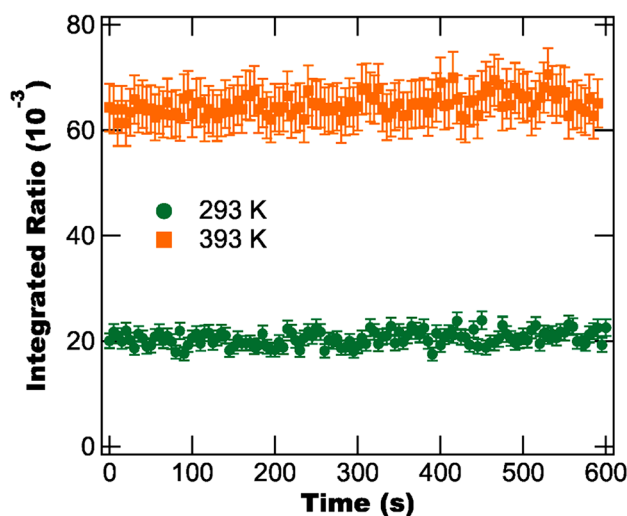


Fig. 16 Integrated ratio (455 nm/482 nm) as a function of time during photodegradation. The ratio is found to be invariant within experimental uncertainty for the entire degradation time period

$$= \frac{\langle I \rangle}{\gamma \hbar \omega}, \quad (13)$$

where σ is the undamaged absorbance cross section, B is the degradation's inverse quantum efficiency, $\langle I \rangle$ is the average pump intensity, γ is the decay rate, and $\hbar \omega$ is the photon energy of the degradation source. Using Eq. 13 and the values from our room temperature (293 K) measurement, we determine a FoM of $1.13(\pm 0.21) \times 10^{21} \text{ cm}^{-2}$, which is on the same order as a large number of organic materials, but almost two orders of magnitude below the most photostable organic materials [87].

With the influence of (acac)₃(phen)'s photodegradation on the emission intensity now determined, we turn to considering how (acac)₃(phen)'s photodegradation affects the intensity ratio between the 455 nm peak and the 482 nm peak. Using the same calculation method described in Sect. 4.4.3, we calculate the integrated peak ratio as a function of temperature for both temperatures as shown in Fig. 16. From Fig. 16, we find that for both 293 and 393 K the integrated ratio is invariant within uncertainty for the entire time frame of photodegradation. This result confirms our initial hypothesis that photodegradation won't influence the intensity ratios.

4.6 Dynamic heating test

Having now determined the temperature dependence of the integrated ratio and shown that the ratio is invariant during ligand photodegradation, we now demonstrate the use of [Y_{1-x}Dy_x(acac)₃(phen)] as a TCT phosphor. To do so, we heat a sample (10 mol%) and measure the luminescence

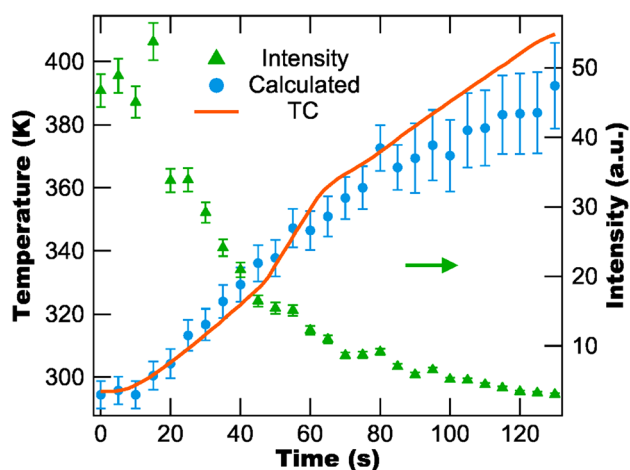


Fig. 17 Temperature as a function of time as measured by a TC and calculated using TCT with $[Y_{0.9}Dy_{0.1}(acac)_3(phen)]$. For comparison, we also include the intensity at 482 nm as a function of time. We find that as the phosphor intensity approaches zero, the calculated temperature becomes more noisy, with the peak accurate temperature near 370 K

spectrum and TC temperature as a function of time using the gated spectroscopy system.

After heating, we compute the integrated intensity ratio as a function of time and calculate the temperature using Eq. 1 and the calibration values. Figure 17 shows the calculated temperature, TC value, and the peak intensity at 482 nm as a function of time. From Fig. 17, we find that the calculated temperature is within uncertainty of the TC temperature for temperatures up to about 370 K after which point the calculated temperatures are more noisy. This noise in temperatures is due to the emission intensity decreasing to the point where the SNR is too small to obtain accurate temperatures.

From Figs. 11 and 17, it should be noted that the maximum accurate temperature is lower in Fig. 17 than in Fig. 11. The reason for this difference is related to the different camera settings between the static calibration measurements and the dynamic heating measurements. During the calibration measurements, the camera gain is adjusted for each temperature to account for the decrease in intensity, while during dynamic heating the gain is fixed for the entire run. This leads to the intensity quickly falling below the accurate detection limit. We are currently working on developing new software which will dynamically adjust the camera gain allowing for a larger temperature range.

5 Conclusions

We develop a TCT phosphor based on a $[Y_{1-x}Dy_x(acac)_3(phen)]$ molecular crystals. The emission (using an

excitation wavelength of 355 nm) is found to contain four peaks originating from the Dy ion and a broad background peak arising from the ligand. The emission from the Dy ion at room temperature is found to have a lifetime ranging from 6 μ s to 16 μ s depending on Dy concentration, and the emission from the ligand at room temperature has a lifetime of approximately 600 ns. This difference in lifetime allows us to remove the ligand's contribution from the emission spectra using time-gated spectroscopy.

Using this technique, we characterize the material's emission for four different concentrations and 14 different temperatures. From these measurements, we find that the material displays both concentration and thermal quenching, which results in decreased emission intensity and luminescence lifetimes at high concentrations and temperatures. With regards to concentration quenching, we find that the peak intensity occurs for a Dy concentration of ≈ 10 mol% and that the lifetime as a function of concentration appears to follow the functionality of Dexter electron exchange quenching. While the estimated ion spacing is found to be the correct order of magnitude for the Dexter exchange mechanism, a more in depth study of the concentration dependence is needed to verify this hypothesis.

We also find that the effect of thermal quenching on $[Y_{1-x}Dy_x(acac)_3(phen)]$'s emission is consistent with results seen in other lanthanide-doped complexes. We find that the emission intensity decays exponentially in the temperature range tested (with the $1/e$ intensity decay temperature being 36.89 ± 0.84 K) and that the emission lifetime obeys Eq. 6 with a thermal quenching barrier of ≈ 0.2 eV. Due to this thermal quenching, we find that—for our current spectroscopy system—we can accurately measure the emission to a temperature of ≈ 423 K.

Additionally, we find that the organic ligand photodegrades under UV illumination resulting in a decrease of the Dy emission peaks as the ligand acts as an “antenna” for the Dy ion. Despite this decrease in luminescence efficiency, due to photodegradation, we find that the ratio between the 455 and 482 nm peaks is unchanged.

With the emission of $[Y_{1-x}Dy_x(acac)_3(phen)]$ characterized, we next demonstrate its use as a TCT phosphor by heating a 10 mol% sample dynamically and measuring the emission as a function of time using our time-gated spectroscopy system. Based on the emission spectra (and using the known calibration parameters) we compute the temperature and compare the calculated value to that measured by a thermocouple with both temperatures found to be within uncertainty of each other up to a temperature of approximately 370 K, with the main limitation due to the current spectroscopy system's software. To remove this limitation, we are currently developing improved software, which will allow for a higher temperature range with our initial target maximum temperature being 423 K.

In summary, we find that $[Y_{1-x}Dy_x(acac)_3(phen)]$ forms organic molecular crystals with appropriate spectroscopic properties for use as a TCT phosphor. Our next step—toward using $[Y_{1-x}Dy_x(acac)_3(phen)]$ as a TCT phosphor during shock compression of heterogeneous materials—is to perform fast laser heating of the phosphor to demonstrate its ability to measure temperature during short time scales. At the same time, we are also testing other Dy^{3+} -doped Yttrium ternary complexes to determine how their thermal performance compares to $[Y_{1-x}Dy_x(acac)_3(phen)]$.

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References

- M. Chen, S. You, K. Suslick, D. Dlott, Hot spots in energetic materials generated by infrared and ultrasound, detected by thermal imaging microscopy. *Rev. Sci. Instrum.* **85**, 023705 (2014)
- M. Chen, S. You, K. Suslick, D. Dlott, Hot spot generation in energetic materials created by long-wavelength infrared radiation. *App. Phys. Lett.* **104**, 061907 (2014)
- R. Pokharel, J. Lind, A. Kanjarla, R. Lebensohn, S. Li, P. Kensei, R. Suter, A. Rollett, Polycrystal plasticity: comparison between grain-scale observations of deformation and simulations. *Annu. Rev. Condens. Matter Phys.* **5**, 317–346 (2014)
- Q. An, W. Goddard, S. Zybin, A. Jaramillo-Botero, T. Zhou, Highly shocked polymer bonded explosives at a nonplanar interface: hot-spot formation leading to detonation. *J. Phys. Chem. C* **117**, 26551–26561 (2013)
- Y. Wu, F. Huang, A microscopic model for predicting hot-spot ignition of granular energetic crystals in response to drop-weight impacts. *Mech. Mater.* **43**, 835–852 (2011)
- P. Rae, H. Goldrein, S. Palmer, J. Field, A. Lewis, Quasi-static studies of the deformation and failure of beta-hmx based polymer bonded explosives. *Proc. R. Soc. A Math. Phys. Eng. Sci.* **458**, 743–762 (2002)
- Y. Gruzdkov, Y. Gupta, Vibrational properties and structure of pentaerythritol tetranitrate. *J. Phys. Chem. A* **105**, 6197–6202 (2001)
- Y. Gupta, Recent developments to understand molecular-changes in shocked energetic materials. *Journal De Physique IV* **5**, 345–356 (1995)
- A. Mellor, D. Wiegand, K. Isom, Hot-spot histories in energetic materials. *Struct. Prop. Energ. Mater.* **296**, 293–298 (1993)
- J. Field, Hot-spot ignition mechanisms for explosives. *Acc. Chem. Res.* **25**, 489–496 (1992)
- J. Field, N. Bourne, S. Palmer, S. Walley, J. Smallwood, Hot-spot ignition mechanisms for explosives and propellants. *Philos. Trans. R. Soc. Lond. Math. Phys. Eng. Sci.* **339**, 269–283 (1992)
- J. Field, G. Swallowe, S. Heavens, Ignition mechanisms of explosives during mechanical deformation. *Proc. R. Soc. Lond. Ser. A* **382**, 231 (1982)
- R. Winter, E. Faber, The role of localized plastic flow in the impact initiation of explosives. *Proc. R. Soc. Lond. Ser. A* **343**, 399–413 (1975)
- M. Chaudhri, J. Field, The role of rapidly compressed gas pockets in the initiation of condensed explosives. *Proc. R. Soc. Lond.* **340**, 113–128 (1974)
- S.N. Luo, B.J. Jensen, D.E. Hooks, K.J. Ramos, J.D. Yeager, K. Kwiatkowski, T. Shimada, D.A. Fredenburg, K. Fezzaa (2012) Ultrafast, high resolution, phase contrast imaging of shock response with synchrotron radiation: opportunities and challenges, in *APS meeting abstracts*
- B.J. Jensen, K.J. Ramos, A.J. Iverson, J. Bernier, C.A. Carlson, J.D. Yeager, K. Fezzaa, D.E. Hooks, Dynamic experiment using impulse at the advanced photon source. *J. Phys. Conf. Ser.* **500**, 042001 (2014)
- B.J. Jensen, S.N. Luo, D.E. Hooks, K. Fezzaa, K.J. Ramos, J.D. Yeager, K. Kwiatkowski, T. Shimada, D.M. Dattelbaum, Ultrafast, high resolution, phase contrast imaging of impact response with synchrotron radiation. *AIP Adv.* **2**, 012170 (2012)
- J. Yeager, S. Luo, B. Jensen, K. Fezzaa, D. Montgomery, D. Hooks, High-speed synchrotron x-ray phase contrast imaging for analysis of low-z composite microstructure. *Compos. Part A Appl. Sci. Manuf.* **43**, 885–892 (2012)
- S.N. Luo, B.J. Jensen, D.E. Hooks, K. Fezzaa, K.J. Ramos, J.D. Yeager, K. Kwiatkowski, T. Shimada, Gas gun shock experiments with single-pulse X-ray phase contrast imaging and diffraction at the advanced photon source. *Rev. Sci. Instrum.* **83**, 073903 (2012)
- DCS facility. <http://www.dcs-aps.wsu.edu/>
- New research opportunities in dynamic compression science, Tech. rep., Institute for Shock Physics, Washington State University (2012)
- A.Z.M. Al-Juboori, Rare earth (Sm^{3+} and Dy^{3+})-doped gadolinium oxide nanomaterials for luminescence thermometry. *Phys. Scr.* **T157**, 014004 (2013)
- A. Heyes, S. Seefeldt, J. Feist, Two-colour phosphor thermometry for surface temperature measurement. *Optics Laser Technol.* **38**, 257–265 (2006)
- A. Khalid, K. Kontis, Thermographic phosphors for high temperature measurements: principles, current state of the art and recent applications. *Sensors* **8**, 5673–5744 (2008)
- K. Kontis, A review of some current research on pressure sensitive and thermographic phosphor techniques. *Aeronaut. J.* **3162**, 495–508 (2007)
- S.W. Allison, G. Gillies, Remote thermometry with thermographic phosphors: Instrumentation and applications. *Rev. Sci. Instrum.* **68**, 2615–2650 (1997)
- M. Cates, S. Allison, S. Jaiswal, D. Beshears, YAG: Dy and YAG: Tm fluorescence to 1700°C. in *The 49th International Instrumentation Symposium—The Instrumentation, Systems, and Automation Society* (Orlando, FL, 2003)
- G. Gross, A. Smith, M. Post, Surface thermometry by laser induced fluorescence. *Rev. Sci. Instrum.* **60**, 12 (1989)
- K. Kontis, Y. Syogenji, N. Yoshikawa, Surface thermometry by laser induced fluorescence of Dy^{3+} :YAG. *Aeronaut. J.* **106**, 453–457 (2002)
- K. Kontis, Surface heat transfer measurements inside a supersonic combustor by laser induced fluorescence. *J. Thermophys. Heat Transf.* **17**, 320–325 (2003)
- R. Hasegawa, I. Sakata, H. Yanagihara, B. Johansson, A. Omrane, M. Alden, Two-dimensional gas phase temperature measurements using phosphor thermometry. *App. Phys. B* **88**, 291–296 (2007)
- G. Sarner, A. Omrane, H. Seyfried, M. Richter, H. Schmidt, M. Alden, Laser diagnostics applied to a full-size fighter jet afterburner, in *European Combustion Meeting* (2005)
- J. Feist, A. Heyes, S. Seefeldt, Thermographic phosphors for gas turbines: instrumentation development and measurement uncertainties, in *11th international symposium on application of laser techniques to fluid mechanics* (Lisbon, Portugal, 2002)
- J.P. Feist, A.L. Heyes, J.R. Nicholls, Phosphor thermometry in an electron beam physical vapour deposition produced

- thermal barrier coating doped with dysprosium, in *Proceedings of the Institution of Mechanical Engineers*, vol. 215 (2001), pp. 333–341
35. J.P. Kumar, G. Ramgopal, Y. Vidya, K. Anantharaju, B.D. Prasad, S. Sharma, S. Prashantha, H. Nagaswarupa, D. Kavyashree, H. Nagabhushana, Green synthesis of $Y_2O_3:Dy^{3+}$ nanophosphor with enhanced photocatalytic activity. *Spectrochim. Acta Part A Mol. Biomol. Spectrosc.* **149**, 687–697 (2015)
 36. S.K. Gupta, M. Mohapatra, V. Natarajan, S.V. Godbole, Photoluminescence investigations of the near white light emitting perovskite ceramic $SrZrO_3:Dy^{3+}$ prepared via gel-combustion route. *Int. J. Appl. Ceram. Technol.* **10**, 593–602 (2013)
 37. M. Jayasimhadri, B.V. Ratnam, K. Jang, H.S. Lee, B. Chen, S.-S. Yi, J.-H. Jeong, L. Rama Moorthy, Combustion synthesis and luminescent properties of nano and submicrometer-size $Gd_2O_3:Dy^{3+}$ phosphors for white leds. *Int. J. Appl. Ceram. Technol.* **8**, 709–717 (2011)
 38. B. Ratnam, M. Jayasimhadri, K. Jang, H. Sueb Lee, S.-S. Yi, J.-H. Jeong, White light emission from $NaCaPO_4:Dy^{3+}$ phosphor for ultraviolet-based white light-emitting diodes. *J. Am. Ceram. Soc.* **93**, 3857–3861 (2010)
 39. J. Kaur, Y. Parganiha, V. Dubey, D. Singh, D. Chandrakar, Synthesis, characterization and luminescence behavior of $ZrO_2:Eu^{3+}$, Dy^{3+} with variable concentration of Eu and Dy doped phosphor. *Superlattices Microstruct.* **73**, 38–53 (2014)
 40. Y. Akita, T. Harada, R. Sasai, K. Tomita, K. Nishiyama, Emission properties of Ln (Eu, Tb, Dy, Er)-doped Y_2O_3 nanoparticles synthesized by surfactant-assembly and their applications in visible color-tuning. *J. Photochem. Photobiol. A Chem.* **299**, 87–93 (2015)
 41. B.R. Anderson, R. Gunawidjaja, P. Price, H. Eilers, Spectroscopic determination of thermal impulse in sub-second heating events using lanthanide-doped oxide precursors and phenomenological modeling. *J. Appl. Phys.* **120**, 083102 (2016)
 42. R. Gunawidjaja, B.R. Anderson, H.D. y Riega, H. Eilers, Sub-second laser heating of thermal impulse sensors, in *AIP conference proceedings* (2016) (pending publication).
 43. W.T. Carnall, P.R. Fields, K. Rajnak, Electronic energy levels in the trivalent lanthanide aquo ions. *J. Chem. Phys.* **49**, 4424–4442 (1968)
 44. W. Carnall, The absorption and fluorescence spectra of rare earth ions in solution, in *Handbook on the Physics and Chemistry of Rare Earths*, chap. 24, eds. by K. Gschneidner, L. Eyring (Elsevier BV, Amsterdam, 1979)
 45. J.-C.G. Bünzli, S.V. Eliseeva, Basics of lanthanide photophysics, in *Lanthanide Luminescence: Photophysical, Analytical and Biological Aspects*, eds. P. Hanninen, H. Harma (Springer, New York, 2010)
 46. J. Bünzli, Lanthanide luminescence for biomedical analyses and imaging. *Chem. Rev.* **110**, 2729–2755 (2010)
 47. N.F.W.F. Sager, F.A. Serafin, Substituent effects on intramolecular energy transfer. I. Absorption and phosphorescence spectra of rare earth β -diketone chelates. *J. Phys. Chem.* **69**, 1092 (1965)
 48. D. Dexter, A theory of sensitized luminescence in solids. *J. Chem. Phys.* **21**, 836 (1953)
 49. H. Xin, M. Shi, X.C. Gao, Y.Y. Huang, Z.L. Gong, D.B. Nie, H. Cao, Z.Q. Bian, F.Y. Li, C.H. Huang, The effect of different neutral ligands on photoluminescence and electroluminescence properties of ternary terbium complexes. *J. Phys. Chem. B* **108**, 10796–10800 (2004)
 50. S.I. Klink, Synthesis and photophysics of light-converting lanthanide complexes, Ph.D. thesis, University of Twente (2000)
 51. M. Latva, H. Takalo, V.-M. Mukkala, C. Matescu, J.C. Rodriguez-Ubis, J. Kankare, Correlation between the lowest triplet state energy level of the ligand and lanthanide(III) luminescence quantum yield. *J. Lumin.* **75**, 149–169 (1997)
 52. J. Feng, L. Zhou, S.-Y. Song, Z.-F. Li, W.-Q. Fan, L.-N. Sun, Y.-N. Yu, H.-J. Zhang, A study on the near-infrared luminescent properties of xerogel materials doped with dysprosium complexes. *Dalton Trans.* **33**, 6593–6598 (2009)
 53. A. Adronov, J.M. Frechet, Light-harvesting dendrimers. *Chem. Commun.* **18**, 1701–1710 (2000)
 54. R.S. Dickens, J.A.K. Howard, C.L. Maupin, J.M. Moloney, D. Parker, R.D. Peacock, J.P. Riehl, G. Siligardi, Ground and excited state chiroptical properties of enantiopure macrocyclic tetranaphthyl lanthanide complexes: controlled modulation of the frequency and polarisation of emitted light. *New J. Chem.* **22**, 891–899 (1998)
 55. J. Bünzli, Luminescent probes, in “Lanthanide Probes in Life Chemical and Earth Sciences: Theory and Practice,” , J. Bunzli and G. Choppin, eds. (Elsevier, 1989), pp. 219–293
 56. N.M. Shavaleev, S.V. Eliseeva, R. Scopelliti, J.-C.G. Bünzli, N-aryl chromophore ligands for bright europium luminescence. *Inorg. Chem.* **49**, 3927–3936 (2010)
 57. S.V. Eliseeva, J.-C.G. Bünzli, Lanthanide luminescence for functional materials and bio-sciences. *Chem. Soc. Rev.* **39**, 189–227 (2010)
 58. L.N. Puntus, K.A. Lyssenko, I.S. Pekareva, J.-C.G. Bünzli, Intermolecular interactions as actors in energy-transfer processes in lanthanide complexes with 2,2'-bipyridine. *J. Phys. Chem. B* **113**, 9265–9277 (2009)
 59. J.-C.G. Bünzli, Review: Lanthanide coordination chemistry: from old concepts to coordination polymers. *J. Coord. Chem.* **67**, 3706–3733 (2014)
 60. J.-C. G. Bünzli, Lanthanide light for biology and medical diagnosis. *J. Lumin.* **170**(Part 3): 866 – 878 (2016) (Light, Energy and Life)
 61. J.-C. G. Bünzli, On the design of highly luminescent lanthanide complexes,” *Coordination Chemistry Reviews* **293–294**, 19 – 47 (2015) (41st International Conference on Coordination Chemistry, Singapore, July 2014)
 62. T. Forster, Delocalized excitation and excitation transfer, in *Modern Quantum Chemistry. Istanbul Lectures. Part III: Action of Light and Organic Crystals*, vol. 3 (Academic Press, New York, 1965), pp. 93–137
 63. T. Forster, 10th spiers memorial lecture. transfer mechanisms of electronic excitation. *Discuss. Faraday Soc.* **27**, 7–17 (1959)
 64. T. Forster, Zwischenmolekulare energiewanderung und fluoreszenz. *Ann. Phys.* **2**, 55 (1948)
 65. T. Forster, Energiewanderung und fluoreszenz. *Naturwissenschaften* **33**, 166–175 (1946)
 66. D. Dexter, J.H. Schulman, Theory of concentration quenching in inorganic phosphors. *J. Chem. Phys.* **22**, 1063 (1954)
 67. D. Dexter, W. Heller, Capture and collision processes for excitons in alkali halides. *Phys. Rev.* **84**, 377 (1951)
 68. W.R. Heller, A. Marcus, A note on the propagation of excitation in an idealized crystal. *Phys. Rev.* **84**, 809 (1951)
 69. S. Sato, M. Wada, Relations between intramolecular energy transfer efficiencies and triplet state energies in rare earth β -diketone chelates. *Bull. Chem. Soc. Jpn.* **43**, 1955–1962 (1970)
 70. J. Tang, P. Zhang, *Lanthanide Single Molecule Magnets* (Springer, Berlin, 2015)
 71. G.-J. Chen, C.-Y. Gao, J.-L. Tian, J. Tang, W. Gu, X. Liu, S.-P. Yan, D.-Z. Liao, P. Cheng, Coordination-perturbed single-molecule magnet behaviour of mononuclear dysprosium complexes. *Dalton Trans.* **40**, 5579 (2011)
 72. W. C. Choy, D. Tao, Rare-earth complexes and its applications in organic light emitting diodes, in *Solid State Chemistry Research Trends*, chap. 2, ed. by R.W. Buckley (Nova Science Publishers Inc, New York, 2007)
 73. H. Brito, O. Malta, M. Felinto, E. Teotonio, Luminescence phenomena involving metal enolates, in *The Chemistry of Metal*

- Enolates*, chap. 3, ed. by J. Zabicky (Wiley, New York, 2009) (The Chemistry of Functional Groups)
74. C.Y. Peng, H.J. Zhang, J.B. Yu, Q.G. Meng, L.S. Fu, H.R. Li, L.N. Sun, X.M. Guo, Synthesis, characterization, and luminescence properties of the ternary europium complex covalently bonded to mesoporous sba-15. *J. Phys. Chem. B* **109**, 15278 (2005)
 75. H. Xin, Y. Ebina, R. Ma, K. Takada, T. Sasaki, *J. Phys. Chem. B* **110**, 9863 (2006)
 76. V.R. Kharabe, A.H. Oza, S.J. Dhoble, Synthesis, PL characterizations and concentration quenching effect in Dy^{3+} and Eu^{3+} activated LiCaBO_3 phosphor. *Luminescence* **30**, 432–438 (2015)
 77. J.J. Joos, K.W. Meert, A.B. Parmentier, D. Poelman, P.F. Smet, Thermal quenching and luminescence lifetime of saturated green $\text{Sr}_{1-x}\text{Eu}_x\text{Ga}_2\text{S}_4$ phosphors. *Opt. Mater.* **34**, 1902–1907 (2012)
 78. O. Meza, E. Villabona-Leal, L. Diaz-Torres, H. Desirena, J. Rodriguez-Lopez, E. Perez, Luminescence concentration quenching mechanism in $\text{Gd}_2\text{O}_3:\text{Eu}^{3+}$. *J. Phys. Chem. A* **118**, 1390–1396 (2014)
 79. G. Ju, Y. Hu, L. Chen, X. Wang, Z. Mu, Concentration quenching of persistent luminescence. *Phys. B Condens. Matter* **415**, 1–4 (2013)
 80. J.R. Lakowicz, *Principles of Fluorescence Spectroscopy* (Springer, New York, 2006)
 81. J. Bierwagen, S. Yoon, N. Gartmann, B. Walfort, H. Hagemann, Thermal and concentration dependent energy transfer of Eu^{2+} in SrAl_2O_4 . *Opt. Mater. Express* **6**, 793–803 (2016)
 82. J. Ueda, P. Dorenbos, A.J. Bos, A. Meijerink, S. Tanabe, Insight into the thermal quenching mechanism for $\text{Y}_3\text{Al}_5\text{O}_{12}:\text{Ce}^{3+}$ through thermoluminescence excitation spectroscopy. *J. Phys. Chem. C* **119**, 25003–25008 (2015)
 83. R. Yu, D.S. Shin, K. Jang, Y. Guo, H.M. Noh, B.K. Moon, B.C. Choi, J.H. Jeong, S.S. Yi, Luminescence and thermal-quenching properties of Dy^{3+} -doped Ba_2CaWO_6 phosphors. *Spectrochim. Acta A* **125**, 458–462 (2014)
 84. J. Feist, A. Heyes, The characterization of $\text{Y}_2\text{O}_3:\text{Sm}$ powder as a thermographic phosphor for high temperature applications. *Meas. Sci. Technol.* **11**, 942–947 (2000)
 85. Y. Bi, C. Chen, Y.-F. Zhao, Y.-Q. Zhang, S.-D. Jiang, B.-W. Wang, J.-B. Han, J.-L. Sun, Z.-Q. Bian, Z.-M. Wang, S. Gao, Thermostability and photoluminescence of $\text{Dy}(\text{III})$ single-molecule magnets under a magnetic field. *Chem. Sci.* **7**, 5020 (2016)
 86. C.D.S. Brites, P.P. Lima, N.J.O. Silva, A. Millan, V.S. Amaral, F. Palacio, L.D. Carlos, Thermometry at the nanoscale. *Nanoscale* **4**, 4799 (2012)
 87. B.R. Anderson, S.-T. Hung, M.G. Kuzyk, Wavelength dependence of reversible photodegradation of disperse orange 11 dyed pmma thin films. *J. Opt. Soc. Am. B* **32**, 1043–1049 (2015)