

# **YBa<sub>2</sub>Cu<sub>3</sub>O<sub>7-δ</sub> films prepared by pulsed laser deposition in O<sub>2</sub>/Ar mixture atmosphere**

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#### **Abstract**

The YBa<sub>2</sub>Cu<sub>3</sub>O<sub>7- $\delta$ </sub>(YBCO) superconducting films were fabricated epitaxially on STO(00 l) substrates by pulsed laser deposition method (PLD) in different  $O_2/Ar$  mixture atmospheres. The surface morphology of the YBCO film is strongly dependent on the  $O_2/Ar$  ratio of the ambient atmosphere. The surface particle density decreases and the particle size increases with the  $O_2/Ar$  ratio decreasing. The lattice parameters and the superconducting transition temperature ( $T_{c0}$ ) are almost independent on the  $O_2$ /Ar ratio and are about the same as 1.168 nm and 90 K, respectively, which can be ascribed to the annealing process in 1 atm  $O_2$  for all the YBCO samples. Based on the above results, in  $O_2/Ar$  ratio of 25:75 atmosphere, YBCO:LaAlO<sub>3</sub> superconducting composite flms with good performance were prepared by co-deposition of PLD with magnetron sputtering method. The experimental results provide a way to control the surface morphology of the YBCO flms deposited by PLD method and provide valuable reference for the preparation of YBCO composite thin flms by co-deposition of PLD with magnetron sputtering method in  $Ar/O<sub>2</sub>$  mixture atmosphere.

**Keywords** PLD · O<sub>2</sub>/Ar mixture atmosphere · YBCO film · Surface morphology · Superconducting transition temperature

## **1 Introduction**

 $YBa<sub>2</sub>Cu<sub>3</sub>O<sub>7-δ</sub>$  (YBCO) superconducting films have been the most intensively investigated due to not only the basic physics but also its application background [[1–](#page-3-0)[4](#page-4-0)]. YBCObased perovskite structure multifunctional heterostructures, such as ferroelectric/superconductor, ferromagnetic/ superconductor, colossal magnetoresistance/superconductor, have attracted widespread attention for the potential novel devices [[5](#page-4-1)[–10\]](#page-4-2). Diferent methods have been used to prepare YBCO thin flms, such as metal–organic chemical vapor deposition (MOCVD) [\[1\]](#page-3-0), metal organic deposition

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 $(MOD)$  [\[11](#page-4-3)], pulsed laser deposition (PLD) [[4–](#page-4-0)[10\]](#page-4-2), chemical solution deposition (CSD) [\[12\]](#page-4-4). Among various techniques, PLD technique has been regarded as one of the most promising methods in preparing multi-component flms with precise control of stoichiometry, reproducibility, and simplicity. However, boulders or droplets composed of spherical particles  $(0.1 - 1 \mu m)$  diameter) and the surface outgrowths composed of misoriented and/or offstoichiometric crystallites are often observed on the surface of YBCO flms [\[13\]](#page-4-5). The surface particles seriously impacts the application of the superconducting flms on multilayered structures, in particular in electronic devices of diodes and transistors  $[14–16]$  $[14–16]$  $[14–16]$  $[14–16]$  $[14–16]$ . Various attempts have been made to solve this problem, such as laser deposition parameters optimization [[17](#page-4-8)[–19\]](#page-4-9), ablation plume processing  $[20, 21]$  $[20, 21]$  $[20, 21]$  $[20, 21]$  $[20, 21]$ , off-axis laser deposition  $[22, 23]$  $[22, 23]$  $[22, 23]$ , and the target preparation and modifcation [\[24,](#page-4-14) [25\]](#page-4-15). However, it is found that the above methods are not very efficient, or even introduce some drawbacks. For example, the region of the optimized deposition parameters is too narrow to fnd, and small fuctuations may infuence the surface quality of the film  $[17]$  $[17]$ . The off-axis method in which the substrate has less chance to receive particles may greatly reduce the deposition rate of the flm [[22](#page-4-12)]. And the target

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modifcation method increases the complexity and technical difficulty of the preparation process  $[24]$ .

In our experiments, Ar was introduced during fabricating YBCO film by PLD method and the impacts of  $O_2/Ar$ ratio on the crystalline quality, surface morphology and transport properties of the YBCO flms are systematically studied. The effect of in-situ annealing procedure in 1 atm  $O<sub>2</sub>$  on the lattice parameter and transport properties of the YBCO film is discussed. In  $O_2/Ar$  ratio of 25:75 atmosphere, YBCO:LaAlO<sub>3</sub> superconducting composite films with good performance were prepared successfully by co-deposition of PLD with magnetron sputtering method.

### **2 Experimental details**

The YBCO films with thickness of  $\sim 100$  nm were deposited on 5 mm  $\times$  5 mm  $\times$  0.5 mm (001)-oriented  $SrTiO<sub>3</sub>(STO)$  substrates by PLD method with a KrF UV excimer laser (248 nm, Coherent) at the base pressure of  $1.0 \times 10^{-4}$  Pa. The distance between the target and substrate was 45 mm. The power density and frequency of the laser were 2.0 J/cm<sup>2</sup> and 2 Hz, respectively. The  $O_2/$ Ar mixture atmosphere with the mass fow rate ratio of 100:0, 75:25, 25:75 was set by adjusting the individual gas flows regulated by two calibrated mass flow controllers. The deposition pressure of the  $O_2/Ar$  gas mixture was maintained at 15 Pa monitored with a resistance vacuum gauge through adjusting the gate valve. The substrate temperature was kept at 795 ℃ during the deposition process. After the deposition process, oxygen was input into the chamber to reach  $1.0 \times 10^5$  Pa before the films were cooled from 795 to 520 ℃ at the rate of 10 ℃/min. For the full oxidation of the YBCO flms, the temperature was maintained at 520 ℃ for 30 min. The samples were then cooled down to room temperature at the rate of 5 ℃/min. The YBCO flms with the same thickness of ~ 100 nm were gained by varying the deposition time from 8 to 12 min when deposited in mixture atmospheres from  $O_2/Ar$  ratio of 100:0 to 25:75. To express concisely, hereafter we use S1, S2 and S3 to denote YBCO flms deposited in diferent  $O_2$ /Ar ratio of 100:0, 75:25, 25:75, respectively. The surface morphology of the flm was studied by scanning electron micrography (SEM) (FEI Nova NanoSEM450). The phase and crystallinity of the YBCO flms were characterized by X-ray difraction (XRD) technique using a Bruker D-8 X-ray difractometer. The superconducting transport properties of  $YBa<sub>2</sub>Cu<sub>3</sub>O<sub>7-δ</sub>$  films were characterized using a physical property measurement system (Quantum design, PPMS-9 T), in which high-purity iridium metal was used as the contact electrodes on YBCO flms.

#### **3 Results and discussions**

 $XRD$  patterns of YBCO films deposited on  $SrTiO<sub>3</sub>$  substrates in  $O_2/Ar$  mixture atmospheres are shown in Fig. [1.](#page-1-0) From which, it can be seen that besides the substrate peaks, only (00 *l*) difraction peaks of the YBCO flm can be seen for all samples, indicating all the flms are highly *c*-axis oriented. Figure [1](#page-1-0)d presents a typical phi-scan pattern of (014) Bragg peak of the YBCO flm, indicative of the epitaxial growth of the YBCO flm on STO (001) substrates. Within the measurement error of our X-ray difraction data, the calculated *c*-axis lattice parameters are almost the same as 1.168 nm for  $S1 \sim S3$  samples [\[26\]](#page-4-16). It is reported that the in-situ annealing process in 1 atm  $O<sub>2</sub>$  is crucial for the oxidation and the oxygen content of  $S1 \sim S3$  samples in spite of the deposition ambient gas. During the oxygen annealing process, surface chemical adsorption and difusion of oxygen into the interior of the YBCO flms occur [\[27](#page-4-17)[–29](#page-4-18)].

The surface particles of  $S1 \sim S3$  samples were identified and counted from scanning electron micrographs (SEM), as shown in Fig. [2a](#page-2-0)–c. It can be seen that there is a big difference in surface morphology (particle numbers and sizes) for  $S1 - S3$ . As for S1, the surface particles are distributed densely and uniformly, much smaller particles with the density of  $2.0 \times 10^8$ /cm<sup>2</sup> and mean size of 0.25  $\mu$ m are presented on the flm surface (shown in Fig. [2a](#page-2-0)), which is the same magnitude as the typical reported densities and sizes [\[30](#page-4-19)]. However, in argon diluted oxygen atmosphere (for S2 and S3), less and larger particles are exhibited on the flm surface



<span id="page-1-0"></span>**Fig. 1** XRD patterns of YBCO samples deposited in mixture atmosphere of different O<sub>2</sub>/Ar ratio **a** 100:0 (S1), **b** 75:25 (S2), **c** 25:75 (S3), respectively. **d** Typical phi-scan pattern of (014) Bragg peak of the YBCO flm



<span id="page-2-0"></span>**Fig. 2** Surface morphology identifed by scanning electron micrographs (SEM) of YBCO samples deposited in mixture atmosphere of diferent O2/Ar ratio **a** 100:0 (S1), **b** 75:25 (S2), **c** 25:75 (S3), respectively

(shown in Fig. [2](#page-2-0)b, c), accompanied with irregular particles, such as needle shaped grains, presented on the flm surfaces [\[31](#page-4-20)]. The EDX spectra of the surface particles and the matrix around the particles have been measured. The results show that the surface particles are *Y*- and Ba-defcient Cu–O particles. It should be mentioned that the shape and color of the visible plume during the deposition is signifcantly diferent for S1 ~ S3 samples. For S1, the plume is more directional and slim with red fame-like of blue color in the center and red color near the boundary. As argon is introduced in the ambient, the plume becomes more difuse, and the color of plume periphery changes from red to bluish. In our previous study, it is reported that the efective energy transfer and the oscillating stabilization time (OST) produced by the interaction between an ambient atom and a Si atom decide the size distribution of Si nanoparticles prepared in varied mixed atomic ratio (MAR) of the ambient gases by PLD method [[32,](#page-4-21) [33\]](#page-4-22). The above results indicate that the interaction between the plume and ambient gas may play an important role. It is reported that the typically natural lifetimes of electronic transitions in the plasma plume are of the order of  $10^{-8}$  s [\[34](#page-4-23)], and the typical velocities of the ions within the plume are around  $10^6$  cm/s [[35\]](#page-4-24). Thus, large spatial dimensions of the visible plume can be explained by the interaction between the plume and ambient gas, as well as the absorption of photons from both the laser and the plasma radiation [[36](#page-4-25)]. And the interaction of the ablation plume with ambient background gases is reported not only a complex hydrodynamic phenomenon but also a chemical process [\[37\]](#page-4-26). Here in the YBCO deposition process,  $O_2$  and Ar can act as reactive and non-reactive gas, respectively. When the oxygen partial pressure becomes lower with the introduction of the inert Ar gas, the elastic scattering of the plume particles by Ar might be enhanced, so the plume seems more difuse rather than directional and slim [[38\]](#page-4-27). Meanwhile, the chemically reactive collisions become weak and the reaction rates become slow for the number of  $O_2$  molecules decreases with the introduction of Ar. Therefore the nucleation and growth of the surface clusters become slower, and also do the flm deposition rates. The particle distribution on the

film surfaces of  $S1 - S3$ , as well as the film deposition rate obtained based on the flm thickness measured by the step proflometer (Bruker, Dektak XT), confrm the above analysis when Ar was introduced to the deposition chamber.

The effect of Ar introduction and the  $O_2/Ar$  ratio on the electrical transport properties of the YBCO flms were studied. The in-plane resistance (*R*) versus temperature (*T*) relation of the YBCO thin flms were presented in Fig. [3](#page-2-1) which measured by a standard four-probe electrical measurements. The results show that the resistance value of the YBCO flm decreases with the  $O_2/Ar$  ratio decreasing, which may be related to the surface particles distribution. It is reported that the surface particles are electrically non-conductive and seriously impacts the application of the superconducting flms, so when the surface particles decrease with the  $O_2/Ar$  ratio decreasing (shown in Fig. [2\)](#page-2-0), the resistance value of the YBCO flm decreases. It can also be seen that the superconducting transition temperature  $(T<sub>c0</sub>)$  is almost the same  $as \sim 90$  K, which is almost independent of the deposition ambient in spite of the signifcant diferences of the surface



<span id="page-2-1"></span>**Fig. 3** Dependence of resistance  $(R)$  on temperature  $(T)$  for  $S1 \sim S3$ with in-situ annealing in 1 atm  $O_2$ 

morphology for  $S1 \sim S3$  samples (shown in Fig. [2\)](#page-2-0). It can be concluded that the oxidation and the electrical transport properties were mainly determined by the in-situ annealing process in  $1.0 \times 10^5$  Pa O<sub>2</sub> atmosphere despite the difference of the deposition ambient gas [[27–](#page-4-17)[29\]](#page-4-18).

Based on the above analysis, surface morphology of YBCO thin film can be modified by adjusting the  $O_2/Ar$ ratio of the ambient atmosphere and good superconducting performance can be gained through in-situ annealing process in  $1.0 \times 10^5$  Pa O<sub>2</sub>. Coincidentally, the O<sub>2</sub>/Ar ratio of 25:75 is commonly used in most of the magnetron sputtering process. It is well known that doping of non-superconducting materials into high temperature superconducting materials can efectively improve the transport properties by introducing artifcial pinning centers into the superconductors. Here, the LaAlO<sub>3</sub>: YBa<sub>2</sub>Cu<sub>3</sub>O<sub>7-δ</sub> superconducting films with thickness of ~ 100 nm were successfully prepared in  $O_2/Ar$  ratio of 25:75 atmosphere by co-deposition of PLD with magnetron sputtering method. The YBCO flm was prepared by pulsed laser deposition method from a single-phase  $YBa<sub>2</sub>Cu<sub>3</sub>O<sub>7-δ</sub>$ target and the LaAlO<sub>3</sub> film was deposited from a LaAlO<sub>3</sub> target by radio frequency sputtering with the corresponding magnetron power of 25 W. The superconducting transition temperature  $(T_c)$  and the critical current density  $(J_c)$ in magnetic field of  $YBa<sub>2</sub>Cu<sub>3</sub>O<sub>7-δ</sub>: LaAlO<sub>3</sub> superconducting$ materials have been investigated. As can be seen in Figs. [4](#page-3-1) and [5,](#page-3-2) the doping of LaAlO<sub>3</sub> have little effect on the  $T_c$  of YBCO flms and can efectively improve the critical current densities of the YBCO flms in magnetic feld higher than  $1$  T at 77 K. From the inset of Fig.  $5$ , it can be seen that the  $J_c$  of YBa<sub>2</sub>Cu<sub>3</sub>O<sub>7-δ</sub>:LaAlO<sub>3</sub> superconducting film decays more slowly than that of YBCO flm in magnetic feld higher than 1 T, which implying the co-deposition of PLD with magnetron sputtering method is a valuable approach



<span id="page-3-1"></span>**Fig. 4** Dependence of resistance (*R*) on temperature (*T*) for YBCO:LaAlO<sub>3</sub> composite film compared with YBCO film



<span id="page-3-2"></span>**Fig. 5**  $J_c/J_c(0)$ -H for YBCO:LaAlO<sub>3</sub> composite film compared with YBCO film at 77 K. Inset shows  $log J_c/J_c(0)$  vs.  $log H$  plot for magnetic field dependence of  $J_c \propto H^{-\alpha}$ 

for the preparation of YBCO composite thin films in  $Ar/O<sub>2</sub>$ mixture atmosphere.

## **4 Conclusions**

The introduction of Ar and the  $O_2/Ar$  ratio on the crystalline quality, surface morphology and electrical transport properties of the YBCO flms deposited by PLD method have been studied intensively. The introduction of Ar has signifcant efect on the visible laser plume and the surface morphology of the flms, but have little efect on the superconducting transition temperature  $T_c$ , which can be attributed to the in-situ annealing in  $1.0 \times 10^5$  Pa O<sub>2</sub>. The experimental results provide a way for efectively controlling the surface morphology of the YBCO flms deposited by PLD method and provide valuable reference for the preparation of YBCO composite thin flms by co-deposition of PLD with magnetron sputtering method in  $Ar/O<sub>2</sub>$  mixture atmosphere.

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