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Flame‑retardant synergistic efect of hydroquinone bis(diphenyl phosphate) and tris(2‑hydroxyethyl) isocyanurate on epoxy resin

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Abstract

In this study, epoxy resin (EP) composites were prepared by the incorporation of hydroquinone bis(diphenyl phosphate) (HDP) and tris(2-hydroxyethyl) isocyanurate (THEIC), and the synergistic efects of HDP and THEIC on the fame retardancy of EP were investigated. The fame retardancy of EP composites was systematically evaluated by measuring limiting oxygen index (LOI), UL-94 vertical burning tests and a cone calorimeter. The results indicate that the synergistic efect of the addition of 15% HDP and 15% THEIC in the composite EP-4 is the best. Compared with neat EP, the LOI of EP-4 has increased from 21.7 to 27.5%, and UL-94 reached V-0. The heat release rate (HRR), total heat release (THR), smoke production rate (SPR) and total smoke production (TSP) of EP-4 in the cone calorimeter test decreased by 81.14, 41.42, 72.57 and 53.61%, respectively. The release rates of $CO₂$ and CO were also signifcantly reduced, indicating a good synergistic efect between HDP and THEIC. Furthermore, the synergistic fame-retardant mechanism between HDP and THEIC has been thoroughly studied by scanning electron microscopy (SEM) and thermogravimetric analysis/infrared analysis (TG-IR). The results show that HDP/ THEIC can efectively strengthen the carbon layer structure resulted from the composite after combustion and increase the fame retardancy of the condensed phase. Meanwhile, PO, PO₂ radicals and NH_3 produced by the degradation of HDP and THEIC are also beneficial to improve the flame retardancy of the gas phase.

Keywords Epoxy resin · Flame retardant · Synergistic efect · Thermogravimetric analysis/infrared spectrometry · Flame-retardant mechanism

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Introduction

As an important thermosetting material, epoxy resin (EP) is widely used in adhesives, coatings, microelectronics manufacturing, aerospace and other felds due to its excellent performance, such as superior adhesion, electrical insulation properties, chemical resistance, mechanical strength and processability. Although EP shows such attractive performance, its high fammability still limits its application in many fields $[1-3]$ $[1-3]$ $[1-3]$. Therefore, how to effectively improve the flame retardancy of EP has been an important research direction in the polymer science and engineering community.

In the past, halogenated fame retardants were often used to improve the fameretardant performance of EP. However, these halogenated fame retardants will release a large number of toxic gases and corrosive substances such as carcinogenic brominated furans, dioxins and hydrogen bromide during combustion [[4,](#page-18-2) [5](#page-18-3)]. With the increasing attention to environmental problems, the application of halogenated flame retardants has been strictly restricted [\[6,](#page-18-4) [7\]](#page-18-5). Nowadays, highefficiency halogen-free flame retardants have become the main research direction. In the halogenated-free fame-retardant system, organophosphorus ester flame retardants have the advantages of low toxicity, high efficiency and good compatibility with materials. Thus, they are being considered to replace halogenated fame retardants. In previous studies, there are reported precedents of organophosphorus fame retardants such as bisphenol-A bis(diphenyl phosphate) and triphenyl phosphate that have been applied in epoxy resin [\[8](#page-18-6)–[10\]](#page-18-7). However, these fame retardants have relatively low melting points and strong mobility and cannot achieve good efects when used alone. They often need to be applied in conjunction with certain synergists to achieve satisfactory fame-retardant performance efects. As a solid organophosphorus ester fame retardant, hydroquinone bis(diphenyl phosphate) (HDP) has a higher melting point and migration resistance, and as far as the authors know, it has not been used as a fame retardant in epoxy resin, so this study is aimed to investigate it as the epoxy resin fame retardant [[11](#page-18-8)].

In recent years, some triazine compounds have been synthesized and used as charring agents. It has been found that triazine compounds can not only improve the carbon layer structure and enhance the condensed phase fame-retardant effect but also the $NH₃$ produced by their decomposition can enhance the gas phase fame-retardant efect [\[12,](#page-18-9) [13](#page-18-10)]. Li combined ammonium polyphosphate, melamine, and triazine charring agent to modify the fame retardant of epoxy resin. The combination of triazine charring agent and ammonium polyphosphate can improve the ability of carbon formation during the combustion of EP [\[14\]](#page-18-11). Lai synthesized a new triazine-based polymer and combined it with melamine pyrophosphate for fame retardant polypropylene. The results showed that the triazine ring produced by the decomposition of triazine-based polymer can be cross-linked to form a graphite-like char, while the decomposition will also release NH_3 to strengthen the effect of the gas phase flame retardancy [[8\]](#page-18-6). Tris(2hydroxyethyl) isocyanurate (THEIC) is a derivative of triazine compounds, which

are commonly used as polyvinyl chloride stabilizers, coating intermediates, fuels, and pharmaceuticals. [[15](#page-18-12), [16\]](#page-18-13) Previously, Li applied ammonium polyphosphate and THEIC as the fame retardants of PP and found that they have a good synergistic efect, which leads to improvement of the limiting oxygen index LOI and UL-94 level of polypropylene and the reduction of the heat release and smoke production during the combustion of polypropylene [[17](#page-18-14)]. It indicates that THEIC has a good efect as a charring agent.

Previously, there have been no attempts to apply HDP and THEIC simultaneously as epoxy fame retardants. Therefore, this study attempted to compound HDP and THEIC into epoxy resin and investigate the fame-retardant synergistic efects of both. Furthermore, the synergistic fame-retardant mechanism between HDP and THEIC has been thoroughly investigated by thermogravimetric analysis (TG), scanning electron microscopy (SEM), thermogravimetric analysis/infrared spectrometry (TG-IR) and Raman spectrometer, etc.

Experimental

Materials

Phosphorus oxychloride $(POC1₃)$, hydroquinone, phenol, anhydrous aluminum trichloride, oxalic acid, sodium hydroxide and triethylenetetramine (TETA) were all of the analytical grade and were acquired from Chengdu Jinshan Chemical Reagent Co. Ltd. (China). EP (E-44) was provided by Nantong Xing Chen Synthetic Material Co. Ltd. (China). THEIC was purchased from Bide Pharmatech Co. Ltd. (China).

Preparation of HDP

11.00 g hydroquinone and 76.65 g phosphorus oxychloride were reacted under the catalysis of anhydrous aluminum trichloride at 100 $^{\circ}$ C for 0.5 h under nitrogen protection, and the excess phosphorus oxychloride was distilled off under reduced pressure to obtain hydroquinone bisphosphorochloride. Hydroquinone bisphosphorochloride reacted with 37.60 g phenol at 120 $^{\circ}$ C for 4 h. After the reaction, the product was washed with oxalic acid solution and sodium hydroxide solution in turn and finally filtered and dried at 80 $^{\circ}$ C to obtain the product. [[18\]](#page-18-15)

Sample preparation

The EP composites were prepared by a thermal curing process. Firstly, THEIC was added into EP and thoroughly stirred at 80 °C, and then HDP was melted at 100 °C and added into the mixture quickly, stirred well and cooled to 40 °C. Afterward, a certain amount of curing agent TETA was added, and the mixture was left to bubble at room temperature for 10 min initially. Finally, the mixture was poured into a preheated mold at 80 °C and cured at room temperature for 2 h and then cured at

Samples	EP $(wt\%)$	HDP $(wt\%)$	THEIC $(wt\%)$	HDP: THEIC	TETA $(wt\%)$	LOI $(\%)$	UL-94	
EP	89.3	$\overline{0}$	$\mathbf{0}$		10.7	21.7	Fail	
$EP-1$	63.5	30	$\mathbf{0}$		6.5	25.2	Fail	
$EP-2$	63.5	25	5	5:1	6.5	24.2	Fail	
$EP-3$	63.5	20	10	2:1	6.5	26.3	$V-1$	
$EP-4$	63.5	15	15	1:1	6.5	27.5	$V-0$	
$EP-5$	63.5	10	20	1:2	6.5	26.2	$V-1$	
$EP-6$	63.5	5	25	1:5	6.5	25.5	Fail	
$EP-7$	63.5	$\overline{0}$	30		6.5	21.4	Fail	

Table 1 Composition and fame retardancy of the samples

80 °C for 4 h. After cooling down, the molds were decoupled to obtain the samples. Table [1](#page-3-0) shows the composition samples.

Characterization

The sample size for measuring the LOI value was $127 \times 6.5 \times 3$ mm³, which was measured on an XZT-100A oxygen index instrument (Kecheng, China). The sample of UL-94 vertical burning tests was performed in a CFZ-3 instrument (Jiangning Analysis Instrument Company, China) with a sample size of $127 \times 12.8 \times 3$ mm³. LOI test and UL-94 vertical burning tests were carried out according to ASTM D2863-97. The fammability of the sample was evaluated by using a cone calorimeter (FTT iConeR). The sample size was $100 \times 100 \times 5$ mm³ for the cone calorimeter test, which was carried out according to the test standard of ISO 5560 with a heat flux of 35 KW/m²a.

The thermal stability of the samples was evaluated by a STA449F3 thermal analyzer (NETZSCH, Germany) under a nitrogen atmosphere from 30 to 700 °C using a heating rate of 10 °C/min.

A Vage3 emission scanning electron microscopy (TESCAN, Czech Republic) was used to observe the microstructure of residual char at 20 kV and 500 magnifcation. The cross-sectional microstructure of the composite was observed at 20 kV and 200 magnifcation. An IE250 energy-dispersive spectroscopy (Oxford, England) was used to analyze the element content of the sample.

The Raman spectra of the residual chars were measured by a Thermo Scientifc DXRxi Raman spectrometer (Thermo Fisher, America) under room temperature from 1000 to 2000 cm−1 with a 450 nm argon laser line excitation source.

A TG-IR instrument comprised the STA449F3 thermal analyzer (NETZSCH, Germany) and a Fourier spectrometer (PerkinElmer, America). The test conditions were in the wavenumber range from 650 to 4000 cm⁻¹ and raised from 30 to 800 °C under an air condition.

Fourier-transformed infrared spectra (FTIR) of the residual chars were obtained by using a Fourier infrared spectrometer (PerkinElmer, America) in the wavenumber range from 400 to 4000 cm^{-1} .

Tensile and fexural tests of the sample were carried out on 119 INSTRON universal testing machines (Instron, America) with two different sizes, $180 \times 20 \times 4$ mm³ and $80 \times 15 \times 4$ mm³, respectively. The tensile and flexural tests were performed according to GB/T 2567–2008 (China).

Results and discussion

Flame retardancy of EP composites

Analysis of LOI and UL‑94 results

The efect of diferent ratios of HDP/THEIC on the fame retardancy of EP is shown in Table [1](#page-3-0) and Fig. [1.](#page-4-0) The results show that the LOI value of neat EP is 21.7% and no rating in the UL-94 test. The LOI value of EP-7 with 30 wt% THEIC only is 21.4%. The LOI value of EP-1 with 30 wt% HDP reached 25.2%, but cannot improve the UL-94 level. The LOI value of the remaining samples shows a pattern of increasing and then decreasing as the amount of THEIC increases, and the same for the UL-94 test. The LOI value of the EP-4 with 15 wt% HDP and 15 wt% THEIC reached up to 27.5% along with a V-0 rating in the UL-94 test, increased by 26.7% compared to neat EP. It can be seen that there is a good synergy between HDP and THEIC, and the efect of the synergy was related to the ratio of addition. There is a good synergistic efect between HDP and THEIC. The possible reason is that HDP can well catalyze the dehydration of THEIC to carbon, so as to increase the residual char amount of the material after combustion, and THEIC can also decompose NH3, thereby strengthening the gas phase fame-retardant efect, and further research is

Fig. 1 The efect of HDP/THEIC weight ratios on the limiting oxygen index (LOI) of composite epoxy resin samples

needed. In order to better explore the synergistic efect between HDP and THEIC, more in-depth research is necessary. EP-4 has the best fame-retardant efect and can best refect the synergistic efect between HDP and THEIC. Therefore, the subsequent analysis will focus on comparing the diferences between EP-4 and EP.

Analysis of cone calorimeter test results

The cone calorimeter test has a good correlation with real fres and is considered the best means to characterize the real burning situation of materials. It can obtain some parameters that can be related to the burning behavior of materials in a fre, parameters include heat release rate (HRR), total heat release (THR), smoke production rate (SPR) and total smoke production (TSP), CO release rate and $CO₂$ release rate [[19\]](#page-19-0). According to UL-94 and LOI test results, EP-4 had the best fame-retardant efect and could better refect the synergistic efect between HDP and THEIC. Therefore, EP-4 and EP were selected for comparison in the cone calorimeter test. The data obtained from the tests are shown in Fig. [2](#page-6-0) and Table [2.](#page-6-1)

It can be seen from Fig. [2](#page-6-0)a, b that the HRR increases rapidly with a peak HRR (pHRR) value of 1553.16 kW/m², and the THR reaches 107.66 MJ/m² once the EP is ignited. These results show that the combustion of EP rapidly exerts heat and releases a large amount of heat at the same time. In contrast, the pHRR of the EP-4 sample is obviously decreased, being only 292.92 kW/m^2 and 81.14% lower than the pHRR of EP, and the THR is only 63.07 $MJ/m²$ and 41.42% lower than the THR of EP. The simultaneous introduction of HDP and THEIC can efectively reduce the heat release rate and total heat release of the composite EP samples.

In the case of a fre, intense smoke is often the primary cause of fatalities, so it is especially important to reduce the smoke amount generated by combustion. Figure [2c](#page-6-0), d shows the graphs of TSP and SPR for the EP and EP-4 samples. It can be seen that the combustion of the EP sample exhibited high peak SPR (pSPR) and TSP, indicating that a lot of fue gas was produced. The pSPR of the EP-4 sample is only 0.0961 m²/s and the TSP is 16.88 m², with a decrease of 72.57 and 53.61%, respectively. In addition, the release of $CO₂$ and CO in the smoke is also suppressed (Fig. [2e](#page-6-0), f), the peak release rate of $CO₂$ and CO of the EP samples is 1.067 and 0.0855 g/s, respectively. While the release rate of $CO₂$ and CO of the EP-4 sample is signifcantly lower, the peak release rate is 0.219 and 0.0314 g/s, respectively. The SPR, TSP, and release rate of $CO₂$ and CO during the combustion of the composites are signifcantly reduced, indicating that the simultaneous addition of HDP and THEIC can effectively reduce the smoke generation during combustion.

It is worth noting that the HRR, SPR, CO release rate, and $CO₂$ release rate profles of both the EP and EP-4 samples show two peaks, and the advent time of the second peak for the EP sample is always earlier than that of the EP-4 sample. The reason for this observed phenomenon might be that the frst peak belongs to the thermal decomposition of EP itself, and the second peak represents the further decomposition of the carbon layer produced by combustion. In Fig. [3](#page-7-0), by comparing the residual carbon of the two samples, it can be found that only a small amount of residual carbon remains after the combustion of the EP sample, while the combustion of the EP-4 sample forms a huge layer of expanded carbon materials. This

Fig. 2 HRR (a), THR (b), SPR (c), TSP (d), CO_2 release rate (e) and CO release rate (f) of curves

		(s)	Samples TTI (s) t-pHRR pHRR(kW/ THR(MJ/ m^2)	m^2	pSPR (m^2/s)	$TSP(m^2)$ FPI		FGI (sm^2/kW) $(kW/m^2/s)$
EP	51	135	1553.16	107.66	0.35	36.39	0.033	11.50
$EP-4$	42	70	292.92	63.07	0.096	16.88	0.14	4.18

Table 2 Combustion parameters obtained from cone calorimeter

indicates that the residual carbon formed by the combustion of EP is completely decomposed, but the combustion of EP-4 results in a fufy carbon layer that is well insulated from heat and can thus slow down the decomposition. So, the second peak

Fig. 3 Digital photographs of the char residues of EP and EP-4 after the cone calorimeter test

reached comes later. In addition, the expanded carbon layer efectively blocks the heat and material exchange between the matrix and the burning area. This is the reason for the decrease in HRR, THR, SPR, TSP, and $CO₂$ and CO release rate of the composite. This further indicates that HDP and THEIC have a good synergistic effect.

In order to give a comprehensive evaluation of the fre safety efects of the composites, the fre performance index (FPI) and fre growth index (FGI) were calculated for both; the results are shown in Table [2](#page-6-1). FPI is defned as the time to ignition (TTI) divided by pHRR. A higher FPI suggests that there is more time for rescue before the violent combustion. FGI is expressed as the pHRR divided by the time to pHRR. A higher FGI represents that the material can be easily ignited and reach a higher pHRR $[20, 21]$ $[20, 21]$ $[20, 21]$ $[20, 21]$ $[20, 21]$. The FPI of EP and EP-4 is 0.033 and 0.14 sm²/kW, respectively, and the FGI is 11.50 and 4.18 kW/m^2 /s, respectively. With the incorporation of HDP and THEIC into EP, the FPI of the EP-4 sample is increased and the FGI is decreased, indicating that the fre safety performance of the composites has been improved.

Thermal stability

The thermogravimetric analysis (TGA) curves and derivative thermogravimetric (DTG) curves for all samples are shown in Fig. [4](#page-8-0) and Table [3.](#page-9-0) The temperature corresponding to a 5% weight loss of the material is defned as the onset decomposition temperature (T_{onset}) . The temperature at the maximum weight loss rate is defined as the fastest decomposition temperature (T_{max}) [\[22](#page-19-3)].

The decomposition of EP is very fast, featuring only a one-step decomposition. The T_{onset} and T_{max} appear at 333.8 and 371.9 °C, respectively. The decomposition is basically fnished at 500 °C, and there is only 11.80% of the initial mass residue left behind at 700 °C. The thermal decomposition of HDP mainly occurs between 250 and 450 °C, and the T_{onset} and T_{max} appear at 277.5 and 354.6 °C, respectively. In addition, there is still a small weight loss process near 540 °C. This is because the decomposition of HDP occurs in two steps; the frst step is caused by the breakage of HDP's own structure, while the decomposition at 540 \degree C is caused by the further

Fig. 4 TGA and DTG curves for EP, HDP, THEIC (**a**, **b**), and EP composites (**c**, **d**)

Table 3 Dat DTG curves THEIC, and

dehydration and decomposition of polyphosphoric acid [\[11](#page-18-8), [23](#page-19-4)]. There is 13.64% mass residue at 700 °C. The decomposition of THEIC is only one step that occurs within the temperature range of 250 to 400 $^{\circ}$ C, and the decomposition is complete without any mass residue.

The TGA curves and DTG curves of the composites are shown in Fig. [4c](#page-8-0), d, and all the samples showed a similar decomposition process, which was concentrated mainly occurs between 200 and 500 °C.

As shown in Table 3 , the T_{onset} of the EP composites is significantly lower than that of the neat EP, which is caused by the lower decomposition temperature of HDP and THEIC. The T_{onset} of all composites decreases as the percentage of THEIC increases, which is due to the lower T_{onset} of THEIC than HDP. Thus, the added THEIC amount has a greater effect on the T_{onset} .

It is noteworthy that the DTG curve of sample the EP-7 sample shows two obvious peaks at both 273.1 and 373.7 \degree C, respectively. The first weight loss peak is due to the decomposition of THEIC, and the second peak at 373.7 \degree C is due to the decomposition of EP. This indicates that adding THEIC alone will only accelerate the decomposition of composites. For the EP-1 sample, the T_{max} appears at 331.2 °C. In comparison, HDP compounded with THEIC results in a higher T_{max} , which indicates that the combination of HDP and THEIC can increase the decomposition of the sample.

Comparing the fnal mass residues of all samples, it can be found that the mass residues of all samples showed frst increase and decrease with the ratio of THEIC. Only the EP-2 and EP-3 samples have more mass residues than the EP-1 sample, other samples have fewer mass residues, but the $EP-2 \sim EP-6$ samples have more mass residues than the EP-7 sample. This indicates that THEIC has a certain carbon formation ability efect after being combined with HDP. Theoretically, the amount of residues char is generally linked to the fame-retardant efect of the composites. However, it is true that the EP-4 sample has the best fame-retardant efect, indicating that the amount of residual char from the decomposition of the sample cannot entirely refect the synergistic efect between THEIC and HDP, and further research is needed.

Condensed phase analysis

Microstructures of the residual chars

The fame-retardant efect of the material is not only related to the number of residual char yields but is also related to the microstructure of the residual carbon. To further study the synergistic efect between HDP and THEIC and the fame-retardant mechanism of the condensed phase, the microstructure of the residual char was characterized by SEM. Figure [5](#page-10-0) shows the microscopic morphology of the residual chars after the combustion of each composite.

The results confrm that the microstructure of the residual chars is closely related to the ratio of HDP and THEIC. The residual chars from EP (a) are full of fssures and have a broken structure. The residual chars of EP-1 (Fig. [5](#page-10-0)b) are loose and fragmented when only HDP is added, while the residual chars of EP-7 (Fig. [5h](#page-10-0)) are dense but with large pores when only THEIC is added, and such a residual char structure is not conducive to slowing down the heat and mass transfer between the unburned matrix and the combustion zone. When HDP and THEIC are added in diferent ratios, the structure of the residual chars is signifcantly changed. As the

Fig. 5 SEM images of the residual char of EP (**a**), EP-1 (**b**), EP-2 (**c**), EP-3 (**d**), EP-4 (**e**), EP-5 (**f**), EP-6 (**g**), EP-7 (**h**)

ratio of THEIC increases, the residual chars gradually become dense and form uniform pores in the carbon layer (EP (Fig. $5c$) ~ EP (Fig. $5g$ $5g$)). This is because HDP produces phosphoric acid derivatives during the thermal decomposition, to catalyze the transformation of THEIC and other hydroxyl-containing compounds from EP decomposition into carbon. Meanwhile, the decomposition of THEIC will produce $NH₃$, which will leave cavities in the residual chars. However, as with the increase of the THEIC ratio, a large amount of $NH₃$ is generated, which leads to the expansion of pore channels and fnally structural damage in the carbon layer. A stable residual char structure can ensure that the carbon layer will not be broken during combustion. The uniform pore channels can isolate the heat generated by the combustion and retain the non-combustible gases produced by the decomposition, dilute the air and prevent further decomposition of the material. It can be seen from Fig. [5](#page-10-0) that the char from EP-4 (Fig. [5e](#page-10-0)) has alveolate structures, and the size and distribution are the most uniform compared to other samples. Therefore, this illustrates the reason why the amount of residual chars from the EP-4 sample is not the highest, but it can bring the best flame-retardant effect among those samples.

Raman spectroscopy can efectively characterize the graphitization degree of the residual chars. The characteristic peaks at 1358 and 1580 cm−1 in the spectra are called D-peak and G-peak, respectively, which represent the vibration of amorphous carbon and crystalline graphite, respectively. Therefore, the ratio of the integrated peak intensity of D to G peak (I_p/I_c) can indicate the graphitization degree of the residual chars, and a lower I_D/I_G value indicates a higher graphitization degree of the residual carbon and a better thermal insulation effect $[24, 25]$ $[24, 25]$ $[24, 25]$ $[24, 25]$. Figure [6](#page-11-0) shows the Raman spectra of the residual chars of EP and EP-4. The I_D/I_C value of the EP-4 sample is 1.403, smaller than that of the EP sample being at 1.865. The results indicate that the addition of HDP and THEIC can increase the degree of graphitization of the residual carbon, thereby enhancing the strength of the carbon layer. Then, a stronger carbon layer can better isolate the heat transfer and avoid further decomposition of the composites.

In conclusion, the combination of THEIC and HDP can increase the graphitization degree of the residual chars and strengthen the structure of the carbon layer. The decomposition of THEIC will also produce $NH₃$ to form uniform pores in the

Fig. 6 Raman spectrum of residual carbon of EP (**a**), EP-4 (**b**)

carbon layer. The dense and porous carbon layer efectively enhances the condensed phase fame-retardant efect of the material and reduces the mass and heat exchange between the composites and the combustion area, thus enhancing the fame-retardant performance of the composites.

Chemical composition of the residual char

EP-4 has the best fame-retardant efect and can best refect the synergistic efect between HDP and THEIC. Therefore, EP-4 and EP were selected as a comparison to study the diference in the chemical composition of residual carbon. Figure [7](#page-12-0) shows the comparison of the elemental composition of the EP-4 before and after the combustion. The presence of N is clear in the EP-4 sample before burning, which is caused by the addition of THEIC. However, N was no longer detected in the sample after combustion, indicating that all N atoms in THEIC were released in the form of gas. In addition, the contents of C and P in the residual char increase, but the content of O decreases. This indicates that a part of the P remains in the residual carbon. This may be a plausible reason for the observed phenomenon: Phosphoric acid derivatives released from the decomposition of HDP are involved in the dehydration of the material into carbon during combustion, which releases water into the air. So, it leads to the decrease of the O content and the increase of the P content. The

Fig. 7 SEM image and elemental analysis of the EP-4 sample before and after combustion

increase of the C content indicates that the carbon layer formed efectively protects the material and prevents further decomposition.

Figure [8](#page-13-0) shows the spectra of the residual char, which gives a further proof of the chemical compositions of the carbon layer. The absorption peaks around 1600 and 1236 cm−1 are attributed to the C=C and C–C stretching vibrations of aromatic compounds. Besides, there are some additional absorption peaks present in the residual carbon IR spectrum of the residual carbon from the EP-4 sample, including the P=O groups (around 1165 cm⁻¹), P–O groups (around 1080 and 752 cm^{-1}), and P–O–C groups (around 985 cm⁻¹). This indicates that the phosphate derivatives from the HDP decomposition catalyze the conversion of EP and THEIC to organophosphates with aromatic structures during the combustion and the char layer is mainly composed of graphite-like complexes, with aromatic structures bridged by P–O–C groups [\[8](#page-18-6), [22](#page-19-3), [26](#page-19-7)].

Gas phase analysis

To further investigate the gas-phase mechanism, a TG-IR analysis of the EP and EP-4 samples was carried out under air conditions. Figure [9](#page-14-0) shows the IR spectra of the decomposition products of the EP and EP-4 samples at diferent temperatures.

The characteristic peaks of typical thermal decomposition products of epoxy resin were observed at 3500 ~ 4000 cm⁻¹, 2800 ~ 3100 cm⁻¹, 2400 ~ 2200 cm⁻¹, 2200 ~ 2000 cm⁻¹, 1735 and 1172 cm⁻¹ [[27](#page-19-8)]. The peaks at 3500 ~ 4000 cm⁻¹ belong to the vibration of O–H groups, which are from phenols and water produced by the EP decomposition. The peaks at $2800 \sim 3100 \text{ cm}^{-1}$ belong to –CH₂–,

Fig. 8 FTIR spectra of the residual char form EP and EP-4

Fig. 9 FTIR spectra of the pyrolysis products at diferent temperatures (**a**, **b**) and 3D TG-FTIR profles (**c**, **d**) obtained for the EP (left column) and EP-4 (right column)

–CH₃ produced by hydrocarbon pyrolysis. The peaks at $2400 \sim 2200$ cm⁻¹ and 2200 ~ 2000 cm⁻¹ were related to the characteristic peaks of CO_2 and CO, respectively. The peaks at 1735 and 1172 cm^{-1} are attributed to C=O and C–O–C groups. The spectra of the EP and EP-4 decomposition products are similar. In addition to the above typical absorption peaks, the absorption peaks at 1304 cm^{-1} are related to P=O. The absorption peaks at 3332, 965, and 930 cm−1 corresponded to NH_3 produced by the decomposition [\[28](#page-19-9)]. This proves that all N atoms in THEIC are released in the form of NH_3 . In summary, PO, PO₂ radicals and NH_3 will be produced during the combustion of EP-4. NH_3 , CO₂, and H₂O can dilute ambient oxygen, and the PO and $PO₂$ radicals generated by the decomposition can react with H and OH to terminate the radical chain generated during the combustion, which inhibits further decomposition of the material and thus has a quenching efect. The results of TG-IR analyses show that the introduction of HDP and THEIC renders the composites also a gas-phase fame-retardant mechanism.

Flame‑retardant mechanism

Based on the results obtained, a possible fame-retardant mechanism is proposed for the combined HDP and THEIC system, as shown in Fig. [10.](#page-15-0)

In the condensed phase, the decomposition of HDP produces phosphate derivatives, which can catalyze the dehydration of THEIC and the hydroxyl-containing compounds produced by the decomposition of EP to form a dense carbon layer, thus inhibiting further combustion of the material. In the gas phase, the decomposition of HDP and THEIC can produce CO_2 , H₂O, NH₃ and other non-combustible gases to dilute the combustible gas and oxygen. In addition, PO and $PO₂$ radicals are produced during the decomposition of HDP can react with a large number of active radicals (H and OH) produced by the decomposition of EP to slow down the further decomposition of the material. Moreover, the gas can make the carbon layer porous and strengthen the fame-retardant efect of the condensed phase. In conclusion, the fame-retardant mechanism of the combined HDP and THEIC system is a synergistic combination of condensed phase and gas phase fame-retardant mechanisms.

Mechanical properties

Analysis of the cross‑sectional morphology of composites

The compatibility of fame retardant and epoxy resin can be analyzed by observing the cross-sectional morphology of the composite. Figure [11](#page-16-0) shows the cross-sectional morphology of all samples. It can be seen that the cross-sectional morphology of EP is relatively smooth, and there are only a few folds on the section. The same is true when only HDP is added, indicating that HDP has good compatibility with EP.

Fig. 10 Schematic of the fame-retardant mechanism of EP-4

Fig. 11 SEM images of cross-sectional morphology of EP (**a**), EP-1 (**b**), EP-2 (**c**), EP-3 (**d**), EP-4 (**e**), EP-5 (**f**), EP-6 (**g**), EP-7 (**h**)

However, with the increase of THEIC ratio, the cross-sectional morphology of the composites becomes rough, indicating that the compatibility between THEIC and EP is poor. The addition of THEIC will affect the mechanical properties of the composites to a certain extent.

Analysis of tensile and fexural tests results

Tensile and fexural tests were used to further analyze the efect of HDP and THEIC addition on the mechanical properties of composites. Figure [12](#page-17-0) shows the comparison of tensile and fexural properties of samples with diferent HDP/THEIC weight ratios.

It can be observed that the tensile and fexural strengths of the samples decrease with the increase of the THEIC ratio, while the tensile and fexural modulus show an increasing trend in general. The decrease in strength indicates that the addition of THEIC reduces the maximum stress that the material can withstand when subjected

Fig. 12 The efect of diferent HDP: THEIC weight ratios on the mechanical properties of the composites

to deformation, while the increase in modulus indicates that the rigidity of the composites increases and less deformation occurs when an external force is applied. The increase in modulus is possibly due to the rigidity of THEIC being higher than that of neat EP. When the rigidity of the particles is higher than the matrix, its addition to the matrix can increase the modulus of the composites. The tensile and fexural strengths of the EP sample are 44.70 and 103.72 MPa, respectively, and the tensile and fexural strengths of the EP-4 sample, the one with the best fame-retardant property is 32.93 and 74.42 MPa, respectively, each with a decrease of 26.33 and 28.25%, respectively, as compared to EP. This is because that THEIC is added in a solid form and does not blend well with EP, thus leading to a decrease in the tensile and fexural strength of the material [\[29](#page-19-10)]. The compatibility of THEIC with EP needs to be improved to reduce the efect of THEIC on the mechanical properties of the material. Chemical modifcation of THEIC may improve its compatibility with EP.

Conclusions

HDP and THEIC were added to the epoxy resin, and the fame-retardant synergistic efect was investigated. THEIC and HDP show excellent synergistic fame-retardant efects. The LOI value of the EP-4 sample with 15 wt% HDP and 15 wt% THEIC reaches up to 27.5% increased by 26.7% compared to neat EP. Also, the fame retardancy of the EP-4 sample reaches a V-0 rating in the UL-94 test. The results of the cone calorimeter test showed that HRR, THR, SPR, TSP, $CO₂$ release rate, and CO release rate are all signifcantly reduced.

The analyses of the residual chars and released gases after the combustion of the EP composites show that the introduction of HDP and THEIC facilitates the formation of uniform cavities in the carbon layer, which can efectively hinder the heat and material exchange between the composite and the combustion area and enhances the condensed phase flame-retardant effect. Moreover, $NH₃$, CO₂, H₂O, PO⁻ and PO_2^- radicals generated by the decomposition of the composites effectively improve the gas phase fame-retardant efect. These results prove that the fame-retardant

mechanism of the EP composites contains the condensed phase and the gas phase actions simultaneously. However, the addition of THEIC will have some infuence on the mechanical properties of EP composites. The reason for this phenomenon is that THEIC is added to EP in the form of solid particles, which has poor compatibility with EP. Microencapsulation of THEIC and its addition to EP may achieve good results.

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