ORIGINAL PAPER

Physicochemical, optical and electrical investigation on poly [(phenylene‑2‑one)‑co‑(thiophene)] novel soluble conductive polymer as‑synthesized through heterogeneous catalysis route

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Received: 30 June 2020 / Revised: 27 September 2020 / Accepted: 13 October 2020 / Published online: 23 October 2020 © Springer-Verlag GmbH Germany, part of Springer Nature 2020

Abstract

Although the use of conductive polymers has invaded the electronics industry, the insolubility parameter presents a major problem against their direct application on the surfaces of materials. Attachment of cyclic compounds to polymer chains is recommended to increase the solubility of conductive polymers. The aim of this study is to synthesize a new type of intrinsic conductive polymer, soluble in common solvents. The technique is based on the copolymerization of thiophene with a synthesized monomer phenylazepane-2-one. The reaction was catalyzed by a solid catalyst, prepared by the acid treatment of natural clay, generating active sites responsible for the adhesion of the thiophene and benzene rings. Proton and carbon nuclear magnetic resonances $({}^{1}H NMR/{}^{13}C NMR)$, ultraviolet spectroscopy (UV–Visible), infrared spectroscopy (IR) and diferential scanning calorimetry (DSC) were used to identify the material obtained poly [(phenylazepane-2-one)-co-(thiophene)] abbreviated poly (PAT). Thermogravimetric analysis (TGA) showed the thermal stability of poly (PAT) before 200 °C. The solubility of poly (PAT) has been tested and confrmed in various common solvents. The indirect bandgap was calculated at 1.12 eV using Tauc formula. Ac electrical conductivity and dielectric permittivity have been studied versus frequency and temperature, showing the semiconductor character of poly (PAT). The results obtained showed that this novel polymer can play a pioneering role to replace conventional insoluble conducting polymers.

Keywords Thiophene · Conductive polymer · Poly (pat) · Bandgap · Permittivity

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Introduction

The intrinsic electrical conductive polymers have become more and more attractive $[1-3]$ $[1-3]$, in the face of the scarcity and high cost of the raw materials making up the inorganic semiconductors $[4–6]$ $[4–6]$ $[4–6]$ $[4–6]$. Their ease of implementation, lightness and low cost make them good candidates for technological applications [\[7,](#page-16-4) [8](#page-16-5)]. π-conjugated polymers consist of alternating single and double carbon–carbon bonds $[9-11]$ $[9-11]$. It is precisely this delocalized π -conjugated system which gives these polymers their particular electronic properties [\[12](#page-17-1)–[14\]](#page-17-2).

Thiophene based polymers are among the most studied and used polymers as π-conjugated materials. So work on thiophene and its derivatives is an important part of conductive polymer research [\[15\]](#page-17-3). Studies on these compounds as conductive polymers began in the early 1980s. In 1986, Elsenbaumer et al. [[16](#page-17-4)], prepared polythiophene making it the frst heterocyclic conjugate polymer able to be shaped. This step decisive for technological applications has been followed in 1988 by the use of polythiophene in the manufacture of the frst real feld-efect transistor based on organic materials [[17](#page-17-5)]. Since then, polythiophene has been the subject of numerous publications and researches using this polymer in the feld of optoelectronics [\[18](#page-17-6), [19](#page-17-7)]. Polythiophene is part of the family of heterocyclic conductive polymers, which is to say that a diferent atom of the carbon atom is present in its cycle, it is a sulfur atom. Its chemical structure gives it interesting properties in terms of stability and resistance to heat. The major disadvantage of this polymer is its low solubility. To deal with this problem, alkyl chains have been introduced at the 3-position of the cycle. These chains provide a relatively high solubility, but this remains a solution to be avoided because of its unexpected and undesirable impact on the fnal polymer properties, which can come from crosslinking phenomena. The synthesis of copolymers has proven to be an efective means of compensating for certain defects in conductive polymers, by being able to modulate: chemical and thermal stability, optical properties and solubility, by varying the nature and composition of the monomers, resulting in a diferent polymer structure [[20](#page-17-8), [21\]](#page-17-9).

Clays can serve as a support for a wide variety of chemical reagents. This is attributed to their ability to concentrate large amounts of reactive species between their aluminosilicate layers [\[22](#page-17-10), [23\]](#page-17-11). Recently, for many chemical reactions, chemically modifed clays have shown a great interest in its use as a catalyst support instead of homogeneous catalysts [\[24](#page-17-12)[–26](#page-17-13)]. The main advantage lies in the ease of separating and recovering this type of catalyst from the reaction medium, it requires only simple fltration, as well as they are reusable and regenerable [\[27](#page-17-14), [28](#page-17-15)].

The clay used in our study is Maghnite, obtained from the mountains of the Maghnia region (west of Algeria) from which its name was derived. It is characterized by a great cation exchange capacity [\[29](#page-18-0)]. Its aluminum/silicon portion is relatively higher when compared to other world clays [[30](#page-18-1)]. The chemically activated Maghnite has shown a large catalytic capacity, which has enabled it to be used in a large number of polymerization reactions, published by the polymer chemistry laboratory of the University of Oran in Algeria [\[31–](#page-18-2)[33](#page-18-3)].

The originality of our work is to synthesize a novel conductive polymer, based on thiophene and a monomer synthesized from caprolactam and phenol (Fig. [1](#page-3-0)). The copolymerization reaction was catalyzed by a green ecological catalyst Maghnite- H^+ . Maghnite- H^+ was prepared by an acid treatment of raw-Maghnite. The insertion of the initiating hydronium ions between the Maghnite layers was shown by X-ray powder difraction (XRD) analysis. The chemical structure of the synthesized monomer and polymer was studied in detail by $H NMR$ and $H^3C NMR$ spectroscopy. Operating conditions such as time, temperature and catalyst content have been well optimized in order to obtain a better yield, and at the same time linear molecular structures. The solubility of poly (PAT) has been tested in dimethylsulfoxide, tetrahydrofuran, *N,N-*dimethylformamide, sulfolane and water. The thermal stability of poly (PAT) has been confrmed at temperatures below 200 °C, these behaviors allow it to meet several industrial requirements. During the diferent stages synthesis of poly (PAT), we tried to follow a strategy which largely conforms to the principles of green chemistry.

Experimental section

Materials

ε-Caprolactam, phenol, thiophene, acetic anhydride, dimethylsulfoxide, *N,N-*dimethylformamide, tetrahydrofuran, sulfolane, toluene, methanol and dichloromethane were purchased from Aldrich Chemical Co Algeria. Raw-Maghnite was procured from Algerian Society of Bentonite (BENTAL). XRD analysis of Maghnite and prepared Maghnite (Maghnite-H⁺) was recorded using Bruker X-Ray diffractometer from Germany at the rate of 2° min⁻¹ in the 2*θ* range of 2.0–80.0°. Various methods of analysis were used to identify and characterize the polymer obtained such as FT-IR spectroscopy (Perkin-Elmer System, USA), UV–Vis spectroscopy (OPTIZEN 2120, Republic of Korea), ¹H-NMR and ¹³C-NMR (300 MHz Bruker, Germany). The thermogravimetric analysis (ATG) was performed by Labsys Evo Setaram (France) with a heating rate of 10 °C/min. The DSC analysis was carried out on a Setaram 92 DSC apparatus (France) under a helium fow rate of 20 ml/min, the sample (4.8 mg) to be analyzed was preheated before registering the spectrum. The rate of heating was $10 \degree C/\text{min}$. The ac electrical conductivity and the dielectric measurements of the poly (PAT) were determined by two-probe method at 1 V, in a frequency range from 20 Hz to 10 MHz, using a GW-Instek 82-LCR-Meter bridge with PC (Taiwan).

Catalyst preparation

Maghnite- H^+ was prepared by treating the raw-Maghnite with a solution of sulfuric acid 0.25 M, at room temperature over a period of two days until saturation was achieved. The cation-exchanged Maghnite was: fltered, resuspended in deionized water continuously until no sulfate ions were detected in the filtrate by $BaCl₂$,

Fig. 1 The synthetic route of poly (PAT)

isolated by filtration, dried in an oven at 105 \degree C and then finely ground. The clay cation exchange capacity (CEC) was found to be 84 meq/100 g of dried Maghnite- H^+ [[12\]](#page-17-1).

Monomer synthesis

Phenylazepane-2-one (PA) was prepared by the reaction of caprolactam with phenol. So in a three-neck fask, 5 g of caprolactam was solubilized in excess of toluene. After stabilizing the temperature at 70 \degree C, 4.25 g of phenol and 4.6 g of acetic anhydride were added to the flask. Then, 0.1 g of the previously prepared Maghnite- H^+ was added to the reaction mixture. The mixture was left for 2 h with gentle stirring. At the end of the reaction, a light yellow precipitate in powder form was recovered by fltration, dried under vacuum and weighed. The yield of the reaction was about 86%.

Polymer synthesis

The copolymerization of thiophene and synthesized phenylazepane-2-one took place in solution under inert atmosphere (nitrogen), using Maghnite-H+ as catalyst. Maghnite-H⁺ was dried at 100 $^{\circ}$ C for one hour before being used. The role of Maghnite-H⁺ was dehydrogenation of thiophene (1 g) and phenylazepane-2-one (4.86 g) dissolved in dichloromethane. The reaction synthesis was carried out during 5 h at 25 °C. Finally, and after removing Maghnite- H^+ from the mixture by filtration, methanol was added slowly to the mixture with gentle stirring to precipitate the polymer. The copolymer obtained was dried under vacuum at room temperature overnight. The yield of the reaction was about 91%. This previously described procedure was repeated several times at diferent temperatures, times and catalyst contents.

Results and discussion

XRD characterization of Maghnite‑H⁺

The X-ray powder diffraction patterns of raw-Maghnite and Maghnite-H⁺ (Fig. [2](#page-5-0)) showed the presence of other crystalline phases in addition to montmorillonite, such as mica, calcite and quartz. Under the efect of the acid treatment of raw-Maghnite, the mica and calcite contents were clearly reduced [\[34](#page-18-4)]. this was deduced from the decrease in the intensity of the corresponding peaks located at 19.74°, 62.14° for mica, and 20.81°, 39.61° for calcite [\[10](#page-16-7)]. The removal of most of the quartz located at 20.81° and 26.54° [\[35](#page-18-5)] is probably due to the mechanical treatment of raw-Maghnite by the sedimentation process. The increase in basal spacing corresponding to the space of montmorillonite layers; from 10.5 \AA in raw-Maghnite to 15.02 \AA in Maghnite- H^+ [[10\]](#page-16-7) was reflected by the shift of the montmorillonite peak towards the small values of θ (8.41° to 5.87°) [[36\]](#page-18-6). This is explained by the replacement of

Fig. 2 X-ray powder difraction of raw-Maghnite and Maghnite-H⁺ **a** structures, **b** montmorillonite peak and **c** XRD spectra.

interlamellar ions such as Mg^{2+} and Ca^{2+} by gigantic hydronium ions of the relatively large molecular radius [[31\]](#page-18-2). In general, we note that acid treatment of Maghnite increased basal spacing of montmorillonite, but did not afect the original Maghnite structure, which was confrmed by the absence of the appearance or disappearance of peaks in XRD spectrum of Maghnite-H+.

Characterization

Figure [3](#page-6-0) shows the optical absorption spectrum of dissolved poly (PAT) in DMSO. The color of the polymer solution was blackened red. The absorption spectrum makes appear three absorption bands located at 324, 347 and 418 nm assigned to; the C=C chromophore present in the aromatic benzene and thiophene rings, the nonbonding electronic level in nitrogen or sulfur atoms and the carbonyl groups of caprolactam, respectively. This curve obtained by UV–Visible spectroscopy makes it possible to calculate the absorption coefficient, which is a necessary parameter in order to then determine the optical bandgap of our synthesized material.

The ¹H-NMR spectra of synthesized monomer and polymer were obtained to identify and confrm their chemical structure. Figure [4](#page-7-0) shows the superposition of the ¹ H-NMR spectra of PA (**a**) and poly (PAT) (**b**). The resonances at 1.69, 2.54 and 3.33 ppm for the PA and 1.71, 2.53 and 3.36 ppm for the poly (PAT) are assigned to caprolactam ring protons. The peaks H_c (2H, d) at 6.66 ppm, H_a (1H, t) at 6.70 and H_b (2H, t) at 6.74 ppm for PA and H_c (2H, s) at 6.47 and H_a (1H, s) at 6.71 ppm for poly (PAT) correspond to phenyl ring protons. The disappearance of the peak H_b

Fig. 3 UV–visible spectrum of poly (PAT)

in the spectrum of the polymer and the conversion of H_a and H_c into singular gave the reason to conceive that the attachment of the thiophene ring on the phenyl ring was at the carbon atom in the meta position. The thiophene ring protons H_g (2H, d) appear at 6.89 ppm. The appearance of the H_g peak as a doublet peak shows that the attachment to the thiophene cycle was in atoms 1 and 4. The small peaks H_h at 7.12 ppm and H_i at 7.92 ppm are attributed to phenyl and thiophene ring protons, respectively, at the end of the polymer chain. The number average molecular weight (M_n) of poly (PAT) was calculated after integration, by comparing the signal areas corresponding to the protons of the thiophene ring in the repeating unit (H_{α}) to that of the terminal group (H_i) , it is about 8900 g/mol. The molar ratio of thiophene and PA in the poly (PAT) chain is 31.71% and 68.29% respectively.

To complete the structural study of synthesized products, 1H NMR analysis was supported by 13 C NMR analysis to identify and confirm the carbon–car-bon bonds. Figure [5](#page-8-0) shows the ¹³C NMR spectra of the monomer (curve above) and the polymer (curve below). The carbons of the phenyl ring are observed at 122.80, 128.86, 117.07 and 141.73 ppm for the monomer $(C_1 C_2 C_3$ and C_4) and at 122.80, 136.12, 117.10 and 141.66 ppm for the polymer $(C'_1, C'_2, C'_3$ and C'_{4}). The displacement of the C_{2} peak from 128.86 in the monomer to 136.12 in

Fig. 4 $\frac{1}{1}$ H NMR spectra of PA (**a**) and poly (PAT) (**b**)

the polymer indicates the conversion of the secondary carbon to a tertiary carbon, due to the creation of a new bond with the thiophene carbon. The signals appeared at 172.45, 50.65, 34.56, 29.92, 24.20 and 29.21 ppm for the monomer (C_5 , C_6 , C_7 , C_8 , C_9 and C_{10}), and at 172.75, 50.29, 34.94, 29.93, 24.91 and 29.22 ppm for the polymer (C'_{5} , C'_{6} , C'_{7} , C'_{8} , C'_{9} and C'_{10}) are assigned to carbon atoms belonging to the caprolactam ring. The attachment of the thiophene ring gave rise to the birth of two peaks C_{11} and C_{12} located, respectively, at 137.09 and 118.87 ppm. The two small peaks α and β saw at 128.88 and 122.14 ppm are, respectively, assigned to the carbons of thiophene and phenyl rings at the ends of the polymer chain. These results are to a large extent compatible with those obtained by ${}^{1}H$ NMR analysis.

Figure [6](#page-9-0) shows the IR absorption spectrum of poly (PAT) from which we will cite the main bands. The two mid-intensity bands at 2931 and 2854 cm^{-1} correspond to the C–H symmetric and asymmetric stretch, respectively. The band at 1729 cm−1 is C=O stretching vibration of the caprolactam ring. The band at 1600 cm−1 is assigned to C=C stretching vibration of the phenyl ring. The band at 1504 cm⁻¹ is assigned to the C=C stretching vibration bond of the thiophene rings. The mid-intensity band at 1311 cm−1 corresponds to the valence vibration

Fig. 5^{13} C NMR spectra of PA **a** and poly (PAT) **b**

of the C–C bond. The band at 1248 cm^{-1} is assigned to the valence vibration of the C-H bond. The vibration band located at 805 cm^{-1} corresponds to the deformation of the C-S-C bond. The IR spectrum was in good agreement with the proposed structure.

Thermal properties

Figure [7](#page-10-0) shows the TGA, DTG and DSC thermograms of poly (PAT) obtained in a helium atmosphere, at a heating rate of 10 °C/min. This polymer has good thermal stability up to a temperature of around 200 $^{\circ}$ C, beyond this temperature; the polymer began to undergo thermal decomposition in a single main stage. This result showed a signifcant improvement in the degradation temperature compared to other synthesized polythiophene; 180 °C [\[37](#page-18-7)], polythiophene composites doped with copper; 170 °C [[38\]](#page-18-8) and similar to that of polythiophene-fullerenes diblock copolymer [[39\]](#page-18-9). The DSC curve shows a small temperature variation at about 175 °C, which represents the glass transition temperature of the polymer, as well as an endothermic peak at about 259 °C due to the melting phenomenon. In the temperature range between 300 and 550 °C, several endothermic and exothermic efects are noticed in the DSC signal, due to the pyrolytic decomposition of the polymer.

Fig. 6 IR spectrum of poly (PAT)

Conductivity measurements

Figure [8](#page-11-0)a shows the evolution of ac electrical conductivity (σ_{ac}), the real part of the permittivity (ε') , the imaginary part of the permittivity (ε'') and the dielectric tangent lost tang (δ) at room temperature. The measurements were made in a frequency range between 20 Hz and 10 MHz. First of all, in Fig [8a](#page-11-0), the ac electrical conductivity increased continuously with increasing frequency applied. This can be since, as the frequency increased, additional electrons are conducted across the interface due to the frequency-assisted electron hopping [\[40\]](#page-18-10). The study of the variation of dielectric permittivity constant as a function of the frequency makes it possible to fundamentally defne the polarization capacity of the poly (PAT) during the application of the feld. Its two real and imaginary parts decreased with increasing frequency; this is due to the dominance of the relaxation process. The values of ε' and ε'' became 7.98 and 0.81, respectively, at the maximum frequency. Tang (δ) is the ratio of ε' to ε'' , it decreased with increasing frequency due to the dominance of the capacitive element against the ohmic component. The relaxation time has been calculated around 1953 µs. The ac electrical conductivity also measured as a function of temperature at

Fig. 7 TGA, DTG and DSC thermograms of poly (PAT)

diferent frequencies (Fig [8b](#page-11-0)). Remarkably, that the ac electrical conductivity of poly (PAT) increased with increasing temperature. This is attributed to the increased likelihood of electrons in the valence band getting enough energy to join the conduction band [[41](#page-18-11), [42\]](#page-18-12). This result confrmed that poly (PAT) exhibits semiconducting material behavior. The increase in ac electrical conductivity also with frequency is due to the existence of polymer defect sites, permitting the movement of charges. By way of comparison, Table [1](#page-12-0) presents the values of electrical conductivity of polythiophene and of some of its derivatives obtained previously Fig. [9](#page-12-1).

Optical analysis

In order to determine the indirect bandgap of poly (PAF) flm, the extrapolation of the plot of $(ahv)^{1/2}$ versus *hv* was carried out, the results are shown in Fig. [9.](#page-12-1) The region of strong absorption was exploited to determine the indirect optical bandgap (Eg), using Tauc formula. The value of the indirect bandgap was obtained 1.12 eV, from the extrapolation of the line to the line base, where the value of the $(ahv)^{1/2}$ is equal to zero. The result obtained allows us to conclude that poly (PAT) is a π -conjugated semiconducting at room temperature ($\langle 2 eV \rangle$.

Fig. 8 a Variation of σ_{ac} , *ɛ'*, *ɛ''* and tang (*δ*) of poly (PAT) with frequency, **b** variation of σ_{ac} of poly (PAT) with temperature at diferent frequencies

Solubility test

The solubility of poly (PAT) in common organic solvents is one of the main objectives expected from the copolymerization of thiophene with PA. Therefore the solubility parameter was evaluated in dimethylsulfoxide, tetrahydrofuran,

Fig. 9 Determination of the indirect bandgap of poly (PAT) from a plot of (*αhν*) 1/2 versus *hν*

Table 1 Conductivity of polythiophene and its derivatives

N,N-dimethylformamide, sulfolane and water. An amount of 1 g of poly (PAT) was introduced into 30 ml of solvent, the mixture was allowed to stir for 30 min at room temperature. The quantity of poly (PAT) which remained insoluble was recovered by fltration, dried and then weighed. The following expression made it possible to calculate the solubility index. The examination results are shown in Fig. [10](#page-13-0).

Solubility index (
$$
\%
$$
) = $\frac{\text{Initial product mass} - \text{Insoluble product mass}}{\text{Initial product mass}}$

According to the values of the calculated solubility index, most of the poly (PAT) was found to be soluble in the solvents tested, relatively little in the water. Poly (PAT) has conjugated chains due to the degree of dehydrogenation, giving magenta solutions of high concentrations. The very good solubility of poly (PAT) despite their high degree of π -conjugation is due essentially to the large unitary groups (caprolactam/benzene), which are meta linked to the carbon atom of the thiophene ring, and also is largely due to the linear molecular structure of poly (PAT);

Fig. 10 Solubility test results

absence of any kind of interchain bonds, that has been confrmed by the structural characterization.

Efect of the amount of Maghnite‑H+, reaction time and temperature on the synthesis of poly (PAT)

In the frst part, we followed the efect of time on the reaction synthesis of poly (PAT) for different amounts of Maghnite-H⁺. Figure [11](#page-14-0)a shows the effect of time on the polymerization yield, investigated by using various weight ratios Maghnite- H^{\dagger} / monomers. The polymerization was carried out in solution (CH_2Cl_2) and at a temperature of 25 °C. The polymerization yield increased over time, reaching 91% at 5 h by using 3% of Maghnite-H⁺. The reaction yield remained constant beyond 5 h. This can be explained by the total consumption of reagents. The remaining 9% is oligomers soluble in CH_2Cl_2 . The continuous increase in reaction yield, depending on the amount of Maghnite- H^+ used, is probably the result of the number of initiating active sites responsible for the induction of the polymerization, this number is proportional to the amount of catalyst used in the reaction.

In the second party, the temperature efect on the polymerization reaction of thiophene with PA was followed. The polymerization was carried out for 5 h in the presence of 3 wt% of Maghnite-H⁺. The results obtained at different temperatures are shown in Fig. [11](#page-14-0)b. From these results, we conclude that the optimal temperature for the reaction is 25 °C (the yields reached a maximum). After this temperature, the reaction yield was reduced, indicating that the temperature 25 °C is the ceiling temperature, at which the rates of polymerization and depolymerization phenomena are equal. These results showed the double role in the opposite direction played by Maghnite- H^+ as a cationic catalyst. Same results have been obtained during the synthesis of polyurethane using Maghnite- H^+ as catalyst [[30\]](#page-18-1).

Fig. 11 Effect of the amount of Maghnite-H⁺, reaction time and temperature on the reaction yield of poly (PAT) synthesis

Figure [12](#page-15-0) shows the proposed reaction mechanism of the copolymerization reaction of thiophene with PA, using Maghnite- H^+ as a catalyst. The initiation step was based on the creation of active sites, by fixing the H^+ ion of Maghnite- H^+ on a thiophene ring sulfur atom, this led to the generation of a sulfonium ion. In the propagation step, the carbon atom adjacent to the sulfonium atom attaches to a phenyl ring carbon. This efect leads to the creation of the carbenium ion on another carbon atom of the phenyl ring, the latter becoming the new active site. This operation continues in an alternative way one by one, where there is each time a displacement of the active site, capable of capturing a free monomer. The process was propagated by chain-growth polymerization. The termination step is reached by adding water, thus deactivating Maghnite-H⁺. The proton of the carbenium ion which is located at the end of the chain returns to the Maghnite, achieving the ionic balance of the Maghnite- H^+ and also allowing its reuse.

Conclusions

The efectiveness of the technique proposed in this study has been successfully demonstrated, in order to synthesize a novel soluble conductive polymer poly (PAT). It was based on the copolymerization of thiophene with PA. PA is a monomer

Fig. 12 Reaction mechanism of the synthesis of PAT using Maghnite-H⁺ as a catalyst

synthesized from phenyl and caprolactam, the presence of its corresponding rings in the backbone gave the poly (PAT) its soluble appearance. The attachment between the phenyl and thiophene rings created alternating polymer chains of single and double bonds. The reaction catalysis method was heterogeneous and green, using Maghnite-H $^+$ as activated clay. Maghnite-H $^+$ was activated by an acidic treatment of raw-Maghnite. The reaction yield was 91% . The IR, ¹H NMR and ¹³NMR analyzes identifed in detail the chemical structure of poly (PAT), showing the linearity of the polymer chains. The attachment between the two monomers was in the meta position atom of the phenyl molecule, and in the 1 and 4 position atoms of the thiophene molecule. The study by TGA and DSC analyses has shown that poly (PAT) is thermally stable below the temperature of 200 °C. UV spectroscopy has shown that poly (PAT) has a narrow bandgap of 1.12 eV, which makes it a π -conjugated semiconducting material. Electrical properties were also evaluated; the electrical conductivity increased with increasing frequency, following polaron-hopping mechanism assisted by frequency. The relaxation time was about 1953 µs. In other words, when compared with many other common conductive polymers, the high intrinsic properties of poly (PAT) would be a signifcant advantage in view of realizing solar cells or field-effect transistors.

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