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Anomalous behaviour of the electrical properties for PVA/ TiO2 nanocomposite polymeric flms

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Abstract

In the present study, the fabricated nanocomposites were prepared by the casting technique by adding titanium dioxide $(TiO₂)$ to polyvinyl alcohol (PVA) with different weight percentages (0.037, 0.185, 0.370, 1.852, 3.704 and 37.037 wt%). SEM micrographs declared that PVA polymer matrix contains a distributed set of cluster of TiO₂ nanoparticles. AC electrical characteristics and dielectric relaxation of nanocomposite polymeric flms in the frequency range 100 kHz–1 MHz were presented as temperature changes from 303 to 413 K. The results indicated that the dielectric performance and electrical properties of the nanocomposites difer anomalously with increasing $TiO₂$ nanoparticles content. It was demonstrated that there is a stronger temperature-reliance of the dielectric properties of the 1.852 wt% of TiO₂/PVA nanocomposite polymeric film. Moreover, there is an interfacial polarization efect, which obeys Arrhenius behaviour that increased nonlinearly with increasing temperature at a fxed frequency. The electrical relaxation dynamics containing dielectric was examined. The dielectric properties were investigated through the electric modulus.

Keywords PVA–TiO₂ nanocomposites \cdot SEM morphology \cdot Dielectric properties \cdot AC electrical conductivity

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Introduction

Over the recent years, polymeric nanocomposites based on ceramic metal oxide have been shown great interest among the scientifc community because of their outstanding achievement in electronic properties [[1,](#page-13-0) [2\]](#page-13-1) such as batteries, solar cells, sensors, supercapacitors and fuel cells. Ceramic materials are generally brittle and very hard to process [\[3](#page-13-2)]. Otherwise, polymer materials at low temperatures are fexible, easy to process and display very high dielectric breakdown [\[4](#page-13-3)]. Electrical properties such as conductivity, dielectric constant and dielectric loss can be improved by mixing polymers with ceramic nanomaterials in the composite structure.

Titanium dioxide (TiO₂) is a transition metal oxide with three distinct crystallization structures such as rutile, anatase and brookite that can be used in many studies because of its excellent optical and electronic properties [\[5](#page-13-4), [6\]](#page-13-5). In general, TiO₂ nanoparticles have been preferred for the photocatalytic and device applications because it has many modifed physical and chemical properties such as thermal stability, non-toxic nature, and best catalytic property with wide band gap $[7–10]$ $[7–10]$ $[7–10]$. Recent attention has concentrated on poly (vinyl alcohol) (PVA) owing to its individual chemical and physical properties. PVA is a semi-crystalline polymer, which is responsible for most physical properties of PVA from the existence of the OH group, which leads to the multiplicity of industrial applications. When PVA polymer is mixed with $TiO₂$, it may interact in an amorphous part or in the crystalline part of the polymer causing a change in its electrical, structural and optical properties.

Therefore, all the information about the infuence of the additives on a certain polymer helps to characterize for a given application $[11]$ $[11]$ $[11]$. The formation of TiO₂ cluster in the PVA matrix is due to the formation of covalent bonds (T_i-O-C) among PVA and TiO₂ nanoparticles $[12]$ $[12]$ $[12]$. Also, the structural adjustment of the PVA matrix is resulting from the formation of Ti–OH bond between $TiO₂$ nanoparticles with the OH groups presented in PVA polymer [[13](#page-14-0), [14](#page-14-1)]. In recent years, intensive work has been carried out on optical, thermal, mechanical and structural characteristics of $PVA/TiO₂$ nanocomposites [\[15\]](#page-14-2), but the studies carried out to study its dielectric properties are very few.

The current work is focused on the preparation of $PVA/TiO₂$ polymeric nanocomposite flms using the method of solution casting. The microstructural study of PVA/TiO₂ nanocomposites was carried out by SEM. The studies of dielectric properties, electric modulus and AC electrical conductivity in a wide range of temperature and frequency have been recorded for describing the molecular motion and dielectric relaxation studies of the nanocomposites.

Fig. 1 SEM micrographs for the surface of $PVA/TiO₂$ nanocomposites

Experimental procedures

Preparation of PVA/TiO₂ polymeric nanocomposite films

PVA (from LOBA Co., the molecular weight of PVA is 77,000 g/mol) and $TiO₂$ (21 nm, Sigma-Aldrich Co.) were utilized without refnement. The concentration of PVA is 45 g/L in aqueous solution. The appropriate amount of PVA in aqueous solution was left on the magnetic stirrer for 48 h at 60 \degree C to have a fully dissolved PVA as a host matrix. Different percentages of $TiO₂$ of pure PVA samples and PVA nanocomposites are as follows: 0.037, 0.185, 0.370, 1.852, 3.704 and 37.037 wt%. Sixty millilitres of PVA aqueous solution was used to prepare all the specimens under investigation. After that, the solution was poured on glass Petri dishes, then left for five days in a drying furnace at 40 \degree C to evaporate water residuals. Pure PVA film was obtained from the previous solution. For $TiO₂/PVA$ polymeric nanocomposite samples; the required quantity of $TiO₂$ to PVA aqueous solution with respect to the following percentages 0.037, 0.185, 0.370, 1.852, 3.704 and 37.037 wt% was introduced. Ultrasonic homogenizer system was operated at 110 watts for a period of 5 min to obtain a good and homogeneous dispersion of TiO₂ nanoparticles in PVA. TiO₂–PVA was poured inside a glassy Petri dishes, then left in drying furnace as mentioned before. The samples were cut in size 2×2 cm², which makes them compatible with the required characterizations and measurements. Finally, the as-prepared samples were left in desiccators.

Devices and measurements

The morphologies of the specimens were done using a *Quanta 250 FEG* (*Field Emission Gun*) microscope on the surface of PVA/TiO₂ nanocomposites.

Programmable automatic LCR meter (Model Fluke PM6306) was used to measure the dielectric properties (real and imaginary parts) of dielectric permittivity (*ε*′, *ε*["]) dissipation factor (tan *δ*) and AC electrical conductivity (σ _{ac}), measured as a function of frequency (100 Hz to 1 MHz) and in the temperature range from 303 to 413 K.

Results and discussion

Structure characterization of PVA/TiO₂ polymeric nanocomposite films

The morphology of PVA, TiO₂ and PVA/TiO₂ nanocomposites with different TiO₂ contents is shown in Fig. [1](#page-2-0). The SEM image of pure PVA shows that the PVA flm surface is smooth with some few pores (Fig. [1a](#page-2-0)). The SEM image of $TiO₂$ shows a homogeneous distribution with spherical particles (Fig. [1b](#page-2-0)), where the average diameter is about 20 nm $[16, 17]$ $[16, 17]$ $[16, 17]$ $[16, 17]$ $[16, 17]$. The white portions display TiO₂ phase in Fig. [1c](#page-2-0)–h. From SEM micrographs, it can be noticed that the morphology of the

Fig. 3 Influence of TiO₂-concentration on ϵ' values of PVA–TiO₂ nanocomposite polymeric films at various frequencies and at 303 K

nanocomposites differs as the content of $TiO₂$ nanoparticles changes which result from cluster formation within the composites. As the concentration of $TiO₂$ increases, the cluster size increases due to the existence of hydroxyl group in the structure of nanocomposites which is the essential factor for the bonding of $TiO₂$ nanoparticles to the polymer leading to stability of the nanocomposites. As shown in Fig. [1](#page-2-0)g and h, when $TiO₂$ nanoparticles increase, the large cluster is formed, and these afect the properties of the flms. Thus, through morphological studies, we can infer that TiO₂ nanoparticles are well dispersed, forming clusters within the PVA matrix and no phase separation occurs [\[18](#page-14-5)].

Dielectric spectra of PVA/TiO₂ polymeric nanocomposite films

The dielectric properties (dielectric constant *ε*′, dielectric loss *ε″* and the loss tangent tan $\delta = \varepsilon''/\varepsilon'$, respectively, at 303 K for PVA–TiO₂ nanocomposites with frequency, are illustrated in Fig. [2](#page-3-0). It is known that the *ε′* value is the energy storage ability, which can be considered as a gauge of the polarization of prepared dielectric material. Therefore, it is noted that the values of *ε*′ are decreased monotonically with increasing frequency and reach a constant value at higher frequencies (Fig. [2](#page-3-0)a). In pure PVA, bipolar polarization returns to the bipolarity-arrangement of hydroxyl groups $[OH]$ that are connected to the chain but in $PVA-TiO₂$ nanocomposite polymeric flms; the resultant polarization is due to the alignment of the ion–dipolar (Ti–OH) interfacial charges that formed complex electric dipoles which, at low frequencies, obey the time-changing electric feld; however, they can no longer obey the varying feld as the frequency increases, thus reducing the values of *ε′* [\[8](#page-13-10), [9\]](#page-13-11). The frequency dependence ε' values of the 37.037 wt% of TiO₂/PVA nanocomposites got to be relatively low (< 15) .

Fig. 4 Plot of electric modulus of **a** real part *M′* and **b** imaginary part *M*″, as a function of frequency for $PVA-TiO₂$ nanocomposite polymeric films at 303 K

Figure [3](#page-5-0) shows the change in dielectric constant, *ε′* of the nanocomposites as a function of $TiO₂$ content at fixed frequencies. These results indicate that the dipolar arrangement of the hydroxyl group in the PVA and other polarization in the nanocomposites changed anomalously upon increasing the $TiO₂$ nanoparticles content. With regard to the nanocomposites, sample no. 4 (PVA–1.852 wt% TiO₂ nanocomposite polymeric flm) has the highest value of the dielectric constant and there are many studies displaying that the dielectric performance changes anomalously and diminishes upon increasing the doping content of inorganic nanofllers [[19\]](#page-14-6). It is noted that the value of *ε*′ decreases at high frequency (1 MHz) for all samples, and this enhances their expediency as low-value nano-dielectric composites. It is clear

Fig. 5 AC electrical conductivity—frequency dependence for the PVA–TiO₂ nanocomposite polymeric films at 303 K

that the dielectric performance of $PVA/TiO₂$ nanocomposites did not possess any apparent association with their structural characteristics. This means that the crystalline polymer (PVA) bipolar structures varied irregularly with $TiO₂$ nanoparticles content. To realize the behaviour of dielectric materials, the characterization of *ε*″ and tan δ is desired. The nanocomposites had dielectric relaxation peak in ε'' and tan δ at low frequency which is attributed to its structural dynamics as shown in Fig. [2](#page-3-0)b and c.

In addition, PVA–37.037 wt% TiO₂ nanocomposite polymeric film can be used as high-quality dielectric material because it has the lowest values of *ε*″ and tan *δ*. It is also clear from Fig. [2b](#page-3-0) and c that 1.852 wt\% of TiO₂/PVA nanocomposite polymeric film has the highest relaxation peak in ε ^{*r*} and tan δ spectra, confirming our earlier discussion of the irregular change in ε'' and tan δ spectra with the structural dynamics in the nanocomposite upon increasing $TiO₂$ content in PVA matrix. MWS relaxation was related to the infuence of the blocked charge at the interfaces between PVA polymer matrix and TiO₂ nanoparticles, while the α -relaxation demonstrated the segmental movement of polymer chains (local chain dynamics) [[12\]](#page-13-9).

The electric modulus spectra of PVA/TiO₂ polymeric nanocomposite films

The electric modulus M^* is calculated as:

$$
M^* = M' + iM'' = \frac{1}{\varepsilon^*(\omega)} = \frac{\varepsilon' + i\varepsilon''}{\varepsilon'^2 + \varepsilon''^2},\tag{1}
$$

where M' is the real part and M' is the imaginary part. Figure [4](#page-6-0) shows the frequency dependence of real *M*′ and imaginary *M*′ parts of the electric modulus for

Fig. 6 The plot of **a** real part ε' , **b** loss part ε'' and **c** tan δ for PVA–1.852 wt% TiO₂ nanocomposite polymeric flms versus log *f* at diferent temperatures

Fig. 7 Temperature (*T*) versus ε' values for PVA–1.852 wt% TiO₂ nanocomposite polymeric films at various frequencies

 $PVA/TiO₂$ nanocomposites at 303 K. These modulus spectra negate the contribution of diferent undesirable efects containing electrode polarization (EP) deliver a bulk response of the material. It is clear from Fig. [4a](#page-6-0) that the relationship between *M*^{$'$ and frequency has shoulders at intermediate and high frequencies, while there} are sharp relaxation peaks in *M″* spectra (Fig. [4b](#page-6-0)); these coincided with the PVA main-chain dynamics [[20\]](#page-14-7). By comparing our results with the previous studies on the *M″* spectra, we detected that the relaxation process changed anomalously upon increased $TiO₂$ nanoparticles content in the nanocomposites, i.e. the interaction of PVA polymer and TiO₂ nanofiller changed irregularly with TiO₂ contents [[21\]](#page-14-8).

AC electrical conductivity of PVA/TiO₂ polymeric nanocomposite films

Figure [5](#page-7-0) displays the plots of AC electrical conductivity, $\sigma_{ac}(\omega)$ with ln ω for PVA–TiO₂ nanocomposite polymeric films at 303 K. The values of σ_{ac} increased linearly with increasing frequency which has diferent slope values. This is due to the semi-crystalline nature of PVA [[22\]](#page-14-9). The values of $\sigma_{ac}(\omega)$ have been changed anomalously with the concentration of fller and based on the nanofller contents.

Dielectric Performance of PVA/TiO₂ polymeric nanocomposite films **under the efect of temperature**

Figure [6](#page-8-0) displays frequency dependence *ε'*, *ε''* and tan δ for 1.852 wt% TiO₂/PVA nanocomposites at diferent temperatures. Dielectric constant *ε*′ values increased by increasing the temperature as shown in Fig. [6](#page-8-0)a, confrming the role of heat in facilitating the free orientation of polar parts in the PVA chain. The variation of *ε*′ as a function of the temperature at different frequencies for PVA–1.852 wt% $TiO₂$

Fig. 8 Variation of electric modulus of **a** real part *M′* and **b** imaginary part *M″* versus frequency for $PVA-1.852$ wt% TiO₂ nanocomposite polymeric films at different temperatures

nanocomposite polymeric flm is depicted in Fig. [7](#page-9-0). The values of *ε*′ increased nonlinearly by increasing the temperature, and this increase in *ε*′ became stronger when the frequency is low. This increase is due to interfacial polarization [[23,](#page-14-10) [24\]](#page-14-11). In addition, at high frequencies (100 kHz, 500 kHz and 1 MHz), the increase in ε values with temperature is almost the same which assures that the bipolar arrangement in the nanocomposite polymeric film stays the same. Figure [6b](#page-8-0) exhibits that the values of *ε*″ increased by increasing temperature; however, for frequencies more than 100 kHz, the values of *ε*″ varied slightly by increasing temperature, which assures their suitability as a fxed dielectric loss material. Figure [6c](#page-8-0) shows the existence of loss tangent peak in the low-frequency region shifted to a higher frequency by increasing temperature $[25]$ $[25]$. Moreover, the values of tan δ increased by increasing temperature. The dielectric losses are large at lower frequencies because of more

Fig. 9 Variation of relaxation time of PVA–1.852 wt% TiO₂ nanocomposite polymeric film as a function of temperatures

time relaxation, but at higher frequencies, the losses decreased by the smaller time relaxation, then the conductivity increases [\[26](#page-14-13)].

Figure [8](#page-10-0)a and b exhibits the variation of M' and M'' versus tan f at different tem-peratures for PVA–1.[8](#page-10-0)52 wt% TiO₂ nanocomposite polymer film. From Fig. 8a, it is obvious that *M*′ values decreased with increasing temperature at a constant frequency. It is observed that M' of 37.037 wt% of TiO₂/PVA was the highest among the tested nanocomposites for all frequency ranges and increased slowly with the increase in frequency, while the M' of the 1.852 wt% of TiO₂/PVA was the lowest among the tested nanocomposites. Moreover, the *M*″ spectra presented a relaxation peak where its intensity increases by increasing temperature with a clear shift to

Fig. 10 AC electrical conductivity—frequency dependence for the PVA–1.852 wt% TiO₂ nanocomposite polymeric flm at diferent temperatures

Fig. 11 *S* parameter versus temperature (*T*) for (0.037 wt%, 1.852 wt% and 37.037 wt%) PVA/TiO₂ nanocomposite polymeric flms

high frequency. This behaviour proposes that the relaxation is activated thermally, and charge carrier hopping occurs [\[27](#page-14-14)]. The values of the modulus τ_M were calculated, by the relation $\tau_M = 1/2\pi f_p$, where (f_p) is the frequency corresponding to the M'' peaks [\[13](#page-14-0)]. This τ_M was referred to the flexible chain segmental motion of the polymer chain. Evidently, the relaxation time decreases with increasing temperature as shown in Fig. 9 [[28\]](#page-14-15).

The variation of AC conductivity for 1.852 wt% of TiO₂/PVA nanocomposite polymeric flm as a function of frequency within the temperature range of 303 and 413 K is shown in Fig. [10](#page-11-1). An increase in conductivity with temperature is noticeable due to the increase in both the dissociation of salts and fexible chain segmental motion. TiO₂ nanoparticles form a new kinetic path across polymer–ceramic boundary leading to increasing in conductivity [\[18](#page-14-5)]

To know the conduction mechanism, the frequency dependence of $\sigma(\omega)$ can be expressed as

$$
\sigma(\text{ac}) = A\omega^s \tag{2}
$$

where *A* is a constant and s is frequency exponent that is between 0 and 1. The value of *s* was calculated at each temperature from the slope of ln σ_{ac} against ln ω . *s* value was calculated from the most linear part of the curve. It was found that its values are between $(0.1 < s < 0.9)$ for all nanocomposites as shown in Fig. [11.](#page-12-0) As noted, the *s* values decreased by increasing temperatures, indicating that the correlated barrier hopping model (CBH) is the conduction mechanism [[13\]](#page-14-0). Polymer chain fexibility is due to changes in the chemical composition of repeated polymer units owing to the arrangement of hydrogen bonds with hydroxyl groups during the polymerization phase, thereby increasing electrical conductivity [[26,](#page-14-13) [29\]](#page-14-16).

Conclusion

In this work, we have synthesized $PVA/TiO₂$ by solution casting and characterization. SEM micrograph confirms that cluster of $TiO₂$ was fully dispersed in the polymer matrix. The effect of $TiO₂$ on the dielectric and electrical properties of PVA nanocomposites was studied. Dielectric constant, dielectric loss, tan *δ*, electric modulus and AC conductivity were changed anomalously by increasing $TiO₂$ content of the nanocomposites. The diference in dielectric constant with frequency displays the existence of dipole polarization processes. Loss tangent and real electric modulus peaks, which exhibit a characteristic frequency, propose the existence of relaxation dipoles in all the nanocomposites. The values of relaxation time decrease with increasing temperature; the AC conductivity increased with increasing frequency and temperature in all nanocomposites. The behaviour of the power factor interpreted the CBH model. Among all nanocomposites, 1.852 wt% of TiO₂/PVA displayed dielectric properties values as well as high conductivity which in turn appeared to be usable in future studies.

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