

# The effect of pressure on open-framework silicates: elastic behaviour and crystal–fluid interaction

G. D. Gatta<sup>1,2</sup>  · P. Lotti<sup>1</sup> · G. Tabacchi<sup>3</sup>

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**Abstract** The elastic behaviour and the structural evolution of microporous materials compressed hydrostatically in a pressure-transmitting fluid are drastically affected by the potential crystal–fluid interaction, with a penetration of new molecules through the zeolitic cavities in response to applied pressure. In this manuscript, the principal mechanisms that govern the *P*-behaviour of zeolites with and without crystal–fluid interaction are described, on the basis of previous experimental findings and computational modelling studies. When no crystal–fluid interaction occurs, the effects of pressure are mainly accommodated by tilting of (quasi-rigid) tetrahedra around O atoms that behave as hinges. Tilting of tetrahedra is the dominant mechanism at low-mid *P*-regime, whereas distortion and compression of tetrahedra represent the mechanisms which usually dominate the mid-high *P* regime. One of the most common deformation mechanisms in zeolitic framework is the increase of channels ellipticity. The deformation mechanisms are dictated by the topological configuration of the tetrahedral framework; however, the compressibility of the cavities is controlled by the nature and bonding configuration of the ionic and molecular content, resulting in different unit-cell volume compressibility in isotypic structures. The experimental results pertaining to

compression in “penetrating” fluids, and thus with crystal–fluid interaction, showed that not all the zeolites experience a *P*-induced intrusion of new monoatomic species or molecules from the *P*-transmitting fluids. For example, zeolites with well-stuffed channels at room conditions (e.g. natural zeolites) tend to hinder the penetration of new species through the zeolitic cavities. Several variables govern the sorption phenomena at high pressure, among those: the “free diameters” of the framework cavities, the chemical nature and the configuration of the extra-framework population, the partial pressure of the penetrating molecule in the fluid (if mixed with other non-penetrating molecules), the rate of *P*-increase, the surface/volume ratio of the crystallites under investigations and the temperature at which the experiment is conducted. An overview of the intrusion phenomena of monoatomic species (e.g. He, Ar, Kr), small (e.g. H<sub>2</sub>O, CO<sub>2</sub>) and complex molecules, along with the *P*-induced polymerization phenomena (e.g. C<sub>2</sub>H<sub>2</sub>, C<sub>2</sub>H<sub>4</sub>, C<sub>2</sub>H<sub>6</sub>O, C<sub>2</sub>H<sub>6</sub>O<sub>2</sub>, BNH<sub>6</sub>, electrolytic MgCl<sub>2</sub>·21H<sub>2</sub>O solution) is provided, with a discussion of potential technological and geological implications of these experimental findings.

**Keywords** Zeolites · Porous materials · High pressure · Compressibility · Sorption phenomena · Crystal–fluid interaction

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✉ G. D. Gatta  
diego.gatta@unimi.it

- <sup>1</sup> Dipartimento di Scienze della Terra, Università degli Studi di Milano, Via Botticelli 23, 20133 Milan, Italy
- <sup>2</sup> CNR-Istituto di Cristallografia, Sede di Bari, Via G. Amendola 122/o, Bari, Italy
- <sup>3</sup> Dipartimento di Scienza e Alta Tecnologia, Università degli Studi dell’Insubria, Via Valleggio 9, 22100 Como, Italy

## Introduction

“Microporous materials” are a class of compounds characterized by open-structures with cavities smaller than 20 Å in diameter. Cavities take the form of channels or cages. Materials with pores larger than 20 Å are the so-called “mesoporous materials”. Zeolites are the most common (in nature) and most used (in industrial processes) microporous

materials. The structure of natural and synthetic zeolites is usually built up by a framework of  $\text{SiO}_4$ – $\text{AlO}_4$ – $\text{PO}_4$ -tetrahedra. The extra-framework population consists of polar molecules (in particular  $\text{H}_2\text{O}$ ) and monovalent or divalent cations, which are commonly exchangeable. In hydrated zeolites, dehydration occurs at temperatures lower than 400 °C and is a usually reversible (and spontaneous) process. The tetrahedral framework may be interrupted by (OH, F) groups, which occupy tetrahedral apexes that are not shared with adjacent tetrahedra (Coombs et al. 1997). The general formula for common Si/Al zeolites is as follows:  $\text{M}_x^+\text{L}_y^{++}[\text{Al}_{(x+2y)}\text{Si}_{n-(x+2y)}\text{O}_{2n}] \cdot m\text{H}_2\text{O}$  (where  $\text{M}^+$  represents monovalent cations;  $\text{L}^{++}$  represents divalent cations; usually  $m < n$ ).

The microporous nature of zeolites governs the four main properties of this class of materials: the selective (and spontaneous) cation-exchange capacity, the catalytic activity (mainly promoted by Brønsted acid sites), the  $T$ -induced reversible hydration/dehydration processes and the most-recently discovered  $P$ -induced sorption of molecules. These properties have made natural or synthetic zeolites an object of attention for a long series of applications, spanning from soil enhancement (e.g. zeolites are used as slow release fertilizers, in areas where environmental issues are of concern), environmental remediation, water treatment, animal feeding, biomedical and veterinary applications (e.g. drug delivery systems), cements and concretes production, gas separation, catalysis in the petroleum industry (e.g. almost all the world's gasoline is produced using zeolites) and nuclear-waste processing (e.g. Komarneni 1985; Mumpton 1999; Kalló 2001; Maxwell and Stork 2001; Ming and Allen 2001; Bish et al. 2003; Ackley et al. 2003; Colella 2011; Gatta et al. 2016a). In this light, zeolites are nowadays considered as an important bulk commodity: the world production of natural zeolites in 2016 was about 2.8 million of tons (price: 100–230 dollar per ton) and the consumption of synthetic zeolites was approximately 1.6 million of tons (U.S. Geological Survey 2017). World reserves of natural zeolites have never been estimated. However, only in the U.S.A., resources are expected to approach 10 trillion tons for zeolite-rich deposits (U.S. Geological Survey 2017).

The peculiar behaviour of zeolites at non-ambient conditions, e.g. at high temperature (HT) or at high pressure (HP), led to a large number of in situ experiments over the last 60 years. More specifically, many experiments were devoted to the behaviour of zeolites in response to applied temperature: the mechanisms of  $T$ -induced dehydration, cation migration and rearrangement of extra-framework population have been investigated extensively, mainly by in situ single-crystal or powder X-ray/neutron-diffraction (e.g. Bish and Carey 2001; Pabalan and Bertetti 2001; Cruciani 2006 and references therein). In situ experiments on zeolites at high pressure have been done only in the last

15–20 years, mainly for technical difficulties, allowing the description of: (1) the compressional behaviour (and phase stability) in response to the applied pressure, along with the  $P$ -induced deformation mechanisms at the atomic scale (e.g. Gatta 2008, 2010a, b and references therein), (2)  $P$ -induced penetration of new molecules and its corresponding volume expansion (e.g. Lee et al. 2002a, b), (3)  $P$ -induced variation of the ionic conductivity of zeolites (e.g. Secco and Huang 1999; Rutter et al. 2000) and (4)  $P$ -induced amorphization processes (e.g. Gillet et al. 1996; Huang and Havenga 2001; Rutter et al. 2001; Greaves et al. 2003; Gulín-González and Suffritti 2004; Goryainov 2005).

The HP-behaviour of a zeolite compressed in a fluid is drastically dependent on the potential crystal–fluid interaction. Gatta (2008, 2010a) and Gatta and Lee (2014) described the relation between compressibility and microporosity, the framework flexibility (through deformation mechanisms at the atomistic level), and the different role played by framework (i.e. Si/Al-ordering, different cross-linking of the “building block unit”) and extra-framework configuration (i.e. nature of cations and absorbed molecules, ionic valence, ionic radii, coordination number) on the behaviour of zeolites when no crystal–fluid interaction occurs. The aim of this work is a comparative analysis, on the basis of previously published data, of the different behaviour of zeolites (natural or synthetic) when compressed in “penetrating” and in “non-penetrating”  $P$ -transmitting fluids (sensu Gatta 2008). More specifically, the effects of crystal–fluid interaction on the elastic behaviour and on the structure rearrangement of zeolites will be described, along with their potential geological and technological implications.

## Experimental methods

The majority of the in situ high-pressure experiments on zeolites have been done by single-crystal and powder X-ray diffraction, as well as IR/Raman spectroscopy, using the so-called Merrill–Basset-type diamond anvil cell (DAC) (Merrill and Bassett 1974; Miletich et al. 2000); only a few experiments have been performed by in situ neutron powder diffraction, using the large-volume Paris–Edinburgh press (i.e. Besson et al. 1992; Colligan et al. 2005; Seryotkin et al. 2005).

HP-experiments with a DAC are usually conducted with the sample compressed hydrostatically in a  $P$ -transmitting fluid.  $P$ -transmitting media can be stable in the liquid or solid state at ambient conditions, or can be gaseous at ambient conditions and then liquefied at low- $T$  before the loading of the DAC, or loaded into the  $P$ -chamber at high- $P$  by means of a gas-loading device. The chemical nature and the behaviour of the  $P$ -transmitting fluid can play an important

role in the HP-experiments on zeolites (Gatta 2008, 2010a; Angel et al. 2007; Gatta and Lee 2014). *P*-fluids can interact or not with the sample. Beside the potential crystal–fluid interaction, which will be extensively described below, it is highly desirable to ensure that the stress applied to the sample is homogeneous, i.e. without any differential stress or induced shear strain (Miletich et al. 2000; Angel et al. 2007). A hydrostatic medium cannot support shear stresses, simply because it has no shear strength. Non-hydrostatic stresses generate inhomogeneous strain in the crystal and, as a consequence, (sample) diffraction peaks broadening with a reduction of the signal-to-noise ratio. In addition, non-hydrostatic conditions can modify the compressional patterns of a given material (deduced on the basis of the unit-cell parameters variation with *P*) and can also suppress or promote phase transitions, including the promotion of *P*-induced amorphization phenomena (e.g. Decker et al. 1979; Kenichi 1999; Haines et al. 2001; Machon et al. 2003; Resel et al. 2004). Zeolites are relatively soft materials, and the occurrence of deviatoric stress usually has a dramatic impact on their behaviour at high pressure.

For experiments on zeolites at high pressure in a DAC, *P*-calibration has mainly been done by the so-called ruby fluorescence method (i.e. detecting the shift in the  $R_1$  emission line of ruby chips included in the compression chamber; *P*-uncertainty:  $\pm 0.05$  GPa, Mao et al. 1986). A further method, especially used for single-crystal experiments, is based on the compressibility pattern of quartz (i.e. with quartz used as an internal standard in the compression chamber; *P*-uncertainty:  $< 0.01$  GPa, Angel et al. 1997). For the few experiments by in situ neutron diffraction with the large-volume press, lead as an internal standard (Colligan et al. 2005) or the calibration curve of the ILL hydraulic-press load vs. *P* (Seryotkin et al. 2005) have been used.

The behaviour of a given material at high pressure is usually described on the basis of its compression pattern (based on the evolution of the unit-cell parameters with *P*) and of its structural rearrangements at the atomic scale (by structure refinements based on the intensity data collected at high pressure). About zeolites, the compressional behaviour is usually well describable by in situ powder or single-crystal experiments. On the other hand, whereas the quality of the HP single-crystal data is usually good enough for structural refinements, the quality of powder data is often not sufficient, and this has led to an intensive use of computational modelling techniques—for example, “rigid unit mode” (RUM)/geometric approaches, classical force-field methodologies, ab initio structural optimizations and molecular dynamics simulations—as a complementary tool to unravel the structure evolution at the atomistic level, as described below. As a matter of fact, all the experiments in a DAC provided intensities, of the Bragg peaks, usually

affected by a series of phenomena. Among those, the most critical are: the beam attenuation due to the absorption of the DAC components, parasitic diffraction of the crystalline components of the DAC (especially diamond anvils, their support plates and metal gasket), shadowing effects due to the DAC components (especially gasket and DAC steel body) (e.g. Miletich et al. 2005). For zeolites, which are usually materials with constituents with poor X-ray diffraction properties, the absorption phenomena are likely the most impacting on the data quality. In addition, in powder experiments an additional phenomenon usually occurs: the preferred orientation of crystallites, as common effect of uniaxial loads.

## Computational modelling

The impressive evolution of high-performance computing (HPC) resources in the last decade, coupled with the concomitant development of theoretical chemistry, has fostered the application of computational techniques to a broad category of problems in zeolite science (Van Speybroeck et al. 2015). Theoretical tools are particularly precious in the study of events occurring at extreme conditions—e.g. high temperature or high pressure—i.e. in cases where obtaining experimental data at atomistic resolution or direct evidences of complex phenomena is often very difficult. In this contribution, we will focus on high-pressure investigations, underlining how modelling, in direct conjunction with experiment, can be used not only to aid structural determinations when refinement of crystallographic data is problematic, but also as a powerful predictive tool able to inspire the design and guide the fabrication of new zeolite materials through the use of high pressure. Nowadays, the scope of computational approaches in exploring pressure effects on open-framework silicates goes well beyond than that of providing the positions of the atoms in the unit cell starting from no empirically based information (Woodley and Catlow 2008). Nevertheless, structural elucidation performed “in silico” over the past two decades has fostered a substantial advance of our understanding about how these materials respond to compression, highlighting a key role of the guest species—molecules and cations—in guiding the deformation of the zeolite framework as a consequence of the applied pressure.

Computational approaches used in zeolite modelling may be categorized into three broad groups, based on the way in which the electronic structure and interatomic interactions of the system are represented and calculated: (1) “rigid unit mode” models and their extensions, namely template-based geometrical approaches; (2) classical force-field techniques—basically, Monte Carlo (MC) and molecular

dynamics (MD) approaches; (3) quantum mechanical methodologies (a.k.a. “ab initio” or “first principles”).

The first category accounts solely for the strongest interactions in the system—covalent bonding and steric exclusion—and is constituted by the “flexibility-centred” methods. As they are based on simplified physical models, not only they are computationally fast and convenient, but they can also help interpreting experimental data and/or theoretical results at higher levels of theory. Application of these methods can provide valuable physical insight into the investigated phenomena, which is especially important in the high-pressure studies of zeolites (e.g. Sartbaeva et al. 2006, 2008; Wells et al. 2011). Specifically, the “rigid unit mode” (RUM) model (Giddy et al. 1993; Hammonds et al. 1994, Dove et al. 2000) treats the zeolite tetrahedra as fundamental, rigid, interacting units and analyses the flexibility of the structure in reciprocal space. Harmonic constraints are applied to the vertices of the tetrahedra, to penalize the separation of two adjacent units. RUMs, in which the polyhedra move as rigid bodies, without undergoing any distortion of their internal geometry, appear at zero frequency and are particularly useful to identify pressure-induced phase transitions. The template-based geometric analysis approach (Wells et al. 2002; Sartbaeva et al. 2006; Wells and Sartbaeva 2012, 2015), instead, works in the real space and considers both the atomic positions, and a set of geometrical templates reproducing the bonding pattern of groups of atoms—tetrahedra, in the case of zeolites. It is a more flexible approach, in the sense that there are still harmonic constraints—which connect atoms to the vertices of the tetrahedral templates—but, unlike in RUM, tetrahedra are not forced by construction to match exactly the input geometry. Rather, the templates and the atomic positions are iteratively relaxed to minimize mutual mismatches and avoid overlap between non-bonded atoms. Using this method, distortions from idealized geometry of the polyhedral templates can be readily visualized and quantified. This approach, implemented in the GASP code and recently reviewed (Wells and Sartbaeva 2012, 2015), has been successfully used for framework materials (Sartbaeva and Wells 2012). In particular, geometric simulations enabled to define an inherently geometrical feature of the zeolite framework: the “flexibility window” (Sartbaeva et al. 2006; Wells et al. 2015), i.e. the interval of unit-cell parameters in which the zeolite tetrahedral units can maintain their ideal shape. This important concept and its usefulness in analysing pressure-induced transformations will be discussed in the following section, together with some applications to high-pressure studies of zeolites.

Methods belonging to the second family are rooted in classical statistical mechanics and the interatomic

interactions are modelled by effective potentials (Sanders et al. 1984; Demontis et al. 1987, 1988, 1990; De Boer et al. 1995). The parameters of the effective potentials are determined by fitting large sets of either experimental or higher level computational data (see, e.g. Gabrieli et al. 2014, 2016). In the study of zeolite materials, the most popular methods are Monte Carlo (MC) and molecular dynamics (MD) (Allen and Tildesley 1987; Demontis et al. 1990; Frenkel and Smit 2001). In the former one, the phase space of the system is sampled stochastically according to a given statistical ensemble, and physical quantities—such as energies, bond distances and cell parameters—are calculated by averaging over the sampled configurations. In MD, atoms are represented as classical particles obeying the Newton equations of motion, thus generating a trajectory; physical quantities are obtained by averaging over simulation time. Another widespread approach is grand-canonical MC, which allows to simulate systems with a variable number of particles. This technique was applied to the study of the pressure-induced H<sub>2</sub>O intrusion in hydrophobic silicalite-1 and LTA—as reported by Desbiens et al. (2005) and Coudert et al. (2009)—and also to the screening of zeolite structures for two important industrial processes: the purification of ethanol obtained from biomasses and isomerization of heavy alkanes in petroleum refining, both performed under moderate pressure conditions (Bai et al. 2015).

While MD can follow the time evolution of a system up to nanosecond scales, with MC it is possible to sample configurations not easily encountered along a standard trajectory. Nonetheless, both of them allow to simulate systems of larger size compared to quantum-mechanical calculations. They are normally applied to zeolites with large unit cells and to dynamic processes such as adsorption, intrusion and diffusion of extra-framework species (e.g. Smit and Maesen 2008; Combariza et al. 2013; Balestra et al. 2015; Gutiérrez-Sevillano et al. 2016; Viani et al. 2016). Although a few examples of first-principles MD applications are available—such as the site-to-site H<sub>2</sub>O diffusion in a partially dehydrated zeolite (Ceriani et al. 2004a)—classical MD still remains the method of choice for this kind of problems. To deal with temperature-induced or pressure-induced structural changes, special techniques are available—for example, constant-pressure molecular dynamics (MD) (Andersen 1980) or the Parrinello–Rahman approach (Parrinello and Rahman 1980, 1981), where the cell parameters are considered as additional dynamical variables, so that the simulation box is allowed to deform. However, very often the system must cross a significant energy barrier; in those cases, the phase transition becomes very unlikely in the accessible simulation time. To overcome this problem, free energy minimisation techniques were devised (Gale 1997); they were applied, e.g. to the monoclinic-to-orthorhombic phase transition in MFI zeolite at high temperature (Grau-Crespo et al.



2002). In another powerful approach, named metadynamics (Laio and Parrinello 2002; Martonák et al. 2003), penalty functions prevent the simulation from visiting again previously sampled configurations. Some examples of application to zeolites are the temperature-induced reconstructive phase transition from anhydrous Li-ABW to eucryptite (Ceriani et al. 2004b), and the simulation of pressure-induced amorphization in  $\beta$ -eucryptite at 3 GPa (Narayanan et al. 2013). In the latter case, simulations revealed that the amorphization mechanism, at the atomistic level, consisted in the tilting and distortion of tetrahedra centred at Al/Si sites, accompanied by changes in the Al coordination and disordering of Li cations.

In the third category of computational approaches, the electronic structure of the system is described quantum mechanically (“from first principles”) either via quantum chemistry methods—i.e. by solving the time-independent Schrödinger equation for the electronic wave function—or through density functional theory (DFT) approaches, where the energy is a unique functional of the electronic density. Since the exact form of the functional is unknown, approximated expressions are adopted (Parr and Yang 1989). In the simplest one, the local density approximation (LDA), the exchange correlation energy functional is calculated using the expression of a homogeneous gas of free electrons. To model zeolites, more accurate functional forms are used, which include dependency from the gradient of the density and are known as generalized gradient (GGA) approximations (e.g. Perdew 1986; Becke 1988; Lee et al. 1988; Perdew et al. 1996; Wu and Cohen 2006; Perdew et al. 2008). Hybrid functionals, which combine DFT with Hartree–Fock exchange (e.g. Becke 1993; Adamo and Barone 1999), can also be adopted; they are more accurate but more demanding than pure-DFT functionals. Also, an approximate treatment of dispersion effects—particularly crucial for modelling, e.g. the sorption of hydrocarbons in zeolites—may be introduced through the use of long-range corrections (e.g. Grimme 2006; Tkatchenko and Scheffler 2009; Grimme 2011).

Several functionals have been proposed over the years, and new ones are continuously developed. In this rapidly evolving scenario, benchmarking plays a crucial role to ensure that the choice of functional would be appropriate for the system under investigation. Benchmarking studies are generally designed to identify a functional (or a group of functionals) able to model a given set of physical quantities—e.g. cell parameters, bond distances and angles, vibrational and elastic properties—for a series of zeolite types by achieving a reasonable compromise between accuracy and cost. In this respect, dispersion-corrected GGA functionals perform particularly well, as it emerges from a series of recent benchmarks. Such detailed benchmark studies were performed both on hybrid functionals (Görtl and Hafner

2012; Görtl et al. 2012; Coudert 2013; Román-Román and Zicovich-Wilson 2015) and on GGA functionals using extensive databases of experimental data on aluminosilicate (Fischer 2015; Fischer et al. 2015; Bryukhanov et al. 2017) and neutral framework zeolites (Fischer et al. 2016; Fischer and Angel 2017).

First principles methodologies can properly describe the breaking and forming of chemical bonds occurring, e.g. in chemical reactions, or in reconstructive phase transitions, but imply a high computational overhead. Fortunately, the crystalline lattice of zeolites can be modelled using either periodic-DFT (e.g. Ballone et al. 2002; Otero Areán et al. 2007; Marx and Hutter 2009) or periodic ab initio methods (Pisani 1996, 1999; Dovesi et al. 2005; Larin et al. 2005; Demichelis et al. 2010), where a simulation box—characterized by a given atomic content, corresponding, e.g. to the crystallographic unit cell of the studied material—is replicated in the three dimensions forming an infinite lattice. Tensorial schemes for ab initio calculations of elastic properties (bulk modulus, shear modulus, Poisson ratio, etc.) at high pressure have been also implemented (e.g. Erba et al. 2017) and tested on garnets and other minerals (Erba et al. 2014a, b).

While standard (static) quantum chemical methods can accurately describe a limited number of atoms, classical approaches are more appropriate for an extended zeolite system. These methodologies can be combined giving rise to hybrid quantum mechanics/molecular mechanics (QM/MM) (see, e.g. Bludský et al. 2005; Morpurgo 2015); hybrid ab initio/DFT variants have also been proposed (see, e.g. Tuma and Sauer 2004, 2006; Piccini et al. 2016). Other approaches aimed at bridging the gap between first principles and classical descriptions are based on the partitioning of the total electron density among individual subsystems (Wesolowski and Warshel 1993; Tabacchi et al. 2005; Mærzke et al. 2009; De Silva and Wesolowski 2012), which maintains an accuracy level suitable for modelling, e.g. dye–zeolite hybrid composites (Zhou et al. 2013). There are, however, situations where both a quantum mechanical treatment of the electronic structure and a description of thermal effects via the atomic motion are needed. These requirements can be satisfied by the first principles molecular dynamics (FPMD) approach, originally proposed by Car and Parrinello (Car and Parrinello 1985; Remler and Madden 1990; Marx and Hutter 2009).

FPMD enables to study the time evolution of a system with first-principle accuracy because the forces on the atoms at each MD time step are obtained using a quantum mechanical description of the electrons. More commonly, the electronic structure is treated with DFT (Marx and Hutter 2009), even though wavefunction-based variants of this technique exist as well. The Car–Parrinello scheme for FPMD defines a fictitious dynamical system in which

the potential energy surface depends on both the nuclear and the electronic degrees of freedom (Car and Parrinello 1985). The electronic wave function coefficients are propagated in time as classical degrees of freedom, and their dynamics generates at each time step the correct adiabatic electronic configuration corresponding to the new ionic positions. Hence, if at the beginning of the simulation the electronic orbitals correspond to the ground state, they will follow the motion of nuclei adiabatically and remain in that state as the nuclear configuration evolves in time. However, the dynamical parameters (fictitious electronic mass and time step) have to be chosen so that the transfer of energy between ions and electrons is kept very small during the simulation (Remler and Madden 1990). This condition is easily satisfied in systems with a large energy gap, such as zeolites.

FPMD has been implemented using different types of basis sets (Lippert et al. 1997; VandeVondele et al. 2005; Marx and Hutter 2009), namely plane waves (PW), projector augmented waves (PAW), and localized functions (VandeVondele et al. 2005). Periodic boundary conditions are generally adopted in all cases. In applications to zeolites, this implies the capability of describing the full crystal from first principles, thus properly reproducing the flexibility properties of the framework—which are especially important in governing the response of zeolites to an applied pressure. Whereas the direct simulation of a pressure-induced phase transition would require variable cell parameters, in many other cases it is convenient to hold the cell parameters fixed at the values experimentally determined at such pressure conditions (where available), which are known to be very accurate. Importantly, no constraints—either symmetry restraints or “frozen” nuclear positions—are normally imposed to the atoms in the simulation cell, which are thus left free to move according to their own potential energy surface determined by the interatomic interactions.

FPMD simulations, due to their high computational cost, are limited to elapsed times of the order of tens of ps, which are too short to observe activated events such as, e.g. chemical reactions, or reconstructive phase transitions. Such processes can be considered rare events on the FPMD time scale. To address these problems, rare-event sampling techniques have been developed, such as the above-mentioned metadynamics (Laio and Parrinello 2002; Barducci et al. 2011). In the case of zeolites, the FPMD extension of metadynamics (Iannuzzi et al. 2003) has been mainly applied to study activated processes occurring at high temperature, e.g. the methanol-to-olefin process catalysed by H-SAPO-34 (De Wispelaere et al. 2015), and the fabrication of zeolite hybrid functional materials (Calzaferrri et al. 2003; Cao et al. 2016) through the intrusion of dye molecules inside the one-dimensional channels of zeolite L (Tabacchi et al. 2016).

Other popular statistical sampling methods are the so-called “bluemoon” ensemble (Carter et al. 1989), the “transition path sampling” (Dellago et al. 1998) or the “nudged elastic band” (Jónsson et al. 1998; Sheppard et al. 2012). In all cases, a basic knowledge of the behaviour of the system at equilibrium conditions—which may generally be obtained via exploratory simulations—is needed. Considering the rapid and continuous increase in computing power, it may be foreseen that these methods will become useful tools in the prediction of the high-pressure behaviour of zeolite materials.

### High-pressure behaviour without any crystal–fluid interaction

To describe the HP-behaviour of a zeolite without any interference of the *P*-fluid, the sample (polycrystalline or single crystal) is compressed in a “non-penetrating” *P*-transmitting medium (sensu Gatta 2008): a fluid made by molecules which cannot penetrate through the structure cavities in response to the applied pressure. Usually, the mix methanol:ethanol = 4:1, glycerol, isopropanol, perfluoroether, fluorinert, or various grades of silicone oils are used as nominally non-penetrating *P*-transmitting fluids (Angel et al. 2007; Gatta 2008; Klotz et al. 2009).

Simulation or structure refinements showed that open-framework materials accommodate the volume compression mainly by *tilting*, *distortion* and *contraction* of the primary building units: the tetrahedra. *Tilting* of tetrahedra occurs around the bridging oxygen atoms that act as “hinges”; as a consequence, this mechanism does not generate distortion of the tetrahedra, but changes the inter-tetrahedral T–O–T angles with *P*. *Distortion* of tetrahedra is reflected by changes in the intra-tetrahedral O–T–O angles, preserving the average T–O bond length. *Contraction* of tetrahedra is expressed by compression of the T–O bond distances. Tilting, distortion and contraction of tetrahedra act simultaneously at any pressure. However, tilting is an energetically less-costly mechanism if compared to distortion or contraction, and this corroborates the experimental evidence that tilting is the dominant mechanism at low-*P* regime, whereas distortion and then contraction becomes dominant at higher pressure, when tilting cannot accommodate the effect of pressure efficiently any further (Gatta 2010a; Gatta and Lee 2014). In other words, there is a hierarchy of the deformation mechanisms (Gatta 2010a), and this is independent of the nature of the cations in tetrahedral coordination (e.g. Si, Al, P, Be, B, ...). As a matter of fact, at least up to 3–5 GPa, tetrahedra behave as “rigid-units” at least to a first approximation, e.g. the estimated fictive bulk modulus (defined as  $K_{P_0, T_0} = -V_0(\partial P/\partial V)_{P_0, T_0} = 1/\beta_{P_0, T_0}$ , where  $\beta_{P_0, T_0}$  is the

volume compressibility coefficient at ambient conditions) of  $\text{SiO}_4$  is 580(24) GPa (Zhang et al. 1998). Significantly, the rigid-unit behaviour is fully supported by available atomistic data obtained from FPMD simulations, and may also hold at above the 5 GPa threshold. For example, the Si and Al tetrahedral units of the zeolite gismondine undergo volume contractions of only 0.05 and 0.08%, respectively, at 7.6 GPa (Betti et al. 2007). Moreover, the above-mentioned hierarchy of deformation mechanisms is further corroborated by recent lattice-dynamic calculations on a low-silica zeolite with LTA framework (Gulín-González et al. 2016).

Tilting of tetrahedra can be observed in all the *HP*-experiments on zeolites, whose structure refinements or simulations are available. A common effect of tilting is the deformation of the secondary building units (SBU) of the zeolitic frameworks, which are represented by “open forms” (e.g. 4-, 5-, 6-, 8-, 10- or 12-membered rings of tetrahedra) or “3D closed forms” (e.g. double six-membered rings,  $\text{T}_5\text{O}_{10}$  units of the “fibrous zeolites group”,  $\text{T}_{10}\text{O}_{20}$  units of the “heulandite group”; Baerlocher et al. 2007). Usually, tilting of tetrahedra produces a continuous rearrangement of the structure, reflected by a monotonic unit-cell volume compression. It is common, for example, to observe that the ellipticity of the zeolitic channels tends to increase monotonically with *P*, via tilting of the tetrahedra that confine the channels, without any “inversion” (Gatta and Lee 2006); an “inversion” in ellipticity at high-*P* usually leads to a phase transition. More complex is the description of the response, at the atomic scale, of zeolites with secondary building units made by 3D closed forms, which behave as “rigid block-units” at a first approximation: at high pressure, their structures respond first by rotation of the SBUs, followed by SBU compression (but keeping the tetrahedra undeformed), and finally by deformation or compression of the tetrahedra themselves. The first two mechanisms are substantially based on tilting and are very efficient to accommodate the effect of pressure at low-medium *P*-regime. Two examples in this respect are those of the “fibrous zeolites group” (NAT, THO, and EDI framework type; SBU:  $\text{T}_5\text{O}_{10}$  or 4 = 1 unit; Baerlocher et al. 2007) and of the “heulandite group” (HEU framework type; SBU:  $\text{T}_{10}\text{O}_{20}$  or 4-4 = 1 unit; Baerlocher et al. 2007). All fibrous zeolites studied at high-*P* (i.e. natrolite, scolecite, thomsonite, edingtonite) show a similar deformation mechanism in response to applied pressure: cooperative rotation (anti-rotation) of the SBU about the axis of the SBU-chains, which in turn gives rise to the compression of the eight-membered rings channels (parallel to the SBU chain axis), with an increase of their ellipticity (Gatta 2005; Gatta and Wells 2004; Gatta et al. 2016b). In fibrous zeolites, the estimated fictive bulk modulus of the SBU is approximately twice those of the zeolites (i.e. ~115 GPa for the SBU vs. ~43 GPa

for natrolite, ~55 GPa for scolecite, ~49 GPa for thomsonite and ~59 GPa for edingtonite) (Gatta 2005). The identical framework topology of natrolite and scolecite (i.e. NAT, Baerlocher et al. 2007) allowed also to speculate about the role played by the extra-framework population on the compressional behaviour of isotopic materials: the framework topology controls the main deformation mechanisms (in this case: cooperative anti-rotation of the SBU by tilting of tetrahedra), but the channel population (i.e. ionic radius of cations, valence, and their bonding configuration) controls the compressibility of the cavities, resulting in different compressibility of unit-cell volume. In addition, fibrous zeolites provide further experimental evidence: ordering of the Si/Al in the tetrahedral sites does not influence structure compressibility, which is virtually identical in orthorhombic (fully ordered Si/Al distribution) and tetragonal (highly disordered Si/Al distribution) edingtonite (Gatta et al. 2004a, b).

In heulandite, the corrugation (and thus the shortening) of the zig-zag chains of SBUs, parallel to [102], is the main deformation mechanism in response to applied pressure, which acts basically by tilting of tetrahedra. In addition, in this case, the estimated fictive bulk modulus of the SBU (i.e. ~63 GPa) is approx. twice that of the zeolite (i.e. ~28 GPa) (Comodi et al. 2001; Gatta et al. 2003). In general, the experimental findings on zeolites with closed-form SBUs show that the compressibility of the structure is the combined effect of soft channels and relatively stiff 3D SBUs compression.

Important quantitative insight into the tilting deformation mechanism may be obtained by investigating how the T–O–T angles evolve as a function of increasing pressure even in structures with “open” SBU. Interestingly, such angles were identified to be responsible for the pressure-induced framework modification up to 8–10 GPa on the basis of integrated experimental-computational data collected for a series of zeolites, e.g. bikitaite (BIK framework type, Ferro et al. 2002; Fois et al. 2002a; Baerlocher et al. 2007), yugawaralite (YUG framework type, Fois et al. 2005a, b; Baerlocher et al. 2007), and gismondine (GIS framework type, Betti et al. 2007; Baerlocher et al. 2007). For example, in gismondine, the average value of the T–O–T angle decreases significantly (from 142.8° to 137.8°) in passing to room pressure to 7.4 GPa (Betti et al. 2007). As a general trend, the decrease of the average T–O–T angle correlates with both applied pressure and volume contraction: namely, the higher the compression, the higher the variation of the angle. This analysis, therefore, brings further support to the intuitive argument that the framework responds to compression by modifying its most flexible element, that is, the junction between tetrahedral units.

A few representative examples can be considered to highlight that tilting can act as the main deformation effect

even when the saturation of one mechanism is achieved at low- $P$ . Gatta et al. (2005) and Gatta and Wells (2006) reported, by experiments and computational modelling, that the framework of levyne (LEV framework type, Baerlocher et al. 2007) reacts, under compression, by tilting of tetrahedra following two distinct deformation mechanisms: the first dominant at  $P < 1$  GPa, through the cooperative rotation of tetrahedra belonging to the double six-membered rings; the second at  $P > 1$  GPa, through compression of the four-membered ring (“joint-unit”) between the aforementioned six-membered rings, as the double six-membered rings have already reached a limit configuration at  $P \sim 1$  GPa. These two mechanisms can explain the anomalous elastic behaviour of levyne, with an increase of the  $c$ -axis length between 0 and 1 GPa followed by a monotonic compression at  $P > 1$  GPa. Up to 5 GPa, structure refinements and simulations show no distortion or compression of the tetrahedra, at a significant level, in levyne structure.

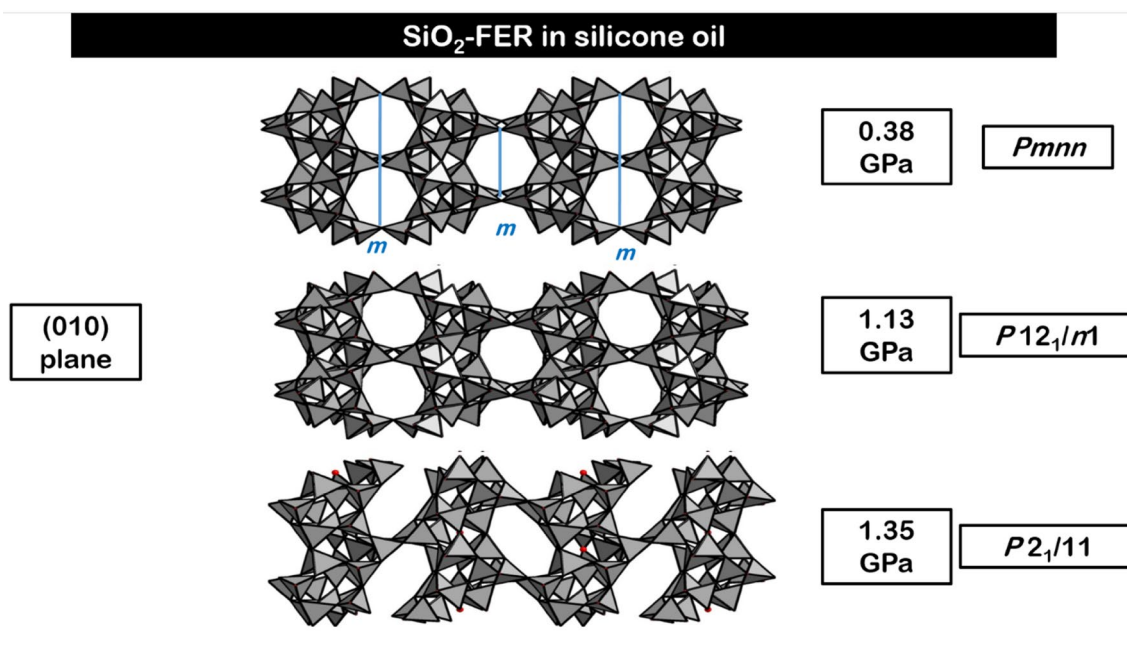
More recently, Comboni et al. (2017) reported the  $HP$ -behaviour of phillipsite (PHI framework type, Baerlocher et al. 2007) up to 9.4 GPa by an in situ single-crystal diffraction experiment. Despite using a hydrous  $P$ -transmitting fluid (i.e. methanol:ethanol:water = 16:3:1), no clear evidence of  $P$ -induced penetration of extra molecules through the zeolitic cavities was observed within the  $P$ -range investigated. However, two different compressional regimes occur. Between 0.0001 and 2.0 GPa, phillipsite behaves as an unusually stiff porous material: the refined bulk modulus is  $K_{P_0, T_0} = 89(8)$  GPa. Between 2.0 and 9.4 GPa, the material turns to be drastically softer and its bulk modulus decreases to  $K_{P_0, T_0} = 18.8(7)$  GPa. The structure refinements proved that, at  $P > 2$  GPa, a  $P$ -induced change in the configuration of the H<sub>2</sub>O molecules, coupled with a change of the tilting mechanisms of the framework tetrahedra, gives rise to a completely different compressional behaviour. Accordingly, the evolution of the monoclinic  $\beta$  angle with pressure shows two distinct trends in the two compressional regimes: with a negative slope between 0.0001 and 2.0 GPa, and a positive slope between 2.0 and 9.4 GPa. The tilting of the tetrahedra, in response to the hydrostatic compression, causes the deformation of the eight-membered ring of tetrahedra which confines the [010] channel and, in turn, the inversion of  $\beta$  vs.  $P$  trend. In this specific case, the saturation of a tilting mechanism and the role played by the extra-framework population concur to a change of the compressional behaviour, without any  $P$ -induced phase transition.

Levyne and phillipsite are two representative examples of zeolites which do not undergo any  $P$ -induced phase transition, despite they experience a change of deformation mechanisms, reflected by a change of the unit-cell volume compressional pattern. The list can be extended, for example, to gobbinsite (GIS framework type, Baerlocher

et al. 2007; Gatta et al. 2010, 2012; Gatta and Lotti 2011), cancrinite (CAN framework type, Baerlocher et al. 2007; Lotti et al. 2012) or nepheline (a feldspathoid, Gatta and Angel 2007). On the other hand, some zeolites with high “framework density” (FD, defined as the number of T atoms in a volume of 1000 Å<sup>3</sup>, Baerlocher et al. 2007) or hydrophobic zeolites (i.e. zeolites nominally without any extra-framework population) tend to react to applied pressure with phase transitions, which are usually displacive in character, and the high-pressure polymorphs can have distorted tetrahedra. A representative example, in this respect, is that of zeolites of the analcime group (i.e. analcime, leucite, wairakite and pollucite; ANA framework type, Baerlocher et al. 2007): all these zeolites experience a first-order phase transition from the high-symmetry low- $P$  polymorph (cubic analcime and pollucite, tetragonal leucite and monoclinic wairakite) to a triclinic high- $P$  polymorph, at relatively low- $P$  ( $\sim 1$  GPa for analcime,  $\sim 2.4$  GPa for leucite,  $\sim 2.5$  GPa for wairakite,  $\sim 0.7$  GPa for pollucite; Gatta et al. 2006, 2008, 2009a; Ori et al. 2008a). The phase transitions are fully reversible upon decompression. For analcime, in particular, structure refinements (based on single-crystal intensity data) revealed that the main deformation mechanisms of the  $HP$ -polymorph act through deformation of the four- and six-membered rings of tetrahedra by tilting of the polyhedra, along with their significant distortion. A spectacular  $HP$ -behaviour with two  $P$ -induced phase transitions in all-silica ferrierite (FER framework type, Baerlocher et al. 2007) was recently reported by Lotti et al. (2015a), by in situ single crystal and powder X-ray diffraction experiments, using penetrating and non-penetrating fluids. Using silicone oil, as a polymeric non-penetrating  $P$ -fluid, the experimental data show a remarkable flexibility of the ferrierite framework at high pressure with two displacive phase transitions, following the path  $Pmnn$ -to- $P12_1/n1$ -to- $P2_1/n11$ , the first at  $\sim 0.7$  GPa and the second at  $\sim 1.24$  GPa (Fig. 1). The transitions are fully reversible. As reported by Lotti et al. (2015a), the two monoclinic space groups do not share a group–subgroup relationship: the multistage phase transition requires an intermediate structure with  $P-1$  symmetry, as common subgroup of both  $P12_1/n1$  and  $P2_1/n11$ . The three polymorphs share a virtually identical bulk compressibility, but with a different anisotropic compressional pattern. Also in this case, the structure evolution in response to applied pressure is mainly governed by tilting, and the phase transitions are the effect of deformation mechanisms saturation, followed by the promotion of new ones in a structure configuration which is energetically more favourable.

Phase transitions can drive to “new materials” which are industrially important: a new zeolite ITQ-50 was obtained by compressing a synthetic all-silica zeolite ITQ-29 in a





**Fig. 1** Structure evolution of the all-silica ferrierite compressed in a non-penetrating fluid (see text for further details)

DAC with a non-penetrating *P*-transmitting fluid (Jordá et al. 2013). As from the crystallographic data it was possible to obtain only the cell parameters of the new material, the structure was obtained with the aid of classical force-field calculation. Remarkably, the pressure-synthesized microporous material showed better performances than the precursor zeolite in the propene/propane separation—an important industrial process.

Besides the modification induced to the framework, it is useful to highlight how the organization of the extra-framework species is modified by compression. For example, the natural zeolite bikitaite hosts in its one-dimensional channels hydrogen bonded chains of H<sub>2</sub>O molecules, which at room conditions display a behaviour typical of a solid phase of water at low dimensionality—hence, the name “one-dimensional ice” (Fois et al. 1999, 2001b)—and an impressive resistance towards dehydration despite not being hydrogen bonded to the framework (Ceriani et al. 2004a). Indeed, upon compression, the one-dimensional ice structure persisted up to 5 GPa, but new hydrogen bonds were formed between the H<sub>2</sub>O chains and the framework oxygen atoms (Ferro et al. 2002). Even more intriguingly, a similar behaviour was found when bikitaite was compressed in penetrating (water-containing) medium up to 4 GPa: the H<sub>2</sub>O chains were preserved up to the highest pressure with no intercalation of additional H<sub>2</sub>O molecules (Seryotkin 2016), highlighting thus the exceptional stability of this confined H<sub>2</sub>O nanostructure. Actually, also the synthetic zeolite Li-ABW hosts in its monodimensional channels the same peculiar

H<sub>2</sub>O wires (Fois et al. 2001a, b); in this case, however, the supramolecular H<sub>2</sub>O architecture is easily disrupted when moderate pressures are applied (Fois et al. 2008a). Theoretical calculations revealed that the reason why these two similar systems display such a different response to compression is the electric polarization of the zeolite framework, which is considerably higher in bikitaite compared to Li-ABW, thus providing a greater electrostatic stabilization to the H<sub>2</sub>O chain (Fois et al. 2001a, 2008a). Porous materials with monodimensional channel systems can host other interesting examples of confined H<sub>2</sub>O nanostructures—such as the H<sub>2</sub>O-triple helix in VPI-5 aluminophosphate (Fois et al. 2002b), or the pressure-induced H<sub>2</sub>O nanotube in zeolite LTL (Lee et al. 2007). The interest in the behaviour of one-dimensional channel materials—at both standard and high-*P* conditions—stems from the possibility of filling them, upon H<sub>2</sub>O evacuation, with guest species of suitable size, for example, drugs (Delle Piane et al. 2014) or luminescent molecules, giving rise to low-dimensionality systems endowed by technologically appealing properties (see, e.g. Calzaferri et al. 2003; Fois et al. 2005c; Manzano et al. 2013; Gigli et al. 2014; Cucinotta et al. 2014, Calzaferri 2017; Gartzia-Rivero et al. 2017).

The key question is: why some zeolites experience *P*-induced phase transitions (with no crystal–fluid interaction) and others not? In addition, which are the limit conditions above which an open-framework structure undergoes a phase transition? Is it possible to predict a transition pressure? The state of the art of the knowledge

does not allow to provide an unambiguous answer to these questions. However, the experimental findings about *P*-induced phase transitions in zeolites (with no crystal–fluid interaction) appear to be well predictable by the “flexibility window” theory. The “flexibility window” (Sartbaeva et al. 2006) is a pervasive property of zeolitic frameworks and it defines a range of densities over which the corner-sharing  $\text{TO}_4$  units, making up the framework, can in principle be made perfectly tetrahedral (i.e. a range of densities over which the tetrahedra retain their holosymmetric shape). This window is limited at high density by contacts between oxygen atoms on neighbouring  $\text{TO}_4$  units, and at low density by extension of the intra-tetrahedral T–O bonds (though not, in general, by linear T–O–T angles). Gatta et al. (2009b) reported that the ANA cubic framework, for example, displays a narrow flexibility window: the framework could be compressed by only 3% in volume before oxygen atoms came into contact. The geometric simulations on the ANA framework (using a tetrahedral geometry appropriate to the Si:Al ratio of natural cubic analcime) (Sartbaeva et al. 2006, 2008; Wells et al. 2011), with the experimental cubic unit-cell parameters measured at high pressure (Gatta et al. 2006), showed that the structures are perfectible over the whole observed range of cubic analcime, from ambient conditions up to around 1 GPa. The theoretical upper and lower limits of the flexibility window in cubic analcime were obtained by simulations using the unit-cell parameters outside the range observed in the experiment. The low-density edge of the window was found to be near the density observed for the analcime structure at ambient conditions (i.e. the ambient structure is close to maximal extension), whereas the high-density edge of the window on compression lies near the last observed cubic structure before the transition. Overall, the *P*-induced phase transition in analcime, observed at about 1 GPa, is well predictable on the basis of the flexibility window of this structure.

### High-pressure behaviour with crystal–fluid interaction

“Penetrating” *P*-transmitting media are a class of fluids which contain molecules able to penetrate, through selective sorption, into the zeolitic cavities in response to applied pressure. This phenomenon implies a drastic effect on the compressional pattern and on the *P*-induced deformation mechanisms at the atomic level. Elemental gaseous media at ambient conditions (e.g. helium, nitrogen, neon, argon, xenon, krypton) and small molecules (e.g.  $\text{H}_2\text{O}$ ,  $\text{CO}_2$ ) are potentially penetrating media. Even some larger molecules, usually used as *P*-transmitting

media (e.g. methanol, ethanol, ethylene, ethylene glycol, acetylene) could be considered as potentially penetrating media. Penetrability of external molecules (or atoms) at high pressure is governed by several variables, among those the most important are: the “free diameters” of the framework cavities, the chemical nature and the configuration of the extra-framework population, the partial pressure of the penetrating molecule in the fluid (if mixed with other non-penetrating molecules, e.g.  $P(\text{H}_2\text{O})$  in a mixture of alcohol– $\text{H}_2\text{O}$ ), the temperature at which the experiment is conducted. A given molecule can, therefore, act as a penetrating medium for some zeolites and as a non-penetrating medium for other zeolites.

The effects of the *P*-induced penetration depend on the nature of the penetrating atoms or molecules through the zeolitic cavities, and, for a comparative analysis, it is convenient to distinguish among monoatomic species, small molecules and larger molecules.

### *P*-induced penetration of monoatomic species

Lee et al. (2010) showed that, when compressed in liquid Ar, natural natrolite (of ideal composition  $\text{Na}_{16}\text{Al}_{16}\text{Si}_{24}\text{O}_{80}\cdot 16\text{H}_2\text{O}$ , NAT framework type, Baerlocher et al. 2007) can incorporate a significant amount of Ar under moderate pressure and temperature conditions (i.e. 60 °C for 10 h). The *P*-induced penetration of Ar gives rise to a spectacular *P*-induced expansion of the unit-cell volume (i.e. ~6.5% larger unit-cell volume than the starting zeolite) at  $P \sim 3.0\text{--}3.5$  GPa, with a new Ar-bearing form of natrolite described as  $\text{Na}_{16}\text{Al}_{16}\text{Si}_{24}\text{O}_{80}\cdot 16\text{H}_2\text{O}\cdot 6\text{Ar}$ . The expansion of the unit-cell volume is governed by the expansion of the eight-membered ring channels along [001], where the extra-framework population lies. The structure refinement of  $\text{Na}_{16}\text{Al}_{16}\text{Si}_{24}\text{O}_{80}\cdot 16\text{H}_2\text{O}\cdot 6\text{Ar}$  showed that  $\text{Na}^+$  remains six-coordinated, with the Ar–Na distances ranging between 2.90 and 3.22 Å and the Ar–O distances at ~3.24 Å. In other words, Ar interacts via short-range van der Waals forces. Natrolite served as an efficient host system for further experiments with monoatomic penetrating species: Xe and Kr. Seoung et al. (2014) reported how Ag-natrolite, ideally  $\text{Ag}_{16}\text{Al}_{16}\text{Si}_{24}\text{O}_{80}\cdot 16\text{H}_2\text{O}$ , adsorbs xenon into its eight-membered ring channels at 1.7 GPa and 250 °C, while  $\text{Ag}^+$  is reduced to metallic  $\text{Ag}^0$  and possibly oxidized to  $\text{Ag}^{2+}$ . The sorption of Xe gives rise to an expansion by 3.2% of the unit-cell volume, with a weak interaction of Xe and Ag (i.e. Ag–Xe ~3.1 Å). Surprisingly, the sorption of xenon is irreversible after pressure release and requires heat to desorb. Using liquid Kr as *P*-transmitting fluid, Ag-natrolite adsorbs Kr at 2.1 GPa and after annealing at 250 °C, with a partial reduction of  $\text{Ag}^+$  to metallic  $\text{Ag}^0$ . The penetration phenomenon leads to a moderate unit-cell volume expansion (only ~1.2%) and appears to be fully reversible under decompression.

In zeolites with larger cavities than those in natrolite, the *P*-induced penetration of monoatomic species does not imply a unit-cell volume expansion, but rather a different compressional behaviour if compared to that obtained in a non-penetrating fluid. For example, Niwa et al. (2013) reported how the synthetic zeolite Linde-type A (LTA framework type, Baerlocher et al. 2007) compressed in liquid He and Ar shows different compressional patterns, which are, in turn, different if compared to those reported for compression in liquid water (penetrating) or in silicone oil (non-penetrating polymeric fluid). Structure refinements were not performed, leaving open questions about the mechanisms at the atomic scale. However, the evolution of unit-cell volume vs. *P* is sufficient to prove that the *P*-induced penetration of He, Ar and H<sub>2</sub>O occurs, with a reduction of the compressibility, if compared to that with a non-penetrating fluid, by 50, 15 and 700%, respectively. The sorption of extra H<sub>2</sub>O molecules affects dramatically the compressibility of Linde-type A. The reduction of compressibility reflects the “pillar effect” that the new intruded atoms or molecules generate, making the zeolitic cavities less compressible (not collapsible).

### *P*-induced penetration of small molecules

Experiments devoted to the *P*-induced penetration of small molecules are probably the most common. The *P*-induced penetration phenomenon was first discovered using water, or hydrous mixtures, by Hazen (1983) and Hazen and Finger (1984), on the basis of the significantly different compressional pattern of a synthetic zeolite (i.e. Linde Na-A; LTA framework type, Baerlocher et al. 2007) in different fluids. Lee et al. (2002a, b) reported the very first structure refinements proving the *P*-induced sorption of extra-H<sub>2</sub>O molecules, from the *P*-transmitting fluid through the zeolitic cavities (the so-called *P*-induced “superhydration”), in a natrolite and its synthetic counterpart K<sub>16</sub>Ga<sub>16</sub>Si<sub>24</sub>O<sub>80</sub>·12H<sub>2</sub>O. Natrolite, ideally Na<sub>16</sub>Al<sub>16</sub>Si<sub>24</sub>O<sub>80</sub>·16H<sub>2</sub>O (NAT framework type, Baerlocher et al. 2007), transforms to Na<sub>16</sub>Al<sub>16</sub>Si<sub>24</sub>O<sub>80</sub>·32H<sub>2</sub>O at ~1.2 GPa (via the intermediate ordered-paranatrolite phase with 24 H<sub>2</sub>O molecules per formula unit, stable between 0.9 and 1.2 GPa, Lee et al. 2005), doubling the number of molecules p.f.u. of H<sub>2</sub>O with a consequent unit-cell volume expansion by ~2.5%, if compared to natrolite stable at ambient *P*. The extra H<sub>2</sub>O molecules intruded in the eight-membered ring channel are bonded to Na, which increases its coordination number. The *P*-induced superhydration effect is completely reversible in natrolite but not in its synthetic counterpart K<sub>16</sub>Ga<sub>16</sub>Si<sub>24</sub>O<sub>80</sub>·12H<sub>2</sub>O, suggesting that the chemical nature of extra-framework (and likely framework) population plays an important role in governing the penetration reversibility. The *P*-induced penetration of H<sub>2</sub>O in natrolite was described with different experimental methods (Colligan et al. 2005; Seryotkin et al.

2005), and it was observed in almost all the fibrous zeolites with spectacular volume expansion (e.g. natrolite, scolecite and thomsonite, e.g. Lee et al. 2002a, b, 2005; Colligan et al. 2005; Gatta 2005; Likhacheva et al. 2006, 2007; Seoung et al. 2013, 2015; Seryotkin et al. 2017), with the exception of edingtonite (EDI framework type, Baerlocher et al. 2007; Gatta et al. 2004a, b): the large Ba-polyhedron fills very efficiently the eight-membered ring channel along [001], which is in an already expanded configuration if compared to the other fibrous zeolites, hindering further penetration of H<sub>2</sub>O molecules.

The role of the cations in natrolite under superhydration conditions was recently explored theoretically by investigating the *P*-induced behaviour of Na-, Rb-, and Cs-natrolite (Kremleva et al. 2013) and of the K-substituted counterpart (Kremleva et al. 2014). With calculations based on density functional theory (DFT), these studies were able to reproduce approximately the critical pressure values at which the corresponding transformations were found to occur in experiments, providing, therefore, valuable insight into their microscopic details. Based on the modelling results, these authors also predicted the possible formation, at high-pressure conditions, of two isomers of superhydrated K-NAT, with either positive or negative chain rotation angles (Kremleva et al. 2014).

An additional *P*-induced expansion phenomenon was observed in laumontite, ideally Ca<sub>4</sub>Al<sub>8</sub>Si<sub>16</sub>O<sub>48</sub>·*n*H<sub>2</sub>O with *n* ≤ 18 (LAU framework type, Baerlocher et al. 2007). Using a partially dry sample with 12 H<sub>2</sub>O m.p.f.u., Lee et al. (2004) showed, using a hydrous *P*-transmitting fluid and a powder sample, that laumontite experiences a phase transition at *P* = 0.2(1) GPa, with a spectacular unit-cell volume increase of ~2.6%. The structure refinements proved that the expansion reflects the transition to a fully hydrated form (with 18 H<sub>2</sub>O m.p.f.u.), by selective sorption of extra H<sub>2</sub>O, which leads to the expansion of the eight-membered ring channels along [001]. Significantly, *P*-induced hydration in laumontite was first investigated by a computer modelling study performed with a classical force field by White et al. (2004). Such a study predicted the occurrence of full hydration at moderate pressures, which was then confirmed by the experiments, evidencing that, when pressure is applied, the structural stability of the laumontite-type framework increases because of the fully occupied H<sub>2</sub>O network.

Fibrous zeolites and (partially dehydrated) laumontite represent rare examples of *P*-induced insertion of H<sub>2</sub>O molecules with spectacular expansion of the unit-cell volume. Usually, the penetration of extra H<sub>2</sub>O molecules from the *P*-fluid is not accompanied by unit-cell volume expansion, as shown, e.g. for: the synthetic Linde-type A (ideally Na<sub>12</sub>Al<sub>12</sub>Si<sub>12</sub>O<sub>48</sub>·26 H<sub>2</sub>O, LTA framework type; Hazen 1983; Hazen and Finger 1984; Arletti et al. 2003; Likhacheva et al. 2009; Niwa et al. 2013); synthetic Li-, (Na,Cs)- and

Cd-RHO zeolites  $[(\text{Li}, \text{Na}, \text{Cs})_{12}(\text{Al}_{12}\text{Si}_{36}\text{O}_{96}) \cdot 44\text{H}_2\text{O}$ , RHO framework type; Lee et al. 2001]; synthetic all-silica zeolite Y (FAU framework type, Colligan et al. 2004); gismondine (ideally  $\text{Ca}_4\text{Al}_8\text{Si}_8\text{O}_{32} \cdot 16\text{H}_2\text{O}$ , GIS framework type; Ori et al. 2008b) and its synthetic K-gallosilicate counterpart (K-GaSi-GIS, Lee et al. 2008); boggsite (ideally  $\text{Ca}_8\text{Na}_3\text{Al}_{19}\text{Si}_{77}\text{O}_{192} \cdot 70\text{H}_2\text{O}$ , BOG framework type; Arletti et al. 2010); synthetic Na-ZSM-5 (i.e.  $(\text{Na}_{4.58}\text{K}_{0.02})(\text{Ca}_{0.18}\text{Mg}_{0.03}\text{Ba}_{0.01}\text{Fe}_{0.05}\text{Sr}_{0.01})(\text{Al}_{4.48}\text{Si}_{91.35})\text{O}_{192} \cdot 28.39\text{H}_2\text{O}$ , MFI framework type, Arletti et al. 2011) and H-ZSM-5 (i.e.  $(\text{H}_{6.8}\text{Na}_{1.1})(\text{Al}_{7.9}\text{Si}_{89.8})\text{O}_{192} \cdot 36\text{H}_2\text{O}$ , MFI framework type, Quartieri et al. 2011); all-silica ferrierite (FER framework type; Lotti et al. 2015a); paulingite (ideally  $(\text{K}, \text{Na}, \text{Ca}_{0.5}, \text{Ba}_{0.5})_{10}(\text{Al}_{10}\text{Si}_{32}\text{O}_{84}) \cdot n\text{H}_2\text{O}$ , with  $n = 27\text{--}44$ , PAU framework type; Gatta et al. 2015); synthetic  $\text{AlPO}_4\text{-5}$  (AFI-framework type; Lotti et al. 2016).

The experiment of Colligan et al. (2004), on a purely siliceous zeolite Y (FAU framework type, Baerlocher et al. 2007), deserves a particular attention. Colligan et al. (2004) compressed a synthetic all-silica zeolite Y (FAU framework type) in silicone oil and in methanol:ethanol:water = 16:3:1 mix using a DAC, and described its *HP*-behaviour on the basis of in situ synchrotron X-ray powder diffraction data, with Rietveld structure refinements and computational modelling. This was probably the first experiment in which a neutral and hydrophobic zeolitic framework was used to describe *P*-induced penetration phenomena, giving the opportunity to examine the effect of pressure on a porous silicate without interferences due to extra-framework charge-balancing cations and their interactions with framework oxygen atoms. Compressed in the mixture of alcohol–water, this zeolite shows a drastically lower compressibility than that observed in (non-penetrating) silicone oil. In addition, in alcohol–water mix, two distinct compressional patterns occurred, with a changeover at 4 GPa. The Rietveld structure refinements proved that new extra-framework sites occur at high pressure, modelled as partially or fully occupied by oxygen atoms of  $\text{H}_2\text{O}$  molecules. The sum of the extra-framework site occupancies increases with pressure, and the pore filling saturation is achieved at about 4.0 GPa. This last experimental finding allowed the authors to explain the change of the compressional behaviour of this zeolite compressed in methanol:ethanol:water = 16:3:1 mix: (1) *P*-induced intrusion of  $\text{H}_2\text{O}$  molecules is the principal process that occurs between 0.0001 and 4 GPa, leading to a drastically lower compressibility if compared to that observed in silicone oil; (2) when the filling of the pores is completed, at  $P > 4$  GPa a new compressional pattern, with higher compressibility, is observed and the effect of hydrostatic compression is predominantly accommodated by framework deformation (mainly via tetrahedral tilting). Unfortunately, no structure refinements were performed in decompression, leaving open questions about the

reversibility of the intrusion process. With this experiment, Colligan et al. (2004) showed how polar molecules (i.e.  $\text{H}_2\text{O}$  molecules) can be intruded in a neutral framework in response to applied pressure.

Some experiments have been performed using liquid  $\text{CO}_2$  as penetrating *P*-transmitting fluid. Natrolite ( $\text{Na}_{16}\text{Al}_{16}\text{Si}_{24}\text{O}_{80} \cdot 16\text{H}_2\text{O}$ ), for example, transforms to  $\text{Na}_{16}\text{Al}_{16}\text{Si}_{24}\text{O}_{80} \cdot 16\text{H}_2\text{O} \cdot 8\text{CO}_2$  at 1.5 GPa (Lee et al. 2011). The penetration of extra  $\text{CO}_2$  molecules through the eight-membered ring channels, running along [001], leads to an expansion of the channels and, in turn, to a spectacular unit-cell volume increase by ~6.8%. The  $\text{CO}_2$ -bearing natrolite stable at high pressure contains ~12 wt % of  $\text{CO}_2$ , and its symmetry decreases from orthorhombic (Sp. Gr. *Fdd2*) to monoclinic (Sp. Gr. *Cc*). The intruded  $\text{CO}_2$  molecules give rise to a rearrangement of the extra-framework population, with a migration of the  $\text{H}_2\text{O}$  molecules toward one side of the channel.  $\text{CO}_2$  molecules interact with both Na and  $\text{H}_2\text{O}$ , and lie in a plane almost perpendicular to the channel direction. The structure refinements after decompression showed that  $\text{Na}_{16}\text{Al}_{16}\text{Si}_{24}\text{O}_{80} \cdot 16\text{H}_2\text{O} \cdot 8\text{CO}_2$  is meta-stable after *P*-release (even after an equilibration time of 1 h).

Haines et al. (2010) showed that the *P*-induced intrusion of extra molecules of  $\text{CO}_2$  in silicalite (MFI framework type) hinders its *P*-induced amorphization (at least up to 20–23 GPa), whereas in a non-penetrating fluid this zeolite showed effects of amorphization at 4–5 GPa.

### ***P*-induced penetration of complex molecules and polymerization phenomena**

Santoro et al. (2013) reported the first evidence of *P*-induced photo-polymerized ethylene in the channels of silicalite (MFI framework type, Baerlocher et al. 2007), using single crystal and powder of  $\text{SiO}_2$ -silicalite compressed in supercritical fluid  $\text{C}_2\text{H}_4$ . The penetration of  $\text{C}_2\text{H}_4$  molecules occurs at 0.5–1.5 GPa, and the polymerization is promoted under ultraviolet (351–364 nm) irradiation. Experimental evidence, based on optical spectroscopy and X-ray diffraction, confirmed that the structure of  $\text{C}_2\text{H}_4$ -bearing silicalite, recovered at ambient pressure, contains single polyethylene chains confined by the zeolitic channels. The  $\text{C}_2\text{H}_4$ -bearing silicalite shows a significant increase of its bulk modulus and density if compared to the parental silicalite, in response to the *P*-induced pore-filling effect.

On the same zeolite, a further experiment was reported by Scelta et al. (2014) devoted to the *P*-induced polymerization of acetylene molecules (to poly-acetylene chains) through the zeolitic cavities. The authors used a multi-methodological approach, based on in situ and ex situ measurements (by IR spectroscopy, Raman spectroscopy and X-ray diffraction), to describe how with a DAC and using only high pressure (~4 GPa; no temperature or ultraviolet irradiation)



it is possible to promote the penetration and re-organization (via polymerization) of  $C_2H_2$  molecules through the zeolite channels.

Silicalite was also selected as the zeolitic host for a further experiment aimed to investigate the  $P$ -induced polymerization in all-silica zeolites: Santoro et al. (2015) reported the  $P$ -induced synthesis of all-transoid polycarbonyl  $[-(C=O)-]_n$  in a zeolite, starting from mixtures of (solid) CO and powder or single crystal of silicalite, compressed in a DAC. Using a multi-methodological approach (based on IR, Raman, single-crystal X-ray diffraction, and ab initio computational methods for calculating the vibrational spectrum of polymerised CO), the authors reported how compressing the zeolite in solid CO, evidence of CO penetration through the zeolitic cavities with a re-organization in a polymeric configuration were found. The experimental findings indicated that the average interaction between confined polymerised CO and the host silicalite is of the van der Waals type, and the resulting IR spectra are compatible with the all-transoid polycarbonyl  $[-(C=O)-]_n$  chains predicted by DFT studies. The ex situ measurements proved that the  $P$ -induced penetration and polymerization of CO is an irreversible process. To the best of our knowledge, this is one of the very rare examples in which the  $P$ -induced penetration of external molecules occurs in a solid host compressed in a solid medium.

The same group of researchers (Santoro et al. 2016) extended their experiments on a different zeolite with a monodimensional system of channels: the all-silica ZSM-22 (TON framework type, Baerlocher et al. 2007). Polycrystalline ZSM-22 compressed (up to 5–10 GPa) in (liquid) acetylene and (liquid) CO, loaded cryogenically in a DAC, experiences similar phenomena as those previously described for silicalite: the intrusion and the subsequent irreversible polymerization of  $C_2H_2$  (to poly-acetylene) and CO (to polycarbonyl  $[-(C=O)-]_n$ ).

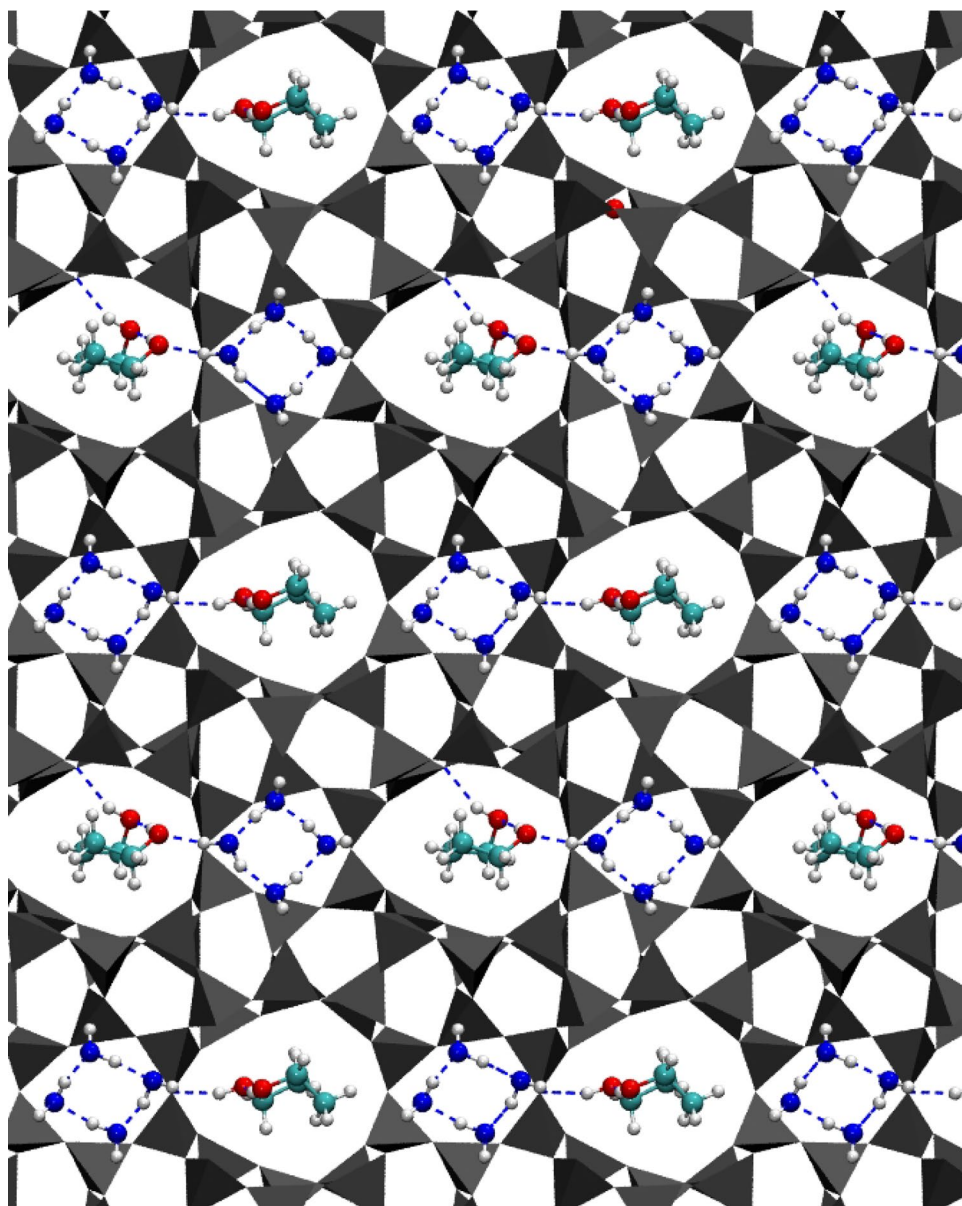
Arletti et al. (2015) investigated the behaviour of a synthetic high-silica mordenite (MOR framework type, Baerlocher et al. 2007) compressed in a non-penetrating fluid (i.e. silicone oil) and a series of potentially penetrating fluids: the mix methanol:ethanol:water = 16:3:1, water:ethanol = 3:1, and ethylene glycol, by in situ (i.e. synchrotron X-ray powder diffraction, Raman spectroscopy) and ex situ measurements (i.e. synchrotron X-ray powder diffraction, IR spectroscopy). The experimental findings showed that: (1) the elastic behaviour with a non-penetrating fluid is consistent with that previously reported (with a  $P$ -induced phase transition from a  $C$ -centred to a primitive space group at  $\sim 1$  GPa, Lotti et al. 2015b) and (2) the  $P$ -induced intrusion of guest molecules, from the  $P$ -fluids through the cavities of this zeolite, occurs for all the (nominally) penetrating fluids and at low pressure. For example, evidence of the ethylene glycol penetration was reported already at 0.1 GPa and this phenomenon appears to

be only partially reversible upon decompression. Whereas methanol or ethanol cannot be intruded at high pressure in Na-mordenite (i.e.  $Na_6Al_6.02Si_{42.02}O_{96} \cdot 19H_2O$ ), as reported by Gatta and Lee (2006) and Lotti et al. (2015b), the absence of the extra-framework population promotes the penetration of these molecules through the empty channels of high-silica mordenite in response to applied pressure.

More recently, Richard et al. (2016) used a combination of in situ Raman spectroscopy and X-ray diffraction to investigate the  $P$ -induced insertion of  $BNH_6$  (ammonia borane, solid at ambient conditions) in the cavities of the hydrophobic silicalite-1F (MFI framework type, Baerlocher et al. 2007). A single crystal of silicalite was compressed in a powder of  $BNH_6$ , using a DAC. The experimental findings showed how  $BNH_6$  molecules penetrate through the cavities of the silicalite-1F structure at very low  $P$  ( $\sim 0.1$  GPa), and the insertion leads to the appearance of new Raman modes. Raman spectra collected at high- $P$  showed how orientational disorder of the  $-BH_3$  and  $-NH_3$  groups, pertaining to the intruded molecules, occur within the  $P$ -range investigated, if compared to the bulk ammonia borane used as  $P$ -transmitting medium. In situ X-ray powder diffraction experiments showed that the compressibility of the  $BNH_6$ -bearing silicalite-1F is three times lower than that of the parental silicalite-1F (with empty cavities), in response to the  $P$ -induced pore-filling effect.

Very recently, Arletti et al. (2017) showed how the hydrophobic all-silica ferrierite (FER framework type, Baerlocher et al. 2007) compressed in the ethanol:water = 1:3 mixture experiences a transition between 0.8 and 1.3 GPa from the orthorhombic (Sp. Gr.  $Pmnn$ ) to monoclinic symmetry (Sp. Gr.  $P2_1/n$ ), coupled with a penetration of the  $P$ -fluid molecules. The X-ray powder diffraction data and modelling (using a dispersion-corrected density functional approximation) showed that the ( $H_2O$ , ethanol)-bearing ferrierite, stable at high pressure, is able to separate the ethanol–water mixture into ethanol dimer wires and  $H_2O$  tetramer squares (Fig. 2). The specific zeolite type—ferrierite—is pivotal for achieving the  $H_2O$ –ethanol organization in such a peculiar two-dimensional arrangement. Ferrierite has two parallel channel systems of different diameter—six- and ten-membered rings—which are perfectly tailored to host, respectively,  $H_2O$  tetramers and ethanol dimers. Surprisingly, the confined supramolecular organization remains stable even upon complete pressure release, which is a key requirement in view of potential technological applications. Indeed, the intrusion of pure  $H_2O$  in all-silica FER was observed by Cailliez et al. (2008) at considerable lower pressures (below 0.30 GPa) using a porosimeter; such an experiment achieved complete filling of the zeolite framework, but  $H_2O$  was reversibly extruded upon pressure release (Cailliez et al. 2008). Notably, powder X-ray diffraction data indicated intrusion of  $H_2O$  molecules at comparable pressure values

**Fig. 2** Supramolecular organization of the H<sub>2</sub>O and ethanol molecules inside the two-dimensional channel system of hydrophobic all-silica ferrierite. The guest molecules are in ball-and-stick representation. Hydrogen bonds are shown as dashed lines. Atom colours: H white, C cyan, O(ethanol) red; O(H<sub>2</sub>O) blue. The structure is projected in the *ab* plane



(0.2 GPa) when ferrierite was compressed in a DAC with a methanol:ethanol:H<sub>2</sub>O = 16:3:1 mixture—in this case, the maximum pressure reached was 1.5 GPa, and H<sub>2</sub>O was retained also at room conditions (Arletti et al. 2014). This suggests that pressures higher than 0.3 GPa are instrumental for the irreversible encapsulation of H<sub>2</sub>O and ethanol molecules in ferrierite, ensuring thus the stability of the supramolecular architecture of dimers and tetramers at room conditions (Arletti et al. 2017).

Previous experiments on the same zeolites were performed by Lotti et al. (2015a) on single crystal and polycrystalline samples, using silicone oil (as a polymeric non-penetrating *P*-fluid) and methanol:ethanol:H<sub>2</sub>O = 16:3:1 mixture, ethylene glycol and 2methyl-2propen-1ol (as potentially penetrating fluids). The experiments with the potentially

penetrating media enhanced the occurrence of *P*-induced intrusion of fluid molecules, with different phase-transition paths and compressibility patterns. However, the disordered distribution of the maxima in the calculated residual electron density map, obtained by single-crystal X-ray structure refinements, did not allow to define a unique scenario of the configuration of the intruded molecules (Lotti et al. 2015a). The starting orthorhombic polymorph was always restored upon decompression with all the used *P*-fluids. The authors highlighted the different *HP*-behaviour observed for single crystal and powder sample using the same *P*-fluid, along with the different *P*-induced phase-transition paths in response to different process kinetics (Lotti et al. 2015a).

The aforementioned experiments on all-silica silicalite (MFI framework type), ZSM-22 (TON framework type),

mordenite (MOR framework type) and ferrierite (FER framework type) were all devoted to describe the  $P$ -induced penetration of molecular guest systems in a zeolitic host. Recently, Arletti et al. (2016) reported the behaviour of an all-silica ferrierite compressed in an electrolytic  $\text{MgCl}_2 \cdot 2\text{H}_2\text{O}$  solution. The  $P$ -induced intrusion was found to occur at very low  $P$  (about 0.19 GPa); the phenomenon is reversible (as proved by the X-ray diffraction before and after the  $P$ -induced intrusion experiment) and it is affected by a moderate hysteresis. The Rietveld refinement, based on the synchrotron diffraction data collected at  $P \sim 0.28$  GPa (above the intrusion pressure), showed that both ions and  $\text{H}_2\text{O}$  molecules of the aqueous solution (used as  $P$ -transmitting fluid) were intruded in the cavities, with an ordered distribution: (1) the  $\text{Mg}^{2+}$  site lies at the centre of the FER cage and it is coordinated by four  $\text{H}_2\text{O}$  molecules (with partial site occupancy) in a square planar configuration, (2) the  $\text{Cl}^-$  site is located in the ten-membered ring channels parallel to [001], coordinated by two  $\text{H}_2\text{O}$  molecules.  $\text{Mg}^{2+}$  and  $\text{Cl}^-$  are partially solvated. The idealized composition of the electrolyte guest solution in the zeolitic cavities is:  $\text{MgCl}_2 \cdot 10\text{H}_2\text{O}$ . At  $P \sim 0.68$  GPa, a phase transition from orthorhombic (Sp. Gr.  $Pmnn$ ) to monoclinic (Sp. Gr.  $P2_1/n$ ) symmetry was also observed.

## Discussion and conclusions

If we consider the behaviour of zeolites when compressed in penetrating and non-penetrating  $P$ -fluids, the first general consideration we can make is that the  $P$ -induced intrusion of new molecules through the zeolitic cavities has a drastic impact of the elastic behaviour and on the structure evolution of a given zeolite. The experimental findings pertaining to compression in non-penetrating fluids showed that:

1. The flexibility observed in this class of open-framework materials, in response to applied  $P$  or  $T$ , is based mainly on tilting of (quasi-rigid) tetrahedra around O atoms that behave as hinges. Tilting of tetrahedra is the dominant mechanisms at low-mid  $P$ -regime, whereas distortion and compression of tetrahedra represent the mechanism which usually dominate the mid-high  $P$ -regime. One of the most common deformation mechanisms in zeolitic framework, able to accommodate the effect of pressure, is the increase, though without inversion, of channels ellipticity. As we can learn from the experiments on isotopic zeolites, the deformation mechanisms are dictated by the topological configuration of the tetrahedral framework and are not influenced by the Si/Al/P distribution or by the extra-framework population. However, the compressibility of the cavities is controlled by the nature and bonding configuration of ionic and molecu-

lar content, resulting in different unit-cell volume compressibility in isotopic structures.

2. The range of compressibility of zeolites is significantly large, with bulk moduli  $\sim 10 < K_{P_0, T_0} < \sim 90$  GPa. The microporosity does not necessarily imply high compressibility, as several zeolites are stiffer than non-zeolitic rock-forming minerals (e.g. quartz, feldspars, feldspathoids, scapolites, micas; Gatta and Lee 2014). In general, zeolites with stuffed channels (i.e. with an extra-framework population) are stiffer than zeolites with empty channels. If the microporosity is represented by the framework density, then the compressibility of zeolites is not directly related to microporosity.

The experimental findings pertaining to compression in penetrating fluids, and thus with crystal–fluid interaction, showed that not all the zeolites experience a  $P$ -induced intrusion of new monoatomic species or molecules from the  $P$ -transmitting fluids. For example, zeolites with well-stuffed channels at ambient conditions (e.g. natural zeolites) tend to hinder the penetration of new species through the channels. There must be several variables that govern the sorption phenomena at high pressure beyond the “free diameters” of the framework cavities, among those: the chemical nature and the configuration of the extra-framework population, the partial pressure of the penetrating molecule in the fluid (if mixed with other non-penetrating molecules, e.g.  $P(\text{H}_2\text{O})$  in a mixture of alcohol– $\text{H}_2\text{O}$ ), likely the rate of the  $P$ -increase, the surface/volume ratio of the crystallites under investigations, the temperature at which the experiment is conducted. As shown by Lotti et al. (2015a), the rate of the  $P$ -increase and the surface/volume ratio of the crystallites can play an important role in governing the crystal–fluid interaction, through  $P$ -induced penetration phenomena, and deserve further investigations.

The re-organization of the intruded molecules in zeolitic hosts, promoted by applied pressure and by other variables (e.g. ultraviolet irradiation, moderate temperature), is one of the most fascinating discoveries in material science over the last decade, with potential technological and geological implications. The aforementioned experiments of Santoro et al. (2013, 2015, 2016), Scelta et al. (2014), Richard et al. (2016) and Arletti et al. (2015, 2017) indicate new routes for creating hybrid host–guest composite materials, where an inorganic framework drives the formation of organic polymer with low dimensionality, acting as a stable host for it. The new hybrid inorganic-host/organic-guest materials display completely different physicochemical properties if compared to the parental zeolites, in which the interesting properties imparted by pressure would be retained also at standard conditions, and could be exploited in applications. In this fascinating perspective, a crucial question to be addressed is whether moderate pressures, e.g. below 1 GPa,

might be sufficient to induce “technologically appealing” irreversible processes, such as the supramolecular organization of simple species in complex patterns, or the formation of one-dimensional polymer chains. Avoiding the use of extreme pressures would be highly desirable for practical applications; for this reason, future investigations along this route should be aimed not only to obtain new composites, but also to determine, or predict, the pressure value at which the transformation would become quantitative and irreversible. Whether supramolecular organization of the included species appears at a certain pressure onset, or if it gradually evolves with increasing pressure, is just one of the questions to be addressed to gather molecular-level knowledge, and hence control, of these processes. A further issue to be investigated is the influence of the composition of the *P*-transmitting media on the intrusion, organization, and transformation processes inside the framework. Additionally, unravelling the molecular details of the pressure-driven penetration of guest species inside the zeolite pores would be a highly challenging task, which would involve a thorough study of the external surfaces of the zeolite material (Hendriks et al. 2017) under non-standard conditions. In this respect, modelling could play a crucial role: the behaviour of surfaces and interfaces under high-temperature conditions has already enabled to capture new phenomena that were not directly accessible to experimental observation (Ceriani et al. 2004a; Fois et al. 2000, 2010b; Tabacchi et al. 2015b)—for example, that the opening of zeolite pores may be enlarged by concerted rotations of the tetrahedral units induced by the intrusion of bulky molecules (Tabacchi et al. 2016). With this premises, we envisage that a combined experimental and modelling endeavour to explore the interfaces of open-framework materials at high-pressure conditions might reveal the occurrence of new, unexpected processes of key relevance for future advances in zeolite science and technology.

The confinement of molecular species inside zeolite channels at normal pressure is already a successful strategy to build supramolecular architectures: materials with innovative properties have been fabricated by constraining molecules in an ordered arrangement within these nanosized spaces. The encapsulation of dyes into the monodimensional system of channels of zeolite L has led to functional composites suitable for a variety of applications—from solar energy technology to nanomedicine (Popovic et al. 2007; Calzaferri 2012; Fois et al. 2012; Cucinotta et al. 2014; Insuwan et al. 2016; Gartzia-Rivero et al. 2017). Host–guest assemblies working as light-harvesting antenna systems are promising materials for the field of molecule-based devices (Calzaferri et al. 2003; Fois et al. 2010a; Martínez-Martínez et al. 2014). These systems usually consist of chromophores hosted in the one-dimensional pores of zeolite L. Like in the antenna systems of natural photosynthetic organisms, upon photon

absorption, a confined dye molecule can transfer its excitation energy to a neighbouring dye molecule, and the energy transfer continues up to the end of the supramolecular chain, where energy could be collected (Calzaferri et al. 2003). To interface the artificial antenna with other components in a working device, the zeolite termination should be functionalized with suitable molecules named “stopcocks” (Maas and Calzaferri 2002), which can also be used to avoid leaking of dye molecules from zeolite channels (Calzaferri 2012; Tabacchi et al. 2015a; Cao et al. 2016; Gartzia-Rivero et al. 2017). Based on this principle, highly versatile composites to be used as tailor-made devices for multiple applications have been created, but their behaviour under non-standard conditions is still to be explored. The study of high-pressure effects on dye–zeolite hybrids would be indispensable for future progress of this technology from fundamental bases. Moreover, by exploiting high pressure, the intrusion of different kinds of dyes might be obtained, with the possibility of organizing these molecules in a complex pattern determined by the topology of the framework and by the related intermolecular interactions, which could ultimately lead to hybrid composites with new, unexpected opto-electronic properties.

High-pressure effects might probably be exploited to improve catalytic processes of relevance for industrial and sustainable applications. These processes may either involve zeolites with strong Lewis acid sites—such as the titanium silicalite TS1 catalyst (Taramasso et al. 1983) adopted in olefin epoxidation processes (Bellussi et al. 1992)—or, more commonly, moderately hydrophilic synthetic zeolites (i.e. with high Si/Al ratio) having protons as extra-framework cations (see, e.g. Fois et al. 2008b). These Brønsted acid sites may activate the intruded molecules. These zeolites are used, for example, in the production of biofuels, as catalysts for cracking and isomerization reactions. Thanks to its framework topology, exhibiting relatively small cavities, hydrophilic ferrierite shows an impressive transition state selectivity for such reactions (see, e.g. Martínez and Corma 2011). However, just because of the small pore diameters, the reaction normally takes place only on the external surface of the zeolite, at its pore opening (Wiedemann et al. 2016). As a consequence of this “pore-mouth catalysis”, only a small proportion of the catalytic sites is actually exploited. Reasonably, moderate pressures could favour the intrusion of the reactant molecules inside the zeolite pores, thus enhancing the performances of the industrial process. Concerning Lewis-acid catalysts, as the olefin epoxidation cycle involves the direct participation of the zeolite framework as active oxygen mediator (Spano et al. 2006), it can be argued that the use of pressure may help to lower the associated free energy barrier, with beneficial effects on catalysis. Besides improving catalytic processes, high-pressure effects might be also effectively exploited to create new zeolite structures



starting from a common, economically convenient parent material. Importantly, the new phases might display better performances than the starting material, as demonstrated by Jordá et al. (2013).

In this broad scenario, modelling studies play a key role. Simulations have become an extremely versatile tool for addressing the complex behaviour of zeolites at high-pressure conditions, and for connecting the experimentally measured response of the system to its features at the atomistic-detail level. Nowadays, the scope of computational investigations is no longer limited to the interpretation of experimental results: they should aim at opening new routes in the various aspects of zeolite technology. Integrated theoretical–experimental approaches would be instrumental in this perspective: they are currently widely adopted and are going to gain further momentum from the growing increase of computing power and the continuous development of faster numerical algorithms.

Concerning the geological implications of the experimental findings on the *P*-induced crystal–fluid interaction in zeolites, it is not difficult to consider the potential role played by zeolites (especially Ca-bearing zeolites, like laumontite) found as among the main mineralogical components of the oceanic basalts and their transformation products, as we know by the ODP—ocean drilling project (e.g. Alt et al. 1986; Sevigny et al. 1992; Yasukawa et al. 2014; <http://iodp.americangeosciences.org/vufind/>). For example, these zeolites may act as potential carrier of H<sub>2</sub>O or CO<sub>2</sub> (or even H<sub>2</sub>S, CH<sub>4</sub>, Ar, Xe, or Kr), sorbed under the combined effect of pressure and temperature during genetic and post-genetic conditions, which can be later released when the oceanic crust is subducted. In addition, whereas it is well known the utilization of synthetic zeolites in petrochemistry (e.g. Vermeiren and Gilson 2009 and references therein), the role played by natural zeolites on generation, migration and accumulation of hydrocarbons (processes that occur at moderate *P/T* conditions) is still unknown. Some zeolite-like minerals, for example, melanophlogite (a polymorph of SiO<sub>2</sub> with an open-framework structure, also called *chltrasil*; Tribaudino et al. 2008, 2010; Gatta et al. 2014), contain CH<sub>4</sub>, CO<sub>2</sub> or N<sub>2</sub> in their structural cavities. Natural methane clathrates—solids condensed at moderate pressure in which a large amount of methane is trapped within a framework of H<sub>2</sub>O—share with zeolites the framework topologies and the host–guest structures. The experimental findings pertaining to the *HP*-behaviour of zeolites can be potentially extended to clathrates. For example, methane clathrates are stable at a higher temperature than liquefied natural gas (i.e. 110–250 K), and this promotes some interest in converting natural gas into clathrates instead of using the conventional liquefaction technology for transportation. The production of methane clathrates from natural gas requires less energy and a

smaller refrigeration plant if compared to liquefied natural gas. Can the application of moderate pressure improve the methane clathrates condensation with a more efficient insertion of CH<sub>4</sub> in the H<sub>2</sub>O framework?

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**G. D. Gatta** is a professor of Mineral Sciences at the Department of Earth Sciences of the University of Milan, Italy. His main research interest is the behaviour of microporous materials under extreme conditions, and the crystal-fluid interaction at high pressure. He was the recipient of the Max Hey Medal of the Mineralogical Society of Britain (2008) and of the Medal for Research Excellence of the European Mineralogical Union (2013). He is Fellow of the Mineralogical Society of America.



**P. Lotti** is a research scientist in Applied Mineralogy at the Earth Sciences Department of the University of Milan, Italy. His research activity is mainly devoted to study the behavior of natural and synthetic microporous compounds at non-ambient (T,P)-conditions. He was the recipient of the “Panichi Award” of the Italian Society of Mineralogy and Petrology (SIMP, 2016) and of the “Nardelli Award” of the Italian Crystallographic Association” (AIC, 2017). He is associate editor of *The American Mineralogist*.



**G. Tabacchi** is a computational chemist and professor of physical chemistry at the Department of Science and High Technology of the Insubria University at Como, Italy. Her research activity includes the simulation of molecules, materials, and interfaces at both normal and high-pressure/high-temperature conditions. Porous materials and host-guest compounds represent her principal research interest, to gather understanding of their peculiar properties—low-dimensionality, size-confinement effects, supramolecular organization and reactivity—at molecular-level detail.