



On the hydrogen storage performance of Cu-doped and Cu-decorated graphene quantum dots: a computational study

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Abstract

Hydrogen gas is a promising renewable energy source. The hydrogen storage performance of two differently modified graphene surfaces, particularly Cu-doped and Cu-decorated circumcoronene (CC), is investigated using density functional theory, 6-311G* basis set and Bader's quantum theory of atoms in molecules (QTAIM). It is found that the Cu-doped CC is able to bind three H₂ molecules on one Cu atom, while the Cu-decorated CC is able to bind up to five H₂ molecules on one Cu atom. Changes in the topology of charge density upon the H₂ adsorption are evaluated under the formalism of QTAIM analysis. The QTAIM analysis of bond critical points as well as the density of states analysis show that the interaction between Cu and adsorbed H₂ molecules can be considered as a physisorption (a van der Waals type interaction). Overall, the results presented in this study point out that the Cu-decorated graphene surfaces are more suitable potential candidates for hydrogen storage than the Cu-doped ones. Furthermore, the inclusion of diffuse functions in the basis set is critically considered.

Keywords Circumcoronene · DFT · Graphene quantum dots · Electronic structure · Hydrogen storage · QTAIM analysis

1 Introduction

In the last two decades, graphene based materials have attracted a huge attention due to their extraordinary chemical, mechanical, electronic and optical properties [1–7]. Graphene, rediscovered and isolated in 2004 [1], is a two dimensional monolayer of sp²-hybridized carbon atoms arranged in a hexagonal honeycomb lattice. In 2007, Schedin et al. [3] have proposed an idea to use graphene materials as adsorbents of individual gas molecules. However, the pristine graphene behaves like a zero-band-gap semiconductor, due to the overlap of its valence and conduction bands at the Brillouin zone [8].

Opening of this band gap is highly desirable because it significantly enhances the electronic properties and chemical reactivity of graphene materials [8]. Opening of this band gap can be performed either by doping (i.e. replacing a carbon atom with the dopant atom) or by decorating (i.e. placing a heteroatom above the carbon layer) of the graphene surfaces (GS) with different heteroatoms [9–18]. Among many transition metals, copper and its oxides have been used as possible dopants or decorating atoms of GS in both theoretical and experimental studies [9, 10, 13, 14, 19–21]. For example, in the theoretical work of Düzenli [10], the Cu-doped GS has been reported as the most efficient H₂O₂ adsorbent. Mohammadi-Manesh et al. [13, 14] have studied the adsorption of H₂S on the GS decorated with Cu, CuO and Cu₂O using density functional theory (DFT), particularly PBE functional. Very recently, a suitability of Cu-doped and Cu-decorated GSs as powerful adsorbents has been demonstrated for a series of organic molecules employing QTAIM analysis [20, 22]. From the synthetic point of view, several experimental attempts to prepare graphene materials decorated with copper atoms [23–25], or covered with Cu²⁺ cations [26] have been reported. Still, it has to be mentioned that the synthesis of Cu–GS units with well defined dimension depends strongly on the reaction time, concentration, voltage, electric current and other conditions [23–25, 27].

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Hydrogen gas is a promising clean and renewable energy source [28]. However, it is difficult to store the hydrogen gas (H_2) under ambient conditions due to weak interactions among the H_2 molecules. The first crucial step in the storage process of the particular gas (in this case H_2) is its efficient adsorption on the adsorbent material. It has been shown that the hydrogen molecule exhibits only weak affinity towards the pristine graphene surface [29]. Nevertheless, a number of theoretical studies of hydrogen storage performance of different doped graphene surfaces are rapidly increasing [29–35]. For example, the hydrogen storage ability of Cr-doped graphene nanoflakes (GS) has been theoretically studied by Xiang et al. [29]. Xiang et al. [29] have reported that three H_2 molecules can be adsorbed on each Cr atom. Similarly, Zhou et al. [31] have reported that three H_2 molecules can be stored on one Li atom of Li-doped GS. Tabtimsai et al. [30] have studied the H_2 adsorption on GS doped with different transition metals. The strongest binding ability towards H_2 has been reported for Os-doped GS, being able to store three H_2 molecules on one Os atom [30]. Faye and coworkers [34] have shown that the presence of boron dopant within the Cu-functionalized (decorated) GS enhances the H_2 adsorption. Still, the reported maximum number of adsorbed H_2 molecules on one Cu atom is three [34]. With respect to what is mention above, one can conclude that the storage ability of doped GS is three H_2 molecules on one transition metal [29–31]. The interaction of Cu atom with the H_2 molecule has been also intensively studied in the past [36].

In the presented paper, the hydrogen storage performance of copper-doped and copper-decorated circumcoronene is studied combining the methods of density functional theory (DFT) and Bader's quantum theory of atoms in molecules (QTAIM) [37]. A comparison between the suitability of these two differently modified GSs as hydrogen adsorbents is discussed as well. The results of this work are presented in Sect. 3 which is subdivided into three main parts. The first subsection is focused on the structural parameters, i.e. bond distances and their changes upon the H_2 adsorption. In the second subsection, calculated adsorption energies and LUMO–HOMO band gaps are considered including the assessment of orbital positioning obtained from the PDOS analysis. QTAIM charge density topology and analysis of bond critical points (BCP) and delocalization indexes (DI) are discussed in the third subsection.

2 Computational details

Circumcoronene (CC) is a polyaromatic hydrocarbon composed of 54 carbon atoms arranged in 19 hexagonal rings. It has been previously used as a model system of the graphene quantum dot (GQD), [20, 22, 29, 30, 38–41] because

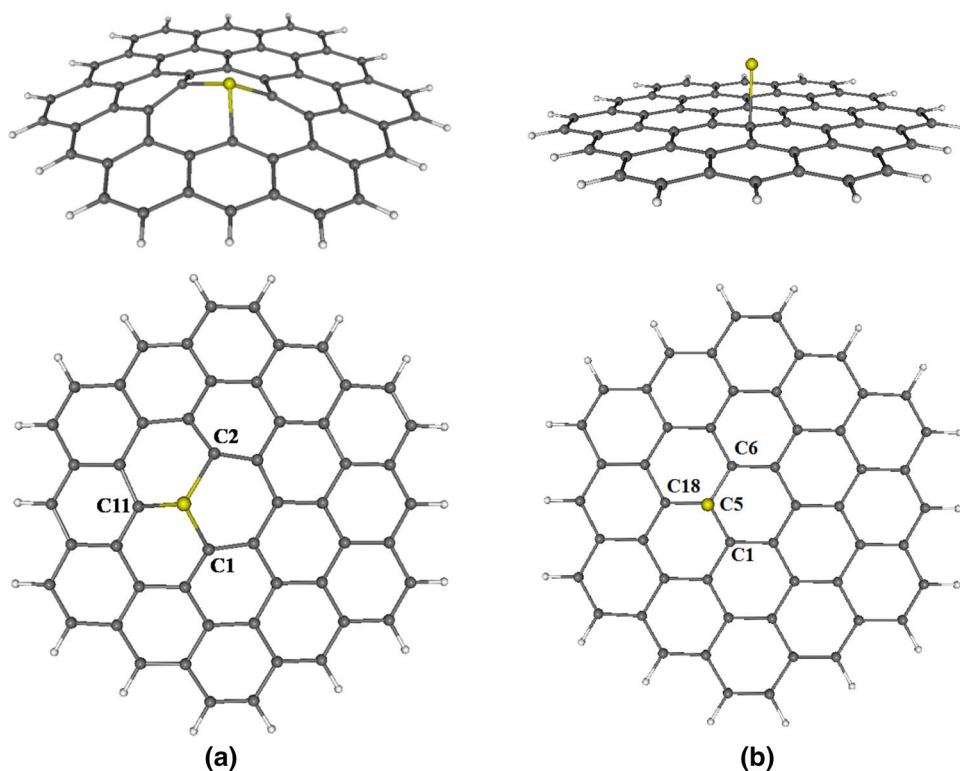
it has a high symmetry and sufficient yet reasonable size from the computational point of view. Optimized geometries (B3LYP/6-311G* level of theory) of Cu-doped and Cu-decorated CC were taken from authors' previous works, [20, 22] considering only the most stable positions of dopant or decorating Cu atom. These geometries were re-optimized using the B3LYP [42–45] functional including the D3 version of Grimme's dispersion correction (GD3) [46] and the same basis set as mentioned above [47–49]. The inclusion of GD3 correction is essential for a proper treatment of adsorption energies and geometries, because standard DFT normally fails in a description of the long-range van der Waals interactions [50–52]. Even though the GD3 correction may overestimate the interaction energies [53], it is still a very good choice when taking into account its accuracy and computational cost [54]. From hereafter, the Cu-doped and Cu-decorated CC will be abbreviated as "Cu-dopCC" and "Cu-decCC", respectively. Structure of the optimized Cu-doped and Cu-decorated CC (including both side- and top-view) is shown in Fig. 1

Hydrogen molecules (H_2) were placed in the vicinity of the Cu atom in both studied systems with the initial distance between the H_2 and Cu being from interval of 2.2–2.5 Å. Number of interacting H_2 molecules was varied from one to five for both systems under study. Subsequent geometry optimizations were performed at the same level of theory, i.e. B3LYP-GD3 [42–46]/6-311G* [47–49]. The energy-based criterion of the SCF convergence was set to 10^{-8} Hartree in all calculations. All systems containing Cu were treated as doublets using the unrestricted Kohn–Sham formalism. Vibrational analysis was employed to confirm that the optimal geometries correspond to energy minima (i.e. no imaginary vibrations). All the aforementioned DFT calculations were carried out using the Gaussian09 program package [55]. QTAIM analysis [37] as implemented in the AIM-All package [56] was performed for every system under study using the Gaussian09 checkpoint files. This analysis was employed to evaluate charge and spin densities at the particular atoms (especially Cu) as well as to estimate the character of interatomic (or intermolecular) interactions via analysis of bond critical points (BCP) and delocalization indexes (DI). Density of states (DOS) and partial density of states (PDOS) calculations were carried out using the GaussSum package [57] to get insight into the molecular orbitals overlapping. Visualization of the optimized structures was performed using the Molekel software suite [58].

To check the results consistency with respect to the basis set used, additional geometry optimizations of the systems with one adsorbed hydrogen molecule (i.e. Cu-dopCC + H_2 and Cu-decCC + H_2) were performed using 6-311G***, 6-311++G** and 6-311++G** [47–49, 59] basis sets.

The basis set superposition error (BSSE) corrected $E_{\text{ads}}^{\text{BSSE}}$ adsorption energies between the Cu-dopCC or Cu-decCC

Fig. 1 Scheme of the B3LYP-GD3/6-311G* optimized Cu-doped (a) and Cu-decorated (b) CC including relevant atom labelling (note that in the case of Cu-decorated CC is carbon C5 directly under Cu atom)



systems and the H_2 molecules were calculated using the equation

$$E_{\text{ads}}^{\text{BSSE}} = E_{\text{CuCC}/nH_2} - (E_{\text{CuCC}}^{\text{BSSE}} + E_{nH_2}^{\text{BSSE}}) \quad (1)$$

where E_{CuCC/nH_2} is the energy of system with n adsorbed hydrogen molecules, $E_{\text{CuCC}}^{\text{BSSE}}$ and $E_{nH_2}^{\text{BSSE}}$ are the BSSE corrected energies of Cu-dopCC or Cu-decCC and n H_2 molecules, respectively, calculated via the counterpoise scheme of Boys and Bernardi [60]. Number of H_2 molecules ranges from one to five. It has to be noted that in some of the counterpoise scheme calculations for evaluating the BSSE corrected energies a quadratically convergent (QC) SCF procedure [61] had to be employed because of their problematic convergence.

3 Results and discussion

3.1 Structural parameters

The calculated distances between the Cu and the hydrogens from adsorbed H_2 molecules as well as between the Cu and the neighbouring or adjacent carbons of CC backbone for all studied systems are compiled in Table 1. Doping the CC backbone with Cu leads to the perturbation of its originally planar structure due to the larger number of electron shells of Cu when compared to carbon [10, 20], see Fig. 1a. On

the other hand, decorating of the CC backbone with Cu preserves its planar (honeycomb) structure [13, 14, 22], see Fig. 1b. In this study, a distance between the Cu atom and CC backbone in Cu-decCC optimized at the B3LYP-GD3/6-311G* level of theory is 2.07 Å (see Cu–C5 value in Table 1). For comparison, the B3LYP/6-311G* Cu–C5 distance reported in one of our previous studies is 2.06 Å [22]. Mohammadi-Manesh et al. have reported the Cu–GS distance of 2.12 Å using the PBE functional with pseudopotentials [13]. Naturally, a Cu atom as an (H_2) acceptor site is sterically significantly less hindered in the Cu-decCC than in the case of Cu-dopCC as can be seen in Fig. 1.

As can be seen from Table 1, the Cu-dopCC is able to adsorb (or store) three H_2 molecules, the fourth one is too distant from the Cu acceptor site (see Cu–H distance of 3.65 Å in Table 1). This is further confirmed by the presence of only three Cu–H bond critical points found in Cu-dopCC + 4 H_2 system, see the section QTAIM charge density topology and bond critical points analysis. The obtained number of adsorbed H_2 molecules corresponds well with the previously published results [29–31, 34] where fourth and/or fifth H_2 molecule are found to be significantly more distant from the dopant atom (Cr, Li, Os, Cu, etc.) than the other three ones. Hence, we can confirm that the ability to store three H_2 molecules seems to be the limit of the doped CC (or doped GS in general). All of the adsorbed H_2 molecules (one, two or three) on the Cu-dopCC are found within the distance of 2.5–2.7 Å from Cu atom. On the other hand, in the case

Table 1 Optimized B3LYP-GD3/6-311G* Cu–H and Cu–C distances in Cu-dopCC and Cu-decCC

System	Distances/Å				
	Cu–H	Cu–C1	Cu–C2	Cu–C11	
Cu-dopCC		1.84	1.86	1.86	
+ H ₂	2.65	1.84	1.87	1.87	
+ H ₂ ^{6-311G**}	2.62	1.84	1.87	1.87	
+ H ₂ ^{6-311++G*}	1.75	2.24	1.96	2.34	
+ H ₂ ^{6-311++G**}	1.71	2.25	1.96	2.35	
+ 2H ₂	2.52, 2.61	1.86	1.88	1.88	
+ 3H ₂	2.49, 2.57, 2.55	1.87	1.89	1.89	
	Cu–H	Cu–C5	Cu–C6	Cu–C1	Cu–C18
Cu-decCC		2.07	2.60	2.60	2.62
+ H ₂	2.64	2.09	2.61	2.62	2.64
+ H ₂ ^{6-311G**}	2.63	2.09	2.61	2.61	2.64
+ H ₂ ^{6-311++G*}	2.76	3.08	3.55	3.11	3.50
+ H ₂ ^{6-311++G**}	2.76	3.08	3.50	3.55	3.11
+ 2H ₂	2.64, 2.64	2.09	2.59	2.63	2.65
+ 3H ₂	2.61, 2.62, 2.61	2.10	2.63	2.62	2.63
+ 4H ₂	2.62, 2.60, 2.60, 2.61	2.11	2.66	2.59	2.65
+ 5H ₂	2.64, 2.66, 2.66, 2.68, 2.68	2.15	2.68	2.66	2.68

For C atoms labelling see Fig. 1. For comparison, the optimized Cu–H and Cu–C distances obtained by using different basis sets are presented for the systems with one adsorbed H₂ molecule

of Cu-decCC up to five H₂ molecules can be adsorbed on the Cu atom: all five Cu–H distances in the Cu-decCC + 5H₂ system are from the interval of 2.64–2.68 Å, see Table 1. This higher H₂ adsorption capacity of Cu-decCC (five) when compared to Cu-dopCC (three) can be mainly assigned to the steric constraints in Cu-dopCC. Naturally, the adsorption of H₂ molecules leads to a small destabilization of the Cu atom upon the CC surface, what is reflected in the increase of Cu–C5 (in Cu-decCC) and Cu–C1, Cu–C2 and Cu–C11 (in Cu-dopCC) bond lengths, see Table 1. These small changes in Cu–C bonds will be further discussed in the section QTAIM charge density topology and bond critical points analysis. For completeness, schematic drawings of Cu-dopCC and Cu-decCC systems with the adsorbed H₂ molecules optimized at the B3LYP-GD3/6-311G* level of theory are shown in Figs. 2 and 3, respectively. Side-views of all the studied systems are provided in Figs. S1 and S2 in electronic supporting information (ESI). The H–H bond lengths in all adsorbed H₂ molecules are around 0.74 Å what corresponds to the H–H bond in the molecular hydrogen.

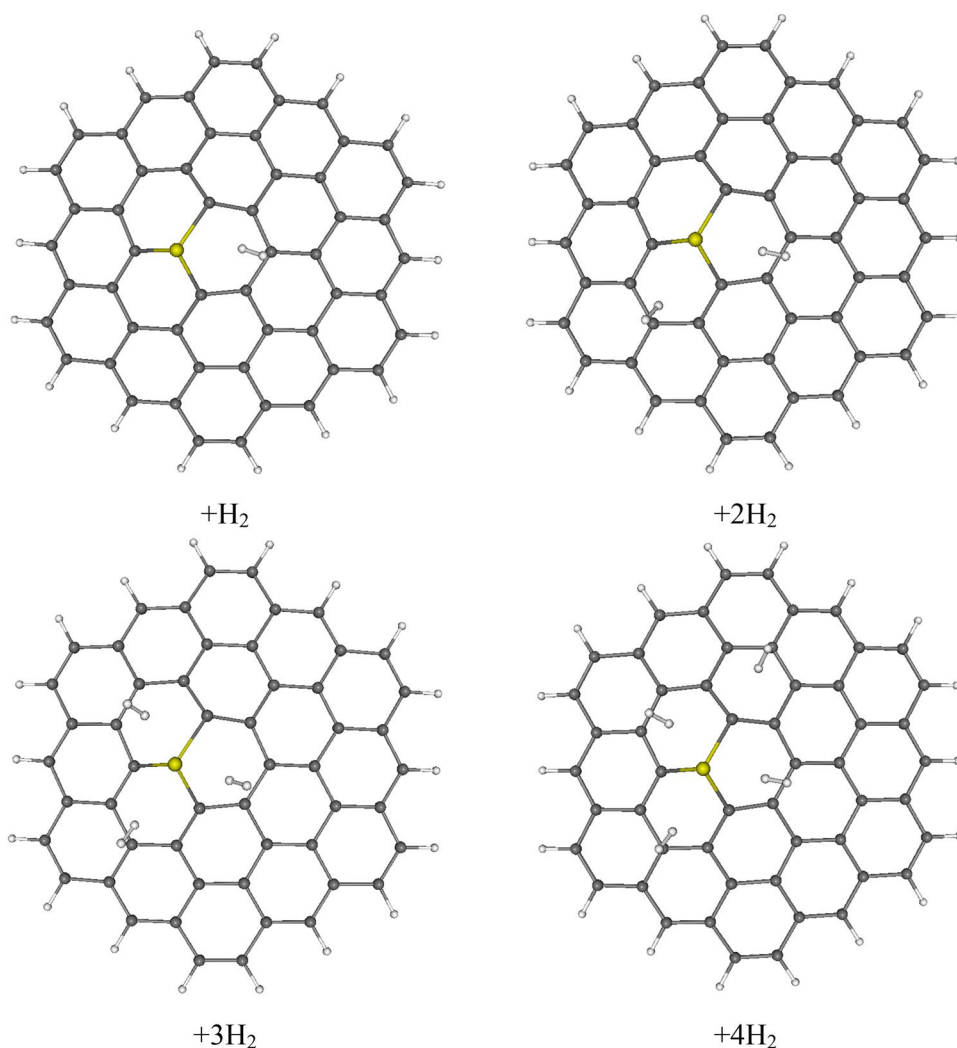
Last but not least, an effect of the different basis sets used (different than 6-311G*) has to be discussed. The inclusion of diffuse functions into the basis set (i.e. employing 6-311++G* and 6-311++G** basis sets) leads to a considerable change in the interactions between the Cu atom and the CC backbone. In the case of the Cu-dopCC system, perturbations in the CC backbone occur which are similar as found for the Cu-doped surfaces reported in one of

our previous works [20]. In addition, the Cu atom in the 6-311++G** optimized Cu-dopCC binds the H₂ molecule in a side-on way (as illustrated in Fig. S3), while when using 6-311G* basis set, the Cu–H₂ interaction has an end-on nature (see Fig. S1). Interestingly, as reported by Ruíz and coworkers [36], the side-on attack of the Cu atom on the H₂ molecule is typical for the ²P excited state of Cu. (The Cu electronic states ²S, ²D, ²P correspond to the valence electron configurations 3d¹⁰4s¹, 3d⁹4s², 3d¹⁰4p¹, respectively [36, 62].) In the case of the Cu-decCC system, the Cu–CC distance becomes destabilized by almost one Å and the H₂ molecule becomes adsorbed above the CC backbone. The diffuse functions within the 6-311++G** basis set are crucial for the assessment of the ²S ground state of Cu, while for the 6-311G* basis set an admixture of ²S and ²D states is identified as the ground state.

3.2 Adsorption energies, LUMO–HOMO band gaps and density of states analysis

The calculated BSSE corrected adsorption energies ($E_{\text{ads}}^{\text{BSSE}}$) and α and β LUMO–HOMO band gaps ($E_{\text{LUMO–HOMO}}$) of all studied systems are compiled in Table 2. The importance of BSSE correction in the calculations of adsorption energies has been already discussed in our previous work [22]. As can be seen from Table 2, the calculated adsorption energies are naturally getting higher with increasing number of adsorbed H₂ molecules. In the case

Fig. 2 Scheme of the optimized B3LYP-GD3/6-311G* Cu-dopCC plus interacting H₂ molecules



of Cu-dopCC system is this increase approximately linear ranging from -6.7 up to -24.8 kJ mol^{-1} , i.e. around 6 kJ mol^{-1} with every H₂ molecule adsorbed, see Table 2. On the other hand, in the case of Cu-decCC system is the increase of $E_{\text{ads}}^{\text{BSSE}}$ values rather exponential with respect to the number of adsorbed H₂ molecules, ranging from -0.3 up to -30.5 kJ mol^{-1} (see Table 2). It has to be mentioned that the B3LYP functional (without GD3 long range correction) fails in calculating adsorption energies, as is written above in Computational details section. The use of B3LYP functional does not provide negative adsorption energies but positive ones. It is also worth to point out that the adsorption energies of hydrogens on Cu atom are an order of magnitude lower than the ones of oxygen, water or methanol reported in our previous studies [20, 22]. This large difference is due to the higher affinity of Cu towards the O atom (e.g. in O₂) [63] than towards the H atom.

Both alpha and beta LUMO–HOMO band gaps ($E_{\text{LUMO–HOMO}}$) as well as spin squared expectation values

(S^2) of the studied systems are also presented in Table 2. Alpha partial density of states (PDOS) of Cu and H atoms in Cu-dopCC and Cu-decCC systems (including the identification of LUMO–HOMO gap) are shown in Figs. 4 and 5, respectively. The $E_{\text{LUMO–HOMO}}$ values of all the systems under study are only negligibly affected by the adsorbed H₂ molecules, as can be seen from Table 2 and Figs. 4 and 5. The differences in $E_{\text{LUMO–HOMO}}$ values are less than 0.1 eV (Table 2) because there are only negligible changes in the d and s valence shells of Cu near Fermi level upon the H₂ molecules adsorption (see the red lines in Figs. 4 and 5 at ca. -4 eV). Nevertheless, all of the LUMO–HOMO band gaps reported here are significantly lower than the band gap in the pristine CC (being around 2.8 eV) [22] due to the doping or decorating with transition metal (in this case Cu) [8, 9]. The larger β LUMO–HOMO band gap (ca. 2.7 eV) in comparison to the α one (ca. 1.3 eV) in Cu-decCC is caused by the presence of one unpaired α 4 s electron of Cu, represented by the small red dashed peak at ca. -3 eV

Table 2 B3LYP-GD3/6-311G* BSSE corrected ($E_{\text{ads}}^{\text{BSSE}}$) adsorption energies, LUMO–HOMO band gap energies ($E_{\text{LUMO-HOMO}}$) in eV and spin contamination (S^2) of all studied systems. The superscripts α and β stand for alpha and beta LUMO–HOMO band gaps, respectively

System	$E_{\text{ads}}^{\text{BSSE}}/\text{kJ mol}^{-1}$	$E_{\text{LUMO-HOMO}}/\text{eV}$	S^2
Cu-dopCC		1.89 $^{\alpha}$, 1.65 $^{\beta}$	0.777
+H ₂	−6.73	1.85 $^{\alpha}$, 1.66 $^{\beta}$	0.778
+2H ₂	−13.48	1.89 $^{\alpha}$, 1.66 $^{\beta}$	0.778
+3H ₂	−18.14	1.92 $^{\alpha}$, 1.69 $^{\beta}$	0.773
+4H ₂	−24.75	1.91 $^{\alpha}$, 1.69 $^{\beta}$	0.773
Cu-decCC		1.41 $^{\alpha}$, 2.67 $^{\beta}$	0.753
+H ₂	−0.36	1.35 $^{\alpha}$, 2.70 $^{\beta}$	0.752
+2H ₂	−3.49	1.30 $^{\alpha}$, 2.71 $^{\beta}$	0.753
+3H ₂	−7.41	1.26 $^{\alpha}$, 2.72 $^{\beta}$	0.753
+4H ₂	−15.80	1.24 $^{\alpha}$, 2.72 $^{\beta}$	0.754
+5H ₂	−30.54	1.32 $^{\alpha}$, 2.71 $^{\beta}$	0.753

of Cu–decCC + n H₂, see small peaks at ca. 2 eV in Fig. 5. These results hand-in-hand with the relatively long Cu–H distances (Table 1) and small changes in LUMO–HOMO gaps (Table 2) point out that the interaction of Cu with H₂ molecules in the studied systems is driven rather by "physical" van der Waals forces than the "chemical" (electronic) ones. The character of these Cu–H interactions will be discussed in more detail via analysis of bond critical points in the coming section.

For completeness, the calculated B3LYP-GD3/6-311G* harmonic frequencies, representing the IR spectra, of the studied systems are presented in Figs. S4 and S5. As illustrated in Fig. S4, the Cu–CC, C–H and H–H vibrations are found in the regions of ca. 1250, 3200 and 4200 cm^{−1}, respectively. For comparison, a calculated B3LYP-GD3/6-311G* energy of H–H vibration in a single hydrogen molecule is 4393 cm^{−1}. Interestingly, an adsorption of H₂ molecules on the Cu-dopCC system has no effect on the

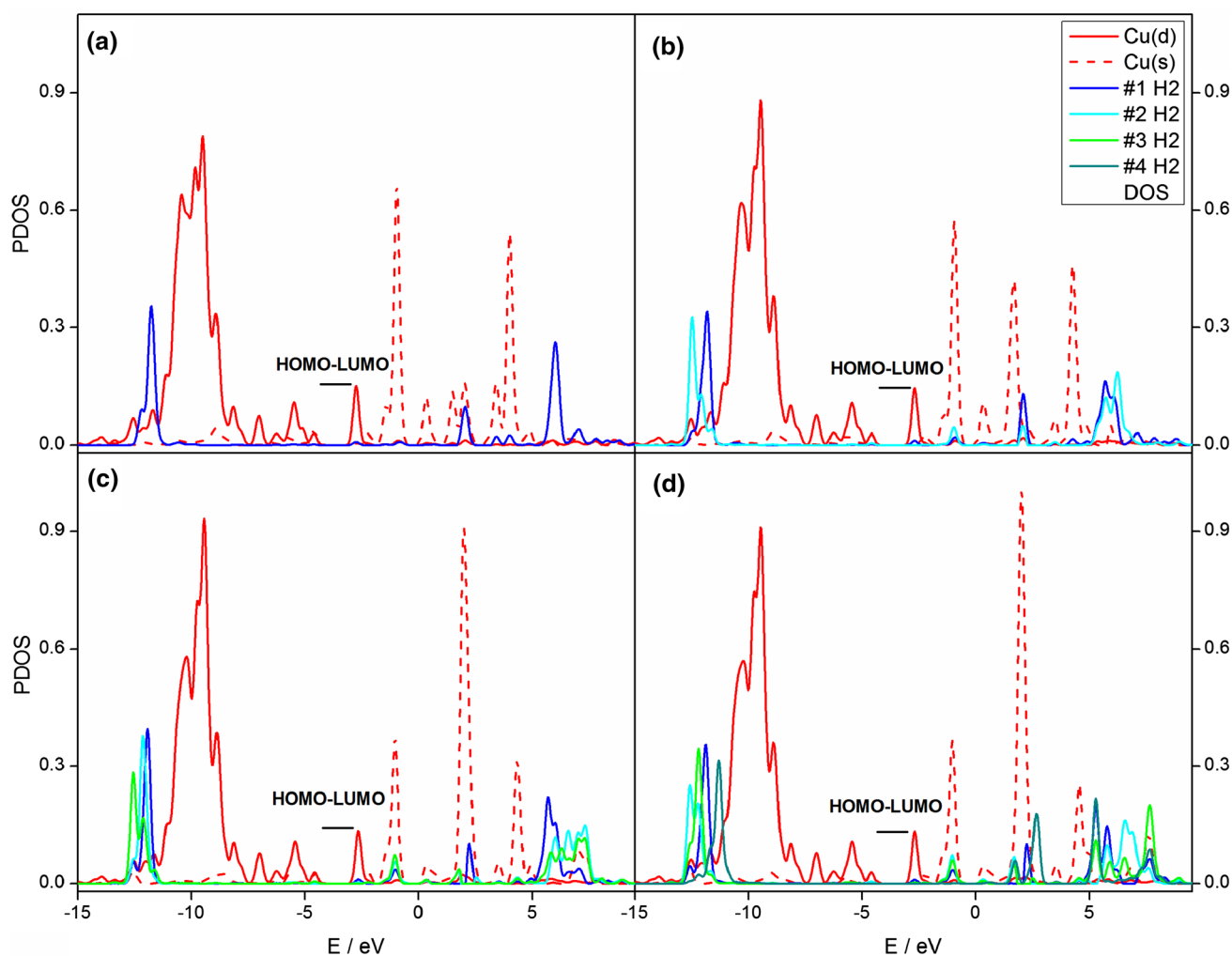


Fig. 4 α PDOS of Cu–H bond in Cu-dopCC + H₂ (a), +2 H₂ (b), +3 H₂ (c), +4 H₂ (d). The particular α LUMO–HOMO band gaps are depicted as well. All plots are obtained at the B3LYP-GD3 level of theory and for comparison purposes the absolute scale is used

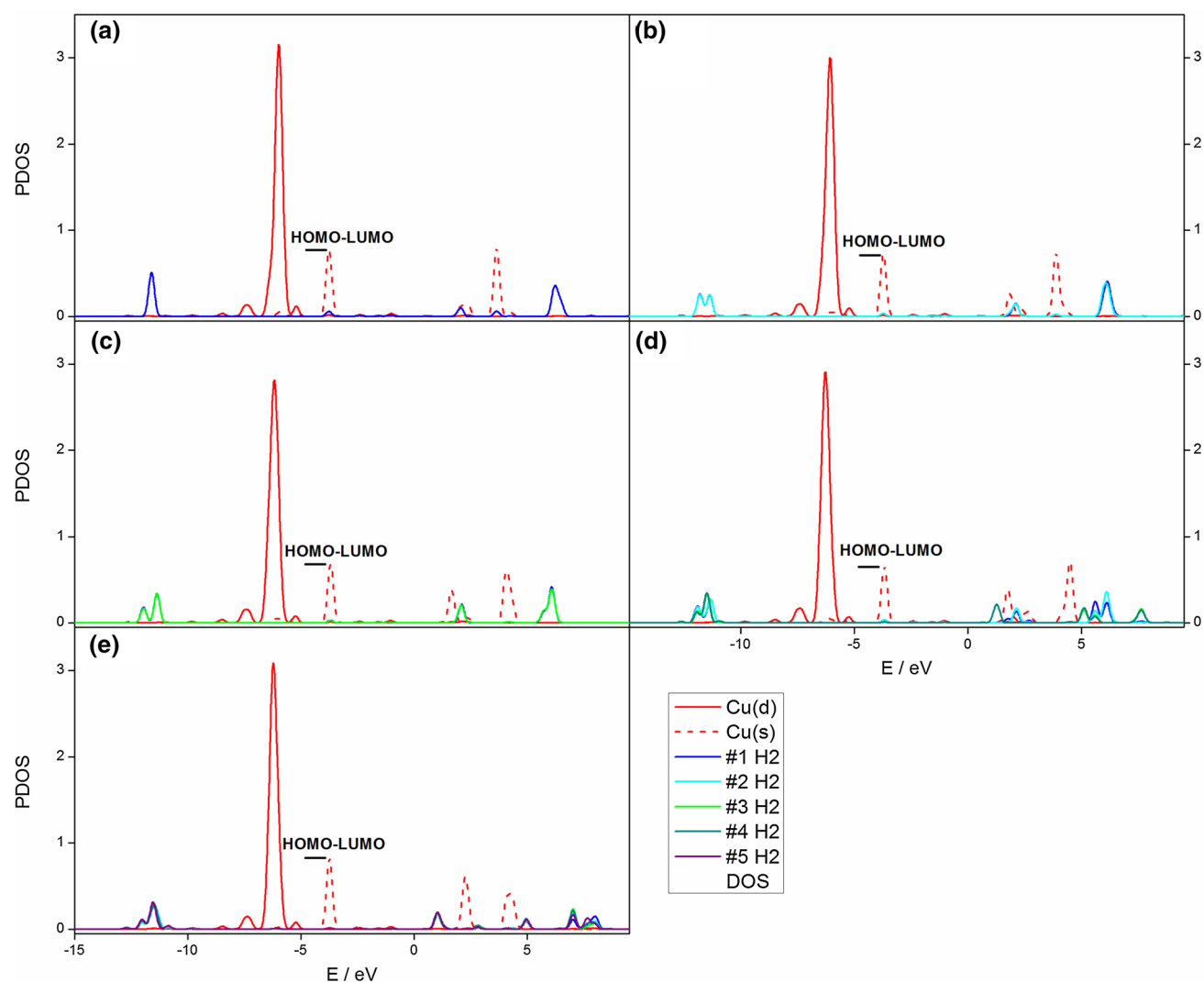


Fig. 5 α PDOS of Cu–H bond in Cu–decCC + H₂ (a), +2 H₂ (b), +3 H₂ (c), +4 H₂ (d), +5 H₂ (e). The particular α LUMO–HOMO band gaps are depicted as well. All plots are obtained at the B3LYP–GD3 level of theory and for comparison purposes the absolute scale is used

calculated Cu–CC vibrations, while in the case of Cu–decCC system, the interaction with H₂ leads to the decrease in IR intensities of these vibrations (Cu–CC), see Fig. S5. Hence, an adsorption of H₂ molecules on the Cu-decorated graphene materials should be experimentally detected in the IR spectra.

3.3 QTAIM charge density topology and bond critical points analysis

Bader’s quantum theory of atoms in molecules (QTAIM) [37] presents a useful tool to analyse the charge and/or spin density distribution using the gradient vector field of the electron density, $\nabla\rho(r)$. The effective atomic charges and spin densities are established upon the formal decomposition of a particular molecule into atomic domains, so-called basins.

The QTAIM charge and spin densities at Cu as a function of the number of adsorbed H₂ molecules in Cu-dopCC and Cu-decCC are shown in Fig. 6. For completeness, the partial QTAIM charges (or spins) at Cu and hydrogen molecules of all studied systems are compiled in Table S1 in ESI. In the case of Cu-dopCC there is only a negligible charge transfer from the adsorbed H₂ molecules: partial charge and spin densities at Cu are around 0.67 and 0.18 e, respectively, (Fig. 6, left) while adsorbed hydrogens have only slightly negative charges (up to -0.019 e see Table S1) which can be considered negligible. For completeness, spin density distribution maps of the Cu-dopCC systems are provided in Fig. S6 in ESI. On the other hand, in the case of Cu-decCC, the positive charge of Cu atom increases with every adsorbed H₂ molecule (see Fig. 6, right) ranging from ca. 0.15 up to ca. 0.30 e. Correspondingly, the spin density of Cu atom

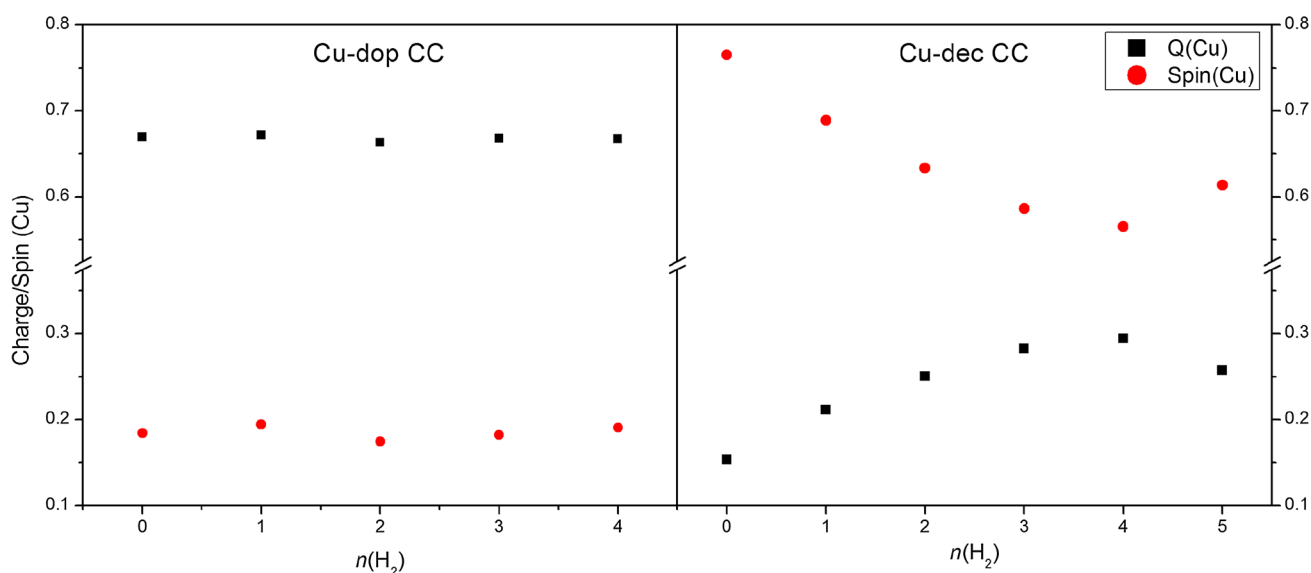


Fig. 6 QAIM charge and spin densities at Cu in Cu-dopCC (left) and Cu-decCC (right) as a function of number of adsorbed H_2 molecules [$n(H_2)$]

decreases upon the adsorption of H_2 molecules (see Fig. 6, right). Spin density distribution maps of the Cu-decCC systems are provided in Fig. S7 in ESI. This charge transfer is reflected in the increased negative charges of adsorbed H_2 molecules (ranging from -0.035 to -0.072 e, see Table S1) as well as in their spin densities (ranging from 0.037 to 0.096 e, see Table S1). From these values, one can clearly see that it is the unpaired electron density which is partially transferred from the Cu atom to the adsorbed H_2 molecules (cf. Figure 6 and Table S1). Interestingly the Cu charge (or spin) dependence of the number of adsorbed H_2 molecules is not divergent, but it has a local maximum (or minimum) in the case of Cu-decCC + 4 H_2 system.

The character (or nature) of chemical bonds can be further inspected using the QAIM analysis of bond critical points (BCPs). The BCP is a saddle point of electron density between two atoms forming a chemical bond [37]. Strength of the chemical bond can be estimated from the charge density in the BCP (ρ_{BCP}). Laplacian of charge density in BCP ($\Delta\rho_{BCP}$) can be either negative or positive depending on whether electronic charge is locally concentrated or depleted [37]. Ellipticity of the electron density (ϵ_{BCP}) is, in general, associated with π character of chemical bonds, hence it can be used as a measure of aromaticity [64]. Delocalization index (DI) is another useful parameter, being a measure of the number of electrons that are shared between two atoms [65]. The BCP electron densities, their Laplacians and DIs of the formed Cu–H bonds in all studied systems are presented in Figs. S8, S9 and 7, respectively. For completeness, the particular values of the BCP characteristics and DIs are given in Table S2 in ESI. Interestingly, the Cu–H bonds of

Cu-dopCC + $n H_2$ systems are getting stronger with every adsorbed H_2 molecule (up to three H_2 molecules) as can be seen from Figs. S8, S9 and 7. However, the particular Cu–H ρ_{BCP} and DI values are still very small, i.e. ca. 0.01 bohr^{-3} and 0.04–0.05, respectively. The situation is similar also in the case of Cu-decCC + $n H_2$ systems: only weak Cu–H bond interaction with ρ_{BCP} of ca. 0.01 bohr^{-3} . On the other hand, the DI values of Cu–H bonds in Cu-decCC + $n H_2$ have the opposite trend than the ones of Cu-dopCC + $n H_2$, as can be seen in Fig. 7. In this case, the strongest Cu–H bond is found in Cu-decCC + H_2 having the DI value of 0.124, even though ρ_{BCP} is still only 0.01 bohr^{-3} (see Fig. 7 and Table S2). For comparison, Afonin et al. [66] have reported that the moderate hydrogen bond between H and O has ρ_{BCP} of ca. 0.04 bohr^{-3} what is four-times larger than our calculated values. This is in agreement with the results presented in the Adsorption energies, LUMO–HOMO band gaps and Density of States analysis section, showing that the interaction between Cu and hydrogens is of van der Waals type.

Another parameter to be monitored upon the adsorption of H_2 molecules are changes in Cu–C bonds. The calculated BCP characteristics (charge densities ρ_{BCP} , Laplacians $\Delta\rho_{BCP}$ and ellipticities ϵ_{BCP}) and DIs of Cu–C bonds in all studied systems are given in Table S3 in ESI. In general, the differences in BCP parameters and DIs of Cu–C bonds are less than 10% in the case of Cu-dopCC + $n H_2$ systems or less than 15% in the case of Cu-decCC + $n H_2$, see Table S3. The larger differences in Cu-decCC + $n H_2$ system when compared to Cu-decCC + $n H_2$ one are due to the larger number of adsorbed hydrogens (five vs. three, respectively). Herein, we show only the dependence of

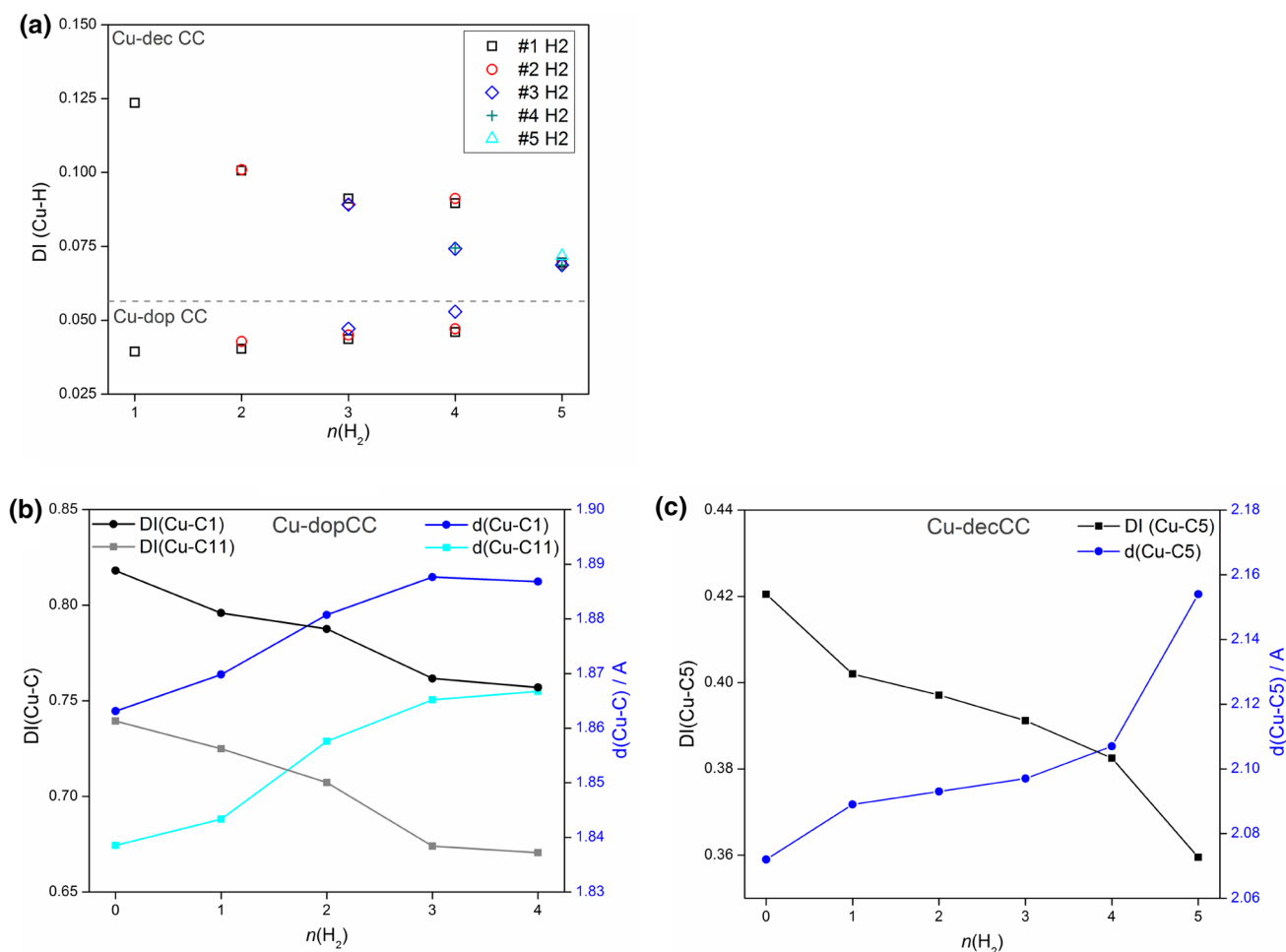


Fig. 7 a B3LYP-GD3/6-311G* delocalization indexes (DI) of Cu–H bonds as a function of number of adsorbed H₂ molecules [$n(\text{H}_2)$], b, c B3LYP-GD3/6-311G* delocalization indexes (DI) and bond dis-

tances (d) of the selected Cu–C bonds of Cu-dopCC and Cu-decCC as a function of number of adsorbed H₂ molecules [$n(\text{H}_2)$]

Cu–C bond lengths and corresponding DI values with respect to the number of adsorbed H₂ molecules [$n(\text{H}_2)$]. These dependences are presented in Fig. 7. (For the sake of simplicity the Cu–C2 bond is omitted in the case of Cu-dopCC system and only Cu–C1 and Cu–C11 are shown. Note that, trends in bond lengths and DI values of Cu–C2 bond are very similar to the ones of Cu–C1 or Cu–C11.) The increase of Cu–C distances (represented by blue and cyan lines) corresponds well with the decrease of Cu–C DI values (black and grey lines) with respect to the growing number of adsorbed H₂ molecules, see Fig. 7. Both of these trends reflect the small weakening of the Cu–C bond upon the H₂ adsorption. Note that, the H–H bonds in all adsorbed H₂ molecules are only slightly weaker (having the DI values of ca. 0.95) when compared to the H–H bond in the molecular hydrogen (DI value of 1.00).

In the case of the calculations taking the diffuse functions into account, the DI value for the Cu–H becomes considerably larger (0.305) in the case of the Cu-dopCC system,

which clearly indicates chemisorption. On the contrary, the DI is smaller for the Cu-decCC system (0.077), because of the larger Cu–H distance, for the 6–311++G** basis set. This actually appears to be a considerable issue not only for the current study but also for other Cu–H interaction studies which employed the DFT/6-311G* protocol [67]

4 Conclusion

In the presented work we have theoretically studied the affinity of Cu-doped and Cu-decorated circumcoronene towards hydrogen molecules to investigate their potential use as H₂ storage materials. The obtained results point out that the adsorption ability of Cu-dopCC is limited to three H₂ molecules on one Cu atom, what is in agreement with previously reported results [29–31, 34]. On the other hand, the Cu-decCC is able to bind up to five H₂ molecules on one Cu atom. The energetic outcome of such adsorption is

within the range of -0.3 to -30 kJ mol⁻¹ with respect to the number of adsorbed hydrogens (see Table 2). In general, the Cu-decCC is found to be more suitable H₂ storage material when compared to the Cu-dopCC, mainly because the Cu in Cu-decCC is sterically less hindered than the one in Cu-dopCC (see Figs. 1, 2, 3). This finding can probably be generalized also to other transition metals used as dopants or decorating atoms in graphene surfaces. However, the PDOS analysis as well as QTAIM analysis of bond critical points show that the interaction between the Cu and adsorbed H₂ molecules in the studied systems is weak and can be considered as a physisorption (see Figs. 4, 5, 6). With respect to what is mentioned above, one can conclude that the Cu-modified (doped or decorated) graphene surfaces present an alternative for currently used H₂ storage materials. However, it has to be mentioned that there are differently modified graphene surfaces, e.g. Cr-doped, [29] Fe-doped or Os-doped [30], exhibiting a better H₂ storage performance than the Cu-modified ones.

Still, it is found that the basis set choice is crucial for the systems studied, where in the case of Cu-dopCC systems the inclusion of diffuse function promises a chemisorption of the H₂ molecule at Cu, while the Cu-CC interaction becomes destabilized and the same occurs for the Cu-H₂ interaction.

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Compliance with ethical standards

Conflict of interest The authors declare that they have no conflict of interest.

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