**ORIGINAL ARTICLE**



# **Tribological behavior of PTFE/Nomex/phenolic composite lubricant under cold forming condition in the bearing assembly process**

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#### **Abstract**

The fabric composites, which are always used as a self-lubricant liner, also work as a forming lubricant during the assembly of the spherical plain bearings. Diferent from previous published works, the tribological behavior of polytetrafuoroethylene (PTFE)/Nomex/phenolic composite lubricant under cold forming condition in the bearing assembly process was investigated in details. A hybrid PTFE/Nomex fabric composite with a volume fraction of 1:3 coated with phenolic resin matrix sticking on a stainless steel 05Cr17Ni4Cu4Nb was prepared for the study. The ring with boss compression test (RCT-B) was selected as a tribometer, and a specifc group of calibration curves were constructed by simulations with various friction factors. To obtain favorable tribological performance of the fabric-composite lubricant, the preparation parameters of the testing workpieces, including number of sandblasting cycles, curing time, and curing temperature, were optimized. To explore the lubricate limit of the optimal fabric-composite lubricant, the RCT-B tests with much larger reduction in height under much higher press speed were furtherly carried out. The failure modes of the fabric-composite lubricant, mainly including peel off, hysteresis, and delamination, were revealed. The failure mechanisms and the applicable limitation of the fabric-composite lubricant were fnally discussed.

**Keywords** Fabric composites · Tribology behavior · Cold forming · Failure modes

## **1 Introduction**

Cold forging is most common when parts are forged without heating the billet. It is a near net-shape manufacturing process. It is a cost-efective method to produce auto components such as shafts  $[1-3]$  $[1-3]$  $[1-3]$ , gears  $[4-6]$  $[4-6]$ , and other parts including inner race [[7\]](#page-11-4), outer race [[8\]](#page-11-5), and spark plug shell [\[9\]](#page-11-6) in complex shape. Lubrication of the billets formed is critical to increase the life of the mating forging dies.

In the manufacture of spherical plain bearings, cold forging process is also used  $[10-12]$  $[10-12]$  in mass production. As for

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spherical plain bearings, hybrid fabric composites [[13\]](#page-11-9) are always employed as the self-lubricating liners [[14,](#page-11-10) [15\]](#page-11-11) to reduce friction and wear. In recent years, fller enhancement, surface modifcation, and the combined technologies have been proposed to improve the tribological properties of different fabric composites.

The addition of nano-titania [\[16\]](#page-11-12), multiwalled carbon nanotubes [[17](#page-11-13)], polystyrene functionalized graphene [\[18](#page-11-14)], graphene oxide [\[19\]](#page-11-15), zirconium diboride particles [\[20](#page-11-16)], and milled pitch-based carbon fbers [\[21](#page-11-17)] can enhance the antiwear ability of fabric composites, and the corresponding appropriate amount of diferent fllers for diferent composites has been determined. Furthermore, hybrids of graphite and MoS2 [\[22](#page-12-0)], nano-Si3N4 and submicron size WS2 [\[23](#page-12-1)], multiwalled carbon nanotubes and graphene oxide [[24](#page-12-2)], boron nitride nanosheets and carbon nanotubes [[25\]](#page-12-3), singlewalled carbon nanotubes and inorganic fullerene-like WS2 [\[26](#page-12-4)] were also used as fillers to improve the tribological performances of fabric composites. Various surface modifcation techniques, including cryogenic treatment [[27\]](#page-12-5), plasma treatment [[28](#page-12-6)], chemical etching [[29\]](#page-12-7), ultrasonic radiation  $[15]$  $[15]$ , chemical coating  $[30]$  $[30]$ , and laser surface texturing  $[31]$  $[31]$ ,

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have been developed to decrease the friction coefficient and wear rate of fabric composites. Recently, many efforts have been dedicated to investigate on the combined methods of surface treatment and fller reinforcement [[32–](#page-12-10)[34](#page-12-11)]. Therefore, the comprehensive performance of fabric composites including tribological properties has been improved.

Among these hybrid fabric composites, the polytetrafuoroethylene (PTFE)/Kevlar [\[35](#page-12-12), [36](#page-12-13)] and PTFE/Nomex [\[37](#page-12-14), [38\]](#page-12-15) composites are always applied as the self-lubricating liners in bearing industry. Through fnite element (FE) method and experiment, the dynamic distributions of the contact stress, wear depth and the worn morphology of the fabric liner were investigated [\[36](#page-12-13)]. To simulate the failure process of spherical plain bearings in the swinging wear condition, the 3D FE model was established based on Archard adhesion wear theory, and the wear mechanism of the fabric liner was revealed with experimental tests [[39\]](#page-12-16). A simplifed 2D FE model was created to predict the wear depth of the fabric liner [[40\]](#page-12-17). The role of transfer flm formation on the tribological properties of fabric composites in spherical plain bearing at low temperature is investigated in details [\[41](#page-12-18)]. Furthermore, a real case study about the failure reasons and wear mechanisms of a joint bearing in helicopter were analyzed, and the dominating phenomena were continuous generation, spalling, extrusion, and regeneration of the PTFE transfer flm occurred between the inner ring and fabric liner [\[42](#page-12-19)].

As mentioned above, there are many excellent works about the tribological behaviors of self-lubricating fabric liners under working condition of bearings used in components, always by using the typical ball on disk or pin on disk testing methods. Actually, during the assembly of spherical plain bearings, the hardened inner ring acts as the inner die when the outer ring is cold deformed, and the fabriccomposite liner sticking on the outer ring works as forming lubricant at that moment, as shown in Fig. [1.](#page-1-0) However, there is very little work on the tribological behavior in this cold assembled process. The present works only focus on the interface friction between the outer ring and forming dies lubricating with Molykote lubricant [[43\]](#page-12-20), and try to simulate the cold forming process for further optimization [[44\]](#page-12-21).

In this study, the tribological behavior of PTFE/Nomex fabric composites under cold forming condition was tried to be investigated. In the following section, the preparation of the testing workpiece according to the manufacturing process of spherical plain bearings is frst described, followed by the key steps of the preparation process. In Sect. [3](#page-3-0), the ring with boss compression test (RCT-B) proposed in previous work [[45\]](#page-12-22) using as a tribometer for cold forming is briefy introduced, and the corresponding calibration curves are constructed by the FE method using the tested data of material 05Cr17Ni4Cu4Nb which is used to make the outer ring in the spherical plain bearing. In Sect. [4,](#page-4-0) the key process parameters, including number of sandblasting cycles, heating curing time, and curing temperature, are orthogonally optimized to achieve a lower shear friction factor, and the determined friction factor could be used to simulate the cold forming process of the spherical plain bearing. In Sect. [5,](#page-6-0) the effects of the reduction in height and press speed on the deformation and damage-prevention properties of the optimized fabric-composite lubricant are experimental tested further, and the failure mechanisms and limitation of the fabric composites working as lubricant in cold forming are discussed. Finally, the conclusions drawn from this research are summarized in Sect. [6.](#page-10-0)

# <span id="page-1-1"></span>**2 Preparation of the testing workpiece**

To carry out the experimental works for the tribological study, the testing workpieces were prepared according to the actual manufacturing process of the spherical plain bearing. In the frst step, a hybrid PTFE/Nomex fabric composites



<span id="page-1-0"></span>

with a volume fraction of PTFE to Nomex:1:3 was rollcoated with phenolic resin matrix, and the mixed composite was precured under a plate curing press. Secondly, the stainless-steel ring specimens were machined from raw bars after heat treatment as required for outer ring in spherical plain bearing, and the specimen surfaces where the composite was to be pasted were sandblasted accordingly. Finally, the mixed composites were pasted to the end surfaces of the ring specimens by using phenolic resin adhesive, and the curing process proceeded under clamping pressure. Once the curing process had been completed, the testing workpiece for the following tribological tests was obtained.

#### **2.1 Precuring of fabric composites**

The fabric was composed of PTFE fbers (weft) and Nomex fbers (weft and warp), and a plain weave shown in Fig. [2a](#page-2-0) was adopted. The diameter of the PTFE and Nomex fbers was 22.44 and 18.06 μm, their breaking elongation was 19.10% and 30.06%, and their initial modulus was 20.35 and 106.30 cN/dtex, respectively. Once precuring had been completed, the mixed composites was formed with hybrid PTFE/Nomex fabric and phenolic resin matrix, as illustrated in Fig. [2b](#page-2-0). The thickness of the composite lubricant could be controlled in the range of 0.38–0.42 mm.

#### <span id="page-2-2"></span>**2.2 Sandblasting of the stainless‑steel specimen**

Martensitic stainless steel (05Cr17Ni4Cu4Nb) was selected, and its chemical composition was  $C(0.07 \text{ wt\%})$ , Si  $(1.00 \text{ wt\%})$ wt%), Mn (1.00 wt%), S (0.03 wt%), P (0.03 wt%), Cr (17.50 wt%), Ni (5.00 wt%), Cu (5.00 wt%), and Nb (0.45 wt%). As shown in Fig. [2](#page-2-0), the stainless steel was heat treated according



<span id="page-2-1"></span>**Fig. 3** Heat treatment process and obtained microstructure

to the heat treatment process for the outer ring of spherical plain bearings. In the solution process, the received bar was heated up to 1030 °C by employing three heat preservation stages at 650 °C, 850 °C, and 1030 °C and then cooled to room temperature in 15 min. In the aging process, the round bars were heated to 620 °C, maintained at that temperature for 4 h, and then cooled. In the microstructure after heat treatment (Fig. [3\)](#page-2-1), the retained and precipitated granular carbides could enhance the dispersion strengthening effect and improve the comprehensive mechanical properties.

After being machined from the heat-treated bars, the top and bottom surfaces of the ring specimens were sandblasted with 24 emeries by using an ACEOP-II shot peening machine. An intermittent shot peening mode was adopted at an interval of 10 s with a spray gun in 6-mm diameter. The flow rate was  $2.0 \text{ m}^3/\text{min}$ , and the shot pressure was set at 0.50 MPa. The ring specimens were slowly sandblasted using circular movements, and various surface topographies could be obtained by difering the number of cycles.



<span id="page-2-0"></span>**Fig. 2** Weave structure (**a**) and schematic (**b**) of the fabric composite. PTFE and Nomex fbers are shown in black and gray with a volume fraction of 1:3

#### **2.3 Final curing process the workpiece**

To realize the fnal curing process, a simple clamping setup shown in Fig. [4](#page-3-1) was adopted. The ring specimen with the PTFE/Nomex/phenolic composites covered on the top and bottom end surfaces was clamped between the PTFE and steel plates. The clamping pressure was maintained at around 2 MPa by tightening four nuts synchronously. The curing temperature was set at 100–160 °C, and the curing time was 1–3 h. In this process, the phenolic resin was cured gradually, and the mixed composites were well pasted on the end surfaces of the ring specimen. Then, the testing workpieces for later tribological evaluation were prepared.

# <span id="page-3-0"></span>**3 Tribological testing method**

Diferent from the typical mechanical friction, here is the friction between the inner ring and the fabric liner when the bearing is running, the friction in metal forming, here is the friction at interface between the inner ring and the fabric liner when the outer ring is cold formed during the assembly, is always involved with very high contact pressure to plasticize the workpiece material [[46\]](#page-12-23). What is more, new surface of workpiece generates during metal forming, and the surface enlargement depending on the workpiece shape [\[47](#page-12-24)] is relatively larger. As for the third index of tribological load, the relative velocity in mechanical friction depends on the running speed of components, while the relative



velocity between die and workpiece is mainly determined by the forming speed.

To characterize tribological behavior in cold forming, more than 20 diferent tribotests for cold forging have been proposed [[48\]](#page-12-25). The ring compression test, which was frstly proposed by Kunogi [[49](#page-12-26)] and further developed by Male and Cockcroft by introducing a set of calibration curves [[50](#page-12-27)], has been one of the most commonly used methods because of its convenience. An alternative method named as RCT-B was put forward [\[45\]](#page-12-22), where the outer diameter of boss of ring specimen before and after deformation could be measured easily and precisely, which could enable more accurate measurement of friction conditions. Therefore, the established RCT-B method in the lab is adopted for the tribological tests of the PTFE/Nomex/phenolic composites under cold forming condition.

#### **3.1 Material flow curves**

To quantify the friction factor in cold forming condition, a corresponding group of calibration curves needs to be constructed according to simulation results. To obtain the material fow curves for the simulation, a group of uniaxial tensile tests of 05Cr17Ni4Cu4Nb stainless steel, which was treated by the same heat treatment process mentioned in Sect. [2.2](#page-2-2), were performed at room temperature. According to the testing data, the following mechanical property parameters were determined: Young's modulus, 198.70 GPa; Poisson's ratio, 0.29; yield strength, 727.12 MPa; and tensile strength, 966.23 MPa.

As shown in Fig. [5](#page-3-2), the stress–strain curves from three tensile samples were plotted, and the data curves of the three samples (C1, C2, and C3) were collected using strain gage and load cell. To describe the hardening behavior of the

<span id="page-3-2"></span>

<span id="page-3-1"></span>**Fig. 4** Elements of the clamping setup **Fig. 5** Stress–strain curves of the 05Cr17Ni4Cu4Nb stainless steel

05Cr17Ni4Cu4Nb stainless steel, a Swift-Voce type law, which is a combination of the unsaturated Swift type law and the saturated Voce type law, was selected. The combined material model was employed because it may improve the prediction accuracy of the material hardening behavior [\[51](#page-12-28)]. On the basis of uniaxial tensile test data, the values of various parameters in the selected Swift-Voce type law could be determined by ftting. The specifc hardening law is derived as follows:

$$
\sigma = 0.0021(\varepsilon + 0.0006)^{2.50} + 0.07
$$
  
× [841.15 + 315.84 × (1 -  $e^{-19.67\varepsilon}$ )] (1)

#### **3.2 Calibration curves**

The ftted Swift-Voce type law was input the FE model of the compression process of the ring with boss with a pressing speed of 1.65 mm/s at room temperature. The basic size ratio of the outer diameter, inner diameter, and height of the ring specimen was 6:3:2, as shown in Fig. [6.](#page-4-1) The nominal dimensions of the outer diameter, inner diameter, height, as well as the height and width of the outer boss were 30.00, 15.00, 10.00, 2.00, and 2.00 mm, respectively.

Simulations with various friction factors, obeying the shear friction rule, were performed to construct the calibration curves. As shown in Fig. [7](#page-4-2), the relationship lines between the reduction in outer diameter of the boss (denoted as  $R_B$ ) and the reduction in height (denoted as  $R_H$ ) under different friction factors are determined according the specifed calculation method in previous work [[45\]](#page-12-22). The variation in the outer diameter of the outer boss for every height reduction was extracted from the simulation results, and the specifc calibration curves within the friction factor ranged from 0 to 0.3 were constructed. Under the reduction in height of 30%, when the friction factor was 0, 0.1, and 0.3, the



<span id="page-4-1"></span>**Fig. 6** Dimensions of the ring specimen



<span id="page-4-2"></span>**Fig.** 7 Calibration curves of four different friction factors (i.e.,  $m=0$ , 0.1, 0.2, and 0.3) and the straight (**a**), concave (**b**), and convex (**c**) profles of the inner diameter of the ring

straight, concave, and convex profles of the inner diameter were as expected, as illustrated in Fig. [7](#page-4-2).

### <span id="page-4-0"></span>**4 Optimization for minimum friction factor**

In the manufacturing process of spherical plain bearings, the outer ring stick with the mixed PTFE/Nomex/phenolic composites is always frstly prepared before assembly using cold extrusion process. As described in Sect. [2](#page-1-1), three parameters including number of sandblasting cycles, heating curing time, and curing temperature can be adjusted during the aforementioned preparation process. Therefore, the preparing process of the mixed composites will be optimized to achieve better lubrication performance based on the RCT-B tests.

## **4.1 Design schemes**

As shown in Table [1](#page-4-3), three controllable process parameters, namely number of sandblasting cycles, heating curing time, and curing temperature, with three levels were established according to the actual process. Correspondingly,

<span id="page-4-3"></span>**Table 1** Controlling parameters and their levels

Parameter	Curing time t(h)	Curing temperature $T({}^{\circ}C)$	Sandblasting cycles $N$
Level 1		100	
Level 2		130	
Level 3		160	

<span id="page-5-0"></span>**Table 2** Testing schemes based on orthogonal design

Scheme	t(h)	$T({}^{\circ}C)$	N	Evaluation index $\theta \times 100\%$
1	1	130	7	45.848
2	1	100	3	46.770
3	1	160	5	48.112
$\overline{4}$	$\overline{c}$	130	5	47.001
5	2	100	7	47.237
6	2	160	3	48.222
7	3	130	3	47.141
8	3	100	5	48.038
9	3	160	7	46.787
k1	46.910	46.663	46.624	
k <sub>2</sub>	47.487	47.349	47.738	
k <sub>3</sub>	47.322	47.707	47.717	
Range	0.577	1.044	1.093	

nine schemes with various process parameters based on the orthogonal design method representing nine diferent conditions are given in Table [2](#page-5-0). Subsequently, four workpieces for each scheme were prepared, and the RCT-B tests were conducted on a H1F200S servo press. The press speed was set at 1.65 mm/s, and four reductions in height (approximately 20%, 22.5%, 25%, and 30%) were employed.

## **4.2 Evaluation index**

After all the RCT-B tests in Table [2](#page-5-0) were completed, the outer diameter of the outer boss and the height of the ring workpiece were measured. The reduction in outer diameter of boss and the reduction in height were calculated. The corresponding data were plotted on the calibration curves, as shown in Fig. [8](#page-5-1). The data points are relatively scattered on the diagram, suggesting the friction condition was considerably afected by the selected process parameters.

To facilitate the subsequent optimization, the ratio between the reduction in outer diameter of the boss and the reduction in height was frst calculated for each compressed workpiece, and then the average ratio of four testing workpieces in each scheme was defned as an evaluation index *θ*, as summarized in Eq. ([2\)](#page-5-2). The larger the  $\theta$  value, the smaller the friction factor and the more favorable the lubricating condition at the interface between the testing workpiece and the compression tools. All the values of evaluation index in diferent schemes are listed in Table [2.](#page-5-0)

<span id="page-5-2"></span>
$$
\theta = \frac{1}{4} \Sigma R_{\text{B}n} / R_{\text{H}n},\tag{2}
$$

where *n* is the number of the testing workpiece,  $R_{\text{Bn}}$  is the reduction in outer diameter of boss corresponding to the *n*th workpiece, and  $R_{Hn}$  is the reduction in height of the *n*th workpiece.

#### **4.3 Optimal results**

By employing the Taguchi method, the mean values *k*1–*k*3 of the evaluation index *θ* of every process parameter at the three levels were calculated and recorded in Table [2](#page-5-0). Regarding the curing temperature, the  $\theta$  value was first decreased and then increased up to the maximum at the third level. As for curing time and number of sandblasting cycles, the *θ* values frst increased and then decreased, and the maximums were reached at the second level. Thus, the optimized specifc values of all the process parameters were determined: number of sandblasting cycles = 5, curing time =  $2 h$ , and curing temperature =  $160^{\circ}$ C.



<span id="page-5-1"></span>**Fig. 8** Testing results: **a** schemes 1–5 and optimized scheme; **b** schemes 6–9 and optimized scheme

Based on the optimal process parameters, three testing workpieces were prepared and also evaluated by the RCT-B method. Similarly, the relative reduction in the outer diameter of the boss and reduction in height were obtained and plotted in Fig. [7](#page-4-2). After optimization, the evaluation index *θ* reached 49.62%, and it was larger than that of all the sampling schemes, which suggests the optimization process was successful. From Fig. [7,](#page-4-2) the friction factor of the optimal scheme could be calibrated as 0.1. In addition, the plotted data points fit the calibration curve of  $m = 0.1$  well, which indicates favorable stability of the mixed composite as a lubricant in cold forming after optimization.

# <span id="page-6-0"></span>**5 Limitation analysis of the mixed composites as forming lubricant**

As mentioned above, the preparation process of the mixed PTFE/Nomex/phenolic composites working as a lubricant for cold forming process has been optimized by the RCT tests under a lower forming speed (1.65 mm/s) and relatively smaller deformation degree (less than 30% of reduction in height). So, it will be interesting to find out the lubricate limit of the optimal mixed composites in cold forming. Here, the macro and micro status of the surface state of the fabric-composite lubricant after deformation will be furtherly investigated, and the unique non-uniformity, non-continuity, and complexity of failure mode of the mixed composites will be discussed.

In this condition, 6 testing schemes with a maximum reduction in height of 43.825% under larger press speeds of 16.50 and 41.25 mm/s were frst tried. As shown in Table [3,](#page-6-1) the average surface expansion ratio of the end surface of testing workpiece in each scheme was calculated; the maximum value reached 1.595. The surface expansion ratio, which also can refect the ability of the mixed composite to expand along with the surface of the ring specimen, can be used as an index to evaluate the reliability of the fabriccomposite lubricant.

As shown in Fig. [9](#page-6-2), no visible surface defects could be observed by naked eyes. Here, the "good" was recorded for the status of the fabric-composite lubricant in Table [3.](#page-6-1) Defnitely, the fabric-composite lubricant expands along the end surface of the ring specimen as the reduction in height increased. The vertical load acts on the phenolic resin in direct contact with upper die to promote it to move along the radial direction on the surface of the ring specimen. The strain diference between the phenolic matrix and the PTFE and Nomex fbers on the contact surface is formed, resulting in interfacial shear stress to make the fbers have the tendency of sliding, while the chemical bonding force and mechanical interlocking force between the fbers and matrix limit their relative motion.

Before compression, the existence of several voids in which bundles of fbers interweave demonstrates poor wettability of phenolic resin to fabric, which is detrimental for mechanical properties and wear resistance of composite, as depicted in Fig. [10](#page-7-0)a. After compression of the workpieces, the polymer macromolecules were compressed to smooth plane, and the initial micro gaps at the boundaries were generally flled. The matrix bonded well with the fabric despite a small number of matrix cracks were observed, as depicted in Fig. [10b](#page-7-0). Overall, the surface of the fabric composite retained favorable integrity and could exert a lubricating efect during the stainless-steel deformation.

To investigate failure modes of the fabric-composite lubricant, the RCT-B tests under much larger reduction in height and much higher press speed would be furtherly performed. The effects of the reduction in height and the press speed would be investigated in details. During the further experimental tests, the load-time curves recorded by the press were extracted, and the contact pressure between the tool and the workpiece with the fabric-composite lubricant was calculated as the maximum load divided by the end surface area of the workpiece after compression of each test.

<span id="page-6-1"></span>

<span id="page-6-2"></span>

**Fig. 9** The compressed workpieces corresponding to schemes S1–S6

<span id="page-7-0"></span>



<span id="page-7-1"></span>**Table 4** Tests under diferent reduction in height



Meanwhile, the temperature of the workpiece during the testing process was tried to be captured by a forward-looking infrared thermal imager, and the maximum temperature of each test was recorded for later analysis.

#### <span id="page-7-3"></span>**5.1 Effect of the reduction in height**

To investigate the efect of the reduction in height, diferent test schemes were designed with the reductions in height from 44.402 to 57.286% at a press speed of 82.50 mm/s, as listed in Table [4](#page-7-1). The surface expansion ratio was increased as fast as the reduction in height was increased, and the maximum value reached 2.0, as shown in Fig. [11](#page-7-2)a. The contact pressure was increased gradually as the reduction in height as expected, cause that it is always increased as the degree of plastic deformation due to the strain hardening efect. The average contact pressure was increased from 962.52 to 1063.40 MPa, as shown in Fig. [11a](#page-7-2). In cold forming, part of the plastic deformation work is transformed into heat energy, and the temperature of the workpiece is positively correlated with the degree of deformation. Here, the highest temperature of the tested specimen was 156.5 °C, as shown in Fig. [11b](#page-7-2).

As shown in Fig. [12](#page-8-0), the representative local appearance of the workpiece surface after compression with reduction in height of 44.402%, including the regions near the inner diameter, the middle, and near the outer diameter of the ring workpiece, was observed by a microscope. The diferent state of phenolic resin at three regions along the radius of workpiece was presented. As reported by Male and Depierre [[52\]](#page-12-29), there is a so called "neutral plane" during the ring compression, and the material flows both



<span id="page-7-2"></span>**Fig. 11** The relationship of surface expansion ratio and contact pressure (**a**), and workpiece temperature (**b**) with the reduction in height



<span id="page-8-0"></span>**Fig. 12** Local appearance of the workpiece surface after compression with reduction in height  $R_H = 44.4\%$ , including the region near the inner diameter (**a**), the middle (**b**), and near the outer diameter (**c**) of the ring

inward and outward from the plane under high friction while the material main flows outward under lower friction along the radial direction. In this case, the determined friction factor of the fabric-composite lubricant was 0.1, so the material mainly fowed outward. Therefore, during the compression the semi-solid phenolic resin was squeezed outwards along the radial direction with the material fow of stainless-steel specimen, the original uniform distribution was changed to random distribution (Fig. [12](#page-8-0)a, b), and much more phenolic resin was accumulated on the outer edge, as seen in Fig. [12](#page-8-0)c.

Regarding this random distribution phenomenon, one of the main contributions comes from the radially distributed interfacial shear stress acting on the fabric composite which was woven in a regular array (Fig. [2\)](#page-2-0). Another contribution should be the diferent bonding properties to phenolic resin matrix between the PTFE and Nomex fbers, the interface between the PTFE fbers and phenolic resin matrix seems relatively easy to debond in the compression process. What is more, the color of accumulated phenolic resin layer in Fig. [12](#page-8-0)c shows a slight char in somewhat [[53\]](#page-13-0). As analyzed in Ref [\[54\]](#page-13-1), phenolic resins are relatively stable up to about 200 to 250 °C and start to char slowly at the elevated temperature. During the test, the captured temperature around 128  $\degree$ C (Fig. [11](#page-7-2)) was relatively lower because the higher flash temperature [\[55](#page-13-2)] was not measured and heat transfer from billet to tools before the temperature capturing.

As further observed in Fig. [12](#page-8-0)b, some PTFE fbers in brown were exposed because of relatively smaller bonding strength with the matrix, and the crack perpendicular to the direction of the fbers locally occurred due to interfacial shear stress [[56](#page-13-3)]. When the reduction in height increased, the local cracks of fbers were coupled and extended to form macroscopic and continuous cracks distributed along the circumferential direction of the ring specimen, intensifying the further deterioration of the fabric composite. Therefore, fshscale cracks appeared with a height in reduction of 45.619%, as shown in Fig. [13a](#page-8-1). Excessive pulled-out and cut-off fibers and numerous cracks with fsh-scale morphology presented on the surface of the fabric-composite lubricant. As observed in Fig. [13](#page-8-1)b, the serious broken and deboned fbers showed like a hole. Some wear debris could be found around the hole, and the broken fbers presented a natural bending shape.

Under the conditions of microscopic observation, a lot of small cracks are there. However, in macro-scale, the fabriccomposite lubricant maintains integrity and protects the end faces of the stainless-steel specimen, providing lubrication effect during compression process (Fig. [14](#page-9-0)a–c).

When the reduction in height increased to 54.220%, the fabric-composite lubricant could not adequately cover the ring specimen (Fig. [14](#page-9-0)d), and the outer edge of the stainlesssteel specimen was exposed. The hysteresis phenomenon refects that the surface expansion of the fabric-composite

<span id="page-8-1"></span>**Fig. 13** SEM observation of composite surface at the height in reduction of 45.619%: **a** fshscale cracks and **b** hole



 165.00 40.684 952.37 Good 247.50 40.684 939.85 Peel of 5 288.75 40.684 947.90 Peel off 313.50 40.684 941.64 Delamination 330.00 40.684 938.06 Delamination 346.50 40.684 946.11 Delamination



**Fig. 14** End faces and sides of workpiece after deformation under various reductions in height at a press speed of 82.50 mm/s

<span id="page-9-0"></span>lubricant could not keep up with that of stainless steel at the interface. As shown in Fig. [14](#page-9-0)e, the hysteresis phenomenon became more serious when the reduction in height reached 57.286%, and the shining outer edge on ring specimen was observed, resulting in metal-to-metal direct contact between the tool and the specimen. As the reduction in height increased further, signifcant hysteresis phenomenon and local peel off occurred, as shown in Fig. [14f](#page-9-0). Regarding the cause of the peel off defect, the fine crack defects formed during the compression were propagated by further deformation especially in the area of high shearing stress. In the peel-off zone, the severe friction condition could be generated because of the direct metal-to-metal contact.

## **5.2 Effect of press speed**

As discussed in Sect. [5.1](#page-7-3), the variation of the reduction in height of the RCT-B tests, followed with the change of the contact pressure and the surface expansion ratio, has been studied. Here, the other characteristic parameter of tribological loads, the relative velocity between tool and workpiece, will be further studied. Defnitely, the relative velocity between tool and workpiece could be directly changed by the press speed. To investigate the efect of press speed, test schemes were arranged with press speeds from 107.25 to 346.50 mm/s at a reduction in height of 40.684%, as given in Table [5](#page-9-1). The captured temperatures of workpieces after compression fuctuated between 134.3 °C and 138.2 °C. It was dominated by the reduction in height, and the effect of the press speed was not significant. The similar slight variation of the measured contact pressure

<span id="page-9-1"></span>

mainly depended on the reduction height, varied from 938.06 to 952.37 MPa, was recorded in Table [5.](#page-9-1) From the appearance of the compressed workpieces, there were no macro defects on the surface of the fabric-composite lubricant even when the press speed reached 165 mm/s.

When the press speed was set at 247.50 mm/s, a large amount of fabric-composite lubricant peeled off at the outer edge of the ring specimen, resulting in a thickness reduction and a diferent surface topography at the damaged zone, as seen from Fig. [15a](#page-9-2). According to the SEM image in Fig. [16](#page-10-1)a, the peel off zone seems smoother after shearing and the other zone shows more cracks. A higher forming speed will introduce a larger relative velocity at the interface, and a larger sliding speed will result in a larger coefficient of friction [\[57](#page-13-4)], and the frictional stress will also become larger, then severe shearing stress generates at the interface. The accumulation of fber cracks and the debonding between fbers and matrix was enhanced by high-level shear stress, and then a large area of peel off took place.

At a higher press speed of 313.50 mm/s, the lubricant layer separated from the steel specimen along the circumference of a specifc radius, and the delamination phenomenon occurred, as shown in Fig. [15](#page-9-2)b. When the press speed was increased up to 346.50 mm/s, delamination became more prominent,

<span id="page-9-2"></span>

**Fig. 15** End faces and sides of compressed workpiece under various press speeds at a reduction in height of 40.68%

<span id="page-10-1"></span>**Fig. 16** SEM images of composite surface under various press speeds, including local peel off zone (a), separation of lubricating layer and the adhesive surface (**b**), and debris (**c**)



(a)  $v=247.50$  mm/s (50X)

(b)  $v=330.00$  mm/s (30X)

(c)  $v=330.00$ mm/s (250X)

and a large area of metal was exposed due to the considerable warping of the lubricating layer, as shown in Fig. [15c](#page-9-2). In the outer area of the ring specimen, the fabric-composite lubricant layer was sheared off and broken into pieces and was squeezed outside the surface.

When the specimen surface compressed with a high press speed of 330.00 mm/s was investigated, the signifcant local delamination phenomenon was easily observed, as shown in Fig. [16](#page-10-1)b. The delamination becomes one of the major failure modes endangering the reliability of fabric-composite lubricants [\[58\]](#page-13-5). The mechanism of delamination is relatively complex, involving highly anisotropic, high stress intensities, and interface discontinuities. Interlaminar stress is a major component of the complex three-dimensional stress leading to delamination, which is caused by the mismatch of Poisson efect caused by diferent fber orientations [\[59](#page-13-6), [60\]](#page-13-7). High relative speed at the interface increases the risk of delamination defects because of the high level of interlaminar stress. Due to the inherently weak interfacial strength and fracture toughness of the matrix, delamination tends to occur in the resin enriched area between the fabric composite lubricant and the ring specimen, but can also occur at geometric discontinuities of the fbers. Delamination results in the appearance of a large number of abrasive particles, mainly including PTFE and Nomex fber debris, resin, and trace metal abrasive debris (see Fig. [16c](#page-10-1)), among which small-size PTFE debris can easily be trapped in surface pits as a secondary source of lubricant for repairing damaged surfaces [\[61\]](#page-13-8). When severe hysteresis or delamination occurs during the deformation process, the integrity of the fabric-composite lubricant is severely damaged. The corresponding lubricating efect is weakened, and a non-uniform friction will be created, resulting in an inclination of the outer boss of the ring specimen. The strong inclination could be easily found in Fig. [15](#page-9-2)c, and this inclination of the outer boss was already proved to be indicative of poor friction condition in cold forging [[45\]](#page-12-22).

## <span id="page-10-0"></span>**6 Summary and outlook**

The spherical plain bearings are always assembled in mass production by using cold extrusion process, and the fabric-composite liner sticking on the outer ring also works as a lubricant during the cold forming of the outer ring. Therefore, the tribological behavior of a PTFE/Nomex/ phenolic composite in cold forming condition was investigated using the RCT-B method. To guide the application in industry, the lubricate limit and the failure modes were also explored. The main conclusions can be summarized as follows:

- 1. The average ratio of the reduction in outer diameter of the boss to the reduction in height was defned as the evaluation index, and the adjustable parameters including number of sandblasting cycles, curing time, and curing temperature were selected as design variables, then an optimal group of process parameters were determined: number of sandblasting cycles=5, curing time = 2 h, and curing temperature =  $160$  °C. After optimization, the maximum evaluation index reached 49.62, and the friction factor of the fabric-composite lubricant was stabilized at 0.1.
- 2. To explore the lubricate limit of the fabric-composite lubricant, the tribological loads including contact pressure, surface expansion ratio, and the relative velocity between tool and workpiece were changed by the variation of the reduction in height and press speed. From the tested results, the fabric-composite lubricant could keep favorable surface integrity when the reduction in height is smaller than 48.6% and the press speed is lower than 165.00 mm/s. In these cases, although there are some local cracks of fbers but no visible surface defects could be observed by naked eyes, and the end faces of the stainless-steel specimen are well protected.
- 3. Three mainly failure modes, including hysteresis, peel off and delamination, were found on the fabriccomposite lubricant in cold forming. The hysteresis phenomenon refects that the surface expansion of the fabric-composite lubricant could not keep up with that of stainless steel at the interface. The peel off defect is mainly caused by that the fne cracks formed during the compression are propagated by further deformation especially in the area of high shear stress. The larger the relative speed at the interface, the higher the interlami-

nar stress generates, and the higher the risk of delamination defects on the fabric-composite lubricant.

4. Considering the limitation of the PTFE/Nomex/phenolic composite working as lubricant in cold forming, it would be interesting to fnd out the solution to improve the comprehensive tribological performance to endure higher contact pressure and larger surface expansion. Due to the facts that fne cracks could be found on the fabric-composite lubricant during the cold compression, the effect of pre-strain on the wear property in service of the fabric composites which are applied as the selflubricant liner in the spherical plain bearings will be investigated in the upcoming works.

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# **Declarations**

**Ethics approval** Not applicable.

**Consent to participate** Not applicable.

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**Competing interests** The authors declare no competing interests.

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