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The efect of topology on the quasi‑static and dynamic behaviour of SLM AlSi10Mg lattice structures

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Abstract

Additive manufacturing (AM) techniques such as selective laser melting (SLM) enable the fabrication of complex metallic lattice structures. By tuning geometric and topological parameters, these structures can be manufactured to exhibit a range of useful properties, including excellent strength-to-weight ratios and energy absorption capabilities. While the efects of these parameters on various aspects of AM lattice performance have been previously studied, such as the efects of manufacturability, material selection and geometric parameters on the quasi-static performance of AM lattice structures, the efect of topology on the dynamic behaviour of SLM AlSi10Mg lattice structures remains relatively unexplored. Lattice structure specimens with fve diferent topologies were manufactured using SLM AlSi10Mg and tested under quasi-static and dynamic loading conditions. The tested topologies were body-centred cubic with (BCCZ) and without (BCC) z-struts; face-centred cubic with (FCCZ) and without (FCC) z-struts; and body and face-centred cubic with z-struts (FBCCZ). A numerical model was developed to investigate failure modes and collapse mechanisms. Specimens were found to fail by the emergence of diagonal shear planes, and the orientation of which was dependent on topology, due to the uneven concentration of stress in struts across the structure. No signifcant rate sensitivity was identifed for any of the tested topologies in the range of tested strain rates. The FCCZ topology was demonstrated to provide the greatest efficiency in terms of both strength-to-weight and stifness-to-weight ratios. These results assist in the characterisation of the dynamic behaviour of SLM AlSi10Mg lattice structures and contribute to their further commercialisation.

Keywords SLM · Lattice structures · Dynamic behaviour · Additive manufacture · AlSi10Mg · Mechanical properties

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1 Introduction

Additive manufacturing (AM) methods, such as selective laser melting (SLM), enable the efficient fabrication of complex and intricate structures that cannot feasibly be obtained by conventional manufacturing methods [[1](#page-18-0)], including lattice structures. Lattice structures are a form of cellular structure that are diferentiated from other cellular structures such as metallic foams by the ordered arrangement of their constituent cells [[2\]](#page-18-1). The usefulness of cellular structures has been understood for hundreds of years [[3\]](#page-18-2), but with the increasing development of AM, metallic lattice structures have recently received much research attention due to their potential for biomedical [[4](#page-18-3)], aero-space [[5](#page-18-4)] and automotive [\[6\]](#page-18-5) applications. Lattice structures may be considered meta-materials with properties and behaviours that are distinct, though related to those of their parent material [\[7](#page-18-6)]. These properties are dictated

by the geometry (size and shape of the structure and its structural elements) and topology (the arrangement and connectivity of structural elements) [[8\]](#page-18-7).

Many studies have been conducted on AM lattice structures [[9\]](#page-18-8), and it has been found that by tuning lattice geometry and topology, lattice structures can be manufactured to achieve a broad range of properties [[10](#page-18-9)] that cannot be achieved by their parent bulk material [[11\]](#page-18-10), including thermal $[12]$ $[12]$ $[12]$, acoustic $[13]$ and mechanical properties $[14]$ $[14]$. Due to their potential for a range of applications and high commercial value, various studies have sought to characterise the quasi-static mechanical properties of lattice structures, including:

- Yan et al. [[15](#page-18-14)] fabricated stainless steel lattice structures with gyroid unit cells to evaluate their manufacturability and performance. They found that structures with 2–8-mm unit cells could be manufactured without the need for support structures and with good conformity to the intended geometry, and that decreasing unit cell size increased yield strength and modulus.
- Leary et al. $[16]$ $[16]$ sought to define the manufacturability of particular strut-based topologies and characterise the mechanical performance of lattice structures fabricated in AlSi12Mg using SLM. Lower limits of manufacturability in terms of strut diameter and inclination angle were identifed, and the general behaviour of diferent unit cell topologies was characterised.
- Leary et al. [[8](#page-18-7)] also investigated the mechanical response, deformation characteristics and failure modes of SLM Inconel 625 lattice structures. The ductility of the material enabled unique insight into transitions between bending and stretch-dominated behaviours for certain cell topologies.
- Köhnen et al. [[17](#page-18-16)] studied stainless steel lattice structures with two diferent topologies they referred to as "f2cc,z" and "hollow spherical" under tensile, compressive and cyclical loads. The f2cc,z specimens were found to deform in a stretch-dominated manner, whereas the hollow spherical specimens displayed bending-dominated deformation behaviour, demonstrating the efects of lattice unit cell topology on the plastic behaviour of lattice structures. Specimen geometry was also found to afect the quality of manufacturing outcomes and fatigue performance.

Although dynamic performance is important for many applications of AM lattice structures, such as fatigue loading of medical implants $[18]$ $[18]$, there is limited data available on their dynamic behaviour. A limited number of studies have sought to defne the dynamic behaviour of metallic cellular structures, including:

- Harris et al. [[19](#page-19-0)] investigated the dynamic compressive performance of stainless steel cellular structures manufactured by selective laser melting (SLM). Hybrid cellular structures were fabricated by increasing the porosity of honeycomb structures by replacing walls with lattice struts, which were then dynamically tested using a split-Hopkinson pressure bar apparatus at strain rates between 5×10^3 and 15×10^3 s⁻¹. Lattice specimens exhibited increased strength at greater strain rates, which was attributed to both the stabilising efect of lateral inertia and wave propagation efects, with wave propagation only occurring above 10^4 s⁻¹.
- Smith et al. [\[20](#page-19-1)] fabricated stainless steel lattice structures with BCC and BCCZ topologies using SLM. Specimens were tested at quasi-static strain rates followed by blast tests at strain rates between 210 and 1710s−1. Under quasi-static conditions, BCC specimens failed progressively, whereas the collapse of BCCZ specimens was buckling-dominated. Specimen blast response was found to have a linear dependence on the applied impulse due to the strain rate sensitivity of the material, and collapse modes were found to be similar between quasi-static and blast specimens.
- McKown et al. [[21\]](#page-19-2) tested stainless steel lattice structures with BCC and BCCZ unit cells under quasi-static and dynamic compressive loads at up to 150 s^{-1} , followed by blast tests at 450 to 1815s−1. Yield stress was found to be moderately sensitive to strain rate, with a 20% increase over the tested range, due to the rate sensitivity of the parent material and very high strain rates. However, microinertia efects were negligible due to the absence of lateral reinforcement in the tested topologies. Failure modes were found to be similar between quasi-static and blast tests.
- Tancogne-Dejean et al. [[22\]](#page-19-3) conducted numerical, quasistatic, and dynamic tests on 316L stainless steel lattice structures with octet-truss unit cell topology. Dynamic testing was performed using a split-Hopkinson pressure bar setup with an average strain rate of approximately 1000 s −1. Lattice specimen strength was found to be signifcantly dependent on strain rate, though the observed rate sensitivity was close to that of the parent material, suggesting this sensitivity was related to material properties rather than structural phenomena.
- Ruan et al. [\[23\]](#page-19-4) studied the compressive behaviour of closed-cell aluminium foams over strain rates ranging from 10^{-3} to 101 s⁻¹ with relative densities between 5 and 20%. Plateau stress was found to be independent of strain rate, and instead related to relative density by a positive power relationship. Specimen failure did not occur uniformly, but in bands, with each subsequent band failing once the previous band has completely crushed.

However, these studies have focused on small selections of unit cell topologies. In this paper, we aim to test a wide range of lattice topologies to elucidate the infuence of strut arrangement and connectivity on the dynamic response of SLM AlSi10Mg lattice structures.

The mechanical performance of cellular structures, including lattice structures, can be generally categorised as either bending-dominated or stretch-dominated [[24](#page-19-5)]. Stretch-dominated structures are characterised by an initial high peak stress followed by a lower plateau stress, whereas bending-dominated structures have a lower initial stifness followed by a relatively constant plateau stress. Due to these behaviours, stretch-dominated structures are generally stifer and stronger than bending-dominated structures, whereas bending-dominated structures are more compliant and have a more consistent stress–strain response [\[25\]](#page-19-6).

The Maxwell criterion [[26](#page-19-7)] provides a means of predicting whether a lattice structure will behave in a bending-dominated or stretch-dominated manner [[8\]](#page-18-7) based on the number of struts (*s*) and nodes (*n*) of a lattice unit cell for a given dimensionality (Eq. 1 and Eq. [2\)](#page-2-0) [\[27\]](#page-19-8). If $M < 0$, the structure is considered *under-stiff*, meaning there are insufficient struts to equilibrate the bending moments at nodes, inducing bending stresses within struts, and resulting in bending-dominated behaviour, whereas if $M=0$, the structure is considered *just-stif*, or if *M*>0, it is considered *over-stif*. In these cases, an adequate number of struts are present to equilibrate bending moments at nodes, meaning struts only experience axial stresses, resulting in stretch-dominated behaviour [\[28](#page-19-9)].

$$
M = s - 2n + 3(2D \text{ truss})
$$
 (1)

$$
M = s - 3n + 6(3D \text{ truss})
$$
 (2)

However, the Maxwell criterion is understood to be a necessary but insufficient condition for stretch-dominated behaviour [\[26](#page-19-7)], and previous studies of lattice structures have found certain lattice topologies to be capable of behaving in a stretchdominated manner, despite being under or just-stif according to the Maxwell criterion [\[8](#page-18-7), [16](#page-18-15), [29](#page-19-10), [30](#page-19-11)]. In this study, lattice structure specimens with fve diferent topologies are tested under quasi-static and dynamic loads to better understand the topological and dynamic efects on their mechanical behaviour.

In this work, lattice specimens with BCC, BCCZ, FCC, FCCZ, and FBCCZ topologies and all geometric parameters kept constant were tested under quasi-static and dynamic loading conditions, with comparison to numerical models of lattice specimens to identify topological and dynamic efects on the compressive mechanical behaviour of SLM AlSi10Mg lattice structures. The results of quasi-static and dynamic testing are discussed, with reference to stress–strain behaviour and qualitative analysis of photography acquired during testing and compared with results of the numerical

model. These fndings assist in the comprehensive characterisation of lattice structures in both quasi-static and dynamic loading regimes and facilitate the further commercialisation of AM lattice structures.

2 Specimen design and manufacture

Certain technical limitations exist regarding the manufacturability of components using SLM and aluminium powder. For example, the relatively high absorptivity of aluminium powder due to internal refections [[31](#page-19-12)] combined with its relatively high difusivity means greater power is required for SLM than other metals, such as titanium [\[16](#page-18-15)]. AlSi10Mg components produced by SLM are also prone to anisotropic mechanical properties [\[32\]](#page-19-13). As a result, although SLM ena-bles the fabrication of complex, intricate structures [[33\]](#page-19-14), it is also subject to manufacturability limitations. To ensure feasibility for this study, specimen designs were based on the manufacturability findings of a previous study [[16\]](#page-18-15).

2.1 Topology and geometry

To investigate the interaction between topological and dynamic efects on the mechanical behaviour of AlSi10Mg lattice structures, specimens were designed with fve diferent unit cell topologies, referred to as BCC (body-centred cubic), BCCZ (body-centred cubic with z-struts), FCC (facecentred cubic), FCCZ (face-centred cubic with z-struts) and FBCCZ (face and body-centred cubic with z-struts). The topological details and relative densities of the CAD models unit cells considered here are presented in Table [1.](#page-3-0) All specimens were designed with 5×7.5 mm cubic cells in each orthogonal direction with 1-mm diameter struts. An example of the CAD geometry of a BCC specimen, including strut diameter and cell size, is presented in Fig. [1.](#page-3-1)

2.2 Manufacture of specimens

Lattice specimens were fabricated using a SLM Solutions 400 W dual laser powder bed SLM 500. The processing parameters used are presented in Table [2](#page-3-2). The chemical composition of the AlSi10Mg powder used is presented in Table [3,](#page-4-0) and the powder particle size distribution is presented in Fig. [2](#page-4-1).

3 Mechanical testing

To enable comparison between quasi-static and dynamic behaviour, as-manufactured lattice specimens were tested under both quasi-static and dynamic loadings. Strain was calculated based on crosshead displacement and specimen height (37.5 mm), and compressive stress was calculated by

Topology	Body- centred cubic (BCC)	Body- centred cubic with z- struts (BCCZ)	Face-centred cubic (FCC)	Face-centred cubic with z- struts (FCCZ)	Face and body- centred cubic with z-struts (FBCCZ)
Image					
Struts (s)	8	12	16	20	28
Nodes (n)	9	9	12	12	13
Maxwell number (M)	-13	-9	-14	-10	-5
Struts oriented in load direction	No	Yes	No	Yes	Yes
Relative density of CAD model (%)	8.5	9.6	6.99	8.01	15.58

Table 1 Lattice specimen unit cell topological details

Fig. 1 CAD representation of BCC lattice specimen, including defnition of geometric parameters

Table 2 SLM processing parameters

dividing the measured load by the nominal cross-sectional area of the specimens (37.5^2 mm^2) .

3.1 Quasi‑static testing

Two specimens of each topology were tested under quasistatic loading using an MTS Landmark with a 100-kN load cell. Specimens were loaded at 10^{-3} s⁻¹ (2.25 mm/min). Time-lapse photography with a frequency of four images per minute was used to record all experiments and identify failure modes.

Fig. 2 Powder particle size distribution acquired by spectrographic analysis

Young's modulus was identifed from the gradient of the linear region of the quasi-static stress–strain curve, yield stress was quantified using the 0.2% strain offset method and ultimate strength was the maximum stress carried by specimens before failure.

Ashby et al. [[34](#page-19-15)] suggest that measuring the Young's modulus of a cellular structure in the conventional manner—the stress–strain ratio during the linear region of elastic deformation—is unrepresentative of the structure's functional stifness, due to local plasticity occurring in the structure at stresses well before yielding. Instead, they recommend that measurement of a cellular structure's modulus be taken from the slope of the unloading curve. For this reason, the unloading modulus of one specimen of each topology was measured at 1% and 2% strain, as well as the conventional Young's modulus, as has been performed in previous studies [[8\]](#page-18-7).

3.2 Dynamic testing

Dynamic testing was performed using an Instron VHS8800 testing machine with a loading capacity of 80 kN. Three specimens of each topology were tested at 5 m/s (corresponding to a strain rate of 133.33 s⁻¹), which was maintained constant during the testing procedure. High-speed photography was used to capture images during testing for identifcation of associated failure modes.

A meaningful measurement of Young's modulus, and therefore yield stress, cannot be extracted from a dynamic stress–strain curve in the same manner as a quasi-static test due to the non-uniform deformation associated with dynamic loading. As a result, dynamic results were analysed in terms of ultimate compressive strength.

3.3 Normalised properties

It is understood that the mechanical performance of cellular structures is dependent on their relative densities [[24](#page-19-5)]. The lattice specimens considered in this study were designed with the same geometric parameters (cell size and strut diameter) with the only variation in topology. However, these geometric and topological parameters afect the relative density of a lattice structure and therefore the mechanical performance. To compare normalised properties, specifc strength and specifc modulus were calculated by dividing specimen properties by their density.

The relative density (ρ^*/ρ_s) of each specimen was calculated based on their mass (m_s) , their nominal volume (V_s) and the density of the material (ρ_s) , which was assumed to be 2.67 g/cm³ (Eq. [3](#page-4-2)).

$$
\frac{\rho^*}{\rho_s} = \frac{m_s}{V_s \rho_s} \tag{3}
$$

3.4 Energy absorption

The energy absorbed during testing per unit volume (W_V) and per unit mass (W_M) , based on the specimen density (ρ) , were calculated using Eq. 4 and Eq. 5 respectively. Calculations were based on stress (σ) and strain (ε) behaviour, measured between the start of the test and strain at failure (ε_f) .

$$
W_V = \int_{0}^{\epsilon_f} \sigma d\varepsilon \tag{4}
$$

$$
W_M = \frac{1}{\rho} \int_{0}^{\epsilon_f} \sigma d\varepsilon \tag{5}
$$

4 Finite element modelling

To investigate collapse behaviour and deformation mechanisms, a fnite element (FE) model was developed to replicate the behaviour of the candidate lattice structures (Fig. [3\)](#page-5-0) using the ABAQUS static and dynamic explicit solvers. Lattice struts were represented using 2-node linear beam elements in space (B31) with circular cross-sections that matched the experimental specimen's topological and geometric parameters (1-mm diameter, 7.5-mm cell size, 5 cells in *x*, *y*, and *z* directions). Beam elements provide a computa-tionally efficient means of modelling lattice structures [[35,](#page-19-16) [36](#page-19-17)] and are appropriate to modelling applications including geometric nonlinearity and material plasticity [[37](#page-19-18)], as is the case when modelling lattice structures. However, there are known limitations of modelling lattice structures using beam elements, such as the inability to directly model geometric defects that result from AM fabrication methods [[37\]](#page-19-18), and do not account for overlapping volumes at strut intersection [\[35](#page-19-16)]. Despite these disadvantages, and due to the exceptional computational efficiency of beam elements [[38\]](#page-19-19) compared to alternative continuum elements [\[39](#page-19-20)], beam elements have been used to efectively model the compressive behaviour of lattice structures in several previous studies [[35](#page-19-16)[–41](#page-19-21)].

Specimens were crushed between plates represented by rigid elements (R3D4) that moved at a constant prescribed velocity with a unit mass. Quasi-static tests were replicated using the static solver and dynamic tests using the dynamic explicit solver with a strain rate of 100 s⁻¹. Mesh density analysis was performed, and a global element size of 10% of cell size (0.75 mm) was found to lead to convergence of results for all topologies.

General contact was set for the whole model, to enable load transfer between the plates and lattice, and to constrain the displacement of struts by contact with each other. Normal behaviour was defned as "hard contact," and friction between the specimens and the rigid plates was modelled by defning the tangential behaviour as "penalty" with a coefficient of friction of 0.1.

An elastic/plastic material model was defined using parameters identifed in a previous study on SLM AlSi10Mg [\[32\]](#page-19-13). Elastic parameters were Young's modulus = 68 GPa and Poisson's ratio $=0.3$. Material plasticity was defined using the Johnson–Cook model (Eq. [6\)](#page-5-1). Constitutive relationships between build orientation angle and strain hardening parameters of SLM AlSi10Mg were previously identifed, and so diferent material models were applied depending on the orientation of struts (Table [4\)](#page-5-2), which was found to reduce error between the experimental and FE model results.

$$
\sigma = \left[A + B\varepsilon^n\right] \left[1 + C\ln(\varepsilon^*)\right] \left[1 - T^{*m}\right] \tag{6}
$$

where σ = plastic flow stress; A = material yield strength; $B = \text{strain}$ hardening coefficient; $\varepsilon = \text{strain}$; *n* = strain hardening exponent; $C =$ strain rate sensitivity; $\dot{\epsilon}^* =$ ratio of given strain rate to the reference strain used to determine strain hardening coefficient and exponent; $T^* =$ non-dimensional temperature, function of room temperature, the material's melting temperature and reference temperature used to determine strain hardening coefficient and exponent; $m =$ thermal softening exponent.

The Johnson–Cook model enables the modelling of material strain rate sensitivity. However, previous studies have concluded that SLM AlSi10Mg can be modelled as nonstrain rate sensitive in the $1-100$ s⁻¹ range [\[32,](#page-19-13) [42](#page-19-22), [43\]](#page-19-23), so no stain rate sensitivity value was prescribed $(C=0)$. Temperature was not considered in this study, so temperature parameters were ignored in the material model $(T=0)$.

Table 4 Johnson–Cook plasticity model parameters for vertical and inclined struts [\[32\]](#page-19-13)

Strut orientation	A (MPa)	B(MPa)	n
Vertical	259.6	1.581.59	0.76
Inclined	264.54	1.278.43	0.71

Fig. 3 Finite element model, including **A** undeformed and **B** deformed state. Lattice specimens are represented using beam elements and crushed between rigid plates

5 Results and discussion

To account for the diferent relative densities of specimens and the efect on mechanical performance, the relative densities of the specimens were identifed and are discussed. Quasi-static and dynamic mechanical testing results are compared, and insights provided by the FE model are explored.

5.1 Relative density

The average relative densities of as-manufactured specimens for all topologies are compared with the predicted value based on specimen CAD in Fig. [4.](#page-6-0) As-manufactured relative densities of body-centred topologies (BCC, BCCZ and FBCCZ) were consistently less than those predicted by the CAD, while FCC and FCCZ as-manufactured relative densities were greater than those predicted by the CAD. SLM-manufactured components are prone to shrinkage during manufacture due to the diference in cooling rates between surface and subsurface [\[44\]](#page-19-24), which may explain the reduced density of as-manufactured body-centred specimens. Unlike the BCC topology, the face-centred topologies (FCC, FCCZ and FBCCZ) share struts between cells. However, for unit cells on the outer surfaces of a lattice structure, the volume of struts that are not shared between unit cells is not accounted for, leading to a greater diference between the CAD relative density and that of the as-manufactured specimens. This diference increases with the number of unit cells, and as specimens had many unit cells on their faces (25 cells per face on six faces) this diference was signifcant compared to other topologies. As the FBCCZ includes both FCC and BCC struts, the diference between the idealised CAD and as-manufactured relative densities is due to that not accounting for shared struts on the specimen faces is

Fig. 4 Comparison of CAD and average relative densities of as-manufactured specimens for all topologies. Error bars for as-manufactured specimens indicate one standard deviation

offset by the error due to shrinkage of struts, meaning the relative density of as-manufactured FBCCZ specimens was less than that predicted by the CAD.

5.2 Quasi‑static results

Selected images of lattice specimens during quasi-static testing are presented in Fig. [5.](#page-8-0) When failure occurred, diagonal failure planes emerged (red highlights in Fig. [5\)](#page-8-0) for all topologies. For BCC and BCCZ specimens, the failure plane was diagonal on two of the parallel vertical planes, but horizontal on the perpendicular planes, whereas for FCC and FCCZ specimens, the failure plane was diagonal on all vertical planes (Fig. [6](#page-9-0)), demonstrating that the orientation of failure planes is dependent on topology. For the specimen with z-struts, deformation was concentrated in the layers in contact with the compression plates, but once the z-struts failed (usually on the upper layer which was in contact with the moving plate) failure planes emerged, with their orientation depending on the presence of BCC or FCC struts. The FBCCZ specimens were not as consistent in their failure mechanism, with no single plane of failure emerging, and failure occurring within random cells through the structure. The FBCCZ specimens were also able to hold together postfailure, unlike the other topologies, due to the signifcant number of struts present, suggesting a potential greater damage tolerance of this topology.

Stress–strain curves for the tested specimens are presented in Fig. [7A.](#page-10-0) Specimens with z-strut topologies (BCCZ, FCCZ and FBCCZ) showed greater stifness and strength than those without, with stress–strain curves that suggest a stretch-dominated response, while both BCC and FCC specimens failed at greater strains than their z-strut counterparts, and behaved in a bending-dominated manner, consistent with previous studies [[10,](#page-18-9) [16\]](#page-18-15). This suggests that the inclusion of z-struts increases strength and stifness, though at the cost of reduced compliance.

All specimens behaved in a brittle manner—once the ultimate strength was reached, the specimens catastrophically failed, and experiments were stopped. Unlike the other topologies, FBCCZ specimens failed progressively, as no single failure plane emerged, as shown in Fig. [5](#page-8-0). Asmanufactured SLM AlSi10Mg has previously been found to be brittle [\[45–](#page-19-25)[47\]](#page-19-26), and SLM lattice structures with similar geometries made from diferent materials such as Inconel and Ti6Al4V have been found to be more ductile [\[8](#page-18-7), [30](#page-19-11)]. This suggests the observed brittle behaviour of the lattice structures is due to the brittleness of the material.

Stress–strain curves normalised by specimen density (specifc stress–strain) are presented in Fig. [7B](#page-10-0) and yield, and ultimate strengths normalised by specimen density (specifc strength) are presented in Fig. [7C](#page-10-0). Although

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Fig. 5 Photographs of initial contact, deformation and failure of ◂ quasi-static specimens during testing. Build direction is upwards and loads are vertical. Red highlights indicate failure planes. The failure plane of the BCCZ specimen occurred on the face not visible to the camera

FBCCZ had the greatest yield and ultimate strengths (Fig. [7A\)](#page-10-0), this is partially due to the greater material presence provided by the extra struts, and when mass is accounted for, FCCZ is the most efficient topology in terms of strength-to-weight ratio. Conversely, although the BCC and BCCZ specimens had greater relative density than the FCC and FCCZ topologies respectively, the superior specific strength of the FCC and FCCZ specimens demonstrates that relative density is not the only contributing factor to lattice mechanical performance, and that topology signifcantly afects qualitative and quantitative performance.

An example of the Young's modulus, 1% and 2% unloading moduli are provided in Fig. [8A,](#page-10-1) and these values for all topologies normalised by specimen densities (specifc modulus) are compared in Fig. [8B.](#page-10-1) The Young's modulus of specimen that were and were not unloaded was very similar, though for BCC, BCCZ and FCC specimens the Young's modulus of the unloaded specimens was greater than that without unloading, whereas for the FCCZ and FBCCZ specimens the opposite was observed. This suggests that unloading reduces the Young's modulus of specimens with FCCZ topologies. The unloading moduli were consistently greater than the Young's modulus for all specimens, and the 2% unloading modulus was always greater than the 1% unloading modulus, suggesting stifness increases with an increase in the strain at which the specimen is unloaded. These fndings support Ashby's assertion that unloading modulus is more representative of a structure's stifness due to the similarity between the unloading moduli and their diference to the Young's modulus.

The effect of the inclusion of z-struts on specific modulus is apparent from Fig. [8B](#page-10-1)—z-struts significantly increase the modulus of lattice topologies, as seen by the superior modulus of the BCCZ and FCCZ topologies compared to the BCC and FCC topologies, respectively. Comparison of Fig. [7C](#page-10-0) (specifc strength) and Fig. [8B](#page-10-1) (specifc modulus) shows that z-struts more signifcantly contribute to the stifness of the lattice than strength, as seen by an increase in specifc modulus provided by the inclusion of z-struts, compared to the increase in specifc strength.

5.3 Dynamic results

High-speed photographs of specimens during dynamic testing are presented in Fig. [9.](#page-12-0) Specimens failed due to the

emergence of diagonal shear planes, as was observed during quasi-static testing. Greater deformation was observed on the loaded (top) face of the specimens with z-struts (BCCZ, FCCZ and FBCCZ), whereas deformation was more evenly distributed through the specimen for specimens without z-struts (BCC and FCC). Failure occurred at greater strains under dynamic loading compared to the quasi-static results due to the increased rate of deformation.

Specimens behaved in a brittle manner under dynamic loading—once softening occurred the specimens catastrophically failed—consistent with the brittle behaviour observed during quasi-static testing. This suggests that conservative safety factors would be necessary for the implementation of as-manufactured SLM AlSi10Mg lattice structures to keep their maximum operating stresses within the linear region of their stress–strain curve.

Dynamic stress–strain curves for selected specimens of all topologies are compared in Fig. [10A](#page-12-1). Similar to the quasi-static stress–strain curves (Fig. [8A\)](#page-10-1), it is shown that the inclusion of z-struts increases the stifness and strength but reduces the strain at which failure occurred.

Comparison of dynamic specifc ultimate compressive strength between topologies (Fig. [10B\)](#page-12-1) shows that FCCZ is the most efficient topology in terms of strength-to-weight ratio under dynamic loadings as well as quasi-static loading. The results were very consistent, with only minor standard deviations for all topologies.

5.4 Comparison of quasi‑static and dynamic results

Stress–strain curves of selected quasi-static and dynamic specimens are compared in Fig. [11A.](#page-13-0) Although the general shape of the curves for a given topology is quite similar, the shallower gradient of the dynamic curves after failure shows that specimens failed over a larger strain compared to quasistatic result, due to the increased strain rate of the dynamic experiments. The peak of the curves generally occurred at greater strain under dynamic loading, which is consistent with the greater strain at failure for dynamically tested specimens, as seen by comparison of the strain in Figs. [5](#page-8-0) and [9.](#page-12-0) This suggests that the general behaviour of all topologies is quite similar between quasi-static and dynamic loading regimes, though the increased rate of the dynamic testing means that the certain phenomena, such as yielding, peak stress and failure, occur at greater strains under increased loading rate.

The scatter plot presented in Fig. [11B](#page-13-0) shows the specifc strength of all specimens over the range of tested strain rates. Although most topologies showed very similar specifc strength at both quasi-static and dynamic strain rates, suggesting a lack of rate sensitivity, FCC specimens seemed to show an increase in specifc strength in the dynamic rate range, whereas FCCZ showed a drop in specifc strength at **Fig. 6** Schematic representation of failure plan orientation on faces of BCC and FCC specimens with respect to the applied load (F)

the greater strain rate, compared to the quasi-static response. However, when mean strength values are compared, as presented in Fig. [11C,](#page-13-0) very little diference is observed between the quasi-static results—BCC, FCC and FCCZ had the greatest variation in mean specifc strength over the tested strain rates, with a 5%, 4.5% and 4.6% change in specifc strength, respectively. BCCZ and FBCCZ were the most consistent across the tested strain rates with only 1.7% and 0.5% diferences, respectively. These results suggest that the strength of all topologies is not signifcantly strain rate sensitive.

5.5 Energy absorption

The energy absorbed per unit volume and per unit mass is presented in Fig. [12A](#page-14-0) and B respectively. The energy absorbed per unit volume (W_v) consistently increased with strain rate for all topologies except FCCZ—12% for BCC, 18% for BCCZ, 23% for FCC and 4% for FBCCZ—whereas FCCZ saw a 12% drop in energy absorbed per unit volume. This is due to the decreased ultimate strength of the dynamically tested specimens seen in Fig. [11A,](#page-13-0) leading to an overall reduction in energy absorption in the dynamic strain rate range. Differences in W_v between topologies reflect differences in modulus and strength values discussed above, as all specimens had the same nominal volume.

When energy absorption was normalised by specimen density (energy absorbed per unit mass, W_M , Fig. [12B](#page-14-0)), FCCZ and FBCCZ topologies were found to signifcantly outperform the BCC, BCCZ and FCC topologies—*WM* for BCC, BCCZ and FCC specimens ranged between 0.65 and 0.86 J/g, whereas that for FCCZ and FBCCZ ranged between 1.4 and 1.6 J/g. The inclusion of z-struts in the BCC topology led to a 12% increase in W_M in the quasistatic range, and a 23% increase in the dynamic range, suggesting that the z-struts provide more mass-efficient energy absorption capabilities. However, a more signifcant increase in W_M was observed due to the inclusion of z-struts in the FCC topology—a 140% increase in W_M was observed in the quasi-static range and a 66% increase in the dynamic range. This demonstrates that the inclusion of z-struts has a more signifcant efect on the FCC topology compared to the BCC topology in terms of energy absorption, and further demonstrates the efficiency of the FCCZ topology, as well as the efficiency of strength- and stifness-to-weight ratios. FBCCZ specimens had lower W_M than the FCCZ specimens in the quasi-static range suggesting the presence of both BCC and FCC struts is inefficient, and that impressive energy absorption can be achieved with the FCCZ topology without the need for BCC struts. FBCCZ specimens had the greatest W_M of all specimens, as they were able to hold together for the greatest strain, as previously discussed, due to the greater presence of material within cells, meaning failure occurred over a greater strain, increasing energy absorption performance.

These results further demonstrate that relative density of a lattice structure's unit cell does not alone dictate its performance, and that topology can be used as a coarse design parameter to radically alter the compressive performance of lattice structures.

Again, all topologies showed an increase in W_M with increased strain rate except FCCZ, which showed a mean decrease of 11%. This improved energy absorption performance is due to stress–strain phenomena occurring at higher strains during dynamic deformation, as previously discussed. As yielding, peak strength and failure occur at higher strains under dynamic loading, due to increased rate of deformation, more energy is absorbed over that greater strain. This demonstrates that SLM lattice structures may be effectively deployed for energy absorption applications under dynamic loading, as their energy absorption performance is increased with the increased strain rate.

The energy absorption capabilities of the tested lattice structures were hampered by the brittleness of the

Fig. 7 Quasi-static results, including **A** stress–strain curves, **B** specifc stress–strain curves and **C** specifc yield and ultimate strengths of all tested topologies

specimens in their as-manufactured state, as no plateau stress could be achieved before catastrophic failure upon yielding, even for the BCC specimens—the most compliant topology. Although results demonstrate that energy absorption efficiency is dependent on topology, the

Fig. 8 A Comparison of Young's modulus, 1% and 2% unloading modulus and **B** specifc Young's modulus of the specimens tested without unloading, and the specifc Young's modulus and 1% and 2% moduli of the specimen tested with unloading

brittleness of the specimens suggests that these particular lattice structures would not be very useful for energy absorption applications in their as-manufactured state, as the continuous deformation observed in the plateau region of compliant cellular structures, which is desirable for energy absorption applications, was absent here. However, the brittleness of SLM AlSi10Mg has been reported previously [[48](#page-19-27), [49\]](#page-19-28), and to overcome this heat treatments including annealing and hot isostatic pressing (HIP) [[50\]](#page-19-29) have been shown to improve ductility at the cost of strength [[51\]](#page-19-30) of bulk SLM AlSi10Mg. Furthermore, heat treatments on lattice structures have demonstrated that mechanical performance can be improved due to reduced porosity [[52](#page-19-31)] and even reduce strain rate sen-sitivity [[53](#page-19-32)]. Although the effect of heat treatment on the performance of SLM AlSi10Mg lattice structures is not within the scope of this study, results from the literature suggest the brittleness of the as-manufactured specimens may be overcome by heat treatment, which would improve their energy absorption performance.

5.6 Numerical modelling

Quasi-static and dynamic stress–strain curves extracted from experiments and simulations are compared in Fig. [13](#page-15-0), along

Fig. 9 High-speed photographs of specimens during dynamic testing ◂ at initial contact, during deformation, and after failure. Build direction is upwards and loads are vertical. Red highlights indicate failure planes

with specific Young's modulus. It can be seen that although there was a good match between the curves for the lattice structures without z-struts (BCC and FCC), the stifness of simulations of the structures including z-struts (BCCZ, FCCZ and FBCCZ) was consistently greater than that of the experimental curves. This fnding is consistent with previous studies comparing the simulated and experimental behaviours of lattice structures. For example, Luxner et al. [[35\]](#page-19-16) found that for lattice structures whose response is highly directionally dependent, as is the case when struts oriented in the build direction are included (z-struts), there is greater error between simulation and experimental curves, as these structures are very sensitive to changes in loading direction. Smith et al. [\[36](#page-19-17)] compared simulated and experimental mechanical properties of BCC and BCCZ lattice structures and also found that there was a better match between the stifness of simulations and experiments of BCC lattice structures compared to BCCZ. These results suggest that the accuracy of numerical modelling of lattice structures is highly dependent on the governing deformation mechanisms [\[35](#page-19-16)], and that structures that rely on buckling for failure (stretch-dominated), such as those topologies including z-struts, tend to have greater variation between simulated and experimental results than structures that fail by bending (bending-dominated).

There are both aleatory and epistemic uncertainties associated with experimental testing of AM lattice structures that result in discrepancies between numerical and experimental results. Variation in the intended and as-manufactured geometries is a known limitation of AM technologies [\[54](#page-19-33)], including SLM [\[55](#page-19-34)], that has been found to signifcantly afect the performance of AM lattice structures [[56\]](#page-19-35). The relative density results of this study (Sect. 5.1) further demonstrate that the magnitude of these discrepancies is dependent on topology. Inconsistencies in as-manufactured geometries mean identifcation of precise mechanical properties of individual struts within lattice structures is difficult $[39]$ $[39]$, particularly strut radius along their length [[37](#page-19-18)]. These geometric discrepancies also have implications for the orientation of loads applied to struts, which is known to have a signifcant efect on buckling performance [[57](#page-19-36)], and is the dominant failure mode for lattice structures with z-struts. Friction between the lattice specimens and compression plates afects the mechanical performance of lattice structures [[58\]](#page-19-37), yet it is understood that there is signifcant variation in the roughness of upward- and downward-facing surfaces of specimens manufactured by SLM [[59\]](#page-19-38). This suggests the friction behaviour on the upper and lower faces of the lattice specimens could be quite diferent, yet this efect remains unquantified and difficult to account for in numerical models. The signifcance of these factors varies depending on topology and means the ability of the numerical model to precisely predict mechanical properties such as stifness and strength is equally topologically dependent.

While there was quantitative discrepancy between the stress–strain behaviour of simulations and experiments, particularly in terms of the stifness of specimens with z-struts, the qualitative behaviour matched quite well. The general shape of the stress–strain curves (Fig. [13](#page-15-0)) of simulations during initial deformation matched well with experimental curves, though discrepancies increased at larger strains due to the absence of a fracture mechanism in the model. The deformation of experimental and simulated lattice specimens is compared in Fig. [14](#page-16-0) and is considered to match quite well—the initial buckling of struts and the concentration of deformation in upper and lower layers in contact with compression plates match between simulations and experiments. Stress contours during the early stages of deformation of the numerical model were found to provide signifcant insight into the causes of macroscopically observed experimental behaviour, particularly with regard to the variations between the tested topologies. The purpose of the numerical model implemented in this study was to provide insight

Fig. 10 Dynamic results, including **A** example stress–strain curves and **B** comparison of mean specifc strength of all topologies. Error bars indicate one standard deviation

Fig. 11 Comparison of average quasi-static (QS) and dynamic (D) results, including **A** stress–strain curves, **B** specifc strength of all specimens over the range of tested strain rates and C specific strength of all specimens, error bars indicate one standard deviation

into the deformation modes observed during experiments, rather than accurate prediction of mechanical properties, and thus, despite quantitative variation between numerical and simulated results, the numerical models were considered to be useful for their intended purpose.

The Von Mises stress contours for static and dynamic explicit simulations are compared in Fig. [15](#page-18-18), which provide insight into the macroscopic behaviour observed in experiments. The stress contours of the BCC and FCC lattice structures elucidate why shear planes emerge with diferent orientations depending on the topology. The BCC specimens show that the struts connecting the diagonally opposite corners carry signifcantly greater stress than other struts within the lattice, and when diagonal shear planes emerge upon failure, they align with these most loaded struts. Similarly with the FCC specimens, it can be seen that stress is more concentrated in the struts connecting opposite corners, though for FCC these most loaded struts span the faces of the lattice structure, rather than across the whole structure, as seen in the BCC specimens. This again explains the orientation of the shear plane formation in FCC lattice structures, as the struts that carry the greatest loads diagonally span the

Fig. 12 Comparison of average quasi-static and dynamic **A** energy absorbed per unit volume and **B** per unit mass

outfaces of the FCC lattice structure, and when shear planes emerge, they align with these most loaded struts.

It is also apparent from the stress contours of the BCC and FCC topologies that stress is greatest in struts furthest from the central *z*-axis of the specimens, suggesting that bending moments are induced within the lattice structures during deformation, and as the moment arm increases in length the further from the centre of the lattice, the struts on the outer faces carry the greatest stress. At a certain load, these induced stresses exceed the strength of the struts, leading to the emergence of diagonal shear planes oriented in alignment with the most loaded struts, resulting in failure. In comparison, the BCCZ and FCCZ specimens both show that when z-struts are included in these topologies these struts carry the greatest stress. This is to be expected, as the z-struts are aligned to the load meaning that they fail by crushing or buckling, whereas the BCC or FCC struts deform by bending, and so require lower stresses for deformation.

Stress contours also show that stresses are most concentrated in the interfacial layers between the compression plates and the lattice structures. This is consistent with the deformation behaviour observed during experiments, as presented in Figs. [5](#page-8-0) and [9](#page-12-0), where the deformation of z-struts in the upper and lower layers is most signifcant. This efect is more pronounced in the dynamic explicit simulations, suggesting that the concentration of stress in the interfacial layers increases with dynamic loading. This phenomenon is commonly observed in dynamically loaded materials and is related to stress wave propagation—as Lu states "high stress brought about by the strong plastic compression waves may cause local plastic collapse" [[60](#page-19-39)]. It is likely these efects would be more pronounced at greater strain rates [\[61](#page-19-40)] and suggest specimens with z-struts may be more susceptible to wave effects, though this is not clearly apparent in the tested strain rate range.

The greater concentration of stress in the impacted layer of cells helps explain the formation of shear planes in specimens with z-struts. For these lattice structures to deform, the z-struts must collapse by either buckling or crushing (depending on their slenderness), and the struts in the interfacial layers see the greatest stresses due to direct contact with the compression plates. Once these z-struts fail, the struts connecting diagonally opposite corners take up the loads, leading to the subsequent concentration of stress in these diagonal struts. Shear planes then emerge aligned to these most loaded struts, as observed in the specimens absent of z-struts.

The stress distributions also help explain the diferences in efficiency between the different lattice topologies, quantifed by specifc strength and energy absorption. The BCC, FCC and BCCZ topologies were found to have the lowest specifc strength and energy absorption capabilities, and the reason for this is demonstrated by the signifcant variation in stress states of the individual lattice struts through the structures. This is particularly pronounced for the BCC and BCCZ stress contours presented in Fig. [15,](#page-18-18) as there is signifcant variation in stress values between the diferent struts. As few struts are carrying signifcant stresses, while most struts are barely stressed at all, this leads to inefficiency of the structure, as much of the material present is underutilised, and is observed as reduced specifc strength and energy absorption efficiency in experiments. The FCCZ topology had the most even distribution of stress amongst its constituent struts, and so was the most efficient by these measures.

The FBCCZ topology can be considered a hybrid of the other tested topologies as it includes BCC, FCC and z-struts. Similar phenomena are observed during early deformation of this topology compared to the others—stress is most concentrated in the z-struts and increases further away from the central *z*-axis and in the upper and lower layers. However, once these struts fail, there are both FCC and BCC struts present to take up loads; no single plane emerges across which stresses are greatest, and so no clear shear planes emerge upon failure. Rather, the FBCCZ specimens failed in an inconsistent manner and more progressively, seen in the deformation images in Figs. [5](#page-8-0) and [9](#page-12-0) and the stress–strain curves in Fig. [11A.](#page-13-0) As a result, FBCCZ specimens had the greatest strength and stifness, and failed more progressively

Fig. 13 Comparison of quasi-static (QS) and dynamic (D) stress–strain curves and specifc Young's modulus between simulations and experiments for all topologies

Fig. 14 Comparison of experimental and simulation deformation behaviour of all topologies

than other topologies, but were less predictable in their failure, the reasons for which can be inferred from the stress contours of the numerical model.

6 Concluding remarks

Despite the research attention AM lattice structures have recently received, mechanical characterisation of the dynamic performance of AlSi10Mg lattice structures remains relatively undefned. This research has sought to overcome this identified deficit by investigating the effects of topology on the dynamic behaviour of SLM AlSi10Mg lattice structures. The fndings of this research contribute to this characterisation and provide a foundation for further research on the broader dynamic behaviour of AM lattice structures.

The key fndings of this work are:

- The as-manufactured relative density of BCC, BCCZ and FBCCZ specimens was found to exceed the expected value, whereas CAD relative densities of FCC and FCCZ topologies were found to underpredict the as-manufactured relative density. This is due to the diferent build orientations of their constituent struts, which afects their fdelity of as-manufactured geometries, and the diferent arrangements of struts and whether struts are shared between cells, which is dependent on topology.
- The inclusion of z-struts in BCC and FCC specimens led to an increase in strength and stifness, but a reduction in compliance, which was consistent for both quasi-static and dynamic testing.
- The FCCZ topology was found to be the most efficient topology in terms of specifc strength and energy absorption, due to the more even distribution of stresses across the structure during deformation.
- Specimens failed by the emergence of diagonal shear planes, the orientation of which is dependent on topology, but increasing strain rate did not alter this failure mechanism. The reason for the orientation of these shear planes was demonstrated by the numerical models that showed that stress is concentrated in struts connecting diagonally opposite corners of the structure, and when shear planes emerge, they are aligned to these most loaded struts.
- Unloading modulus increased with the strain at which it was measured and was consistently greater than the Young's modulus for all topologies.
- Stress–strain behaviour was largely consistent across the tested strain rate range for all topologies, although deformation generally occurred at greater strains under dynamic loading due to the greater rate of deformation during dynamic testing.
- Although some variation in specific strength was noticed between quasi-static and dynamic results, mean values were very similar, suggesting there is no signifcant rate sensitivity of the tested lattice structures

Fig. 15 Von Mises stress distribution (MPa) in static and dynamic ◂ explicit numerical models of lattice for all topologies

• The accuracy of numerical models was found to vary depending on topology, with the stifness of simulations of lattice structures including z-struts consistently exceeding the experimentally observed stifness, due the sensitivity of their mechanical response to the orientation of loads. However, qualitative behaviour was found to be consistent with experimentally observed behaviour, and the models provide insight into the deformation modes observed during experimental testing. Specifcally, the concentration of stress in specifc struts and the implications for the emergence of shear planes, and the distribution of stress across the structure elucidate the causes of failure and the macroscopically observed efficiency of the tested lattice topologies.

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Martin Leary: conceptualization; methodology; writing—original draft; writing—review and editing; supervision; project administration; funding acquisition.

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