Table 2 Mg-Se Lattice Parameter Data

	Commentation in the second sec	Lattice pa	arameters,		
Phase	at.% Se	a	т <i>с</i>	Comment	Reference
(Mg)	0	0.32093	0.52107	At 25° C	[King1]
MgSe		0.546		At RT	[27Bro]
(Se)	100	0.43655	0.49576	At 25° C	[King1]

Cited References

27Bro: E. Broch, "Precision Determination of Lattice Parameters of MgO, MgS, MgSe, MnO, and MnSe Compounds," Z. Phys. Chem., 127, 446-454 (1927) in German. (Crys Structure; Experimental)

67Per: V.P. Perminov, "Structural Characteristics and Crystal Chemistry of Binary Magnides," Poroshk. Metall., (5), 89-97 (1967) in Russian; TR: Sov. Powder Metall. Met. Ceram., (5), 409-416 (1967). (Crys Structure; Review)

Mg-Se evaluation contributed by A.A. Nayeb-Hashemi and J.B. Clark, Department of Metallurgical Engineering, The University of Missouri-Rolla, MO 65401. This work was supported by ASM International. Literature searched through 1984. Professor Clark is the ASM/NBS Data Program Category Editor for binary magnesium alloys.



The Ag-Ni (Silver-Nickel) System

107.8682 58.69

By M. Singleton and P. Nash llinois Institute of Technology

Equilibrium Diagram

The Ni-Ag system is an apparently simple monotectic system that has not been studied much since the evalua-

tion of [Hansen]. The equilibrium phase diagram (Fig. 1) is based on the evaluation of [Hansen], with some modifications. Invariant reaction temperatures and compositions are given in Table 1. The melting points of the pure



Reaction	Compositions of the respective phases, at.% Ag			Reaction type
$ \begin{array}{c} \hline L_1 \rightleftharpoons (Ni) + L_2 \dots & 3 \\ L \rightleftharpoons (Ni) + (Ag) \dots & 99.679 \end{array} $	1	96.11	1435	Monotectic
	>1	99.8	960	Eutectic

Table 1 Special Points of the Assessed Ni-Ag Phase Diagram

Table 2 Solubility of Ni in (Ag)

Reference	Composition, at.% Ag	Reaction °C	
[30Tam]		922	
	99.846	860	
	99.879	785	
	99.919	702	
	99.941	640	
	99.952	600	
	99.967	510	
	99.978	400	
[76Lad]		912	
	99.813	881	
	99.831	865	
	99.862	834	
	99.890	808	
	99.903	788	
	99.916	767	
	99.936	750	
	99.924	748	
	99.941	733	
	99.937	724	
	99.957	674	
	99.974	631	

elements in Fig. 1 have been taken from [Melt] as $1455 \,^{\circ}$ C for Ni (reported as $1453 \,^{\circ}$ C in [Hansen] and $1451 \,^{\circ}$ C in [77Sie]) and 962 $\,^{\circ}$ C for Ag (reported as 960.5 $\,^{\circ}$ C in [Hansen] and 961 $\,^{\circ}$ C in [77Sie]).

The Ag-rich liquidus was investigated by [61Ste], who used a liquid sampling technique followed by accurate chemical analysis to obtain the data points shown in Fig. 1. These data were shown by [61Ste] to fit the following analytical representation:

$$\log_{10} N = \frac{-A}{T} + B \tag{Eq 1}$$

where A = 4800, B = 1.4 (for 1370 °C > T > 1000 °C), N is the mole fraction, and T is the absolute temperature.

By extrapolating this expression, the eutectic composition at the eutectic temperature (960 °C, taken from [Hansen]) was fixed in Fig. 1 at 99.679 at.% Ag. Similarly, the liquidus composition at the monotectic temperature (1435 °C, taken from [Hansen, 77Sie]) was fixed at 96.11 at.% Ag.

Calculation of the Ag-rich eutectic composition is possible, assuming a eutectic temperature of 960 °C and employing the following relation [81Gas]:

$$X_{\rm Ni} = (T_{\rm M} - T_{\rm E}) x \frac{\Delta H_{\rm m}}{T_{\rm M}^2 (1-k)R}$$

where R is the gas constant = 8.3143 J/mol·K; $X_{\rm Ni}$ is the atom fraction of Ni; $T_{\rm M}$ is the melting point of Ag, K; $T_{\rm E}$ is the eutectic temperature, K; $\Delta H_{\rm m}$ is the enthalpy of fusion of pure Ag, J/mol; and k is the distribution coefficient (equal to zero if ideal mixing is assumed).

Assuming ideal mixing and taking ΔH_m to be 11 300 J/mol [83Cha], Eq 2 yields a eutectic composition of 99.822 at.% Ag, which is in reasonable agreement with the value extrapolated from the experimental data (99.679 at.% Ag) and used in Fig. 1.

The position of the monotectic point was placed by [Hansen] at approximately 3 at.% Ag on the basis of earlier work using solidified layer analyses [07Pet, 13Ces]. Figure 1 is drawn accordingly. Data pertaining to the liquid miscibility gap are largely unavailable. The phase boundary is estimated (dashed line) in Fig. 1. The two data points included in Fig. 1 are taken from work by [74Pop] on Fe-Ni-Ag and Co-Ni-Ag ternary melts.

The limit of solid solubility of Ni in (Ag) was determined by [30Tam] through measurements of the specific magnetization of equilibrated and quenched alloys. Data are presented in Table 2 with the more recent data of [76Lad]. Although the solid solubility of Ag in (Ni) decreases with decreasing temperature, no quantitative data are available. The estimate of 1 at.% Ag obtained by [Hansen] for the maximum solubility is used in Fig. 1.

Crystal structure data for the Ni-Ag system are provided in Table 3.

Metastable Phases

[69Ric] used condensation of co-evaporated Ni and Ag onto an amorphous backing to extend the maxium solid solubility of Ni in (Ag) to approximately 14 at.% Ni at the eutectic temperature.

Thermodynamics

[61Ste] reasoned that, for a dilute solution obeying Henry's law in which pure solid metal is in equilibrium with the solution:

$$\ln N = -\Delta H/RT \pm \Delta S/R \tag{Eq 3}$$

where N is the mole fraction solute (Ni), ΔH is the sum of the differential heat of solution and heat of fusion, R is the gas constant, T is the absolute temperature (1000 °C <T < 1375 °C), and ΔS is the sum of the entropy of fusion and the excess partial molar entropy.

The data plotted in Fig. 2 indicate that Eq 3 is valid for compositions up to $N = 3 \times 10^{-2}$. Figure 2 is based on the data of [61Ste], who reported the entropy data in "e.u" without defining the units. It has been assumed that an "e.u" is an entropy unit, which is 1 cal/mol·K.

Measurement of the gradient of the curve in Fig. 2 yields:

$$\Delta H = 93\,004 \text{ J/mol} \tag{Eq 4}$$

$$\Delta S = 26.8 \text{ J/mol} \cdot \text{K}.$$

Using values for the heat and temperatures of fusion taken from [58Kub], the differential heat of solution ΔH

 Table 3
 Ni-Ag Crystal Structure Data

Phase	Composition, at.% Ag	Pearson symbol	Space group	Strukturbericht designation	Prototype	Reference
Ni	0 to 1	cF4	Fm3m	A1	Cu	[King1]
	99.8 to 100	cF4	Fm3m	A1	Cu	[King1]



and the excess partial molar entropy were calculated as $75\,402$ J/mol and 16.8 J/mol·K, respectively.

Later work by [76Lad] on the solubility of Ni in (Ag) presented in Table 1 fit the expression (similar to Eq 3 [61Ste]):

$$C_{a} = \exp S/k \cdot \exp - \Delta H/kT \tag{Eq 5}$$

where C_o is the atom fraction solute (Ni); ΔS is the excess entropy per atom of solute, J/mol·K; ΔH is the enthalpy per atom of solute, J/mol; and k is Boltzmans constant = 1.3805 J/K.

The data fit yielded:

 $C_{a} = \exp 6.445 \exp - 33722/T$

giving

 $\Delta S/k = 6.445, H/k = 33722$

which in turn yields:

 $\Delta S = 8.898 \text{ J/mol} \cdot \text{K}$

 $\Delta H = 102.350 \text{ J/mol}$

These values are on the order of those determined by [61Ste].

[68Gup] studied the diffusion of Ni in liquid Ag in approximately the same range of temperatures and compositions as [61Ste] and represented the behavior analytically as:

$$D_{\rm Ni} = (7.4 \pm 1.6) \times 10^{-4} \exp(-(8120 \pm 620)/\rm{RT}) \ \rm{cm^2/s}$$
(Eq 6)

Using this expression, the activation energy for Ni diffusion in liquid Ag was calculated as $34\,015 \pm 2\,513$ J/mol,

from which value the partial molar energy $\Delta_{sol}G_{Ni}$ of solution of Ni in liquid Ag was determined to be 101 793 J/mol. The positive sign of $\Delta_{sol}G_{Ni}$ is consistent with the experimental observation that the solubility of Ni in (Ag) is very limited [Elliott], because a large positive $\Delta_{sol}G_{Ni}$ implies solvent-solute immiscibility.

Suggestions for Further Work

Although the general form of the Ni-Ag equilibrium diagram is agreed upon, there is a lack of information regarding certain regions of the diagram. In particular, the variation of solid solubility of Ag in (Ni) with temperature and the extent of the liquid/liquid miscibility gap need to be determined.

Cited References

- 13Ces: P. de Cesaris, "The Ternary Alloys of Nickel, Copper and Silver," *Gass. Chim. Ital.*, 43(2), 365-379 (1913) in Italian. (Equi Diagram; Experimental)
- 30Tam: G. Tammann and W. Oelson, "Dependence of Concentration of Saturated Mixed Crystals on Temperature," Z. Anorg. Chem., 186, 264-266 (1930) in German. (Equi Diagram; Experimental)
- 51Kle: O.J. Kleppa and J.A. Weir, "The Solubility of Copper in Liquid Lead Below 950°," J. Am. Chem. Soc., 73, 4848-4850 (1951). (Thermo; Experimental)
- 58Kub: O. Kubaschewski and E.Ll. Evans, *Metallurgical Ther*mochemistry, 3rd. ed., Vol. 1, Int. Ser. Monographs Metal Phys. and Phys. Met., Pergamon Press, New York (1958). (Thermo; Compilation)
- 61Ste: D.A. Stevenson and J. Wulff, "Liquid-Solid Phase Distribution Studies in the Systems Iron-Lead, Cobalt-Lead, Chromium-Tin and Nickel-Silver," *Trans. Metall. AIME*, 221, 271-275 (1961). (Equi Diagram, Thermo; Experimental)
- 68Gup: Y.P. Gupta, "Solute Diffusion in Liquid Metals," Advances in Physics, 16(62), 333-350 (1968). (Thermo; Experimental)
- 69Ric: R. Ricci-Bitti and J. Dixmier, "Production of Metastable Phases and Solid Solutions by Co-Evaporation of Elements Non-Miscible in the Liquid State," *Solid State Commun.*, 18, 1345-1346 (1969) (Equi Diagram, Meta Phases; Experimental)
- 74Pop: S.I. Popel and V.N. Kozhurkov, "Surface Properties and Mutual Solubility of Iron-Nickel-Silver and Cobalt-Nickel-Silver Melts," *Izv. Akad. Nauk. SSSR Metal*, 2, 49-52 (1974) in Russian. (Equi Diagram; Experimental)
- 76Lad: J. Ladet, J. Beradini, and F. Cabane-Brouty, "Behavior of Iron, Cobalt, and Nickel in the Vicinity of Dislocations in Silver and Copper Single Crystals," Scr. Metall., 10, 195-199 (1976) in French. (Equi Diagram, Thermo; Experimental)
- 77Sie: T.A. Siewert and R.W. Heine, "Recent Look at the Ag-Cu-Ni System," *Metall. Trans. A*, *8*, 515-518 (1977). (Equi Diagram; Experimental)
- 81Gas: D.R. Gaskell, Introduction to Metallurgical Thermodynamics, McGraw-Hill, New York (1981). (Thermo; Theory)
- 83Cha: M.W. Chase, "Heats of Transition of the Elements," Bull. Alloy Phase Diagrams, 4(1), 124 (1983).

Ag-Ni evaluation contributed by **M. Singleton** and **P. Nash**, Illinois Institute of Technology, Metallurgical and Materials Engineering, 10 West 33rd Street, Chicago, IL 60616. This work was funded by NASA, Grant No. NAG3-302 through ASM International. Literature searched through 1983. Professor Nash is the ASM/NBS Data Program Category Editor for binary nickel alloys.