## **Table 2 Mg-Se Lattice Parameter Data**



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# **The Ag-Ni (Silver-Nickel) System**

**107.8682 58.69** 

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# **Equilibrium Diagram**

**The Ni-Ag system is an apparently simple monotectic system that has not been studied much since the evalua-** **tion of [Hansen]. The equilibrium phase diagram (Fig. 1) is based on the evaluation of [Hansen], with some modifications. Invariant reaction temperatures and compositions are given in Table 1. The melting points of the pure** 





#### **Table 1 Special Points of the Assessed Ni-Ag Phase Diagram**

#### **Table 2 Solubility of Ni in (Ag)**



elements in Fig. 1 have been taken from [Melt] as  $1455^{\circ}$ C for Ni (reported as 1453 °C in [Hansen] and 1451 °C in [77Sie]) and 962 °C for Ag (reported as 960.5 °C in [Hansen] and  $961 °C$  in [77Sie]).

The Ag-rich liquidus was investigated by [61Ste], who used a liquid sampling technique followed by accurate chemical analysis to obtain the data points shown in Fig. 1. These data were shown by [61Ste] to fit the following analytical representation:

$$
\log_{10} N = \frac{-A}{T} + B \tag{Eq 1}
$$

where  $A = 4800, B = 1.4$  (for 1370 °C >  $T > 1000$  °C), N is the mole fraction, and  $T$  is the absolute temperature.

By extrapolating this expression, the eutectic composition at the eutectic temperature (960 °C, taken from [Hansen]) was fixed in Fig. 1 at 99.679 at.% Ag. Similarly, the liquidus composition at the monotectic temperature (1435  $^{\circ}$ C, taken from [Hansen, 77Sie]) was fixed at 96.11 at.% Ag.

Calculation of the Ag-rich eutectic composition is possible, assuming a eutectic temperature of  $960 °C$  and employing the following relation [81Gas]:

$$
X_{\rm Ni} = (T_{\rm M} - T_{\rm E}) x \frac{\Delta H_{\rm m}}{T_{\rm M}^2 (1 - k) R}
$$

where *R* is the gas constant =  $8.3143$  J/mol·K;  $X_{\text{Ni}}$  is the atom fraction of Ni;  $T_M$  is the melting point of Ag, K;  $T_E$  is the eutectic temperature, K;  $\Delta H_m$  is the enthalpy of fusion of pure Ag, J/mol; and  $k$  is the distribution coefficient (equal to zero if ideal mixing is assumed).

Assuming ideal mixing and taking  $\Delta H_m$  to be 11 300 J/mol [83Cha], Eq 2 yields a eutectic composition of 99.822 at.% Ag, which is in reasonable agreement with the value extrapolated from the experimental data (99.679 at.% Ag) and used in Fig. 1.

The position of the monotectic point was placed by [Hansen] at approximately 3 at.% Ag on the basis of earlier work using solidified layer analyses [07Pet, 13Ces]. Figure 1 is drawn accordingly. Data pertaining to the liquid miscibility gap are largely unavailable. The phase boundary is estimated (dashed line) in Fig. 1. The two data points included in Fig. 1 are taken from work by [74Pop} on Fe-Ni-Ag and Co-Ni-Ag ternary melts.

The limit of solid solubility of Ni in (Ag) was determined by [30Tam] through measurements of the specific magnetization of equilibrated and quenched alloys. Data are presented in Table 2 with the more recent data of [76Lad]. Although the solid solubility of Ag in (Ni) decreases with decreasing temperature, no quantitative data are available. The estimate of I at.% Ag obtained by IHansen] for the maximum solubility is used in Fig. 1.

Crystal structure data for the Ni-Ag system are provided in Table 3.

## **Metastable Phases**

[69Ric] used condensation of co-evaporated Ni and Ag onto an amorphous backing to extend the maxium solid solubility of Ni in (Ag) to approximately 14 at.% Ni at the eutectic temperature.

## **Thermodynamics**

161Ste] reasoned that, for a dilute solution obeying Henry's law in which pure solid metal is in equilibrium with the solution:

$$
\ln N = -\Delta H / RT \pm \Delta S / R \tag{Eq 3}
$$

where  $N$  is the mole fraction solute (Ni),  $\Delta H$  is the sum of the differential heat of solution and heat of fusion,  $R$  is the gas constant, T is the absolute temperature (1000  $\degree$ C < T < 1375 °C), and  $\Delta S$  is the sum of the entropy of fusion and the excess partial molar entropy.

The data plotted in Fig. 2 indicate that Eq 3 is valid for compositions up to  $N = 3 \times 10^{-2}$ . Figure 2 is based on the data of f61Ste], who reported the entropy data in "e.u" without defining the units. It has been assumed that an "e.u" is an entropy unit, which is  $1 \text{ cal/mol} \cdot \text{K}$ .

Measurement of the gradient of the curve in Fig. 2 yields:

$$
\Delta H = 93\,004 \text{ J/mol} \tag{Eq 4}
$$

$$
\Delta S = 26.8 \text{ J/mol} \cdot \text{K}.
$$

Using values for the heat and temperatures of fusion taken from [58Kub], the differential heat of solution  $\Delta H$ 

**Table 3 Ni-Ag Crystal Structure Data** 

<b>Phase</b>	Composition. at.% Ag	Pearson svmbol	<b>Space</b> group	Strukturbericht designation	Prototype	Reference
		cF4 cF4	Fm3m Fm3m	Al	Cu Cu	[King1] [King1]



and the excess partial molar entropy were calculated as 75 402 J/mol and 16.8 J/mol. K, respectively.

Later work by [76Lad] on the solubility of Ni in (Ag) presented in Table 1 fit the expression (similar to Eq 3 161Ste]):

$$
C_o = \exp S/k \cdot \exp - \Delta H/kT \tag{Eq 5}
$$

where  $C_o$  is the atom fraction solute (Ni);  $\Delta S$  is the excess entropy per atom of solute, J/mol·K;  $\Delta H$  is the enthalpy per atom of solute, J/mol; and k is Boltzmans constant  $=$ 1.3805 J/K.

The data fit yielded:

 $C<sub>o</sub>$  = exp 6.445 exp - 33 722/T

giving

 $\Delta S/k = 6.445, H/k = 33722$ 

which in turn yields:

 $\Delta S = 8.898$  J/mol · K

 $\Delta H = 102.350$  J/mol

These values are on the order of those determined by 161Ste].

168Gup] studied the diffusion of Ni in liquid Ag in approximately the same range of temperatures and compositions as [61Ste| and represented the behavior analytically as:

$$
D_{\text{Ni}} = (7.4 \pm 1.6) \times 10^{-4} \text{ exp }(- (8120 \pm 620)/\text{RT}) \text{ cm}^2/\text{s}
$$
\n(Eq 6)

Using this expression, the activation energy for Ni diffusion in liquid Ag was calculated as  $34015 \pm 2513$  J/mol, from which value the partial molar energy  $\Delta_{sol}G_{Ni}$  of solution of Ni in liquid Ag was determined to be 101 793 J/mol. The positive sign of  $\Delta_{sol}G_{Ni}$  is consistent with the experimental observation that the solubility of Ni in (Ag) is very limited [Elliott], because a large positive  $\Delta_{sol}G_{Ni}$  implies solvent-solute immiscibility.

# **Suggestions for Further Work**

Although the general form of the Ni-Ag equilibrium diagram is agreed upon, there is a lack of information regarding certain regions of the diagram. In particular, the variation of solid solubility of Ag in (Ni) with temperature and the extent of the liquid/liquid miscibility gap need to be determined.

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