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# The C-Eu (Carbon-Europium) System

12.011

151.96

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No phase diagram is available for the Eu-C system, although some data are available on compounds that are formed in the Eu-C system. [58Spe] did not include Eu in their investigation of the crystal structure of RE-C compounds, but later investigators reported structure data for several Eu-C compounds. [70Lap] reported the preparation of a monocarbide which they designated  $\text{EuC}_{1-x}$ , where  $x$  represents the amount of C that is missing in the compound. They claimed a cubic NaCl-type structure and reported lattice parameters of  $0.5145 \pm 3$  nm in the presence of excess metal and  $0.5141 \pm 1.1$  nm in the presence of excess C. These lattice parameters (Table 1) appear to fit well with those of the cubic  $\text{Fe}_4\text{N}$ -type  $\text{RE}_3\text{C}$  structures.

After unsuccessful attempts to prepare  $\text{Eu}_3\text{C}$ ,  $\text{Eu}_2\text{C}_3$ , and  $\text{EuC}_2$  from the elements, [72Col] reacted >99.9% pure Eu with HCN and  $(\text{CN})_2$  at elevated temperatures and obtained the compounds  $\text{Eu}_2\text{C}_3$ ,  $\text{EuC}_2$ , and  $\text{EuN}_x\text{C}_y$ . X-ray data were obtained with Fe-filtered  $\text{CoK}_\alpha$  radiation using a 114.6 mm Debye-Scherrer camera.  $\text{Eu}_2\text{C}_3$  was found to crystallize with the cubic  $\text{Pu}_2\text{C}_3$ -type structure, in good agreement with the structure of neighboring sesquicarbides. Their data for  $\text{Eu}_2\text{C}_3$  are listed in Table 1.

Other investigators reported some lattice parameter data for  $\text{EuC}_2$ , but a complete structure analysis has not been reported. [68Fai] synthesized  $\text{EuC}_2$  from the elements in sealed Mo bombs in an argon-filled C tube furnace.  $\text{EuC}_2$  gave anomalous X-ray patterns, more complex than those of  $\text{LaC}_2$ ,  $\text{CeC}_2$ , or  $\text{NdC}_2$ , but reproducible with four different samples. This pattern was indexed tentatively as a

bct with  $a = 1.215$  and  $c = 0.7290$  nm, accounting for all but two low index lines. [64Geb] produced  $\text{EuC}_2$  by two methods: (1) reduction of  $\text{Eu}_2\text{O}_3$  (99.9% pure, weight?) with spectrographic grade C (graphite) placed in a graphite crucible; and (2) melting of 2-to-1 mol mixture of C and Eu in a stainless steel bomb in a resistance tube furnace. X-ray powder diffraction patterns of the product using  $\text{CuK}_\alpha$  radiation in both a 114.59-mm powder camera and a Siemens diffractometer showed the bct  $\text{CaC}_2$ -type structure, along with a second phase with a different type of structure. The second phase could be obtained in larger quantities in samples containing a lower C-to-Eu ratio and heated to a somewhat lower temperature.

The low-angle lines of this impurity could be indexed on the basis of an orthorhombic cell with  $a = 0.876$ ,  $b = 1.123$ , and  $c = 0.719$  nm, but the high-angle lines were not indexed satisfactorily [66Geb].

[82Sak] synthesized  $\text{EuC}_2$  by two methods: (1) 99% pure Eu and spectrographic grade C (graphite) were sealed under vacuum in a silica capsule and heated at 1000 °C for 30 h; and (2) 99.999%  $\text{Eu}_2\text{O}_3$  was mixed with C (graphite), pelletized, and heated in Mo under a flow of argon at 1800 °C for 5 h. An excess of 5 at.% C (graphite) was used to remove all of the oxygen. Powder X-ray diffraction patterns were taken using Ni-filtered  $\text{CuK}_\alpha$  radiation with an internal standard of Si. Phase transformation was examined by differential thermal analysis in the temperature range 27 to 727 °C. The dicarbides obtained by each method had bct  $\text{CaC}_2$ -type structure, but each contained a second phase that could not be removed

Table 1 Eu-C Crystal Structure and Lattice Parameter Data

Phase	Composition range, at.% C	Pearson symbol	Space group	Strukturbericht designation	Prototype	Lattice parameters, nm		Density, g/cm <sup>3</sup>	Reference
						a	c		
(Eu) . . . . .	0	<i>cI2</i>	<i>Im3m</i>	A2	W	0.45827	...	5.244	[86Gsc]
$\text{Eu}_3\text{C}$ . . . . .	~25 to ~33	<i>cF5</i>	<i>Fm3m</i>	$L'_1$	$\text{Fe}_4\text{N}$	0.5145(3) (a)	...	7.646(a)	[70Lap]
						0.5141(3) (b)	...	7.722(b)	[70Lap]
$\text{Eu}_2\text{C}_3$ . . . . .	~60	<i>cI40</i>	$\bar{I}43d$	$D5_c$	$\text{Pu}_2\text{C}_3$	0.8368	...	7.530	[72Col]
$\alpha\text{EuC}_2$ . . . . .	66.7	<i>tI6</i>	<i>I4/mmm</i>	$C11_a$	$\text{CaC}_2$	0.4082(36)	0.6701(56)	5.236	[64Geb, 82Sak]
$\beta\text{EuC}_2$ . . . . .	66.7	<i>cF12</i>	<i>Fm3m</i>	C1	$\text{CaF}_2$	0.5961(1)	...	5.519	[68Mat]
$\text{EuC}_6$ . . . . .	85.7	<i>hP14</i>	$P6_3/mmc$	...	$\text{EuC}_6$	0.4314(3)	0.9745(8)	4.737	[75Gue, 80Elm]
(C) . . . . .	100	<i>hP4</i>	$P6_3/mmc$	A9	C (graphite)	0.24612	0.67090	2.266	[Pearson2]

(a) Eu-rich. (b) C-rich.

by annealing. The lattice parameters for  $\text{EuC}_2$  with the  $\text{CaC}_2$ -type structure are listed in Table 1.

[67Kri] attempted to measure the  $\alpha \rightleftharpoons \beta$  transformation temperature of  $\text{EuC}_2$  in a sealed Ta bomb, but could not attribute the phenomena they observed at 355 °C to  $\text{EuC}_2$ , because their X-ray pattern showed predominant lines for Eu. However, [82Sak] reported that tetragonal  $\text{EuC}_2$  transforms to an fcc form at 350 °C, which is close to the temperature where [67Kri] had observed a change. This is accepted as the transformation temperature of  $\text{EuC}_2$ , but it is considerably lower than those of other  $\text{RE}_2\text{C}_2$  compounds.  $\beta\text{EuC}_2$  has fcc structure of the  $\text{CaF}_2$  type. No melting temperature data were found. [67Kri] were not able to determine the  $\text{EuC}_2$ -C eutectic temperature by thermal analysis, due to the high vapor pressure of Eu over  $\text{EuC}_2$ . Because Eu exhibits variable valence tendencies, the properties and parameters of its compounds do not follow the systematic variations encountered between other rare earth systems.

[75Gue] and [80Elm] studied intercalation of Eu metal in C (graphite). Using direct action of metal vapor on graphite in metal or glass tubes sealed under vacuum, they obtained a 25% yield of intercalation with Eu at 500 °C after 20 days. The core of their sample remained pure graphite. They reported that the first-stage compound,  $\text{EuC}_6$ , has a hexagonal unit cell with the space group  $P6_3/mmc$ . The reported lattice parameters are in Table 1.

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## The C-Gd (Carbon-Gadolinium) System

12.011

157.25

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No phase diagram is available for the Gd-C system. [58Spe] reported that Gd forms a tri-rare-earth carbide with a range of solubility (see "The Carbon-Rare Earth Systems," in this issue). Their report included lattice spacings for the C-rich  $\text{Gd}_3\text{C}$  compound, which is ferromagnetic at room temperature. See Table 1 for structure details of this cubic  $\text{Fe}_4\text{N}$ -type form. [73Hub] prepared

specimens of trigonal " $\text{Gd}_2\text{C}$ " by arc melting 99.9 wt.% Gd and spectrographic quality C (graphite) together under a Zr-gettered inert gas atmosphere. After remelting several times, samples were crushed and X-ray powder data were obtained using a 114.59-mm camera and Ni-filtered  $\text{CuK}_\alpha$  radiation. Some free Gd was detected in the samples, and on the basis of ultraviolet emission spec-