The C-Dy (Carbon-Dysprosium) System

12.011

162.50

By K.A. Gschneidner, Jr. and F.W. Calderwood lowa State University

No phase diagram is available for the Dy-C system, but some structural data have been reported. [58Spe] found that at compositions near Dy_3C , the Dy atoms are arranged in a cubic close-packed structure, with the C atoms distributed randomly in the octohedral interstices of this cubic structure, forming an NaCl defect structure or an Fe₄N-type structure (see The Carbon-Rare Earth Systems," in this issue). Structure details for Dy₃C are given in Table 1 for both the C-rich compound and the Dy-rich phase reported by [79Aok]. The Fe_4N -type phase is the high-temperature form that can exist at room temperature in a metastable state. The stable room-temperature form has the approximate composition Dy_2C and the rhombohedral CdCl₂-type structure. The lattice parameter for trigonal Dy_2C is given in Table 1. No melting or decomposition temperature is available for this phase.

The sesquicarbide, Dy_2C_3 , was reported by [58Spe] to have the bcc Pu_2C_3 -type structure with lattice parameters as shown in Table 1 for the Dy-rich and C-rich compositions. There is no information on the melting point or deformation temperature of this compound.

[58Spe] found that the room-temperature form of DyC₂ has the bct CaC₂-type structure (Table 1). [67Kri] confirmed this structure and reported that the transition to cubic DyC₂ occurs at 1295 \pm 20 °C. Other investigators reported lower and higher transformation temperatures (1250 °C [73Mcc] and 1300 to 1317 °C [76Loe]), and the average of these reported temperatures, 1290 \pm 30 °C, is in good agreement with that indicated by systematics.

No report was found that detailed the high-temperature structure of DyC_2 , although investigators who measured the transformation temperature mentioned a cubic structure. Because the βREC_2 structures that have been deter-

mined (RE = La, Ce, Eu, Tb, and Lu) all have the CaF₂type cubic form, we believe that DyC_2 also has this form. A third structure has been reported for DyC_2 , but high pressures are required to form it from the $C11_a$ CaC₂-type structure. See Table 1 and "The Carbon-Rare Earth Systems," in this issue, for details.

[67Kri], using thermal analysis methods, found the DyC_2 -C eutectic temperature to be 2290 \pm 25 °C. [70Yup] and [71Kos] reported the melting point of DyC_2 as 2250 \pm 60 and 2245 °C, respectively, but because these values are lower than the reported value of the eutectic, some adjustment is necessary. In this situation, we considered these melting temperatures actually to be eutectic temperatures, and we included them in the calculation of the average eutectic temperature. Treatment of the available data for the REC₂-C eutectic temperatures by systematics indicates that the calculated average for DyC_2 , 2260 \pm 25 °C, must be close to the true value. Other than the two reported melting temperatures for DyC_2 , which were treated as eutectic temperatures, no melting-point data are available for this compound, and the lack of corresponding data for the neighboring dicarbides makes an application of systematics impossible.

Cited References

- 58Spe: F.H. Spedding, K.A. Gschneidner, Jr., and A.H. Daane, "The Crystal Structures of Some of the Rare-Earth Carbides," J. Am. Chem. Soc., 80, 4499 (1958).
- 67Kri: N.H. Krikorian, T.C. Wallace, and M.G. Bowman, "Phase Relationships of the High-Carbon Portion of Lanthanide-Carbon Systems," *Propriétés Thermodynamiques Physiques et Structurales des Dérivés Semi-Metalliques*, Centre National Recherche Scientifique, Paris, 489 (1967).

68Ato: M. Atoji, "Magnetic Structure of DyC2," J. Chem. Phys.,

Table 1 Dy-C Crystal Structure and Lattice Parameter Data

	Composition range, at.% C	Pearson symbol	Space group	Struktur- bericht designa- tion	Prototype	Lattice parameters, nm			Density,	
Phase						а	b	с	g/cm ³	Reference
(αDy)	0	hP2	$P6_3/mmc$	A3	Mg	0.35915		0.56501	8.551	[86Gsc]
$(\alpha' \mathbf{D} \mathbf{y}) \dots$	0	(a)	•••	•••		0.3595	0.6184	0.5678	8.551	[86Gsc]
$(\beta Dy) \dots$	0	cI2	Im3m	A2	W	0.398(b)	•••	•••	8.56	[86Gsc]
$\alpha Dy_2 C \dots$	~33	hR3	$R\overline{3}m$	C19	$CdCl_2$	0.6312(c)	•••		8.437	[81Ato]
$\beta Dy_3 C \dots$	${\sim}25$ to ${\sim}33$	cF5	Fm3m	L'_1	Fe_4N	0.5001(d)	•••		8.419(d)	[79Aok]
						0.5079(5) (e)	•••	• • •	8.543(e)	[58Spe]
Dy_2C_3	~54 to ~60	cI40	$I\overline{4}3d$	$D5_c$	Pu_2C_3	0.8198(2) (d)	•••	• • •	8.517(d)	[58Spe]
						0.8215(1) (e)	• • •	•••	8.650(e)	[70Kru]
α DyC ₂	~66.7	tI6	I4/mmm	$C11_a$	CaC_2	0.3669(2)	•••	0.6171(4)	7.459	[58Spe, 67Kri,
										68Ato, 76Ada, 80Sak, 81Sak]
$\beta Dy C_2(f) \dots$	~66.7	cF12	Fm3m	C1	CaF_2				•••	[67Kri]
$\alpha' Dy C_2(g)$.	~66.7	o??			LuC_2	1.355	2.697	0.7100	• • •	[68Kru]
(C)	100	hP4	$P6_3/mmc$	A9	C (graphite)	0.24612	•••	0.67090	2.266	[Pearson2]

(a) Orthorhombic distortion, $T \le 86$ K. (b) Determined by extrapolation to 0% solute of a versus composition data for Dy-Mg alloys. (c) $\alpha = 32.99^{\circ}$. (d) Dy-rich. (e) C-rich. (f) Structure for this form has not been reported, but is assumed because it is in agreement with known βREC_2 data. (g) High-pressure form. 48, 3384 (1968).

- 68Kru: M.C. Krupko, N.H. Krikorian, and T.C. Wallace, "Polymorphism in Heavy Rare-Earth Dicarbides at High Pressures and Temperatures," Proc. 7th Rare-Earth Research Conference, 28-30 Oct, Coronado, CA, 197 (1968).
- 70Kru: M.C. Krupko and N.H. Krikorian, "High-Pressure Synthesis of New Heavy Rare-Earth Carbides," Proc. 8th Rare-Earth Research Conference, Vol. 2, T.A. Henrie and R.E. Lindstrom, Ed., National Technical Information Service, Springfield, VA, 382 (1970).
 70Yup: V.L. Yupko, G.N. Makarenko, and Yu.B. Paderno,
- 70Yup: V.L. Yupko, G.N. Makarenko, and Yu.B. Paderno, "Physical Properties of Carbides of Rare-Earth Metals," *Tugoplavkie Karbidy*, G.V. Samsonov, Ed., Naukova Dumka, Kiev (1970) in Russian; TR: *Refractory Carbides*, N.B. Vaughn, Translator, Consultants Bureau, New York, 251 (1974).
- 71Kos: T.Ya. Kosolapova, G.N. Makarenko, and L.T. Domasevich, "Characteristics of Chemical Bonding in Rare-Earth Carbides," *Zh. Prikl. Khim.*, 44, 953 (1971) in Russian; TR: *Russ. J. Appl. Chem.*, 44, 965 (1971).
- 73Mcc: I.J. McColm, T.A. Quigley, and N.J. Clark, "The Cubic-Tetragonal Transformation in Metal Dicarbides—II. Lanthanide Dicarbides and Some Mixed Lanthanide Dicarbide Solid Solutions," J. Inorg. Nucl. Chem., 35, 1931 (1973).
- 76Ada: G.-Y.Adachi, Y. Shibata, K. Ueno, and J. Shiokawa, "Heats of the Tetragonal-Cubic Transformation in Rare-Earth

Dicarbides and Mixed Rare-Earth Dicarbide Solid Solutions," J. Inorg. Nucl. Chem., 38, 1023 (1976).

- **76Loe:** I.R. Loe, I.J. McColm, and T.A. Quigley, "Cubic to Tetragonal Transformations in Dicarbides—III. Enthalpies and Strain Energies of HoC₂-NdC₂ and GdC₂-LaC₂ Solid Solutions," J. Less-Common Met., 46, 217 (1976).
- **79Aok:** Y. Aoki and D.E.G. Williams, "Some Properties of Alloys with Compositions Close to DyC_{0.33}," J. Less-Common Met., 65, 35 (1979).
- 80Sak: T. Sakai, G.-Y. Adachi, and J. Shiokawa, "Magnetic Properties of CeC₂ and Its Solid Solutions, Ce_xDy_{1-x}C₂," *Mater. Res. Bull.*, 15, 1001 (1980).
- 81Ato: M. Atoji, "Neutron Diffraction Studies of Tb_2C and Dy_2C in the Temperature Range 4 to 296 K," J. Chem. Phys., 75, 1434 (1981).
- 81Sak: T. Sakai, G.-Y. Adachi, T. Yoshida, and J. Shiokawa, "Magnetic and Electrical Properties of Rare-Earth Dicarbides and Their Solid Solutions," J. Less-Common Met., 81, 91 (1981).
- 86Gsc: K.A. Gschneidner, Jr. and F.W. Calderwood, "Intra Rare Earth Binary Alloys: Phase Relationships, Lattice Parameters and Systematics," in *Handbook on the Physics and Chemistry* of Rare Earths, Vol. 8, K.A. Gschneidner, Jr. and L. Eyring, Ed., North-Holland Physics Publishing, Amsterdam, 1-161 (1986).

C-Dy evaluation contributed by K.A. Gschneidner, Jr., Director, and F.W. Calderwood, Rare-Earth Information Center, Ames Laboratory, Iowa State University, Ames, IA 50011. This program was supported by the Department of Energy through the Joint Program on Critical Compilation of Physical and Chemical Data coordinated through the Office of Standard Reference Data, National Bureau of Standards. Additional support was contributed by: Molycorp, Inc., Union Oil Co. of California, Los Angeles, CA; Reactive Metals & Alloys Corp., West Pittsburg, PA; Ronson Metals Corp., Newark, NJ; and Santoku Metal Industry Co., Ltd., Kobe, Japan. Literature searched through Aug 1984. Professor Gschneidner is the ASM/NBS Data Program Category Editor for binary rare-earth alloys.

The C-Er (Carbon-Erbium) System

12.011

167.26

By K.A. Gschneidner, Jr. and F.W. Calderwood lowa State University

No phase diagram is available for the Er-C system, although some information has been reported for the compounds and their structures. Tri-erbium carbide, ErC_x (where 0.25 < x < 0.65) forms a cubic Fe₄N-type structure in which the anion sites are partially occupied by C atoms. This structure exists immediately below the melting point, but if x is close to 0.5, the structure transforms at lower temperatures to a trigonal CdCl₂-type structure (see "The Carbon-Rare Earth Systems," in this issue). [81Ato] stated that if x is much different from 0.5, the cubic structure is retained at all temperatures. Lattice parameters for cubic Er₃C are presented in Table 1, but no values were given for trigonal Er₂C. No melting point or transformation temperature data are available for these compositions.

An intermediate compound of Er having a lower than cubic symmetry can be obtained by arc melting. [58Spe] stated that erbium sesquicarbide is isostructural with Y_2C_3 . The structure of the latter compound was eventually determined by [71Bau] to be tetragonal $Sc_{15}C_{19}$ type and the composition to be $Y_{15}C_{19}$ instead of Y_2C_3 . [74Bau] determined the equilibrium concentration of erbium "sesquicarbide" to be $\text{Er}_{15}\text{C}_{19}$ and reported pseudocubic tetragonal lattice parameters for this structure (Table 1). It was shown by [70Kru1], [70Kru2], and [80Nov] that Er_2C_3 can be obtained with the cubic Pu_2C_3 -type structure by a solid-state synthesis at elevated temperatures and pressures (30 to 90 kbar, 1200 to 1400 °C). The lattice parameter is included in Table 1. No melting information is available for these Er intermediate compounds.

In the region between 55 and 60 at.% C, the phase relationships in the Er-C phase diagram may be quite similar to those observed in the Y-C system (see "The C-Y (Carbon-Yttrium) System," to be published).

The dicarbide, ErC_2 , was reported by [58Spe] to have the bct CaC_2 -type structure. The average value of lattice parameters from several sources is listed in Table 1. An $\alpha \rightleftharpoons \beta$ transformation from the tetragonal to a cubic form occurs at a temperature above 1300 °C; the lowest of four values is 1275 °C reported by [73Mcc], and the average of these values is 1310 ± 25 °C. This is in good agreement with the systematic trends of the $\alpha \rightleftharpoons \beta$ transformation of REC₂ compounds. In view of the $\alpha \rightleftharpoons \beta$ transformation,