Waterborne Acrylic-Epoxy Coatings

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INTRODUCTION

In the early 1990s, the drive for higher performance
coatings with lower volatile organic compound
(VOC) content led to the commercial introduction of coatings with lower volatile organic compound waterborne acrylic-epoxy coatings. These are two-com**p** onent coatings, with one componentcontaining a carboxyl-functional acrylic latex, and the other component containing an epoxy emulsion. Upon mixing, cure is believed to proceed via carb oxyl-epoxy reaction and/ or epoxy homopolymerization. It is interesting that although these reactions usually proceed slowly under ambient conditions, the applied coating has attractive properties. The resulting crosslinked system is responsible for upgraded performance over acrylic latex coa tings in properties such as hardness, mar, abrasion resistance, chemical resistance, and water resistance. These coatings also display faster dry and improved exterior durability over many epoxy-amine-based systems. Other attractive features include low organic solvent content and low odor. Therefore, acrylic-epoxy coatings are finding utility as topcoats in moderate duty industrial as well as high performance architectural applications. Examples of the latter applications are institutional (e.g., schools, locker areas, laboratories, etc.) wall surfaces of metal, masonry, plaster, and gypsum wallboard. References 1-5 provide additional information on the history, chemistry, and traditional performance of waterborne acrylic-epoxy coatings.

Both liquid epoxy resin emulsions and solid epoxy resin dispersions in water have been used to crosslink these systems. Liquid epoxy resins are lower molecular weight (i.e., < 1000) systems emulsified in water with the aid of a surfactant. Because of their relatively low molecular weight and viscosity, these systems exhibit good handling, flow, and coalescence, with little or no coalescing solvent; however, they usually take longer to dry, expecially dry hard. Solid epoxy dispersions are dispersed particles of higher molecular weight, solid epoxy resin in water. The molecular weight distribution of these resins is typically: 18% < 1000, 79% between 1000 and 10,000, and only 3% > 10,000. Coating systems with solid epoxy resin dispersions usually dry faster due to the lacquer dry of the higher molecular weight component, but they contain

Waterborne two-compo n *ent acrylic-epoxy coatings are gaining popularity as topcoats in moderate du*ty industrial and high per*formance architectural (HIPA C) applications, This increased popularity is due*

to their attractive handling, application, and per*formance properties, along with their low solvent content md odor, The obje eth~es of thi~ work were to characterize the cure and property development* of these coatings, evaluate performance properties *of cured films, and investigate a new epoxy resin dispersion in existing acrylic-epoxy formulations.* These evaluations confirmed that existing acrylicepoxy coatings have long pot life and short dry times while displaying a range of chemical resis*tance and physical properties. IR spectroscopy and differential scanning calorimetry* (DSC) *re*sults indicated that the extent of cure at ambient *conditions over a 21-day period was minimal; however, dynamic mechanical analysis* (DMA) *and solvent swell results did ittu~trate noticeable crosslink density development under these condition~, DSC res~dte demonetrated more complete* reaction and cure after heating. Direct substitu*tion of a novel epoxy resin dispersion into these formulas resulted in lower required solvent content, shorter d~ time, higher gtoss, higher crosslink density, and improved water and scrub resistance.*

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10% co solvent to aid processing, assist flow, and permit coalescence of the film. From this description, the tradeoffs between liquid epoxy and solid epoxy resins in coating formulations are obvious.

A novel waterborne epoxy resin⁶⁻⁸ has been developed which offers the most attractive features of both liquid and solid waterborne epoxy resin systems. This product will be designated AP ER throughout this paper. The product is produced by emulsifying traditional Bisphenol A liq uid epoxy resin, initiating homopolymerization, and stopping the polymerization when the desired degree of polymerization and molecular weight distribution has oc curred. AP-ER epoxy resin has a unique molecular weight distribution that is quite different than both liquid and solid epoxy resins: 48% < 1000, 27% between 1000 and 10,000, and 25% > 10,000. This molecular weight distribu tion is responsible in part for its unique properties. The high molecular weight component of the material can allow for a fast, lacquer type drying characteristic, while its low glass transition temperature $(0^{\circ}C)$ and minimum film formation temperature (4° C) allows fast dry with no co-solvent. *Table* i illustrates a general comparison of all three epoxy resin types. Previous research^{$7-8$} with this new material primarily has been with amine-based cure

Table 2--Parameters for Acrylic Resins Used in Starting Formulas

Property	Maincoat AE58ª	Ecocnvl9790 ^b		
	42.5	42.0		
Water content (%)	57.5	43.5		
Solvent content (%)	O	14.5		
	8.6	8.5		
	700	15.000		
Equivalent weight (on solids)	1700	660		
ta). Rohm and Haas tha Resolution				

Table 3--Parameters for Epoxy Resins Used in Starting Formulas

ties and property development of commercial acrylic epoxy coatings and published start ing formulations.

were:

systems; however, it would seem plausible that this resin may be suitable in acrylic-epoxy formulations. Potential advantages of this coating would include fast dry time and lower solvent con tent, while maintaining other performance properties. Therefore, the objectives of this project

(1) Characterize the proper-

(2) Study cure development of these coatings in relation to property development.

(3) Assess if and how the new epoxy resin may be used in waterborne acrylic-epoxy coatings.

EXPERIMENTAL

Materials

Two commercially available acrylic epoxy coatings were evaluated, along with two suggested starting formulas prepared within our laboratories. The commercial coat ings are designated CC-1 and CC-2 while the control starting formulas are designated SF-1 and SF-2. *Tables 2* and 3 list the properties of the acrylic and epoxy resins utilized in the published starting formulas that are listed in *Tables* 4 and 5. *Table* 6 lists formulation parameters of the coatings evaluated. It should be noted that CC-1 contains solid epoxy dispersion while CC-2, SF-1, and SF-2 contain liquid epoxy emulsion.

One objective of this work was to determine if and how the new epoxy resin may be used in waterborne acrylicepoxy coatings. To pursue this objective, AP ERwas sub stituted at various concentrations into the CC-1 and SF-1 coatings for the standard epoxy resins in those systems. Since CC-1 is a commercial coating and little is known about the formulation other than the parameters listed in *Table* 6, mix ratios of CC-1 part A to AP-ER evaluated were 8:1, 4:1, and 2:1 by volume. Since the SF-1 formula was known *(Table 4)*, AP-ER was substituted on a weight basis. Mix ratios of SF-1 part A to AP-ER were 905.5:41 g, 905.5:82 g, and 905.5:165 g, respectively. These substitutions were performed by simply making a direct replace ment of the AP ER for the existing epoxy component (part B) in each of the formulas.

Experimental Procedures

The coatings were mixed prior to application per the manufacturers' recommended procedures, including the suggested induction time of 30 min prior to application. The coatings were applied to cold rolled steel panels with zinc phosphate treatment (Bonderite 952) using a 6 mil bird bar. Free films were prepared by applying the coat ings to a Tedlar film and then peeling the cured film from the Tedlar prior to testing. Unless otherwise noted, the

Table 4-Formula for SF-1

applied coatings were allowed to cure for 21 days at 21°C, 50% relative humidity prior to testing. Table 7 lists the testing procedures that were followed to evaluate coating performance. Property development was characterized by assessing Persoz hardness and MEK double rubs as a function of time after application. ASTM D 3730 is a "Standard Guide for Testing High Performance Interior Architectural Wall Coatings (HIPAC)." This standard provides test methods and requirements for coatings used in these applications and was used as a guideline for the evaluation procedures and criteria.

In addition to coating property development on metal substrates, characterization of free films was performed 1, 7, 14, and 21 days after application to assess crosslink density, T_g development, and extent of cure. Cure development was studied via dynamic mechanical analysis (DMA), differential scanning calorimetry (DSC), IR spectroscopy, and solvent swells. DMA was performed using an RSA II from Rheometric Scientific. The specimen was tested in the tensile deformation mode at an applied frequency of one cycle/second over a temperature range of -100 to 200 °C, heating in 6° C increments with a 60 -sec soak time at each temperature to ensure isothermal conditions. The DSC analysis was performed using a 2920 DSC from TA Instruments heating from -130 to 250° C at a rate of 10° C/min. DSC data were analyzed using Universal V2.5H software from TA instruments. Gas flow rates were 50 ml/min of dry nitrogen and 25 ml/min of helium. Sample sizes were 10-20 mg. Samples were scanned, cooled, and then rescanned to determine the extent of cure. IR spectra were obtained using a Nicolet Magna 860 at 4 wavenumbers resolution with 128 signal average scans. A Fresnel ATR accessory

Table 5-Formula for SF-2

from CIC Photonics was attached and the coatings were applied to an Amtir 1 ATR crystal.

Solvent swells were performed on free film samples approximately 2 cm by $\hat{2}$ cm by 40 microns. The test was performed by immersing the free film samples in MEK for one hour and, immediately upon removal, measuring the increase in length of the sides. These data were used to calculate a percent volume increase, which in turn was used to calculate a theoretical crosslink density. Crosslink density was calculated in terms of Mc, molecular weight between crosslinks, using the following equation:

$$
Mc = \frac{p}{v_e} = \frac{V_o}{v_e'} = \frac{(v_2^{0.33} - v_2/2)}{-(\ln(1 - v_2) + v_2 + \chi v_2^2)/v_1}
$$

where: ρ is the polymer density assumed to be $1.0 g/ml$, v_e is the crosslink density in moles/ml, v_e' is the moles of elastically effective network chains in volume V_0 of unswollen polymer film, v_1 is the molar volume of solvent (90.1 ml/mole for MEK), χ is the Flory-Huggins interaction parameter of solvent with network polymer (estimated to be 0.5 for this system), and v_2 is the volume fraction of polymer in the swollen film at equilibrium swelling as mathematically defined by:

$$
v_2 = 1/(1 + f)^3
$$
 and $f = (x_2 - x_1)/x_1$

where x_1 is the original length of a side of the polymer film (before swelling) and x_2 is the length after swelling.⁹ The crosslink density was calculated from DMA data by:

Table 6-Formulation Parameters of Waterborne **Acrylic-Epoxy Coatings**

	CC-1	$CC-2$	SF-1	$SF-2$
Volume solids (%) 39	-209	44	34	34
		230	230	-230

Test of	HIPAC - (30-min Induction - 21 Day Cure)	ASTM Method	
Application method 6-mil bird bar applicator			
Drying time (hr) Set-to-touch		D 5895	
	Tack-free	D 5895	
	Dry-hard	D 5895	
	Dry-through	D 5895	
Film appearance Gloss (20°)		D 523	
	$Gloss(60^{\circ})$	D 523	
	Gloss (85 ^o)	D 523	
		D 2197A	
	Wet scrape (kg) $(24 \text{ hr} / 21 \degree C)$	D 2197A	
Solvent resistance MEK double rubs			
	MEK immersion		
		D 1308	
4-hr exposure	5% sol. Sodium hydroxide	D 1308	
4-hr recovery	-5% sol. Sulfuric acid	D 1308	
	5% sol. Nitric add	D 1308	
	Chlorox	D 1308	
	Mustard	D 1308	
	Ketchup	D 1308	
	Coffee	D 1308	
	Lipstick	D 1308	
		D 4585	
	Xenon arc 500 hr ($\Delta E/60^{\circ}$ gloss retention - %)	G26	
		D 2794	
	Gardner impact (in./lb) Reverse	D 2794	
		D 1737	
		D 3363	
	Persoz	D 4366	
Scrub resistance Cycles to failure		D 2486	
	Abrasion resistance Taber (1000 g, 1000 cycles, CS17) mg, loss	D 1044	

Table 7--Procedures to Evaluate Coating Performance

$$
Mc = \frac{\rho}{v_e} = \frac{3RT}{E'}
$$

where R is the ideal gas constant, and E' is the modulus of the film at temperature $T(100^{\circ}C)$.

RESULTS

Benchmark Evaluations

The benchmarking results comparing the two commercial paints and two control formulas are presented in *Table 8*.

Several properties are notable for the entire group of four coatings. Generally, they display:

- -Long pot life (24-36 hr)
- -- Short dry time (Set-to-touch in 15-45 min, Dry hard in 2-5 hr)
- Good chemical resistance (200 MEK double rubs, four hour chemical spot tests)
- Fairly good hardness (Pencil hardness HB H, Persoz hardness 142-249)
- Fairly good weather resistance (500 hr Xenon arc Weatherometer with $\Delta E < 2$, 60° gloss retention of 68-98%)

With some of the other properties evaluated, a range of performance was observed between the four coatings. SF-1 withstood a 24 hr water immersion with no dramatic changes, while the other three coatings displayed blister ing. After 1000 hours' exposure in a Cleveland Humidity cabinet, CC-1 and SF-1 had no obvious effects, while the other two coatings displayed blistering after only 24 hr of exposure. SF-1 had a direct impact flexibility of >160 in. lb, while the other coatings had impacts of <30 in.-lb. Finally, scrub resistance ranged from 1024 cycles up to 4925 cycles, again illustrating a wide range of performance. Comparing these results with properties listed in ASTM D 3730, these coatings generally displayed moderate performance relative to industrial coating requirements and high performance relative to architectural applica tions, which suits their target markets and applications quite well as stated in the Introduction.

Substitution of the New Epoxy Resin into CC-1 andSF-1

Qne objective of this work was to determine if and how the new epoxy resin may be used in acrylic epoxy coat ings. To pursue this objective, the AP ER resin was substi tuted at various concentrations into CC-1 and SF-1 as described in the Experimental section. The comparisons of the AP ER substituted coatings versus those of the initial formulas CC-1 and SF-1 are reported in *Tables* 9 and 10, respectively. Substituting AP-ER into the CC-1 coating improved dry time, gloss, water resistance, and scrub resistance, with other properties generally remaining corn parable. Dry hard times were reduced from 2.25 down to 0.75 hr, while dry through times decreased from 6.5 down to 3.5 hr. The dry through time for the coating at 4:2 AP ER concentration was >12 hr probably due to excessive epoxy in the system, which does not harden. Gloss values of 60° increase from 66.5 up to the 85-90 range, another attractive feature for topcoat applications. AP ER improved the 24-hr water immersion from blistering of the coating to virtually no effect on the coating or substrate. Finally, the scrub resistance was improved from 1024 up to the 5000 cycle range, again an attractive feature, especially for inte riot HIPAC topcoat applications. The SF-1 formulations with AP-ER similarly have decreased dry times and increased gloss. The decreased dry time with AP ER could be expected due to its unique molecular weight distribu

tion (described in the Introduction), which offers the coalescing attributes of liquid epoxy resin, while displaying the lacquer drying characteristics of solid epoxy disper sions without the need for co solvent. The cause for im provements in gloss, water resistance, and scrub resistance is not quite as evident at this time. Further evidence may come from analytical studies presented and discussed later.

It should be noted that, in these studies, AP-ER was directly substituted for the epoxy component of CC 1 and SF-1 with no other adjustments to the formulation. Substi-

Table 8--Coating Benchmark Test Results

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tution of AP-ER in CC-1 resulted in a VOC reduction from 209 to approximately 180 g/L due to the elimination of solvent in the epoxy component. Good film formationwas still observed. The acrylic component, having a high $T_{\rm g}$, requires co-solvent or other means of plasticization to film form. We did not explore further VQC reduction by reformulating the acrylic component but this is a fertile area for additional investigation and development. It is possible that co-solvent levels in these systems, as in Part A of SF-1, and Parts A or B of SF-2 could be reduced when incorporating AP ER, while obtaining good film formation.

Property and Cure Development

The development of Persoz hardness and resistance to MEK double rubs over the first 30 days after application are presented in *Table* 11 and illustrated in *Figures* 1 and 2. Hardness continues to increase steadily and consider ably over this time period. The prolonged duration of hardness development may be due to the slow crosslinking reaction and/or the slow release of co solvents. However, resistance to MEK develops at a faster rate, at least for coatings CC-1, CC-2, and SF-1, reaching 200 or greater double rubs within four to 10 days after application. This would suggest that crosslinking is developing, which is responsible for increased MEK resistance, while solvent release from the applied coatings is slow, thus causing the films to remain plasticized and soft. Considering the pro posed crosslinking mechanisms of carboxyl-epoxy reaction and epoxy homopolymerization, both of which are slow at ambient conditions, it is surprising that MEK resistance develops this quickly. In order to assess the rate and extent of T_g and crosslink development, cure of SF-1 films was investigated using IR spectroscopy, DMA, DSC, and solvent swell up to 21 days after application.

The IR spectra of the SF-1 coating at 1, 7, 14, and 21 days after application are presented in *Figure* 3. These spectra show no obvious signs of reaction, including little, if any, epoxy decrease (epoxy band at 916 cm^{-1}) over this period. This is in contrast to the reported literature on these coatings, $1-5$ which states that they will undergo carboxyl-epoxy reaction to develop crosslinking. However, as noted earlier, this reaction is very sluggish at room temperature and therefore these results may not be alto gether surprising. Nonetheless, these results do seem to be in contrast to the ability of SF-1 to develop MEK resistance.

Table 12 lists results from the solvent swells, DMA and DSC characterizations and analysis for SF-1. During the first heating in DSC evaluations of the ambient cured SF-1 coatings (Figure 4A), a large exotherm was observed (noted as residual cure in *Table* 12), suggesting an incomplete cure under ambient conditions. The exotherm was \sim 30 J/ g and it occurred between 80 and 200°C and stayed fairly constant in magnitude over the first 21 days after coating

Table 10-Coating Test Results for Substitution of New Epoxy Resin into SF-1

application. The T_g of the ambient cured material increased from 9 to 25°C over this 21-day period, as indicated by both DMA T_g onset results and DSC T_g midpoint results, which agree fairly well. After the first DSC heating, the SF-1 samples were cooled and then reheated in DSC (noted in

Table 11-Coating Property Development

Table 12 as second heat). During this second heating (Figure 4B), no residual exotherms were observed, the T_g s increased from the first heating, and the T_g s stabilized at \sim 40 $\rm ^{\circ}$ C (i.e., they no longer were a function of the ambient cure time). To illustrate the cure during this first heating and the increase in T_g , Figure 4 illustrates the DSC scans for the first and second heating of SF-1 performed seven
days after application. The DSC results indicate that the coatings did not reach their full potential cure under ambient conditions but were able to fully cure during their

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Cure Days	Solvent Swell		DMA	DMA	DMA	DSC	DSC ^o	DSC ^D
	Vol % Increase	Mc Calc	Mc Calc	T_0 Onset (°C).	T_0 Mid (°C)	T_0 Mid (°C)	Residual Cure	T_0 Mid (°C)
14 21	125 82 80 91	(g/mol) 171993. 27560 25505 43150	(g/mol) 24492. 21402 16723. 14907	5 15 25 25	33 40 46 46	Q 10 25	(J/g) -35.5 28.3 -26.31 27.3	2nd Heat -37 41 41 43

Table 12-Cure Evaluations of SF-1 Coatings Using Solvent Swells, DMA, and DSC

ia). Defermined as difference inexotherm observed from DSC soan during first heating minus exotherm observed from soan of second heating molexotherm u
(b). Typbserved during second heating.

Table $13 - T_g$ and Mc results from DMA analysis of CC-1 and SF-1 with AP-ER

first DSC heating. It is likely that heating is required for these coatings to reach their maximum potential cure and crosslink density. The lack of incomplete cure at ambient conditions is most likely due to the naturally slow reaction of these systems at ambient conditions, combined with vitrification of the system at room temperature. However, it must be noted that only a small amount of chemical reaction is needed in thermosetting systems to achieve noticeable crosslink density of the system. These results illustrate that some crosslink density was achieved in these systems, evidently enough to impart a substantial level of chemical resistance and other properties (*Table 8*) required for them to perform adequately in their intended moderate duty industrial and high performance architectural applications.

Mc values (molecular weight between crosslinks) were calculated by applying the results of solvent swell and DMA analysis to the equations presented in the Experi mental section. Lower Mc numbers indicate higher crosslink density. The first observation that must be noted is that Mc values calculated from DMA results are all lower than those calculated from solvent swell data. As noted from the DSC analysis previously discussed, these coatings are not fully cured (reacted) at room temperature but do undergo substantially more curing when heated. Therefore, the samples characterized via DMA undergo crosslinking during the DMA heating phase, thus increasing crosslink density and lowering Mc. In contrast, the solvent swell samples were cured and analyzed only at room temperature and therefore their crosslink density is understandably lower and their Mc values higher. Nonetheless, it appears that Mc for these coatings stabilizes in the $15,000 - 25,000$ g/mol range after 14 to 21 days. This indicates that SF-1 seemingly has crosslink density sufficient to obtain good solvent and chemical resistance, which agrees with the applied coating properties reported in *Table 8.*

Although cure development of CC-1 and SF-1 with AP-ER was not followed per se, crosslink densities were ob tained from applied coatings cured for 21 days at 21°C, 50% relative humidity. *Figure* 5 illustrates an example of these DMA results. The T_g and Mc values of these coatings are reported in *Table* 13 along with results for SF 1 at the same time after application. ${\mathbb T}_{\mathbf g}$ values are all around 56° C just slightly above the T_g for SF-1, 46°C. However, crosslink density increases considerably with the substitution of AP ER, with Mc values decreasing from around 15,000 $25,000$ to approximately 2300 g/mol. This dramatic increase in crosslink density may be responsible for the improvements in water and scrub resistance when AP ER is substituted into these systems.

SUMMARYAND CONCLUSIONS

These evaluations confirmed that commercial acrylic ep oxy coatings have long pot life $(\geq18$ hr), short dry times (set-to-touch < 45 min; dry hard < 5 hr), good hardness, moderate chemical resistance, and fairly good weather ability. A range of performance was observed with water resistance, flexibility, and scrub resistance. As a whole, these coatings generally displayed moderate performance relative to industrial coating requirements and high per formance relative to architectural applications.

Substitution of AP-ER into these coafings for the specified epoxy component resulted in quicker dry times, higher gloss, better water resistance, and dramatically improved scrub resistance. This modification also holds promise to reduce solvent content of the system due to its ability to film form with lower co-solvent concentrations.

Hardness development of the acrylic epoxy coatings occurred rather slowly and was still improving 30 days after application; however, resistance to MEK double rubs developed in most of the coatings over a four 10-day period. This suggests that ample crosslinking is occurring to impart chemical resistance properties but slow release of volatiles from the coating results in sluggish hardness development. Analysis of film curing via IR spectroscopy indicated little chemical change in the system, including little to no reaction of the epoxy. However, DMA and solvent swell studies indicate crosslink density formation of $Mc = 15,000-25,000$. Over a 30-day cure at 21^oC and 50% RH, T_g values also increase from about 5 to 25°C. DSC analysis of room temperature cured systems confirmed incomplete cure; however, upon a second DSC heating, this exotherm no longer appeared and the system was fully cured. Although room temperature drying did not result in full cure, properties were still good. Substituting AR-EP into the system increased crosslink density of Mc to approximately 2300 g/mol.

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