# **The AI-B (Aluminum-Boron) System**

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## **Equilibrium Diagram**

Due to contradictory evidence and misleading speculations, evaluation of the A1-B system has hitherto proven unsatisfactory. Early investigations of the constitution of the system A1-B by [26Hae], later discussed by [26Mei], claimed the existence of a eutectic  $L \leftrightarrow Al + AlB_2$  at 560 °C at 30 to 35 at.% B. AlB was said to form peritectically from  $Al(L)$  and  $AlB_{12}$  at 1100 ~ The assessed A1-B phase diagram is presented in Fig. 1. The Al-rich portion ( $>33$  at.% Al) is essentially based on a review by [84Sig], who reported the eutectic  $L \leftrightarrow (Al) + AlB_2$  at a maximum solid solubility of 0.0045 at.% of B in A1 at 659.6 <sup>o</sup>C, the temperature corroborated by DTA experiments of [93Ips]. Agreement exists on the peritectic mode of formation of  $AIB<sub>2</sub>$  from the dodecaboride and the liquid. Whereas the peritectic point and temperature were given at 0.55 at.% B and 975 °C by [61Ser] and [84Sig], at ~2.4 at.% B and 980 °C by [90Car], and at 0.74 at.% B and ~927 °C by [76Mon], results of [36Hof2], [53Lih], [66Gru], and [72Sir] claimed the existence of AlB<sub>2</sub> up to temperatures between 1350 and 1500 °C. Differential thermal analysis experiments  $(Al<sub>2</sub>O<sub>3</sub>$  crucibles, 5 K/min,  $Ar + 5\% H_2$  atmosphere) [93Ips] confirmed the peritectic formation of AlB<sub>2</sub> at 1030  $\pm$  5 °C; this value is used throughout this assessment. The boron-rich  $(>66$  at.% B) part of the phase diagram refers to a review by [89Ner] including some features from [84Sig], [89Hig], and [90Carl and results from a reinvestigation of the AI-B binary by [93Dus] including Pirani melting point measurements with a calibrated microoptical pyrometer and confirming the incongruent melting of  $\alpha AIB_{12}$  at 2050  $\pm$ 35 °C.



Attempts to establish the solubility of Al in  $\beta$ -boron from Xray powder and single crystal studies on an arc-melted alloy with nominal composition  $AIB_{12}$  revealed a maximum solubility of-3 at.% A1 [89Hig, 93Dus].

Despite the continuing controversy on the existence of " $\beta$ AlB<sub>12</sub>" and "AlB<sub>10</sub>," all the experimental evidence leads to the conclusion that these phases are impurity (carbon) stabilized and that  $\text{AlB}_2$  and  $\text{AlB}_{12}$  are the only true binary aluminum boride compounds. Experimental evidence from the formation of ternary aluminum borides  $AI_{1-x}M_{\lambda}B_{12}$  ( $M = Be$ , Mg, V, Cr, Mn, Fe, Co, Ni, Cu) with the  $\gamma AIB_{12}$ -type structure [81Hig, 87Higl, 87Hig2] furthermore backs the conclusion that  $\alpha$ AlB<sub>12</sub> is the only thermodynamically stable modification of the aluminum dodecaboride. Binary  $\gamma AIB_{12}$  might thus be accepted as a metastable compound or as a ternary product stabilized by the small amounts of  $M$  present in Al flux grown crystallized material. The stability of  $\alpha AIB_{12}$  at 850 °C confirmed by [93Dus] is in contrast to observations by [67Ato], who claimed no formation of AlB<sub>12</sub> below 920 °C. A similar instability of AlB<sub>2</sub> at room temperature reported by [53Lih], however, is believed to arise from chemical attack rather than from thermodynamic instability.

A new phase in the A1-B system was formed after implantation of aluminum by boron ions at peak concentrations between 6 and 60 at.% [92Var]. The composition of this phase  $Al_2B_3$  is near 60 at.% B. It is stable to at least  $525^{\circ}$ C after heat treatment in vacuum. The accepted invariant reactions in the A1-B binary are listed in Table 1.



### **Table 2 AI-B Crystal Structure Data**





### **Crystal Structures and Lattice Parameters**

**Crystal structure and lattice parameter data for A1-B phases are given in Tables 2 and 3, respectively. Table 4 results from an attempt to resolve the existing confusion about true and/or impu-**

**rity-stabilized A1-B phases. Thus Table 4 lists the original phase name in contrast to the phase name accepted throughout this assessment based on proper characterization. Table 4 in chronological sequence reflects on the various methods of preparation and characterization of the "A1-B" phases mentioned in the literature including information on the aluminum** 

### **Table 3 AI-B Lattice Parameter Data**



(a) Metastable or metal impurity stabilized.

### **Table 4 Preparation and** Crystal Structures **of Aluminum Borides**



#### **Phase name** Proper Crystal Lattice parameters, nm in reference phase name system and the system of the system **Reference** in reference phase name system a a b c Preparation Comment  $[53$ Lih] ................. AlB<sub>2</sub> AlB<sub>2</sub> AlB<sub>2</sub> ... ... ... ... ... ... ... Starting L + AlB<sub>12</sub>  $\rightarrow$ materials: Al, AlB<sub>2</sub> at  $\widetilde{T}_p$  ><br>B<sub>2</sub>O<sub>3</sub>, S, KClO<sub>3</sub>, 1400 °C, at RT  $B_2O_3$ , S, KClO<sub>3</sub>, 1400 °C, at RT: 2<br>1350 to 1450 °C AlB<sub>2</sub>  $\rightarrow$  "AlB<sub>4</sub>"  $\text{AlB}_2 \rightarrow \text{``AlB}_4"$  $+$  Al  $[55Sam]$ ...............  $AIB_{12}$   $\alpha AIB_{12}$  Tetragonal 1.01 ... 1.43 ...<br> $[56Fe]$ ................  $AIB_2$   $AIB_2$  Hexagonal 0.3009 ... 0.3262 Graphite-[56Fel] ................ AIB 2 AIB 2 Hexagonal 0.3009 ... 0.3262 Graphite- Impurities: A1, C, crucible, 800 °C  $B_4C$ <br>
...  $\alpha = 65^\circ$ , cell of  $[58Par]$  .................  $\beta B$   $\beta B$  Rhombohedral 1.014 ... ... ... monoclinic AIB<sub>12</sub> [39Hal2] is transformed into  $\beta$ **B**<br>Comparison of [58Koh] .............. A1B12 ~B Monoclinic ......... Comparison of monoclinic  $\text{AlB}_{12}$ with  $\beta B$  $\alpha$ AIB<sub>12</sub>  $\alpha$ AIB<sub>12</sub> Tetragonal 1.028 ... 1.430 Graphite-crucible ... 6AIB<sub>12</sub> ... 1.430 Graphite-crucible ...  $\begin{array}{lllllllll}\n\beta AIB_{12}^{\phantom{124} \phantom{124} } & & & \alpha_{13} B_{48} \bar{C}_2 & & \text{Teragonal} & & 1.255 & \dots & & 1.018 & \text{Graphite-crucible} & \dots \\
AB_{10} & & & AB_{24} C_4 & & \text{Orthorhombic} & & 0.8881 & & 0.9100 & & 0.5690 & \text{Graphite-crucible} & \text{Chemical}\n\end{array}$ Graphite-crucible analysis: 0.37 wt.% C  $[60Kohl]$ ..............  $\alpha AIB_{12}$   $\alpha AIB_{12}$  Tetragonal 1.0161 ... 1.4283 Graphite-crucible Melting point at  $2150 \pm 50$  °C  $\beta AIB_{12}$   $Al_3B_{48}C_2$  Orthorhombic 1.2340 1.2631 1.0161 Graphite-crucible ...<br>Pseudotetragonal 1.255 ... 1.018 ... Pseudotetragonal 1.255 ... 1.018 ... ... ...<br>
Hexagonal 0.3009 ... 0.3262 ... No details of  $\cdots$  $[60K_0h2]$  ............. AlB<sub>2</sub> AlB<sub>2</sub> Hexagonal 0.3009 ... 0.3262 ... preparation  $\Delta I B_{10}$   $\Delta I B_{24}C_4$  Orthorhombic 0.888 0.910 0.569 Graphite-crucible ...<br>  $\alpha A I B_{12}$  Tetragonal 1.016 ... 1.428 Graphite-crucible ...  $\alpha A1B_{12}$   $\alpha A1B_{12}$  Tetragonal 1.016 ... 1.428 Graphite-crucible ...<br>  $\beta A1B_{12}$   $A1_3B_{48}C_2$  Orthorhombic 1.234 1.263 1.016 Graphite-crucible ...  $\overrightarrow{A1}_3\overrightarrow{B}_{48}\overrightarrow{C}_2$  Orthorhombic 1.234 1.263 1.016 Graphite-crucible ...<br>  $\gamma$ AIB<sub>12</sub> Orthorhombic 1.66 1.75 1.02 Graphite-crucible Observed at  $\gamma AIB_{12}$   $\gamma AIB_{12}$  Orthorhomblc 1.66 1.75 1.02 Graphite-crucible single-crystal examinations of  $\alpha$ AlB<sub>12</sub><br>Starting  $[61Bec]$ ................  $AIB_2$   $AIB_2$  Hexagonal ... ... ... ...  $AI_2O_3$ -crucible, Starting  $H_2$ , Ar, vac., 900 materials: 98%  $H_2$ , Ar, vac., 900 Al, 96 to 99% B, above  $950 °C$ :  $AlB_2 \rightarrow Al(L) +$  $\hat{B}$ Al $B_{12}$ [61Koh] ..............  $\gamma AIB_{12}$   $\gamma AIB_{12}$  Orthorhombic 1.656 1.753 1.016 Graphite-crucible ...  $\alpha AIB_{12}$   $\alpha AIB_{12}$  Tetragonal 1.015 ... 1.429  $A_1O_3$ -crucible, ... [61Ser]................ He  $\beta$ AIB<sub>12</sub> Al<sub>3</sub>B<sub>48</sub>C<sub>2</sub> Orthorhombic 1.255 ... 1.018 Preparation No chemical without C analysis for C  $AIB$ ,  $AIB$ , stable below 975 °C<br>Single-crystal [63Wil] ................  $AIB_{10}$   $AIB_{24}C_4$  Orthorhombic 0.888 0.910 0.569 Graphite-crucible obtained from [58Koh]<br>Chemical [64Mat1]............. A1B 2 A1B 2 Hexagonal ......... Chemical analysis:  $Al_0 9B_2$ <br>Chemical and ~A1B m ~A1B12 Tetragonal ...... Graphite- Chemical and crucible, 1700 crystallographic<br>
<sup>o</sup>C, cooling rate analysis as °C, cooling rate analysis a<br>  $\sim$ 450 °C/h [36Nár] ~450 °C/h  $A1_3B_{48}C_2$   $A1_3B_{48}C_2$  Orthorhombic 1.234 1.263 0.508 ... Low-temperature phase (LT - mod. A) Orthorhombic 0.617 1.263 1.016 ... Low-temperature phase (LT - mod. B) Tetragonal 0.882 ... 0.509 ... Hightemperature phase (HT)  $AIB_{24}C_4$   $AIB_{24}C_4$  Orthorhombic 0.888 0.910 0.569 ... Earlier called " $AIB_{10}$ "

Table 4 Preparation and Crystal Structures of Aluminum Borides (continued)



**Table 4 Preparation and Crystal Structures of Aluminum Borides (continued)** 

# **Section II: Phase Diagram Evaluations**





Reference	Phase name in reference	Proper phase name	Crystal system	$\boldsymbol{a}$	Lattice parameters, nm b	$\mathfrak{c}$	Preparation	Comment
$[87Kro]$	$AlB_{24}C_4$	$AlB_{24}C_4$	Orthorhombic	0.8914	0.9095	0.5676	$\cdots$	Morphologic studies
	$Al_3B_{48}C_2$	$Al_3B_{48}C_2$	Tetragonal	0.884	$\ldots$	0.506	$\cdots$	High- temperature phase from [64Mat]
[89Pug]	$AlB_{10}$	AlB <sub>24</sub> $C_4$	Orthorhombic	$\cdots$	$\ddot{\phantom{0}}$	.	Oxidation of $B_4C$ -Al, surface etching in ethanol solution of iodine with perhydrol	Melting point at $2100^{\circ}$ C, oxidation starts above 700 °C
[89Ner]	$\gamma AIB_{12}$	$\gamma AIB_{12}$	Orthorhombic	$\cdots$		$\cdots$	$\ddotsc$	Assumption that $\gamma AIB_{12}$ is no binary phase
	AlB <sub>2</sub>	AlB <sub>2</sub>	$\ddot{\phantom{0}}$	.	$\ddotsc$	$\ldots$	$\cdots$	Claimed to be stable up to 1450 °C
[89Hig]	$\text{AlB}_{31}$	βB	Hexagonal	1.0965	.	2.3868	Arc melting of $\alpha AIB_{12}$	Solubility of $-3$ at.% Al in $\beta B$
[92Var]	AI <sub>2</sub> B <sub>3</sub>	Al <sub>2</sub> B <sub>3</sub>	Hexagonal	1.840	$\cdots$	0.896	Ion implantation of Al by B	After vacuum heat treatment a nonequilibrium precipitate is stable up to 525 $^{\circ}C$

**Table 4 Preparation and Crystal Structures of Aluminum Borides (continued)** 

(a) All compounds obtained from monocrystal specimens grown from B-C solution m AI melt.

**Table 5 Electrical Conductivity of Compounds in the AI-B System by [87Kis]** 

<b>Phase</b>	Temperature, K	E. еV
	100 to 192	0.18
	192 to 227	0.23
	227 to 357	0.363
	100 to 203	0.217
	$200 \text{ to } 400$	$0.6$ to $1.2$
	110 to 119	0.18
	119 to 238	0.28
	238 to 435	0.075

boron carbides as carbon usually was the major source of impurity. Comparison of the lattice parameter data given in Table 4 reveals the various attempts towards a proper and unambiguous crystallographic description of the individual and in many cases mislabeled phases.

### **Miscellaneous**

Melting points of AlB<sub>2</sub>, AlB<sub>24</sub>C<sub>4</sub>("AlB<sub>10</sub>"),  $\alpha$ AlB<sub>12</sub>, and  $\text{Al}_3\text{B}_{48}\text{C}_2$  (" $\beta \text{Al}_3\text{B}_{12}$ ") were found by [60Gia] to be 1655 ± 50  $^{\circ}$ C, 2423 ± 50  $^{\circ}$ C, 2163 ± 50  $^{\circ}$ C, and 2214 ± 50  $^{\circ}$ C, respectively. No details, however, were given concerning the correction for black body condition.

Vickers (HV) and Knoop (KV) microhardness as well as crack resistance for single crystal faces of  $\alpha AlB_{12}$ ,  $\gamma AlB_{12}$ ,  $AlB_{24}C_4$ ("AlB<sub>10</sub>"), and Al<sub>3</sub>B<sub>48</sub>C<sub>2</sub> (" $\beta$ AlB<sub>12</sub>") were studied in detail by [86Dub]. The authors used the fracture toughness  $K<sub>1c</sub>$  as a criterion for crack resistance. These mechanical properties increase in the following order:  $\alpha AlB_{12} \rightarrow \gamma AlB_{12} 2$  $\text{AlB}_{24}\text{C}_{4}(" \text{AlB}_{10}") \rightarrow \text{Al}_{3}\text{B}_{48}\text{C}_{2}$  (" $\beta \text{AlB}_{12}$ "). The Young

modulus E (for 100% dense pellets) of  $\alpha AIB_{12}$  is similar to B4C (400 to 600 GPa) [91Gos]. Knoop hardness was measured for various crystal faces by [60Gia] under a load of 100 g for AlB<sub>2</sub> (9.6 GPa for (00.1)),  $\alpha$ AlB<sub>12</sub> (22.45 GPa for (110), 23.8 GPa for (101), 22.1 GPa for (221)), and  $\gamma AIB_{12}$  (23.55 GPa for (100)). These values have to be compared with 22.9 GPa for the (100) face of  $\alpha$  and  $\gamma AIB_{12}$  at a load of 2 N [86Kis] and with the values of Knoop-microhardness on (101) faces of  $\alpha AIB_{12}$ between 20.2 and 22.6 GPa found by [80Oka]. Under a load of 30 g, [55Sam] found  $36.2 \pm 1.7$  GPa as Knoop hardness for  $\alpha$ AlB<sub>12</sub>. Fracture toughness was reported by [86Kis] for  $\alpha$  and  $\gamma$ AlB<sub>12</sub> to be 2.2 MPa m<sup>1/2</sup> for (100).

Thermal conductivity  $\lambda$  of  $\alpha$  and  $\gamma AIB_{12}$  is reported by [86Kis] to be 38.7 W/m $\cdot$ K at 315 K in contrast to  $\sim$ 14 W/m $\cdot$ K measured and calculated by [91Kek] and  $\sim$  5 W/m $\cdot$ K measured by [91Gos].

Experimental values for the electrical resistivity at room temperature are between  $3.1 \times 10^{-2}$  and  $7.7 \times 10^{-2}$  Qm for AlB<sub>2</sub> [72Sir],  $5.92 \times 10^2 \Omega m$  for  $\alpha$ AlB<sub>12</sub>,  $3.85 \times 10^5 \Omega m$  for  $\gamma$ AlB<sub>12</sub>,  $2.64 \times 10^3$  to  $1 \times 10^6$   $\Omega$ m for  $Al_3B_{48}C_2$  (" $\beta AlB_{12}$ "), and  $2.02 \times$ 10<sup>5</sup>  $\Omega$ m for AlB<sub>24</sub>C<sub>4</sub> ("AlB<sub>10</sub>") [87Kis]. Resistivity of  $\alpha$ AlB<sub>12</sub> as a function of temperature and porosity was reported by [55Sam]. No superconductivity was observed for  $\text{AlB}_2$  as low as  $1.5 K$  [72Sir]. All phases in the Al-B system are p-type semiconductors [86Kis]. According to [72Sir], the measured values of the Hall coefficient (R) are positive, ranging from  $1 \times 10^{-3}$ to  $1.5 \times 10^{-4}$  cm<sup>3</sup>/C. [87Kis] also calculated the activation energy of electrical conductivity  $E_a$  in various temperature ranges from measurements of the temperature dependence of  $E_a$ . The results are listed in Table 5.

<b>Phase</b>	Temperature,	$\Delta H$ <sup>V</sup> <sub>1</sub> , k.J/mol	$\Delta_f S_T^0$ , .I/mol·deg	Reference
	298	$-66.9 \pm 12.6$	$\cdots$	[67Dom]
	298	$-66.9$	34.7	$[76$ Jon
		$-146.4$ to $-154.8$		[77Kan]
	298	$-200.8 \pm 41.8$	$\cdots$	[67Dom]
	298	$\cdots$	88.4	[88Kap]
	1673 to 2273	$-171.5$	50.2	[77Kan]
		$-196.6$ to $-209.2$	$\cdots$	[77Kan]
	1673 to 2273	$-223.8$	49.4	[77Kan]

**Table 6 Thermodynamic Data in the AI-B System** 

Oxidation of  $\alpha$ AlB<sub>12</sub> single crystals in air starts at 1000 °C forming  $9\text{Al}_2\text{O}_3$   $2\text{B}_2\text{O}_3$  and  $\text{B}_2\text{O}_3$  [80Oka].

The dissolution of  $\alpha AIB_{12}$  was studied as a function of dispersity in hydrochloric, sulfuric, nitric acids, KOH in various concentrations and in hydrogen peroxide at room temperature and at the boiling point of the solvent [87Ner].

From infrared (IR) and electron spin resonance (ESR) spectroscopy on single crystals of  $\alpha AIB_{12}$  and  $\gamma AIB_{12}$ , the refractive index was determined; the A1-B compounds can be regarded as wide zone semiconductors [88Pri].

The energy of an interstitial solid solution of boron in aluminum and the energy of mixing were calculated using the pseudopotential method by [81 Bog].

A calculation of the band structure of  $\text{AlB}_2$  in all main symmetry directions was performed by [77Arm] with the LCAO technique in order to discuss the stability and bonding in metallic diborides. A density-of-states calculation confirmed that electrons are transferred from A1 into B orbitals, which are mainly  $\pi$  bonding in the two-dimensional boron layer.

A calculation of the electronic structure of  $\text{AlB}_2$  using the LAPW method is currently in progress [93Rip].

### **Thermodynamics**

Thermodynamic data have been assembled by various research groups [67Dom, 76Jon, 77Kan. 81Bat, 84Sig, 87Kis, 88Kap]. They are given in Table 6.

Measurements of the enthalpy of  $AIB<sub>2</sub>$  in the temperature range 310 to 1200 K by [87Kis] give the temperature dependence  $H^0(T) - H^0(298) = -12\,335.54 + 34.75153T + 22.2118$  $\times$  10<sup>-3</sup> $T^2$  (J/mol).

The heats of combustion are  $-3711.2$  kJ/mol for AlB<sub>2</sub> and  $-14916.0 \text{ kJ/mol}$  for  $\alpha$ AlB<sub>12</sub> [67Dom].

The coefficients for the specific heat are  $C_p = a + bT - cT^2$ (J/mol-K). For AlB<sub>2</sub>,  $a = 50.17$ ,  $b = 22.3 \times 10^{-3}$ ,  $c = 7.87 \times$ 10<sup>5</sup> [76Jon], and  $a = 34.75$ ,  $b = 44.4 \times 10^{-3}$ ,  $c = 0$  [87Kis]. For  $\alpha$ AlB<sub>12</sub>,  $a = 211.28$ ,  $b = 115.07 \times 10^{-3}$ ,  $c = 85.34 \times 10^{5}$ [88Kap].

The calculated Gibbs energy for the reaction  $AIB_2(s) \rightarrow$ 2B(polycrystalline) + AI(L) equals 201 133.25- 152.143 T J/mol. For the reaction  $AI(s) + 2B(\beta) \rightarrow AIB_2$ ,  $\Delta G$  equals  $-66$  944 + 5.335 T J/mol [84Sig].

The integral heat of dissolution of boron in molten aluminum at 1773 K was examined by [81Bat] between 0 and 18 at.% B; the composition dependence was claimed to be  $\Delta H_{\text{sol}} =$ 9292.7  $x_B$  (J/g-atom).

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### **Addendum**

The A1-B System, contributed by O.N. Carlson, was published *in Bulletin of Alloy Phase Diagrams,* 11(6), 560-566 [90Carl. The present evaluation supersedes the earlier evaluation.

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